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## **Development of surface functionalization strategies for 3Dprinted polystyrene constructs**

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### **Abstract**

There is a growing interest in 3D printing to fabricate culture substrates; however, the surface properties of the scaffold remain pertinent to elicit targeted and expected cell responses. Traditional 2D polystyrene (PS) culture systems typically require surface functionalization (oxidation) to facilitate and encourage cell adhesion. Determining the surface properties which enhance protein adhesion from media and cellular extracellular matrix (ECM) production remains the first step to translating 2D PS systems to a 3D culture surface. Here we show that the presence of carbonyl groups to PS surfaces correlated well with successful adhesion of ECM proteins and sustaining ECM production of deposited human mesenchymal stem cells, if the surface has a water contact angle between 50° and 55°. Translation of these findings to custom-fabricated 3D PS scaffolds reveals carbonyl groups continued to enhance spreading and growth in 3D culture. Cumulatively, these data present a method for 3D printing PS and the design considerations required for understanding cell-material interactions

### **Keywords**

polystyrene; surface chemical modification; plasma treatment; protein adhesion; 3D printing

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CONFLICT OF INTEREST

JPF is a founding member of, and has a financial interest in, 3DBioWorks, which focuses on developing bioreactor technology.

### **INTRODUCTION**

Polystyrene (PS) has served as the fundamental substrate for adherent cell culture for more than 50 years.<sup>1</sup> However, the lack of surface functionality which cells recognize and spread on has dictated the need to modify PS surfaces to facilitate anchorage and develop 3D models to more accurately mimic real biological systems.<sup>2</sup> Typical tissue culture polystyrene (TCPS) exhibits high surface oxidation, providing a functionalized surface for cells to adhere and proliferate.<sup>3</sup> A number of plasma chemistries have been used to functionalize PS surfaces, including air,<sup>4</sup> ammonia,<sup>5</sup> oxygen,<sup>6</sup> and multiple organic compounds.<sup>7</sup> Surface activation with plasma has been leveraged to incorporate biofunctional groups and increase surface wettability, a mechanism thought to facilitate cell and protein adhesion.<sup>8</sup> Linking together surface chemistry, material properties, and biological response provides a viable path to translating 2D in vitro culture surfaces to 3D scaffolds. The fundamental basis of these protein–surface interactions lies in engineering the surface chemistry to direct protein adsorption.<sup>9</sup> Building *in vitro* surfaces to mimic tissue niches through protein adhesion, cell patterning, and guiding differentiation provides a unique opportunity to sequester cells and generate functional tissues.10,11

The widespread use of PS as a 2D culture substrate, and demonstrated improvements 3D culture provides, suggests transitioning PS to a 3D culture platform. However, the applications of 3D-printed PS as a cell-contacting growth surface have been limited,<sup>12</sup> likely due to the difficulty in liquefying PS without thermally degrading the polymer.13 Previous approaches tend to rely on fabricating large objects by sintering smaller objects (e.g., a packed-bead bioreactor)<sup>14</sup> or microfluidic approaches.<sup>15-17</sup> Transforming the standard PS culture surface to a complex bioreactor scaffold would aid in expanding culture populations for clinical and research applications.18 However, transitioning PS from 2D to 3D cell culture requires surface modification to facilitate protein and cell adhesion, where understanding surface chemical properties and cell response could lead to both universal and targeted culture constructs.

The combination of 3D printing and surface chemical modification provides the potential for a highly tunable process to select culture environments for select cell types. Directed surface modification has been shown to influence human mesenchymal stem cell (hMSC) differentiation, but further exploration would fully develop the link between surface chemical cues and cell phenotype.19 Variable phenotype expression of hMSCs has been established through surface modification, where carbonyl (carboxyl) containing surfaces showed evidence for preference to the chondrogenic phenotype and high ECM protein association with the surfaces (particularly fibronectin).<sup>20</sup> In addition, fibronectin and integrin have been seen to associate preferentially with carbonyl and amine containing surfaces.<sup>21</sup> Surface wettability influences the adhesive force between proteins and the substrate, resulting in conformational changes at the interface<sup>22</sup> which can be sensed by adhered cells and influences spreading and proliferation.23 Building surface chemistry and properties to selectively elicit and pattern protein adhesion provides a viable path to engineering cell–material interactions.<sup>24-26</sup> Amine groups have been used to direct hMSC osteogenic differentiation, attributed to differential activation of focal adhesion kinases by preferentially associating ECM proteins.<sup>27</sup>

Significant progress in developing 2D surfaces targeting growth and differentiation has been achieved, but the biomimicry of 3D scaffolds and ability to control the physiological environment through fluid dynamics,  $28$  mechanical stresses,  $29$  oxygen content,  $30$  cellular distribution,  $31$  and overall geometry  $32$  warrant a transition to 3D culture. Transforming PS from standard flat cell culture dishes to a 3D scaffold could provide the means to better replicate the in vivo environment by improving cell–cell interactions and enabling higher culture densities.<sup>33</sup> 3D printing provides a unique avenue to rapidly manufacture and test models with biologically relevant structures; a platform that could establish clinical niches in vitro for highly replicative screening of novel pharmaceuticals, cancer therapies, and disease models.  $34-38$  The main objective presented in this work explores engineering surface chemistry for 3D-printed PS scaffolds to target hMSC expansion. First, we sought to establish the surface characteristics generated by our plasma system. Next, we determined the link between surface properties and protein adhesion. Then, we investigated hMSC spreading, growth, and ECM expression on the surfaces and ultimately show the concept translates to a 3D model. We demonstrate that carbonyls on moderately hydrophilic surfaces help to promote hMSC proliferation and phenotype maintenance, likely through associated surface ECM proteins expressed by the cells and deposited from media. These strategies present a viable 3D PS scaffold capable of supporting hMSC expansion with wide applicability.

### **MATERIALS AND METHODS**

#### **2D substrate preparation**

Silicon wafers of 15 mm<sup>2</sup> were wiped clean with ethanol (EtOH), sonicated for 5 min in acetone, then methanol, and dried with nitrogen. Fifty microliters of 10% (w/w) PS (>50,000 molecular weight atactic flakes, Polysciences, Inc., Warrington, PA)/toluene was spun cast at 314.2 rad/s (3000 RPM) for 40 s with a spin coater. These samples were used for surface characterization. For cell and protein analyses, cover glass slides were wiped clean with EtOH and coated 10% PS solution following the same spin casting method:  $22 \text{ mm} \times 40 \text{ mm}$ and  $25 \times 40$  mm used 150 μL of solution and  $24$  mm  $\times 50$  mm used  $250$  μL of solution. Protein analyses used only the smallest glass slides.

### **Surface activation**

A low-temperature plasma activated samples by creating a dielectric barrier discharge (DBD) in a 4 mm ID glass capillary tube attached to a 20 kHz AC high-voltage source [Figure 1(A)]. Ultrapure helium was used as a discharge gas with acetonitrile vapor or oxygen added at a concentration of 2.5% (v/v). Ignition frequency was visually tuned with the lowest power selected achieving uniform and consistent plasma ignition (0.14–0.34 kV). Samples were translated under the plasma jet with servo motors. For 3D-printed objects, each side was exposed to plasma treatment.

### **Goniometer wettability measurements (water contact angle)**

Two microliters of deionized water was dispensed onto leveled samples with an inverted syringe pump. A video camera recorded the pendant droplet touching and stabilizing on the

surface. The baseline water–surface interface was determined and Young-Laplace (Sessile) contact angle measured on at least three drops.

#### **X-ray photoelectron spectroscopy (XPS)**

Quantitative XPS measurements determined elemental compositions with a survey scan of  $0-1100$  eV, a pass energy of 160 eV, and a resolution of 1.0 eV. The binding stoichiometry was determined from high-resolution scans with a pass energy of 40 eV and a resolution of 0.1 eV. A Gaussian–Lorentzian distribution was used for peak fitting, with a full width at half maximum constraint of 1.7 eV on six spectra per sample.

### **Total protein determination**

Samples were covered in fetal bovine serum (FBS) for 1 h at 37°C, then rinsed with phosphate-buffered saline (PBS), placed on ice, and covered with cold radioimmunoprecipiration assay (RIPA) buffer for 5 min. Isolated protein was centrifuged at  $14,000g$  for 15 min at 4 $\degree$ C. The supernatant was tested on three samples in technical triplicate with a Pierce BCA Protein Assay following standard microplate procedures.

#### **Polyacrylamide gel electrophoresis (PAGE) protein separation and Coomassie staining**

Protein was isolated following the methods explained above with Halt Protease Inhibitor (ThermoFisher, Waltham, MA) added to RIPA, including a 1:100 FBS:RIPA control. Mini-Protean TGX (Bio-Rad, Hercules, CA) gels were loaded with Laemmli buffer and samples, as described.<sup>39</sup> Gels were run for 45 min at 120 V in  $1 \times$  Tris-Glycine buffer and fixed in 5% acetic acid, 45% deionized distilled water (ddH<sub>2</sub>O, >18.2 M $\Omega$  resistivity), and 50% methanol for 30 min. Gels were washed three times for 5 min in ddH<sub>2</sub>O with protein bands visualized and imaged with Coomassie blue.

### **Proteomics through mass spectrometry (MS)**

Protein was isolated through electrophoresis, as described. Gel lanes were separated, cut into 10 equal pieces, digested with sequencing grade trypsin, and peptides extracted with acetonitrile-formic acid buffer.<sup>40</sup> Liquid-chromatography (LC)-MS/MS was performed using a nano-LC system (Easy nLC1000) connected to Q Exactive mass spectrometer (ThermoFisher). Peptides were eluted at 300 nL/min using a series of linear gradients of acetonitrile in 0.1% formic acid. Data sets were searched against UniProt human database using MaxQuant software (version  $1.5.5.1$ ).<sup>41</sup> Groups with peptide counts less than the total number of test groups (6), nonspecies, or obvious contaminants were eliminated.

### **Cell culture practices**

hMSCs (RoosterBio, Frederick, MD) were thawed and expanded in Dulbecco's Modified Eagle Medium with 10% FBS, MEM non-essential amino acids 0.1 mmol/L, and penicillin (100 U/mL)–streptomycin (100  $\mu$ g/mL) (growth media). 90% confluent hMSCs were washed with pH 7.4 PBS and lifted with a 5 min incubation with 0.25% trypsin–EDTA and neutralized with growth media. Passage 2–4 hMSCs were used during experiments, with cells treated as passage 1 upon receipt. FIfty thousand hMSCs per 30 minute UV sterilized

sample were used unless noted. Cell counting was accomplished using the Trypan Blue exclusion method with a hemocytometer.

#### **Confocal imaging and image processing**

Cells were lifted as described and resuspended in the media of interest (growth media) or RoosterBio High-Performance Basal Media (no serum media). hMSCs were seeded onto surfaces and grown overnight. Cytochalasin-D-treated hMSCs had their media replaced with growth media containing 10  $\mu$ mol/L cytochalasin-D for 1 h prior to fixing.<sup>42-45</sup> Samples were chemically fixed using 4% formaldehyde and 1% sucrose solution and permeabilized with a 300 mmol/L sucrose, 50 mmol/L sodium chloride, 6 mmol/L magnesium chloride, 20 mmol/L HEPES, and 0.5% Triton-X-100 solution. Actin was stained with 2.5% AlexaFluor 594 Phalloidin in PBS, counterstained diamidino-2-phenylindole (DAPI) or VectaSheild (Vector Labs, Burlingame, CA). A confocal microscope (SP5 X, Leica Microsystems, Buffalo Grove, IL) captured Z-stack images, which were processed with a MATLAB (MathWorks, Natick, MA) program. The program read in image files from the specified directory, and separated the image stacks into individual image locations (i.e., groups of individual z-stacks). Small debris and back-ground fluorescence were eliminated with size and intensity exclusion thresholds. For each image stack, the average spread area was calculated by dividing the total area covered by phalloidin (red channel) by the number of detected nuclei (blue channel). Running this process for each stack was used to generate statistics (Supporting Information, Appendix 1).

### **Flow cytometry**

hMSCs were seeded as described and grown for 6 days, with media replaced on the third day. Cells were lifted with StemPro Accutase (ThermoFisher), diluted with an equal volume of growth media, and resuspended in 1% FBS in PBS (flow buffer). To stain, a PE-Negative Cocktail (CD45, CD34, CD11b, CD19, and HLA-DR) and -Positive Cocktail (FITC CD90, PerCPCy 5.5 CD105, and APC CD73) from a Human MSC Analysis kit (BD Biosciences, Franklin Lakes, NJ) was used, following manufacturer's recommendations. hMSC Isotype controls were run and UltraComp eBeads (ThermoFisher) single color controls were used for compensation. Data collection was performed on an FACS Canto II (BD Biosciences) and analysis was performed with Cytobank Community software (Cytobank Inc. Santa Clara, CA).

### **DNA quantification**

Cells were dissociated from three samples with Trypsin and 0.25% EDTA and resuspended in PBS. DNA was isolated following the procedures for DNeasy Blood & Tissue Kit (Qiagen, Hilden, Germany) and quantified in technical triplicate using a Quant-iT PicoGreen (ThermoFisher) kit, following the manufacture's protocol. Fluorescence measurements were read on an M5 SpectraMax plate reader (Molecular Devices, San Jose, CA) with an excitation wavelength of 490 nm and emission read at 538 nm.

### **RT-PCR**

RNA was isolated with a Trizol phase separation technique. Samples were covered with Trizol, gently mixed, and transferred to a microcentrifuge tube. Chloroform was added and vigorously mixed. Phase layers were separated by centrifugation at 12,000g for 15 min at 4°C, with the aqueous phase transferred. 70% EtOH was added to 35% of the final volume. Samples were transferred to a RNeasy mini column (Qiagen). Complete RNA isolation followed kit instructions. cDNA conversion was completed using a High Capacity cDNA Archive Kit (ThermoFisher) following manufacturers procedures. A  $C$ T gene expression assay was completed, comparing genes of interest to GAPDH (endogenous control). A 7900HT RT-PCR system (Applied Biosystems, Foster City, CA) cycled samples: 2 min at 50°C, 10 min at 95°C, 40 cycles of 15 s at 95°C, and 1 min at 60°C.

### **3D scaffold fabrication**

A 3D Bioplotter (EnvisionTEC GmbH, Gladbeck, Germany) fabricated scaffolds. A sugar– glass support material (EnvisionTEC GmbH) was heated to 150°C and extruded through a 0.3 mm ID needle tip at 70–80 kPa at 10–30 mm/s. The PS was melted at 155°C and extruded through a 0.4 mm ID needle tip at 900 kPa at 3–5 mm/s. The build surface was maintained at 60°C. The extrusion temperature of the PS is below the thermal decomposition temperature of the material  $(\sim 163^{\circ}C^{13})$  to avoid degradation of the material. At the completion of prints, the support was removed with ddH2O, scaffolds washed overnight in ddH<sub>2</sub>O, cleaned in 100% EtOH for 30 min, and air dried.

### **Scanning electron microscopy (SEM)**

A FEI Quanta 200 FEG-ESEM (Hillsboro, OR) was used to image PS scaffolds under high vacuum at 5.0 keV. Samples were coated with Au–Pd prior to imaging to reduce charging artifacts.

#### **Gel permeation chromatography (GPC)**

The molecular weight of printed and stock PS was determined by GPC (Waters Alliance Separations Module e2695, Waters 2414 Refractive Index Detector, and Waters HSPgel columns in series (HR MB-L and HR 3.0 columns, 6.0 mm ID  $\times$  15 cm), Waters, Milford, MA). Three samples of each were dissolved in tetrahydrofuran with molecular weight determined against PS standards in at least technical duplicate.

#### **Differential scanning calorimetry (DSC)**

The glass transition temperature was determined by measuring the thermal energy change as a function of temperature change with a TA DSC Q100 (TA Instruments, New Castle, DE). Three ~10 mg samples of printed and stock PS were sealed in hermetic aluminum pans. Samples were rapidly equilibrated at 160°C and kept isothermal for 5 min. The temperature was ramped at 10°C per min to −20°C and back to 160°C and heat flow recorded.

### **Statistical analysis**

Statistical significance was determined through a one-way two-sided ANOVA at the 95% confidence interval using the Games-Howell comparison method in Minitab 18 (Minitab, Inc. State College, PA) unless noted.

### **RESULTS**

#### **Surface characterization of 2D treated surfaces**

In this work, we sought to develop a surface modification method which facilitates hMSC adhesion and growth on custom PS scaffolds. This work was performed on 2D surfaces to validate our method. Wettability analysis [Figure 1(B,C)] shows that treated surfaces experience a significant increase in wettability as compared to the no treatment (NT) control (92.3°). 2.5% oxygen (O2) and 2.5% acetonitrile (ACN) plasmas reduced contact angles to 54.6° and 26.4°, respectively, while TCPS surfaces maintained a native contact angle of 50.0°. Further surface characterization, accomplished through XPS, confirmed the NT surface as primarily carbon containing, with moderate oxygen and little nitrogen content observed in TCPS [Figure 1(D)]. Nitrogen incorporation in  $O_2$  and He surfaces were similar. Total oxygen incorporation was similar in TCPS-, 100% helium plasma (He)-, and O<sub>2</sub> plasma-treated samples, followed by ACN. ACN-treated samples had much greater nitrogen incorporation. Specific bonding stoichiometry indicate that oxygen present in He- and  $O_2$ treated surfaces contained both single-bonded oxygen and carbonyl groups [Figure 1(E) and Supporting Information Table SI]. The carbonyl content of He- and  $O<sub>2</sub>$ -treated surfaces exceeded, though not statistically different, the quantities present in TCPS. ACN-treated surfaces contained the greatest quantity of single- and double-bonded oxygen, as well as potential surface amines groups and malononitrile.

### **Characterization of adhered surface protein**

The contact angle decreased relative to initial values following serum exposure, and was statistically similar across all surfaces, except for ACN-treated surfaces [Figure 2(A)]. ACN-treated surfaces were observed to have an increase in contact angle. All contact angle changes were statistically different from initial values (two-sample t test,  $p < 0.001$ ) for all test surfaces at the 95% confidence interval). A Pierce BCA analysis revealed O<sub>2</sub>treated samples were the only group to contain a statistically greater quantity of adhered protein relative to the NT control, but no different from the other surfaces [Figure 2(B)]. Coomassie staining confirmed the adhered protein quantities similarities, where the darker bands of the  $O_2$ - and TCPS-treated samples indicate greater protein presence [Figure 2(C)]. Additionally, darker high-molecular-weight bands appear present in these samples. To better quantify protein differences, mass spectrometry was performed on the PAGE gel separated proteins. The proteins have been grouped by family type and the top 10 protein families present ranked [Figure 3(A,B)]. Based on peptide counts (e.g., instances of matching protein fragments),  $O_2$ -treated surfaces tend to associate better with  $(1)$  proteins generally, attracting more and a wider variety and (2) greater relative amounts of ECM proteins (23.1% of O2, 19.7% of TCPS, 14.8% of NT, 11.5% of He, 10.8% of FBS, and 5.1% of ACN). Specifically, collagen family proteins were only found to be a top 10 present protein by

peptide count in the  $O_2$  (#6, 3.6%) surfaces and TCPS surfaces (#10, 3.0%). These both represent a greater fraction than in the diluted FBS control (#10, 2.6%).

#### **Cellular interaction with protein-coated surfaces**

Surfaces containing the highest quantities of carbonyl groups tended to facilitate the greatest spreading (Figure 4). Protein containing media and coating the surfaces with media increased cell spreading, while the use of a serum-free media and an actin inhibitor (cytochalasin-D) reduced cell spreading. NT surfaces tended to prevent cellular spreading, where cells remained rounded. Quantified hMSC phenotype (Table I) shows that the cells generally expressed (>95%) of the positive markers and lacked expression of negative markers (83%–90%).  $O_2$ -treated samples exhibited the highest amount of phenotypically expected hMSCs. To contrast this, He-treated samples, with lower wettability and similar oxygen content, exhibited higher ratios of hMSCs expressing the differentiation markers in the Negative Cocktail Mix. The ACN-treated surfaces showed decreases in the positive and negative markers.

### **Cellular response to 2D surfaces**

DNA quantification showed significantly higher growth (DNA fold change) after 6 days [Figure 5(A)]. The He- and  $O_2$ -treated surfaces showed the greatest increase in DNA, followed by treatment with ACN. PCR showed similar collagen I [Figure 5(B)] and fibronectin [Figure 5(C)] mRNA expression of hMSCs attached to the plasma-treated samples and TCPS over the course of 6 days. Expression levels were significantly greater than on the NT surface (collagen I, expression on the NT surface was inconclusive). As well, CD29 [Figure 5(D)] expression was significantly greater from hMSCs on the treated surfaces relative to the NT surfaces (except for He, where expression was inconclusive).

#### **Translation of results to 3D model**

To explore the applicability of this method on 3D-printed PS, we developed a method for extrusion deposition modeling of PS without thermally degrading the polymer [Figure 6(A)]. Scaffolds with 90° [Figure 6(B)], 60° [Figure 6(C)], and 85.5° [Figure 6(D)] rotations between individual layers were printed. SEM imaging of the 90° matching demonstrates open internal pore structure [Figure 6(E)]. When the 85.5° surface was treated with the  $O<sub>2</sub>$ plasma, hMSCs adhered and spread within the scaffolds. At 3 days, hMSCs were observed to bridge fibers [Figure 6(F)] and remained spread on fibers with expected morphology after 6 days [Figure 6(G)]. hMSC seeded at 5000 cells/cm<sup>2</sup> continued to expand in static culture up to 7 days [two-sample *t* test *p* value between NT and  $O_2$  scaffolds = 0.011, Figure 6(H)]. No differences in glass transition temperature [two-sample *t* test,  $p = 0.212$ , Figure 6(I)], high or low fraction molecular weights [two-sample t test,  $p = 0.825, 0.210$ , Figure 6(J,K)], or molecular weight area fractions (two-sample *t* test,  $p = 0.905$ ; Supporting Information Table SII) were observed between the printed and stock material, indicating our method did not thermally degrade the material during extrusion.

### **DISCUSSION**

Our first goal was to decouple the 3D geometry and surface chemistry by understanding the wetting and chemical properties generated by the DBD plasma treatment. The He test group was included as a process control with  $O<sub>2</sub>$  and ACN surfaces intended to increase wettability and incorporate carbonyl and amine groups, respectively. NT samples set a baseline for the DBD groups and TCPS provided a comparison to standard culture surfaces. Targeting contact angles within the 40°−60° range have been shown to best promote cell adhesion [Figure 1 (B,C)], independent of surface chemistry.<sup>8</sup> TCPS, He, and  $O_2$  surfaces achieved this target range and ACN and NT surfaces were statistically outside this range. As well, the differences in contact angle between the He- and  $O_2$ -treated surfaces may be attributable different, subtle (10s of nm) structures on each<sup>23,46</sup> or differences in surface charge.<sup>47,48</sup>

Previous work suggests that carbonyl incorporation on surfaces tends to correlate positively with cell growth<sup>7</sup> and surface wettability influences protein conformation.<sup>49</sup> The moderate oxygen and low nitrogen content in TCPS is consistent with the manufacturing process: a corona discharge plasma in an oxygen rich environment.<sup>50</sup> O<sub>2</sub> and He chemical similarity [Figure 1(D,E)] likely due to the experimental DBD system depositing reactive species from the ambient atmosphere. However, significant malononitrile deposition was observed on ACN substrates, likely a by-product of plasma polymerization of the monomer as well as degradation of the vaporized acetonitrile in the energetic plasma discharge. Malononitrile is thought to inhibit metabolic rate of mammalian cells and form thiocyanate in tissues,  $51$ causing detrimental effects on seeded cells. The developed custom DBD system promotes surface changes, controlled by changing the input gasses, and effectively modifies the PS targets with reactive ionized species.

Having established surface chemical differences between the groups, we next investigated how these surface properties impacted protein adhesion. It is well established that proteins will tend to adhere on surfaces, regardless of underlying composition through nonspecific adsorption.52 However, protein conformational differences, types of bound proteins, and relative concentrations may differ due to surface chemistry.53 The interaction between the surface chemistry, wettability, and proteins likely dictates preferences toward individual protein adhesion types and conformation.<sup>54</sup> By engineering surfaces to attract or repel proteins of interest, targeted cell expansion and niche generation may be possible. The first step to realizing differences in our treated surfaces included repeating wettability measurements on serum-coated surfaces [Figure 2(A)]. We found the surface wettability significantly increased on all surfaces, except the ACN surfaces. These data indicate that the adhered proteins may undergo a conformational change depending on the surface chemistry, as seen previously.<sup>22,53</sup> Total protein quantification [Figure 2(B)], PAGE gel separation [Figure 2(C)], and MS analysis (Figure 3) support differences in total protein content and composition. As with all protein interacting surfaces, the Vroman Effect is certainly in play.55 We sought to investigate how hMSCs will begin to interact with the surface. It has been previously observed that at an hour, ~10% of seeded cells will begin to attach to TCPS,<sup>56</sup> beginning to spread on the surfaces.<sup>57</sup> As well, this time is near the peak (areal mass) of attached protein, forming a uniform layer, and cells may sense this attached protein as a single uniform structure.<sup>58</sup> The protein layer is fully formed, though it may change over

time, with higher molecular weight material generally replacing lower weight material.<sup>59</sup> Particular proteins of interest contain the peptide sequence arginine–glycine–aspartic acid  $(RGD)$ , help to promote cell adhesion through integrin proteins.<sup>60</sup> ECM proteins, such as collagen, contain the RGD sequence and have been shown to influence proliferation and osteoblastic differentiation of hMSCs through cellular pathway activation and mechanical substrate deformation.<sup>61-64</sup> Determining the surface proteins informs ultimate hMSC response. Other than collagen, the remaining proteins mentioned in the table are not known to have definitively beneficial cell adhesion affects. Kininogen and apolipoprotein have been observed to inhibit cell adhesion processes.<sup>65,66</sup> ITIH proteins work to stabilize hyaluronic acid.67 Glycoproteins play a role in cell-cell interactions, but not ECM interactions directly.<sup>68,69</sup> Serum albumin works to inhibit endothelial apoptosis.<sup>70</sup> The serpin superfamily inhibit enzymes, such as those in the coagulation cascade.<sup>71</sup> Macroglobulin and antiproteinases act in the foreign body response.<sup>72,73</sup> Serotransferrin, vitamin D-binding protein, hemoglobin, and fetoprotein are all linked to carrying metabolites.<sup>74-77</sup> Collagen stands apart from this group, as the only ECM component protein. Improved protein adhesion on the carbonyl containing surfaces could be due to their electronegativity relative to single-bonded oxygen, encouraging hydrogen bond formation associating with proteins.78 Conversely, ACN-treated surfaces appear to resist protein adhesion, specifically ECM proteins, which may be linked to the extremely high surface wettability<sup>79</sup> or surface chemistry and likely acidic surface environment.<sup>80</sup>

Protein adhesion differences were compared against hMSC spread area, a measure of surface attractiveness, binding, and phenotype. Quantified cell spreading (Figure 4) suggests the carbonyl groups on surfaces with contact angles between 50° and 55° act with proteins either in the media or during a coating to facilitate attachment, potentially through an amenable conformational preference or increased presence of ECM proteins.49 Surfaces lacking these traits may lack the native biological residues necessary for optimal protein adhesion and cell anchorage, attachment, and spreading. All the treated surfaces should allow hMSCs to initially attach and grow, and that the carbonyls on surfaces are likely coordinating with proteins to encourage the cell attachment and spreading. For example, more ECM proteins were observed on  $O_2$  surfaces and these had higher spreading as compared to He and NT surfaces. Both the use of a serum free media and treating the cells with an actin inhibitor (cytochalasin-D) reduced cell spread area as compared to both growth media conditions. This correlation indicates cell spreading is influenced by an actinmediated mechanism with the underlying surface proteins, guided by the underlying surface properties. Though the cells likely do not sense the chemistry directly, the coordination of surface chemistry and protein adhesion certainly facilitates cell adhesion, patterning, and growth.81,82 The underlying surface properties can be used to influence hMSC adhesion characteristics by modulating underlying adhered protein.

To ensure the hMSCs maintained their stemness, flow cytometry was performed with a set of standard positive and negative multipotency markers (Table I). Moderately hydrophilic surfaces (TCPS,  $O_2$ ) supported a high fraction of phenotypically expected cells, suggesting some interaction between the underlying surface chemistry, adhered protein, and differentiation potential. The adhered hMSCs on these surfaces were found to align with previously set guidelines of surface marker expression.<sup>83</sup> In contrast, the ACN surface

showed decreased positive and increased lacking of negative markers, indicating a reduction in multipotency.

The next objective was to evaluate hMSC growth and ECM protein expression differences of hMSCs grown on the test surfaces. Cumulative results suggest that adding oxygen to the plasma tends to improve hMSC–surface interactions, where treatment with ACN tends to diminish these (Figure 5). The hMSCs expanded significantly more on the treated surfaces versus the untreated  $[Figure 5(A)]$  and continued to express collagen I, fibronectin, and CD29 [Figure 5(B-D)]. Collagen I and fibronectin are two representative ECM matrix proteins.84 CD29 (Integrin β1), in part, helps to regulate hMSC ECM adhesion, migration, and focal adhesion kinase pathway activation.85,86 The diminishing presence of ECM production suggests the cells may lack the ability to modify the surrounding environment on the NT surface. $87$  This may be caused by the underlying protein layer directed by the treated PS surface properties. The carbonyl containing surfaces tended to allow continued matrix production over the time studied.

The final goal was to evaluate the translatability of our 2D findings to a 3D model, working toward bringing PS from a 2D surface to 3D scaffold with directed functionalization. The 2D work details the importance of the  $O<sub>2</sub>$  treatment and was tested in 3D as a proof-ofconcept The method [Figure 6 (A)] is capable of producing a wide range of geometries and heights [Figure 6(B-D)] with an open internal architecture [Figure 6(E)] without thermally degrading the polymer [Figure 6(I-K) and Supporting Information Table SII]. hMSCs were observed to spread and interact with the scaffold [Figure 6(F,G)], utilizing both the polymer filaments and filling in the void spaces between the fibers. The trends indicate that the plasma treatment improved growth [Figure 6(H)]. O<sub>2</sub>-treated scaffolds demonstrate a method to transform PS to a 3D culture surface which hMSCs attach and grow with future work seeking to investigate this further. Extrusion 3D-printed PS scaffolds have previously isolated and expanded lymphoma cancer cells, with the scaffold isolating the cells of interest<sup>36</sup> and PS punched scaffold have expanded hMSCs in a perfusion bioreactor.<sup>12</sup> Continued customization of PS scaffolds would define surface chemistry, geometry, and flow properties for individual cell populations. Further work seeks to understand how hMSCs sense these 3D culture platform, and the influences on directed differentiation, leading to highly specialized PS bioreactors.

### **CONCLUSION**

Here we propose a set of criteria to transform PS from a 2D to 3D culture surface. Incorporating carbonyls on surfaces with water contact angles between 50° and 55° tended to facilitate ECM protein and hMSC adhesion, produce greater numbers of cells, and an increased proportion of phenotypically expected cells. The data suggest that protein–cell coordination at the surface was influenced by the underlying surface properties, where the high electronegativity of the carbonyl bonds and moderate wettability likely improves hMSC response to adhered proteins. Extremely wetting surface detracted from these findings. Transitioning from 2D to 3D, namely a surface containing ~6% carbonyl groups on an  $O<sub>2</sub>$ -treated PS base with water contact angle between 50 $^{\circ}$  and 55 $^{\circ}$ , would serve well as a growth substrate in 3D. Initial demonstration shown here presents a viable path for

3D PS bioreactors. Transitioning PS from a 2D to 3D culture surface would allow for culture systems utilizing flow, complex geometry, and densified culture, better replicating the in vivo environment and potential to generate clinically relevant population sizes for transplantation and research. Particularly, understanding how surface chemistry influences protein and cell adhesion could be used for building complex engineered environments to elicit targeted responses from cells.

### **Supplementary Material**

Refer to Web version on PubMed Central for supplementary material.

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#### **FIGURE 1.**

A: Diagram of the DBD system with plasma flowing over the sample. B: Example contact angle images of the NT-,  $O_2$ -, and ACN-treated surfaces. C: Quantified contact angles. Activating the surfaces with He,  $O_2$ , and ACN reduced contact angles. Groups with different letters indicate statistical difference ( $p < 0.05$ ), data are shown as mean  $\pm$  standard deviation,  $n = 5$ . D: Survey XPS data shows surface oxygen and nitrogen presence on the DBD-treated surfaces and TCPS and nitrogen on the DBD-treated surfaces. Nitrogen presence in the DBD surfaces is likely a by-product of plasma treatment in ambient atmosphere. The greatest nitrogen presence was seen on the ACN-treated surfaces. Oxygen and nitrogen presence was accompanied by a reduction in carbon. E: High-resolution DBD-treated surfaces showed single bonded oxygen and carbonyl groups. TCPS showed both single bonded oxygen and carbonyl groups. NT surfaces remained purely carbon. Nitrogen presence on the ACNtreated surfaces were primarily bound to carbon as imine groups, with 13% of nitrogen on the surface was present as malononitrile. XPS data are shown as mean ± standard deviation; six scans per sample; single sample for NT, ACN, and TCPS; two samples for He and  $O_2$ . Groups with different letters indicate statistical difference; asterisk indicates identical values (Tukey multiple comparison,  $p < 0.05$ ).

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**ACN** 



### **FIGURE 2.**

A: Quantified contact angles across the surfaces following serum exposure. Contact angles were reduced for all surfaces except on ACN-treated surfaces. This indicates increased surface wettability and surface energy following exposure to serum. Groups with different letters indicate statistical difference ( $p < 0.05$ ), data are shown as mean  $\pm$  standard deviation,  $n = 5$  except TCPS  $n = 4$ , ACN  $n = 3$ . B: Total protein content following coating. O<sub>2</sub> surfaces showed statistically greater protein adhesion than the NT surface, but not different from any other surface. Data are shown as mean  $\pm$  standard deviation,  $n = 3$ . C: Coomassie stained gel. Visually, the  $O_2$  surface band appears darker than the NT surface across the whole lane. Similar bands are observed across all treatment samples and the source serum.



### **FIGURE 3.**

A: Table showing the top 10 proteins present by peptide count, grouped by identified protein family. ECM proteins are called out separately and represent percent of total peptides counted. Collagen was only present in the top 10 for FBS, TCPS, and  $O_2$ , indicating a possible correlation between carbonyls on surfaces and ECM protein association. Conversely, ACN surfaces had the lowest percentage of ECM proteins present. B: Graphical representation of the percentages of counted peptides. Colors in figures match those in the table for each group.



### **FIGURE 4.**

A: Average spread area of hMSCs on treated surfaces in various media. The data indicates that proteins tend to associate with the carbonyl containing surfaces, facilitating cell spreading through an actin-related mechanism. Surfaces precoated with serum generally had greater spread areas, where media lacking protein showed less spread cells. The addition of cytochalasin-D, an actin inhibitor, to the media prior to imaging generally reduced spread area of the hMSCs. Note: ANOVA was performed over the whole dataset, groups with different letters indicate statistical difference ( $p$  < 0.05), and data are shown as mean  $\pm$ standard deviation,  $n = 16$  except NT cytochalasin-D treatment,  $n = 15$ .

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### **FIGURE 5.**

A: Normalized DNA content to each D0 value. The He and  $O_2$  surfaces showed the greatest DNA content fold change, followed by ACN, TCPS, and NT surfaces. This indicates surface activation successfully facilitated hMSC proliferation. Data are shown as mean  $\pm$ standard deviation,  $n = 3$ . B: mRNA expression of collagen I. All surfaces showed detectable expression except for the NT surface. C: mRNA expression of fibronectin. All surfaces showed greater expression than the NT surface on day 6. D: mRNA expression of CD29. All surfaces showed detectable expression except for the He surface. Statistically greater expression was seen on TCPS,  $O_2$ , and ACN surfaces compared to the NT surface at day 6. For CT mRNA expression, depressed expression on the NT surface at day 6 may indicate reduced ability to modify the surrounding environment and express ECM proteins, likely related to the underlying protein content directed by the surface chemistry. Note: groups with different letters indicate statistical difference ( $p < 0.05$ ) for ANOVA was performed on each time point. For B-D, data are in technical triplicate and displayed as mean ± standard deviation.

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### **FIGURE 6.**

A: A schematic of the 3D printing method. Card stock is placed onto the temperature controlled base plate with the support material printed on top of this. The extrusion head is then replaced automatically and PS is extruded onto the support material. 3D-printed PS scaffolds with layer offset angles of (B) 90°, (C) 60°, and (D) 85.5°. E: An SEM image of the 90° inner geometry of the PS scaffold. F: A single hMSC is seen bridging two PS fibers in a 85.5° scaffold at 3 days. The green lines indicate the edges of the PS fibers, DNA is stained blue with DAPI, and F-actin is stained red with phalloidin. G: hMSCs on a 85.5° scaffold display expected morphology on a  $O_2$ -treated scaffold at 6 days. The image is a projected z-stack over the curved surface of a single PS fiber. DNA is stained blue with DAPI and F-actin is stained red with phalloidin. H: DNA fold change from day 0 after 7 days of growth. hMSCs were initially seeded at 5,000 cells/cm<sup>2</sup>. Trends indicate that the 85.5° scaffold  $O_2$  scaffold supports hMSC growth throughout the scaffold and warrants further development ( $p = 0.011$ , data shown as mean  $\pm$  standard deviation  $n = 3$ ). To characterize the PS before and after printing, matching batches of source (stock) and

extruded (printed) material were evaluated. I: Differential scanning calorimetry was used to determine the glass transition temperature of the material. No difference  $(p = 0.212)$ was observed between samples ( $n = 3$ ). (J) High molecular weight fraction and (K) low molecular weight fraction of stock and printed polystyrene as determined by gel permeation chromatography ( $n = 3$ ). Samples were run against a standard curve to determine molecular weight. No difference was found between high ( $p = 0.825$ ) and low ( $p = 0.210$ ) fractions of the stock or printed material.

### **TABLE I.**

### hMSC Phenotype Assessment Through Flow Cytometry



Carbonyl containing surfaces tended to have greater percentages of hMSCs displaying expected surface markers, where ACN-treated surfaces tended to have reduce the percentage of expected phenotype hMSCs.