

Contents lists available at ScienceDirect

Japanese Dental Science Review



journal homepage: www.elsevier.com/locate/jdsr

Does speed-sintering affect the optical and mechanical properties of yttria-stabilized zirconia? A systematic review and meta-analysis of *in-vitro* studies

Hengyi Liu^a, Masanao Inokoshi^{a,*}, Kaiqi Xu^a, Watcharapong Tonprasong^a, Shunsuke Minakuchi^a, Bart Van Meerbeek^b, Jef Vleugels^c, Fei Zhang^{b,c}

^a Department of Gerodontology and Oral Rehabilitation, Graduate School of Medical and Dental Sciences, Tokyo Medical and Dental University, 1-5-45 Yushima, Bunkyo, Tokyo 113-8549, Japan

^b KU Leuven (University of Leuven), Department of Oral Health Sciences, BIOMAT & UZ Leuven (University Hospitals Leuven), Dentistry, Kapucijnenvoer 7 blok a, B-3000 Leuven, Belgium

^c KU Leuven (University of Leuven), Department of Materials Engineering, Kasteelpark Arenberg 44, BE-3001 Leuven, Belgium

ARTICLE INFO

Keywords: Zirconia Rapid sintering Dental material Yttria Optical property Mechanical property

ABSTRACT

Zirconia restorations are increasingly popular in dental treatment. Yttria-stabilized zirconia (YSZ) needs to be sintered for clinical applications and novel speed-sintering protocols are being developed for chairside treatments. Whether the properties of speed-sintered YSZ meet clinical requirements, however, remains unclear. Therefore, we conducted a systematic review and meta-analysis on the influence of speed-sintering on the optical and mechanical properties of dental YSZ according to the Preferred Reporting Items for Systematic Reviews and Meta-Analyses (PRISMA) 2020 guidelines. A literature search was conducted using PubMed, Embase, and Web of Science databases for relevant articles published between January 1, 2010 and February 28, 2022 in English, Chinese, or Japanese. After full-text evaluation and quality assessment, 26 articles were selected. Meta-analysis revealed that speed-sintering does not significantly affect the CIEDE2000-based translucency parameter, contrast ratio, three-point flexural strength, biaxial flexural strength, or fracture toughness of YSZ (p < 0.01) compared to conventional sintering. However, the CIELab-based translucency parameter of conventionally sintered YSZ is higher than that of speed-sintered YSZ. The descriptive analysis indicated that speed-sintering does not affect the hardness of YSZ compared to that of conventionally sintered YSZ. The results indicate that speed-sintering is suitable for preparing YSZ for dental restorations.

1. Introduction

After having initially been used as a femoral head replacement material, 3-mol% yttria-stabilized zirconia (YSZ) was first applied in dentistry for crown-and-bridge prosthodontics in 2004 [1]. Compared to conventional metal and porcelain-fused-to-metal restorations, full-ceramic zirconia restorations have better aesthetic performance and eliminate the risk of complications due to metal allergies [2]. Furthermore, their mechanical properties are superior to those of other ceramic restorations such as lithium-disilicate glass-ceramics and hybrid ceramics [3].

Pure zirconia is a ceramic that exists in three polymorphs depending on the temperature: monoclinic (m), from room temperature to 1170 °C; tetragonal (*t*), from 1170 to 2370°C; and cubic (*c*), above 2370 °C [4,5]. However, monoclinic zirconia (*m*-ZrO₂) has inferior mechanical properties, which limits its clinical application. In order to stabilize tetragonal zirconia (*t*-ZrO₂) at room temperature, Y₂O₃, CaO, MgO, or CeO₂ stabilizers are incorporated in the zirconia-crystal structure during sintering. Among these stabilizers, yttria (Y₂O₃) is the most popular for dental zirconia. Depending on the Y₂O₃ content, YSZ is categorized as tetragonal zirconia polycrystals (TZP), containing 3-mol% Y₂O₃ and being referred to as 3Y-TZP, or partially stabilized zirconia (PSZ), containing 4–6 mol% Y₂O₃ and being referred to as 4–6Y-PSZ. YSZ has excellent mechanical properties because of the stress-induced *t*→*m* transformation, which increases the toughness of the material. Along with the fine-grained microstructure, YSZ exhibits much higher strength

* Corresponding author. E-mail address: m.inokoshi.gerd@tmd.ac.jp (M. Inokoshi).

https://doi.org/10.1016/j.jdsr.2023.08.007

Received 9 November 2022; Received in revised form 2 August 2023; Accepted 22 August 2023

1882-7616/© 2023 Published by Elsevier Ltd on behalf of The Japanese Association for Dental Science. This is an open access article under the CC BY-NC-ND license (http://creativecommons.org/licenses/by-nc-nd/4.0/).

and toughness than other technical ceramics [6–8]. YSZ can slowly and spontaneously degrade in moist environments, such as in the oral cavity and even at room temperature. This process, known as low-temperature degradation, can cause micro- and macro-cracking on the YSZ surface, which may influence the aesthetics and mechanical properties of YSZ in clinical practice [9–11].

For clinical applications, YSZ needs to be sintered to full density. Conventional sintering procedures are time- and energy-consuming. For example, monolithic 3Y-TZP requires sintering times and temperatures of 4-12 h and 1350-1550 °C, respectively, depending on the manufacturer's instructions [12,13]. Recently, there has been a drive towards the chairside fabrication of dental restorations. To our knowledge, there is no clear definition of speed-sintering of dental zirconia. However, speed-sintering typically consists of a fast heating step, a short dwell time at the sintering temperature, and a fast cooling step, enabling sintering in less than 240 min. Speed-sintering of YSZ within only 10-30 min has even been reported [14]. Notably, speed-sintering facilitates the clinical preparation of zirconia restorations in a single visit. Dental restorations can be designed, shaped, and sintered chairside by taking digital impressions using an intraoral scanner, customizing the restoration using computer-aided design (CAD) and computer-aided manufacturing/milling (CAM), and rapidly sintering the restoration by speed-sintering. As well as being less energy- and time-intensive than conventional sintering processes, speed-sintering also saves dentists' time by eliminating the need for temporary restorations. Additionally, patients benefit from this one-visit approach since they do not have to return for another appointment to set the restoration [15]. Nevertheless, the physical and mechanical properties and the microstructure of speed-sintered YSZ may differ from those of conventionally sintered YSZ [16], which means it remains unclear whether speed-sintered YSZ is suitable for clinical use.

Conventional sintering furnaces are based on conduction, radiation, and convection heating [17]. Conventional sintering of YSZ usually involves indirect heating via resistive heating elements (including molybdenum disilicide), which heat up the air in the furnace that in turn heats up the material surface. Heat slowly transmits from the surface to the restoration to the interior due to the low thermal conductivity (2.5–2.8 W/m•°C) of zirconia-based materials [1]. Because cracking can occur during rapid heating and cooling [18,19], a low heating rate (5–10 °C/min) and long dwell time (2–4 h) are typically used to achieve uniform heat distribution and optimum sintering [16,20]. In contrast, techniques such as microwave sintering [18,21], induction heating [16, 22], flash sintering [23–26], cold sintering [27–29], spark plasma sintering [30,31], hot-pressing [32], and selective laser sintering [33] can shorten the total sintering time and reduce the energy consumption. The application of some of these procedures to YSZ is currently being investigated.

Recently, a few manufacturers have synthesized YSZ via microwave sintering and induction heating. In both cases, electromagnetic energy is used to directly heat the material. Microwave sintering uses electromagnetic radiation transmitted from a magnetron. For ceramic materials, a frequency of 2.45 GHz and wavelength of 122 mm are typically used. Heat is generated by dipole oscillation or grain-boundary polarization within the ceramic, and some energy is transported by ionic conduction in the oscillating electric field [34,35]. During induction heating, energy is transmitted via an electric current produced by an electroconductive susceptor body under an alternating magnetic field. Zirconia-based restorations can be sintered in a fluctuating magnetic field when placed at the center of a susceptor body surrounded by a copper induction coil [16,22]. These novel speed-sintering methods enable zirconia-based restorations to be heated at rates of up to 342 °C/min, reducing the total sintering time to as low as 10 min [36, 37]. Some studies have reported that speed-sintered YSZ demonstrates comparable optical and mechanical properties, such as flexural strength, fracture toughness, hardness, and Weibull modulus, to those of conventionally sintered YSZ [21,22,36-40]. However, other studies

have indicated that speed-sintering can lower the flexural strength, light transmittance, and translucency parameter of YSZ [6,16,41–44]. Such contrasting reports make it difficult to predict the clinical performance of speed-sintered restorations.

As a prospective technique for producing YSZ dental restorations, speed-sintered YSZ needs to be examined comprehensively and the fundamental properties of speed-sintered YSZ should be documented sufficiently. Therefore, the objective of this systematic review and metaanalysis is to investigate the influence of speed-sintering on the optical and mechanical properties of YSZ. The null hypothesis is that there is no difference between the optical and mechanical properties of conventionally sintered and speed-sintered YSZ.

2. Materials and methods

The systematic review and meta-analysis were conducted according to the Preferred Reporting Items for Systematic Reviews and Meta-Analyses (PRISMA) 2020 guidelines [45] and registered in the International Prospective Register of Systematic Reviews (PROSPERO) network under number CRD42022335235. The population, intervention, comparison, outcome, and study design (PICOS) were as follows:

P (population): Monolithic YSZ with regular shapes (for example, disc and bar).

I (intervention): Speed-sintering.

C (comparison): Conventional sintering.

O (outcome): Changes in the optical and mechanical properties.

S (study design): In-vitro studies.

The focus question of this study is as follows:

"Does speed-sintering influence the optical and mechanical properties of YSZ?".

2.1. Eligibility criteria

The studies included in the systematic review and meta-analysis were identified through a search of three online databases and the reference lists of all identified articles, with the latter being peerreviewed publications. The inclusion criteria were as follows: (1) invitro study; (2) monolithic YSZ in a regular shape (disc or bar); (3) at least one speed-sintering procedure; (4) conventionally sintered YSZ as a control; (5) comparison of the optical and mechanical properties of speed-sintered and conventionally sintered YSZ; and (6) measurement of the optical and mechanical properties of speed-sintered and conventionally sintered YSZ. Studies were excluded if they met one or more of the following exclusion criteria: (1) *in-vivo* study; (2) zirconia framework with other materials, such as porcelain, metal, or acrylic; (3) zirconia crown or veneer; (4) zirconia containing a stabilizer other than Y₂O₃ (like CeO₂ or MgO); (5) no proper control; (6) no evaluation of the optical or mechanical properties prior to property-influencing processes such as low-temperature degradation and sandblasting; (7) data not related to differences in properties between speed-sintered and conventionally sintered YSZ; (8) no details of sintering; and (9) reviews, case reports, protocols, and clinical guidelines.

2.2. Search strategy



Fig. 1. Schematic of study selection according to the Preferred Reporting Items for Systematic Reviews and Meta-Analyses (PRISMA) statement.

(induction heat*)) OR (induction heating)) OR (induction heated)). Similar terms were employed in the Embase and Web of Science searches. The last database search was conducted on March 1, 2022.

2.3. Study screening and selection, and data extraction

Titles and abstracts were independently reviewed by two authors (H.

L. and K.X.), and studies were selected for full-text screening if they met most of the inclusion criteria. After full-text evaluation, two shortlists of potentially appropriate studies were provided by both authors, and a third author (W.T.) was consulted if consensus was not reached. The final decision about the included studies was made based on the discussion between the two authors (H.L. and K.X.) with the suggestion of the third author (W.T.). The coefficient of inter-examiner agreement

Table 1

Characteristics of the included studies.

Ref.	Year	Material tested	Type of zirconia	Conventional sintering program	Heating rate of conventional sintering	Speed sintering program	Heating rate of speed sintering	Tested properties
Almazdi et al.	2012	e-max ZirCAD ^a	3Y-TZP	1500 $^\circ \text{C}$ for 8 h	N/A	1500 $^\circ\text{C}$ for 30 min	N/A	Three-point flexural strength
Kim and Kim [42]	2017	Rainbow Shade A2 ^b	3Y-TZP	1500 °C for 2 h (Total time: 8 h)	N/A	1500 °C for 30 min (Total time: 2 h)	N/A	Translucency parameter (CIEDE2000)
Jerman et al. [13]	2020	Ceramill ZI ^c Ceramill Zolid ^c Ceramill Zolid	3Y-TZP 3Y-TZP 4Y-PSZ	1450 °C for 120 min 1450 °C for 120 min 1450 °C for 120 min	N/A N/A N/A	1580 °C for 10 min 1580 °C for 10 min 1580 °C for 10 min	N/A N/A N/A	Three-point flexural strength
Presenda	2015	HT+ ^c Lava ^d	3Y-TZP	1300/1400 °C for	10 °C/min to	1200/1300 °C for	100 °C/min to 1200/	Fracture
et al. [60]		Vita ^e	3Y-TZP	120 min 1300/1400 °C for	1300/1400 °C 10 °C/min to	10 min 1200/1300 °C for	1300 °C 100 °C/min to 1200/	toughness; Hardness
		T73YS ^f	3Y-T7P	120 min 1300/1400 °C for	1300/1400 °C 10 °C/min to	10 min 1200/1300 °C for	1300 °C 100 °C/min to 1200/	The difess
Dekkon [94]	2021	CopranZri ^h	3V T7D	120 min 1500 °C for 90 min	1300/1400 °C	10 min 1500 °C for 30 min	1300 °C	Translucency
Pekkali [04]	2021	Copranzii	51-1 <i>L</i> P	1500 C 101 90 IIIII	950 °C, 6 °C/min to 1500 °C	1500 C 101 50 mm	1100 °C, 20 °C/min to 1500 °C	parameter (CIELab)
		CopraSupreme ⁿ	4Y-PSZ	1500 °C for 90 min		1500 °C for 30 min		
Presenda et al. [56]	2017	N/A ^f	3Y-TZP	1400 °C for 2 h	10 °C/min to 1400 °C	1200/1300 °C for 10 min	100 °C/min to 1200/ 1300 °C	Fracture toughness;
Yılmaz Savaş and Akın [53]	2022	InCoris TZI ^g	3Y-TZP	1500 °C for 120 min (Total time: 8 h)	N/A	1540 °C for 25 min/ 158 °C for 10 min (Total time: 130 min)	N/A	Hardness Translucency parameter (CIEDE2000); Contrast ratio
Kim et al. [58]	2013	Lava Frame Zirconia ^d	3Y-TZP	1450 °C for 20 min to 40 h	10 °C/min to 1450 °C	1500 °C for 20 min	50 °C/min to 1500 °C	light transmittance
		KaVo Everest ZS- blanks ⁱ	3Y-TZP	1450 °C for 20 min to 40 h	7 °C/min to 1450 °C	1500 °C for 20 min	50 °C/min to 1500 °C	
Marinis et al. [39]	2013	KaVoʻ	3Y-TZP	1450 °C for 2 h (Total time: 10 h)	N/A	1520 °C for 35 min (Total time: 2 h)	N/A	Fracture toughness
		Lava 3 M ^d	3Y-TZP	1500 °C for 2 h (Total time: 8.5 h)	N/A	1520 °C for 35 min (Total time: 2 h)	N/A	
		Crystal HS ^j	3Y-TZP	1510 °C for 2 h (Total time: 10 h)	N/A	1520 °C for 35 min (Total time: 2 h)	N/A	
Presenda et al. [59]	2017	Lava Zirconia ^d	3Y-TZP	1400 °C for 2 h (Total time: 380 min)	10 °C/min to 1400 °C	1200 °C for 10 min (Total time: 30 min)	100 °C/min to 1200 °C	Fracture toughness;
		TZ-3YSE ^f	3Y-TZP	1400 °C for 2 h (Total time: 380 min)	10 °C/min to 1400 °C	1200 °C for 10 min (Total time: 30 min)	100 °C/min to 1200 °C	Hardness
Presenda et al. [99]	2015	Lava Zirconia ^d	3Y-TZP	1400 °C for 2 h	10 °C/min to 1400 °C	1200 °C for 10 min	100 °C/min to 1200 °C	Hardness
		TZ-3YSE ^f	3Y-TZP	1400 $^\circ C$ for 2 h	10 °C/min to 1400 °C	1200 $^\circ \text{C}$ for 10 min	100 °C/min to 1200 °C	
Jansen et al.	2019	Ceramill ZI ^c	3Y-TZP	1450 $^\circ \text{C}$ for 120 min	N/A	1570/1590 °C for 10 min	N/A	Light transmittance;
		Ceramill Zolid ^c	3Y-TZP	1450 $^\circ \text{C}$ for 120 min	N/A	1570/1590 °C for 10 min	N/A	Three-point flexural strength
		Ceramill Zolid HT+ ^c	4Y-PSZ	1450 $^\circ \text{C}$ for 120 min	N/A	1570/1590 °C for 10 min	N/A	
Al-Zordk and Saker [85]	2020	DD Bio ZX2 ^k	3Y-TZP	1450 $^\circ \mathrm{C}$ for 120 min	8 °C/min to 900 °C, 3 °C/min to 1450 °C	1450 $^\circ \text{C}$ for 50 min	60 °C/min to 990 °C, 13 °C/min to 1450 °C	Translucency parameter (CIELab):
		DD Cubex2 ^k	5Y-PSZ	1450 $^\circ \text{C}$ for 120 min	8 °C/min to 900 °C, 3 °C/min to 1450 °C	1450 $^\circ \text{C}$ for 50 min	60 °C/min to 990 °C, 13 °C/min to 1450 °C	Contrast ratio
		Zolid FX Preshaded ^c	5Y-PSZ	1450 $^\circ C$ for 120 min	20 °C/min to 900 °C, 10 °C/min to 1450 °C	1450 $^\circ \text{C}$ for 60 min	60 °C/min to 990 °C, 13 °C/min to 1450 °C	
		Zolid FX White ^c	5Y-PSZ	1450 $^\circ \text{C}$ for 120 min	20 °C/min to 900 °C, 10 °C/min to 1450 °C	1450 $^\circ \text{C}$ for 60 min	60 °C/min to 990 °C, 13 °C/min to 1450 °C	
Liu et al. [36]	2022	Katana HT ^l	4Y-PSZ	1500 °C for 120 min (Total time: \sim 7 h)	10 °C/min to 1500 °C	1515 °C for 30 min (Total time: ~90 min)	35 °C/min to 1515 °C	Translucency parameter (CIELab):
		Katana STML ¹	5Y-PSZ	1550 °C for 120 min (Total time: ~7 h)	10 °C/min to 1550 °C	1560 °C for 30 min/ 1560 °C for 6.9 min (Total time: ~90 min/18 min)	35 °C/min to to 1500 °C /342 °C/min to 1300 °C, 180 °C/ min to 1450 °C,	Biaxial flexural strength

24 °C/min to 1560 °C

(continued on next page)

Table 1 (continued)

Tuble I (colain	iiii)							
Ref.	Year	Material tested	Type of zirconia	Conventional sintering program	Heating rate of conventional sintering	Speed sintering program	Heating rate of speed sintering	Tested properties
		Katana UTML ¹	6Y-PSZ	1550 °C for 120 min (Total time: \sim 7 h)	10 °C/min to 1550 °C	1560 °C for 30 min (Total time: ~90 min)	35 °C/min to 1560 °C	
		Zpex 4 ^f	4Y-PSZ	1450 °C for 120 min (Total time: ~7 h)	10 °C/min to 1450 °C	1450 °C for 30 min (Total time: ~90 min)	35 °C/min to 1450 °C	
		Zpex Smile ^f	5Y-PSZ	1450 °C for 120 min (Total time: ~7 h)	10 °C/min to 1450 °C	1450 °C for 30 min (Total time: \sim 90 min)	35 °C/min to 1450 °C	
Yang et al. [86]	2020	Copran Zr-I ^h	3Y-TZP	1500 $^\circ\text{C}$ for 90 min	10 °C/min to 950 °C, 6 °C/min to 1500 °C	1500 °C for 30 min	50 °C/min to 1100 °C, 20 °C/min to 1500 °C	Translucency parameter (CIEDE):
		Copran Zr-I Ultra-T white ^h	3Y-TZP	1500 $^\circ \text{C}$ for 90 min	10 °C/min to 950 °C, 6 °C/min to 1500 °C	1500 $^\circ \text{C}$ for 30 min	50 °C/min to 1100 °C, 20 °C/min to 1500 °C	Biaxial flexural strength;
		Copran Zr-I Ultra-T A2 ^h	3Y-TZP	1500 $^\circ\text{C}$ for 90 min	10 °C/min to 950 °C, 6 °C/min to 1500 °C	1500 $^\circ \text{C}$ for 30 min	50 °C/min to 1100 °C, 20 °C/min to 1500 °C	flexural strength; Hardness
		Cecron HT ^g	3Y-TZP	1520 $^\circ\text{C}$ for 130 min	22 °C/min to 880 °C, 11 °C/min to 1520 °C	1540 $^\circ \text{C}$ for 35 min	70 °C/min to 1540 °C	
		Cecron XT ^g	5Y-PSZ	1520 $^\circ\text{C}$ for 130 min	22 °C/min to 880 °C, 11 °C/min to 1520 °C	1540 $^\circ \text{C}$ for 35 min	70 °C/min to 1540 °C	
Cokic et al. [22]	2020	Katana STML ¹	5Y-PSZ	1550 °C for 2 h (Total time: 6.8 h)	10 °C/min to 1550 °C	1560 °C for 16 min (Total time: 28 min)	350 °C/min to 1300 °C, 150 °C/min to 1500 °C, 10 °C/ min to 1560 °C	Translucency parameter (CIEDE); Contrast ratio; Biaxial flexural strength; Fracture toughness:
Ai et al. [61]	2015	N/A ^m	3Y-TZP	1350 °C for 2 h (the first sintering step: 900 °C for 30 min)	25 °C/min to 900 °C, 5 °C/min to 1350 °C	1350 °C for 30 min (First step: 900 °C for 30 min)	25 °C/min to 900 °C, 20 °C/min to 1350 °C; 25 °C/min to 1350 °C	Hardness Three-point flexural strength; Fracture toughness; Hardness
Borrell et al. [34]	2013	3Y-TZP-B ^f	3Y-TZP	1400 $^\circ \text{C}$ for 60 min	5 °C/min to 1400 °C	1100/1200/1300/ 1400 °C for 5/10/ 15 min	25 °C/min to 1100/ 1200/1300/1400 °C	Fracture toughness;
Luz et al. [18]	2021	Vipi Block ⁿ	3Y-TZP	1530 °C for 2 h (Total time: 600 min)	N/A	1450 °C for 15 min (Total time: 105 min)	30 °C/min	Translucency parameter (CIELab); Biaxial flexural strength
Liu et al. [87]	2022	Copran Zr-I Ultra- T ^h	3Y-TZP	${\sim}1500~^\circ\text{C}$ for 60 min (Total time: 300 min)	N/A	~1500 °C for 15 min (Total time: 90 min)	N/A	Translucency parameter (CIELab)
		Cercon ht ^g	3Y-TZP	~1540 °C for 90 min (Total time: 270 min)	N/A	~1540 °C for 15 min (Total time: 60 min)	N/A	
		Cercon xt ^g	5Y-PSZ	\sim 1540 °C for 90 min (Total time: 270 min)	N/A	~1540 °C for 15 min (Total time: 60 min)	N/A	
Li et al. [54]	2019	ST-3Y-TZP A2 preshaded Zirconia Blank ^o	3Y-TZP	1480 °C for 2 h (Total time: 15 h)	20 °C/min to 1480 °C	1580 °C for 2–20 min (Total time: 12–40 min)	400 °C/min to 1200 °C, 190 °C/min to 1480 or 1580 °C	Light transmittance; Three-point flexural strength; Fracture toughness; Hardness
Lawson and Maharishi	2020	Katana STML ¹	5Y-PSZ	1550 °C for 2 h (Total time: 7 h)	N/A	N/A (Total time: 18/30 min)	N/A	Translucency parameter
[10]			51-P32	(Total time: 7 h)	IN/ A	30 min)	1N/ A	Three-point
		Zpex Smile ^t	5Y-PSZ	1550 °C for 2 h (Total time: 7 h)	N/A	N/A (Total time: 30 min)	N/A	flexural strength
Moratal et al. [19]	2021	3Y-TZP ^f	3Y-TZP	1400 $^\circ \text{C}$ for 60 min	10 °C/min to 1400 °C	1200/1300 °C for 10 min	30 °C/min to 1200/ 1300 °C	Contrast ratio; Fracture

toughness; Hardness

(continued on next page)

Table 1 (continued)

Ref.	Year	Material tested	Type of zirconia	Conventional sintering program	Heating rate of conventional sintering	Speed sintering program	Heating rate of speed sintering	Tested properties
Jiang et al. [52]	2015	Lava Zirconia ^d Zenostar Zirconia ⁿ	YSZ YSZ	1500 °C for 2 h 1490 °C for 2 h	N/A N/A	1420 °C for 30 min 1420 °C for 30 min	15 °C/min to 1420 °C 15 °C/min to 1420 °C	Translucency parameter (CIElab)
Han et al. [55]	2018	Domestic TZP ^o	YSZ	1530 °C for 8.2 h	N/A	1578 °C for 13.5 min	N/A	Biaxial flexural strength; Fracture toughness; Hardness
Ribeiro et al. [57]	2019	Experimental TZP	3Y-TZP	1530 °C for 2 h	10 °C/min to 800 °C, 5 °C/min to 1530 °C	1350/1450 °C for 15/30 min	25 °C/min to 1000 °C, 15 °C/min to 1300 °C, 5 °C/min to 1350/1450 °C	Fracture toughness; Hardness
		Vita In-Ceram 2000 YZ Cubes 4019 ^e	3Y-TZP	1530 $^\circ \rm C$ for 2 h	10 °C/min to 800 °C, 5 °C/min to 1530 °C	1350/1450 °C for 15/30 min		

^a Ivoclar Vivadent, Schaan, Liechtenstein; ^bGenoss, Suwon, South Korea; ^cAmann Girrbach, Pforzheim, Germany; ^d3M Oral Care, St.Paul, MN,USA; ^eVita Zahnfabrik, Bad Säckingen, Germany; ^fTosoh, Tokyo, Japan; ^hWhitepeaks Dental Solutions, Hamminkeln, Germany; ^gDentsply Sirona, Bensheim, Germany; ⁱKavo Dental, Biberach, Germany; ^jDLMS, Scottsdale, AZ, USA; ^kDental Direct, Spenge, Germany; ^lKuraray Noritake Dental, Tokyo, Japan; ^mYixing ultrafine powder company, Yixing, China; ⁿVipi Wieland, Francisco Beltrão, Brazil; ^oUpcera, Shenzhen, China; ^pZirconzahn, South Tyrol, Italy.

(kappa coefficient) was calculated for the two authors (H.L. and K.X.).

Data were extracted by two investigators (H.L. and K.X.). The following data were acquired from the included studies: author(s), publication year, title, publication journal, sample size, brand and commercial name of YSZ, type of material, sample shape (bar or disc), type of speed sintering, evaluation method, and numerical results of the study. For meta-analysis, the confidence intervals (CIs) were converted to means and standard deviations if the study provided CIs only, and the groups were combined into one if multiple conventional sintering or speed-sintering protocols were investigated in one study. The abovementioned conversions were performed using spreadsheet software (Microsoft Office Excel 365 for Windows) according to the Cochrane Handbook for Systematic Reviews of Interventions [46]. One author (H. L.) contacted the corresponding author(s) of the studies via e-mail when some essential data, such as mean values, standard deviation, or sample size, were missing. Studies were excluded from the meta-analysis if the required information could not be obtained.

2.4. Risk of bias assessment

Risk of bias was assessed by two reviewers (H.L. and K.X.) based on the modified Consolidated Standards of Reporting Trials (CONSORT) checklist [47]. The following items were evaluated for quality assessment: (1) structured summary in the abstract; (2a) adequate information on scientific background and rationale; (2b) clear and well-formulated objectives and/or hypotheses; (3) specific information on the intervention enabling study replication; (4) pre-specified primary and secondary measures of outcome; (5) sample-size justification; (6) method of specimen randomization; (7) mechanism of specimen randomization; (8) person who performed specimen randomization; (9) person blinded before intervention practices; (10) statistical analysis used; (11) complete dataset; (12) description of the study limitations; (13) listed funding or other supports mentioned; and (14) study protocol available for potential readers. Each item was evaluated as reported or not reported according to the study description.

Another risk of bias evaluation was based on and adapted from previous studies [48–50]. The following parameters were utilized to evaluate the qualities of the included articles: (1) sample-size justification; (2) randomization; (3) sintering; (4) sample preparation; (5) adequate statistical analysis; (6) measuring procedures of each experiment; and (7) tests executed by a single-blinded operator. If the parameter was clearly reported, the study received a "0"; if the parameter was written but detailed information was missing, the study received a "1"; if a parameter was not mentioned at all, the study

received a "2". Studies were considered low-, medium-, and high-risk if they had a total sum of values in the range of 0–4, 5–9, and 10–14, respectively.

2.5. Data analyses

Meta-analysis was applied to a parameter if that parameter was included in at least three different studies and at least five different datasets were available, and the studies revealing comprehensive metaanalytic data of that parameter accounted for at least 50% of the studies. Therefore, the data for the CIELab-based translucency parameter, CIEDE2000-based translucency parameter, contrast ratio, three-point flexural strength, biaxial flexural strength, and fracture toughness were considered for meta-analysis. Other properties such as light transmittance and hardness were included in the systematic review, but are presented as descriptive data because some information for metaanalysis was lacking.

Mean values (mean), standard deviations (SD), and sample sizes (*n*) of each study were included in the meta-analysis. The data for the meta-analysis were combined in a random-effects model, and a heterogeneity study was conducted by Q and I^2 tests. p < 0.1 in the Q test and $I^2 > 50\%$ in the I^2 test indicated statistically high heterogeneity [51]. R statistical software (version 4.1.2, R Foundation for Statistical Computing, Vienna, Austria) was used in this study, and the significance level for the analyses was set as $\alpha = 0.05$.

3. Results

3.1. Search and selection

Fig. 1 shows a schematic of the study search and selection process according to the PRISMA statement. The initial electronic searches of PubMed, Web of Science, and Embase identified 142, 865, and 130 studies, respectively. A total of 224 studies were rejected because of duplications. After removing the duplicate studies, 870 studies were excluded based on the titles and abstracts reviewed, and the remaining 43 studies were selected for full-text analysis. Finally, 26 studies were included in the systematic review, among which 25 were written in English and one was written in Chinese [52]. The kappa coefficients for screening and eligibility were 0.98 and 0.91, respectively.

Detailed information on the included articles is provided in Table 1, where the authors, year, material tested, type of YSZ, and details of speed-sintering and conventional sintering for each article are presented.

Table 2

Parameters examined in the included studies for descriptive analysis and meta-analysis.

Ref.	CIELab-based translucency	CIEDE2000-based translucency	Contrast ratio	Light transmittance	Three-point flexural	Biaxial flexural	Fracture toughness	Hardness
					strength	strength		
Almazdi et al. [21]	-	-	-	-	0	-	-	-
Kim and Kim [42]	-	0	-	-	-	-	-	-
Jerman et al. [13]	-	-	-	-	0	-	-	-
Presenda et al. [60]	-	-	-	-	-	-	*	*
Pekkan (2021)[84]	0	-	-	-	-	-	-	-
Presenda et al. [56]	-	-	-	-	-	-	0	0
Yılmaz Savaş and Akın [53]	-	0	0	-	-	-	-	-
Kim et al. [58]	-	-	-	0	-	-	-	-
Marinis et al. [39]	-	-	-	-	-	-	0	-
Presenda et al. [59]	-	-	-	-	-	-	*	*
Presenda et al. [99]	-	-	-	-	-	-	-	*
Jansen et al. [41]	-	-	-	*	*	-	-	-
Al-Zordk and Saker (2020)[85]	Ō	-	0	-	-	-	-	-
Liu et al. [36]	o(+)	-	-	-	-	0	-	-
Yang et al. [86]	*	-	-	-	*	*	-	*
Cokic et al. [22]	○(+)	-	0	-	-	0	○(+)	○(+)
Ai et al. [61]	-	-	-	-	*	-	*	*
Borrell et al. [34]	-	-	-	-	-	-	*	*
Luz et al. [18]	○(+)	-	-	-	-	○(+)	-	-
Liu et al. [87]	*	-	-	-	-	-	-	-
Li et al. [54]	-	-	-	*	0	-	0	0
Lawson and Maharishi		0	-	-	0	-	-	-
)[16]								
Moratal et al. [19]	-	-	*	-	-	-	*	*
Jiang et al. [52]	0	-	-	-	-	-	-	-
Han et al. [55]	-	-	-	-	-	0	0	0
Ribeiro et al. (2019)	-	-	-	-	-	-	0	0
[3/] Tatal	0	0	4	0	7	F	11	10
Total	8	3	4	3	/	5	11	12
information articles	6	3	3	1	4	4	6	5
Percentage of full information articles	75%	100%	75%	33.3%	57.1%	80%	54.5%	41.7%
Number of datasets	22	7	6	1	9	7	8	6

 \circ : With full information for meta-analysis; *: Lack of essential information for meta-analysis; \circ (+): Information directly obtained from the author.

Table 3

Risk of bias in the included studies according to the modified Consolidated Standards of Reporting Trials (CONSORT) checklist. Each item is marked as reported (*) or not (empty box) according to the description of each study.

Ref.	1	2a	2b	3	4	5	6	7	8	9	10	11	12	13	14
Almazdi et al. [21]	*	*	*	*	*						*	*	*		
Kim and Kim [42]	*	*	*	*	*						*	*	*	*	
Jerman et al. [13]	*	*	*	*	*						*	*	*	*	
Presenda et al. [60]	*	*	*	*	*									*	
Pekkan [84]	*	*	*	*	*						*	*	*	*	
Presenda et al. [56]	*	*	*	*	*							*		*	
Yılmaz Savaş and Akın [53]	*	*	*	*	*	*	*	*			*	*	*		
Kim et al. [58]	*	*	*	*	*						*	*	*	*	
Marinis et al. [39]	*	*	*	*	*	*					*	*		*	
Presenda et al. [59]	*	*	*	*	*							*		*	
Presenda et al. [99]	*	*	*	*	*							*		*	
Jansen et al. [41]	*	*	*	*	*						*	*	*	*	
Al-Zordk and Saker [85]	*	*	*	*	*						*	*	*		
Liu et al. [36]	*	*	*	*	*	*					*	*	*	*	
Yang et al. [86]	*	*	*	*	*						*			*	
Cokic et al. [22]	*	*	*	*	*						*	*		*	
Ai et al. [61]	*	*	*	*	*									*	
Borrell et al. [34]	*	*	*	*	*									*	
Luz et al. [18]	*	*		*	*						*	*		*	
Liu et al. [87]	*	*	*	*	*						*		*	*	
Li et al. [54]	*	*	*	*	*						*	*			
Lawson and Maharishi [16]	*	*	*	*	*						*	*	*	*	
Moratal et al. [19]	*	*	*	*	*									*	
Jiang et al. [52]	*	*	*	*	*						*	*			
Han et al. [55]	*	*	*	*	*						*	*		*	
Ribeiro et al. [57]	*	*	*	*	*							*		*	

Table 4

Risk of bias in the included studies adapted and modified from previous studies.

Ref.	Sample size	Random	Sintering	Sample preparation	Statistical analysis	Measuring procedures	Operator	Total	Risk of bias
Almazdi et al. [21]	2	2	1	0	0	0	2	7	Medium
Kim and Kim [42]	2	2	1	0	0	0	2	7	Medium
Jerman et al. [13]	2	2	1	0	0	0	2	7	Medium
Presenda et al. [60]	2	2	1	0	2	0	2	9	Medium
Pekkan [84]	2	2	0	0	0	0	2	6	Medium
Presenda et al. [56]	2	2	1	1	2	0	2	10	High
Yılmaz Savaş and Akın	0	0	1	0	0	0	2	3	Low
Kim et al [58]	2	2	1	0	0	0	2	7	Medium
Marinis et al. [39]	0	2	1	0	0	0	2	5	Medium
Presenda et al. [59]	2	2	1	0	2	0	2	9	Medium
Presenda et al. [99]	2	2	1	0	2	0	2	9	Medium
Jansen et al. [41]	2	2	1	0	0	0	2	7	Medium
Al-Zordk and Saker [85]	2	2	0	0	0	0	2	6	Medium
Liu et al. [36]	0	2	0	0	0	0	2	4	Low
Yang et al. [86]	2	2	0	0	0	0	2	6	Medium
Cokic et al. [22]	2	2	0	0	0	0	2	6	Medium
Ai et al. [61]	2	2	1	0	2	1	2	10	High
Borrell et al. [34]	2	2	1	0	2	0	2	9	Medium
Luz et al. [18]	2	2	0	1	0	0	2	7	Medium
Liu et al. [87]	2	2	0	1	0	0	2	7	Medium
Li et al. [54]	2	2	0	1	2	0	2	9	Medium
Lawson and Maharishi	2	2	1	0	0	0	2	7	Medium
Moratal et al. [19]	2	2	1	0	2	1	2	10	High
Jiang et al. [52]	2	2	1	0	0	0	2	7	Medium
Han et al. [55]	2	2	1	0	0	0	2	7	Medium
Ribeiro et al. [57]	2	2	0	0	2	0	2	8	Medium

The main focus of this review is to investigate the effect of speedsintering on the optical and mechanical properties of YSZ (Table 2). The optical properties comprise the CIELab-based translucency parameter, CIEDE2000-based translucency parameter, contrast ratio, and light transmittance, and the mechanical properties include threepoint flexural strength, biaxial flexural strength (piston-on-three-ball

	Spee	d sinte	ring	Conv	ention	al sinteri	ng St	andardized	Mean			
Study	Total	Mean	SD	Total	Mean	SD		Difference	9	SMD	95%-CI	Weight
Pekkan 2021 3Y 0.7mm	5	8.22	0.2900	5	8.63	0.2900				-1.28	[-2.70; 0.15]	4.7%
Pekkan 2021 3Y 1.0mm	5	6.41	0.1300	5	6.98	0.7300		-+-		-0.98	[-2.33; 0.37]	4.8%
Pekkan 2021 3Y 1.3mm	5	5.09	0.1300	5	5.26	0.3700				-0.55	[-1.83; 0.72]	4.9%
Pekkan 2021 4Y 0.7mm	5	13.07	0.2100	5	12.75	0.1500		+		1.58	[0.07; 3.10]	4.6%
Pekkan 2021 4Y 1.0mm	5	11.24	0.1300	5	11.17	0.1000				0.54	[-0.73; 1.82]	4.9%
Pekkan 2021 4Y 1.3mm	5	9.73	0.1400	5	9.64	0.3600		-+-		0.30	[-0.95; 1.55]	4.9%
Pekkan 2021 5Y 0.7mm	5	11.59	0.4900	5	13.02	0.4200				-2.83	[-4.82; -0.83]	4.0%
Pekkan 2021 5Y 1.0mm	5	10.43	0.7800	5	11.76	0.6100				-1.71	[-3.28; -0.15]	4.5%
Pekkan 2021 5Y 1.3mm	5	9.40	0.4000	5	10.82	0.4900				-2.87	[-4.88; -0.85]	3.9%
AI-Zordk and Saker 2020 5Y FXP	10	11.03	0.0500	10	11.01	0.0600		+		0.35	[-0.54; 1.23]	5.3%
AI-Zordk and Saker 2020 5Y FXW	10	10.53	0.1400	10	10.84	0.1000		-+-		-2.44	[-3.65; -1.23]	5.0%
AI-Zordk and Saker 2020 5Y DDC	10	11.06	0.1300	10	11.14	0.1000		+		-0.66	[-1.57; 0.24]	5.3%
AI-Zordk and Saker 2020 3Y DDB	10	10.06	0.1400	10	9.88	0.1300		-+-		1.28	[0.30; 2.26]	5.2%
Liu et al. 2022 4Y KATANA HT	12	17.65	0.2500	12	18.14	0.3800		+		-1.47	[-2.39; -0.55]	5.3%
Liu et al. 2022 5Y KATANA STML	24	23.46	0.3700	12	23.15	0.5000		+		0.73	[0.01; 1.44]	5.5%
Liu et al. 2022 6Y KATANA UTML	12	25.03	0.4400	12	25.40	0.4800		-		-0.78	[-1.61; 0.06]	5.4%
Liu et al. 2022 4Y Zpex 4	12	20.98	0.2700	12	21.11	0.4000		+		-0.37	[-1.18; 0.44]	5.4%
Liu et al. 2022 5Y Zpex Smile	12	15.34	0.5400	12	24.25	0.3500				-18.90	[-24.78; -13.03]	1.1%
Cokic et al. 2020 5Y	7	31.28	0.3600	7	32.29	0.6400		-+-		-1.82	[-3.13; -0.51]	4.8%
Luz et al. 2021 3Y	4	13.00	1.0000	4	29.00	0.8000		- 1		-15.35	[-25.71; -4.99]	0.4%
Jiang et al. 2015 YSZ Lava	5	19.32	0.3700	5	19.22	0.1500				0.32	[-0.93; 1.57]	4.9%
Jiang et al. 2015 YSZ Zenostart	5	18.53	0.2400	5	18.49	0.3800		-		0.11	[-1.13; 1.35]	4.9%
Random-effects model	178			166				•		-0.84	[-1.54; -0.14]	100.0%
Heterogeneity: $l^2 = 82\%$, $\tau^2 = 2.1611$, p	< 0.01								10 2			
							-20 -1	0 0	10 2	0		

Fig. 2. Forest for CIELab-based translucency based on the meta-analysis (SD, standard deviation; CI, confidence interval; SMD, standardized mean difference).

	Spee	d sinte	ring	Conv	entiona	al sintering	Standardized Mean			
Study	Total	Mean	SD	Total	Mean	SD	Difference	SMD	95%-CI	Weight
Kim and Kim 2017 3Y 0.5mm	9	11.43	0.1500	9	11.52	0.1500	-	-0.57	[-1.52; 0.38]	14.9%
Kim and Kim 2017 3Y 1.0mm	9	7.50	0.1100	9	7.87	0.1300	—	-2.93	[-4.34; -1.51]	14.8%
Kim and Kim 2017 3Y 1.5mm	9	5.28	0.1200	9	5.31	0.2000	-	-0.17	[-1.10; 0.75]	14.9%
Yilmaz Savas and Akin 2022 3Y	20	7.26	0.5600	10	7.37	0.3400	-	-0.21	[-0.98; 0.55]	14.9%
Lawson and Maharishi 2020 5Y KATANA STML	20	7.63	0.2200	10	7.88	0.2500	-	-1.06	[-1.87; -0.25]	14.9%
Lawson and Maharishi 2020 5Y Prettau Anterior	10	3.96	0.2600	10	7.88	0.2700		-14.16	[-19.11; -9.22]	13.5%
Lawson and Maharishi 2020 5Y Zpex Smile	10	5.17	0.1200	10	8.47	0.1700 —		-21.48	[-28.91; -14.04]	12.0%
Random-effects model	87			67				-5.22	[-10.96; 0.53]	100.0%
Heterogeneity: $l^2 = 92\%$, $\tau^2 = 57.3117$, $p < 0.01$										
							-20 -10 0 10	20		

Fig. 3. Forest for CIEDE2000-based translucency according to the meta-analysis (SD, standard deviation; CI, confidence interval; SMD, standardized mean difference).



Fig. 4. Forest for contrast ratio based on the meta-analysis (SD, standard deviation; CI, confidence interval; SMD, standardized mean difference).

method), fracture toughness, and hardness.

3.2. Risk of bias

The risk of bias in the evaluated studies is presented in Tables 3 and 4. The modified CONSORT scale demonstrated that all the articles had sufficient information on the structured summary, scientific background, intervention, and pre-specified measures of outcome, and most of the included studies contained information on the specific objectives, statistical methods, limitations, and sources of funding or other support. However, only Yılmaz Savaş and Akın [53], Marinis et al. [39], and Liu et al. [36] reported how the sample size was determined. Only Yılmaz Savaş and Akın [53] reported the method used to implement the

random-allocation sequence and the author who conducted the randomization. The remaining items were not noted in any of the included studies. Regarding the evaluation of risk of bias, as adapted and modified from previous studies, most of the studies exhibited a medium risk of bias, two studies demonstrated a low risk of bias, and three studies presented a high risk of bias.

3.3. Meta-analysis

Six meta-analyses were conducted for the CIELab-based translucency parameter, CIEDE2000-based translucency parameter, contrast ratio, three-point flexural strength, biaxial flexural strength (piston-on-threeball method), and fracture toughness. Some studies were included in the

	Spee	d sinterin	g	Conv	entional s	sintering	Standardized Mean			
Study	Total	Mean	SD	Total	Mean	SD	Difference	SMD	95% -CI	Weight
Almazdi <i>et al.</i> 2012 3Y	10	1080.08	79.3700	10	1108.33	162.5500		-0.21	[-1.09; 0.67]	12.8%
Jerman <i>et al.</i> 2020 3Y ZI	16	1020.00	161.0000	16	904.00	162.0000		0.70	[-0.02; 1.42]	12.9%
Jerman et al. 2020 3Y Zolid	16	669.00	66.0000	16	665.00	96.0000		0.05	[-0.65; 0.74]	12.9%
Jerman <i>et al.</i> 2020 4Y HT+	16	990.00	197.0000	16	596.00	60.0000		2.64	[1.66; 3.61]	12.7%
Li <i>et al.</i> 2019 3Y	30	1092.00	156.0000	10	1172.00	73.0000		-0.56	[-1.28; 0.17]	12.9%
Lawson and Maharishi 2020 5Y KATANA STML	20	823.63	86.4600	10	761.36	6.8000		0.85	[0.06; 1.64]	12.9%
Lawson and Maharishi 2020 5Y Prettau Anterior	10	557.62	46.1200	10	787.38	50.4800	— <mark>—</mark> —	-4.55	[-6.34; -2.76]	11.8%
Lawson and Maharishi 2020 5Y Zpex Smile	10	493.66	65.7400	10	789.77	6.5600		-6.07	[-8.33; -3.81]	11.1%
Random-effects model	128			98				-0.77	[-2.68; 1.14]	100.0%
Heterogeneity: $I^2 = 92\%$, $\tau^2 = 7.2024$, $\rho < 0.01$										
							-5 0 5			

Fig. 5. Forest for three-point flexural strength according to the meta-analysis (SD, standard deviation; CI, confidence interval; SMD, standardized mean difference).

H. Liu et al.



Fig. 6. Forest for biaxial flexural strength based on the meta-analysis (SD, standard deviation; CI, confidence interval; SMD, standardized mean difference).

meta-analysis more than once, because two or more different types of YSZ or specimen thicknesses were evaluated.

For CIELab-based translucency, three articles involving 22 comparisons were analyzed using a random-effects model. The results indicated that the CIELab-based translucency parameter of conventionally sintered samples was higher than that of speed-sintered samples (mean difference (MD) = -0.84; 95% CI = -1.54 to -0.14; p = 0.0183) (Fig. 2). The meta-analysis exhibited high heterogeneity: $I^2 = 82\%$ and Q = 119.5 (p < 0.0001).

For CIEDE2000-based translucency, three articles combining seven datasets were included in the meta-analysis and subsequently evaluated using a random-effects model. The results indicated that there is no significant difference in CIEDE2000-based translucency between speed-sintered and conventionally sintered samples (MD = -5.22; 95% CI = -10.96 to 0.53; p = 0.0750) (Fig. 3). The heterogeneity was high: $I^2 = 92\%$ and Q = 70.98 (p < 0.0001).

For contrast ratio, six datasets from three studies were analyzed using a random-effects model. The results demonstrated that the contrast ratio of speed-sintered samples was comparable to that of conventionally sintered samples (MD = 0.06; 95% CI = -0.50 to 0.63; p = 0.8250) (Fig. 4). The heterogeneity was intermediate: $I^2 = 56\%$ and Q = 11.35 (p = 0.0449).

For three-point flexural strength, four articles with eight datasets were included in the meta-analysis. The specimen sizes were $25 \times 4.5 \times 2.5 \text{ mm}$ [21], $25 \times 4 \times 1 \text{ mm}$ [13], $20 \times 4 \times 1.2 \text{ mm}$ [54], and $16 \times 4 \times 1.2 \text{ mm}$ [16]. One study followed ISO 2006 [21] and one study followed ISO 6872 [16]. Jerman et al. [13] reported that ISO 6872:2019 was not strictly followed because the increased material

thickness might hide the influence of phase transformation on the flexural strength. A random-effects model was utilized to assess the three-point flexural strength of different samples. The results indicated that the three-point flexural strength of speed-sintered YSZ bars was comparable to that of conventionally sintered YSZ bars (MD = -0.77; 95% CI = -2.68 to 1.14; p = 0.4302) (Fig. 5). The heterogeneity was high: $I^2 = 92\%$ and Q = 89.99 (p < 0.0001).

For biaxial flexural strength (piston-on-three-ball method), data from four studies combining eight estimations were examined using a random-effects model. The specimen sizes were Φ 14.5 × 1.2 mm [36], 12 × 12 × 1.2 mm [22], and Φ 12 × 1.2 mm [18,55]. Three studies followed ISO 6872:2015 [18,22,36] and one study followed ISO 6872:2008 [55]. The results demonstrated that the biaxial flexural strength of speed-sintered YSZ discs was similar to that of conventionally sintered YSZ discs (MD = -0.07; 95% CI = -0.52 to 0.39; p = 0.7691) (Fig. 6). The heterogeneity was intermediate: $I^2 = 69\%$ and Q = 22.42 (p = 0.0021).

For fracture toughness, six studies with eight datasets were included in the meta-analysis. The indentation method was employed to measure the fracture toughness in five studies [22,54–57], whereas one study [39] used the single edge V-notch beam method. The datasets were analyzed using a random-effects model. The results implied that speed-sintered YSZ has a similar fracture toughness to conventionally sintered YSZ (MD = -0.85; 95% CI = -1.80 to 0.00; p = 0.0502) (Fig. 7). The heterogeneity was intermediate: $I^2 = 70\%$ and Q = 23.40(p = 0.0015).



Fig. 7. Forest for fracture toughness according to the meta-analysis (SD, standard deviation; CI, confidence interval; SMD, standardized mean difference).

Outcomes of the par nalvsis.	ameters invest	igated in	the included studies for descriptive	Parameter	Ref.
Parameter	Ref.	Year	Outcome	investigated	
Translucency parameter (CIELab)	Yang et al. [86]	2020	Translucency parameters of conventionally sintered ZW and XT higher than those of other conventionally sintered materials; No significant differences between the translucency parameters of	Hardness	Moratal et al. [19 Presenda
	Liu et al.	2022	conventionally sintered and rapid- sintered UW, UT, and HT. Translucency parameters of		
	[87]		conventionally sintered samples comparable to those of corresponding speed-sintered samples, except XT, whose translucency parameter was calculated using the colorimeter		Presenda et al. [56
Contrast ratio	Moratal et al. [19]	2021	Contrast ratio of conventionally sintered NK00 similar to that of		Presenda
Light transmittance	Kim et al. [58]	2013	microwave-sintered NK00. Light transmittances of microwave- sintered Lava and KaVo higher than those of conventionally sintered Lava and KaVo.		
	Jansen et al. [41]	2019	Light transmittances of conventionally sintered ZD and HT+ (all thicknesses) and ZI (thickness of 2.5–3 mm) higher than those of corresponding speed- sintered samples		Presenda et al. [99
	Li et al. [54]	2019	Light transmittance of rapid- sintered sample (R-1) lower than those of conventionally sintered and rapid-sintered samples (R-2 and R- 2, representiable)		
Three-point flexural strength	Jansen et al. [41]	2019	No significant differences between three-point flexural strengths of conventionally sintered and speed- sintered samples for a given type of		Yang et a [86]
	Yang et al. [86]	2020	zirconia. No significant differences between three-point flexural strengths of conventionally sintered and speed- sintered samples for a given type of		[22] Ai et al. [61]
	Ai et al. [61]	2015	zirconia. Three-point flexural strength of microwave-sintered sample higher than that of conventionally sintered sample		Borrell e [34]
Biaxial flexural strength	Yang et al. [86]	2020	No significant differences between biaxial flexural strengths of conventionally sintered and speed- sintered samples for a given type of zirconia.		Li et al. [
Fracture toughness	Presenda et al. [60]	2015	No significant differences between biaxial flexural strengths of conventionally sintered and microwave-sintered LAVA and TOSOH.		Moratal et al. [19
			Fracture toughness of microwave- sintered VITA lower than that of conventionally sintered VITA.		Han et a [55]
	Presenda et al. [59]	2017	No considerable differences between fracture toughnesses of conventionally sintered and speed- sintered samples.		Ribeiro e [57]
	Ai et al. [61]	2015	Flexural toughness of microwave- sintered sample higher than that of conventionally sintered sample.		
	Borrell et al. [34]	2013	At 1400 °C, flexural toughness of conventionally sintered and microwave (5 min)-sintered samples higher than those of		

neter tigated	Ref.	Year	Outcome
			microwave (10 and 15 min)-
			sintered samples.
	Moratal	2021	Fracture toughness of microwave
	et al. [19]		(1200 °C)-sintered NK00 higher
			than those of conventionally
			sintered and microwave (1300 °C)-
	Duccon do	2015	sintered NKOU samples.
less	ot al [60]	2015	the hardnesses of conventionally
			sintered and microwave sintered
			LAVA and VITA
			Hardness of microwave-sintered
			TOSOH higher than those of other
			microwave-sintered samples.
	Presenda	2017	3Y-TZP hardness values:
	et al. [56]		conventionally sintered, 13.9
			\pm 0.4 GPa; microwave-sintered,
			13.9 ± 0.8 GPa (1200 $^\circ\text{C})$ and 14.7
			\pm 0.6 GPa (1300 °C).
	Presenda	2017	LAVA hardness values:
	et al. [59]		conventionally sintered, 12.9
			\pm 0.2 GPa; microwave-sintered,
			13.0 ± 0.3 GPa.
			conventionally sintered 13.0
			± 0.1 GPa: microwave-sintered
			\pm 0.1 Gra, increasing the sintered, 13.4 \pm 0.4 GPa.
	Presenda	2015	LAVA hardness values:
	et al. [99]		conventionally sintered, 15.9
			\pm 0.5 GPa, whereas that of
			microwave-sintered LAVA was 15.8
			\pm 0.6 GPa.
			LAB zirconia hardness values:
			conventionally sintered, 17.0
			\pm 0.4 GPa; microwave-sintered,
	I	0000	16.8 ± 0.6 GPa.
	Yang et al.	2020	No significant differences between
	[00]		sintered and speed-sintered
			samples
	Cokic et al.	2020	Hardness of speed-sintered sample
	[22]		higher than that of conventionally
			sintered sample.
	Ai et al.	2015	Hardness of microwave-sintered
	[61]		sample higher than that of
			conventionally sintered sample.
	Borrell et al.	2013	At 1400 °C, hardness of
	[34]		conventionally sintered sample
			lower than that of microwave-
	Lietal [54]	2010	Sintered Sample.
	LI Et al. [34]	2019	sintered sample 13.4 ± 0.2 GPa
			rapid-sintered samples, 13.4 ± 0.2 of a,
			(R-1), 13.4 ± 0.3 (R-2), and 13.3
			± 0.1 GPa (R-3).
	Moratal	2021	Hardness of microwave (1300 °C)-
	et al. [19]		sintered sample higher than those of
			conventionally sintered and
			microwave (1200 $^\circ\text{C})\text{-sintered}$
			samples.
	Han et al.	2018	Hardness of rapid-sintered sample
	[55]		higher than that of conventionally
			sintered sample.
	Ribeiro et al.	2019	Hardness of microwave-sintered
	[57]		coprecipitated powder-made
			conventionally sintered sample
			No statistical differences between
			hardness values of conventionally
			sintered and microwave-sintered

pre-sintered zirconia samples.

3.4. Descriptive analysis

A descriptive analysis of the evaluated studies is presented in Table 5. The studies included in the descriptive analysis suggest that the translucency parameters (CIELab) and contrast ratios of most of the speed-sintered samples were similar to those of conventionally sintered samples. Kim et al. [58] reported that the light transmittance of microwave-sintered YSZ was higher than that of conventionally sintered YSZ. In contrast, Jansen et al. [41] demonstrated that the light transmittance of speed-sintered YSZ. Li et al. [54] reported that the light transmittance of a sample prepared using a 12-min rapid-sintering protocol was inferior to those prepared with conventional sintering, 25-min rapid-sintering, and 40-min rapid-sintering protocols.

Most of the studies included in the descriptive analysis indicated that the three-point or biaxial flexural strengths of the speed-sintered samples were comparable to those of conventionally sintered samples. Presenda and colleagues [59,60] and Borrell et al. [34] reported that the fracture toughness of microwave-sintered samples was comparable to or lower than that of conventionally sintered samples. In contrast, Ai et al. [61] and Moratal et al. [19] reported that microwave-sintered samples presented a higher fracture toughness than conventionally sintered samples. Almost all included studies revealed that speed-sintered samples had similar or higher hardness than the corresponding conventionally sintered samples, except for Ribeiro et al. [57], who reported that the hardness of microwave-sintered samples using a co-precipitated powder was lower than that of conventionally sintered samples.

4. Discussion

This systematic review and meta-analysis disclosed specific differences in the optical and mechanical properties between speed-sintered and conventionally sintered YSZ. The meta-analysis revealed that the investigated properties of speed-sintered YSZ are comparable to those of conventionally sintered YSZ, except for the CIELab-based translucency parameter. The descriptive analysis indicated that the hardness of speedsintered samples is similar or higher than that of conventionally sintered samples, except in one study [19], which showed that conventionally sintered samples had a lower hardness than microwave-sintered samples. Overall, the null hypothesis was partially accepted.

In this study, optical properties were compared for four parameters, namely, CIELab-based translucency, CIEDE2000-based translucency, contrast ratio, and light transmittance. These parameters are determined based on spectral transmittance and reflectance data in a certain wavelength range, as acquired using a spectrophotometer or colorimeter, as well as color coordinates, such as lightness (L*), red/green coordinates (a*), yellow/blue coordinates (b*), chroma (C*), and hue angle (H*) [62]. CIELab-based translucency ranges from 0 for a totally opaque material to 100 for a totally transparent material; that is, a higher CIELab-based translucency parameter indicates higher translucency. Alp et al. [63] demonstrated that CIEDE2000-based translucency is suitable for non-uniform optical materials, whereas CIELab-based translucency can only be applied to optically uniform materials. In addition, the color difference on black versus white backgrounds can be evaluated when an extremely translucent material is investigated using the CIEDE2000-based translucency parameter; instead, the CIELab-based translucency parameter would be 100 for a transparent material. The contrast ratio ranges from 0 for an absolutely transparent sample to 1 for a completely opaque sample [64]. Light transmittance always ranges from 0% (totally opaque) to 100% (totally transparent) [65].

Generally, the translucency of YSZ increases with an increase in sintering temperature and time [66-70], because larger grained microstructures with fewer grain boundaries and a higher volume fraction of c-ZrO₂ are formed. The birefringence effect, which occurs in non-symmetric crystal structured materials, will result in reflection and

refraction at the grain boundaries if adjacent grains have different crystallographic orientations. t-ZrO₂ has high birefringence, which reduces light transmittance, whereas c-ZrO₂ is isotropic, which reduces light scattering [71,72]. Therefore, zirconia ceramics with fewer grain boundaries cause less light reflection, and c-ZrO₂-containing ceramics transmit more light. As a consequence, fewer grain boundaries or larger grains and a high fraction of c-ZrO₂ are conducive to the translucency of YSZ. Recent studies, however, have shown that the translucency of YSZ sintered at 1450 °C is higher than that of YSZ sintered at 1350 and 1550 °C [73,74]. In contrast, the translucency of YSZ sintered at 1600 °C is similar to that of YSZ sintered at 1450 °C [66,67,72]. Theoretically, the optical properties of YSZ sintered at 1550 °C should be superior or at least not inferior to those of YSZ sintered at 1450 °C [75, 76]. Nevertheless, some studies have reported the opposite results [73, 74]. This might be due to microcrack formation during $t \rightarrow m$ phase transformations above a critical *t*-ZrO₂ grain size with higher sintering temperatures and longer sintering times, which could negatively affect light transmittance [73,74,77,78]. Inokoshi et al. [79] reported that a significant fraction of *m*-ZrO₂ was obtained when 3Y-TZP was sintered at higher temperatures for longer times, and the translucency of 3Y-TZP might be affected by the increase in m-ZrO₂ fraction. Meanwhile, the phase composition has a considerable impact on the translucency of zirconia [80]; Cho Too et al. [75] reported that zirconia with a higher fraction of c-ZrO₂ had higher translucency, and Pekkan et al. [71] suggested that m-ZrO2 itself could lower translucency because of the difference in refractive index between t-ZrO2 and c-ZrO2. Another hypothesis is that the amount of light scattering increases with an increase in grain size when the grain size is similar to the light wavelength. In contrast, the amount of light scattering decreases with an increase in grain size when the grain size is substantially larger than the light wavelength [58,81-83]. Moreover, some studies have reported that longer sintering times (up to 2 h) can lead to YSZ with better optical properties [70,73,74]. However, no difference was noticed between the optical properties of the YSZ samples when sintered longer than 2 h [70, 74]. Therefore, short-time sintered YSZ with optical properties similar to those of long-time sintered YSZ can be achieved by increasing the sintering temperature within reasonable limits. Although some studies have reported that the optical properties of some speed-sintered YSZ materials are similar to those of the corresponding conventionally sintered samples, contradictory results have also been reported [16,22,36, 42,52,53,84–87]. This indicates that speed-sintering should not be used to sinter commercial YSZ without testing or approval from the respective manufacturer.

Compared to other dental ceramics, YSZ has become popular in the field of dentistry because of its superior mechanical properties. Among the different types of YSZ, 3Y-TZP exhibits the highest flexural strength, typically ranging from 850 to 1500 MPa [66,75,88–90]; 4Y-PSZ tends to show similar flexural strength to 3Y-TZP, ranging from 900 to 1450 MPa [36,75,88–91]; 5Y-PSZ demonstrates significantly lower flexural strength at 650–1000 MPa [36,66,75,88–91]; and 6Y-PSZ has the lowest flexural strength at 500–700 MPa [36,90,91]. In this study, the mechanical properties of YSZ were characterized in terms of four different parameters, namely, three-point flexural strength, biaxial flexural strength (piston-on-three-ball method), fracture toughness, and hardness.

Stawarczyk et al. [69] reported that YSZ with a higher fracture strength (>1100 MPa) was obtained when sintered at 1400–1550 °C, whereas YSZ with a lower flexural strength (<1000 MPa) was obtained when sintered at \leq 1350 or \geq 1600 °C. Some other studies have partially agreed with this conclusion, while some comparisons of these studies have revealed that the flexural strength of YSZ sintered at 1350 or 1600 °C is comparable to that of those sintered at 1400–1550 °C [66, 70,72,92]. Theoretically, higher sintering temperatures result in larger grains with higher transformability/toughness and possibly fewer internal defects, which would enhance the mechanical properties of YSZ [93]. Nevertheless, increasing the grain size above a critical size causes a

Speed sintering protocols recently reported by manufacturers.

Manufacturer	Product	Recommended furnaces for speed sintering	Conventional sintering	Speed sintering	Limitations of speed sintering program	Reference
Ivoclar Vivadent, Schaan, Liechtenstein	IPS e-max ZirCAD prime; IPS e-max ZirCAD MT Muti; IPS e-max ZirCAD MT; IPS e-max ZirCAD LT; IPS e-max ZirCAD MO; IPS e-max ZirCAD prime Esthetic (only applies to third- party furnaces).	Programat S1 1600 (Except IPS e-max ZirCAD Prime Esthetic); Third-party furnaces.	440–590 min (Programat S1 1600); 550 min (third-party furnaces)	Programat S1 1600: 90–265 min (up to 3 units); 75–175 min (single crown); 165–270 min (14 units, For IPS e-max ZirCAD MO/LT) Third-party furnaces: ~290 min	For single units or 3-unit bridges (except IPS e- max ZirCAD MO/LT)	https://www.ivoclar. com/en_gb/products/a- products/ips-e.max- zircad-prime
Dentsply Sirona, Charlotte, NC, USA	Cercon ht; Cercon kt; Cercon Base; Cecron ht ML; Cecron xt ML.	Multimat2Sinter; heat DUO; Sirona HTC-speed sintering furnace; Third-party furnaces	~280 min (Multimat2Sinter, heat DUO, Sirona HTC-speed sintering furnace) 240–560 min + slow cooling to 200 °C (third-party furnaces)	Multimat2Sinter, heat DUO, Sirona HTC-speed sintering furnace: ~175 min (Cercon ht/ht ML/Base: up to 6 units; Cercon xt/xt ML: up to 3 units). Third-party furnaces: ~3 h. (Cercon ht/ht ML/ Base: up to 6 units; Cerconxt/xt ML: up to 3 units)	Up to 6 units for Cercon ht/ht ML/Base; Up to 3 units for Cercon xt/xt ML	https://www. dentsplysirona.com/en- us/categories/lab/ zirconia.html#zirconia- materials
Dentsply Sirona	inCoris Zi meso; inCoris TZI; inCoris TZI C; CEREC Zirconia; CEREC Zirconia Meso; CEREC Zirconia Plus	inFire HTC; inFire HTC Speed; CEREC SpeedFire	4–8 h	Speed: 1–3 h Superspeed: 10–15 min	3 single crowns or one 3- unit bridge.	https://www. dentsplysirona.com/en/ service-contact/ download-center.html; https://www. dentsplysirona.com/en- us/categories/cerec/ sinter-with-cerec.html; https://www. dentsplysirona.com/ja- jp/explore/cerec/sinter- with-cerec.html.
Dental Direkt, Spenge, Germany	DDBioZ; DDBioZX2; DDcubeONE; DDcubeX2.	Dekema Austromat 664; Dekema Austromat 674	564–660 min (T _{max} : 1450 °C)	\sim 2 h + rapid cooling (T_{max} : 1450 °C, Austromat 664); \sim 2.5 h + rapid cooling (T_{max} : 1450 °C, Austromat 674)	One plane, maximum 3 crowns per sintering process (Austromat 664); One plane, maximum 6 crowns per sintering process (Austromat 674) Only single crowns with maximum 4 mm wall thickness can be used for speed sintering programs.	https://www. dentaldirekt.de/en/ products/materials/ zirconium-dioxide
3 M Oral Care, St.Paul, MN, USA	3 M Chairside Zirconia	CEREC speedfire; Programat CS4.	~2.5 h (T _{max} : 1500 °C, Programat CS4 only)	20–40 min (<i>T</i> _{max} : 1565 °C for Programat CS4)	Only single crowns can be used for Programat CS4 furnace; 3 single crowns or 3-unit bridges can be used for CEREC speedfire.	https://www.3 m.com/ 3 M/en_US/p/d/ b5005087087/
Zirkonzahn, South Tyrol, Italy	ICE Translucent; Prettau 4 Anterior; Prettau 4 Anterior Dispersive; Prettau; Prettau 2; Prettau 2 Dispersive; Anatomic Coloured	Zirkonofen 600/V2; Zirkonofen 600/V3; Zirkonofen 700; Zirkonofen 700 Ultra-Vakuum; Zirkonofen Turbo;	8–12 h (1500–1600 °C)	Speed sintering [†] : ~4.5 h (1500 °C) Fast sintering [†] : ~1.5 h (1500–1600 °C) Ultra-fast sintering [§] : ~1 h (1500 °C)	[†] For ICE Translucent (single crowns and thin- walled bridges), Prettau 4 Anterior (single crowns only), Prettau 4 Anterior Dispersive (Single crowns only), Anatomic Coloured; [‡] For Prettau 4 Anterior, Prettau 4 Anterior Dispersive, Prettau, Prettau 2 and Prettau 2 Dispersive; [§] For Prettau 4 Anterior,	https://zirkonzahn.com/ us/download-section

(continued on next page)

Prettau 4 Anterior Dispersive. H. Liu et al.

Table 6 (continued)

Manufacturer	Product	Recommended furnaces for speed sintering	Conventional sintering	Speed sintering	Limitations of speed sintering program	Reference
Kuraray Noritake, Tokyo, Japan	KATANA HT/LT; KATANA ML; KATANA HTML; KATANA STML; KATANA STML; KATANA UTML; KATANA YML; KATANA Zirconia Block	Noritake KATANA F-2 N; CEREC speedfire†; Third-party furnaces	7 h (1550 °C)	Speed sintering 1: ~90 min (1515–1560 °C); Speed sintering 2: ~54 min (T _{max} : 1600 °C, excluding KATANA HT/ LT and KATANA Zirconia Block) Super speed sintering†: ~18 min	Up to 3 crowns or 3 units bridges. †For KATANA Zirconia Block only.	https://www. kuraraynoritake.eu/en/ labside/zirconia; https://www. kuraraynoritake.com/ world/product/ cad_materials/ katana_zirconia.html
Metoxit, Thayngen, Switzerland	Z-CAD HD; Z-CAD HTL; Z-CAD ONE4ALL; Z-CAD ONE4ALL MULTI; Z-CAD SMILE.	N/A	~400–700 min (T _{max} : 1450 °C)	~105 min (T _{max} : 1530 °C)	Single crowns or bridges up to 3 units.	https://metoxit.com/ dental-en/cad-cam-en/
Pritidenta, Leinfelden- Echterdingen, Germany	priti multidisc ZrO ₂ multicolor (Multi Translucent/High Translucent/Extra Translucent); priti multidisc ZrO ₂ monochrome (High Translucent/Extra Translucent/ Opaque); priti multibloc ZrO ₂ multicolor (High Translucent/Extra Translucent)	N/A	~7 h (<i>T</i> _{max} : 1450 °C)	Speed sintering 1^{\dagger} : ~4 h (T_{max} : 1500 °C, except priti multibloc ZrO ₂ multicolor Extra Translucent) Speed sintering 2^{\dagger} : ~1 h (only for priti multibloc ZrO ₂ multicolor Extra Translucent)	[†] Suitable for single crowns or bridges up to 3 units; [‡] Suitable for single crowns only.	https://pritidenta.com/ en/products/cadcam- materials/
Whitepeaks Dental Solutions, Hamminkeln, Germany	Copran Zri; CopraSupreme; CopraSmile; CopraSupreme Hyperion; CopraClassic HS	N/A	${\sim}275$ min (T _{max} : 1500 °C) + cooling time (unregulated in the closed furnace)	~70 min (1500 °C) + cooling time (unregulated in the closed furnace)	Speed program not recommended to achieve maximum translucency.	https://www.white- peaks-dental.com/en/ downloads/
Sagemax, Federal Way, WA, USA	NexxZr T Multi; NexxZr+ Multi	Programat* S1, Programat* S1 1600 Programat* S2	~9.6 h (<i>T</i> _{max} : 1500 °C)	~4.8 h (<i>T</i> _{max} : 1530 °C)	Up to 5 units	https://eifu.sagemax. com/en

spontaneous $t \rightarrow m$ transformation during cooling [94], and may enhance crack propagation due to internal stresses [95]. Additionally, at high sintering temperatures (for example, 1650 °C for 3Y-TZP), the grain size and amounts of *c*-ZrO₂ and *m*-ZrO₂ increase owing to partial $t \rightarrow c$ and $t \rightarrow m$ transformations [79]. Consequently, the flexural strength of the sintered YSZ may be lower than that sintered at lower temperatures (1400–1550 °C) because of the lower proportion of t-ZrO₂ [69]. Prolonged sintering times can also contribute to better mechanical properties [54,93]. Recent studies have not provided compelling evidence for a positive influence of speed-sintering on fracture strength, which indicates that the strength of YSZ may be affected by CAD/CAM fabrication [40,96]. However, the yttrium distribution is affected by the sintering temperature, which might be another effect causing a difference in the mechanical and optical properties of YSZ [97].

Recent studies have revealed that the fracture toughness of 3Y-TZP is approximately 3.7-4.5 MPa·m^{1/2}; that of 4Y-PSZ is approximately 3.5-4.0 MPa·m^{1/2}; that of 5Y-PSZ is lower at 2.4-3.2 MPa·m^{1/2}; and that of 6Y-PSZ is just 2.2 MPa·m^{1/2} [88–90]. Positive correlations have been reported between the fracture toughness and sintering temperature [79], sintering time [54,79], and cooling rate [88]. However, speed-sintering cannot guarantee a higher fracture toughness than conventional sintering, which indicates that a high sintering rate may have a strong negative influence on fracture toughness [30]. YSZ with lower hardness can occasionally be obtained at higher sintering temperatures [79], which suggests that a larger grain size decreases the hardness of YSZ [98]. In addition, the increased fraction of m-ZrO₂ on the outer surface of YSZ may affect the hardness of YSZ because the $t \rightarrow m$ transformation at the YSZ surface would produce compressive stresses, leading to higher hardness [70]. However, the $t \rightarrow m$ transformation could also reduce the hardness because of the presence of microcracks. Nevertheless, our meta-analysis did not find significant differences in the fracture toughness and hardness of conventionally sintered and speed-sintered YSZ.

Some studies have investigated "superspeed" sintering, which is characterized by an extremely short sintering time, i.e., a total sintering time of 10–30 min, with an initial heating rate of 100–400 °C/min [16, 22,36,37,53,54,59]. Most of these studies demonstrated that the optical properties of superspeed-sintered YSZ are comparable to those of conventionally sintered YSZ. Nevertheless, the mechanical properties of superspeed-sintered YSZ are different from those of conventionally sintered YSZ. Lawson and Maharishi [16] and Liu et al. [36] reported that the flexural strength of superspeed-sintered 5Y-PSZ was higher than that of conventionally sintered 5Y-PSZ due to the smaller grain size. Other studies revealed that the hardness and fracture toughness of superspeed-sintered YSZ were higher and lower than those of conventionally sintered YSZ, respectively [22,54,59]. Few studies have focused on the differences in the optical and mechanical properties of superspeed- and speed-sintered YSZ samples. Yilmaz Savaş and Akın [53] reported that the translucency parameter and contrast ratio of superspeed-sintered 3Y-TZP were similar to those of speed-sintered

3Y-TZP in a group of samples processed without dipping in coloring liquid. Liu et al. [36] demonstrated that the translucency parameter of superspeed-sintered Katana STML (5Y-PSZ; Kuraray Noritake Dental, Tokyo Japan) was comparable to that of speed-sintered Katana STML. However, the biaxial flexural strength of superspeed-sintered Katana STML was considerably higher than that of speed-sintered Katana STML. The current study showed that superspeed-sintered YSZ can be a viable material for dental treatment, which may facilitate one-day or chairside restorative treatment.

Recently, some manufacturers have reported speed-sintering protocols with total sintering times that are significantly shorter than those of their conventional sintering protocols (Table 6). However, most speed-sintering protocols have two main limitations. The first is that only a few furnaces can be used for speed-sintering. Some manufacturers have suggested the use of specific speed-sintering furnaces. Another limitation is that speed-sintering cannot be used for the fabrication of long bridge restorations. According to the manufacturers, speedsintering is only suitable for single crowns or bridges of up to three units. Although manufacturers have not explicitly stated the reasons for this, there are several possible reasons why speed-sintering programs may not be suitable for long-span restorations. For example, the limited chamber size of the designated furnaces and zirconia blocks used in speed-sintering may mean that they are not able to accommodate long bridges. Additionally, the rapid sintering and cooling processes inherent to speed-sintering could lead to increased shrinkage and cracking in long bridges compared to that in single crowns or shorter bridges. The presence of such cracks might adversely affect the overall quality and structural integrity of long-span restorations. Nevertheless, further research is crucial to fully comprehend the feasibility and limitations of using speed-sintering protocols with long-span dental restorations, and a comprehensive investigation is warranted to thoroughly examine how speed-sintering protocols affect long-span restorations.

Our study has some limitations since detailed data could not fully be obtained. Another notable concern is that a considerable proportion of the selected studies did not specify the method employed for sample size calculation. Consequently, it is challenging to ascertain whether the sample sizes in each study were adequate to draw reliable conclusions. Some studies had suboptimal study designs, which could have introduced bias into the overall findings of this review. In addition, only three databases were searched. Some relevant studies in other databases were unavailable. Some studies have proposed a manual search of the related studies on websites such as Google Scholar; however, herein, a manual search was not conducted. Our review suggests that speed-sintered YSZ could be a viable and acceptable option for restorative dental treatments. Once *in-vitro* experimental studies yield appropriate results for speed-sintered YSZ, clinical or *in-vivo* studies are needed to assess the clinical behavior of this material.

5. Conclusion

A systematic review and meta-analysis of the effects of speedsintering on the optical and mechanical properties of YSZ are reported. The meta-analysis revealed that speed-sintering does not considerably influence the CIEDE2000-based translucency, contrast ratio, three-point flexural strength, biaxial flexural strength, or fracture toughness of YSZ compared to those of conventionally sintered YSZ. However, the CIELab-based translucency of conventionally sintered YSZ is higher than that of speed-sintered YSZ. The descriptive analysis indicated that speed-sintering does not affect the hardness of YSZ compared to that of conventionally sintered YSZ. The abovementioned results suggest that speed-sintered YSZ could be a feasible material for crowns and short-span restorations. It is important to note that most of the papers selected for this study did not provide a clear explanation of their sample size determination method. Given this limitation, a cautious interpretation of the findings is warranted, along with more well-designed studies on the influence of speed-sintering on YSZ.

However, based on the available evidence, speed-sintered YSZ demonstrates potential as a choice for restorative dental procedures.

Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

Acknowledgments

The authors thank Prof. Yu Zhang and Dr. Stevan M. Cokic for providing the data for meta-analysis. This work was supported by JST SPRING, Grant Number JPMJSP2120, and the JSPS Grant-in-Aid for Scientific Research (C) JP22K10071.

References

- Anusavice KJ. Dental ceramics. In: Anusavice KJ, Shen C, Rawls HR, editors. Phillips' science of dental materials. twelfth ed..,. St. Louis: Elsevier/Saunders; 2013. p. 418–73.
- [2] Harada R, Takemoto S, Hattori M, Yoshinari M, Oda Y, Kawada E. The influence of colored zirconia on the optical properties of all-ceramic restorations. Dent Mater J 2015;34(6):918–24.
- [3] Jassim ZM, Majeed MA. Comparative evaluation of the fracture strength of monolithic crowns fabricated from different all-ceramic CAD/CAM materials (an. *Vitr* Study) Biomed Pharm J 2018;11(3):1689–97.
- [4] Zhang Y, Lawn BR. Novel zirconia materials in dentistry. J Dent Res 2018;97(2): 140–7.
- [5] Ozdogan A, Ozdemir H. The effects of repetitive firing processes on the optical, thermal, and phase formation changes of zirconia. J Adv Prosthodont 2020;12(1): 9–14.
- [6] Lümkemann N, Stawarczyk B. Impact of hydrothermal aging on the light transmittance and flexural strength of colored yttria-stabilized zirconia materials of different formulations. J Prosthet Dent 2021;125(3):518–26.
- [7] Denry I, Kelly JR. State of the art of zirconia for dental applications. Dent Mater 2008;24(3):299–307.
- [8] Li J, Zhang X-H, Cui B-C, Lin Y-H, Deng X-L, Li M, et al. Mechanical performance of polymer-infiltrated zirconia ceramics. J Dent 2017;58:60–6.
- [9] Nakamura T, Nakano Y, Usami H, Wakabayashi K, Ohnishi H, Sekino T, et al. Translucency and low-temperature degradation of silica-doped zirconia: A pilot study. Dent Mater J 2016;35(4):571–7.
- [10] Wille S, Zumstrull P, Kaidas V, Jessen LK, Kern M. Low temperature degradation of single layers of multilayered zirconia in comparison to conventional unshaded zirconia: Phase transformation and flexural strength. J Mech Behav Biomed Mater 2018;77:171–5.
- [11] Yang H, Xu Y-L, Hong G, Yu H. Effects of low-temperature degradation on the surface roughness of yttria-stabilized tetragonal zirconia polycrystal ceramics: A systematic review and meta-analysis. J Prosthet Dent 2021;125(2):222–30.
- [12] Abd El-Ghany OS, Sherief AH. Zirconia based ceramics, some clinical and biological aspects: Review. Futur Dent J 2016;2(2):55–64.
- [13] Jerman E, Wiedenmann F, Eichberger M, Reichert A, Stawarczyk B. Effect of highspeed sintering on the flexural strength of hydrothermal and thermo-mechanically aged zirconia materials. Dent Mater 2020;36(9):1144–50.
- [14] Wiedhahn K, Fritzsche G, Wiedhahn C, Schenk O. Zirconia crowns the new standard for single-visit dentistry? Int J Comput Dent 2016;19(1):9–26.
- [15] Blatz MB, Conejo J. The current state of chairside digital dentistry and materials. Dent Clin North Am 2019;63(2):175–97.
- [16] Lawson NC, Maharishi A. Strength and translucency of zirconia after high-speed sintering. J Esthet Restor Dent 2020;32(2):219–25.
- [17] Oghbaei M, Mirzaee O. Microwave versus conventional sintering: a review of fundamentals, advantages and applications. J Alloy Compd 2010;494(1–2): 175–89.
- [18] Luz JN, Kaizer MdR, Ramos NdC, Anami LC, Thompson VP, Saavedra GdSFA, et al. Novel speed sintered zirconia by microwave technology. Dent. Mater. 2021;37(5): 875–81.
- [19] Moratal S, Gil-Flores L, Salvador MD, Suarez M, Penaranda-Foix FL, Borrell A. Study of colored on the microwave sintering behavior of dental zirconia ceramics. J. Asian Ceram Soc 2021;9(1):188–96.
- [20] Sulaiman TA, Abdulmajeed AA, Donovan TE, Vallittu PK, Närhi TO, Lassila LV. The effect of staining and vacuum sintering on optical and mechanical properties of partially and fully stabilized monolithic zirconia. Dent Mater J 2015;34(5):605–10.
- [21] Almazdi AA, Khajah HM, Monaco Jr EA, Kim H. Applying microwave technology to sintering dental zirconia. J Prosthet Dent 2012;108(5):304–9.
- [22] Cokic SM, Vleugels J, Van Meerbeek B, Camargo B, Willems E, Li M, et al. Mechanical properties, aging stability and translucency of speed-sintered zirconia for chairside restorations. Dent Mater 2020;36(7):959–72.
- [23] Cologna M, Francis JSC, Raj R. Field assisted and flash sintering of alumina and its relationship to conductivity and MgO-doping. J Eur Ceram Soc 2011;31(15): 2827–37.

H. Liu et al.

- [24] Muccillo R, Muccillo ENS. An experimental setup for shrinkage evaluation during electric field-assisted flash sintering: application to yttria-stabilized zirconia. J Eur Ceram Soc 2013;33(3):515–20.
- [25] Todd RI, Zapata-Solvas E, Bonilla RS, Sneddon T, Wilshaw PR. Electrical characteristics of flash sintering: thermal runaway of Joule heating. J Eur Ceram Soc 2015;35(6):1865–77.
- [26] Raj R. Joule heating during flash-sintering. J Eur Ceram Soc 2012;32(10): 2293–301.
- [27] Galotta A, Sglavo VM. The cold sintering process: a review on processing features, densification mechanisms and perspectives. J Eur Ceram Soc 2021;41(16):1–17.
- [28] Guo H, Guo J, Baker A, Randall CA. Cold sintering process for ZrO₂-based ceramics: significantly enhanced densification evolution in yttria-doped ZrO₂. J Am Ceram Soc 2017;100(2):491–5.
- [29] Guo H, Bayer TJM, Guo J, Baker A, Randall CA. Cold sintering process for 8 mol% Y₂O₃-stabilized ZrO₂ ceramics. J Eur Ceram Soc 2017;37(5):2303–8.
- [30] Salamon D, Maca K, Shen Z. Rapid sintering of crack-free zirconia ceramics by pressure-less spark plasma sintering. Scr Mater 2012;66(11):899–902.
- [31] Fregeac A, Ansart F, Selezneff S, Estournès C. Relationship between mechanical properties and microstructure of yttria stabilized zirconia ceramics densified by spark plasma sintering. Ceram Int 2019;45(17B):23740–9.
- [32] Madeira S, Buciumeanu M, Carvalho O, Silva FS. Influence of sintering pressure on the microstructure and tribological properties of low temperature fast sintered hotpressed Y-TZP. Ceram Int 2019;45(5):5883–93.
- [33] Ferrage L, Bertrand G, Lenormand P. Dense yttria-stabilized zirconia obtained by direct selective laser sintering. Addit Manufact 2018;21:472–8.
- [34] Borrell A, Salvador MD, Peñaranda-Foix FL, Cátala-Civera JM. Microwave sintering of zirconia materials: Mechanical and microstructural properties. Int J Appl Ceram Technol 2013;10(2):313–20.
- [35] (Ćurković) Veseli L, Gabelica R, Žmak I, Ropuš I, Vukšić I. A review of microwaveassisted sintering technique. Trans FAMENA 2021;45(1):1–16.
- [36] Liu H, Inokoshi M, Nozaki K, Shimizubata M, Nakai H, Cho Too TD, et al. Influence of high-speed sintering protocols on translucency, mechanical properties, microstructure, crystallography, and low-temperature degradation of highly translucent zirconia. Dent Mater 2022;38(2):451–68.
- [37] Kaizer MR, Gierthmuehlen PC, dos Santos MBF, Cava SS, Zhang Y. Speed sintering translucent zirconia for chairside one-visit dental restorations: optical, mechanical, and wear characteristics. Ceram Int 2017;43(14):10999–1005.
- [38] Kauling AE, Güth J-F, Erdelt K, Edelhoff D, Keul C. Influence of speed sintering on the fit and fracture strength of 3-unit monolithic zirconia fixed partial dentures. J Prosthet Dent 2020;124(3):380–6.
- [39] Marinis A, Aquilino SA, Lund PS, Gratton DG, Stanford CM, Diaz-Arnold AM, et al. Fracture toughness of yttria-stabilized zirconia sintered in conventional and microwave ovens. J Prosthet Dent 2013;109(3):165–71.
- [40] Zimmermann M, Ender A, Mehl A. Influence of CAD/CAM fabrication and sintering procedures on the fracture load of full-contour monolithic zirconia crowns as a function of material thickness. Oper Dent 2020;45(2):219–26.
- [41] Jansen JU, Lümkemann N, Letz I, Pfefferle R, Sener B, Stawarczyk B. Impact of high-speed sintering on translucency, phase content, grain sizes, and flexural strength of 3Y-TZP and 4Y-TZP zirconia materials. J Prosthet Dent 2019;122(4): 396–403.
- [42] Kim H-K, Kim S-H. Comparison of the optical properties of pre-colored dental monolithic zirconia ceramics sintered in a conventional furnace versus a microwave oven. J Adv Prosthodont 2017;9(5):394–401.
- [43] Nakamura T, Nakano Y, Usami H, Okamura S, Wakabayashi K, Yatani H. In vitro investigation of fracture load and aging resistance of high-speed sintered monolithic tooth-borne zirconia crowns. J Prosthodont Res 2020;64(2):182–7.
- [44] Khaledi AAR, Vojdani M, Farzin M, Pirouzi S. The effect of sintering program on the compressive strength of zirconia copings. J Dent 2018;19(3):206–11.
 [45] Page MJ, McKenzie JE, Bossuyt PM, Boutron I, Hoffmann TC, Mulrow CD, et al.
- [45] Page MJ, McKenzie JE, Bossuyt PM, Boutron I, Hoffmann TC, Mulrow CD, et al. The PRISMA 2020 statement: an updated guideline for reporting systematic reviews. BMJ 2021:372:n71.
- [46] Higgins JPT, Li T, Deeks JJ. Choosing effect measures and computing estimates of effect. In: Higgins JPT, Thomas J, Chandler J, Cumpston M, Li T, Page MJ, et al., editors. Cochrane handbook for systematic reviews of interventions. 2nd ed..,. Chichester (UK): John Wiley & Sons; 2019. p. 143–76.
- [47] Faggion Jr CM. Guidelines for reporting pre-clinical in vitro studies on dental materials. J Evid Based Dent Pr 2012;12(4):182–9.
- [48] Pereira GKR, Venturini AB, Silvestri T, Dapieve KS, Montagner AF, Soares FZM, et al. Low-temperature degradation of Y-TZP ceramics: a systematic review and meta-analysis. J Mech Behav Biomed Mater 2016;55:151–63.
- [49] Zhang C-Y, Agingu C, Tsoi JKH, Yu H. Effects of aging on the color and translucency of monolithic translucent Y-TZP ceramics: a systematic review and meta-analysis of in vitro studies. BioMed Res Int 2021;2021:8875023.
- [50] Zhang L-X, Hong D-W, Zheng M, Yu H. Is the bond strength of zirconia-reinforced lithium silicate lower than that of lithium disilicate? - a systematic review and meta-analysis. J Prosthodont Res 2022;66(4):530–7.
- [51] Solá-Ruíz MF, Rico-Coderch A, Montiel-Company JM, Fons-Badal C, Verdejo-Solá B, Agustín-Panadero R. Influence of the chemical composition of monolithic zirconia on its optical and mechanical properties. systematic review and metaregression. J Prosthodont Res 2022;66(2):193–207.
- [52] Jiang Y, Yang Y, Zhan W, Hu G, Yang Q. Translucency of dental zirconia ceramics sintered in conventional and microwave ovens (in Chinese with English abstract). Hua Xi Kou Qiang Yi Xue Za Zhi 2015;33(6):642–5.
- [53] Yılmaz Savaş T, Akın C. Effects of sintering protocol and dipping time on the optical properties of monolithic zirconia. J Prosthet Dent 2022;127(5). 801.E1-801.E8.

- [54] Li L, Zhao C, Du Z, Qiu Y, Si W. Rapid-sintered dental zirconia for chair-side onevisit application. Int J Appl Ceram Technol 2019;16(5):1830–5.
- [55] Han Y, Zheng D, Wang Q, Si W. Effect of chairside rapid sintering process on microstructure and mechanical properties of TZP zirconia ceramics. J Biomater Tissue Eng 2018;8(1):95–102.
- [56] Presenda Á, Salvador MD, Penaranda-Foix FL, Catalá-Civera JM, Pallone E, Ferreira J, et al. Effects of microwave sintering in aging resistance of zirconia-based ceramics. Chem Eng Process 2017;122:404–12.
- [57] Ribeiro ASL, Arata A, de Lima NB, Ussui V, Lazar DRR. Comparison of a laboratorial scale synthesized and a commercial yttria-tetragonal zirconia polycrystals ceramics submitted to microwave sintering. Int J Appl Ceram Technol 2019;16(5):2020–7.
- [58] Kim M-J, Ahn J-S, Kim J-H, Kim H-Y, Kim W-C. Effects of the sintering conditions of dental zirconia ceramics on the grain size and translucency. J Adv Prosthodont 2013;5(2):161–6.
- [59] Presenda Á, Salvador MD, Vleugels J, Moreno R, Borrell A. Fretting fatigue wear behavior of Y-TZP dental ceramics processed by non-conventional microwave sintering. J Am Ceram Soc 2017;100(5):1842–52.
- [60] Presenda Á, Salvador MD, Peñaranda-Foix FL, Moreno R, Borrell A. Effect of microwave sintering on microstructure and mechanical properties in Y-TZP materials used for dental applications. Ceram Int 2015;41(5B):7125–32.
- [61] Ai Y, Xie X, He W, Liang B, Fan Y. Microstructure and properties of Al₂O_{3(n)}/ZrO₂ dental ceramics prepared by two-step microwave sintering. Mater Des 2015;65: 1021–7.
- [62] Shiraishi T, Wood DJ, Shinozaki N, van Noort R. Optical properties of base dentin ceramics for all-ceramic restorations. Dent Mater 2011;27(2):165–72.
- [63] Alp G, Subaşi MG, Seghi RR, Johnston WM, Yilmaz B. Effect of shading technique and thickness on color stability and translucency of new generation translucent zirconia. J Dent 2018;73:19–23.
- [64] Mosquim V, Ferrairo BM, Vertuan M, Magdalena AG, Fortulan CA, Lisboa-Filho PN, et al. Structural, chemical and optical characterizations of an experimental SiO₂–Y-TZP ceramic produced by the uniaxial/isostatic pressing technique. J Mech Behav Biomed Mater 2020;106:103749.
- [65] Ueda K, Güth J-F, Erdelt K, Stimmelmayr M, Kappert H, Beuer F. Light transmittance by a multi-coloured zirconia material. Dent Mater J 2015;34(3): 310–4.
- [66] Sen N, Sermet IB, Cinar S. Effect of coloring and sintering on the translucency and biaxial strength of monolithic zirconia. J Prosthet Dent 2018;119(2). 308.E1-308. E7.
- [67] Sanal FA, Kilinc H. Effect of shade and sintering temperature on the translucency parameter of a novel multi-layered monolithic zirconia in different thicknesses. J Esthet Restor Dent 2020;32(6):607–14.
- [68] Jiang L, Liao Y, Wan Q, Li W. Effects of sintering temperature and particle size on the translucency of zirconium dioxide dental ceramic. J Mater Sci Mater Med 2011; 22(11):2429–35.
- [69] Stawarczyk B, Özcan M, Hallmann L, Ender A, Mehl A, Hämmerlet CHF. The effect of zirconia sintering temperature on flexural strength, grain size, and contrast ratio. Clin Oral Invest 2013:17(1):269–74.
- [70] Ebeid K, Wille S, Hamdy A, Salah T, El-Etreby A, Kern M. Effect of changes in sintering parameters on monolithic translucent zirconia. Dent Mater 2014;30(12): e419–24.
- [71] Pekkan G, Pekkan K, Bayindir BC, Özcan M, Karasu B. Factors affecting the translucency of monolithic zirconia ceramics: a review from materials science perspective. Dent Mater J 2020;39(1):1–8.
- [72] Cardoso KV, Adabo GL, Mariscal-Muñoz E, Antonio SG, Filho JNR. Effect of sintering temperature on microstructure, flexural strength, and optical properties of a fully stabilized monolithic zirconia. J Prosthet Dent 2020;124(5):594–8.
- [73] Attachoo S, Juntavee N. Role of sintered temperature and sintering time on spectral translucence of nano-crystal monolithic zirconia. J Clin Exp Dent 2019;11(2): e146–53.
- [74] Juntavee N, Attashu S. Effect of sintering process on color parameters of nano-sized yttria partially stabilized tetragonal monolithic zirconia. J Clin Exp Dent 2018;10 (8):e794–804.
- [75] Cho Too TD, Inokoshi M, Nozaki K, Shimizubata M, Nakai H, Liu H, et al. Influence of sintering conditions on translucency, biaxial flexural strength, microstructure, and low-temperature degradation of highly translucent dental zirconia. Dent Mater J 2021;40(6):1320–8.
- [76] Kanpalta B, Burduroğlu D, Kara Ö. Effect of artificial aging on the translucency of monolithic zirconia materials sintered at different temperatures. J Prosthet Dent 2022;128(1):91. E1-91.E6.
- [77] Nogueira AD, Della, Bona A. The effect of a coupling medium on color and translucency of CAD–CAM ceramics. J Dent 2013;41(Suppl 3):e18–23.
- [78] Della Bona A, Nogueira AD, Pecho OE. Optical properties of CAD–CAM ceramic systems. J Dent 2014;42(9):1202–9.
- [79] Inokoshi M, Zhang F, De Munck J, Minakuchi S, Naert I, Vleugels J, et al. Influence of sintering conditions on low-temperature degradation of dental zirconia. Dent Mater 2014;30(6):669–78.
- [80] Zhang F, Vanmeensel K, Batuk M, Hadermann J, Inokoshi M, Van Meerbeek B, et al. Highly-translucent, strong and aging-resistant 3Y-TZP ceramics for dental restoration by grain boundary segregation. Acta Biomater 2015;16:215–22.
- [81] Lee Y-K. Influence of scattering/absorption characteristics on the color of resin composites. Dent Mater 2007;23(1):124–31.
- [82] Casolco SR, Xu J, Garay JE. Transparent/translucent polycrystalline nanostructured yttria stabilized zirconia with varying colors. Scr Mater 2008;58 (6):516–9.

- [83] O'Brien WJ, Johnston WM, Fanian F. Double-layer color effects in porcelain systems. J Dent Res 1985;64(6):940–3.
- [84] Pekkan K. Effect of sintering regimes and thickness on optical properties of zirconia ceramics for dental applications. Int J Appl Ceram Technol 2021;18(4):1354–64.
- [85] Al-Zordk W, Saker S. Impact of sintering procedure and clinical adjustment on color stability and translucency of translucent zirconia. J Prosthet Dent 2020;124 (6):788. E1-788.E9.
- [86] Yang C-C, Ding S-J, Lin T-H, Yan M. Mechanical and optical properties evaluation of rapid sintered dental zirconia. Ceram Int 2020;46(17):26668–74.
- [87] Liu Y-C, Lin T-H, Lin Y-Y, Hu S-W, Liu J-F, Yang C-C, et al. Optical properties evaluation of rapid sintered translucent zirconia with two dental colorimeters. J Dent Sci 2022;17(1):155–61.
- [88] Kim H-K. Effect of a rapid-cooling protocol on the optical and mechanical properties of dental monolithic zirconia containing 3–5 mol% Y₂O₃. Materials 2020;13(8):1923.
- [89] Jerman E, Lümkemann N, Eichberger M, Zoller C, Nothelfer S, Kienle A, et al. Evaluation of translucency, Marten's hardness, biaxial flexural strength and fracture toughness of 3Y-TZP, 4Y-TZP and 5Y-TZP materials. Dent Mater 2021;37 (2):212–22.
- [90] Čokić SM, Cóndor M, Vleugels J, Van Meerbeek B, Van Oosterwyck H, Inokoshi M, et al. Mechanical properties-translucency-microstructure relationships in commercial monolayer and multilayer monolithic zirconia ceramics. Dent Mater 2022;38(5):797–810.

- [91] Inokoshi M, Shimizubata M, Nozaki K, Takagaki T, Yoshihara K, Minakuchi S, et al. Impact of sandblasting on the flexural strength of highly translucent zirconia. J Mech Behav Biomed Mater 2020;115:104268.
- [92] Öztürk C, Can G. Effect of sintering parameters on the mechanical properties of monolithic zirconia. J Dent Res Dent Clin Dent Prospects 2019;13(4):247–52.
- [93] Juntavee N, Attashu S. Effect of different sintering process on flexural strength of translucency monolithic zirconia. J Clin Exp Dent 2018;10(8):e821–30.
- [94] Bravo-Leon A, Morikawa Y, Kawahara M, Mayo MJ. Fracture toughness of nanocrystalline tetragonal zirconia with low yttria content. Acta Mater 2002;50 (18):4555–62.
- [95] Chevalier J, Olagnon C, Fantozzi G. Subcritical crack propagation in 3Y-TZP ceramics: static and cyclic fatigue. J Am Ceram Soc 1999;82(11):3129–38.
- [96] Fraga S, Amaral M, Bottino MA, Valandro LF, Kleverlaan CJ, May LG. Impact of machining on the flexural fatigue strength of glass and polycrystalline CAD/CAM ceramics. Dent Mater 2017;33(11):1286–97.
- [97] Matsui K, Horikoshi H, Ohmichi N, Ohgai M, Yoshida H, Ikuhara Y. Cubicformation and grain-growth mechanisms in tetragonal zirconia polycrystal. J Am Ceram Soc 2003;86(8):1401–8.
- [98] Trunec M. Effect of grain size on mechanical properties of 3Y-TZP ceramics. Ceram-Silik 2008;52(3):165–71.
- [99] Presenda Á, Salvador MD, Moreno R, Borrell A. Hydrothermal degradation behavior of Y-TZP ceramics sintered by nonconventional microwave technology. J Am Ceram Soc 2015;98(12):3680–9.