

Experimental Study of Compatibility between the Eco-friendly Insulation Mixed Gas $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ and EPDM and CR Materials

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Cite This: *ACS Omega* 2024, 9, 7958–7966

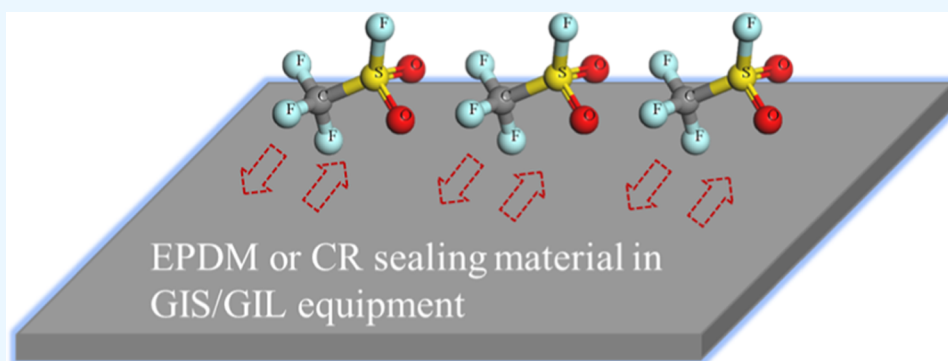


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ABSTRACT: As a greenhouse gas with strong global warming potential, the use of SF_6 needs to be reduced as much as possible. Researching environmentally friendly insulation (EFI) gases to replace SF_6 in power electrical equipment is an effective way to reduce its usage. $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$, as a newly proposed EFI gas, has certain potential to replace SF_6 . Compatibility of $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ gas with rubber sealing materials commonly used in electrical equipment is still unknown. In this article, the compatibility of $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ with the ethylene-propylene-diene monomer (EPDM) and chloroprene rubber (CR) was investigated experimentally. It was found that $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ would slightly decompose under the influence of EPDM and CR rubber under certain conditions. The surface morphology of EPDM changed slightly under the influence of $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$, and it was similar to the influence of SF_6 . While the surface morphology of CR deteriorated significantly with obvious defects. The mechanical properties of EPDM were not significantly affected by $\text{CF}_3\text{SO}_2\text{F}$, which is similar to the influence of SF_6 . But CR was affected greatly by $\text{CF}_3\text{SO}_2\text{F}$ gas. Permanent deformation compression and surface morphology are two effective indicators for characterizing the compatibility between gas and rubber sealing materials. This research provides a reference for the application of $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ as a new EFI gas in power equipment.

INTRODUCTION

SF_6 gas is widely used in gas-insulated electrical equipment (GIE) because of its excellent insulating properties and arc extinguishing ability.^{1,2} However, SF_6 is also a greenhouse gas with the strongest greenhouse effect. According to IPCC's Sixth Assessment Report Climate Change, SF_6 has an atmospheric lifetime of 1000 years and a global warming potential (GWP) of 24,300.³ This means that every 1 kg of SF_6 emitted is equivalent to 24.3 tons of CO_2 emitted. Therefore, the development of environment-friendly insulation (EFI) gases to replace SF_6 has become an important and hot research issue in the power electrical industry.^{4,5}

$\text{CF}_3\text{SO}_2\text{F}$ is a newly proposed EFI gas with huge potential to replace SF_6 . According to existing studies, its dielectric strength is about 40% higher than that of SF_6 , and its GWP is 86% lower than SF_6 . Moreover, the liquefaction temperature of $\text{CF}_3\text{SO}_2\text{F}$ at 0.1 MPa is about $-23\text{ }^\circ\text{C}$, much lower than existing EFI gases such as $\text{C}_4\text{F}_7\text{N}$, $\text{C}_3\text{F}_{10}\text{O}$, etc.^{6,7} When it is

mixed with N_2 or CO_2 at higher pressures, the synergistic effect of the $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ mixture is better than that of $\text{CF}_3\text{SO}_2\text{F}/\text{CO}_2$, indicating that the $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ mixture is more suitable for high-voltage GIE.⁸ It is previously found that the 40% $\text{CF}_3\text{SO}_2\text{F}/60\% \text{N}_2$ gas mixture can achieve equal insulation strength with SF_6 .⁹

However, in addition to the insulating properties of $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ gas, it is also necessary to consider the material compatibility between gases and solid materials when considering the use of $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ gas.^{10–12} As sealing

Received: October 11, 2023

Revised: December 10, 2023

Accepted: January 25, 2024

Published: February 6, 2024



materials in the GIE, current rubber materials should be compatible with $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ gas. Otherwise, it may cause gas leakage and cause insulation breakdown accidents. Therefore, investigation is needed to investigate the compatibility between $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ and rubber materials commonly used in GIE.

In recent years, researchers at home and abroad have carried out in-depth studies of the compatibility between various EFI gases and rubber materials. Zheng et al. found that the chemical reaction between $\text{C}_4\text{F}_7\text{N}$ gas and ethylene-propylene-diene monomer (EPDM) occurs under the thermal acceleration test.¹³ Cheng et al. found that $\text{C}_3\text{F}_{10}\text{O}$ decomposes into gases such as $\text{C}_3\text{F}_6\text{O}$, C_3F_6 , and C_3HF_7 after the thermal acceleration test, and the rubber itself is also embrittled.¹⁴ Lan et al. investigated the compatibility between the $\text{C}_6\text{F}_{12}\text{O}/\text{N}_2$ gas mixture and nitrile rubber and found that the two undergo a strong chemical reaction at high temperatures.¹⁵ Wu et al. investigated the compatibility between the $\text{C}_4\text{F}_7\text{N}/\text{CO}_2/\text{O}_2$ ternary gas mixture and EPDM and found that O_2 exacerbates the reaction between $\text{C}_4\text{F}_7\text{N}$ and EPDM.¹⁶ Wang et al. studied the compatibility of $\text{C}_4\text{F}_7\text{N}/\text{CO}_2$ and its decomposition gases with rubber materials and found that nitrile rubber is less compatible with $\text{C}_4\text{F}_7\text{N}/\text{CO}_2$.¹⁷ From the above research, it can be seen that the study of compatibility plays an important role in the practical application of EFI gases; therefore, it is necessary to carry out research on compatibility between $\text{CF}_3\text{SO}_2\text{F}$ gas and rubber sealing materials.

The compatibility between $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ gas and two kinds of rubber, EPDM, and chloroprene rubber (CR), which are commonly used in GIE, is studied by a thermal acceleration experiment. The compatibility between those two is evaluated by the decomposition of $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ gas under the influence of rubber and the performance deterioration of rubber under the influence of $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ gas. Experimental results could provide compatibility perspective evidence for evaluating the feasibility of $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ gas as one of the SF_6 alternatives.

EXPERIMENTAL METHOD

Preparation of Samples. Due to the problem of liquefaction of pure $\text{CF}_3\text{SO}_2\text{F}$ gas at high pressures, a mixture of $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ was selected for the test. The mole ratio of $\text{CF}_3\text{SO}_2\text{F}$ at 40% was used, which can make the insulation strength of the $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ mixture equivalent to that of SF_6 . A control group using SF_6 gas was also studied. The $\text{CF}_3\text{SO}_2\text{F}$ gas used in this study was supplied by Beijing Yuji Co. with a purity of 99%. N_2 and SF_6 gases were conventional commercial supplies with a purity higher than 99.99%.

The rubber materials used in this study were EPDM and CR. They are commonly used sealing materials in high-voltage electrical equipment.¹⁸ The rubber materials were provided by Henan Pinggao Electric Co. The rubber materials were prepared into corresponding tensile and compression samples for testing according to ISO 815-1 and ISO 37.^{19,20} The dimensional drawings of the two specimens are shown in Figure 1.

Test Method. As a long-term indicator, compatibility is usually studied using the thermal acceleration test.²¹ In this test, the chemical reaction rate is accelerated by increasing the temperature, thereby characterizing the aging condition of material under decades of service within an acceptable test time range. The selection of temperature is crucial in the thermal acceleration test. It is necessary to increase the temperature to simulate a sufficiently long operating time, but the temperature should not be too high to cause material

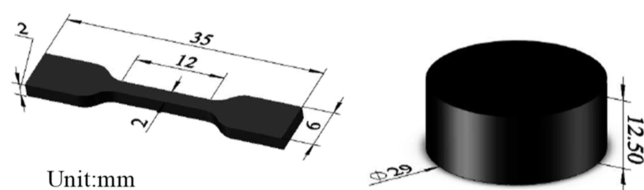


Figure 1. Dimensions of the rubber sample.

cracking. In this study, a gas–solid compatibility test platform was set up to carry out thermal acceleration tests using a metal-sealed container and a high-temperature test chamber. The composition and proportion of $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ mixed gas after the test were detected by gas chromatography–mass spectrometry (GC–MS). The surface morphology of the rubber and the surface element after the test were detected by field emission scanning electron microscopy and energy-dispersive spectrometry. The mechanical properties of rubber after testing were detected by a universal testing machine. The compatibility of $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ gas with EPDM and CR was determined by the above tests, as shown in Figure 2.

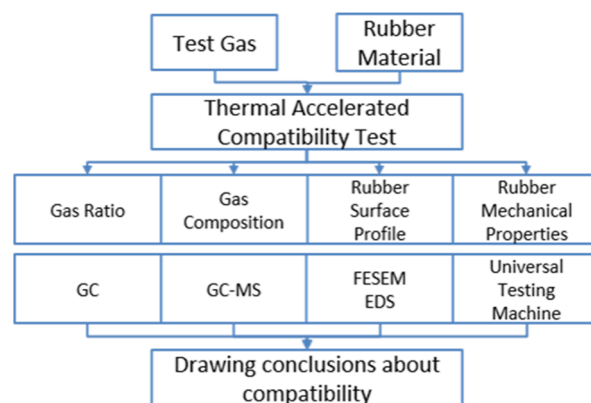


Figure 2. Compatibility test flowchart.

Test Conditions. To set a suitable thermal acceleration test temperature, we mainly referred to the ISO 23529:2004 and IEC 62271-1:2017.^{22,23} For the selection of temperature gradient for comparison, the test temperature was selected at three temperature levels, namely 70, 85, and 100 °C.

To set the suitable test pressure for $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ mixed gas, we referred to IEC 62271-200:2011. The gas pressure inside the compartment should be determined according to the manufacturer and the user.²⁴ Since $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ has not yet been put into practical industrial applications, we referred to the current use of SF_6 gas-insulated equipment. The test pressure of $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ was set at an absolute pressure of 0.4 MPa.⁵ For the control group, SF_6 gas with an absolute pressure of 0.4 MPa was used. The test duration was selected as 7 days with reference to ISO 23529:2004. The compatibility between $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ gas and two types of rubber was evaluated by detecting the decomposition of the gas under the influence of rubber and the chemical reaction of rubber under the influence of gas before and after the test.

RESULTS AND DISCUSSION

Gas Composition Changing Law. The gas–solid compatibility tests were performed using the test methods. After tests, the composition of $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ gas mixture was

detected by GC–MS. The chromatographic column used was GS-GASPRO 60 m, and the inlet temperature was 100 °C. The total flow rate was 67.9 mL/min. The column flow rate was 3.09 mL/min. The linear velocity was 44.7 cm/s. The purging flow rate was 3 mL/min. The split ratio was 20:1. The heating program was controlled by an initial temperature of 35 °C, held for 5 min, then increased to 150 °C at a rate of 8 °C/min, and held for 10 min. The MS conditions were set at 200 °C for the ion source and 200 °C for the interface.

The results showed that the decomposition gas was generated at 70, 85, and 100 °C. The gas peaks of the decomposed gases were more obvious after the temperature was increased. Taking the GC–MS detection results of the gas at 100 °C as an example, Figures 3 and 4 show the GC–MS

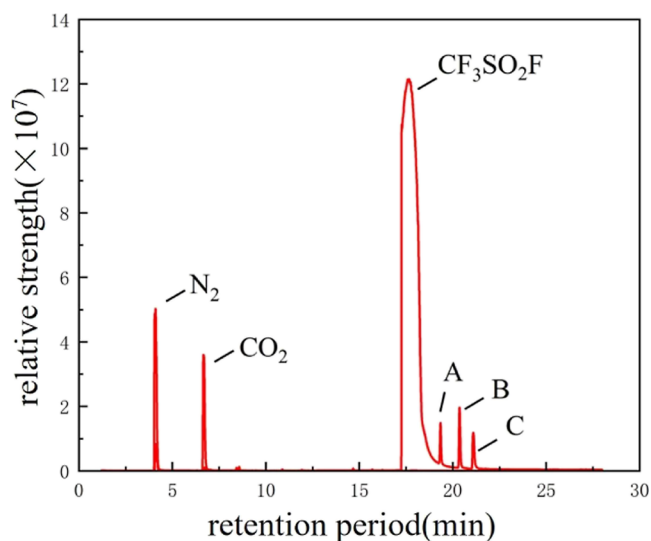


Figure 3. GC–MS results of EPDM and $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ at 100 °C.

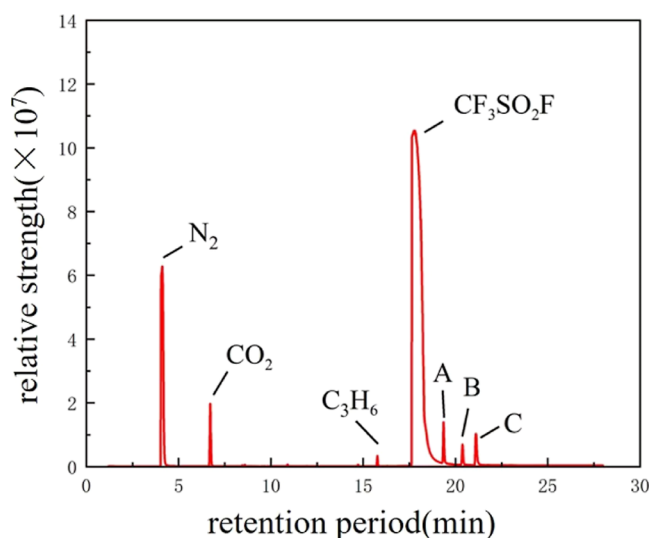


Figure 4. GC–MS results of CR and $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ at 100 °C.

detection results of the gas after the test of EPDM and CR at 100 °C, respectively, which show that EPDM and CR will lead to the generation of decomposition gases after the test with $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$. The gases A–C in Figures 2 and 3 are $\text{C}_2\text{F}_6\text{O}_5\text{S}_2$, $\text{C}_2\text{H}_3\text{ClF}_2$, and CF_4 , respectively. $\text{C}_2\text{F}_6\text{O}_5\text{S}_2$ may be formed by the combination of two $\text{CF}_3\text{SO}_2\text{F}$ molecules after

breakage of the S–F bond, and $\text{C}_2\text{H}_3\text{ClF}_2$ is formed after reacting with rubber. CF_4 may be formed due to the removal of the CF_3 group from $\text{CF}_3\text{SO}_2\text{F}$.

Comparing the results in Figures 5 and 6, it can be seen that no decomposition gases were detected after the compatibility

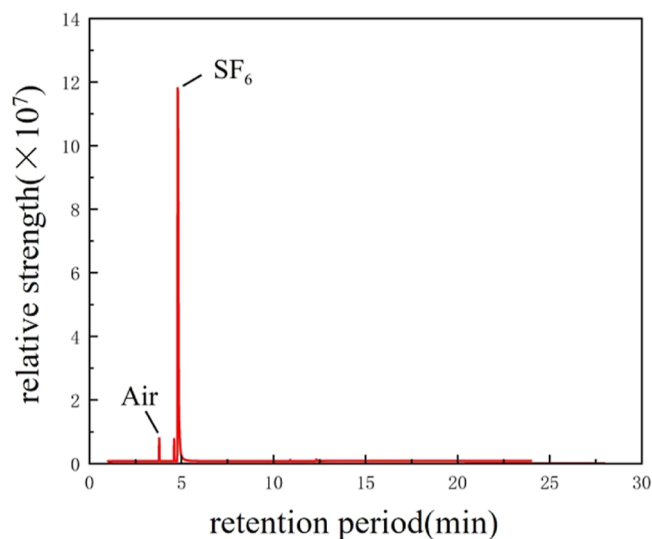


Figure 5. GC–MS test results of EPDM and SF_6 at 100 °C.

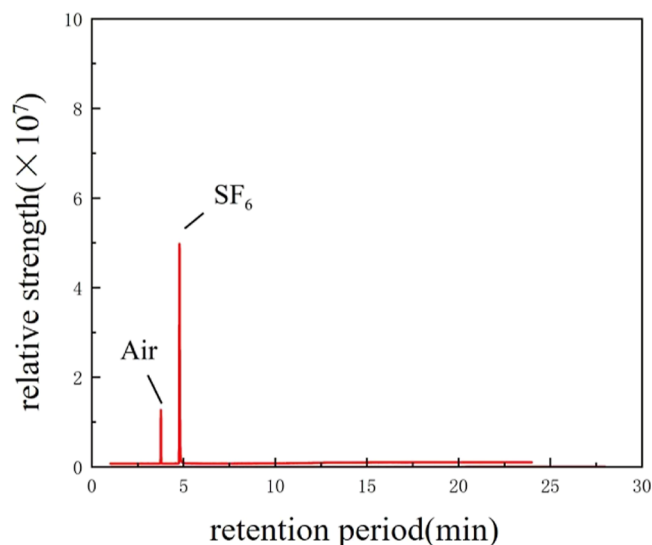


Figure 6. GC–MS test results of CR and SF_6 at 100 °C.

tests of SF_6 with EPDM and CR, which suggests that the rubber itself does not form decomposition gases under the same test conditions, and that the decomposition gases are due to the reaction between EPDM, CR, and $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ at high temperatures. This indicates that there is a compatibility problem between $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ and EPDM/CR. Tests at high temperatures show that decomposition gases formed between $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ and rubber, which led to a decrease in the gas ratio of $\text{CF}_3\text{SO}_2\text{F}$ and resulted in a decrease in its insulating ability.

Gas Ratio Changing Law. After the tests, the gas proportion of $\text{CF}_3\text{SO}_2\text{F}$ was detected by GC, and the test results are shown in Table 1. It can be seen that for the effect of EPDM on the proportion of $\text{CF}_3\text{SO}_2\text{F}$, it decreased by 1.55% at 70 °C, 2.06% at 85 °C, and 5.61% at 100 °C. For the

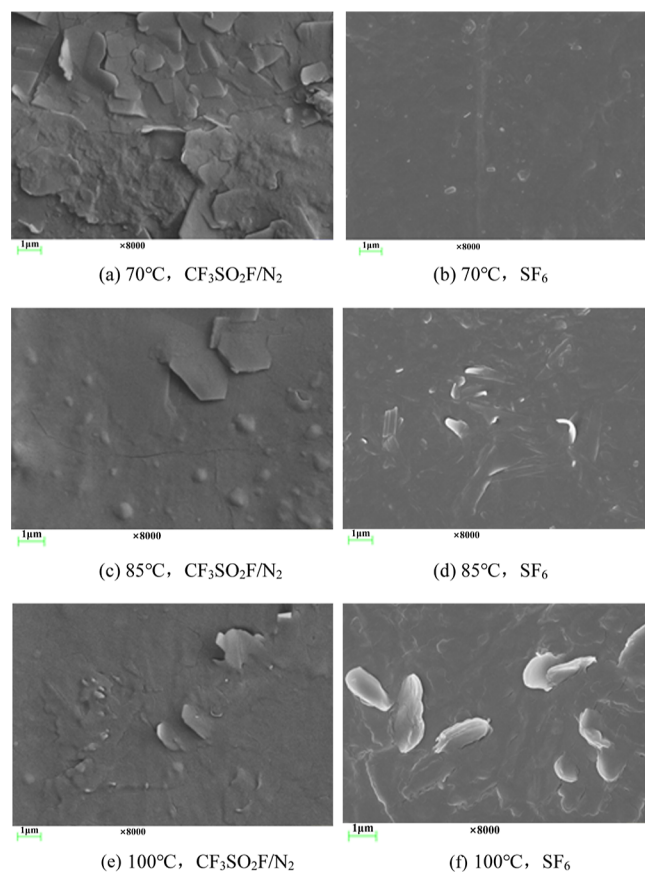
Table 1. Mole Ratio of $\text{CF}_3\text{SO}_2\text{F}$ before and after the Compatibility Test

test group	test temperature/ $^{\circ}\text{C}$	ratio before test/%	ratio after test/%
EPDM	70	39.38	37.83
	85	40.72	38.66
	100	40.69	35.08
CR	70	40.75	37.34
	85	40.06	36.94
	100	40.11	35.13

effect of CR on the proportion of $\text{CF}_3\text{SO}_2\text{F}$, it also decreased, with 3.41% at 70 $^{\circ}\text{C}$, 3.12% at 85 $^{\circ}\text{C}$, and 4.98% at 100 $^{\circ}\text{C}$. The results show that the proportions of $\text{CF}_3\text{SO}_2\text{F}$ after the compatibility test with CR are approximately the same as those of EPDM. This indicates that the effect of EPDM and CR on $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ is nearly the same.

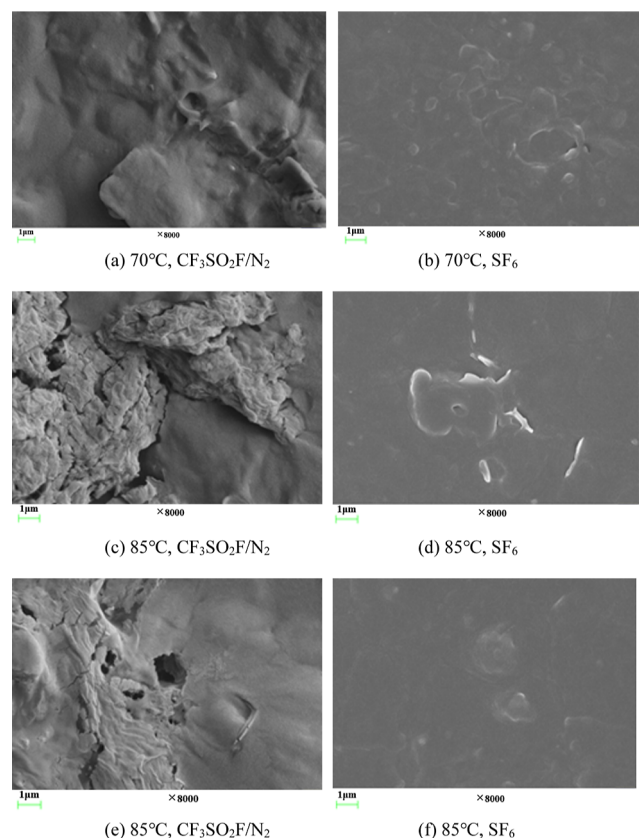
It also shows that the effect of both EPDM and CR on the proportion decrease of $\text{CF}_3\text{SO}_2\text{F}$ becomes more obvious with the increase of the test temperature. The thermal acceleration test speeds up the reaction rate by increasing the temperature, thus reflecting in a short time what happens over a longer period of time. Therefore, the test results show that both EPDM and CR will cause a decrease in the gas proportion of $\text{CF}_3\text{SO}_2\text{F}$ when coexisting with $\text{CF}_3\text{SO}_2\text{F}$ for a long period of time.

Surface Morphology Changing Law. After the tests, the surface morphology of EPDM and CR was analyzed, as shown in Figure 7. At 70 $^{\circ}\text{C}$, the surface of EPDM after testing with $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ is seen to have some crystals precipitated. While

**Figure 7.** Surface morphology of EPDM after tests.

in the SF_6 control group, the surface morphology of EPDM is more smooth without obvious defects. The surface of EPDM is seen to have some crystals precipitated in both the test group with $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ and the control group with SF_6 at 85 and 100 $^{\circ}\text{C}$. It is seen that the erosion of $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ on the surface of EPDM is similar to the erosion of SF_6 on the surface of EPDM.

The surface morphology of CR after tests is shown in Figure 8. It is seen that the surface morphology of CR in $\text{CF}_3\text{SO}_2\text{F}/$

**Figure 8.** Surface morphology of CR after tests.

N_2 at 70 $^{\circ}\text{C}$ has obvious unevenness, while there is no obvious phenomenon in the control group with SF_6 . At 85 $^{\circ}\text{C}$, CR in $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ shows obvious cracks and defects on the surface, while the surface of CR in SF_6 is still intact. At 100 $^{\circ}\text{C}$, CR in $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ shows pits on the surface, but the CR in the control group still shows no obvious defects. This indicates that the erosion of CR by $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ is significantly higher than that in the control group.

In addition, the surface elements of EPDM and CR were analyzed, and the elemental content on the rubber surface was detected using the point scanning method. The surface elemental contents of EPDM with $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ and SF_6 after thermal acceleration tests are shown in Tables 2 and 3, respectively. Seven elements, namely C, O, Al, Si, S, Ca, and Zn, were detected on the surface of EPDM in the SF_6 control group. Among them, C is the element in the main chain of EPDM rubber. O, Al, Ca, and Zn are some metal oxide additives added to the rubber in the production process. S comes from vulcanizing agents, and Si comes from SiO_2 reinforcing agents. In the $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ test group, it can be clearly seen that in addition to the seven elements detected in the control group, three elements were also detected, namely,

Table 2. Surface Elemental Content after Tests of EPDM and CF₃SO₂F/N₂

element	70 °C test group/%	85 °C test group/%	100 °C test group/%
C	94.99	96.10	95.59
O	3.45	2.63	2.45
F	0.30	0.48	0.72
Na	0.12		0.10
Al	0.08	0.09	0.07
Si	0.15	0.13	0.13
S	0.21	0.00	0.16
Cl		0.06	0.06
Ca	0.45	0.23	0.48
Zn	0.25	0.27	0.21

Table 3. Surface Elemental Content after Tests of EPDM and SF₆

element	70 °C test group/%	85 °C test group/%	100 °C test group/%
C	92.06	92.13	90.00
O	6.24	5.24	6.22
Al		0.25	0.16
Si	0.43	0.40	
S	0.00	0.50	1.09
Ca		0.82	1.76
Zn	1.26	0.66	0.77

Table 4. Surface Elemental Content after Tests of CR and CF₃SO₂F/N₂

element	70 °C test group/%	85 °C test group/%	100 °C test group/%
C	90.41	83.95	80.74
O	4.09	7.14	8.76
F	1.04	3.69	4.99
Na	0.14	0.15	0.14
Mg	0.46	1.21	1.18
Al	0.09	0.08	0.09
S	0.56	1.06	1.28
Cl	2.84	2.24	2.37
Ca	0.16	0.15	0.09
Fe		0.07	0.07
Zn	0.21	0.25	0.30

Table 5. Surface Elemental Content after Tests of CR and SF₆

element	70 °C test group/%	85 °C test group/%	100 °C test group/%
C	86.08	85.86	84.60
O	5.67	5.76	7.15
Mg	0.26	0.35	0.29
Al	0.22	0.12	0.13
S	0.50	0.50	
Cl	6.07	6.36	7.27
Ca		0.29	
Zn	1.20	0.76	0.57

F, Na, and Cl. Na is also likely to come from the metal oxide additives contained in the rubber itself. The small amount of Cl detected may be due to the Cl impurities in the test gas adhering to the rubber surface or reacting with the rubber. The source of element F, which was not included in the control group, was CF₃SO₂F gas. This suggests that a chemical reaction between CF₃SO₂F and EPDM has occurred, resulting in the detection of the element F attached to the surface of EPDM.

The surface elemental contents of CR with CF₃SO₂F/N₂ and SF₆ after tests are shown in Tables 4 and 5. It is seen that the element F was not detected in the control group but

detected in the test group with CF₃SO₂F/N₂. This also indicates that chemical reactions between CF₃SO₂F and CR occurred, resulting in the attachment of the element F on the rubber surface.

The results of surface morphology tests showed that the surface morphology degradation of EPDM affected by CF₃SO₂F erosion was similar to that of SF₆, whereas CR was more severely affected by CF₃SO₂F erosion than by SF₆. Surface elemental tests also showed that both EPDM and CR are chemically reactive with CF₃SO₂F. The compatibility between EPDM and CF₃SO₂F is better than that of CR in terms of changes in rubber surface morphology.

Mechanical Properties Changing Law. Mechanical properties of rubber can reflect its sealing properties. The tensile and compression properties of the rubber materials were measured in this study. Figures 9 and 10 show the change

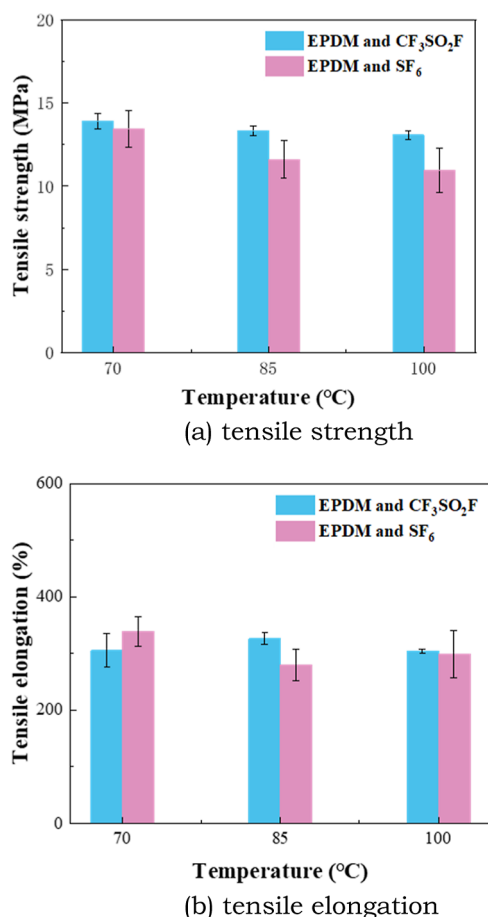


Figure 9. Tensile property of EPDM after the tests.

in tensile properties of EPDM and CR affected by $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ and SF_6 . It can be seen from Figure 9 that the tensile strength of EPDM in the $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ gas is higher than that in SF_6 . The tensile elongation of EPDM did not change significantly with temperature increase. Moreover, there is no obvious difference between the change of tensile elongation of EPDM in $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ and that in SF_6 . This indicates that the tensile properties of EPDM are less affected by $\text{CF}_3\text{SO}_2\text{F}$ and similar to that in SF_6 .

It can be seen from Figure 10 that the tensile strength of CR in $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ gas decreases slightly with an increase of temperature. The tensile strength of CR in $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ is higher than that in SF_6 at 70 $^{\circ}\text{C}$, and the gap is gradually narrowed with the increase of temperature. There is also no obvious difference between the change in tensile elongation of CR in $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ and that in SF_6 . It can be seen that the tensile property of EPDM and CR does not deteriorate significantly under the influence of $\text{CF}_3\text{SO}_2\text{F}$. The tensile property of EPDM and CR under the influence of $\text{CF}_3\text{SO}_2\text{F}$ is similar to that under SF_6 influence.

Figures 11 and 12 show the changes in compression properties of EPDM and CR under the influence of $\text{CF}_3\text{SO}_2\text{F}$ and SF_6 . Permanent deformation compression (PDC) and stiffness at 25% deformation are two main indicators of the

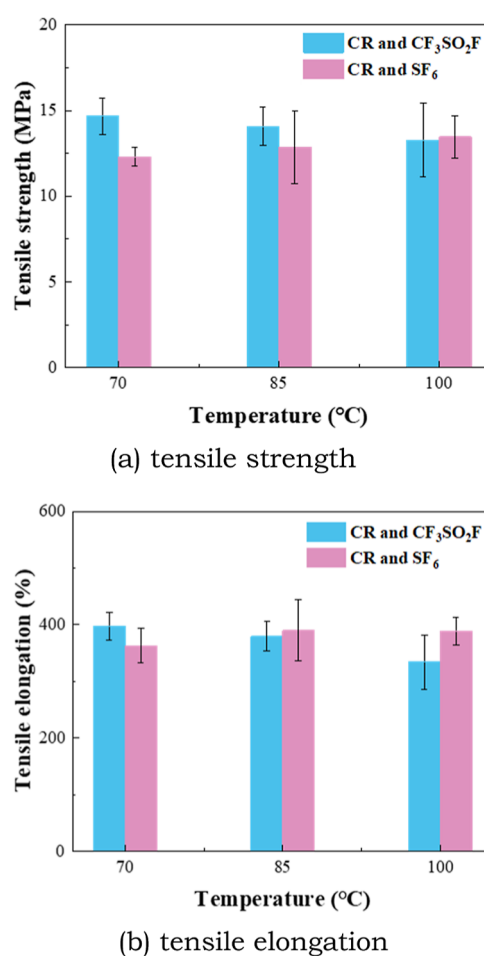
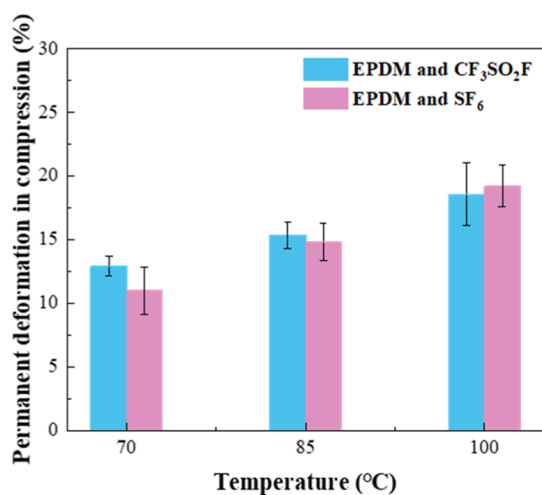


Figure 10. Tensile property of CR after the tests.

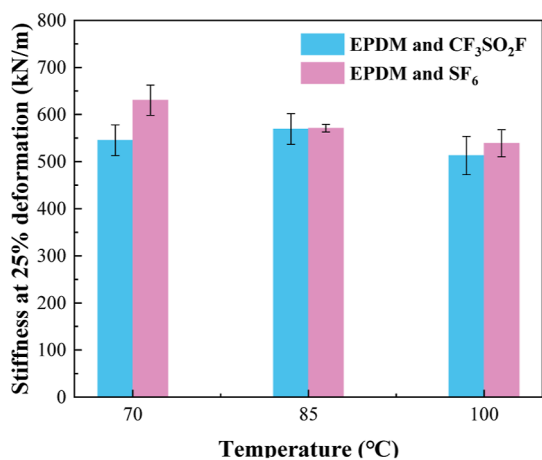
compression performance of rubber materials. As shown in Figure 11, the PDC of EPDM in $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ and SF_6 gas both increased gradually with temperature. But the gap of PDC of EPDM in $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ and SF_6 is small. It is seen from Figure 11 that the stiffness at 25% deformation of EPDM does not change much with temperature. The influence of $\text{CF}_3\text{SO}_2\text{F}$ on the stiffness at 25% deformation of EPDM is also similar to that of SF_6 .

As shown in Figure 12, the PDC of CR in $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ or SF_6 gas is significantly higher than that of EPDM, and the stiffness at 25% deformation of CR is significantly lower than that of EPDM. This indicates that the compression performance of CR is less better than EPDM. The PDC of CR in $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ increases significantly with temperature, but the PDC of CR in SF_6 changes little with temperature. The stiffness at 25% deformation of CR in $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ decreases significantly with temperature increase. But it changes little with temperature increase in SF_6 gas. This indicates that the compression performance of CR is greatly affected by $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ gas, but it will not be affected by SF_6 . The result corresponds well with the surface test results in Figure 8.

The tensile property of EPDM and CR in $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ is similar to that in SF_6 , and the effect of the temperature is not obvious. The PDC property of EPDM in $\text{CF}_3\text{SO}_2\text{F}/\text{N}_2$ is also similar to that in SF_6 , but the PDC property is significantly affected by temperature changes. For rubber sealing materials, compression performance is the main performance to be concerned about, so it can be assumed that temperature is the



(a) permanent deformation compression

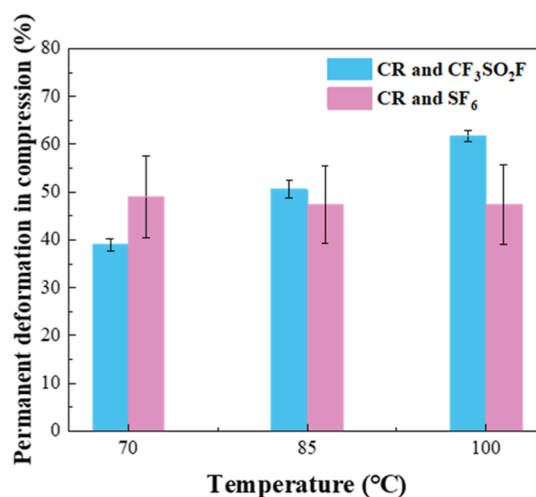


(b) stiffness at 25% deformation

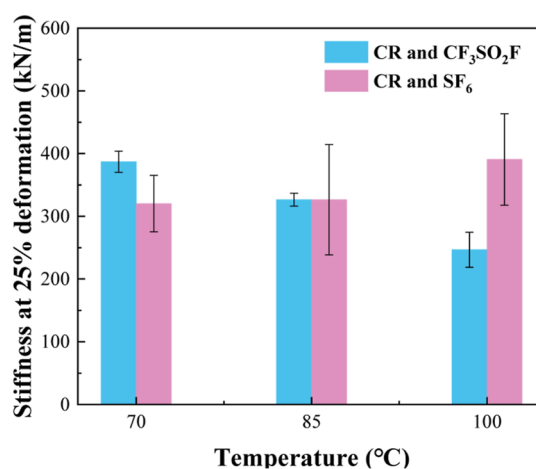
Figure 11. Compression property of EPDM after the tests.

dominant influence on the mechanical properties of EPDM. However, for CR, the influence of CF₃SO₂F on the compression performance is not negligible. This may be due to a chemical reaction between CF₃SO₂F gas and the surface of CR rubber, which can be confirmed by the surface measurement results.

Discussion. Through previous experimental research, we can obtain the compatibility between the EFI gas CF₃SO₂F and the EPDM and CR rubber used in existing power equipment, which is of great significance for its practical engineering application. However, at the scientific level, it is usually more concerning what kind of reaction occurs between CF₃SO₂F and the surface of rubber, which is essentially needed to be addressed in terms of gas–solid compatibility. However, the rubber used in power equipment is a polymer material with very complex molecular formula, and additives such as metal oxides are also added to the rubber to improve its sealing performance and reliability. This makes it difficult to analyze the specific reactions that occur between CF₃SO₂F gas and rubber. In this section, a brief analysis of the reaction mechanism will be conducted based on the experimental results.



(a) permanent deformation compression



(b) stiffness at 25% deformation

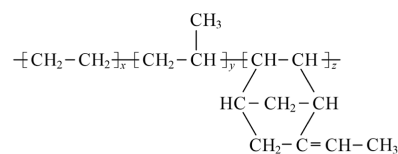
Figure 12. Compression property of CR after the tests.

The molecular formulas of EPDM and CR rubbers and CF₃SO₂F gas are shown in Figure 13. From the data in Tables 2–5, it can be seen that in addition to basic elements such as C and Cl, these rubbers also contain elements such as O, Zn, Mg, Al, Ca, etc., which are mainly from additives. Comparing the experimental group with the SF₆ control group, it can be seen that F will appear after the experiment in both EPDM and CR, and the proportion of F content increases with an increase of temperature. This indicates that CF₃SO₂F gas reacts with the rubber surface. For EPDM, there was no significant change in the C element and a slight decrease in the O element content, indicating that CF₃SO₂F may have reacted with the additives in it. For CR, there was a significant decrease in C element, indicating a reaction between the rubber body and the gas, which is consistent with the experimental test results. For the specific reaction process, further simulation calculation research is needed in the future.

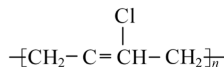
CONCLUSIONS

In this study, the compatibility between CF₃SO₂F/N₂ mixed gas and commonly used EPDM and CR rubber materials is investigated experimentally. The conclusions are as follows.

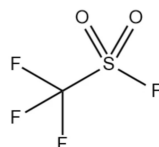
- (1) At the test temperatures of 70, 85, and 100 °C, the decomposition of CF₃SO₂F gas occurs after coexisting



(a) Structural formula of EPDM rubber



(b) Structural formula of CR rubber

(c) Structural formula of CF₃SO₂F**Figure 13.** Structural formulas of different rubbers and CF₃SO₂F gas.

with EPDM and CR for 7 d. The increase of temperature will lead to more serious decomposition of CF₃SO₂F gas. The mole proportion of CF₃SO₂F gas in the CF₃SO₂F/N₂ mixture will decrease accordingly. This is unfavorable for practical use, and it is necessary to focus on the changes in the mole proportion of CF₃SO₂F gas.

- (2) The surface morphology of CR is degraded by CF₃SO₂F, and a large number of defects appear on the surface. The mechanical properties of CR are greatly affected by CF₃SO₂F. It means that CF₃SO₂F gas is incompatible with CR rubber, and CR is not suitable as a sealing material for equipment containing CF₃SO₂F gas. The change in EPDM surface morphology in CF₃SO₂F/N₂ is similar to that in SF₆. From this perspective, EPDM can be considered a sealing material for equipment containing CF₃SO₂F gas, but the temperature of the sealing ring needs to be strictly controlled, otherwise it will affect the performance of EPDM.
- (3) The tensile property of EPDM and CR sealing rubber materials is less affected by temperature, but the compressive property is more affected by temperature, especially the PDC property. PDC and surface morphology are two more effective indicators for characterizing the compatibility between the insulating gas and rubber sealing materials. This provides a reference for future research on the gas–solid compatibility between other rubber sealing materials and EFI gases.

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<https://pubs.acs.org/10.1021/acsomega.3c07958>

Notes

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ACKNOWLEDGMENTS

The authors thank the National Key R&D Program of China (Grant no. 2021YFB2401400) and the State Grid Corporation of China Technology Project (Grant no. SGAHDK00LJJS2200332) for supporting this study and the Testing Center of Wuhan University for surface testing services.

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