



# Article $Ti_3C_2T_x$ Coated with $TiO_2$ Nanosheets for the Simultaneous Detection of Ascorbic Acid, Dopamine and Uric Acid

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**Abstract:** Two-dimensional MXenes have become an important material for electrochemical sensing of biomolecules due to their excellent electric properties, large surface area and hydrophilicity. However, the simultaneous detection of multiple biomolecules using MXene-based electrodes is still a challenge. Here, a simple solvothermal process was used to synthesis the  $Ti_3C_2T_x$  coated with  $TiO_2$  nanosheets ( $Ti_3C_2T_x@TiO_2$  NSs). The surface modification of  $TiO_2$  NSs on  $Ti_3C_2T_x$  can effectively reduce the self-accumulation of  $Ti_3C_2T_x$  and improve stability. Glassy carbon electrode was modified by  $Ti_3C_2T_x@TiO_2$  NSs ( $Ti_3C_2T_x@TiO_2$  NSs/GCE) and was able simultaneously to detect dopamine (DA), ascorbic acid (AA) and uric acid (UA). Under concentrations ranging from 200 to 1000  $\mu$ M, 40 to 300  $\mu$ M and 50 to 400  $\mu$ M, the limit of detection (LOD) is 2.91  $\mu$ M, 0.19  $\mu$ M and 0.25  $\mu$ M for AA, DA and UA, respectively. Furthermore,  $Ti_3C_2T_x@TiO_2$  NSs/GCE demonstrated remarkable stability and reliable reproducibility for the detection of AA/DA/UA.

Keywords: dopamine; ascorbic acid; uric acid;  $Ti_3C_2T_x$ ;  $TiO_2$ ; solvothermal; electrochemical sensor

# 1. Introduction

MXene, a type of two-dimensional (2D) material, is characterized by the formula  $M_{n+1}X_nT$ . In this formula, M represents transition metals, X could be either carbon or nitrogen, while T refers to surface functional groups like =O, –F, and –OH [1,2], and it attracts attention as a promising material for electrochemical modification [3,4]. In particular, the abundant functional groups, high specific surface area and excellent electrical properties of MXene provide promising applications for the electrochemical detection of various biomolecules, especially for dopamine (DA), ascorbic acid (Vitamin C or AA) and uric acid (UA). The stability of physiological functions can be influenced by these three important substances which coexist in body fluids [5]. As a poly-hydroxyl molecule, AA plays a dual role in promoting antibody production and neutralizing the harmful impact of free radicals [6]. DA, a catecholamine neurotransmitter, contributes to nervous system regulation [7]. UA serves as a purine metabolite, commonly present in living organisms [8]. Several common diseases are attributed to fluctuations in the concentrations of these biomolecules, including mental illness, AIDS, Parkinson's disease, skin rashes and gout [9,10]. Thus, it is important to discover an appropriate approach for swiftly and precisely identifying AA, DA and UA.

Recently, the use of MXene as a modified electrode material has been shown to enable accurate detection of AA, DA and UA [11,12]. However, two factors limit the application of MXene: On the one hand, the unstable multilayer structure of MXene is



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**Copyright:** © 2024 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (https:// creativecommons.org/licenses/by/ 4.0/). prone to self-accumulation, which reduces the active sites and lowers the electron transfer rate [13]. On the other hand, the functional groups on the surface of MXene caused by wet etching are unable to form covalent bonds with the biomolecules, thus reducing the detection sensitivity [14]. Therefore, several strategies have been developed to resolve the above problems, such as the use of intercalating agents and surface modification/functionalization [15,16]. Notably, the preparation of composites (or heterojunctions) by introducing other active materials into MXenes is considered to be the best strategy. For instance, a novel MXene/DNA/Pd/Pt material was synthesized by Zheng et al., and the sensor exhibited high selectivity against AA and  $H_2O_2$  [17]. Amara et al. fabricated a graphitic pencil which is modified by perylene diimide/MXene ( $Ti_3C_2T_x$ ), realizing the specific detection of DA [18]. However, the performance of MXene-based materials in simultaneously detecting AA/DA/UA remains unsatisfactory due to the significant concentration differences among different biomolecules [19], their similar oxidation potentials [20] and the strong electrostatic repulsion [21].  $TiO_2$  is recognized for its considerable potential for use in the biological sector, thanks to its affordability, customizable attributes, lack of toxicity, simple preparation process, consistent quality and improved durability [22,23].

Furthermore, the nanocomposite of MXene@TiO<sub>2</sub> prevents the aggregation of MXene, and it improves the stability of the electrochemical sensor and achieves reliable reproducibility [24]. So far, different forms of TiO<sub>2</sub> have been synthesized, including nanoparticles, thin films structures, nanorods and so on [25,26]. In our previous work [27], it was mentioned that materials that have a high specific surface area can enable the precise detection of AA/DA/UA because of the presence of numerous active sites. Therefore, the growth of TiO<sub>2</sub> nanosheets on the MXene surface can be used as a means to effectively realize the simultaneous detection of biological small molecules.

Here,  $Ti_3C_2T_x$  coated with TiO<sub>2</sub> nanosheets ( $Ti_3C_2T_x@TiO_2$  NSs) is synthesized through a simple solvothermal process. By incorporating TiO<sub>2</sub> nanosheets,  $Ti_3C_2T_x@TiO_2$  NSs, modified glassy carbon electrode ( $Ti_3C_2T_x@TiO_2$  NSs/GCE) enables detection of AA/DA/UA individually and simultaneously. Additionally,  $Ti_3C_2T_x@TiO_2$  NSs/GCE demonstrates outstanding stability and reliable reproducibility.

#### 2. Result and Discussion

# 2.1. Microstructural Characterizations of $Ti_3C_2T_x@TiO_2$ NSs Samples

As shown in Figure S1, comparing the standard cards (JCPDS Card No. 52-0875), it can be seen that the black curves show peaks at 19.1°, 34°, 36.77°, 38.94°, 41.7°, 48.3° and 60.14° corresponding to the (004), (101), (103), (104), (105), (107) and (110) crystalline facet characteristic peaks of  $Ti_3AlC_2$ . The red curve shows the peaks at  $9.0^\circ$ ,  $18.6^\circ$ ,  $27.9^\circ$  and  $(61.1^{\circ})$ , which correspond to the (002), (006), (008) and (110) facets of  $Ti_3C_2T_x$  [28]. Notably, the characteristic peak at 39.3° disappears, indicating the etching of the Al element [29]. Upon etching of the Al element,  $TiO_2$  grows on the MXene surface due to oxidation, where peaks at 36.9°, 37.8° and 38.6° (corresponding to the (103), (004) and (112) crystal planes of TiO<sub>2</sub>) align with characteristic peaks of the (103) crystal planes of  $Ti_3C_2T_x$ , leading to widened peaks at  $35^{\circ}-40^{\circ}$ . Prior to annealing (blue curve), the shift of the (002) peak toward lower angles suggests further delamination of  $Ti_3C_2T_x$ . Additionally, comparing the standard cards (JCPDS card NO. 21-1272), two faint peaks at 38.6° and 44.1° emerge, corresponding to the (112) and (200) planes of anatase  $TiO_2$ . Following annealing (green curve), another peak appears at position 23.5°, corresponding to the (101) planes of anatase TiO<sub>2</sub>. After solvothermal treatment, the results show that the lamellae structure of  $Ti_3C_2T_x$ remains stable, while a crystallized phase of anatase TiO<sub>2</sub> has emerged.

Figure S2a depicts the closely aligned layered structure of the pristine  $Ti_3AlC_2$ . As shown in Figure S2b, the  $Ti_3C_2T_x$  exhibits a layered accordion-like structure following treatment with HF (24 h), with each lamella measuring around 20 nm thick and a layer spacing of approximately 300 nm. Figure 1a–c illustrate the comparison of samples with varying amounts of TiCl<sub>3</sub>. When the TiCl<sub>3</sub> concentration is low ( $Ti_3C_2T_x@TiO_2$  NSs(0.6), Figure 1a), sparse NSs can be observed on the surface of  $Ti_3C_2T_x$  lamellae. As the amount of TiCl<sub>3</sub> increases (Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs(0.8), Figure 1b), the density of TiO<sub>2</sub> NSs on the surface of Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> lamellae gradually increases. At the maximum TiCl<sub>3</sub> concentration (Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs(1.0), Figure 1c), highly dense NSs, and even nanospheres composed of NSs, are formed on the surface of Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> lamellae, indicating an excess of TiCl<sub>3</sub>. After annealing, the morphology remains relatively unchanged (Figure 1d–f). The TEM images of Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs(0.8) are presented in Figure 1g,h. Figure 1g shows abundant NSs growing on the surface of Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> lamellae, with lengths of 30–100 nm and thicknesses of 3–5 nm. A facet spacing of 0.253 nm wide can be observed in the edge portion of the material, corresponding to the (002) crystallographic facet of Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>. The lattice stripe spacing (0.352 nm) of the NSs is correlated to the (101) crystal plane of anatase TiO<sub>2</sub> [30] (Figure 1h). The SAED image of Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs is demonstrated in Figure 1i (red). The diffraction circle diameters in the SAED plots are the same as those diffracted from the (101) and (103) crystal surfaces of anatase TiO<sub>2</sub>, while the six bright and sharp spots exhibit a six-fold symmetry, representing the crystal planes (0110), (1010) and (1100) of Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> (Figure 1i, inset).



**Figure 1.** SEM images of  $Ti_3C_2T_x@TiO_2$  NSs(0.6): (a)  $Ti_3C_2T_x@TiO_2$  NSs(0.8), (b)  $Ti_3C_2T_x@TiO_2$  NSs(1.0), (c) SEM images of  $Ti_3C_2T_x@TiO_2$  NSs(0.6), (d)  $Ti_3C_2T_x@TiO_2$  NSs(0.8), (e)  $Ti_3C_2T_x@TiO_2$  NSs(1.0), (f) after annealed at 350 °C, (g) TEM image, (h) HR-TEM and (i) SAED image of  $Ti_3C_2T_x@TiO_2$  NSs(0.8).

Figure 2a shows the full XPS spectra of  $Ti_3C_2T_x$  and  $Ti_3C_2T_x@TiO_2$  NSs. The red curve in Figure 2a represents  $Ti_3C_2T_x$ , exhibiting characteristic peaks for C 1s, Ti 2p, O 1s and F 1s at 282 eV, 457 eV, 529 eV and 683 eV, respectively. The black curve in Figure 2a corresponds to  $Ti_3C_2T_x@TiO_2$  NSs, where the characteristic peak at 682 eV for F 1s is absent. This observation suggests that the  $Ti^{3+}$  of  $TiCl_3$  has replaced the –F functional groups. The Ti 2p profile (Figure 2b) reveals the energy levels of 461.2 and 457.2 eV, corresponding to Ti  $2p_{1/2}$  and Ti  $2p_{3/2}$ , which indicate the formation of anatase  $TiO_2$ . Concerning the C 1s profile (Figure 2c), characteristic peaks occur at 285.8, 283.2 and 281.7 eV, corresponding to O-C=O, C–O, and C–C, respectively. Furthermore, the O 1s profile (Figure 2d) shows peaks at 529.5 and 527.3 eV that represent C–O and Ti–O bonds in  $TiO_2$ , respectively. These findings provide evidence for the disappearance of –F functional groups and the concomitant generation of  $TiO_2$  on the surface of  $Ti_3C_2T_x$ .

The process of morphological evolution involved in the fabrication of  $Ti_3C_2T_x$  @TiO<sub>2</sub> NSs is illustrated in Figure 3. Initially, Ti<sup>4+</sup> ions are generated through the reaction of Ti<sup>3+</sup> of

TiCl<sub>3</sub> with dissolved oxygen in the solution [31,32] (Equations (S1) and (S2)). Subsequently, these positively charged Ti<sup>4+</sup> ions uniformly adhere to the negatively charged surface of Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>. Concurrently, Ti<sup>4+</sup> ions react with ethylene glycol to form titanium glycolate crystal nucleuses (Ti(OCH<sub>2</sub>CH<sub>2</sub>O)<sub>2</sub>) [33] (Equation (S3)). As the solvothermal process proceeds, a portion of Ti(OCH<sub>2</sub>CH<sub>2</sub>O)<sub>2</sub> undergoes hydrolysis, leading to the development of a small quantity of anatase TiO<sub>2</sub> on the surface of Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>. In the final annealing step, the remaining unreacted Ti(OCH<sub>2</sub>CH<sub>2</sub>O)<sub>2</sub> is completely transformed into anatase TiO<sub>2</sub>, while the carbon-containing species are entirely eliminated (Equation (S4)).



Figure 2. XPS spectra of the (a) survey, (b–d) Ti 2p, C 1s and O 1s for the Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs.



**Figure 3.** Growth mechanism of  $TiO_2$  nanosheet on  $Ti_3C_2T_x$  surface.

Figure 4a shows the electrochemical performance of Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs/GCE, Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>/ GCE and GCE using CV curves in 0.1 M KCl solution with 0.5 mM K<sub>3</sub>[Fe (CN)<sub>6</sub>]. The relative peak currents, listed in decreasing order, are as follows: Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs(0.8)/GCE > Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs(1.0)/GCE > Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs(0.6)/GCE > Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs(1.0)/GCE > Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs(0.6)/GCE > GCE. As shown in Figure 4b, the impedance of the different electrodes was obtained by EIS (voltage amplitude of 10 mV,  $10^{-2}$ – $10^{6}$  Hz). The electron transfer process occurring at the electrode surface was indicated by the semicircular arc observed at higher frequencies. The charge transfer resistance ( $R_{ct}$ ) of the electrodes can be represented by the semicircular arc diameter of the fitted plot line. Based on the Randles equivalent circuit (Figure 4b, inset), Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs(0.8)/GCE exhibits the lowest charge transfer impedance ( $R_{ct} = 531 \Omega$ ), followed by Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>/GCE (868.7 Ω), Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs(0.6)/GCE (608.7 Ω) and Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs(1.0)/GCE (1183 Ω).



**Figure 4.** (a) CV curves and (b) EIS curve of  $Ti_3C_2T_x@TiO_2$  NSs/GCE,  $Ti_3C_2T_x/GCE$  and GCE; (c) CV curves of  $Ti_3C_2T_x@TiO_2$  NSs(0.8)/GCE with different scanning rates. The CV curves of  $Ti_3C_2T_x@TiO_2$  NSs/GCE,  $Ti_3C_2T_x/GCE$  and GCE are detected in (d) 100  $\mu$ MAA, (e) 100  $\mu$ M DA and (f) 100  $\mu$ M UA solutions. In all the experiments, detection was in 0.1 M KCl solution with 0.5 mM K<sub>3</sub>[Fe (CN)<sub>6</sub>].

As shown in Figure 4c, which displays the CV curves, the peak current of  $Ti_3C_2T_x@TiO_2$  NSs(0.8)/GCE rises in a linear fashion as the scan rate increases (20–100 mV/s). Furthermore, the redox peaks of the electrodes display a good linear relationship with the square root of scan rate (the inset of Figure 4c). The active surface area of  $Ti_3C_2T_x@TiO_2$  NSs/GCE was calculated using the Randles–Sevcik equation:

$$Ip = 2.69 \times 10^5 \times n^{3/2} \times AC^* \times D^{1/2} \times \upsilon^{1/2}$$
(1)

where  $I_p$  represents the peak oxidation current, n is the number of electron transfers (n = 1), A is the active surface area, D is the diffusion coefficient ( $7.6 \times 10^{-6} \text{ cm}^2 \text{ s}^{-1}$  for [Fe(CN)<sub>6</sub>]<sup>4+</sup>),  $C^*$  is the concentration of [Fe(CN)<sub>6</sub>]<sup>4+</sup> and v is the scanning rate. The active surface area of Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs(0.8)/GCE was calculated as approximately 0.74 cm<sup>2</sup>, which is nine times greater than that of bare GCE (0.08 cm<sup>2</sup>). The TiO<sub>2</sub> NSs grown on the surface of Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> effectively increased the area of catalytic activity. As shown in Figure 4d, the reaction peak could not be detected by Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs/GCE is employed. The peak current, listed in descending order, is as follows: Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs(0.8)/GCE > Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub>

NSs(0.6)/GCE > Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs(1.0)/GCE > GCE. As shown in Figure 4e, all five electrodes can detect the oxidation peak of DA, with the peak current ordered as follows: Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs(0.8)/GCE > Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>/GCE > GCE > Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs(1.0)/GCE > Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs(0.6)/GCE. The five oxidation peaks of UA are clearly distinguished, with the peak current ordered as follows: Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs(0.6)/GCE > Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs(0.6)/GCE > Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs(0.8)/GCE > Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs(0.8)/GCE > Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs(0.8)/GCE > Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs(0.6)/GCE > Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs(0.6)/GCE > Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs(0.8)/GCE > ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs(0.6)/GCE > GCE (Figure 4f). Since Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs(0.8)/GCE showed excellent electrochemical detection capability, it was used as the modified electrode material for the additional experiments.

The oxidation kinetics of AA, DA and UA on the  $Ti_3C_2T_x@TiO_2$  NSs/GCE electrode were studied using CV (pH 7.4) under various scan rates (20 mV/s–100 mV/s). The peak currents of AA, DA and UA show strong linear correlation with the scan rate (Figure 5a–c), while the peak potentials are gradually biased toward large overpotentials, indicating kinetic limitations in the reaction [34]. As shown in Figure 5d–f, the linear correlation coefficient (R<sup>2</sup>) between the oxidation current and square root of the scan rate was determined to be 0.9868, 0.9801 and 0.9700 for AA, DA and UA. Hence, it can be confirmed that the oxidation reaction of AA, DA and UA on  $Ti_3C_2T_x@TiO_2$  NSs/GCE was controlled by a diffusion process [35].



**Figure 5.** CV curves recorded in (**a**) 100  $\mu$ M AA, (**b**) 100  $\mu$ M DA and (**c**) 100  $\mu$ M UA, with increasing scan rates on Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs/GCE. (**d**–**f**) The square root of the scan rates vs. peak current of AA/DA/UA. Error bars based on *S*/*N* = 3.

# 2.3. The Use of $Ti_3C_2T_x@TiO_2$ NSs/GCE for Separate and Simultaneous Detection of AA, DA and UA

Figure 6 demonstrates the separate detection of AA, DA and UA on  $Ti_3C_2T_x@TiO_2$ NSs/GCE in 0.1 M PBS (pH 7.4) through the use of DPV. Figure 6a–c show that  $Ti_3C_2T_x@TiO_2$ NSs/GCE exhibits the individual detection of AA, DA and UA under concentrations ranging from 40 to 300  $\mu$ M, 40 to 500  $\mu$ M and 50 to 600  $\mu$ M, respectively. Figure 6d–f show the R<sup>2</sup>a of AA, DA and UA as 0.9994, 0.9941 and 0.9975, respectively. The LOD for AA, DA and UA was determined to be 1.12  $\mu$ M, 0.35  $\mu$ M and 0.47  $\mu$ M, respectively.

Figure 7a demonstrates the simultaneous determination of AA and DA through DPV on  $Ti_3C_2T_x@TiO_2$  NSs/GCE. A noticeable difference was observed in the peak potentials of AA (0.27 V) and DA (0.46 V). For AA (80–280  $\mu$ M) and DA (10–60  $\mu$ M), a direct correlation was identified between peak currents and concentrations, resulting in R<sup>2</sup> values of 0.9845 and 0.9922, respectively (Figure 7b,c). Similarly, Figure 7d illustrates the distinct separation of DA (0.33 V) and UA (0.54 V) oxidation peak potentials. As shown in Figure 7e,f, a linear



**Figure 6.** DPV curves of (**a**) AA, (**b**) DA and (**c**) UA, separately detected on  $Ti_3C_2T_x@TiO_2$  NSs/GCE, and the corresponding oxidation concentration vs. peak current. (**d**–**f**). Error bars based on *S*/*N* = 3.



**Figure 7.** DPV curves based on different concentrations when detecting (**a**) AA and DA and (**d**) DA and UA on  $Ti_3C_2T_x@TiO_2$  NSs/GCE in 0.1 M PBS (pH 7.4). (**b**,**c**) Calibration curves for the current vs. concentrations of AA and UA, (**e**,**f**) calibration curves for the current vs. concentrations of DA and UA. Error bars based on *S*/*N* = 3.

It should be noted that when AA and UA coexisted in solution, no discernible peak splitting phenomenon was observed (Figure S5a–c). At the same time, the oxidation

potential was shifted higher upon the addition of AA or UA. This phenomenon is attributed to the close oxidation potential of AA and UA; the offset of the AA and UA peaks is due to the catalytic oxidation of UA by AA [36].

The DPV curves of the simultaneous determination of AA, DA and UA using  $Ti_3C_2T_x@TiO_2$  NSs/GCE are shown in Figure 8. As depicted in Figure 8a, the oxidation peak potentials of AA, DA and UA were observed to be 0.06 V, 0.37 V and 0.55 V, respectively. Additionally, Figure 8b-d highlight the linear relationship between peak currents and concentrations, with observed ranges of 200–1000  $\mu$ M (R<sup>2</sup> = 0.9875) for AA, 40–300  $\mu$ M  $(R^2 = 0.9875)$  for DA and 50–400  $\mu$ M ( $R^2 = 0.9991$ ) for UA. The LOD was determined to be  $2.91 \mu$ M,  $0.19 \mu$ M and  $0.25 \mu$ M for AA, DA and UA, respectively. Subsequently, this work was compared with previous work on electrochemical sensors (Table 1). There are several key advantages in our work: Firstly, our synthesis method is simple, and the preparation conditions are gentle, in contrast to the complex synthesis processes and high-temperature preparation conditions commonly found in the existing literature [37-39]. Secondly, our use of non-precious metals is more cost-effective than the use of precious metals; even though precious metals can enhance material sensitivity, they also escalate manufacturing costs [40,41]. Materials such as MXene and TiO<sub>2</sub> present a promising low-cost alternative. Lastly, our work demonstrates a lower Limit of Detection (LOD) for AA when using MXenebased materials. While the negative electronegativity of the MXene surface traditionally hinders AA detection, the TiO<sub>2</sub> synthesized via the solvothermal method effectively optimizes the surface charge of MXene, enabling the simultaneous detection of AA, DA and UA, with a particularly noteworthy LOD of 2.9 µM for AA.



**Figure 8.** (a) DPVs recorded for  $Ti_3C_2T_x@TiO_2$  NSs/GCE for different concentrations of AA (200–1000  $\mu$ M), DA (40–300 $\mu$ M) and UA (50–400 $\mu$ M). (**b**–**d**) Linear calibration curves for the oxidation peak currents vs. concentrations of AA, DA and UA. Error bars based on *S*/*N* = 3.

It is noteworthy that the inclusion of DA exhibited an ability to separate the overlapping peaks of AA and UA. The oxidation products of AA and UA absorb or electropolymerize on the electrodes and reduce their detection sensitivity [42,43]. When DA is added, the oxidation products of DA react with AA, thereby mitigating electrode contamination. Additionally, DA is recognized as a catalyst for both AA and UA [44,45]. Upon the addition of DA, it underwent reactions with AA and UA to generate intermediate complexation products, effectively lowering the activation energy of AA and increasing the activation energy of UA. Consequently, the peaks corresponding to AA and UA resurfaced during simultaneous detection.

**Table 1.** The performance of sensors for the electrochemical simultaneous detection of AA, DA and UA has been reported recently.

| Electrode   | Linear Range (µM) |          |          | LOD (µM) |      |      | Refs      |
|---|-------------------|----------|----------|----------|------|------|-----------|
|   | AA                | DA       | UA       | AA       | DA   | UA   | - 10015   |
| FGP   | 10-1800           | 1–300    | 5-800    | 4.2      | 0.42 | 2.2  | [46]      |
| P-4-ABA/GCE   | 20-800            | 5-100    | 1-80     | 5.0      | 1.0  | 0.5  | [47]      |
| HNP-PtTi  | 200-1000          | 4-500    | 100-1000 | 24.2     | 3.2  | 5.3  | [48]      |
| p-TA/nano-Au/GCE  | 2.1 - 50.1        | 0.6-340  | 1.6-110  | 14.8     | 0.05 | 0.1  | [49]      |
| Fe <sub>3</sub> O <sub>4</sub> /Co <sub>3</sub> O <sub>4</sub> /mc@g-C <sub>3</sub> N <sub>4</sub> /GCE | 500-8000          | 1–70     | 5-100    | 12.55    | 0.21 | 0.18 | [50]      |
| SPGNE   | 4-4500            | 0.5-2000 | 0.8-2500 | 0.95     | 0.12 | 0.20 | [8]       |
| PG/GCE  | 9-2314            | 5-710    | 6-1330   | 6.45     | 2.00 | 4.82 | [51]      |
| Pt@g-C <sub>3</sub> N <sub>4</sub> /N-CNTs/GCE  | 2-2000            | 5-100    | 1-110    | 29.44    | 0.21 | 2.99 | [52]      |
| Co/MoSe <sub>2</sub> /PPy@CNF   | 30-3212           | 1.2-536  | 10-1071  | 6.32     | 0.45 | 0.81 | [37]      |
| $Ti_3C_2T_x/TiO_2$ NSs  | 200-1000          | 40-300   | 50-400   | 2.9      | 0.19 | 0.25 | This work |

FGP—Few-layer graphene/Pt; GCE—glassy carbon electrode; HNP—hierarchical nanoporous; P-4-ABA—Poly (4-aminobutyric acid); p-TA/nano-Au—gold nanoparticles on poly (3-amino-5-mercapto-1,2,4-triazole) film; Fe<sub>3</sub>O<sub>4</sub>—iron oxide; Co<sub>3</sub>O<sub>4</sub>—cobalt oxide; SPGNE—screen-printed graphene electrode; g-C<sub>3</sub>N<sub>4</sub>—graphitic carbon nitride; N-CNTs—N-doped carbon nanotubes; PG—pristine graphene; and Co/MoSe<sub>2</sub>/PPy@CNF—cobalt-doped 2D-MoSe<sub>2</sub>/polypyrrole hybrid-based carbon nanofibers.

Based on previous studies [40,41], simultaneous detection of AA/DA/UA faces significant challenges due to mutual interference, such as strong electrostatic interactions and overlapping oxidation potentials [36,38,53]. Through the solvothermal process,  $Ti_3C_2T_x@TiO_2$ NSs exhibit two distinct advantages that enhance their ability to detect these molecules simultaneously. First, the replacement of –F by additional  $Ti^{3+}$  ions significantly changes the surface charge of  $Ti_3C_2T_x$ , reducing the mutual repulsion of the analytes. Secondly, a considerable number of  $TiO_2$  NSs are grown on the  $Ti_3C_2T_x$  lamellae surface, which decreases the ion diffusion length [54] and increases the specific surface area of the  $Ti_3C_2T_x@TiO_2$ NSs. Finally, the formation of  $Ti_3C_2T_x$  and  $TiO_2$  heterojunctions facilitates the transport of charge carriers [39]. As a result, the integration of  $Ti_3C_2T_x$  and  $TiO_2$  allows for the creation of a highly precise and sensitive sensor capable of detecting biological small molecules with high resolution.

### 2.4. Reproducibility, Interference Immunity Testing and Stability Analysis

Several potential coexisting substances were detected using the amperometric method to verify the interference resistance of  $Ti_3C_2T_x@TiO_2$  NSs/GCE (Figure 9a–c). The continuous addition of the interfering substance did not affect the detection current of the AA/DA/UA. Therefore,  $Ti_3C_2T_x@TiO_2$  NSs/GCE demonstrated excellent anti-interference capability for detecting AA, DA and UA. Five consecutive measurements were conducted in the solution containing 200  $\mu$ M AA, 60  $\mu$ M DA and 40  $\mu$ M UA, investigating the stability of  $Ti_3C_2T_x@TiO_2$  NSs/GCE. The results shown in Figure 9d indicate that the peaks of the five consecutive measurements remained almost unchanged, with a calculated RSD of 0.54%. This suggests that  $Ti_3C_2T_x@TiO_2$  NSs/GCE exhibits good stability under testing conditions. Furthermore, a DPV test was performed after the electrode was exposed to air for 5 days to evaluate the environmental stability of  $Ti_3C_2T_x@TiO_2$  NSs/GCE. As shown in Figure 9f, the simultaneous detection of AA/DA/UA was utilized on five



different  $Ti_3C_2T_x@TiO_2$  NSs/GCE electrodes in 0.1 M PBS using DPV. The oxidation peak currents of AA (200 M), DA (40 M) and UA (60 M) were not significantly altered.

**Figure 9.** Amperometric responses of the mixture upon addition of 10  $\mu$ M (**a**) AA, (**b**) DA, (**c**) UA and 1000  $\mu$ M other chemicals on Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs/GCE; (**d**) DPV curves detected 5 consecutive times on Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs/GCE for AA, DA and UA; (**e**) DPV curve change of Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs/GCE containing 200  $\mu$ M AA, 40  $\mu$ M DA and 60  $\mu$ M UA (5 days) in 0.1 M PBS; (**f**) peak current using five different Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs/GCE electrodes at fixed concentrations of AA (200  $\mu$ M), DA (40  $\mu$ M) and UA (60  $\mu$ M) in PBS. Error bars based on *S*/*N* = 3.

### 3. Experimental Section

# 3.1. Chemicals

Kaixi Materials Co., Ltd. In Shandong, China supplied  $Ti_3AlC_2$  powder (94 wt%, ~300 mesh). Sinopharm Chemical Reagent Co., Ltd. in Beijing, China provided potassium chloride (KCl), ethyl alcohol (99.9%), titanium trichloride (TiCl<sub>3</sub>), hydrofluoric acid (40 wt% HF), acetone, potassium ferricyanide (K<sub>3</sub>[Fe (CN)<sub>6</sub>]), ethylene glycol (EG), ascorbic acid, dopamine, uric acid, dibasic sodium phosphate (Na<sub>2</sub>HPO<sub>4</sub>) and potassium dihydrogen phosphate (KH<sub>2</sub>PO<sub>4</sub>). Praxair Gas Co., Ltd. supplied nitrogen (N<sub>2</sub>) and argon gas (Ar). Yue Ci Electronic Technology Co., Ltd. in Shanghai, China provided aluminum oxide powder (1.0, 0.3 and 0.05 µm). Analytical grade chemicals were employed in this work without additional purification.

#### 3.2. Apparatus

An electrochemical workstation was used to obtain the electrochemical properties of the modified electrode. The model of the electrochemical workstation is CHI, 660E (Dalian, China). The surface morphology of the material was obtained by scanning electron microscopy (SEM) using the Nova 230, FEI (Hillsboro, OR, USA) model. The ultrastructure of the material was obtained by transmission electron microscopy, model Tecnai G2 F30 S-TWIN, FEI (USA). The high-resolution electron microscopy was obtained using JEM-2010 (HRTEM, JEOL Japan Electronics Co., Ltd, Akishima-shi in Japan). The structure and phase of the materials were analyzed using X-ray powder diffraction (XRD), model TTRIII, Rigaku (Tokyo, Japan) with Cu K $\alpha$  radiation ( $\lambda$  = 1.5406 Å) and X-ray photoelectron spectroscopy (XPS), using the VG Multilab 2009 (Manchester, UK) model.

# 3.3. Synthesis of $Ti_3C_2T_x$ @TiO<sub>2</sub> NSs

Scheme 1 shows the preparation process of  $Ti_3C_2T_x@TiO_2$  NSs.  $Ti_3C_2T_x$  was obtained by the one-step wet etching method [55]. The  $Ti_3AlC_2$  powder (1 g) was added gradually to a solution of HF (40 mL, 40 wt%) and allowed to react at 35 °C for 24 h with stirring at a speed of 500 rpm. The etched substance was rinsed with deionized water (DI) until the pH reached above 6.0. High-speed centrifugation (300 rpm, 3 min) allows the powder to be separated from the DI. Finally, the gelatinous  $Ti_3C_2T_x$  was dehydrated under vacuum freezing conditions (-80 °C, 2 h). A solution was prepared by mixing 30 mL of EG with an equal volume (k mL) of TiCl<sub>3</sub> and DI (k = 0.6, 0.8, 1.0) and stirring it for approximately 30 min. Subsequently, 20 mg of as-prepared  $Ti_3C_2T_x$  was added to the solution and sonicated for 30 min to create a homogeneous suspension. The suspension was moved to a stainless steel autoclave and subjected to heat treatment at 150 °C for 12 h before being cooled to room temperature. The precipitates were washed with ethyl alcohol, and dried for 2 days. Finally, the precipitates were annealed at 350 °C in flowing Ar for 2 h. The obtained granular powders were marked as  $Ti_3C_2T_x@TiO_2$  NSs(0.6),  $Ti_3C_2T_x@TiO_2$  NSs(0.8), and  $Ti_3C_2T_x@TiO_2$  NSs(1.0), respectively.



**Scheme 1.** The synthesis process of  $Ti_3C_2T_x@TiO_2$  NSs.

#### 3.4. Fabrication of the $Ti_3C_2T_x@TiO_2$ NSs Electrode

Aluminum oxide powder with particle sizes of 1.0, 0.3 and 0.05  $\mu$ m was first placed on a nylon cloth for polishing the glassy carbon electrode (GCE). The electrodes were then washed continuously for 5 min using acetone, ethanol and deionized water (DI), respectively, until it was ensured that a smooth electrode surface was obtained. Then, 10 mg of Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs was dispersed in 10 mL of DI using the ultrasonic method. Next, 3  $\mu$ L of the aforementioned Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs solution was dropped on the GCE surface and allowed to dry under nitrogen atmosphere to form a flat film. For comparison, the preparation process of the remaining electrodes was performed in a similar way.

# 3.5. Electrochemical Measurement

The main purpose of this work is to discuss the electrochemical detection performance of modified electrodes for AA/DA/UA in human physiological situations. Therefore, pH 7.4 was chosen as the detection solution condition in this work. The pH of the solution was regulated by PBS. GCE or  $Ti_3C_2T_x/GCE$  or  $Ti_3C_2T_x@TiO_2$  NSs/GCE served as the working electrode. Ag/AgCl was used as the reference electrode. Platinum (Pt) served as the counter electrode. The electrochemical detection capability was evaluated using differential pulse voltammetry (DPV). Furthermore, the time-amperometric method was used to assess the ability to resist interferences. The interferents included NaCl, KCl, K<sub>2</sub>SO<sub>4</sub>, Na<sub>2</sub>SO<sub>4</sub>, glucose, glutamate, glycine and lysine.

## 4. Conclusions

In this work,  $Ti_3C_2T_x$  coated with TiO<sub>2</sub> nanosheets was synthesized using a simple solvothermal process. The  $Ti_3C_2T_x@TiO_2$  NSs-modified GCE has been proven to enable the simultaneous detection of AA at concentrations ranging from 200 to 1000  $\mu$ M, DA at concentrations ranging from 40 to 300  $\mu$ M, and UA at concentrations ranging from 50 to 400  $\mu$ M. The LOD achieved for these analytes is 2.91  $\mu$ M (AA), 0.19  $\mu$ M (DA) and 0.25  $\mu$ M (UA). Furthermore, the  $Ti_3C_2T_x@TiO_2/GCE$  exhibits excellent resistance to interference, good repeatability and high selectivity. Two key advantages of the  $Ti_3C_2T_x@TiO_2$  are demonstrated to enhance their capability for simultaneous detection. Firstly, the substitution of -F ions with additional  $Ti^{3+}$  ions leading to a negative to neutral surface charge

change of  $Ti_3C_2T_x$ , thereby reducing electrostatic repulsion. Secondly, the growth of a mass of  $TiO_2$  nanosheets on the surface of  $Ti_3C_2T_x$  significantly increases the specific surface area and charge transport properties of the composite material. This work provides a novel approach for AA, DA and UA recognition in MXene-based composites. In future electrochemical detection, researchers may utilize targeted nanomaterial modifications between MXene lamellae to achieve precise detection of various analytes. This approach not only mitigates the self-accumulation of MXene but also increase the specific active sites, which is conducive to improving the selectivity and sensitivity of MXene-based materials.

**Supplementary Materials:** The following supporting information can be downloaded at: https: //www.mdpi.com/article/10.3390/molecules29122915/s1. Figure S1. XRD patterns of Ti<sub>3</sub>AlC<sub>2</sub>, Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> and Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>@TiO<sub>2</sub> NSs. Figure S2. SEM images of (a) Ti<sub>3</sub>AlC<sub>2</sub>, (b) Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>. Figure S3. (a) DPV results in the presence of 1 mM AA at different concentrations of DA (80–500  $\mu$ M) (b) the relationship between the corresponding oxidation peak current value and concentration of DA (c) DPV results in the presence of 20  $\mu$ M DA at different concentrations of AA (150–400  $\mu$ M), (d) the relationship between the corresponding oxidation peak current value and concentration of AA. Figure S4. (a) DPV results in the presence of 150  $\mu$ M UA at different concentrations of DA (50–400  $\mu$ M) (b) the relationship between the corresponding oxidation peak current value and concentration of DA (c) DPV results in the presence of 100  $\mu$ M DA at different concentrations of UA (150–400  $\mu$ M), (d) the relationship between the corresponding oxidation peak current value and concentration of DA. Figure S5. (a) DPV results in the presence of 300  $\mu$ M DA at different concentrations of LA (200–700  $\mu$ M) and UA (100–350  $\mu$ M), (b) DPV results in the presence of 500  $\mu$ M AA at different concentrations of UA (40–100  $\mu$ M), (c) DPV results in the presence of 100  $\mu$ M UA at different concentrations of AA (40–500  $\mu$ M).

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