

Review

Conductive Polymer-Based Electrodes and Supercapacitors: Materials, Electrolytes, and Characterizations

Zahra Roohi 1,2 [,](https://orcid.org/0000-0001-7867-5102) Frej Mighri [1](https://orcid.org/0000-0002-0321-7606) and Ze Zhang 2,3,[*](https://orcid.org/0000-0003-4331-2058)

- ¹ Department of Chemical Engineering, Faculty of Sciences and Engineering, Université Laval, Quebec, QC G1V 0A6, Canada; zahra.roohi.1@ulaval.ca (Z.R.); frej.mighri@gch.ulaval.ca (F.M.)
- ² Division of Regenerative Medicine, Saint-François d'Assise Hospital, Research Center of CHU de Québec—Université Laval, Quebec, QC G1L 3L5, Canada
- ³ Department of Surgery, Faculty of Medicine, Université Laval, Quebec, QC G1V 0A6, Canada
- ***** Correspondence: ze.zhang@fmed.ulaval.ca

Abstract: New materials and the interactions between them are the basis of novel energy storage devices such as supercapacitors and batteries. In recent years, because of the increasing demand for electricity as an energy source, the development of new energy storage materials is among the most actively studied topics. Conductive polymers (CPs), because of their intrinsic electrochemical activity and electrical conductivity, have also been intensively explored. While most of the high capacitance reported in the literature comes from hybrid materials, for example, conductive polymers composed of metal oxides and carbon materials, such as graphene and carbon nanotubes, new chemistry and the 3D structure of conductive polymers remain critical. This comprehensive review focuses on the basic properties of three popular conductive polymers and their composites with carbon materials and metal oxides that have been actively explored as energy storage materials, i.e., polypyrrole (PPy), polyaniline (PANi), and polythiophene (PTh), and various types of electrolytes, including aqueous, organic, quasi-solid, and self-healing electrolytes. Important experimental parameters affecting material property and morphology are also discussed. Electrochemical and analytical techniques frequently employed in material and supercapacitor research are presented. In particular, cyclic voltammetry (CV) and electrochemical impedance spectroscopy (EIS) are discussed in detail, including how to extract data from spectra to calculate key parameters. Pros and cons of CP-based supercapacitors are discussed together with their potential applications.

Keywords: energy storage; conductive polymer; supercapacitor; electrochemistry; polypyrrole

1. Introduction

Global economic development and population growth increase fossil fuel consumption and pollution, forcing us to search for clean energy sources and new energy conversion and storage methods. Electrochemical energy is an essential part of the clean energy portfolio.

The most common electrochemical energy storage systems are batteries, fuel cells, and electrochemical supercapacitors (ESCs or SCs). Among these systems, SCs are characterized by their high power density, long life cycle, and fast charging.

The capacitance of supercapacitors can reach thousands of Farads (F). Unlike batteries, charge storage in supercapacitors is controlled by surface reaction instead of ion diffusion in the material. Therefore, supercapacitors are better than batteries in terms of power density in the same volume. Supercapacitors can be used in smart applications such as wearable devices, sensors, and portable energy storage systems. A higher specific surface area in supercapacitors can provide not only higher capacitance but also higher energy density [\[1\]](#page-27-0). While the capacitance of conventional capacitors is in the range of micro- and milli-Farads (μ F and mF), in supercapacitors it can reach thousands of Farads. Figure [1](#page-1-0) shows the history of supercapacitor development and the revolution of active materials [\[2\]](#page-27-1).

Citation: Roohi, Z.; Mighri, F.; Zhang, Z. Conductive Polymer-Based Electrodes and Supercapacitors: Materials, Electrolytes, and Characterizations. *Materials* **2024**, *17*, 4126. [https://doi.org/10.3390/](https://doi.org/10.3390/ma17164126) [ma17164126](https://doi.org/10.3390/ma17164126)

Academic Editor: Federico Bella

Received: 21 June 2024 Revised: 31 July 2024 Accepted: 9 August 2024 Published: 20 August 2024

Copyright: © 2024 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license [\(https://](https://creativecommons.org/licenses/by/4.0/) [creativecommons.org/licenses/by/](https://creativecommons.org/licenses/by/4.0/) $4.0/$).

SCs provide an excellent balance between power density and energy density by SCs provide an excellent balance between power density and energy density by bridging the gap between batteries that have high energy density and traditional capacitors that have high power density $[3-5]$ $[3-5]$. They can be used as an energy storage source in outlying places where there is no public energy network because wiring costs are too high, or in places where wiring is dangerous. They can also be used as a power supply in portable

and in provided as a power supply in portable devices such as cellphones, notebooks, and electric or hybrid vehicles [\[6\]](#page-27-4). The Ragone chart in Figure [2](#page-1-1) shows the difference between the level of energy density and power density of sity of different energy storage devices [7]. different energy storage devices [\[7\]](#page-27-5). sity of different energy storage devices [7]. SCs provide an excellent balance between power density and energy density by between batteries that the gap between batteries that the gap that have help that have have have have the gap traditional capacity and the gap of the gap of

from Ref. [7] with permission from Elsevier. **Figure 2.** Ragone chart: Power density vs. energy density of energy storage devices. Reproduced **Figure 2.** Ragone chart: Power density vs. energy density of energy storage devices. Reproduced

energy in its structure. SCs have been developed since the 1950s of the 20th century. The first SC had a capacity of around 1 F, which was patented in 1971 by Standard Oil of Ohio (SOHIO). The first commercial product was marketed by Panasonic in 1982 and was An SC, also called an ultracapacitor, is an electrical component that can store electrical named Gold Cap [\[7\]](#page-27-5). The Gold Cap had high equivalent series resistance (ESR). In 1992, a commercial SC, called electrochemical double-layer capacitor, or EDLC at the time, with low ESR, was introduced to the market by Maxwell Laboratories, with the brand Boost Cap and a nominal capacity of about 1 kF [\[7\]](#page-27-5). Today's SCs have capacities higher than several thousand F and are used in devices that need to release a huge amount of energy in a short
electronic devices time. They can also be used in electric and hybrid vehicles, fuel cell vehicles, electronic devices, energy harvesting systems, solar cells, and wind turbines [\[8\]](#page-27-6).
The commonly used water-solution is one of the commonly used water-solution is one of the common variable poly several thousand F and are used in devices that need to release a huge amount of energy in a nominal capacity of about 1 kF [7]. Today s SCs have capacities higher than several

The selection of electrode material and appropriate electrolytes, such as polyvinyl
 $\frac{1}{2}$ alcohol (PVA), which is one of the commonly used water-soluble polymeric electrolytes, alcohol (FVA), which is one of the commonly used water-soluble polymeric electrolytes,
should follow some criteria [\[9](#page-27-7)[,10\]](#page-27-8). Compared with aqueous electrolytes, organic and ionic gel electrolytes can increase the working potential of supercapacitors because of their ger electrolytes can include the working potential or supercupacitors because of their wide potential window [\[11](#page-27-9)[,12\]](#page-27-10). The critical factors that have to be taken into consideration include operating potential, chemical stability, ionic conductivity, solubility, working temperature window, and viscosity, as well as mechanical properties [\[13\]](#page-27-11). m _{udd} m _b m ₁, m ₁. The critical m ₁, m ₁. The compared with aqueous electrolytes, organic and this

Enforcement which which is another factor that should be taken into consideration. This factor is expect stating is another ratter that should be taken the exhibiteration. This factor is related to the energy storage mechanism, which can be categorized in three different groups. The first group is the EDLC type, such as carbon materials, in which the charge/discharge process is based on physical absorption of ions at the interface of the electrode and electrolyte. trolyte. Stability of these kinds of materials can decrease significantly during long cycles. The second group has battery-type and pseudocapacitive behaviors, such as metal oxides, metal carbides, and metal nitrides, which is based on self-activation reactions. During a self-activation process, specific capacitance increases with respect to the initial value, but drops to the initial value or less than 100% after finishing the self-activation step. The third group belongs to conductive polymers, of which the energy storage mechanism is based on reversible redox reactions, i.e., the doping-dedoping of p-type or n-type counter ions that produce Faradic current. But because of the volume expansion/contraction during charge/discharge cycles, capacity retention is less than 100% for this group of materials [\[14\]](#page-27-12). Figure 3 shows a brief classification of di[ffe](#page-2-0)rent factors that can affect cycle stability.

Figure 3. Factors affecting cyclic stability of supercapacitors. Reproduced from Ref. [\[14\]](#page-27-12) with mission from the Royal Society of Chemistry. permission from the Royal Society of Chemistry.

Conducting polymers with conjugated structures, such as polypyrrole (PPy), poly-Conducting polymers with conjugated structures, such as polypyrrole (PPy), polyaniline (PANi), and polythiophene (PTh) and their derivatives, are interesting materials to be used as active materials in supercapacitors because of their unique electrochemical properties. They can also be composed with other electrode materials, such as graphene, properties. They can also be composed with other electrode materials, such as graphene, metal oxides, and carbon nanotubes (CNT), to adjust their properties to achieve special metal oxides, and carbon nanotubes (CNT), to adjust their properties to achieve special functionalities [\[15](#page-27-13)[–17](#page-28-0)]. functionalities [15–17].

Other than electrolytes and electrode materials, the structural design of an electrode also plays a key role in supercapacitor performance. For example, a 3D structure increases the surface area and can facilitate the charge and discharge process [\[18\]](#page-28-1). The substrate on which the electrode material is loaded also plays an important role. It can improve flexibility, stretchability, self-healing ability, and waterproof functions of supercapacitors [\[19](#page-28-2)[–21\]](#page-28-3).

To design a high-performance supercapacitor, it is critical to select the electrode material, electrolyte, and structure according to its application. Current research shows that the available supercapacitors are still not satisfactory in large-scale applications. There are important issues to be resolved, such as energy density, power density, durability, and good mechanical properties [\[13\]](#page-27-11). Study of the electrochemical process between electrolyte and electrode material, electrical double-layer configuration, charge–discharge, mass transfer mechanism, and all electrochemical reactions related to pseudocapacitance is of significance to establish the theoretical foundation of polymer-based supercapacitors [\[22](#page-28-4)[–24\]](#page-28-5).

Despite many achievements in this field, the research about supercapacitors, especially flexible supercapacitors, needs to be improved for large-scale industrial applications. Developing novel materials with high electrochemical capacitance, long durability, good mechanical properties, and fast charge/discharge ability is essential in the case of flexible supercapacitors. Based on updated information, this review provides the state of the art of conductive polymer-based electrode materials and supercapacitors, focusing on material chemistry, fundamentals governing conductive polymer conductivity, critical factors affecting capacitance, typical characterization methods, and how to extract data to calculate typical performance parameters describing electrodes and supercapacitors. It provides researchers in the field with updated information and furnishes beginners with a relatively comprehensive background and tools.

2. Supercapacitors Based on Conducting Polymers

Conducting polymers (CPs) are exciting candidates for supercapacitor electrodes. PPy, PANI, and PTh are the main CPs that have been heavily investigated in recent years [\[25](#page-28-6)[,26\]](#page-28-7).

A supercapacitor consists of two electrodes, a separator, and electrolyte. The charge can be stored at the interface of electrode and electrolyte in the form of an electrical double layer, and its capacitance can be calculated from Equation (1):

$$
C = \frac{A\varepsilon}{d} \tag{1}
$$

where:

C: capacitance between two electrodes;

A: area of the two surfaces of the electrodes that face each other;

ε: dielectric constant of the electrolyte; and

d: separation distance between two electrodes.

Energy density, power density, and cycle life are the critical parameters to evaluate the performance of a supercapacitor, which can be calculated from the below equations. Based on Equation (2), two different types of capacitance in their structure (C_p and C_n) are connected in series and make the total capacitance (C_T) [\[27\]](#page-28-8).

Total capacitance:

$$
\frac{1}{C_T} = \frac{1}{C_p} + \frac{1}{C_n} \tag{2}
$$

where:

Cp: positive electrode/electrolyte; and Cn: negative electrode/electrolyte.

Actually, capacitance shows the amount of charge in the presence of electrical potential, which depends on the amount and the surface area of the active material. Capacitance can be calculated based on weight (C_m, F/g), area (C_s, F/cm²), or volume (C_v, F/cm³) [\[22\]](#page-28-4), as shown in Equations (3)–(5).

$$
C_v = \frac{1}{\theta r \Delta V} \int_{V_0}^{V_0 + \Delta V} I dV = \frac{I \Delta t}{\theta \Delta V}
$$
(3)

$$
C_m = \frac{1}{mr\Delta V} \int_{V_0}^{V_0 + \Delta V} I dV = \frac{I\Delta t}{m\Delta V}
$$
(4)

$$
C_s = \frac{1}{sr\Delta V} \int_{V_0}^{V_0 + \Delta V} I dV = \frac{I\Delta t}{s\Delta V}
$$
(5)

where r: scan rate (V/s) ;

 V_0 : lower potential limit (V); ∆V: potential window (V); I: current (A); and ∆t: discharge time (s).

Current and test duration must be normalized by weight, volume, or area.

Other parameters that are very important in supercapacitors are output power and stored energy, which are shown in the Equations (6) and (7) [\[22\]](#page-28-4):

$$
E_X = \frac{1}{2} C_X \Delta V^2 \tag{6}
$$

$$
P_X = \frac{1}{4R} \Delta V^2 = \frac{E_X}{\Delta t}
$$
 (7)

 $X = m$ for mass, s for area, or v for volume

2.1. Electric Double-Layer Capacitors or EDLC Supercapacitors

EDLCs are the most common supercapacitors in the marketplace. The charge transfer mechanism is based on electrostatic interactions. In these types of supercapacitors, a layer called a Helmholtz double layer is formed at the interface between electrode and electrolyte, where energy (charge) is electrostatically stored, as illustrated in Figure [4.](#page-4-0) Because there is no electron exchange and redox reaction between electrode and electrolyte, the current discharged by EDLCs is non-faradaic. The critical parameters that can affect EDCL capacitance are electrode surface area and Helmholtz layer thickness. The most used electrode material is activated carbon (AC) because of its large surface area and low price. The dielectrics used in EDCLs can be liquid electrolytes, and there is not any electrochemical reaction on the electrode during the charging/discharging process [\[6\]](#page-27-4).

Inner Helmholtz layer

 \mathbf{F} **Figure 4. Figure 1. H Figure 4.** Helmholtz double layer in ELDC supercapacitors.

2.2. Pseudo-Supercapacitors or Faradaic Supercapacitors

In these types of supercapacitors, a reversible faradaic process of redox reaction, or reversible electrochemical doping-dedoping, and reversible adsorption occur on the or reversible electrochemical doping-dedoping, and reversible adsorption occur on the electrode surface. Reversible adsorption of hydrogen can occur on the surface of a gold or platinum electrode. A reversible redox reaction can occur on the surface of metal oxide electrodes, such as $RuO₂$, $MnO₂$, $CoO₃$, NiO, and Fe₃O₄. And reversible electrochemical doping-dedoping can occur on the surface of conductive polymer electrodes, such as PPy, PANi, and PTh. All of these processes generate faradaic current [\[27](#page-28-8)[–29\]](#page-28-9). or platinum electrode. A reversible redox reaction can occur on the surface of metal oxide
electrodes, such as RuO₂, MnO₂, CoO₃, NiO, and Fe₃O₄. And reversible electrochemical
doping–dedoping can occur on the sur

With the electrochemical or faradaic process, working voltage is extended and specific capacitance is increased [\[30\]](#page-28-10). Conway et al. have shown that a faradaic supercapacitor can have a capacitance of 10 - 100 times higher than electrostatic supercapacitance [\[31\]](#page-28-11).

As mentioned before, in these types of supercapacitors, the charging/discharging process occurs on the electrode surface in the form of both reversible redox reactions and an electric double layer. Because of the doping and dedoping process during redox reactions, the electrode is stressed and degrades faster than electrostatic supercapacitors. The difference in charge/discharge mechanisms between electrostatic supercapacitance and pseudocapacitance is shown in Equations (8) and (9). Actually, in an electrostatic capacitor (Equation (8)) [\[23\]](#page-28-12), there is no charge transfer and ion exchange across the electrode and electrolyte interface, while in pseudocapacitors (Equation (9)), a reversible chemical reaction occurs at the electrode surface.

$$
E_p + E_n + A^- + C^+ \xrightarrow{\text{Charging}} E_p // A^- + E_n // C^+ \tag{8}
$$

$$
\left\{ \left[X\right]_{n}^{0} \right\}_{m} + A^{-} \xleftarrow{\text{Charging}(-e^{-})} \left\{ \left[X\right]_{n}^{+} A^{-} \right\}_{m} \tag{9}
$$

2.3. Hybrid Supercapacitors

By combining pseudocapacitance and electrostatic capacitance, another type of supercapacitor can be created, which are called hybrid supercapacitors. With this combination, one can increase both volumetric and gravimetric energy densities. Consequently, a higher amount of current can be produced. Additionally, due to the faradaic reaction from the pseudocapacitance part, energy density is also increased. Right now, hybrid supercapacitors still need to be improved and are not commercially available [\[32\]](#page-28-13). Table [1](#page-5-0) compares the characteristics of these three types of supercapacitors and lithium-ion batteries.

Table 1. Important characteristics of different electrochemical energy storage systems [\[7](#page-27-5)[,33,](#page-28-14)[34\]](#page-28-15).

3. Electrode Material

To design a high-performance supercapacitor, choosing the right active material is very important [\[35\]](#page-28-16). Selection of polymers, polymer chain modification, and polymer blending with other materials are important techniques that are essential to create new polymer structures or compositions based on their final usage. These modifications can improve mechanical, electrical, and electrochemical properties [\[36\]](#page-28-17).

The electrode materials chosen for supercapacitors must have critical characteristics, such as high conductivity, high electrical capacity, good mechanical property, high corrosion resistivity, good chemical stability, being environmentally friendly, and low cost. Among the various materials, conducting polymers have attracted significant attention because of their unique and fascinating characteristics.

3.1. Conducting Polymers (CPs)

CPs are organic materials with a conjugated band system that can conduct electricity without any conductive filler. The most common CPs include PPy, PANi, and PTh. CPs can be synthesized through chemical polymerization in the presence of oxidant and dopant or through electrochemical polymerization in the presence of dopants and monomers in a specific solvent [\[37\]](#page-28-18). To increase the conductivity and electrochemical performance of CPs, they have to be doped with appropriate dopants [\[38\]](#page-28-19). The doped CPs can be n-type or p-type, and the charge/discharge process occurs simultaneously with the doping and dedoping processes. Equations (10) and (11) show the charging process in the n-doping and p-doping states, respectively [\[38\]](#page-28-19). The discharge process is the reverse of these equations.

$$
C_p \rightarrow C_p^{n+}(A^-)_n + ne^- (p \, doping) \tag{10}
$$

$$
C_p + ne^- \rightarrow (C^+)_n C_p^{n-} \ (n \ doping) \tag{11}
$$

The polymer's nature determines the type of dopant. PPy and PANi cannot be ndoped because of inappropriate potential, and they can only be p-doped. However, PTh is p- and n-dopable. Generally, the specific capacitance of n-doped PTh is lower than that of the p-doped form. In low potential (less than −2.0 V vs. Ag/AgCl), PTh is n-doped. However, n-doped PTh is unstable and can't be used in practice, even its conductivity is comparable with that of the p-doped form [\[38\]](#page-28-19). Therefore, choosing the right material for a specific application is very important [\[38\]](#page-28-19).

The conductivity of these polymers can be tuned from 10^{-10} up to 10^4 S/cm with different dopants or doping levels [\[39](#page-28-20)[–41\]](#page-28-21). Based on their electrical conductivity and redox states, they can be used as the active materials of pseudocapacitors. They can have a large specific capacitance, such as 1284 F/g for PANI [\[42\]](#page-28-22), 480 F/g for PPy [\[43\]](#page-28-23), and 210 F/g, for PEDOT [\[44\]](#page-28-24). Table [2](#page-7-0) shows a list of typical conducting polymers with their chemical structures [\[45\]](#page-29-0). *Materials* **2024**, *17*, x FOR PEER REVIEW 8 of 36

Table 2. Typical conducting polymers and structures. *Materials* **2024**, *17*, x FOR PEER REVIEW 8 of 36

Service Contractor

Service Construction

Service Contractor

S

<u>Service Communication</u>

<u>Service Communication</u>

Service Control Control

<u>Service Communication</u>

A-

A-

+

 $+$

Service Construction

+ A-

Service Construction

Table 2. *Cont.* Poly(3,4-ethylene dioxythiophene) (PEDOT)

3.1.1. Polypyrrole

Among CPs, PPy is very attractive because of its balanced properties, such as thermal stability, high conductivity, environmental stability, and ease of synthesis [\[40\]](#page-28-25). PPy is readily synthesized on various substrates of various shapes to provide a high electrochemical performance [\[46\]](#page-29-1). The charge storage capability of PPy can be increased by improving the ion diffusion rate and contact surface area [\[47\]](#page-29-2). Based on these unique advantages, PPy has been considered a promising candidate to fabricate flexible and light-weight supercapacitors. However, the poor cyclic stability, low electrical conductivity of PPy composites, and the poor mechanical properties of pure PPy are still significant challenges [\[48\]](#page-29-3).

To overcome such problems, many different techniques have been designed, for example, PPy with designed morphology or PPy composed with carbon materials or metal oxides. The electrical capacitance and cyclic stability of PPy must be improved before it can be used in energy storage devices.

Polymerization conditions, such as the type of polymerization, temperature, potential range, oxidant, solvent, etc., are very important. The ideal polymerization of pyrrole occurs when the α carbons (Figure [5\)](#page-8-0) are oxidized, either by chemical oxidants or through an oxidative electrical potential, into cation radicals. These cation radicals then join one with the other and deprotonate to form a dimer. The dimers will be oxidized again to form cation radicals and join to form oli[gom](#page-29-4)ers [49], and so on. However, PPy can have a high degree of cr[ossli](#page-29-5)[nki](#page-29-6)ng [50–52]. As show[n i](#page-8-0)n Figure 5, there are two reaction sites, α (positions 2 and 5) and β (positions 3 and 4), on a pyrrole ring. Chain propagation at α sites leads to a regular and linear structure, while propagation involving β sites leads to irregular structures and crosslinks. Since high temperatures can increase the probability of the reactions at the β position, if a highly ordered structure of PPy is desired, a low reaction temperature is preferred to reduce the degree [of c](#page-29-7)rosslinking [53].

Polymerization conditions, such as the type of polymerization, temperature, poten-

Figure 5. (a) Pyrrole and the possible structures of pyrrole dimers, (b) α - α and α - β couplings leading to chain branching and crosslinking in PPy structure. Figure [5a](#page-8-0) is reproduced from Ref. [\[53\]](#page-29-7) with to chain branching and crosslinking in PPy structure. Figure 5a is reproduced from Ref. [53] with permission from Elsevier. permission from Elsevier.

PPy has different oxidation states. As shown in Figure 6, [it c](#page-8-1)an switch reversibly tween the first three states during the redox process. There is no change in covalent bond between the first three states during the redox process. There is no change in covalent bond in the first three states. The last state, which is called the overoxidized state, is irreversible. in the first three states. The last state, which is called the overoxidized state, is irreversible. The irreversible state occurs because one or more oxygen atoms are covalently attached to The irreversible state occurs because one or more oxygen atoms are covalently attached to the pyrrole ring, leading to the loss of the structural conjugation and hence the decrease the pyrrole ring, leading to the loss of the structural conjugation and hence the decrease in conductivity. In fact, undoped PPy can easily react with atmospheric oxygen and become overoxidized.

Polymers are insulators in general. However, in the case of conducting polymers, Polymers are insulators in general. However, in the case of conducting polymers, when they are doped with the proper counterions, they become electrically conductive $[54,55]$ $[54,55]$. When doped, PPy obtains a dynamic electrical structure that can transport electricity from one side to another with the help of polarons or bipolarons $[56]$.

Next, the critical parameters that affect the final conductivity of the CPs will be cussed in detail. These parameters include the experimental conditions, such as the discussed in detail. These parameters include the experimental conditions, such as the polymerization method, polymerization temperature, electrode material in case of electrochemical polymerization, solvent, counterion/electrolyte, monomer substitutions, oxidant

type, nature of the counterions, molar ratio of dopant/monomer, and application temperature of the CPs [\[57\]](#page-29-11).

Dian and Lacroix have shown that during polymerization, nucleophiles can decrease conductivity and mechanical properties. Water molecules can act as nucleophiles and attack the pyrrole ring to form carboxyl groups, breaking up the conjugation of polymer chains [\[58\]](#page-29-12).

Counterions also have a dramatic effect on conductivity [\[59\]](#page-29-13). The presence of bulky functional groups on the pyrrole ring can decrease the degree of conjugation and, consequently, conductivity. To avoid such negative effects, it is needed that these groups be separated at least by four carbons in the polymeric chain, and it is better to use a flexible chain as a linker between the pyrrole ring and the functional group [\[60\]](#page-29-14). Based on some studies, thermal conductivity can be increased by stretching the polymer because of polymer chain alignment [\[61\]](#page-29-15). Stretching makes polymer chains aligned in the stretch direction. As a result, thermal transfer becomes more efficient along the oriented polymer chains (Figure [7\)](#page-9-0) [\[62\]](#page-29-16).

Figure 7. Stretch effect on polymeric chain, showing: (a) 3D unit cell of the polymer in unstretched state, (b) projection of the unstretched cell to XY plane, (c) stretched sample. Colors represent different ferent polymer chains. Reproduced from Ref. [62] with the permission of American Physical Society. polymer chains. Reproduced from Ref. [\[62\]](#page-29-16) with the permission of American Physical Society.

 S_{y} interests the conductivity of P₂, B_{y} reducing the temperature, S_{y} the conductivity of PPy increases. This increment is because of reducing the number of side
reactions leading to assessed ation and non-planar structures [61,62] Synthesis temperature affects the conductivity of PPy. By reducing the temperature, reactions leading to overoxidation and non-planar structures [\[61](#page-29-15)[,63\]](#page-29-17).

side reactions leading to overoxidation and non-planar structures [61,63]. Doping level directly affects conductivity. By increasing the doping level, conductivity Doping level directly affects conductivity. By increasing the doping level, conductiv-can increase as high as 15 orders of magnitude [\[64\]](#page-29-18). A dopant can enter the PPy structure during or after polymerization to create a conjugation defect along the polymeric backbone. ture during or after polymerization to create a conjugation defect along the polymeric This defect is involved in the double-bond rearrangement of the conjugated system by forming a polaron or a bipolaron, which is the charge carrier that transfers the charge along the chain [\[65,](#page-29-19)[66\]](#page-29-20). The conductivity normally increases with an increase in doping level that is affected by the dopant/monomer molar ratio and the nature of the dopant. Dopant can α be either added into polymerization solution or generated by the oxidant. For example be either added into polymerization solution or generated by the oxidant. For example,

by using ferric chloride (FeCl₃) as the oxidant, it forms a donor–acceptor complex in the conjugated system and leaves a chlorine anion as a counterion.

The polarons act as charge carriers and are formed at a low concentration of the dopant. The mobility of the polarons is not high, and consequently, the conductivity of the PPy is relatively low as well. By increasing the amount of dopant, more Cl[−] anions are inserted into pyrrole rings to create more polarons. When the polymer chain becomes crowded with polarons, they form bipolarons that have a higher energy level. This process increases the number of charge carriers and also their mobility, leading to an increase in conductivity [\[67\]](#page-29-21).

Application temperature also affects conductivity. It has been shown that a hightemperature application environment can increase the conductivity of conductive polymers. It was found that at low temperatures, charge carriers are electrostatically bound to the counterions, such as Cl[−] anions, and therefore, they are not expected to move fast. However, at higher temperatures, the charge carriers have enough energy to overcome the electrostatic attachment to the counterions, and consequently, they can easily move, leading to higher attachment to the counterions, and consequently, they can easily move, leading to higher conductivity [\[68\]](#page-29-22). Electron spin resonance (ESR) studies show that by increasing the temperature in doped PPy, the g-value of the polymer increases linearly from 1.9504 to
temperature in doped PPy, the g-value of the polymer increases linearly from 1.9504 to 1.9925 in a temperature range of $30-200\degree C$ [\[69\]](#page-29-23). The g-value is a factor that has direct relation to the specific energy of a particle; this value in a free electron is 2.0023, and the relation to the specific energy of a particle; this value in a free electron is 2.0023, and the g-values of free radicals are very close to this value [\[69\]](#page-29-23). the g-values of free radicals are very close to this value [69]. 1.9925 to 1.9926 1.1 y, the g-value of 1.90 $\%$ [69]. The g-value incluses incluse it and 1.904 that 1.904 that 1.904 that

Polarons and bipolarons are a kind of self-localized defection associated with their Polarons and bipolarons are a kind of self-localized defection associated with their quantum state in an energy gap. The quantum state and energy gap of these two particles quantum state in an energy gap. The quantum state and energy gap of these two particles are different in terms of spin and charge. A polaron has a spin of $\pm 1/2$ and an electric charge of $\pm e$; however, a bipolaron is spinless and its electric charge is $\pm 2e$ [\[70](#page-29-24)[,71\]](#page-29-25).

Transition from the polaron level $(t < 80 \text{ fs})$ to the bipolaron level $(t > 100 \text{ fs})$ occurs after the adiabatic removing of an electron. Figure [8](#page-10-0) shows the time evolution of this transition by decreasing energy levels and energy gap. The small oscillation of the energy levels is because of lattice oscillation by hole perturbation [\[70,](#page-29-24)[72\]](#page-29-26).

Figure 8. Time evolution of the energy level inside and around gap relating to an adiabatic transition **Figure 8.** Time evolution of the energy level inside and around gap relating to an adiabatic transition of polaron levels (t < 80 fs) into bipolaron levels (t > 100 fs). Reproduced from [72] with permission of polaron levels (t < 80 fs) into bipolaron levels (t > 100 fs). Reproduced from [\[72\]](#page-29-26) with permission from Springer Nature. from Springer Nature.

The separation of energy bands determines what is metal, semiconductor, or insulator, as illustrated in Figure [9.](#page-11-0) In metals, the band gap between the valance band (VB) and conducting band (CB) is zero (Eg = 0 Ev); that is why electrons can move freely in metals. However, in insulators, this gap is larger than 3.16 eV, so large that the electrons do not However, in insulators, this gap is larger than 3.16 eV, so large that the electrons do not have enough energy to jump from VB to CB. In semiconductors, the gap between VB and have enough energy to jump from VB to CB. In semiconductors, the gap between VB and CB is decreased by a treatment known as doping that generates electrons and holes as CB is decreased by a treatment known as doping that generates electrons and holes as

charge carriers in inorganic semiconductors and polarons and bipolarons as charge carriers in CPs [\[73–](#page-29-27)[75\]](#page-30-0).

Figure 9. Electronic bands illustrating (a) undoped; (b) polaron; (c) bipolaron; and (d) fully doped states of PPy. Reproduced from Ref. [74], CC-BY 3.0*.* states of PPy. Reproduced from Ref. [\[74\]](#page-30-1), CC-BY 3.0.

3.1.2. Polyaniline Polyaniline, or Pani, also called aniline black, is another CP that has been used in the has bee

Polyaniline, or PANi, also called aniline black, is another CP that has been used in energy storage applications. It can be synthesized by chemical and electrochemical polymerization methods. Its light weight, high conductivity, low cost, and good mechanical properties make PANi an interesting material to be used in supercapacitors. Figure [10](#page-11-1) shows the chemical structure of PANi.

Figure 10. Polyaniline chemical structure*.* **Figure 10.** Polyaniline chemical structure.

PANi can be deposited on the surface of metal oxides (MOs)/carbon to form a the redox process, it can be used in electrochromic supercapacitors. Comparing the three popular CPs, PPy and PTh are more stable than PANi and can be synthesized directly in doped form [27]. According to the studies on the PANi synthesized under different conditions, only the one that has the potential window of $0.8-1.0$ V can be used in supercapacitors, and any PANi with a potential window of less than 0.6 V is not in the supercapacitor class. This is because that energy density is proportional to the square of the cell voltage ($E = (CV^2)/2$), meaning that a low potential does not provide enough energy to a supercapacitor [\[77,](#page-30-3)[78\]](#page-30-4). PANi can be doped in n-type or p-type. Doping can be done during polymerization or after polymerization [\[79\]](#page-30-5). Some dopants in PANi can attach covalently to the polymeric chain and cannot be removed during the subsequent electrochemical redox process, which blocks a portion of the capacitance of PANi. Dedoping can work in favor of electroactivity because it can form structural micropores, consequently increasing capacitance. On the other hand, initial doping can affect the morphology of CPs, so that directly affects the overall performance of the supercapacitor [\[80](#page-30-6)-84]. PANi/MOs/nanocarbon ternary hybrid [\[76\]](#page-30-2). Because of the color change of PANi during

Morphology is one of the crucial factors that can affect the electrochemical behav-iors of PANi. Since the electrode-electrolyte interface area plays a significant role [\[85\]](#page-30-8), by increasing the surface porosity of the PANi electrode, the specific capacitance of the supercapacitor can be increased. For example, Sharma et al. synthesized a nanoporous

PANi with a specific capacitance of 410 F/g [\[86\]](#page-30-9). Another method by which we can increase the surface area and then the specific capacitance is to deposit PANi on a substrate of high surface area, such as a highly ordered metal oxide, porous carbon, or a template, such as a nanofiber template that can be removed after polymerization [\[87](#page-30-10)[–90\]](#page-30-11).

Nanostructured PANi can be easily synthesized by chemical and electrochemical methods. PANi has excellent electrochemical cyclability and reversibility, which make it appropriate to be used in a supercapacitor [\[79\]](#page-30-5). There are many parameters that can affect the electrochemical properties of PANi during both chemical and electrochemical polymerizations, such as polymer chain length, crosslink degree, types of dopant, doping level, morphology, and diffusion pathways. These characteristics can be controlled with different parameters, such as temperature, concentration of the solution, and dopant [\[91\]](#page-30-12). It was reported that a crosslinked PANi electrode retained 100% of its specific capacitance after 1000 cycles [\[92\]](#page-30-13).

There are generally three main types of PANi nanocomposites, i.e., carbon/PANi, metal oxide/PANi, and metal oxide/carbon/PANi, which have been investigated for use in pseudocapacitors.

In the case of carbon/PANi composite, carbon nanomaterials provide a high specific area, and PANi provides good electrical conductivity. PANi can be composited with carbon nanotubes, either in the form of single-walled (SWCNT) or multi-walled (MWCNT), to benefit from both the high surface area and high conductivity of the nanotubes [\[93–](#page-30-14)[95\]](#page-30-15). A disordered polymeric chain can limit diffusion of electroactive species and redox site accessibility. In order to solve these kinds of problems, it is better to use ordered mesoporous carbon as a template to synthesize PANi on the external surface of ordered mesoporous carbon, which can lead to a high surface area structure and consequently, high capacitance [\[96](#page-30-16)[,97\]](#page-30-17). The presence of appropriate functional groups at the surface of carbon nanotubes can form active sites for the growth and attachment of PANi, which is in favor of higher capacitance. Graphene with a 2D structure is an important type of carbon nanomaterial that has been widely investigated in PANi/graphene nanocomposites [\[98\]](#page-30-18). The nature of interactions between PANi and graphene affects the electrical capacity of the composite. When this interaction is non-covalent, such as Π- Π interaction (PANi and graphene have rich Π system), the cyclability of the supercapacitor drops by up to 45% after 1000 cycles [\[99\]](#page-30-19). PANi molecules can be deposited on the graphene sheet vertically with ordered alignment, leading to $1665 \frac{F}{g}$ specific capacitance [\[100\]](#page-30-20).

Metal oxides are also good candidates for supercapacitors because they show pseudocapacitance behavior in a wide potential range. Metal oxides such as $RuO₂$, $MnO₂$, $V₂O₅$, $Fe₂O₃$, NiO, MoO₃, and WO₃ can be used in PANi-based supercapacitors [\[79\]](#page-30-5). In another approach, metal oxides can be used as oxidizing agents to initialize the polymerization. With this method, PANi has been shown to grow with the metal oxide and form a nanocomposite [\[101\]](#page-30-21). In such composites, PANi acts as a conductive material that compensates for the low electrical conductivity of metal oxides. The best example is $MnO₂$, which has been used in batteries and supercapacitors and suffers from low conductivity. PANi effectively improved the electrochemical performance of the PANi/MnO₂ nanocomposite [\[102](#page-30-22)[,103\]](#page-30-23).

There are also ternary composites made of PANi, metal oxide, and carbon, of which, all the advantages of the components, such as high conductivity, high specific surface area and high accessible electrochemical sites, are combined to achieve a high performance [\[104](#page-31-0)[,105\]](#page-31-1). In these composites, metal oxides are attached to the graphene surface to form a good substrate, on which PANi can grow vertically. A research group has shown a specific capacitance of 1360 F/g with a graphene/ZrO₂/PANi system [\[106,](#page-31-2)[107\]](#page-31-3).

3.1.3. Polythiophene (PTh)

PTh is a conducting polymer with high environmental stability and tunable conductivity. PTh can be synthesized in the form of powder or film [\[108\]](#page-31-4). There are many different methods to synthesize PTh, including chemical and electrochemical [\[109\]](#page-31-5), photochemical [\[110\]](#page-31-6), ultrasonic-assisted, and template-assisted syntheses [\[110,](#page-31-6)[111\]](#page-31-7). PTh is synthesized from thiophenes, as showed by Figure [11.](#page-13-0)

Figure 11. Polythiophene-forming process*.* **Figure 11.** Polythiophene-forming process.

Normally, n is between 2 and 4, A^- is a counterion required to maintain the oxidation state, and m is the number of repeat units and is related to molecular weight. The electrochemical polymerization of PTh is a process considered a paradox, meaning that at the potential that thiophene needs to be oxidized, the synthesized polymeric chain can be overoxidized, changing the final chemical and physical properties. At a constant current or potential, the synthesized material is a mixture of the oxidized and overoxidized polymers. Some research groups have been able to reduce the overoxidation by using bithiophene or terthiophene instead of thiophene. However, the results have shown that the PTh that is synthesized with these initial dimers or trimers has lower conductivity [\[112–](#page-31-8)[114\]](#page-31-9). Fu et al. synthesized a PTh film with the specific capacitance of 110 F/g by the electrochemical polymerization method [\[115\]](#page-31-10). Senthil Kumar et al. reported a PTh film with a specific capacitance of 70 F/g by the chemical polymerization method [\[116\]](#page-31-11). Figure [12](#page-13-1) shows the 12 shows the reaction mechanism of PTh formation in chemical polymerization [117]. reaction mechanism of PTh formation in chemical polymerization [\[117\]](#page-31-12).

From John Wiloy & Sons $f(x)$ is the sons. **Figure 12.** Mechanism of polythiophene polymerization. Reproduced from [117] with permission **Figure 12.** Mechanism of polythiophene polymerization. Reproduced from [\[117\]](#page-31-12) with permission from John Wiley & Sons. from John Wiley & Sons.

relatively lower than that made of PANi and PPy [\[38\]](#page-28-19). PTh can also be n-doped or pdoped. Consequently, they can be used as the negative electrode material or the positive electrode material. Comparing n-doped and p-doped materials, the n-doped PTh has lower capacitance and conductivity than the p-doped ones [38]. [Th](#page-28-19)e n-doped PTh is produced at low potential and is sensitive to oxygen and water. Consequently, it can be easily oxidized and can severely self-discharge, which degrades the electrochemical performance of the elect[rode](#page-31-13) [\[11](#page-31-14)8,119]. In order to overcome this problem, an electron-withdrawing group can be added to the thiophene ring so that the n-doped PTh can be synthesized at less negative $potential [120].$ The specific capacitance of the electrode made of PTh derivatives (ex. PEDOT) is The specific capacitance of the electrode made of PTh derivatives (ex. PEDOT) is

Among the three CPs mentioned above, PPy often shows a high capacitance and cyclic stability in the literature, while PANi may show a high conductivity and PEDOT a high chemical stability. For example, flexible electrodes made of PPy/FeCl₃ nanorods coated on cotton fabrics recorded 578 F/g at 0.2 A/g [\[121\]](#page-31-16); PEDOT deposited on a carbon paper showed 126 F/g at 1mA/cm² [\[122\]](#page-31-17); and nanostructured PANi deposited on graphite electrodes demonstrated 460 F/g at zero scan rate (by interplotting) [\[123\]](#page-31-18). However, the

values in the literature often cannot be directly compared because of variations in other parameters, such as current density, scanning rate, area vs. weight, frequency, etc.

4. Electrode Arrangement: Two vs. Three Electrodes

Electrode arrangement plays a crucial role in the characterization of electrochemical devices. The choice between two-electrode or three-electrode systems depends on the purpose of the measurement.

4.1. Three-Electrode System

Three-electrode systems are commonly used in the research and development of electrode materials. The material of interest is used as the working electrode, together with a reference electrode, such as silver chloride $(Ag/AgCl)$, placed near the working electrode, and a counter- (also called an auxiliary) electrode that should have a sufficient surface area, such as a platinum mesh. A potentiostat ensures that most current passes through the working and counter electrodes, while the working potential is measured between the reference and working electrodes. This means that only a half cell (working electrode) is measured. This configuration can also accurately control the potential between reference and working electrodes. A three-electrode setup is often preferred during the initial testing and development stages of flexible electrodes [\[124\]](#page-31-19). This is because the precise control and accurate measurement of electrical potential provided by the three-electrode system are critical in understanding the electrochemical behaviors of the flexible materials [\[125\]](#page-31-20). Issues such as mechanical deformation, which can alter the electrochemical response, are more easily identified and mitigated using a three-electrode configuration [\[126\]](#page-31-21).

4.2. Two-Electrode System

In a two-electrode system, there are only working and counter electrodes. The current passes between the two, and the voltage drop is measured. One can combine the reference lead with the counter-electrode lead to convert a three-electrode system into a two-electrode system. The main advantage of a two-electrode system is its simplicity and ease of implementation. Obviously, what is studied is the full cell instead of a single electrode. That is why this configuration is often used to study the performance of devices such as supercapacitors and batteries. It is particularly useful for large-scale energy storage devices where the primary focus is on overall performance rather than detailed mechanistic studies. However, this simplicity comes with certain drawbacks. The two-electrode setup does not allow for independent control or monitoring of the potential at the working electrode, which can lead to inaccurate measurements due to the potential drop across the counter-electrode. This can particularly affect the performance evaluation of flexible electrodes, where uniform current distribution and minimized resistance are crucial [\[127\]](#page-31-22). Flexible electrodes are typically implemented in a two-electrode configuration for wearable and portable devices due to their simplicity and low profile [\[128\]](#page-31-23).

In conclusion, three-electrode systems offer precise control and measurement of electrochemical activities and are preferred in material research, and two-electrode systems measure entire electrochemical cells and are used to characterize full cell devices.

5. Electrolyte

The electrolyte contains salts and solvents and plays a key role in supercapacitor performance. Important parameters are ion type, size, and concentration; solvent type; interactions between solvent and ions; interactions between electrolyte and electrode; and electrolyte potential window. The electrolyte can affect power and energy density and self-discharge of supercapacitors. Figure [13](#page-15-0) below shows the electrolyte classification. Each electrolyte has some advantages and disadvantages. For example, aqueous electrolytes have high conductivity and high capacitance, but their working voltage is limited because of the relatively low decomposition potential of water (1.23 V) [\[129\]](#page-31-24). On the other hand, an organic electrolyte can be used at a higher potential, but they suffer from lower ionic

conductivity. Solid-state electrolytes don't have leakage problem, but sometimes they suffer from low ionic conductivity.

of the relatively low decomposition potential of water (1.23 V) [129]. On the other hand,

Figure 13. Classification of electrolytes for electrochemical supercapacitors. Reproduced from [13] **Figure 13.** Classification of electrolytes for electrochemical supercapacitors. Reproduced from [\[13\]](#page-27-11) with permission from Royal Society of Chemistry. with permission from Royal Society of Chemistry.

5.1. Liquid Electrolyte 5.1. Liquid Electrolyte

Liquid electrolyte is classified into two main groups, i.e., aqueous electrolyte and Liquid electrolyte is classified into two main groups, i.e., aqueous electrolyte and non-aqueous electrolyte. non-aqueous electrolyte.

Aqueous electrolytes consist of acidic, alkaline, and neutral water solutions. Because Aqueous electrolytes consist of acidic, alkaline, and neutral water solutions. Because they have narrow potential windows $(1.0-1.3 \text{ V})$ [\[129\]](#page-31-24), they are generally not used in commercial products but rather, in the research area, because they are inexpensive and easy to use without any special conditions. The important parameters that affect solvent conductivity and specific capacitance include ion (cation and anion) size in bare and hydrated states, and ion mobility. The potential window and corrosion degree of the electrolyte also have to be considered when choosing an appropriate electrolyte for a system. Table [3](#page-16-0) shows a list of aqueous electrolytes, together with their performance in by stem. There of shows a list of aqueous electrolytes, together while their performance in supercapacitors. The most frequently used acidic, alkaline, and neutral aqueous electrolytes H_1, H_2, H_3, H_4, H_5 and H_6 and H_7 respectively. The operation voltage of the operation v are made of H_2SO_4 , KOH, and Na₂SO₄, respectively. The operation voltage of these point for the considered as which decomposition voltage and which freezing and boiling points must be considered. One molar H_2 SO₄ is the most common acidic solution reported in the literature because, in this concentration, it has the maximum ion conductivity, which can increase the specific capacitance of supercapacitors. EDLCs, in the presence of H_2 SO₄, have higher specific capacitance than neutral aqueous electrolytes and organic electrolytes [\[130–](#page-31-25)[132\]](#page-31-26). electrolytes is about 1 V. Parameters such as water decomposition voltage and water

Generally, the specific capacitance and energy densities in EDLCs do not have a huge difference when using an alkaline (KOH) aqueous electrolyte or an acidic (H₂SO₄) electrolyte [\[132\]](#page-31-26). A KOH solution with a concentration of 6 M has high ionic conductivity and so has frequently been used among other alkaline solvents. Compared with alkaline and acidic solvents, neutral electrolytes have lower ionic conductivity, which causes lower specific capacitance, but on the other hand, they can provide a higher operating voltage [\[133](#page-32-0)[,134\]](#page-32-1). Table [3](#page-16-0) shows the performance of some aqueous electrolytes.

Electrode Type	Electrolyte	Cell Voltage (V)	Specific Capacitance (F/g)	Energy Density (W h/kg)	Power Density (W/kg)	Temp. $(^{\circ}C)$	Refs.
Mesoporous $MnO2$	1 M Li ₂ SO ₄	1	284.24 at 1 mV/s	28.8	70	RT	$[135]$
Mesoporous $MnO2$	1 M Na ₂ SO ₄	1	278.8 at $1 \,\mathrm{mV/s}$	28.4	70	RT	$[135]$
$MnO2$ nano flowers	1 M LiOH	0.6	363 at 2 mV/s		$\overline{}$	$\overline{}$	$[136]$, 2015
MnO ₂ Qcarbon nanofibers composites	0.5 M Na ₂ SO ₄	0.85	551 at 2 mV/s (75 1C)			$0 - 75$	$[137]$, 2013
AC	0.5 M Na ₂ SO ₄	1.6	135 at $0.2 A/g$	10		$\overline{}$	$[133]$, 2010
AC	4 M NaNO ₃ -EG	2	22.3 at 2 mV/s	$14 - 16$	500	$0 - 60$	$[138]$, 2014
AC	1 M NaNO ₃	1.6	116 at 2 mV/s		$\overline{}$	RT	$[139]$, 2014
AC	Na ₂ SO ₄ /0.5 M	1.6	135 at $0.2 A/g$	10			$[133]$
AC fibers	$1 M H_2SO_4$	0.9	280 at $0.5 A/g$	$\overline{}$	$\overline{}$	RT	$[140]$, 2014
PANi-grafted rGO/AC	1 M H ₂ SO ₄	0.8	1045.51 at 0.2 A/g	8.3	60,000	$\overline{}$	$[141]$, 2014
Graphene/mPANi	1 M H ₂ SO ₄	0.7	749 at $0.5 A/g$	11.3	106.7	$\overline{}$	$[142]$, 2014
PPy thin films	$0.5 M H_2SO_4$	1	510 at 0.25 mA/cm ²	133	758	$\overline{}$	$[143]$, 2014
Pristine flexible PPy membrane	Solid PVA/ H_2SO_4/EG	0.7	191.7 at $0.5 A/g$	14.1	181.9	RT	$[144]$, 2021

Table 3. Aqueous electrolyte-based supercapacitors and their performance.

Abbreviations: Temp: temperature; RT: room temperature; AC: activated carbon; PANi: polyaniline; rGO: reduced graphene oxide; PPy: polypyrrole; mPANi: mesoporous PANi film on ultra-thin graphene nanosheets; EG: ethylene glycol.

Another group is organic electrolytes, which have a high operation potential window between 2.5 and 2.8 V. This high operation voltage can improve both energy density and power density. Also, because they are less corrosive, organic electrolytes provide the possibility of using cheaper material for current collectors. However, the supercapacitors using organic electrolytes have a higher price and lower specific capacitance because of lower ion conductivity, flammability, and toxicity. Also, a complicated process of purification and assembly is required to remove residual impurities that can cause electrolyte degradation and self-discharging [\[145\]](#page-32-12). Generally, an organic electrolyte consists of a conducting salt dissolved in an organic solvent. Like for aqueous electrolytes, the nature of organic electrolytes, such as ion size, ion–solvent interaction, conductivity, and viscosity, can affect supercapacitor performance. The most used organic electrolytes in the literature are propylene carbonate (PC), ethylene carbonate-dimethyl carbonate (EC-DMC), and ethylene carbonate-diethyl carbonate (EC-DEC) [\[146–](#page-32-13)[148\]](#page-32-14). Table [4](#page-17-0) shows different organic electrolyte-based supercapacitors and their performance.

Ionic liquid electrolytes are another type of liquid electrolyte, consisting of melted cations and anions. Ionic liquids have high thermal, chemical, and electrochemical stability; very low volatility; non-flammability; and highly tunable physical and chemical properties [\[11](#page-27-9)[,149\]](#page-32-15). Because of these attractive properties, they have attracted much attention.

Abbreviations: Temp: temperature; RT, room temperature; CNT, carbon nanotube; HFIP, 1,1,1,3,3,3hexafluoropropan-2-ol; AC, activated carbon; ADN, adiponitrile; TEAODFB, tetraethylammonium difluoro(oxalato)borate; TEMABF4: triethylmethylammonium tetrafluoroborate; PS, 1,3-propylene sulfite; TiC-CDC: titanium carbide-derived carbon; SBPBF4, spiro-(1,10)-bipyrrolidinium tetrafluoroborate; EC: ethylene carbonate; DEC: diethyl carbonate; DMC: dimethyl carbonate; EA, ethyl acetate; PANi, polyaniline.

5.2. Solid and Quasi-Solid Electrolytes

A major type of solid electrolyte is polymer-based electrolytes, further classified into three different groups, including solid polymer electrolytes (SPEs), gel polymer electrolytes (GPEs), and inorganic electrolytes.

The most important advantage of solid electrolytes is the simplified packaging process, because they are liquid-leakage-free and can be used as an electrolyte and an electrode separator at the same time. Examples of solid electrolytes are Li_3PS_4 , $Li_7P_3S_{11}$, $Li_{10}GeP_2S_{12}$, and metal oxides like LiNbO₃, and LiTaO₃ [\[163\]](#page-33-3).

SPE electrolytes are composed of polymers and salts without any solvent. The parameters that should be considered to develop a solid-state electrolyte are thermal, chemical, and electrochemical conductivity, mechanical stability, and ionic conductivity.

Due to liquid in GPEs, some studies call this type of electrolyte a quasi-solid-state electrolyte [\[164,](#page-33-4)[165\]](#page-33-5). A GPE is composed of a polymeric matrix, such as polyvinyl alcohol (PVA) and polyacrylic acid (PAA), and a liquid electrolyte that can be an aqueous electrolyte or an organic solvent. PVA is a linear polymer that can be dissolved in an aqueous solution, such as an alkaline solution (ex. KOH), a strong acid (ex. H_2SO_4), or a neutral solution (ex. LiCl), to make a hydrogel electrolyte. PVA hydrogel electrolyte has many interesting properties, such as high hydrophilicity, film-making properties, and non-toxicity. In addition, it has a low cost, and the preparation process is really easy [\[166\]](#page-33-6). Poly(acrylate) and PAA have also been studied as polymer hosts of electrolytes, and the resulting polymer hydrogels showed increased proton conduction when the protons entered an aqueous medium from polymer side chains [\[167](#page-33-7)[,168\]](#page-33-8).

A research group has shown the order of specific capacitance of a $RuO₂$ electrode in the presence of different electrolytes, including poly(2-acrylamido-2-methyl-1-propanesulfonic acid (PAMPS), potassium polyacrylate (PAAK), PAA, as the following [\[167,](#page-33-7)[169\]](#page-33-9):

 $PAMPS/H_2O > 1 M H_2SO_4$ (aqueous electrolyte) > $PAA/H_2SO_4 > PAAK/H_2SO_4 >$ $PAMPS/H_2SO_4$

As it is obvious that the PAMPS/H2O electrolyte has the best performance among the others. This is because of the sulfonate groups at the side chains that can provide the best proton accommodation and, as a result, the highest capacitance [\[169\]](#page-33-9)

Another type of quasi-solid-state electrolyte is called organogel electrolytes, which use organic solvents to replace water in order to increase the working voltage of the cell. Different polymers, such as polyethylene oxide (PEO) [\[170\]](#page-33-10), poly(methyl methacrylate) (PMMA) [\[171\]](#page-33-11), polyvinylprrolidone (PVP), polyether ether ketone (PEEK) [\[172\]](#page-33-12), and copolymers [\[173,](#page-33-13)[174\]](#page-33-14), can be used to host organic solvents, such as polycarbonates (PC), ethyl cellulose (EC), dimethyl carbonate (DMC), dimethyl sulfoxide (DMSO), and dimethylformamide (DMF). This system can increase the cell voltage up to 2.5–3.0 V, which is higher than aqueous electrolytes that have cell voltages up to 1.3 V [\[169\]](#page-33-9). However, organogel electrolytes suffer from relatively low ionic conductivity.

The next group of quasi-solid-state electrolytes is ionic liquids (ILs), which have a large working potential window of up to 4.0 V [\[169\]](#page-33-9). When ILs are used as electrolytes, the IL is incorporated into a polymeric host. The properties of the electrolyte depend on the interactions between the IL and the host polymer. The polymers that can be used with ILs are the same as those used in organogel electrolytes, as mentioned above.

The last group of solid-state electrolytes is inorganic electrolytes. Inorganic electrolytes are not bendable and flexible, but on the other hand, they have good mechanical and thermal stability. A research group has reported a glass–ceramic electrolyte that is used as both an electrolyte and a separator and has high Li-ion conductivity [\[175\]](#page-33-15).

5.3. Redox-Active Electrolytes

In these types of electrolytes, a redox reaction can occur in the electrolyte and increase capacitance [\[176\]](#page-33-16). This kind of electrolyte is prepared by adding suitable redox additives to an electrolyte system to improve the performance of the supercapacitor system. Some examples of these redox additives are $Na₂MoO₄$, $Ce₂(SO₄)₃$, and 1,4dihydroxyanthraquinonedone [\[177\]](#page-33-17).

Heteropoly acids, such as phosphotungstic acid (PWA), can be used as redox-active electrolytes, which can increase proton conductivity and provide multiple redox–electron transfers [\[178](#page-33-18)[,179\]](#page-33-19). The main problem with this type of electrolyte is self-discharging. The reason for it is the migration of the electrolyte between electrodes. In order to inhibit this migration, an ion-exchange membrane (e.g., Nafion) can be used as a separator or to use a redox-active electrolyte that can be converted to an insoluble form during the charging– discharging process [\[180\]](#page-33-20). These types of electrolytes can increase specific capacitance and, consequently, energy and power densities. They can provide an electron source and increase the speed of the Faradic reaction in pseudocapacitance electrodes. On the other hand, they can decrease cycle stability [\[181](#page-33-21)[,182\]](#page-33-22). Potassium iodide (KI) and hydroquinone (HQ) are other examples of redox-active electrolytes.

Non-aqueous redox-active electrolytes are another type of redox-active electrolyte that can increase cell working voltage and energy density. The last group is solid redox-active electrolytes, which have almost the same electron transfer mechanism as the liquid ones.

5.4. Self-Healing Electrolytes

While solid and quasi-solid electrolytes are preferred in wearable and flexible electronics because of their low risk of leaking, they face the risk of structural damage during repeated deformation. Self-healing electrolytes can, ideally, automatically repair small damages such as cracks. The so-called self-healing electrolytes are largely based on two mechanisms: the release of pre-stored reactive chemicals inside a material (electrolyte), trigged by damages such as a crack. The released chemicals will eventually fill and bond to the crack. The re-bonding of the racks can also come from the electrolyte itself without involving the pre-stored chemicals. In such an electrolyte, the bonding between material molecules is dynamic and reversible. When the force generated by deformation exceeds the ultimate strength of the electrolyte, the bonds among electrolyte molecules are broken. However, when deformation disappears, if the broken molecules can return to their proximity, the bonds automatically reform among the molecules. A self-healing electrolyte is a special case of self-healing materials, which has been nicely reviewed recently [\[183](#page-33-23)[–185\]](#page-33-24).

6. Characterization Methods

Electrical conduction in CPs is much more complicated than in classical semiconductors because the doping of CPs involves a redox reaction that changes the chemical and electronic structures of the polymers. Also, CPs have a one-dimensional lattice that makes them amorphous, and the charge carriers are not electrons and often change with the degree of doping and the type of dopant. The appropriate methods used to evaluate electrode materials include cyclic voltammetry (CV), electrochemical impedance spectroscopy (EIS), conductivity measurement (ex: the four-point probe method), scanning electron microscopy (SEM), X-ray photoelectron spectroscopy (XPS), and Fourier transform infrared spectroscopy (FTIR).

6.1. Electrochemical Analysis

Based on electrode materials, their energy storage mechanisms can be different, which further affect cyclic voltammogram and charge/discharge curve shapes. As mentioned before, there are three categories of electrode materials. The first type is the ECDL type, in which a reversible adsorption/desorption process of electrolyte ions is formed at the interface between the electrode and electrolyte; no electrochemical reaction or phase change occurs so that their cyclic stability is generally good. The second type shows pseudoca-

pacitance, in which charge storage is based on a reversible redox reaction on the surface of the material. Compared to the EDLC type, this type of electrode material has a larger specific capacitance, but its cyclic stability is relatively poor. In the third type, or batterytype materials, the electrochemical reaction is controlled by electrolyte ion diffusion that undergoes intercalation reactions of phase change ions. Battery-type materials show high energy density but poor cyclic stability [\[14\]](#page-27-12). The difference in electrochemical behaviors of electrode materials is shown in Figure [14.](#page-20-0)

Figure 14. Comparison of electrochemical behaviors of different types of electrode materials based **Figure 14.** Comparison of electrochemical behaviors of different types of electrode materials based on CV cures (up) and GCD curves (down). (a) EDLCs, (b) and (c) pseudocapacitors, (d) battery-type. Reproduced from [\[14\]](#page-27-12) with permission from Royal Society of Chemistry.

6.1.1. Electrochemical Impedance Spectroscopy (EIS) 6.1.1. Electrochemical Impedance Spectroscopy (EIS)

EIS is a non-destructive method to measure impedance (Z) as a function of the alternat-Ing current amplitude in a range of frequencies. The usual electrochemical measurements,
such as cyclic voltammetry (CV), which uses DC potential, does not provide enough infor- $\frac{1}{2}$ and $\frac{1}{2}$ voltammetry (CV) , which uses DC potential, does not provide enough more mation about the electrochemical reactions that occurred at the interface of an electrode and electrolyte. EIS, which can be performed in a broad range of frequencies in different potentials, gives valuable information about electrical features, such as charge transfer resistivity (R_c), solution resistivity (R_s), diffusion resistivity (Warburg impedance), and equivalent electric circuit simulation. In a simple system, the equivalent circuit model is called a Randle circuit (Figure 15). ing current amplitude in a range of frequencies. The usual electrochemical measurements,

In the EIS method, a small AC potential is applied to DC potential. Since no electrochemical reaction is 100% reversible when the applied potential is reversed, using a high amplitude of AC potential can distort the electrochemical stability of the system [\[186\]](#page-33-25). To minimize such perturbation caused by AC potential, a very small AC signal is applied on a constant DC potential. The variation of impedance with frequency is often displayed in two ways: a Bode plot and a Nyquist plot (Figure [16\)](#page-21-1).

Figure 15. Components of a simple electrochemical system, showing: (a) electric and electrochemical phenomena occurred in the solution and interface area; (**b**) modeled circuit based on these phenomena. C_d : double layer capacitor, R_C: charge transfer resistivity, W: Warburg impedance, R_s: solution *Figure 15.* Components of a simple electrochemical system, showing. (a) electric and electrochemical phenomena.
phenomena occurred in the solution and interface area; (**b**) modeled circuit based on these phenomena. resistivity.

Figure 16. Bode and Nyquist plots representing the essential elements in a circuit that simulates the **Figure 16.** Bode and Nyquist plots representing the essential elements in a circuit that simulates the impedance measured by EIS. These elements are (a) a capacitor, (b) a resistor, (c) a capacitor in series with a resistor, and (**d**) a capacitor in parallel with a resistor. Red curves correspond to left Y axis, and and green curves correspond to right Y axis. green curves correspond to right Y axis.

and ω is frequency. In a Nyquist plot, the imaginary part of impedance (Z_{Im}) is plotted versus the real part of impedance (Z_{Re}) for different values of ω . Figure [16](#page-21-1) shows the most important components in a circuit and the Bode and Nyquist plots representing them. In a Bode plot, $log \ Z \$ and ϕ are both plotted against $log \ \omega$, where Z is impedance In a relatively short time of measurement, a Nyquist plot can provide information about the inner state of a system in terms of electrolyte properties, diffusion layer, ion migration, electrode surface property, and electrode reactions (Figure [17\)](#page-22-0). This method ingration, electrode surface property, and electrode reactions (rigure 17). This method
is independent of leakage current and fluctuating voltage. At low frequencies, migration phenomena are dominant; at middle frequencies, electrode reaction is dominant; and inductive effect and structure porosity are dominant at high frequencies [\[187\]](#page-34-0).

about the inner state of a system in terms of a system in terms of electrolyte properties, diffusion layer, io

Figure 17. Typical impedance spectrum of a battery in an exaggerated Nyquist plot. Here, R_s is the double layer capacitances of the electrodes, σ is the Warburg coefficient, R_3 is the intersection of the series resistance, R_1 and R_2 are the charge transfer resistances of the electrodes, C_{d11} and C_{d12} are the Warburg impedance. Reproduced from Ref. [\[187\]](#page-34-0), CC BY-NC-ND 4.0.

Every electrochemical cell has its special phenomena and, consequently, its impedance curve models, which can be expressed analytically with an equivalent circuit consisting of electrical elements such as capacitors, resistors, and inductors. This equivalent circuit can simulate the internal electrical properties of a cell. To run an impedance test, one needs to adjust an appropriate frequency range, dc potential, ad potential, and curve type.

6.1.2. Cyclic Voltammetry

Cyclic voltammetry (CV) is a potential sweep technique frequently used in electrochemical study and it provides abundant information about the redox reactions at an electrode. It is among the most widely practiced electrochemical methods. Figure [18](#page-23-0) shows a linear (t, time) complete scan where the backward scan starts when the forward scan reaches the switching potential E_{λ} . The potential at any time point is given by [\[188\]](#page-34-1):

$$
(0 < t \le \lambda) E = Ei - vt
$$

$$
(t > \lambda) E = Ei - 2v\lambda + vt
$$

where:

v: scan rate (v/s); E_{λ} : switching potential; E_i : initial potential; and λ: run time at switching potential.

trode, and the peak in positive scan shows oxidation.

Figure 18. Cyclic voltammetry. (a) Cyclic potential sweep, (b) resulting cyclic voltammogram. produced from [187] with permission from John Wiley & Sons. Reproduced from [\[187\]](#page-34-0) with permission from John Wiley & Sons.

The peak in the negative scan represents a reduction taking place at the working electrode, and the peak in positive scan shows oxidation.

By this voltammogram, the capacitance of an electrode material can be calculated. The charge (Q) accumulated on the electrode is calculated by the area under the CV curve in calculated based on the equation below: one direction from E_1 to E_2 (see Figure [19\)](#page-23-1), and the capacitance (C_i) of the electrode can be

$$
C_i = \left| \frac{Q}{E_2 - E_1} \right|
$$

Figure 19. Cyclic voltammogram and determination of capacity.

6.1.3. Cyclic Charge–Discharge (CCD)

Figure 19. Cyclic voltammogram and determination of capacity*.* 6.1.3. Cyclic Charge–Discharge (CCD) constant current in a determined voltage window. A loop of charge–discharge is called a cycle. The capacitance (C) of each cycle is calculated in farad (F) based on the following α densities, and cycle stability. Note that α CCD is a technique frequently used to evaluate specific capacitance, power and energy densities, and cycle stability. Normally, a charge–discharge cycle is performed under a equation [\[189\]](#page-34-2):

$$
C=\frac{Q}{V}
$$

where Q is the charge in coulomb (C), and V is the voltage window. Both are the function of a cycle.

Q is calculated based on the equation below [\[190\]](#page-34-3):

$$
Q = \frac{I \times \Delta t}{m (or A)}
$$

where I/m or I/A is current density per weight (A/g) or current density per area (mA/cm 2), and ∆t is discharging time. So, the calculation of C is based on the following equation where ∆V is the voltage window:

$$
C_s = \frac{I \times \Delta t}{m (or A) \times \Delta V}
$$

Energy and power densities of the electrode are calculated based on the following equations:

Energy density
$$
(Wh \cdot Kg^{-1}) = \frac{1}{2} C_s \Delta V^2 \frac{1000}{3600}
$$

Power density $(W \cdot Kg^{-1}) = \frac{E}{t_d} \times 3600$

where C_s = specific capacitance (F/g or mF/cm²); ΔV (V) is the maximum potential window; E is the energy density; and t (s) is the discharging time.

6.2. Four-Point Probe for Resistivity Measurement

The four-point probe technique is most commonly used to measure surface or sheet resistivity. The set-up consists of four small metal tips lined up with the same separation, as shown in Figure [20.](#page-24-0) A constant current is applied between the two outer tips, and the voltage drop is measured between the inner tips. The resistivity of the surface is given by the equation below, where C_F is the correction factor based on the size of the specimen relative to the dimension of the probe and the ratio of the thickness of the conducting layer to probe separation [\[191\]](#page-34-4).

 $\cdot C_F$

Figure 20. Four-point probe to measure membrane resistivity. **Figure 20.** Four-point probe to measure membrane resistivity.

6.3. Infrared Spectroscopy 6.3. Infrared Spectroscopy

Infrared spectroscopy (IR) is a conventional method to study the chemical nature of substances, including polymers. IR is based on the interactions between the molecular substances, including polymers. IR is based on the interactions between the inforced and structures of a material and the electromagnetic radiation in the infrared range. When IR structures of a material and the electromagnetic radiation in the infrared range. When IR radiation is in contact with a test material, the chemical bonds of the material absorb IR energy at characteristic frequencies according to the chemical bond vibration modes, such as stretching, bending, and scissoring [\[192\]](#page-34-5). The balls represent the atoms connected by chemical bonds that act as springs. Based on how the atoms move, the vibration mode can be determined. The movement of an atom toward or away from other atoms along the line of the spring represents a stretching vibration. Stretching can be either symmetric or asymmetric. A molecule with three or more atoms can experience multiple modes of vibration, as summarized in Figure [21.](#page-25-0) Infrared spectroscopy (IR) is a conventional method to study the chemical nature of

Figure 21. Molecular vibration modes. **Figure 21.** Molecular vibration modes.

as summarized in Figure 21. The summarized in Figure 21. The summarized in Figure 21. The summarized in Figure

Each chemical group has particular interactions with infrared radiation, which Each chemical group has particular interactions with infrared radiation, which causes causes energy absorptions at characteristic frequencies that differ, one chemical group energy absorptions at characteristic frequencies that differ, one chemical group from another. from another. The region below 1800 cm−1 is the fingerprint of a material, where the peaks most useful information, and can be complicated to interpret. Figure [22](#page-25-1) shows the typical most useful intermation, and can be complicated to interpret. Figure 22 shows the typical infrared absorption positions and intensities of important chemical bonds. The region below 1800 cm⁻¹ is the fingerprint of a material, where the peaks carry the

 F_{F} Figure 22. Group F_{F} and F_{F} in F_{F} in F_{F} is F_{F} in F_{F} i [193], CC BY 4.0*.* from [\[193\]](#page-34-6), CC BY 4.0. **Figure 22.** Group frequency and fingerprint regions of the mid-infrared spectrum. Reproduced

6.4. X-ray Photoelectron Spectroscopy (XPS)

XPS is a non-destructive surface analysis method that allows identifying the chemical and electronic structures of the top-most layer of a material. It gives information about the electronic valance structures of chemical bonds. The atomic composition of a sample can also be determined by this method. The under-test sample is excited by a high-energy X-ray source. While the X-ray penetration depth into a sample is deep, the free path (λ) of the photoelectrons escaping from the sample is very short, e.g., 5 to 20 Å, leading to a sampling $h = \frac{1}{2}$ depth of 15 to 60 A (3A). By changing the take-on angle, the sampling depth can be further reduced. Consequently, this method gives surface rather than bulk chemical information. In this method, atoms absorb X-ray energy, then an electron from the K-shell, which is the shell and the shell and the shell and the K-shell, which is the lowest energy shell of the atom, is ejected, and its kinetic energy (KE) is measured. The binding energy (BE) of that photoelectron can be calculated based on the KE and the energy of the incident beam (h*v*) [\[194\]](#page-34-7). Figure [23](#page-26-0) shows the energy-ejecting mechanism. depth of 15 to 60 Å (3λ). By changing the take-off angle, the sampling depth can be further

ured. The binding energy (BE) of that photoelectron can be calculated based on the KEE α

anism.

Figure 23. Electron excitation from K-shell's atoms. Reproduced from [194] with permission from **Figure 23.** Electron excitation from K-shell's atoms. Reproduced from [\[194\]](#page-34-7) with permission from Pavan M. V. Raja & Andrew R. Barron, Rice University (CC BY 4.0). Pavan M. V. Raja & Andrew R. Barron, Rice University (CC BY 4.0).

the number of detected electrons with a specific BE is proportional to the number of atoms of the corresponding element in the testing sample, which provide the relative percentage of each element i[n](#page-26-1) that sample. Table 5 shows the electron BE and the orbitals from which
the electrons are sisated α is electron and species. Because the BE of photoelectrons is characteristic for each element (except hydrogen), the electrons are ejected.

Table 5. Some important elemental binding energies. Reproduced from [\[194\]](#page-34-7) with permission from Pavan M. V. Raja & Andrew R. Barron, Rice University (CC BY 4.0).

6.5. Scanning Electron Microscopy (SEM)

In a material surface study, scanning electron microscopy (SEM) is one of the more important methods that can provide useful information about the material surface structure. Compared to optical microscopy, which is the oldest method with a maximum magnification of 1000 times a sample's original size, SEM works based on electrons, and the magnification of the image can reach up to 300,000 times that of the sample's original size. SEM is a non-destructive method in which a high-energy beam of electrons is applied to the surface of a sample and penetrates about $1 \mu m$ of depth to generate the secondary electrons (SEs) and backscattered electrons (BSEs) that are used to produce an image of sample topography. In order to have a clear, sharp image, parameters such as brightness and intensity can be adjusted by the operator. Low accelerating voltages (less than 5 kV) can provide images with rich surface details; however, high accelerating voltages (15–30 kV) that can penetrate underneath the surface will reflect signals that include some details about the interior of the specimen [\[195\]](#page-34-8).

7. Conclusions

Multiple factors affect the overall performance of electrodes and supercapacitors. Only carefully designed strategies can lead to high capacitance, balanced energy and power densities, and acceptable cyclic stability. Nanostructured CP electrodes provide a high number of redox reaction sites at the electrode/electrolyte interface and, consequently, a high capacitance. Such nanostructures can be formed through special polymerization methods or by coating the nanostructured carbon and metal oxide surfaces. Nanostructures may also improve the cyclic stability of CP electrodes because of less stress in thin-walled

structures compared with thick structures that have undergone the same strain. Currently, metal oxides and carbon-based nanomaterials, such as graphene and carbon nanotubes, are the leaders. A $Co₃O₄$ electrode achieved 3560 F/g, while a PPy/CoO electrode recorded 2225 F/g , demonstrating the advantage of combining with metal oxide [\[196\]](#page-34-9). To compete with carbon and metal oxide-based supercapacitors, CP electrodes are light, less expensive, capable of being flexible, and environmentally friendly. These advantages make CPbased electrodes and supercapacitors suitable for wearable, disposable, and low-energydensity personal electronics. Supercapacitors made of CP may also be used in medical implants because CPs such as PPy and PEDOT are biocompatible. However, CP-based supercapacitors still need to improve their energy density and cyclic stability. To be used in medical implants, biocompatible electrolytes are mandatory. Industry must identify the best composition and structure based on specific applications. Another challenge is how to form and maintain excellent contact between CP and metallic current collectors, particularly when repeated deformation occurs. Obviously, there is still a gap between the research and application of CP-based supercapacitors, which needs our continuous efforts.

Author Contributions: Z.R.: methodology, writing of first manuscript, format analysis, and revision. F.M.: resource, format analysis, supervision, and revision. Z.Z.: conceptualization, resource, format analysis, validation, supervision, revision, and funding. All authors have read and agreed to the published version of the manuscript.

Funding: This work was funded by the Discovery Grant (RGPIN-2017-06138) of the Natural Sciences and Engineering Research Council of Canada.

Data Availability Statement: Not applicable.

Conflicts of Interest: The authors declare no conflict of interest.

References

- 1. Chen, C.; Zhang, Y.; Li, Y.; Dai, J.; Song, J.; Yao, Y.; Gong, Y.; Kierzewski, I.; Xie, J.; Hu, L. All-wood, low tortuosity; aqueous, biodegradable supercapacitors with ultra-high capacitance. *Energy Environ. Sci.* **2017**, *10*, 538–545.
- 2. Kumar, N.; Kim, S.B.; Lee, S.Y.; Park, S.J. Recent advanced supercapacitor: A Review of storage mechanisms, electrode materials, modification, and perspectives. *Nanomaterials* **2022**, *12*, 3708. [\[CrossRef\]](https://doi.org/10.3390/nano12203708)
- 3. Ahmed, S.; Ahmed, A.; Basha, B.; Hussain, S.; Uddin, I.; Gondal, M.A. Critical review on recent developments in conducting polymer nanocomposites for supercapacitors. *Synth. Met.* **2023**, *295*, 117326.
- 4. Hao, L.; Yu, D. Progress of conductive polypyrrole nanocomposites. *Synth. Met.* **2022**, *290*, 117138.
- 5. Majeed, A.H.; Mohammed, L.A.; Hammoodi, O.G.; Sehgal, S.; Alheety, M.A.; Saxena, K.K.; Dadoosh, S.A.; Mohammed, I.K.; Jasim, M.M.; Salmaan, N.U. A review on polyaniline: Synthesis, properties, nanocomposites, and electrochemical applications. *Int. J. Polym. Sci.* **2022**, *2022*, 19.
- 6. Poonam, K.; Sharma, A.; Arora, S.K. Tripathi. Review of supercapacitors: Materials and devices. *J. Energy Storage* **2019**, *21*, 801–825.
- 7. Libich, J.; Máca, J.; Vondrák, J.; Čech, O.; Sedlaříková, M. Supercapacitors: Properties and applications. *J. Energy Storage* 2018, 17, 224–227.
- 8. REN21. *Renewable Energy Policy Network for 21st Century, Renewables 2016 Global Status Report*; REN21: Paris, France, 2016; ISBN 978-3-9818107-0-7.
- 9. Wu, Z.S.; Winter, A.; Chen, L.; Sun, Y.; Turchanin, A.; Feng, X.; Müllen, K. Three-dimensional nitrogen and boron co-doped graphene for high-performance all-solid-state supercapacitors. *Adv. Mater.* **2012**, *24*, 5130–5135.
- 10. Lu, X.; Wang, G.; Zhai, T.; Yu, M.; Xie, S.; Ling, Y.; Liang, C.; Tong, Y.; Li, Y. Stabilized TiN nanowire arrays for high-performance and flexible supercapacitors. *Nano Lett.* **2012**, *12*, 5376–5381.
- 11. Armand, M.; Endres, F.; MacFarlane, D.E.; Ohno, H.; Scrosati, B. Ionic-liquid materials for the electrochemical challenges of the future. *Nat. Mater.* **2009**, *8*, 621–629.
- 12. Van Aken, K.L.; Beidaghi, M.; Gogotsi, Y. Formulation of ionic-liquid electrolyte to expand the voltage window of supercapacitors. *Angew. Chem. Int. Ed.* **2015**, *54*, 4806–4809.
- 13. Zhong, C.; Deng, Y.; Hu, W.; Qiao, J.; Zhang, L.; Zhang, J. A review of electrolyte materials and compositions for electrochemical supercapacitors. *Chem. Soc. Rev.* **2015**, *44*, 7484–7539. [\[PubMed\]](https://www.ncbi.nlm.nih.gov/pubmed/26050756)
- 14. Wu, Q.; He, T.; Zhang, Y.; Zhang, J.; Wang, Z.; Liu, Y.; Zhao, L.; Wu, Y.; Ran, F. Cyclic stability of supercapacitors: Materials, energy storage mechanism, test methods, and device. *J. Mater. Chem. A* **2021**, *9*, 24094–24147.
- 15. Zhang, K.; Zhang, L.L.; Zhao, X.S.; Wu, J. Graphene/polyaniline nanofiber composites as supercapacitor electrodes. *Chem. Mater.* **2010**, *22*, 1392–1401.
- 16. Antiohos, D.; Folkes, G.; Sherrell, P.; Ashraf, S.; Wallace, G.G.; Aitchison, P.; Harris, A.T.; Chen, J.; Minett, A.I. Compositional effects of PEDOT-PSS/single walled carbon nanotube films on supercapacitor device performance. *J. Mater. Chem.* **2011**, *21*, 15987–15994.
- 17. Zhu, M.; Huang, Y.; Deng, Q.; Zhou, J.; Pei, Z.; Xue, Q.; Huang, Y.; Wang, Z.; Li, H.; Huang, Q.; et al. Highly flexible, freestanding supercapacitor electrode with enhanced performance obtained by hybridizing polypyrrole chains with MXene. *Adv. Energy Mater.* **2016**, *6*, 1600969.
- 18. Ghosh, S.; Inganäs, O. Conducting polymer hydrogels as 3D electrodes: Applications for supercapacitors. *Adv. Mater.* **1999**, *11*, 1214–1218.
- 19. Chen, T.; Peng, H.; Durstock, M.; Dai, L. High-performance transparent and stretchable all-solid supercapacitors based on highly aligned carbon nanotube sheets. *Sci. Rep.* **2014**, *4*, 3612.
- 20. Zang, X.; Zhu, M.; Li, X.; Li, X.; Zhen, Z.; Lao, J.; Wang, K.; Kang, F.; Wei, B.; Zhu, H. Dynamically stretchable supercapacitors based on graphene woven fabric electrodes. *Nano Energy.* **2015**, *15*, 83–91.
- 21. Wang, S.; Liu, N.; Su, J.; Li, L.; Long, F.; Zou, Z.; Jiang, X.; Gao, Y. Highly stretchable and self-healable supercapacitor with reduced graphene oxide based fiber springs. *ACS Nano* **2017**, *11*, 2066–2074.
- 22. Wang, Z.; Zhu, M.; Pei, Z.; Xue, Q.; Li, H.; Huang, Y.; Zhi, C. Polymers for supercapacitors: Boosting the development of the flexible and wearable energy storage. *Mater. Sci. Eng. R Rep.* **2020**, *139*, 100520.
- 23. Chatterjee, D.P.; Nandi, A.K. A review on the recent advances in hybrid supercapacitors. *J. Mater. Chem. A* **2021**, *9*, 15880.
- 24. Wang, G.; Zhang, L.; Zhang, J. A review of electrode materials for electrochemical supercapacitors. *Chem. Soc. Rev.* **2012**, *41*, 797–828.
- 25. Pattananuwat, P.; Aht-ong, D. Controllable morphology of polypyrrole wrapped graphene hydrogel framework composites via cyclic voltammetry with aiding of poly (sodium 4-styrene sulfonate) for the flexible supercapacitor electrode. *Electrochim. Acta* **2017**, *224*, 149–160.
- 26. Fu, D.; Li, H.; Zhang, X.M.; Han, G.; Zhou, H.; Chang, Y. Flexible solid-state supercapacitor fabricated by metal-organic framework/graphene oxide hybrid interconnected with PEDOT. *Mater. Chem. Phys.* **2016**, *79*, 166–173.
- 27. Conway, B.E. Transition from "Supercapacitor" to "Battery" Behavior in Electrochemical Energy Storage. *J. Electrochem. Soc.* **1991**, *138*, 1539.
- 28. Han, G.; Liu, Y.; Zhang, L.; Kan, E.; Zhang, S.; Tang, J.; Tang, W. MnO² nanorods intercalating graphene oxide/polyaniline ternary composites for robust high-performance supercapacitors. *Sci. Rep.* **2014**, *4*, 4824.
- 29. Zhou, C.; Zhang, Y.; Li, Y.; Liu, J. Construction of high-capacitance 3D CoO@Polypyrrole nanowire array electrode for aqueous asymmetric supercapacitor. *Nano Lett.* **2013**, *13*, 2078–2085.
- 30. Zhang, Y.; Feng, H.; Wu, X.; Wang, L.; Zhang, A.; Xia, T.; Dong, H.; Li, X.; Zhang, L. Progress of electrochemical capacitor electrode materials: A review. *Int. J. Hydrogen Energy* **2009**, *34*, 4889–4899.
- 31. Conway, B.E.; Birss, V.; Wojtowicz, J. The role and utilization of pseudocapacitance for energy storage by supercapacitors. *J. Power Sources* **1997**, *66*, 1–14.
- 32. Mensah-Darkwa, K.; Zequine, C.; Kahol, P.K.; Gupta, R.K. Supercapacitor energy storage device using biowastes: A sustainable approach to green energy. *Sustainability* **2019**, *11*, 414. [\[CrossRef\]](https://doi.org/10.3390/su11020414)
- 33. Yu, G.; Xie, X.; Pan, L.; Bao, Z.; Cui, Y. Hybrid nanostructured materials for high-performance electrochemical capacitors. *Nano Energy* **2013**, *2*, 213–234.
- 34. Miller, J.R.; Simon, P. Fundamentals of electrochemical capacitor design and operation. *Electrochem. Soc. Interface.* **2008**, *17*, 31–32.
- 35. Zheng, Y.; Yang, Y.; Chen, S.; Yuan, Q. Smart, stretchable and wearable supercapacitors: Prospects and challenges. *CrystEngComm* **2016**, *18*, 4218–4235.
- 36. Wang, Y.; Zhu, C.; Pfattner, R.; Yan, H.; Jin, L.; Chen, S.; Molina-Lopez, F.; Lissel, F.; Liu, J.; Rabiah, N.I.; et al. A highly stretchable, transparent, and conductive polymer. *Sci. Adv.* **2017**, *3*, 1602076. [\[CrossRef\]](https://doi.org/10.1126/sciadv.1602076)
- 37. Lota, K.; Khomenko, V.; Frackowiak, E. Frackowiak. Capacitance properties of poly(3,4-ethylenedioxythiophene)/carbon nanotubes composites. *J. Phys. Chem. Solids* **2004**, *13*, 2419. [\[CrossRef\]](https://doi.org/10.1016/j.jpcs.2003.10.051)
- 38. Snook, G.A.; Kao, P.; Best, A.S. Conducting-polymer-based supercapacitor devices and electrodes. *J. Power Sources* **2011**, *196*, 1–12. [\[CrossRef\]](https://doi.org/10.1016/j.jpowsour.2010.06.084)
- 39. Janata, J.; Josowicz, M. Conducting polymers in electronic chemical sensors. *Nat. Mater.* **2003**, *2*, 19–24. [\[CrossRef\]](https://doi.org/10.1038/nmat768) [\[PubMed\]](https://www.ncbi.nlm.nih.gov/pubmed/12652667)
- 40. Sadki, S.; Schottland, P.; Brodie, N.; Sabouraud, G. The mechanisms of pyrrole electropolymerization. *Chem. Soc. Rev.* **2000**, *29*, 283–293.
- 41. Lee, K.; Cho, S.; Sung, H.P.; Heeger, A.J.; Lee, C.W.; Lee, S.H. Metallic transport in polyaniline. *Nature* **2006**, *441*, 65–68. [\[CrossRef\]](https://doi.org/10.1038/nature04705)
- 42. Peng, C.; Hu, D.; Chen, G.Z. Theoretical specific capacitance based on charge storage mechanisms of conducting polymers: Comment on "Vertically oriented arrays of polyaniline nanorods and their super electrochemical properties". *Chem. Commun.* **2011**, *47*, 4105–4107. [\[CrossRef\]](https://doi.org/10.1039/c1cc10675a)
- 43. Zang, J.; Bao, S.J.; Li, C.M.; Bian, H.; Cui, X.; Bao, Q.; Sun, C.Q.; Guo, J.; Lian, K. Well-aligned cone-shaped nanostructure of polypyrrole/RuO2 and its electrochemical supercapacitor. *J. Phys. Chem. C* **2008**, *112*, 14843–14847. [\[CrossRef\]](https://doi.org/10.1021/jp8049558)
- 44. Bobacka, J.; Lewenstam, A.; Ivaska, A. Electrochemical impedance spectroscopy of oxidized poly(3,4-ethylenedioxythiophene) film electrodes in aqueous solutions. *J. Electroanal. Chem.* **2000**, *88*, 423–429. [\[CrossRef\]](https://doi.org/10.1016/S0022-0728(00)00206-0)
- 45. Wallace, G.G.; Teasdale, P.R.; Spinks, G.M.; Kane-Maguire, L.A.P. *Conductive Electroactive Polymers: Intelligent Materials Systems*, 2nd ed.; CRC Press: Boca Raton, FL, USA, 2003. [\[CrossRef\]](https://doi.org/10.1201/9781420067156)
- 46. Chen, Y.; Du, L.; Yang, P.; Sun, P.; Yu, X.; Mai, W. Significantly enhanced robustness and electrochemical performance of flexible carbon nanotube-based supercapacitors by electrodepositing polypyrrole. *J. Power Sources* **2015**, *287*, 68–74. [\[CrossRef\]](https://doi.org/10.1016/j.jpowsour.2015.04.026)
- 47. Qian, T.; Yu, C.; Wu, S.; Shen, J. A facilely prepared polypyrrole-reduced graphene oxide composite with a crumpled surface for high performance supercapacitor electrodes. *J. Mater. Chem. A* **2013**, *1*, 6539–6542. [\[CrossRef\]](https://doi.org/10.1039/c3ta11146f)
- 48. Wang, K.; Wu, H.; Meng, Y.; Wei, Z. Conducting polymer nanowire arrays for high performance supercapacitors. *Small* **2014**, *10*, 14–31. [\[CrossRef\]](https://doi.org/10.1002/smll.201301991)
- 49. Villarreal, I.; Morales, E.; Otero, T.F.; Acosta, J.L. Electropolymerization kinetics of pyrrole in aqueous solution on graphite felt electrodes. *Synth. Met.* **2001**, *123*, 487–492. [\[CrossRef\]](https://doi.org/10.1016/S0379-6779(01)00343-5)
- 50. Thomas, O.D.; Peckham, T.J.; Thanganathan, U.; Yang, Y.; Holdcroft, S. Sulfonated polybenzimidazoles: Proton conduction and acid-base crosslinking. *J. Polym. Sci. Part A Polym. Chem.* **2010**, *48*, 3640–3650. [\[CrossRef\]](https://doi.org/10.1002/pola.24147)
- 51. Cadby, A.J.; Yang, C.; Holdcroft, S.; Bradley, D.D.C.; Lane, P.A. Limiting intersystem crossing in conjugated polymers by molecular design. *Adv. Mater.* **2002**, *14*, 57–60. [\[CrossRef\]](https://doi.org/10.1002/1521-4095(20020104)14:1%3C57::AID-ADMA57%3E3.0.CO;2-0)
- 52. Thomas, R.; Durix, S.; Sinturel, C.; Omonov, T.; Goossens, S.; Groeninckx, G.; Moldenaers, P.; Thomas, S. Cure kinetics, morphology and miscibility of modified DGEBA-based epoxy resin—Effects of a liquid rubber inclusion. *Polymer* **2007**, *48*, 1695–1710. [\[CrossRef\]](https://doi.org/10.1016/j.polymer.2007.01.018)
- 53. Tan, Y.; Ghandi, K. Kinetics and mechanism of pyrrole chemical polymerization. *Synth. Met.* **2013**, *175*, 183–191. [\[CrossRef\]](https://doi.org/10.1016/j.synthmet.2013.05.014)
- 54. Zhang, L.; Meng, S.; Zhang, Z. Electroactivity and stability of polylactide/polypyrrole composites. *J. Biomater. Sci. Polym. Ed.* **2011**, *22*, 1931–1946. [\[CrossRef\]](https://doi.org/10.1163/092050610X529164)
- 55. Shirakawa, H. The discovery of polyacetylene Film: The dawning of an era of conducting polymers (Nobel lecture)—Die Entdeckung der Polyacetylenfilme—Der Beginn des Zeitalters leitfähiger Polymere (Nobel-Aufsatz). *Angew. Chem. Int. Ed.* **2001**, *14*, 2574–2580.
- 56. MacDiarmid, A.G.; Mammone, R.J.; Kaner, R.B.; Porter, S.J. The concept of 'doping' of conducting polymers: The role of reduction potentials. *Philos. Trans. R. Soc. Lond. A* **1985**, *314*, 3–15. [\[CrossRef\]](https://doi.org/10.1098/rsta.1985.0004)
- 57. Nalwa, H.S. *Handbook of Organic Conductive Molecules and Polymers, Volume 2, Conductive Polymers: Synthesis and Electrical Properties*; Volume 2 Edition; Wiley: Hoboken, NJ, USA, 1997.
- 58. Shown, I.; Ganguly, A.; Chen, L.C.; Chen, K.H. Conducting polymer-based flexible supercapacitor. *Energy Sci. Eng.* **2015**, *3*, 2–26. [\[CrossRef\]](https://doi.org/10.1002/ese3.50)
- 59. Paul, R.K.; Pillai, C.K.S. Melt/solution processable conducting polyaniline with novel sulfonic acid dopants and its thermoplastic blends. *Synth. Met.* **2000**, *114*, 27–35. [\[CrossRef\]](https://doi.org/10.1016/S0379-6779(00)00206-X)
- 60. Higgins, S.J. Conjugated polymers incorporating pendant functional groups—Synthesis and characterisation. *Chem. Soc. Rev.* **1997**, *26*, 247–257. [\[CrossRef\]](https://doi.org/10.1039/cs9972600247)
- 61. King, G.; Higgins, S.J. Synthesis and characterisation of novel substituted benzo[c]thiophenes and polybenzo[c]thiophenes: Tuning the potentials for n- and p-doping in transparent conducting polymers. *J. Mater. Chem.* **1995**, *30*, 305.
- 62. Liu, J.; Yang, R. Tuning the thermal conductivity of polymers with mechanical strains. *Phys. Rev. B Condens. Matter Mater. Phys.* **2010**, *81*, 174122. [\[CrossRef\]](https://doi.org/10.1103/PhysRevB.81.174122)
- 63. Mithra, L.M.M.; Cao, Y.; Cho, S.; Sutar, D.; Lee, K.; Menon, R.; Subramanyam, S.V. Electrical transport and reflectance studies on polypyrrole-CF3SO3- in the vicinity of metal-insulator transition. *Synth. Met.* **2001**, *119*, 437–438. [\[CrossRef\]](https://doi.org/10.1016/S0379-6779(00)01284-4)
- 64. Malhotra, B.D.; Singhal, R. Conducting polymer based biomolecular electronic devices. *Pramana J. Phys.* **2003**, *61*, 331–343. [\[CrossRef\]](https://doi.org/10.1007/BF02708313)
- 65. Rohwerder, M.; Michalik, A. Conducting polymers for corrosion protection: What makes the difference between failure and success? *Electrochim. Acta* **2007**, *53*, 1300–1313. [\[CrossRef\]](https://doi.org/10.1016/j.electacta.2007.05.026)
- 66. Mahmud, H.N.M.E.; Kassim, A.; Zainal, Z.; Yunus, W.M.M. Electrochemical formation of polypyrrole-carboxymethylcellulose conducting polymer composite films. *J. Mater. Sci. Technol.* **2005**, *21*, 661–665.
- 67. Othman, N.; Talib, Z.A.; Kassim, A.; Shaari, A.H.; Liew, J.Y.C. Electrical properties of polypyrrole conducting polymer at various dopant concentrations. *Malays. J. Fundam. Appl. Sci.* **2009**, *5*, 29–33. [\[CrossRef\]](https://doi.org/10.11113/mjfas.v5n1.284)
- 68. Shaktawat, V.; Jain, N.; Saxena, R.; Saxena, N.S.; Sharma, K.; Sharma, T.P. Temperature dependence of electrical conduction in pure and doped polypyrrole. *Polym. Bull.* **2006**, *57*, 535–543.
- 69. Mofdal, M.E.; Al-Jumaili, B.E.; Talib, Z.A.; Muh'd, I.B.; Tagbo, M. The evaluation of temperature effect on peak-to-peak line width (∆Hpp) of conjugating polymer polypyrrole (Ppy). *J. Phys. Conf. Ser.* **2019**, *1178*, 012021. [\[CrossRef\]](https://doi.org/10.1088/1742-6596/1178/1/012021)
- 70. Lima, M.P.; Magela, E.; Silva, G. Dynamical evolution of polaron to bipolaron in conjugated polymers. *Phys. Rev. B Condens. Matter Mater. Phys.* **2006**, *74*, 14–16. [\[CrossRef\]](https://doi.org/10.1103/PhysRevB.74.224304)
- 71. Xie, S.; Mei, L.; Lin, D.L. Transition between bipolaron and polaron states in doped heterocycle polymers. *Phys. Rev. B* **1994**, *50*, 13364.
- 72. De Oliveira, J.R.; Silva, G.M.E. Interchain interaction effects on polaron-bipolaron transition on conducting polymers. *J. Mater. Sci.* **2008**, *43*, 585–590. [\[CrossRef\]](https://doi.org/10.1007/s10853-007-1665-z)
- 73. Molapo, K.M.; Ndangili, P.M.; Ajayi, R.F.; Mbambisa, G.; Mailu, S.M.; Njomo, N.; Masikini, M.; Baker, P.; Iwuoha, E.I. Electronics of conjugated polymers (I): Polyaniline. *Int. J. Electrochem. Sci.* **2012**, *7*, 11859–11875. [\[CrossRef\]](https://doi.org/10.1016/s1452-3981(23)16509-6)
- 74. Namsheer, K.; Rout, C.S. Conducting polymers: A comprehensive review on recent advances in synthesis, properties and applications. *RSC Adv.* **2021**, *11*, 5659–5697. [\[CrossRef\]](https://doi.org/10.1039/d0ra07800j)
- 75. Wu, C.Q.; Sun, X. Nonlinear optical properties of conducting polymers. *Synth. Met.* **1991**, *43*, 3213–3216. [\[CrossRef\]](https://doi.org/10.1016/0379-6779(91)91271-B)
- 76. Navathe, G.J.; Prasad, S.R.; Mane, A.M.; Barge, S.H.; Dongale, T.D.; Shaikh, V.; Karanjkar, M.M.; Teli, S.B.; Patil, P.S.; Prasad, N.R. A critical review on design and development of new generation energy storage devices. *ES Energy Environ.* **2022**, *17*, 11–32. [\[CrossRef\]](https://doi.org/10.30919/esee8c739)
- 77. Liu, H.; Wang, Y.; Gou, X.; Qi, T.; Yang, J.; Ding, Y. Three-dimensional graphene/polyaniline composite material for highperformance supercapacitor applications. *Mater. Sci. Eng. B Solid-State Mater. Adv. Technol.* **2013**, *178*, 293–298. [\[CrossRef\]](https://doi.org/10.1016/j.mseb.2012.12.002)
- 78. Gobal, F.; Faraji, M. Electrodeposited polyaniline on Pd-loaded TiO₂ nanotubes as active material for electrochemical supercapacitor. *J. Electroanal. Chem.* **2013**, *691*, 51–56. [\[CrossRef\]](https://doi.org/10.1016/j.jelechem.2012.12.008)
- 79. Eftekhari, A.; Li, L.; Yang, Y. Polyaniline supercapacitors. *J. Power Source* **2017**, *347*, 86–107. [\[CrossRef\]](https://doi.org/10.1016/j.jpowsour.2017.02.054)
- 80. Bian, C.; Yu, A. De-doped polyaniline nanofibres with micropores for high-rate aqueous electrochemical capacitor. *Synth. Met.* **2010**, *160*, 1579–1583. [\[CrossRef\]](https://doi.org/10.1016/j.synthmet.2010.04.019)
- 81. Wu, W.; Pan, D.; Li, Y.; Zhao, G.; Jing, L.; Chen, S. Facile fabrication of polyaniline nanotubes using the self-assembly behavior based on the hydrogen bonding: A mechanistic study and application in high-performance electrochemical supercapacitor electrode. *Electrochim. Acta* **2015**, *152*, 126–134. [\[CrossRef\]](https://doi.org/10.1016/j.electacta.2014.11.130)
- 82. Gawli, Y.; Banerjee, A.; Dhakras, D.; Deo, M.; Bulani, D.; Wadgaonkar, P.; Shelke, M.; Ogale, S. 3D polyaniline architecture by concurrent inorganic and organic acid doping for superior and robust high rate supercapacitor performance. *Sci. Rep.* **2016**, *6*, 21002. [\[CrossRef\]](https://doi.org/10.1038/srep21002)
- 83. Giri, S.; Ghosh, D.; Das, C.K. In situ synthesis of cobalt doped polyaniline modified graphene composites for high performance supercapacitor electrode materials. *J. Electroanal. Chem.* **2013**, *697*, 32–45. [\[CrossRef\]](https://doi.org/10.1016/j.jelechem.2013.01.038)
- 84. Yang, Z.; Peng, H.; Wang, W.; Liu, T. Crystallization behavior of poly(ε-caprolactone)/layered double hydroxide nanocomposites. *J. Appl. Polym. Sci.* **2010**, *116*, 2658–2667. [\[CrossRef\]](https://doi.org/10.1002/app.31787)
- 85. Grover, S.; Goel, S.; Marichi, R.B.; Sahu, V.; Singh, G.; Sharma, R.K. Polyaniline all solid-state pseudocapacitor: Role of morphological variations in performance evolution. *Electrochim. Acta* **2016**, *196*, 131–139. [\[CrossRef\]](https://doi.org/10.1016/j.electacta.2016.02.157)
- 86. Sharma, V.; Sahoo, A.; Sharma, Y.; Mohanty, P. Synthesis of nanoporous hypercrosslinked polyaniline (HCPANI) for gas sorption and electrochemical supercapacitor applications. *RSC Adv.* **2015**, *5*, 45749–45754. [\[CrossRef\]](https://doi.org/10.1039/c5ra03016a)
- 87. Fan, L.Z.; Hu, Y.S.; Maier, J.; Adelhelm, P.; Smarsly, B.; Antonietti, M. High electroactivity of polyaniline in supercapacitors by using a hierarchically porous carbon monolith as a support. *Adv. Funct. Mater.* **2007**, *17*, 3083–3087. [\[CrossRef\]](https://doi.org/10.1002/adfm.200700518)
- 88. Cho, S.; Shin, K.H.; Jang, J. Enhanced electrochemical performance of highly porous supercapacitor electrodes based on solution processed polyaniline thin films. *ACS Appl. Mater. Interfaces.* **2013**, *5*, 9186–9193. [\[CrossRef\]](https://doi.org/10.1021/am402702y)
- 89. Rajagopalan, B.; Hur, S.H.; Chung, J.S. Surfactant-treated graphene covered polyaniline nanowires for supercapacitor electrode. *Nanoscale Res. Lett.* **2015**, *10*, 1–9. [\[CrossRef\]](https://doi.org/10.1186/s11671-015-0888-1)
- 90. Miao, Y.E.; Fan, W.; Chen, D.; Liu, T. High-performance supercapacitors based on hollow polyaniline nanofibers by electrospinning. *ACS Appl. Mater. Interfaces.* **2013**, *5*, 4423–4428. [\[CrossRef\]](https://doi.org/10.1021/am4008352) [\[PubMed\]](https://www.ncbi.nlm.nih.gov/pubmed/23586693)
- 91. Eftekhari, A. Synthesis of nanostructured large particles of polyaniline. *J. Appl. Polym. Sci.* **2006**, *102*, 6060–6063. [\[CrossRef\]](https://doi.org/10.1002/app.25232)
- 92. Mujawar, S.H.; Ambade, S.B.; Battumur, T.; Ambade, R.B.; Lee, S.H. Electropolymerization of polyaniline on titanium oxide nanotubes for supercapacitor application. *Electrochim. Acta* **2011**, *56*, 4462–4466. [\[CrossRef\]](https://doi.org/10.1016/j.electacta.2011.02.043)
- 93. Eftekhari, A. (Ed.) *Nanostructured Conductive Polymers*; Wiley: Hoboken, NJ, USA, 2010. [\[CrossRef\]](https://doi.org/10.1002/9780470661338)
- 94. Uppugalla, S.; Male, U.; Srinivasan, P. Design and synthesis of heteroatoms doped carbon/polyaniline hybrid material for high performance electrode in supercapacitor application. *Electrochim. Acta* **2014**, *146*, 242–248. [\[CrossRef\]](https://doi.org/10.1016/j.electacta.2014.09.047)
- 95. Liu, T.; Finn, L.; Yu, M.; Wang, H.; Zhai, T.; Lu, X.; Tong, Y.; Li, Y. Polyaniline and polypyrrole pseudocapacitor electrodes with excellent cycling stability. *Nano Lett.* **2014**, *14*, 2522–2527. [\[CrossRef\]](https://doi.org/10.1021/nl500255v)
- 96. Yan, Y.; Cheng, Q.; Wang, G.; Li, C. Growth of polyaniline nanowhiskers on mesoporous carbon for supercapacitor application. *J. Power Sources* **2011**, *169*, 7835–7840. [\[CrossRef\]](https://doi.org/10.1016/j.jpowsour.2011.03.088)
- 97. Woo, S.W.; Dokko, K.; Nakano, H.; Kanamura, K. Incorporation of polyaniline into macropores of three-dimensionally ordered macroporous carbon electrode for electrochemical capacitors. *J. Power Sources* **2009**, *190*, 596–600. [\[CrossRef\]](https://doi.org/10.1016/j.jpowsour.2009.01.050)
- 98. Moussa, M.; El-Kady, M.F.; Zhao, Z.; Majewski, P.; Ma, J. Recent progress and performance evaluation for polyaniline/graphene nanocomposites as supercapacitor electrodes. *Nanotechnology* **2016**, *27*, 442001. [\[CrossRef\]](https://doi.org/10.1088/0957-4484/27/44/442001)
- 99. Jo, S.; Park, Y.H.; Ha, S.G.; Kim, S.M.; Song, C.; Park, S.Y.; In, I. Simple noncovalent hybridization of polyaniline with graphene and its application for pseudocapacitor. *Synth. Met.* **2015**, *209*, 60–67. [\[CrossRef\]](https://doi.org/10.1016/j.synthmet.2015.06.004)
- 100. Li, Y.; Zhao, X.; Yu, P.; Zhang, Q. Oriented arrays of polyaniline nanorods grown on graphite nanosheets for an electrochemical supercapacitor. *Langmuir* **2013**, *29*, 493–500. [\[CrossRef\]](https://doi.org/10.1021/la303632d) [\[PubMed\]](https://www.ncbi.nlm.nih.gov/pubmed/23205664)
- 101. Jaidev; Jafri, R.I.; Mishra, A.K.; Ramaprabhu, S. Polyaniline-MnO₂ nanotube hybrid nanocomposite as supercapacitor electrode material in acidic electrolyte. *J. Mater. Chem.* **2011**, *21*, 17601–17605. [\[CrossRef\]](https://doi.org/10.1039/c1jm13191e)
- 102. Eftekhari, A.; Molaei, F. Carbon nanotube-assisted electrodeposition. Part I: Battery performance of manganese oxide films electrodeposited at low current densities. *J. Power Sources* **2015**, *274*, 1306–1314. [\[CrossRef\]](https://doi.org/10.1016/j.jpowsour.2013.10.136)
- 103. Eftekhari, A.; Molaei, F. Carbon nanotube-assisted electrodeposition. Part II: Superior pseudo-capacitive behavior of manganese oxide film electrodeposited at high current densities. *J. Power Sources* **2015**, *274*, 1315–1321. [\[CrossRef\]](https://doi.org/10.1016/j.jpowsour.2013.10.144)
- 104. Mu, B.; Zhang, W.; Shao, S.; Wang, A. Glycol assisted synthesis of graphene-MnO² -polyaniline ternary composites for high performance supercapacitor electrodes. *Phys. Chem. Chem. Phys.* **2014**, *16*, 7872–7880. [\[CrossRef\]](https://doi.org/10.1039/c4cp00280f)
- 105. Lu, K.; Jiang, R.; Gao, X.; Ma, H. Fe3O4/carbon nanotubes/polyaniline ternary composites with synergistic effects for high performance supercapacitors. *RSC Adv.* **2014**, *4*, 52393–52401. [\[CrossRef\]](https://doi.org/10.1039/c4ra11088a)
- 106. Eftekhari, A. Glycerol biosensor based on glycerol dehydrogenase incorporated into polyaniline modified aluminum electrode using hexacyanoferrate as mediator. *Sens. Actuators B Chem.* **2001**, *80*, 283–289. [\[CrossRef\]](https://doi.org/10.1016/S0925-4005(01)00916-9)
- 107. Eftekhari, A. Electropolymerization of aniline onto passivated substrate and its application for preparation of enzyme-modified electrode. *Synth. Met.* **2004**, *145*, 211–216. [\[CrossRef\]](https://doi.org/10.1016/j.synthmet.2004.05.016)
- 108. Vu, Q.T.; Pavlik, M.; Hebestreit, N.; Pfleger, J.; Rammelt, U.; Plieth, W. Electrophoretic deposition of nanocomposites formed from polythiophene and metal oxides. *Electrochim. Acta* **2005**, *51*, 1117–1124. [\[CrossRef\]](https://doi.org/10.1016/j.electacta.2005.05.052)
- 109. Pringle, J.M.; Forsyth, M.; MacFarlane, D.R.; Wagner, K.; Hall, S.B.; Officer, D.L. The influence of the monomer and the ionic liquid on the electrochemical preparation of polythiophene. *Polymer* **2005**, *46*, 2047–2058. [\[CrossRef\]](https://doi.org/10.1016/j.polymer.2005.01.034)
- 110. Patil, B.H.; Jagadale, A.D.; Lokhande, C.D. Synthesis of polythiophene thin films by simple successive ionic layer adsorption and reaction (SILAR) method for supercapacitor application. *Synth. Met.* **2012**, *162*, 1400–1405. [\[CrossRef\]](https://doi.org/10.1016/j.synthmet.2012.05.023)
- 111. Reyman, D.; Guereca, E.; Herrasti, P. Electrodeposition of polythiophene assisted by sonochemistry and incorporation of fluorophores in the polymeric matrix. *Ultrason. Sonochem.* **2007**, *14*, 653–660. [\[CrossRef\]](https://doi.org/10.1016/j.ultsonch.2006.10.006)
- 112. Krische, B.; Zagorska, M.; Hellberg, J. Bithiophenes as starting monomers for polythiophene syntheses. *Synth. Met.* **1993**, *58*, 295–307. [\[CrossRef\]](https://doi.org/10.1016/0379-6779(93)91139-S)
- 113. Krische, B.; Zagorska, M. Polythiophene synthesis by electropolymerization of thiophene and bithiophene. *Synth. Met.* **1989**, *33*, 257–267. [\[CrossRef\]](https://doi.org/10.1016/0379-6779(89)90472-4)
- 114. Roncali, J. Conjugated poly(thiophenes): Synthesis, functionalization, and applications. *Chem. Rev.* **1992**, *92*, 711–738. [\[CrossRef\]](https://doi.org/10.1021/cr00012a009)
- 115. Fu, C.; Zhou, H.; Liu, R.; Huang, Z.; Chen, J.; Kuang, Y. Supercapacitor based on electropolymerized polythiophene and multi-walled carbon nanotubes composites. *Mater. Chem. Phys.* **2012**, *132*, 596–600. [\[CrossRef\]](https://doi.org/10.1016/j.matchemphys.2011.11.074)
- 116. Senthilkumar, B.; Thenamirtham, P.; Selvan, R.K. Structural and electrochemical properties of polythiophene. *Appl. Surf. Sci.* **2011**, *257*, 9063–9067. [\[CrossRef\]](https://doi.org/10.1016/j.apsusc.2011.05.100)
- 117. Yagci, Y.; Yilmaz, F.; Kiralp, S.; Toppare, L. Photoinduced polymerization of thiophene using iodonium salt. *Macromol. Chem. Phys.* **2005**, *206*, 1178–1182. [\[CrossRef\]](https://doi.org/10.1002/macp.200500084)
- 118. Skompska, M.; Mieczkowski, J.; Holze, R.; Heinze, J. In situ conductance studies of p- and n-doping of poly(3,4 dialkoxythiophenes). *J. Electroanal. Chem.* **2005**, *245*, 135–143. [\[CrossRef\]](https://doi.org/10.1016/j.jelechem.2004.11.008)
- 119. Levi, M.D.; Gofer, Y.; Aurbach, D.; Lapkowski, M.; Vieil, E.; Serose, J. Simultaneous voltammetric and in situ conductivity studies of n-doping of polythiophene films with tetraalkylammonium, alkali, and alkaline–earth cations. *J. Electrochem. Soc.* **2000**, *147*, 1096–1104. [\[CrossRef\]](https://doi.org/10.1149/1.1393319)
- 120. Bongini, A.; Barbarella, G.; Favaretto, L.; Sotgiu, G.; Zambianchi, M.; Mastragostino, M.; Arbizzani, C.; Soavi, F. New n-dopable thiophene based polymers. *Synth. Met.* **1999**, *101*, 13–14. [\[CrossRef\]](https://doi.org/10.1016/S0379-6779(98)00773-5)
- 121. Xu, J.; Wang, D.; Fan, L.; Yuan, Y.; Wei, W.; Liu, R.; Gu, S.; Xu, W. Fabric electrodes coated with polypyrrole nanorods for flexible supercapacitor application prepared via a reactive self-degraded template. *Org. Electron.* **2015**, *26*, 292–299.
- 122. Du, H.Y.; Liu, X.X.; Ren, Z.; Liu, P.P. Capacitance characteristic of PEDOT electrodeposited on different substrates. *J. Solid State Electrochem.* **2018**, *22*, 3947–3954.
- 123. Gojgić, J.; Petrović, M.; Jugović, B.; Jokić, B.; Grgur, B.; Gvozdenović, M. Electrochemical and electrical performances of high energy storage polyaniline electrode with supercapattery behavior. *Polymers* **2022**, *14*, 5365. [\[CrossRef\]](https://doi.org/10.3390/polym14245365) [\[PubMed\]](https://www.ncbi.nlm.nih.gov/pubmed/36559731)
- 124. Fontana-Escartin, A.; Puiggalí-Jou, A.; Lanzalaco, S.; Bertran, O.; Alemán, C. Manufactured flexible electrodes for dopamine detection: Integration of conducting polymer in 3D-printed polylactic acid. *Adv. Eng. Mater.* **2023**, *21*, 2100002.
- 125. Yu, A.J.Y.; Nyein, H.Y.Y.; Gao, W. Flexible electrochemical bioelectronics: The rise of in situ bioanalysis. *Adv. Mater.* **2020**, *32*, 1902083.
- 126. Elbadawi, A.W.B.M.; Ong, J.J.; Pollard, T.D.; Gaisford, S. Additive manufacturable materials for electrochemical biosensor electrodes. *Adv. Funct. Mater.* **2021**, *31*, 2006407.
- 127. Poizot, P.; Gaubicher, J.; Renault, S.; Dubois, L.; Liang, Y. Opportunities and challenges for organic electrodes in electrochemical. *Energy Storage Chem. Rev.* **2020**, *120*, 6490–6557. [\[CrossRef\]](https://doi.org/10.1021/acs.chemrev.9b00482)
- 128. Liu, K.; Duan, T.; Zhang, F.; Feng, M.; Wang, R.; Tian, X.; Li, H.; Jiang, B. Flexible electrode materials for emerging electronics: Materials, fabrications and applications. *J. Mater. Chem. A.* **2024**, *12*, 20606–20637. [\[CrossRef\]](https://doi.org/10.1039/D4TA01960A)
- 129. Iqbal, M.Z.; Zakar, S.; Haider, S.S. Role of aqueous electrolytes on the performance of electrochemical energy storage device. *J. Electroanal. Chem.* **2020**, *585*, 113793. [\[CrossRef\]](https://doi.org/10.1016/j.jelechem.2019.113793)
- 130. Jiménez-Cordero, D.; Heras, F.; Gilarranz, M.A.; Raymundo-Piñero, E. Grape seed carbons for studying the influence of texture on supercapacitor behaviour in aqueous electrolytes. *Carbon* **2014**, *71*, 127–138. [\[CrossRef\]](https://doi.org/10.1016/j.carbon.2014.01.021)
- 131. Torchała, K.; Kierzek, K.; Machnikowski, J. Capacitance behavior of KOH activated mesocarbon microbeads in different aqueous electrolytes. *Electrochim. Acta* **2012**, *86*, 260–267. [\[CrossRef\]](https://doi.org/10.1016/j.electacta.2012.07.062)
- 132. Zhang, X.; Wang, X.; Jiang, L.; Wu, H.; Wu, C.; Su, J. Effect of aqueous electrolytes on the electrochemical behaviors of supercapacitors based on hierarchically porous carbons. *J. Power Sources* **2012**, *216*, 290–296. [\[CrossRef\]](https://doi.org/10.1016/j.jpowsour.2012.05.090)
- 133. Demarconnay, L.; Raymundo-Piñero, E.; Béguin, F. A symmetric carbon/carbon supercapacitor operating at 1.6 v by using a neutral aqueous solution. *Electrochem. Commun.* **2010**, *12*, 1275–1278. [\[CrossRef\]](https://doi.org/10.1016/j.elecom.2010.06.036)
- 134. Fic, K.; Lota, G.; Meller, M.; Frackowiak, E. Novel insight into neutral medium as electrolyte for high-voltage supercapacitors. *Energy Environ. Sci.* **2012**, *5*, 5842–5850. [\[CrossRef\]](https://doi.org/10.1039/c1ee02262h)
- 135. Li, S.; Qi, L.; Lu, L.; Wang, H. Facile preparation and performance of mesoporous manganese oxide for supercapacitors utilizing neutral aqueous electrolytes. *RSC Adv.* **2012**, *2*, 3298–3308. [\[CrossRef\]](https://doi.org/10.1039/c2ra00991a)
- 136. Krishnan, S.G.; Reddy, M.V.; Harilal, M.; Vidyadharan, B.; Misnon, I.I.; Rahim, M.H.A.; Ismail, J.; Jose, R. Characterization of MgCo2O⁴ as an electrode for high performance supercapacitors. *Electrochim. Acta* **2015**, *161*, 312–321. [\[CrossRef\]](https://doi.org/10.1016/j.electacta.2015.02.081)
- 137. Wang, J.G.; Yang, Y.; Huang, Z.H.; Kang, F. Effect of temperature on the pseudo-capacitive behavior of freestanding MnO2@carbon nanofibers composites electrodes in mild electrolyte. *J. Power Sources* **2013**, *224*, 86–92. [\[CrossRef\]](https://doi.org/10.1016/j.jpowsour.2012.09.075)
- 138. Ramasamy, C.; Del Val, J.P.; Anderson, M. An analysis of ethylene glycol-aqueous based electrolyte system for supercapacitor applications. *J. Power Sources* **2014**, *248*, 370–377. [\[CrossRef\]](https://doi.org/10.1016/j.jpowsour.2013.09.078)
- 139. Abbas, Q.; Pajak, D.; Fra, E.; Béguin, F. Effect of binder on the performance of carbon/carbon symmetric capacitors in salt aqueous electrolyte. *Electrochim. Acta* **2014**, *140*, 132–138. [\[CrossRef\]](https://doi.org/10.1016/j.electacta.2014.04.096)
- 140. Jin, Z.; Yan, X.; Yu, Y.; Zhao, G. Sustainable activated carbon fibers from liquefied wood with controllable porosity for highperformance supercapacitors. *J. Mater. Chem. A* **2014**, *2*, 11706–11715. [\[CrossRef\]](https://doi.org/10.1039/c4ta01413h)
- 141. Liu, X.; Shang, P.; Zhang, Y.; Wang, X.; Fan, Z.; Wang, B.; Zheng, Y. Three-dimensional and stable polyaniline-grafted graphene hybrid materials for supercapacitor electrodes. *J. Mater. Chem. A* **2014**, *2*, 15273–15278. [\[CrossRef\]](https://doi.org/10.1039/c4ta03077j)
- 142. Wang, Q.; Yan, J.; Fan, Z.; Wei, T.; Zhang, M.; Jing, X. Mesoporous polyaniline film on ultra-thin graphene sheets for high performance supercapacitors. *J. Power Sources* **2014**, *247*, 197–203. [\[CrossRef\]](https://doi.org/10.1016/j.jpowsour.2013.08.076)
- 143. Shinde, S.S.; Gund, G.S.; Dubal, D.P.; Jambure, S.B.; Lokhande, C.D. Morphological modulation of polypyrrole thin films through oxidizing agents and their concurrent effect on supercapacitor performance. *Electrochim. Acta* **2014**, *119*, 1–10. [\[CrossRef\]](https://doi.org/10.1016/j.electacta.2013.10.174)
- 144. Zhang, Q.; Li, Y.; Zhu, J.; Lan, L.; Li, C.; Mao, J.; Wang, F.; Zhang, Z.; Wang, L. Ultra-low temperature flexible supercapacitor based on hierarchically structured pristine polypyrrole membranes. *Chem. Eng. J.* **2021**, *420 part 1*, 129712.
- 145. Shukla, A.K.; Sampath, S.; Vijayamohanan, K. Electrochemical supercapacitors: Energy storage beyond batteries. *Curr. Sci.* **2000**, *79*, 1656–1661.
- 146. Wu, Z.S.; Ren, W.; Wen, L.; Gao, L.; Zhao, J.; Chen, Z.; Zhou, G.; Li, F.; Cheng, H.M. Graphene anchored with Co₃O₄ nanoparticles as anode of lithium ion batteries with enhanced reversible capacity and cyclic performance. *ACS Nano* **2010**, *4*, 3187–3194. [\[CrossRef\]](https://doi.org/10.1021/nn100740x)
- 147. Higgins, T.M.; McAteer, D.; Coelho, J.C.M.; Sanchez, B.M.; Gholamvand, Z.; Moriarty, G.; McEvoy, N.; Berner, N.C.; Duesberg, G.S.; Nicolosi, V.; et al. Effect of percolation on the capacitance of supercapacitor electrodes prepared from composites of manganese dioxide nanoplatelets and carbon nanotubes. *ACS Nano* **2010**, *4*, 3187–3194. [\[CrossRef\]](https://doi.org/10.1021/nn5038543)
- 148. Lim, E.; Kim, H.; Jo, C.; Chun, J.; Ku, K.; Kim, S.; Lee, H.I.; Nam, I.S.; Yoon, S.; Kang, K.; et al. Advanced hybrid supercapacitor based on a mesoporous niobium pentoxide/carbon as high-performance anode. *ACS Nano* **2014**, *8*, 8968–8978. [\[CrossRef\]](https://doi.org/10.1021/nn501972w)
- 149. Clare, B.; Sirwardana, A.; MacFarlane, D.R. Synthesis, purification and characterization of ionic liquids. *Top. Curr. Chem.* **2010**, *290*, 1–40. [\[CrossRef\]](https://doi.org/10.1007/128_2008_31) [\[PubMed\]](https://www.ncbi.nlm.nih.gov/pubmed/21107793)
- 150. Yu, X.; Ruan, D.; Wu, C.; Wang, J.; Shi, Z. Spiro-(1,1′)-bipyrrolidinium tetrafluoroborate salt as high voltage electrolyte for electric double layer capacitors. *J. Power Sources* **2014**, *265*, 309–316. [\[CrossRef\]](https://doi.org/10.1016/j.jpowsour.2014.04.144)
- 151. Cai, K.; Mu, W.; He, T.; Hou, J. Investigation of the electrode molding technologies for the carbon-based supercapacitors. *J. Solid State Electrochem.* **2012**, *16*, 2541–2546. [\[CrossRef\]](https://doi.org/10.1007/s10008-012-1657-z)
- 152. Brandt, A.; Isken, P.; Lex-Balducci, A.; Balducci, A. Adiponitrile-based electrochemical double layer capacitor. *J. Power Sources* **2012**, *204*, 213–219. [\[CrossRef\]](https://doi.org/10.1016/j.jpowsour.2011.12.025)
- 153. Francke, R.; Cericola, D.; Kötz, R.; Waldvogel, S.R. Novel electrolytes for electrochemical double layer capacitors based on 1,1,1,3,3,3-hexafluoropropan-2-ol. *Electrochim. Acta* **2012**, *62*, 372–380.
- 154. Väli, R.; Laheäär, A.; Jänes, A.; Lust, E. Characteristics of non-aqueous quaternary solvent mixture and Na-salts based supercapacitor electrolytes in a wide temperature range. *Electrochim. Acta* **2014**, *121*, 294–300. [\[CrossRef\]](https://doi.org/10.1016/j.electacta.2013.12.149)
- 155. Sevilla, M.; Fuertes, A.B. Direct synthesis of highly porous interconnected carbon nanosheets and their application as highperformance supercapacitors. *ACS Nano* **2014**, *8*, 5069–5078. [\[CrossRef\]](https://doi.org/10.1021/nn501124h)
- 156. Han, J.; Xu, G.; Ding, B.; Pan, J.; Dou, H.; MacFarlane, D.R. Porous nitrogen-doped hollow carbon spheres derived from polyaniline for high performance supercapacitors. *J. Mater. Chem. A* **2014**, *2*, 5352–5357. [\[CrossRef\]](https://doi.org/10.1039/c3ta15271e)
- 157. Perricone, E.; Chamas, M.; Cointeaux, L.; Leprêtre, J.C.; Judeinstein, P.; Azais, P.; Béguin, F.; Alloin, F. Investigation of methoxypropionitrile as co-solvent for ethylene carbonate based electrolyte in supercapacitors. A safe and wide temperature range electrolyte. *Electrochim. Acta* **2013**, *93*, 1–7. [\[CrossRef\]](https://doi.org/10.1016/j.electacta.2013.01.084)
- 158. Jung, N.; Kwon, S.; Lee, D.; Yoon, D.M.; Park, Y.M.; Benayad, A.; Choi, J.Y.; Park, J.S. Synthesis of chemically bonded graphene/carbon nanotube composites and their application in large volumetric capacitance supercapacitors. *Adv. Mater.* **2013**, *25*, 6854–6858. [\[CrossRef\]](https://doi.org/10.1002/adma.201302788) [\[PubMed\]](https://www.ncbi.nlm.nih.gov/pubmed/24105733)
- 159. Jänes, A.; Eskusson, J.; Thomberg, T.; Lust, E. Supercapacitors based on propylene carbonate with small addition of different sulfur containing organic solvents. *J. Electrochem. Soc.* **2014**, *16*, A1284–A1290. [\[CrossRef\]](https://doi.org/10.1149/2.0691409jes)
- 160. Zhang, H.; Wang, J.; Chen, Y.; Wang, Z.; Wang, S. Long-term cycling stability of polyaniline on graphite electrodes used for supercapacitors. *Electrochim. Acta* **2013**, *105*, 69–74. [\[CrossRef\]](https://doi.org/10.1016/j.electacta.2013.04.114)
- 161. Sun, W.; Wang, Y. Graphene-based nanocomposite anodes for lithium-ion batteries. *Nanoscale* **2014**, *6*, 11528–11552. [\[CrossRef\]](https://doi.org/10.1039/c4nr02999b)
- 162. Yiğit, D.; Güllü, M.; Yumak, T.; Sinağ, A. Heterostructured poly(3,6-dithien-2-yl-9H-carbazol-9-yl acetic acid)/TiO₂ nanoparticles composite redox-active materials as both anode and cathode for high-performance symmetric supercapacitor applications. *J. Mater. Chem. A* **2014**, *2*, 6512–6524. [\[CrossRef\]](https://doi.org/10.1039/c4ta00109e)
- 163. Chen, C.; Eichel, R.A.; Notten, P.H.L. Ionic conductivity of metal oxides: An essential property for all-solid-state lithium-ion batteries. In *Metal Oxide-Based Thin Film Structures: Formation, Characterization and Application of Interface-Based Phenomena*; Pryds, N., Esposito, V., Eds.; Elsevier: Amsterdam, The Netherlands, 2018; pp. 361–408. [\[CrossRef\]](https://doi.org/10.1016/B978-0-12-811166-6.00016-9)
- 164. Łatoszyńska, A.A.; Zukowska, G.Z.; Rutkowska, I.A.; Taberna, P.L.; Simon, P.; Kulesza, P.J.; Wieczorek, W. Non-aqueous gel polymer electrolyte with phosphoric acid ester and its application for quasi solid-state supercapacitors. *J. Power Sources* **2015**, *274*, 1147–1154. [\[CrossRef\]](https://doi.org/10.1016/j.jpowsour.2014.10.094)
- 165. Fan, L.Q.; Zhong, J.; Wu, J.H.; Lin, J.M.; Huang, Y.F. Improving the energy density of quasi-solid-state electric double-layer capacitors by introducing redox additives into gel polymer electrolytes. *J. Mater. Chem. A* **2014**, *2*, 9011–9014. [\[CrossRef\]](https://doi.org/10.1039/c4ta01408a)
- 166. Fei, H.; Yang, C.; Bao, H.; Wang, G. Flexible all-solid-state supercapacitors based on graphene/carbon black nanoparticle film electrodes and cross-linked poly(vinyl alcohol)-H2SO⁴ porous gel electrolytes. *J. Power Sources* **2014**, *266*, 488–495. [\[CrossRef\]](https://doi.org/10.1016/j.jpowsour.2014.05.059)
- 167. Ko, J.M.; Nam, J.H.; Won, J.H.; Kim, K.M. Supercapacitive properties of electrodeposited polyaniline electrode in acrylic gel polymer electrolytes. *Synth. Met.* **2014**, *266*, 488–495. [\[CrossRef\]](https://doi.org/10.1016/j.synthmet.2014.01.011)
- 168. Nam, H.-S.; Wu, N.-L.; Lee, K.-T.; Kim, K.M.; Yeom, C.G.; Hepowit, L.R.; Ko, J.M.; Kim, J.-D. Electrochemical capacitances of a nanowire-structured MnO2 in polyacrylate-based gel electrolytes. *J. Electrochem. Soc.* **2012**, *159*, A899–A903. [\[CrossRef\]](https://doi.org/10.1149/2.112206jes)
- 169. Zhong, C.; Deng, Y.; Hu, W.; Sun, D.; Han, X.; Qiao, J.; Zhang, J. Electrolytes for electrochemical supercapacitors. In *Electrolytes for Electrochemical Supercapacitors*; Zhong, C., Deng, Y., Hu, W., Sun, D., Han, X., Qiao, J., Zhang, J., Eds.; CRC Press: Boca Raton, FL, USA, 2016. [\[CrossRef\]](https://doi.org/10.1201/b21497-6)
- 170. Pal, B.; Yang, S.; Ramesh, S.; Thangadurai, V.; Jose, R. Electrolyte selection for supercapacitive devices: A critical review. *Nanoscale Adv.* **2019**, *1*, 3807–3835.
- 171. Yuksel, R.; Sarioba, Z.; Cirpan, A.; Hiralal, P.; Unalan, H.E. Transparent and flexible supercapacitors with single walled carbon nanotube thin film electrodes. *ACS Appl. Mater. Interfaces.* **2014**, *6*, 15434–15439. [\[CrossRef\]](https://doi.org/10.1021/am504021u)
- 172. Chiu, K.F.; Su, S.H. Lithiated and sulphonated poly(ether ether ketone) solid state electrolyte films for supercapacitors. *Thin Solid Films* **2013**, *544*, 144–147. [\[CrossRef\]](https://doi.org/10.1016/j.tsf.2013.03.135)
- 173. Hsueh, M.F.; Huang, C.W.; Wu, C.A.; Kuo, P.L.; Teng, H. The synergistic effect of nitrile and ether functionalities for gel electrolytes used in supercapacitors. *J. Phys. Chem. C* **2013**, *1*, 3807–3835. [\[CrossRef\]](https://doi.org/10.1021/jp4031128)
- 174. Syahidah, S.N.; Majid, S.R. Super-capacitive electro-chemical performance of polymer blend gel polymer electrolyte (GPE) in carbon-based electrical double-layer capacitors. *Electrochim. Acta* **2013**, *112*, 678–685. [\[CrossRef\]](https://doi.org/10.1016/j.electacta.2013.09.008)
- 175. Francisco, B.E.; Jones, C.M.; Lee, S.H.; Stoldt, C.R. Nanostructured all-solid-state supercapacitor based on Li 2S-P 2S 5 glassceramic electrolyte. *Appl. Phys. Lett.* **2012**, *100*, 103902. [\[CrossRef\]](https://doi.org/10.1063/1.3693521)
- 176. Fic, K.; Frackowiak, E.; Béguin, F. Unusual energy enhancement in carbon-based electrochemical capacitors. *J. Mater. Chem.* **2012**, *22*, 24213–24223. [\[CrossRef\]](https://doi.org/10.1039/c2jm35711a)
- 177. Zhang, L.; Yang, S.; Chang, J.; Zhao, D.; Wang, J.; Yang, C.; Cao, B. A review of redox electrolytes for supercapacitors. *Front. Chem.* **2020**, *8*, 1–7. [\[CrossRef\]](https://doi.org/10.3389/fchem.2020.00413)
- 178. Suárez-Guevara, J.; Ruiz, V.; Gomez-Romero, P. Hybrid energy storage: High voltage aqueous supercapacitors based on activated carbon-phosphotungstate hybrid materials. *J. Mater. Chem. A* **2014**, *2*, 1014–1021. [\[CrossRef\]](https://doi.org/10.1039/c3ta14455k)
- 179. Tian, Q.; Lian, K. In situ characterization of heteropolyacid based electrochemical capacitors. *Electrochem. Solid State Lett.* **2010**, *13*, 4. [\[CrossRef\]](https://doi.org/10.1149/1.3247071)
- 180. Zhang, M.; Zhou, Q.; Chen, J.; Yu, X.; Huang, L.; Li, Y.; Li, C.; Shi, G. An ultrahigh-rate electrochemical capacitor based on solution-processed highly conductive PEDOT:PSS films for AC line-filtering. *Energy Environ. Sci.* **2016**, *9*, 2005–2010. [\[CrossRef\]](https://doi.org/10.1039/c6ee00615a)
- 181. Chen, W.; Rakhi, R.B.; Alshareef, H.N. Capacitance enhancement of polyaniline coated curved-graphene supercapacitors in a redox-active electrolyte. *Nanoscale* **2013**, *5*, 4134–4138. [\[CrossRef\]](https://doi.org/10.1039/c3nr00773a)
- 182. Maiti, S.; Pramanik, A.; Mahanty, S. Interconnected network of MnO₂ nanowires with a "cocoonlike" morphology: Redox couplemediated performance enhancement in symmetric aqueous supercapacitor. *ACS Appl. Mater. Interfaces.* **2014**, *6*, 10754–10762. [\[CrossRef\]](https://doi.org/10.1021/am502638d)
- 183. Liao, H.; Zhong, W.; Li, T.; Han, J.; Sun, X.; Tong, X.; Zhang, Y. A review of self-healing electrolyte and their applications in flexible/stretchable energy storage devices. *Electrochim. Acta* **2022**, *404*, 139730.
- 184. Zhang, Z.; Gao, Y.; Gao, Y.; Jia, F.; Gao, G. A self-adhesive, self-healing zwitterionic hydrogel electrolyte for high-voltage zinc-ion hybrid supercapacitors. *Chem. Eng. J.* **2023**, *452*, 139014.
- 185. Lv, H.; Chu, X.; Zhang, Y.; Liu, Q.; Wu, F.; Mu, D. Self-healing solid-state polymer electrolytes for high-safety and long-cycle lithium-ion batteries. *Mater. Today* 2024. [\[CrossRef\]](https://doi.org/10.1016/j.mattod.2024.06.018)
- 186. Preethichandra, D.M.G.; Sonar, P. Electrochemical impedance spectroscopy and its applications in sensor development and measuring battery performance. *IEEE Sens. J.* **2022**, *22*, 10152–10162. [\[CrossRef\]](https://doi.org/10.1109/JSEN.2021.3119650)
- 187. Kanoun, O. (Ed.) *Impedance Spectroscopy: Advanced Applications: Battery Research, Bioimpedance, System Design*; De Gruyter: Berlin, Germany, 2018; p. 8.
- 188. Bard, A.J.; Faulkner, L.R. (Eds.) *Electrochemical Methods: Fundamentals and Applications*, 2nd ed.; Wiley: Hoboken, NJ, USA, 2000.
- 189. *White Paper:Testing Super-Capacitors Application Note Rev. 2.0 5/31/2018 Copyright 1990–2018*; Gamry Instruments, Inc.: Warminster, PA, USA, 2018.
- 190. Sethi, M.; Shenoy, U.S.; Bhat, D.K. A porous graphene-NiFe₂O₄ nanocomposite with high electrochemical performance and high cycling stability for energy storage applications. *Nanoscale Adv.* **2020**, *2*, 4229–4241. [\[CrossRef\]](https://doi.org/10.1039/d0na00440e) [\[PubMed\]](https://www.ncbi.nlm.nih.gov/pubmed/36132772)
- 191. Yagüe, J.L.; Guimerà, A.; Villa, R.; Agulló, N.; Borrós, S. A new four-point probe design to measure conductivity in polymeric thin films. *Afinidad* **2013**, *70*, 166–169.
- 192. Bellamy, L.J. *The Infra-Red Spectra of Complex Molecules*, 3rd ed.; Chapman and Hall: London, UK, 1975; Volume 1.
- 193. Infrared: Interpretation, LibreTexts—Chemistry. Available online: [https://chem.libretexts.org/Bookshelves/Physical_](https://chem.libretexts.org/Bookshelves/Physical_and_Theoretical_Chemistry_Textbook_Maps/Supplemental_Modules_(Physical_and_Theoretical_Chemistry)/Spectroscopy/Vibrational_Spectroscopy/Infrared_Spectroscopy/Infrared:_Interpretation) [and_Theoretical_Chemistry_Textbook_Maps/Supplemental_Modules_\(Physical_and_Theoretical_Chemistry\)/Spectroscopy/](https://chem.libretexts.org/Bookshelves/Physical_and_Theoretical_Chemistry_Textbook_Maps/Supplemental_Modules_(Physical_and_Theoretical_Chemistry)/Spectroscopy/Vibrational_Spectroscopy/Infrared_Spectroscopy/Infrared:_Interpretation) [Vibrational_Spectroscopy/Infrared_Spectroscopy/Infrared:_Interpretation](https://chem.libretexts.org/Bookshelves/Physical_and_Theoretical_Chemistry_Textbook_Maps/Supplemental_Modules_(Physical_and_Theoretical_Chemistry)/Spectroscopy/Vibrational_Spectroscopy/Infrared_Spectroscopy/Infrared:_Interpretation) (accessed on 8 August 2024).
- 194. Raja, P.M.V.; Barron, A.R. Physical methods in Chemistry and Nano Science. LibreTexts, 7 September 2023. Available online: <https://chem.libretexts.org/@go/page/55831> (accessed on 5 August 2024).
- 195. Mohammed, A.; Abdullah, A. Scanning electron microscopy (SEM): A review. In Proceedings of the 2018 International Conference on Hydraulics and Pneumatics—HERVEX, Băile Govora, Romania, 7–9 November 2018; pp. 77–85.
- 196. Tundwal, A.; Kumar, H.; Binoj, B.J.; Sharma, R.; Kumar, G.; Kumari, R.; Dhayal, A.; Yadav, A.; Singh, D.; Kumar, P. Developments in conducting polymer-, metal oxide-, and carbon nanotube-based composite electrode materials for supercapacitors: A review. *RSC Adv.* **2024**, *14*, 9406–9439. [\[PubMed\]](https://www.ncbi.nlm.nih.gov/pubmed/38516158)

Disclaimer/Publisher's Note: The statements, opinions and data contained in all publications are solely those of the individual author(s) and contributor(s) and not of MDPI and/or the editor(s). MDPI and/or the editor(s) disclaim responsibility for any injury to people or property resulting from any ideas, methods, instructions or products referred to in the content.