

Comparative *In Vivo* Biocompatibility of Cellulose-Derived and Synthetic Meshes in Subcutaneous Transplantation Models

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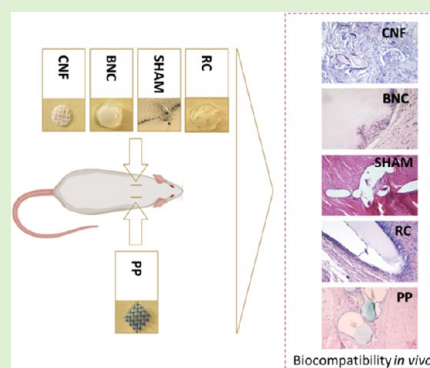


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ABSTRACT: Despite the increasing interest in cellulose-derived materials in biomedical research, there remains a significant gap in comprehensive *in vivo* analyses of cellulosic materials obtained from various sources and processing methods. To explore durable alternatives to synthetic medical meshes, we evaluated the *in vivo* biocompatibility of bacterial nanocellulose, regenerated cellulose, and cellulose nanofibrils in a subcutaneous transplantation model, alongside incumbent polypropylene and polydioxanone. Notably, this study demonstrates the *in vivo* biocompatibility of regenerated cellulose obtained through alkali dissolution and subsequent regeneration. All cellulose-derived implants triggered the expected foreign body response in the host tissue, characterized predominantly by macrophages and foreign body giant cells. Porous materials promoted cell ingrowth and biointegration. Our results highlight the potential of bacterial nanocellulose and regenerated cellulose as safe alternatives to commercial polypropylene meshes. However, the *in vivo* fragmentation observed for cellulose nanofibril meshes suggests the need for measures to optimize their processing and preparation.



INTRODUCTION

The biomedical field is on a quest for sustainable and biocompatible materials that integrate seamlessly with the human body.^{1,2} With the increasing prominence of such materials, extensive research has focused on understanding the complex interplay between implanted biomaterials and host tissue.^{3,4} The ideal postimplantation scenario anticipates these materials to induce a transient inflammatory response,⁴ which aids the body's long-term healing mechanisms without causing undesirable short- or long-term complications.⁵

Historically, synthetic meshes, particularly those composed of polypropylene (PP), have led surgical treatments, such as hernia and pelvic organ prolapse repairs, since the 1950s.^{6–8} However, the widespread use of PP meshes is now being challenged or halted due to inconsistencies in outcomes,^{6,7,9} including significant patient-reported complications and adverse reactions.^{7,10–12} Regulatory interventions, such as the market removal of certain PP mesh products,¹³ highlight these concerns. Nevertheless, the search for the ideal mesh—combining biocompatibility, mechanical durability, and resistance—continues.^{5,7,14,15}

Cellulose is frequently noted for its economic viability and inherent nontoxicity.^{1,16,17} As the scientific community gravitates toward sustainable methodologies,^{18–21} cellulose emerges as a promising material. However, despite its broad consideration in the biomedical field, there remains a notable gap in the literature regarding in-depth *in vivo* analyses of

different cellulose forms, especially those derived from diverse sources and processing methods. These issues are crucial in evaluating cellulose's potential to replace PP meshes in biomedical applications.

Bacterial nanocellulose (BNC), regenerated cellulose (RC), and cellulose nanofibrils (CNFs) represent distinct cellulose structures, each derived from different sources and production methods, and therefore, exhibit inherently different properties. BNC, produced by specific bacterial genera such as *Komagataeibacter*,^{1,21,22} forms nanofibril networks, which are promising for their biocompatibility, moisture retention, and suitability for biomedical applications, including skin, bone, and vascular grafts,^{23–25} as well as wound dressings.^{26–28}

In contrast, RC is traditionally sourced from wood fibers,²⁹ agricultural^{30,31} or forestry streams,³² which are dissolved and then regenerated.^{2,33–38} RC is commonly used in the textile industry, where it is valued for its mechanical durability in the form of filaments, films and wearables.¹ Lastly, CNF, extracted from plant cell walls, offers high mechanical strength and a high surface area due to its high level of deconstruction.^{16,17}

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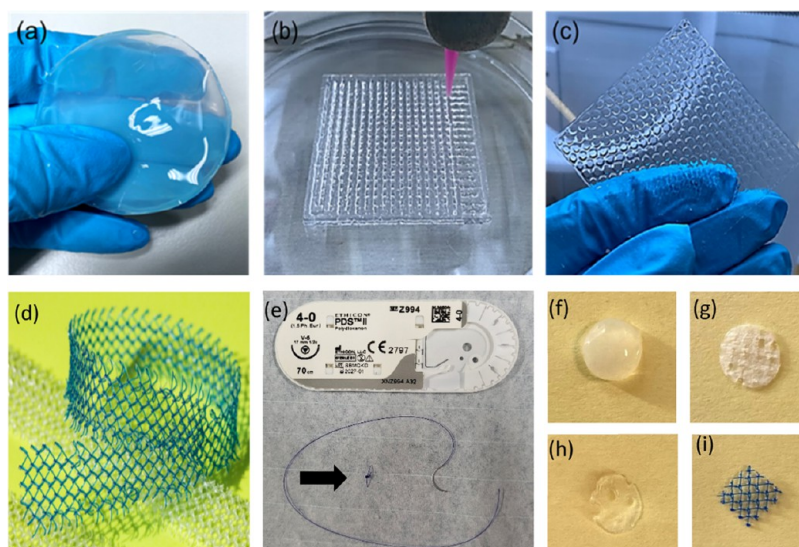


Figure 1. Cellulose-derived samples in wet condition: (a) bacterial nanocellulose (BNC), (b) 3D-printed cellulose nanofibrils (CNF), (c) regenerated cellulose (RC). (d) Commercial polypropylene (PP) based mesh (Photograph by Kalle Kataila, Aalto University). (e) Polydioxanone (PDS, Johnson & Johnson, New Brunswick, NJ). Standardized 8 mm samples: (f) BNC, (g) 3D-CNF, (h) RC, (i) PP. Black arrow = PDS suture.

This makes CNF a suitable candidate for creating customizable structures with adjustable shape, porosity, and microstructure.^{39–44}

To the best of our knowledge, no *in vivo* biocompatibility studies have been published for alkali-dissolved (NaOH/H₂O, ZnO) RC, nor has there been an analysis comparing the main features of BNC, RC, CNF, and PP under the same conditions. Therefore, we assessed the *in vivo* biocompatibility of cellulose-derived meshes using a rat subcutaneous implantation model with 80 healthy female Sprague–Dawley rats.⁴⁵ Acknowledging the limitations in achieving identical material structures across the tested cellulose forms (e.g., geometry, porosity, density), we aimed to test the hypothesis that BNC, RC, and CNF—due to their different sources, production methods, and resulting unique properties—could potentially replace PP, introducing sustainable, versatile, and biocompatible meshes for biomedical applications.

MATERIALS AND METHODS

Materials. Figure 1 displays images of all samples tested in this study before cutting. They included commercial PP meshes, implants made from different cellulose forms and polydioxanone (PDO) used as a sham procedure.

Polypropylene (PP). Gynecare TVT Exact PP mesh was obtained in a sterile package from Johnson & Johnson (New Brunswick, NJ). The mesh was cut, autoclaved (at 121 °C for 15 min), and utilized as control samples for *in vivo* tests (Figure 1d,i).

Bacterial Nanocellulose (BNC). The strain used for BNC production, *Komagataeibacter medellinensis*, was provided by the School of Engineering, Universidad Pontificia Bolivariana, Colombia.⁴⁶ D-(+)-glucose, sodium phosphate dibasic (Na₂HPO₄), peptone, yeast extract, citric acid, sodium bromide, sodium hypochlorite, and sodium hydroxide were purchased from Sigma-Aldrich (St. Louis, MO). Milli-Q water (purified using a Millipore Synergy UV unit, Burlington, MA) was used throughout the experiments (18.2 MΩ cm).

First, glucose, yeast extract, peptone, and Na₂HPO₄ were mixed in a dry mass ratio of 8:2:2:1. Milli-Q water was then added to achieve a final volume of 1 L, containing 20 g of glucose, 5 g of yeast extract, 5 g of peptone, and 2.5 g of Na₂HPO₄. All components were fully dissolved, and the pH of the medium was adjusted to 4.5 with citric acid. The container was sterilized in an autoclave at 121 °C for 15 min

and then cooled to room temperature. The bacterial strain was added to the culture medium and gently shaken to ensure homogeneity. The BNC culture medium was poured into sterilized containers and incubated at 28 °C for 10 days.

The resulting BNC biofilm was washed several times with deionized water and left in it for 24 h, with the water changed several times daily to remove residual components from the growth medium. The bacterial cellulose pellicle was then purified with 0.1 M NaOH at 60 °C for 4 h, followed by several washes in hot deionized water for 6 h.⁴⁷ Finally, the BNC was stored in autoclaved Milli-Q water at 4 °C.

Regenerated Cellulose (RC). The cellulose source used was Avicel PH-101 microcrystalline cellulose (50 μm particle size, degree of polymerization 400), purchased from Sigma-Aldrich, Ireland, and used without modifications. Sodium hydroxide (NaOH, purity 99.6%) from VWR Chemicals and zinc oxide (ZnO, pro-analysis grade) from Sigma-Aldrich were used to prepare the cellulose solvent. Sulfuric acid (H₂SO₄) and sodium sulfate (Na₂SO₄), both reagent grade from Sigma-Aldrich were used to prepare the regeneration bath. All solutions were prepared using Milli-Q water (Millipore Corporation, resistivity 18 MΩ·cm).

Cellulose (7% w/w), previously dried under vacuum (200 mbar, 60 °C, 12 h), was dissolved in a 2.3 M NaOH solution with the addition of ZnO, maintaining a ZnO/NaOH mass ratio of 0.167. This cellulose was added to the precooled NaOH/ZnO aqueous solution at −5 °C. The mixture was stirred at 300–700 rpm in a vessel with a cooling jacket using a water/propylene glycol 1:1 mixture, maintaining a temperature of −5 °C for 24 h. After stirring, an opaque and viscous solution was obtained, which was then frozen at −17 °C for 12 h. As reported,³⁸ this freezing step enhances the interaction between NaOH hydrated shells and the reactive hydroxyl groups of cellulose, facilitating dissolution. Upon thawing,¹⁹ a transparent and fully dissolved ink was obtained.

The cellulose ink was used immediately after thawing to produce the RC films using ink extrusion with a nozzle gauge and infill density that were carefully controlled. Extrusion was performed using a 250 μm diameter nozzle at an extrusion pressure ranging from 6–15 kPa and printing speed of 11 mm/s. The films were extruded on glass Petri dishes to obtain samples of 1 mm thickness with a 75% infill density. This process involved repeated injections of the RC ink along the horizontal plane, leading to merging and consolidation of individual printed lines into a single, uniform surface. The printed film patches were coagulated for 1 h in a coagulation solvent composed of 10% w/w H₂SO₄ and 10% w/w Na₂SO₄, washed with

Milli-Q water until achieving a stable pH, autoclaved, and stored in Milli-Q water at 4 °C.

The use of low-viscosity RC inks, with cellulose concentration below 12 wt % in alkali, allowed to produce smooth and even films at room temperature. As demonstrated in our previous studies,¹⁷ these low-concentration inks are ideal for creating uniform films showing no visible surface textures.

3D-Printed Cellulose Nanofibrils (CNFs). TEMPO-oxidized cellulose nanofibrils (TOCNF) were produced using TEMPO-mediated oxidation, followed by the disintegration of never-dried, fines-free, and fully bleached hardwood (birch) fibers sourced from a Finnish pulp mill, as described previously.⁴³ The cellulose fibers were immersed in Milli-Q water with the addition of 0.013 mmol/g TEMPO and 0.13 mmol/g sodium bromide. Subsequently, 5 mmol/g sodium hypochlorite was added, and the pH was adjusted to 10 using 0.1 M sodium hydroxide. The mixture was maintained at room temperature and stirred for approximately 6 h. The resulting fibers were washed several times with deionized water until a neutral pH was achieved. The fibers were then fibrillated using a microfluidizer (M-110P, Microfluidics Inc., Newton, MA) at a pressure of 1400 bar with a single pass.

A BIOX bioprinter from CELLINK (Sweden) equipped with pneumatic multiprintheads was used to 3D print lattice structures, as shown in Figure 1b. The TOCNF was placed into 3 mL clear pneumatic syringes and extruded through a 22-gauge (410 μm diameter) sterile blunt needle. Lattice structures were printed on a 100 mm diameter glass Petri dish, subjected to UV-sterilization and autoclaving at 121 °C for 15 min following freeze-drying to ensure complete sterilization.

Polydioxanone (PDO). A monofilament, slowly resorbable suture⁴⁸ (PDS, Johnson & Johnson, New Brunswick, NJ) was used as a sham procedure (Figure 1e).

Design Criteria. All implants were standardized to a uniform size of 8 mm diameter. Circular samples were cut from cellulose-derived meshes using a sterile biopsy punch, and square samples were cut from PP (Figure 1f–i). The CNF and RC implants were 3D-printed with similar thickness and infill density to replicate the structure of the PP-based control mesh. Although the overall geometries were consistent, small variations in the viscosity of RC and CNF inks resulted in differences in interlayer adhesion and void coalescence, subsequently leading to small square infill patterns that merged and round out into circular features.

Material Characterization. The characterizations of each material were thoroughly examined by the authors in previous publications.^{47,49,50} All cellulose-derived materials used in the study have similar tensile strength and high wettability, measured by the water contact angles (less than 90°).⁵¹ RC films presented an elastic modulus ranging from 4.5 to 8 GPa and a tensile strength of 120–170 MPa.⁴⁹ Those for CNF and BNC corresponded to 10 and 10–18 GPa for the elastic modulus and 200–300, and 200–250 MPa for the tensile strength, respectively.^{47,52}

TEMPO-oxidized CNF has a surface charge of about 1.36 mmol_{COOH}/g.⁵³ The introduction of carboxyl groups increases hydrophilicity and enhances cell adhesion, proliferation, and compatibility with host tissue.⁵⁴ Indeed, hydrophilic surfaces generally promote better interaction with cells and reduce adverse immune responses.¹⁷ Based on XRD, FTIR, and GPC analyses,^{44,47,49,53} RC is primarily composed of cellulose type I while the CNF and BNC cellulose correspond to the cellulose polymorph. For BNC, trace amounts of proteins were also detected, which is not expected to impact the materials' performance *in vivo*.⁴⁷

Methods. Scanning Electron Microscopy (SEM). The microstructure of the surface and cross-section of implants were observed using a scanning electron microscope (Carl Zeiss AG, Oberkochen, Germany) operated at an accelerating voltage of 2 kV. Samples were fixed on metal stubs with double-sided carbon tape and sputter-coated with a 3–4 nm layer of gold–palladium alloy using a LEICA EM ACE600 (Leica Camera AG, Wetzlar, Germany) sputter coater.

Animal Tests. A total of 80 healthy female Sprague–Dawley rats, aged 10 to 12 weeks, were used in this study. Based on previous

literature^{55,56} and the principles of the 3Rs,⁵⁷ a sample size of 10 animals per group was calculated to achieve 80% power to detect an effect size of 1 with a standard deviation of 0.8 in the histological scoring of the inflammatory host response between groups, using a semiquantitative scoring system.⁵⁸ An α level of 0.05 was assumed, and group comparisons were performed using the Wilcoxon signed-rank test. The rats were housed in pairs under equal conditions with free access to food, water, and environmental stimuli. They were acclimatized for 1 week before the study and provided with weekly access to a large activity cage during the follow-up period as part of the refinement. The experimental design was approved by the National Animal Ethics Committee (ESAVI/4488/2021). Animal housing adhered to the European Directive (Directive 2010/63/EU) and Finnish legislation (497/2013).

Procedures. Each rat was implanted with two subcutaneous implants: PP as a control on the left side of the spine, and either BNC, RC, CNF or PDO (PDS, Johnson & Johnson, New Brunswick, NJ) as a sham procedure, on the right side (Table 1). Prior to

Table 1. Experimental Groups in a Rat Subcutaneous Transplantation Model

group	animals/group	implant left	implant right	suture material ^a
1	20	PP ^b	BNC ^c	4–0 PDO ^d
2	20	PP ^b	RC ^e	4–0 PDO ^d
3	20	PP ^b	CNF ^f	no
4	20	PP ^b	SHAM ^d	4–0 PDO ^d

^aImplants were secured to the underlying tissue with one suture.

^bPolypropylene (PP). ^cBacterial nanocellulose (BNC). ^dPolydioxanone (PDS, Johnson & Johnson, New Brunswick, NJ). ^eRegenerated cellulose (RC). ^f3D-printed cellulose nanofibrils (CNFs).

induction of anesthesia, the rats were weighed and premedicated with buprenorphine. Anesthesia was induced in an induction chamber with 4.5% isoflurane and 600 mL/min air flow. Upon loss of consciousness, the rat was transferred to a mask with 2–2.3% isoflurane and 600 mL/min air flow, then positioned ventrally on a heated plate (37 °C). The back was shaved and disinfected with 80% alcohol, following standard aseptic protocols.

Two vertical incisions (1–1.5 cm) were made through the skin, 2–3 cm on each side of the spine caudal to the scapulae, and subcutaneous pockets were created by blunt dissection. Each mesh, except for the fragile CNF, was secured to the underlying tissue with one suture of a resorbable, monofilament suture (4–0 PDO, PDS, Johnson & Johnson, New Brunswick, NJ) to prevent implant migration and ease identification at excision (Table 1). The skin was closed with simple interrupted 4–0 poliglecaprone sutures (Monocryl, Johnson & Johnson, New Brunswick, NJ) (Figure 2). Each rat was uniquely marked for individual identification using a specific combination of earmarks made by an ear punch.

After surgery, the rats were monitored and kept on a 37 °C plate until they fully recovered. For pain relief, they received subcutaneous injections of carprofen (Rimadyl, Zoetis Animal Health, Denmark), a nonsteroidal anti-inflammatory drug, administered before surgery and continued for 2 days postoperatively. The rats were checked twice daily for the first 7 days after surgery, and subsequently, once daily until euthanasia. General behavior and movement were assessed on a scale of 0–2 adapted from Carstens & Moberg.⁵⁹ 0 indicating normal behavior and movement, 1 indicating ungroomed appearance, partial piloerection, hyporexia, or abnormal stance, and 2 indicating severe piloerection, anorexia, or recumbency. Pain levels were evaluated using the Rat Grimace Scale,⁶⁰ also on a scale of 0–2. Wounds were evaluated on a scale of 0–5⁶¹ and photographed on days 1, 3, 5, 7, 30, and 90.

Follow-Up, Macroscopic Evaluation, and Tissue Collection. Follow-up assessments were conducted at one and three months, with ten samples of each material evaluated at each time point. At the conclusion of the follow-up period, the rats were euthanized using CO₂ asphyxiation, followed by decapitation. After euthanasia, the scar

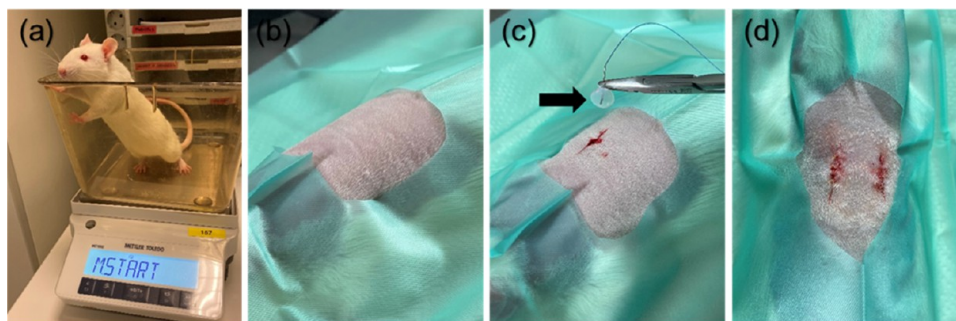


Figure 2. Implantation procedure. (a) Weighing of a rat prior to anesthesia. (b) An anesthetized rat at ventral recumbency on a heated plate (37 °C) with back shaved and disinfected with 80% alcohol. (c) Two vertical incisions were made on each side of the spine caudal to the scapulae, a subcutaneous pocket created by blunt dissection, and an implant (bacterial nanocellulose, BNC) secured to the underlying tissue. (d) Routine skin closure with simple interrupted sutures. *Black arrow* = BNC implant.

and surrounding tissue were assessed for local inflammation, infection, seroma, or abscess formation, graded on a scale of 0 to 1 (0 = absent, 1 = present).

The implanted mesh, or PDO as a sham procedure, was excised with 3 mm margins, including skin, subcutaneous tissue, and muscle. Tissue samples were fixed in 4% paraformaldehyde and processed for paraffin embedding using an automated tissue processor (Tissue Tek VIP, Sakura Finetek, Torrance, CA). Samples were then sectioned longitudinally into 5 μ m slices with a microtome (Microm cool-cut HM 355S, Thermo Scientific, Kalamazoo, MI) for histological analysis. The sections were stained with hematoxylin and eosin (H&E) following standard procedures,⁶² and assigned coded identification numbers.

Histopathology. Histological assessment was conducted blindly by our experienced pathologist (H.S. author) and a veterinary surgeon (N.M.M.P. author). A customized scale adapted from a semi-quantitative scoring system (EN ISO 10993-6 Annex E)⁵⁸ was developed to differentiate the degree of various host tissue responses. The biocompatibility of the implanted mesh materials was assessed by analyzing the host inflammatory response, including cellularity, foreign body reaction (FBR), and cell ingrowth. The integral classification score comprised the presence and for foreign body reaction, the number of different cell types per unit area within the implanted mesh and the tissue-implant interface. Specific evaluations included:

- **Foreign Body Reaction (FBR):** Presence of macrophages and foreign body giant cells (FBGCs) (Table 2).

Table 2. Foreign Body Reaction to an Implant Adapted from EN ISO 10993-6 Annex E⁵⁸

cell type/response	score				
	0	1	2	3	4
foreign body reaction					
macrophages	0	1–40/hpf ^a	40–80/hpf ^a	heavy infiltrate, 80–180/hpf ^a	packed, >180/hpf ^a
foreign body giant cells	0	1–2/hpf ^a	3–5/hpf ^a	5–10/hpf ^a	sheets, >10/hpf ^a

^aAbbreviations: high-power (400 \times) field.

- **Acute Inflammation:** Presence of neutrophils.
- **Chronic Inflammation:** Presence of lymphocytes.
- **Granulation Tissue:** Presence of plasma cells, sporadic lymphocytes, and granulocytes.
- **Scar Formation:** Presence of fibrosis.

Granulation tissue, characterized by proliferating capillaries and variable inflammatory reactions, was noted to regress and transform into scar tissue with reduced cellularity. These histological character-

istics were evaluated in at least three sections per tissue sample and double-counted. Scoring was performed on a scale from 0 to 4 for FBR (0 = absent, 1 = mild, 2 = moderate, 3 = heavy infiltrate, 4 = packed)⁵⁸ (Table 2) and from 0 to 1 for acute and chronic inflammation, granulation tissue, and scar formation (0 = absent, 1 = present). The thickness of fibrosis was evaluated as recommended by EN ISO 10993-6 Annex E⁵⁸ (0 = absent, 1 = narrow band = 1–2 cell layers in thickness, 2 = moderately thick band = < 10 cell layers in thickness, 3 = thick, contiguous band along length of tissue, 4 = extensive, thick zone with effacement of local architecture).

Statistical Analysis. A proportional-odds cumulative logit model was employed, utilizing mesh type, time point, and their interaction as fixed factors, to compare the severity of tissue reactions among different explants. This model assessed the probability of higher severity in tissue reactions. Odds ratios with 95% confidence intervals were computed to quantify differences relative to the PP mesh.

To further evaluate tissue reaction severity within individual rats across different meshes (or sham for PDO), within-rat differences were calculated compared to the PP mesh. Descriptive tabulation of these differences was initially performed by mesh type (or sham), followed by formal analysis using a cumulative logit model with mesh as the sole fixed factor. Pairwise odds ratios derived from this model facilitated comparisons between meshes. Statistical significance was defined as $p < 0.05$. All analyses were conducted using SAS software, version 9.4 (SAS Institute Inc., Cary, NC).

RESULTS AND DISCUSSION

PP and RC Meshes Are Less Porous than Those Made from BNC and CNF. Surface and cross-sectional analyses of the meshes were conducted using SEM prior to implantation to investigate their porosity, morphology, and microstructural differences. The porosity of the solid phase of the cellulose-derived meshes in cross-section SEM images indicated the highest porosity for CNF and the smallest for RC (Figure 3). BNC, characterized by physical entanglement, exhibited a relatively dense surface network while displaying a highly porous layered cross-section. These findings align with previous studies on fibrillated cellulose.¹⁶

CNF showed larger openings and pores on both its surface and cross-section, resulting in a rougher texture compared to BNC, RC, and PP. Previous research on porous nanocelluloses has indicated that myoblast cells tend to attach less effectively to surfaces with higher roughness.⁴⁴ However, attachment behavior can vary significantly depending on factors such as cell type and surface charge. In contrast, RC, following regeneration and coagulation, exhibited fewer pores and formed a more compact microstructure that resembled the homogeneous, smooth surface characteristic of PP (Figure 3).

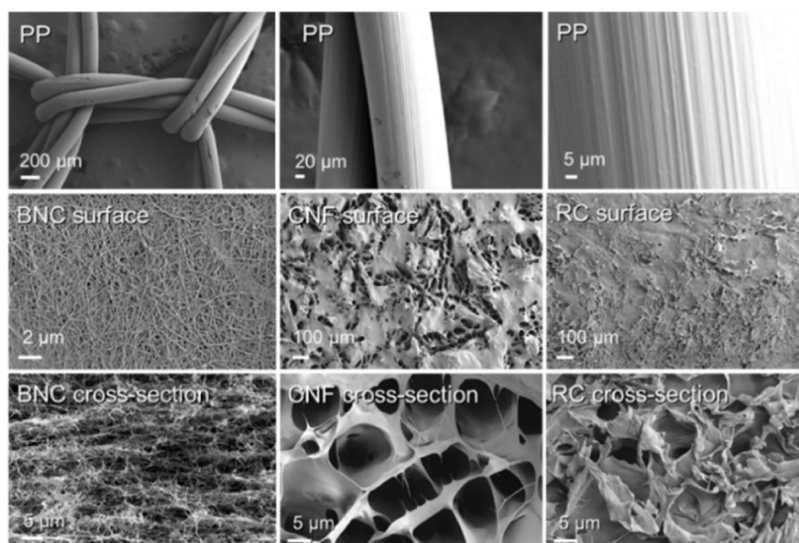


Figure 3. Scanning electron microscopy (SEM) of the meshes preimplantation. The surface and cross-section of polypropylene (PP), bacterial nanocellulose (BNC), 3D-printed cellulose nanofibrils (CNFs), and regenerated cellulose (RC).

Table 3. Wound Reactions in a Rat Subcutaneous Transplantation Model

follow-up	wound reaction	BNC ^a	RC ^b	CNF ^c	SHAM ^d	PP ^e	total number of wound reactions
7 days	mild erythema and swelling			1		6	7/158 (4.4%)
1 month	mild erythema and swelling					1	5/158 (3.2%)
	erythema and clear discharge abscess		1			1 2	

^aBacterial nanocellulose. ^bRegenerated cellulose. ^c3D-printed cellulose nanofibrils. ^dPolydioxanone (PDS, Johnson&Johnson, New Brunswick, NJ).

^ePolypropylene.

In addition to considering the porosity of the solid fraction of the meshes, another crucial factor is the open area of the structure. The PP mesh consisted of a grid of solid, nonporous filaments arranged in a woven structure with large open areas (Figure 1d). In its wet state, BNC forms a highly microporous membrane (Figure 1a), similar to RC films (Figure 1c). The cellulose meshes, created through direct ink writing of cellulose nanofibrils, exhibit a grid-like structure characterized by smaller square-like open areas. These filaments comprise the solid phase and possess inherent microporosity and permeability. The configuration of these open areas (square) on the mesh surface, along with the biomaterial's topography, play a pivotal role in influencing the intensity of the host tissue's response.^{11,63}

Pore size is considered the most important characteristic of the solid phase of the material as it determines interactions with cells and potential integration.⁶⁴ Cells and host tissues favor a porous microstructure due to enhanced transport of oxygen and nutrients.⁶⁴ Differences in porosity allow cells to penetrate through the pores of BNC, RC, and CNF, unlike the solid microstructure of PP. The mesh configuration of CNF was intentionally designed to increase porosity and surface area. The open structure of the mesh allows for better cell adhesion, proliferation, and tissue integration, crucial for certain biomedical applications, such as wound healing.⁴³

It is noteworthy that PP and CNF meshes were stored and implanted in a dry condition to ensure sterility. The wet strength of cellulose is generally lower than its dry strength.⁵⁰ However, both BNC and RC retained a significant level of their strength in wet conditions.

The final performance of the materials in the host tissue depend not only on their porosity, but also their mechanical properties and chemical composition, surface charge, and wettability.⁵¹ All the cellulose-derived materials used in the study have similar tensile strengths^{47,49,52} and high wettability, and were expected to promote tissue interaction.⁵¹ The pore size and surface charge of the same materials can be adjusted to promote cell interactions. Bioinert nanomaterials can be rendered bioactive by adjusting the magnitude of their surface charge density. This adjustment can trigger interactions with oppositely charged proteins and ions from the surrounding biological environment, facilitating cell membrane interactions and enhancing cell adhesion and proliferation.⁶⁵

The morphology of cellulose-derived materials is influenced by the source of cellulose and the synthesis processes used for creating meshes, patches, and other cellulosic materials. Cellulose regenerated through solvent dissolution typically exhibits smooth macroscale morphologies with minimal roughness.⁵⁰ In contrast, CNF dispersed in water can be regenerated to mimic the morphology of RC while allowing for small surface roughness.⁶⁶ In this study, both CNF and RC demonstrated smooth macroscale surfaces, although CNF, composed of smaller building blocks, potentially display higher accessibility for cell-material interactions.⁶⁷ In the case of BNC, production through bacterial metabolism limits control over the microscale surface features.⁶⁸ The specific methods selected for transforming BNC, CNF, and RC were chosen for their simplicity and energy efficiency. Therefore, comparisons should consider the factors mentioned above. Each system is discussed on its own merits regarding viable procedures for manufacturing meshes and patches.

Table 4. Cell Response to Different Implants in a Rat Subcutaneous Transplantation Model^a

parameter	BNC ^b	RC ^c	CNF ^d	SHAM ^e	PP ^f
follow-up 1 month					
number of explants	9	6	7	7	39
foreign body reaction ^g					
1	3 (33.3%)	3 (50%)	2 (28.6%)	3 (42.9%)	21 (53.8%)
2	4 (44.4%)	1 (16.7%)	2 (28.6%)	2 (28.6%)	6 (15.4%)
3	1 (11.1%)		1 (14.2%)		
4					
acute inflammation ^h		1 (16.7%)	1 (14.3%)	1 (14.3%)	4 (10.3%)
chronic inflammation ⁱ					1 (2.6%)
granulation tissue ^j	1 (11.1%)	1 (16.7%)	1 (14.3%)	2 (28.6%)	7 (17.9%)
scar ^k		1 (16.7%)		1 (14.3%)	6 (15.4%)
excluded ^l	1 (10%)	4 (40%)	3 (30%)	3 (30%)	1 (2.6%)
follow-up 3 months					
number of explants	7	7	6	8	30
foreign body reaction ^g					
1	4 (57.1%)	7 (100%)	1 (16.7%)	5 (62.5%)	11 (36.7%)
2	2 (28.6%)		2 (33.3%)	2 (25%)	12 (40%)
3	1 (14.3%)		2 (33.3%)		1 (3.3%)
4			1 (16.7%)		
acute inflammation ^h					
chronic inflammation ⁱ					1 (3.3%)
granulation tissue ^j					3 (10%)
scar ^k				1 (12.5%)	3 (10%)
excluded ^l	3 (30%)	3 (30%)	4 (40%)	2 (20%)	10 (25%)

^aThe numbers in the parameter column represent the severity of the cell response. Scores from 0 to 4 (0 = absent, 1 = mild, 2 = moderate, 3 = severe, 4 = packed) were used for evaluation of foreign body reaction^g and scores from 0 to 1 (0 = absent, 1 = present) for evaluation of the other cell responses. The numbers in the treatment columns represent the number of samples where some degree of cell response was observed. ^bBacterial nanocellulose. ^cRegenerated cellulose. ^d3D-printed cellulose nanofibrils. ^ePolydioxanone (PDS, Johnson&Johnson, New Brunswick, NJ). ^fPolypropylene. ^gGiant cells and macrophages. ^hNeutrophils. ⁱLymphocytes. ^jPlasma cells, sporadic lymphocytes, and granulocytes. ^kConnective tissue. ^lTechnical issues, nonrepresentative samples, premature euthanasia due to study-unrelated cause.

While it is possible to tailor the morphology and macromolecular structure of cellulose to create structures with similar visual features—such as through mechanical modification or the fabrication of 3D-printed meshes—it is important to note that these processing steps can significantly impact the mechanical and microstructural characteristics of the final structures, as well as sterility.

Implanted Cellulose-Derived Meshes Do Not Compromise Rat Well-Being Or Survival. Overall, 79 out of 80 implanted rats (98.8%) survived the study period and exhibited expected weight gain. One rat was euthanized prematurely 9 days postoperatively due to a tumor-like lesion on the right hind leg. The data from this rat was excluded from the study.

Of the 158 wounds assessed, 146 (92.4%) healed without complications (Figure S1a–e). The majority (83.3%) of wound reactions observed at 7 days and one month postimplantation were associated with wounds implanted with PP meshes (Figure S1f–h, Table 3) All wound reactions resolved either spontaneously or with local treatment. No wound reactions were noted at three months postimplantation.

The higher incidence of wound reactions in wounds implanted with PP meshes compared to previous reports^{15,69–71} underscores the importance of consistent macroscopic evaluation of wounds, as conducted in this study.

Two abscesses (S1h, Table 3) were detected in the wounds implanted with PP, attributable to intraoperative aseptic issues, but did not affect the overall well-being or survival of the rats. These findings are consistent with previous *in vivo* studies involving subcutaneous implantation of BNC,^{72,73} and CNF,³⁹

where no clinical signs of infection were reported at the implantation sites.

BNC and RC Resist Degradation *In Vivo*. At explantation, BNC, RC, and PP meshes were easily retrieved and remained cohesive implying minimal degradation in 90 days. In contrast, CNF meshes were more challenging to visualize from host tissue, suggesting notable implant degradation and/or fragmentation. All implanted meshes integrated into the surrounding tissues, without any macroscopic signs of adverse reactions around the implanted sites.

The *in vitro* degradation profile of BNC was analyzed in a previous study.⁴⁹ After 28 days of exposure to pH 7.4 and pH 5, BNC exhibited mass losses of less than 2 and 4%, respectively, primarily due to fibril detachment during the washing step and characterization, indicating its potential for long-term tissue support.⁴⁹ Similarly, chitosan-modified TOCNF samples showed a maximum weight loss of about 10% after 28 days.⁴⁷ For RC, the initial dissolution in alkali promotes gelation.⁵⁰ The *in vitro* and *in vivo* degradation profiles are expected to differ due to the chemical, physical, and mechanical interactions, particularly related to host tissue morphology, integration with the host tissue, and mechanical tension during tissue healing. Based on previous studies,^{43,47} highly crystalline nanocellulose resists degradation and weight unless subjected to enzymatic, hydrolytic, or autocatalytic oxidation, which are not expected to occur *in vivo*.^{43,47}

All Implanted Materials Induced a Foreign Body Reaction in Host Tissue. A total of 126 samples from 79 rats were included in the analysis. Thirty-two samples were

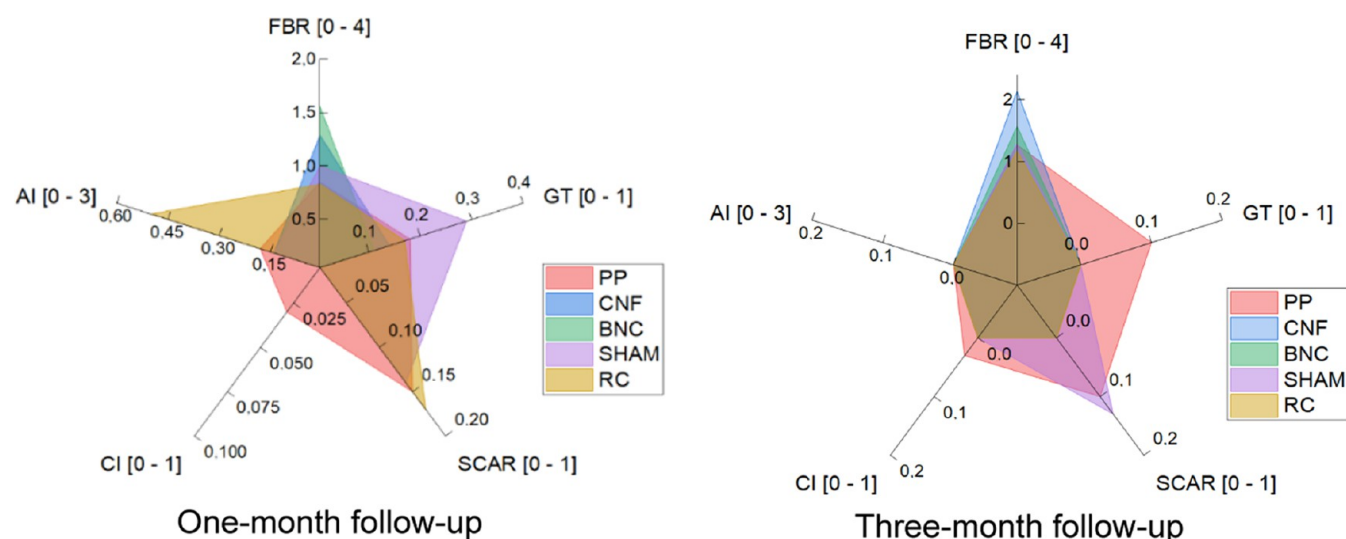


Figure 4. Cell response to bacterial nanocellulose (BNC), regenerated cellulose (RC), 3D-printed cellulose nanofibrils (CNFs), polydioxanone (PDS, Johnson&Johnson, New Brunswick, NJ) as sham procedure, and polypropylene (PP) in a rat subcutaneous transplantation model at two different time points. FBR = foreign body reaction, GT = granulation tissue, AI = acute inflammation, CI = chronic inflammation.

excluded from histopathological evaluation: 7 due to non-representativeness and 25 due to technical processing issues (Table 4). For a detailed histopathological assessment, refer to Figure S2 in the Supporting Information.

Bacterial Nanocellulose (BNC). The majority (8/9, 88.9%) of implanted BNC meshes exhibited mild-to-severe FBR at one-month postimplantation, with all samples (7/7, 100%) showing similar reactions at three months. At one month, the FBR in BNC implants was significantly more severe compared to the PP meshes ($p = 0.0150$, OR 6.02, 95% CI 1.43–25.36). However, by comparing one- and three-month follow-ups, the difference in FBR diminished ($p = 0.482$, OR 1.74, 95% CI 0.37–8.26) (Table 4, Figure 4). This observed inflammatory response aligns with findings from our recent *in vitro* study, where BNC induced monocyte activation but suppressed the proinflammatory macrophage-like phenotype induced by 12-O-Tetradecanoylphorbol-13-acetate (TPA).⁴⁷

Our results are consistent with previous *in vivo* studies on BNC implantation, such as intradermal implantation in rabbits for 28 days⁷⁴ and subcutaneous implantation in sheep for 1–32 weeks,⁷⁵ which showed a chronic phase of inflammation with foreign body giant cells (FBGCs). In contrast, studies by Pértile et al.⁷² and Helenius et al.⁷³ did not report FBR in rats implanted with BNC subcutaneously for periods ranging from 3 to 12 months. Differences noted may be attributed to variations in material characteristics (shape, size, porosity) and animal models used.⁷⁶ The porous nature of BNC facilitates cell migration and predisposes to inflammatory reactions in host tissue. While the BNC samples used in our study were washed with NaOH (aq.), residual traces cannot be completely ruled out.

Regenerated Cellulose (RC). The majority (4/6, 66.7%) of the implanted RC meshes exhibited a mild-to-moderate FBR at one month postimplantation, while a mild FBR was observed in all samples (7/7, 100%) at three months. The severity of FBR in the RC meshes was notably similar to that observed with PDO and PP, displaying a mostly mild-to-moderate FBR at one-month postimplantation and a slightly milder FBR at three months (Table 4, Figure 4). This similarity in host tissue reaction can be attributed to the analogous microstructure of

RC and PP meshes. Our results align with previously published studies,^{77,78} where PP mesh induced a mild but persistent FBR both as an intraperitoneal mesh in rats⁷⁷ and in pelvic reconstructive surgery in humans.⁷⁸

3D-Printed Cellulose Nanofibrils (CNFs). A mild-to-severe FBR was observed in the majority (5/7, 71.4%) of CNF meshes at one-month postimplantation. By three months, all CNF samples exhibited a moderate-to-severe or packed FBR (Table 4, Figure 4). Unlike RC meshes, where the initial FBR diminished over time, the FBR in CNF meshes intensified, with a significantly greater severity at three months ($p = 0.0030$, OR 19.14, 2.78–131.83) compared to PP.

In an *in vitro* study, Ajdary et al.⁴³ demonstrated the high biocompatibility of 3D-printed TOCNF-based patches, finding that drug loading supported cardiac cell proliferation for 28 days. Although the number of samples in our study was limited, the FBR observed was more severe than in previous *in vivo* biocompatibility studies of TOCNF.^{39–41} Outstanding biocompatibility and wound healing efficacy of TOCNF combined with gelatin and aminated silver nanoparticles (Ag-NH₂NPs) were reported in a 14-day *in vivo* wound healing study with mice.⁴¹ No FBR was elicited in a subcutaneous injection of TOCNF in a 12-week rat model.³⁹ Furthermore, a subcutaneous rat transplantation study of a 3D aerogel blend of CNF and gelatin over 8 weeks showed significant improvement in FBGC reaction and acute inflammation between four and 8 weeks postimplantation.⁴⁰

The CNF meshes were not easily identified at the three-month follow-up. The challenging visualization and the more severe FBR relative to PP were likely attributable to the mesh's morphology and surface chemistry when implanted in a dry condition. Degradation depends on both the implant's morphology and surface chemistry, as well as the mode of preparation.³⁹ Mesh construction and composition appear to be more crucial in determining FBR after implantation than merely the reduction of the material itself.⁵ The rough surface and fragmentation of CNF might contribute to the development of a pronounced late FBR.⁷⁶ Previous studies indicate that myoblast cells exhibit reduced attachment to surfaces with higher roughness;⁴⁴ however, cell attachment behavior is

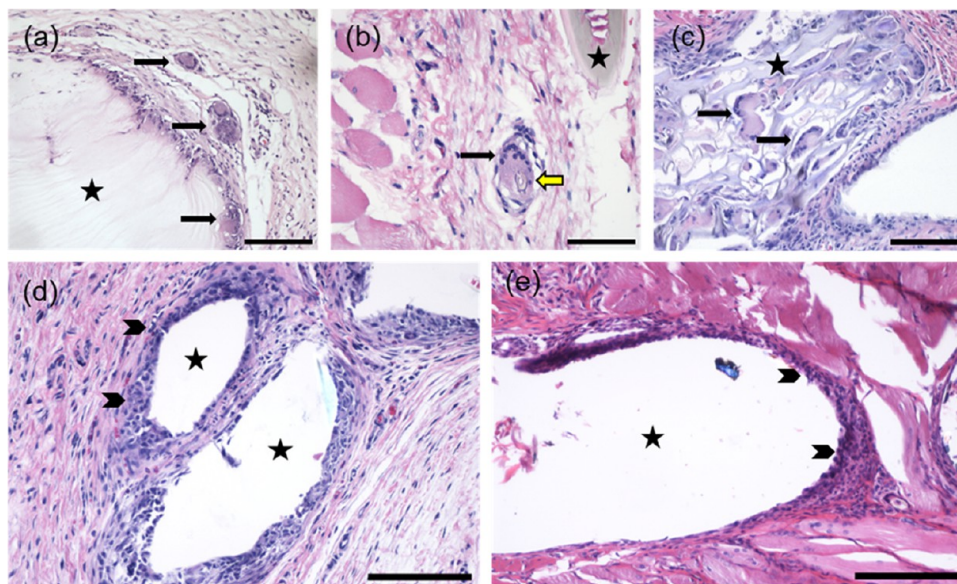


Figure 5. Histopathological analysis of different implants at day 30. (a) Foreign body giant cells in the border zone surrounding bacterial nanocellulose (BNC). (b) A foreign body giant cell phagocytosing regenerated cellulose (RC). (c) Biomaterial fragmentation and foreign body giant cell infiltration into 3D-printed nanofibrils (CNFs). (d) Macrophages in the border zone surrounding polypropylene (PP) and (e) polydioxanone (PDS, NJ). Black arrow = foreign body giant cell, yellow arrow = phagocytosed biomaterial, arrowhead = macrophages, asterisk = implanted mesh/mesh hole or, in case of sham, suture material. H&E staining. Bars 200 μm (a, c–e) and 100 μm (b).

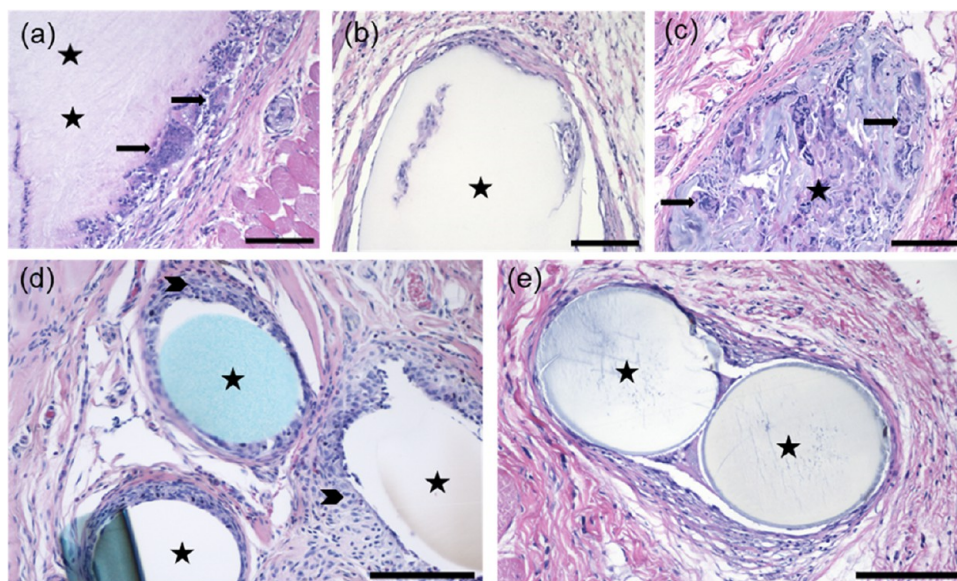


Figure 6. Histopathological analysis of different implants at day 90. (a) Foreign body giant cells in the border zone surrounding bacterial nanocellulose (BNC). (b) Macrophages in the border zone surrounding and infiltrating into regenerated cellulose (RC). (c) Biomaterial fragmentation and foreign body giant cell infiltration into 3D-printed nanofibrils (CNFs). (d) Macrophages in the border zone surrounding polypropylene (PP) and (e) polydioxanone (PDS, Johnson&Johnson, New Brunswick, NJ). Black arrow = foreign body giant cell, arrowhead = macrophages, asterisk = implanted mesh/mesh hole or, in case of sham, suture material. H&E staining. Bars 200 μm .

highly influenced by the specific cell type and the overall surface charge.

No significant difference was detected when comparing the severity of FBR of BNC, RC, CNF, and PDO with PP within individual rats. The FBR detected in our study is consistent with the expected host tissue reaction to a foreign material. Implantation of a biomaterial typically initiates an inflammatory process aimed at preventing tissue damage, isolating and destroying the foreign material, and initiating the repair process. The acute inflammatory reaction generally subsides

within a week, whereas a chronic inflammatory response can last for up to 4 weeks, leading to a granulation phase, including FBR.^{4,78}

Cellulose-Derived Meshes Induce Cell Ingrowth to Implanted Materials. With BNC, RC, and CNF, the FBR was observed not only at the implant periphery, but also penetrating the material (Figures 5 and 6). Cell ingrowth was multifocally present at the tissue-implant interface, with some FBGCs engulfing the biomaterial (Figure 5b). Dense cell ingrowth, phagocytosis of the implanted material and implant

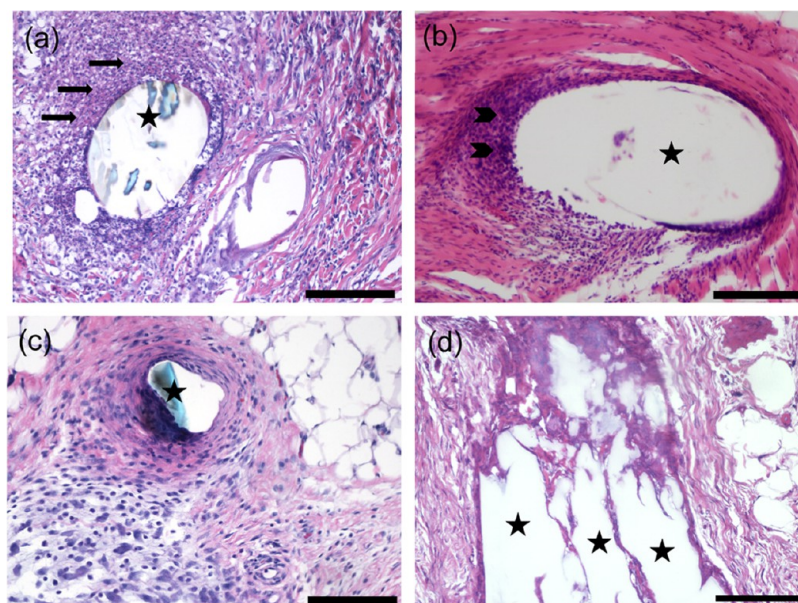


Figure 7. Histopathological analysis of different cell reactions surrounding implants with the exception of foreign body reaction. (a) Acute inflammation, day 30, polypropylene (PP). (b) Chronic inflammation, day 90, polypropylene (PP). (c) Granulation tissue with variable inflammatory reaction, day 90, polypropylene (PP). (d) Scar tissue, day 30, regenerated cellulose (RC). Black arrow = neutrophils, arrowhead = lymphocytes, and asterisk = implanted mesh/mesh hole. H&E staining. Bars: 200 μm in (a, c, d) and 400 μm in (b).

fragmentation were consistently more extensive in CNF meshes, displaying a more indistinct border zone between the material and the surrounding tissues compared to the other explants at given time points (Figures 5c and 6c). Over time, integration with the host tissue was observed. Cellular penetration is expected to be more intense through porous implants, in agreement with Pèrtille⁷² and Helenius et al.⁷³ In contrast, no cellular infiltration or fragmentation of the implanted material was observed with PP and PDO (Figures 5 and 6).

Macrophages are considered to be the most important cellular mediators of FBRs in biodegradable materials. When macrophages are not effective in removing the foreign material, they fuse into FBGCs.^{63,79} FBGCs persist in the tissues as long as the biomaterial is detected, eventually leading to degradation of the implanted material.^{4,64} A successfully implanted biocompatible biomaterial will integrate with the host tissue,⁷⁹ as shown in the present study with BNC, RC, and CNF (Figures 5 and 6).

None of the implanted materials induced signs of necrosis, extensive fibrosis, or marked inflammatory reactions. Localized collections of neutrophils were detected in a total of 7/126 samples (5.6%) (Table 3, Figure 7a) either around the implant (3/7, 42%), or distant from the implantation site in subcutaneous tissue (4/7, 58%) indicating intraoperative aseptic failure. No signs of acute inflammation were observed in any of the samples three months postimplantation. In addition, no signs of chronic inflammation were observed in any of the cellulose-derived samples nor PDO in the given time points. Mild neovascularization with minimal capillary proliferation and focal, 1–3 buds of vessels; mild fatty infiltrate around the implants, and sporadic fibrocytes were generally displayed. Scar formation with mild to moderate fibrosis was observed in a total of 11/126 samples (8.7%) (Table 3, Figure 7d). Importantly, no extensive fibrosis, nor signs of necrosis were detected. No significant differences were observed when comparing acute and chronic inflammation, granulation tissue,

and scar formation between BNC, RC, CNF and PDO implants and PP, nor between the implants within the same individual.

Biocompatibility of BNC, RC, and CNF Meshes and Their Potential To Replace PP Meshes. An appropriate host response and active cell ingrowth were detected in all implanted cellulose-derived meshes in the host tissue. The presence of macrophages and FBGCs was not indicative of an adverse FBR. No signs of necrosis, extensive fibrosis, or marked inflammatory reactions were observed in any of the tissue samples. Implants reached a steady state of tolerance within the surrounding host tissue, suggesting that BNC, RC, and CNF meshes exhibit potential biocompatibility and are suitable for biointegration.

All BNC and RC meshes were easily identified at one- and three-month postimplantation. Although cellulose is considered biodegradable, it does not readily degrade *in vivo* due to the absence of cellulases in animals.¹⁷ By contrast, van Ho et al.³⁹ observed degradation of cellulose nanocrystals after 12 weeks in a subcutaneous rat injection model. Based on our findings, we anticipate no or slow degradation of BNC and RC, making them suitable implants for applications in which prolonged structural integrity is desired, and degradation is not a requirement, such as alternatives to PP meshes in current use. Cellulose nanocrystals exhibit limited stability under moisture-rich conditions.⁸⁰ They were not included in the present study, since their use would be restricted to reinforcing existing matrices rather than serving as the primary material.

The elasticity and strength of the RC films, CNF mesh, and BNC are all key factors in their suitability for different biomedical applications. The uniform texture of RC films and their mechanical properties make them suitable for applications where flexibility and smooth surface interaction are necessary. Meanwhile, the CNF mesh's mechanical robustness and porosity make it more suitable to highly interactive tissue environments.⁸¹

CONCLUSIONS

The biocompatibility of bacterial nanocellulose (BNC) and cellulose nanofibrils (CNF) was confirmed in this study. The biocompatibility of alkali-dissolved (NaOH/H₂O, ZnO) regenerated cellulose (RC) was demonstrated *in vivo*. All implanted cellulose-derived materials exhibited expected, consistent host tissue responses to a foreign material, comparable to PP. However, the *in vivo* response differed: RC, with analogous microstructure to that of PP, showed similar, mild-to-moderate foreign body reaction (FBR), consisting mostly of macrophages and foreign body giant cells at one-month postimplantation, and a milder FBR at three months. On the contrary, the FBR detected with BNC implants was significantly more severe one-month post-implantation and with CNF at three months when compared to PP. The rough surface of CNF might have contributed to the pronounced late FBR. All implanted cellulose-derived meshes induced cell ingrowth. Differences in porosity allowed cells to penetrate through the pores of BNC, RC, and CNF, unlike the solid microstructure of PP and PDO. CNF showed extensive implant fragmentation and phagocytosis likely due to morphology effects and implantation in dry conditions.

All cellulose-derived materials showed potential for future clinical applications as scaffolds for tissue repair; however, the *in vivo* fragmentation observed for the CNF material needs to be taken into consideration if produced and used as in the current study. Our results provide valuable insights into mesh materials, laying a foundation for future *in vivo* investigations that should consider the specific morphologies, procedures and number of implants tested.

ASSOCIATED CONTENT

Data Availability Statement

The data of histopathological evaluation used to support the findings of this study is included within the [Supporting Information](#) file (S2).

Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acs.biomac.4c00984>.

Wound grading after subcutaneous implantation of bacterial nanocellulose (BNC), regenerated cellulose (RC), 3D-printed cellulose nanofibrils (CNFs), polypropylene (PP) and polydioxanone (PDO) in a rat subcutaneous transplantation model ([PDF](#))

Cell response to different implants in a rat subcutaneous transplantation model ([PDF](#))

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The manuscript was written through contributions of all authors. All authors have given approval to the final version of the manuscript.

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Notes

The authors declare no competing financial interest.

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ABBREVIATIONS

BNC, bacterial nanocellulose; RC, regenerated cellulose; CNFs, cellulose nanofibrils; SEM, scanning electron microscopy; FBR, foreign body reaction; FBGC, foreign body giant cell

REFERENCES

- (1) Seddiqi, H.; Oliaei, E.; Honarkar, H.; Jin, J.; Geonzon, L. C.; Bacabac, R. G.; Klein-Nulend, J. Cellulose and its derivatives: towards biomedical applications. *Cellulose* **2021**, *28*, 1893–1931.
- (2) Dufresne, A. Nanocellulose: a new ageless bionanomaterial. *Mater. Today* **2013**, *16* (6), 220–227.
- (3) Apelgren, P.; Sämfors, S.; Säljö, K.; Mölne, J.; Gatenholm, P.; Troedsson, C.; Thompson, E. M.; Kölby, L. Biomaterial and biocompatibility evaluation of tunicate nanocellulose for tissue engineering. *Biomater. Adv.* **2022**, *137*, No. 212828.
- (4) Anderson, J. M.; Rodriguez, A.; Chang, D. T. Foreign body reaction to biomaterials. *Semin. Immunol.* **2008**, *20* (2), 86–100.
- (5) Weyhe, D.; Belyaev, O.; Müller, C.; Meurer, K.; Bauer, K. H.; Papapostolou, G.; Uhl, W. Improving Outcomes in Hernia Repair by the Use of Light Meshes – A Comparison of Different Implant Constructions Based on a Critical Appraisal of the Literature. *World J. Surg.* **2007**, *31* (1), 234–244.
- (6) Baylón, K.; Rodríguez-Camarillo, P.; Elías-Zúñiga, A.; Díaz-Elizondo, J. A.; Gilkerson, R.; Lozano, K. Past, present and future of surgical meshes: A review. *Membranes* **2017**, *7* (3), 47.
- (7) Dällenbach, P. To mesh or not to mesh: a review of pelvic organ reconstructive surgery. *Int. J. Women's Health* **2015**, *7*, 331–343.
- (8) Usher, F. C.; Oshner, J.; Tuttle, L. L., Jr. Use of Marlex mesh in the repair of incisional hernias. *Am. Surg.* **1958**, *24* (12), 969–974.
- (9) Mangir, N.; Aldemir, D. B.; Chapple, C. R.; MacNeil, S. Landmarks in Vaginal Mesh Development: Polypropylene Mesh for Treatment of SUI and POP. *Nat. Rev. Urol.* **2019**, *16*, 675–689.
- (10) Abhari, R. E.; Izett-Kay, M. L.; Morris, H. L.; Cartwright, R.; Snelling, S. J. B. Host-biomaterial interactions in mesh complications after pelvic floor reconstructive surgery. *Nat. Rev. Urol.* **2021**, *18* (12), 725–738.
- (11) Margulies, R. U.; Lewickt-Gaupp, C.; Fenner, D. E.; McGuire, E. J.; Clemens, J. Q.; DeLancey, J. O. L. Complications requiring reoperation following vaginal mesh procedures for prolapse. *Am. J. Obstet. Gynecol.* **2008**, *199* (6), 678.e1–678.e4.
- (12) Farmer, L.; Ayoub, M.; Warejcka, D.; Southerland, S.; Freeman, A.; Solis, M. Adhesion formation after intraperitoneal and extraperitoneal implantation of polypropylene mesh. *Am. Surg.* **1998**, *64* (2), 144–146.
- (13) U.S. Food and Drug Administration. FDA takes action to protect women's health, orders manufactures of surgical mesh intended for transvaginal repair of pelvic organ prolapse to stop selling all devices. Online2021. Available from URL: <https://www.fda.gov/news-events/press-announcements/fda-takes-action-protect-womens-health-orders-manufacturers-surgical-mesh-intended-transvaginal>.
- (14) Iakovlev, V. V.; Guelcher, S. A.; Bendavid, R. Degradation of Polypropylene *in vivo*: A Microscopic Analysis of Meshes Explanted from Patients. *J. Biomed. Mater. Res., Part B* **2017**, *105* (2), 237–248.
- (15) Voskerician, G.; Jin, J.; White, M. F.; Williams, C. P.; Rosen, M. J. Effect of biomaterial design criteria on the performance of surgical meshes for abdominal hernia repair: a pre-clinical evaluation in a chronic rat model. *J. Mater. Sci. Mater. Med.* **2010**, *21* (6), 1989–1995.
- (16) Li, T.; Chen, C.; Brozena, A. H.; Zhu, J. Y.; Xu, L.; Driemeier, C.; Dai, J.; Rojas, O. J.; Isogai, A.; Wågberg, L.; Hu, L. Developing Fibrillated Cellulose as a Sustainable Technological Material. *Nature* **2021**, *590* (7844), 47–56.
- (17) Lin, N.; Dufresne, A. Nanocellulose in Biomedicine; Current status and future prospect. *Eur. Polym. J.* **2014**, *59*, 302–325.
- (18) Hildebrandt, J.; Thrän, D.; Bezema, A. The circularity of potential bio-textile production routes: Comparing life cycle impacts of bio-based materials used within the manufacturing of selected leather substitutes. *J. Cleaner Prod.* **2021**, *287*, No. 125470.
- (19) Reyes, G.; King, A. W. T.; Koso, T. V.; Penttilä, P. A.; et al. Cellulose dissolution and gelation in NaOH (aq) under controlled CO₂ atmosphere: supramolecular structure and flow properties. *Green Chem.* **2022**, *24*, 8029–8035.
- (20) Isikgor, F. H.; Becer, C. R. Lignocellulosic biomass: a sustainable platform for the production of bio-based chemicals and polymers. *Polym. Chem.* **2015**, *6*, 4497–4559.
- (21) Jadczyk, K.; Ochędzan-Siodłak, W. Bacterial cellulose: Biopolymer with novel medical applications. *J. Biomater. Appl.* **2023**, *38* (1), 51–63.
- (22) Ross, P.; Mayer, R.; Benziman, M. Cellulose biosynthesis and function in bacteria. *Microbiol. Rev.* **1991**, *55* (1), 35–58.
- (23) Cañas-Gutiérrez, A.; Martínez-Correa, E.; Suárez-Avenidaño; Arboleda-Toro, D.; Castro-Herazo, C. Influence of bacterial nanocellulose surface modification on calcium phosphates precipitation for bone tissue engineering. *Cellulose* **2020**, *27*, 10747–10763.
- (24) Pang, M.; Huang, Y.; Meng, F.; Zhuang, Y.; Liu, H.; Du, M.; Ma, Q.; Wang, Q.; Chen, Z.; Chen, L.; Cai, T.; Cai, Y. Application of bacterial cellulose in skin and bone tissue engineering. *Eur. Polym. J.* **2020**, *122*, No. 109365.
- (25) Bao, L.; Tang, J.; Hong, F. F.; Lu, X.; Chen, L. Physicochemical properties and *in vitro* biocompatibility of three bacterial nanocellulose conduits for blood vessel applications. *Carbohydr. Polym.* **2020**, *239*, No. 116246.
- (26) Hou, S.; Xia, Z.; Pan, J.; Wang, N.; Gao, H.; Ren, J.; Xia, X. Bacterial Cellulose Applied in Wound Dressing Materials: Production and Functional Modification – A Review. *Macromol. Biosci.* **2023**, *24* (2), No. e2300333.
- (27) Aboelnaga, A.; Elmasry, M.; Adly, O. A.; Elbadawy, M. A.; Abbas, A. H.; Abdelrahman, I.; Salah, O.; Steinvall, I. Microbial cellulose dressing compared with silver sulphadiazine for the treatment of partial thickness burns: A prospective, randomised, clinical trial. *Burns* **2018**, *44* (8), 1982–1988.
- (28) Napavichayanun, S.; Yamdech, R.; Aramwit, P. The safety and efficacy of bacterial nanocellulose wound dressing incorporating sericin and Polyhexamethylene biguanide: *in vitro*, *in vivo* and clinical studies. *Arch. Dermatol. Res.* **2016**, *308* (2), 123–132.
- (29) Tu, H.; Zhu, M.; Duan, B.; Zhang, L. Recent Progress in High-Strength and Robust Regenerated Cellulose Materials. *Adv. Mater.* **2021**, *33*, No. 2000682.
- (30) Phinichka, N.; Kaenthong, S. Regenerated cellulose from high alpha cellulose pulp of steam-exploded sugarcane bagasse. *J. Mater. Res. Technol.* **2018**, *7* (1), 55–65.
- (31) Reyes, G.; Pacheco, C. M.; Isaza-Ferro, E.; González, A.; Pasquier, E.; Alejandro-Martín, S.; Arteaga-Peréz, L. E.; Carrillo, R. R.; Carrillo-Varela, I.; Mendonça, R. T.; Flanigan, C.; Rojas, O. J. Upcycling agro-industrial blueberry waste into platform chemicals and structured materials for application in marine environments. *Green Chem.* **2022**, *24*, 3794–3804.
- (32) Björquist, S.; Aronsson, J.; Henriksson, G.; Persson, A. Textile qualities of regenerated cellulose fibers from cotton waste pulp. *Text. Res. J.* **2018**, *88* (21), 2485–2492.
- (33) Zainul Armir, N. A.; Zulkifli, A.; Gunaseelan, S.; Palanivelu, S. D.; Saller, K. M.; Othman, M. H. C.; Zakaria, S. Regenerated

Cellulose Products for Agricultural and Their Potential: A Review. *Polymers* **2021**, *13* (20), 3586.

(34) Budtova, T.; Navard, P. Cellulose in NaOH–water based solvents: a review. *Cellulose* **2016**, *23* (1), 5–55.

(35) Rosenau, T.; Potthast, A.; Sixta, H.; Kosma, P. The chemistry of side reactions and byproduct formation in the system NMMO/cellulose (Lyocell process). *Prog. Polym. Sci.* **2001**, *26* (9), 1763–1837.

(36) Swatloski, R. P.; Spear, S. K.; Holbrey, J. D.; Rogers, R. D. Dissolution of Cellulose with Ionic Liquids. *J. Am. Chem. Soc.* **2002**, *124* (18), 4974–4975.

(37) Liebert, T. Cellulose Solvents – Remarkable History, Bright Future. In *Cellulose Solvents: For Analysis, Shaping and Chemical Modification*, ACS Symposium Series, 2010; Vol. 1033, Chapter 1, pp 3–54.

(38) Väisänen, S.; Ajdary, R.; Altgen, M.; et al. Cellulose dissolution in aqueous NaOH–ZnO: cellulose reactivity and the role of ZnO. *Cellulose* **2021**, *28*, 1267–1281.

(39) Ho, H. V.; Makkar, P.; Padalhin, A. R.; Le, T. T. T.; Lee, S. Y.; Jaegyoung, G.; Lee, B.-T. Preliminary studies on the *in vivo* performance of various kinds of nanocellulose for biomedical applications. *J. Biomater. Appl.* **2020**, *34* (7), 942–951.

(40) Mirtaghavi, A.; Baldwin, A.; Tanideh, N.; Zarei, M.; Muthuraj, R.; Cao, Y.; Zhao, G.; Geng, J.; Jin, H.; Luo, J. Crosslinked porous three-dimensional cellulose nanofibers-gelatine biocomposite scaffolds for tissue regeneration. *Int. J. Biol. Macromol.* **2020**, *164*, 1949–1959.

(41) Liu, R.; Dai, L.; Si, C.; Zeng, Z. Antibacterial and hemostatic hydrogel via nanocomposite from cellulose nanofibers. *Carbohydr. Polym.* **2018**, *195*, 63–70.

(42) Saito, T.; Kimura, S.; Nishiyama, Y.; Isogai, A. Cellulose Nanofibers Prepared by TEMPO-Mediated Oxidation of Native Cellulose. *Biomacromolecules* **2007**, *8* (8), 2485–2491.

(43) Ajdary, R.; Ezazi, N. Z.; Correia, A.; Kemell, M.; Huan, S.; Ruskoaho, H. J.; Hirvonen, J.; Santos, H. A.; Rojas, O. J. Multifunctional 3D-printed patches for long-term drug release therapies after myocardial infarction. *Adv. Funct. Mater.* **2020**, *30* (34), No. 2003440.

(44) Ajdary, R.; Huan, S.; Ezazi, N. Z.; Xiang, W.; Grande, R.; Santos, R.; Rojas, O. J. Acetylated Nanocellulose for Single-Component Bioinks and Cell Proliferation on 3d-Printed Scaffolds. *Biomacromolecules* **2019**, *20* (7), 2770–2778.

(45) Seifalian, A.; Digesu, A.; Khullar, V. The use of animal models in preclinical investigations for the development of a surgical mesh for pelvic organ prolapse. *Int. Urogynecol. J.* **2024**, *35* (4), 741–758.

(46) Castro, C.; Cleenwerck, I.; Trček, J.; Zuluaga, R.; De Vos, P.; Caro, G.; Aguirre, R.; Putaux, J. L.; Gañán, P. *Gluconacetobacter medellinensis* sp. nov., cellulose- and non-cellulose-producing acetic acid bacteria isolated from vinegar. *Int. J. Syst. Evol. Microbiol.* **2013**, *63* (Part 3), 1119–1125.

(47) Ajdary, R.; Abidnejad, R.; Lehtonen, J.; Kuula, J.; Raussi-Lehto, E.; Kankuri, E.; Rardy, B.; Rojas, O. J. Bacterial nanocellulose enables auxetic supporting implants. *Carbohydr. Polym.* **2022**, *284*, No. 119198.

(48) Molea, G.; Schonauer, F.; Bifulco, G.; D'Angelo, D. Comparative study on biocompatibility and absorption times of three absorbable monofilament suture materials (Polydioxanone, Poliglecaprone 25, Glycomer 631). *Br. J. Plast. Surg.* **2000**, *53*, 137–141.

(49) Ajdary, R.; Reyes, G.; Kuula, J.; Raussi-Lehto, E.; Mikkola, T. S.; Kankuri, E.; Rojas, O. J. Direct Ink Writing of Biocompatible Nanocellulose and Chitosan Hydrogels for Implant Mesh Matrices. *ACS Polym. Au* **2022**, *2* (2), 97–107.

(50) Reyes, G.; Ajdary, R.; Kankuri, E.; Kaschuk, J. J.; Kosonen, H.; Rojas, O. J. Cellulose gelation in NaOH_(aq) by CO₂ absorption: Effects of holding time and concentration on biomaterial development. *Carbohydr. Polym.* **2023**, *302*, No. 120355.

(51) Hubbe, M. A.; Gardner, D. J.; Shen, W. Contact angles and wettability of cellulosic surfaces: A review of proposed mechanisms and test strategies. *BioResources* **2015**, *10* (4), 8657–8749.

(52) Mokhena, T. C.; Sadiku, E. R.; Mochane, M. J.; Ray, S. S.; John, M. J.; Mtibe, A. Mechanical properties of cellulose nanofibril papers and their bionanocomposites: A review. *Carbohydr. Polym.* **2021**, *273*, No. 118507.

(53) Reyes, G.; Lundahl, M. J.; Alejandro-Martín, S.; Arteaga-Pérez, L. E.; Oviedo, C.; King, A. W. T.; Rojas, O. J. Coaxial Spinning of All-Cellulose Systems for Enhanced Toughness: Filaments of Oxidized Nanofibrils Sheathed in Cellulose II Regenerated from a Protic Ionic Liquid. *Biomacromolecules* **2020**, *21* (2), 878–891.

(54) Nikolits, I.; Radwan, S.; Liebner, F.; Dietrich, W.; Egger, D.; Chariyev-Prinz, F.; Kasper, C. Hydrogels from TEMPO-Oxidized Nanofibrillated Cellulose Support *In Vitro* Cultivation of Encapsulated Human Mesenchymal Stem Cells. *ACS Appl. Bio Mater.* **2023**, *6* (2), 543–551.

(55) Ai, F.-F.; Mao, M.; Zhang, Y.; Kang, J.; Zhu, L. The *in vivo* biocompatibility of titanized polypropylene lightweight mesh is superior to that of conventional polypropylene mesh. *NeuroUrol. Urodyn.* **2020**, *39* (1), 96–107.

(56) Charan, J.; Kantharia, N. D. How to calculate sample size in animal studies? *J. Pharmacol. Pharmacother.* **2013**, *4* (4), 303–306.

(57) Hubrecht, R. C.; Carter, E. The 3Rs and Humane Experimental Technique: Implementing Change. *Animals* **2019**, *9* (10), 754.

(58) EN ISO 10993–6. *Biological Evaluation of Medical Devices. Part 6: Tests for Local Effects after Implantation (ISO 10993- 6:2016)*, Finnish version of EN ISO 10993–6. Finnish Standards Association: Helsinki, Finland, 2016.

(59) Carstens, E.; Moberg, G. P. Recognizing Pain and Distress in Laboratory Animals. *ILAR J.* **2000**, *41* (2), 62–71.

(60) Sotocina, S. G.; Sorge, R. E.; Zaloum, A.; Tuttle, A. H.; Martin, L. J.; Wieskopf, J. S.; Mapplebeck, J. C. S.; Wei, P.; Zhan, S.; Zhang, S.; Zhang, S.; McDougal, J. J.; King, O. D.; Mogil, J. S. The Rat Grimace Scale: A Partially Automated Method for Quantifying Pain in the Laboratory Rat via Facial Expressions. *Mol. Pain* **2011**, *7* (1), No. 55.

(61) Bailey, I. S.; Karran, S. E.; Toyn, K.; Brough, P.; Ranaboldo, C.; Karran, S. J. Community surveillance of complications after hernia surgery. *Br. Med. J.* **1992**, *304* (6825), 469–471.

(62) Cardiff, R. D.; Miller, C. H.; Munn, R. J. Manual Hematoxylin and Eosin Staining of Mouse Tissue Sections. *Cold Spring Harb. Protoc.* **2014**, *2014* (6), 655–658.

(63) Klopffleisch, R.; Jung, F. The pathology of the foreign body reaction against biomaterials. *J. Biomed. Mater. Res., Part A* **2017**, *105* (3), 927–940.

(64) Atashrouz, S.; Hatampoor, A.; Yadegari, A.; Ghasemi, H.; Tayebi, L.; Rasoulianboroujeni, M. Mathematical modeling of oxygen transfer in porous scaffolds for stem cell growth: The effects of porosity, cell type, scaffold architecture and cell distribution. *Mater. Chem. Phys.* **2019**, *222*, 377–383.

(65) Eichhorn, S. J.; Etale, A.; Wang, J.; Berglund, L. A.; Li, Y.; Cai, Y.; Chen, C.; Cranston, E. D.; Johns, M. A.; Fang, Z.; Li, G.; Hu, L.; Khandelwal, M.; Lee, K.-Y.; Oksman, K.; Pinitsoontorn, S.; Quero, F.; Sebastian, A.; Titirici, M. M.; Xu, Z.; Vigolini, S.; Frka-Petesic, B. Current international research into cellulose as a functional nanomaterial for advanced applications. *J. Mater. Sci.* **2022**, *57*, 5697–5767.

(66) Tamo, A. K. Nanocellulose-based hydrogels as versatile materials with interesting functional properties for tissue engineering applications. *J. Mater. Chem.* **2024**, *12*, 7692–7759.

(67) Zhang, B. Z.; Duan, W.; Wang, Y.; Dai, L.; Cai, B.; Kong, L.; Fan, J.; Zhang, G.; Wang, L.; Wu, W.; Ning, R. Recent advances of cellulose nanofiber-based materials in cell culture: From population to single-cell. *TrAC, Trends Anal. Chem.* **2023**, *166*, No. 117159.

(68) Ullah, M. W.; Alabbosh, K. F.; Fatima, A.; Ul Islam, S.; Manan, S.; Ul-Islam, M.; Yang, G. Advanced biotechnological applications of bacterial nanocellulose-based biopolymer nanohybrids: A review. *Adv. Ind. Eng. Polym. Res.* **2024**, *7* (1), 100–121.

(69) Vogels, R. R. M.; van Barneveld, K. W. Y.; Bosmans, J. W. A. M.; Beets, G.; Gijbels, M. J. J.; Schreinemacher, M. H. F.; Bouvy, N. D. Long-term evaluation of adhesion formation and foreign body

response to three new meshes. *Surg. Endosc.* **2015**, *29* (8), 2251–2259.

(70) Riccetto, C.; Miyaoka, R.; de Fraga, R.; Barbosa, R.; Dambros, M.; Teixeira, A.; Palma, P. Impact of the structure of polypropylene meshes in local tissue reaction: *in vivo* stereological study. *Int. Urogynecol. J.* **2008**, *19* (8), 1117–1123.

(71) Bellón, J.; Contreras, L. A.; Buján, J.; Palomares, D.; Carrera-San Martín, A. Tissue response to polypropylene meshes used in the repair of abdominal wall defects. *Biomaterials* **1998**, *19* (7), 669–675.

(72) Pértile, R. A.; Moreira, S.; Gil da Costa, R. M.; Correia, A.; Guãrdao, L.; Gartner, F.; Vilanova, M.; Gama, M. Bacterial Cellulose: Long-Term Biocompatibility Studies. *J. Biomater. Sci., Polym. Ed.* **2012**, *23* (10), 1339–1354.

(73) Helenius, G.; Bäckdahl, H.; Bodin, A.; Nannmark, U.; Gatenholm, P.; Risberg, B. *In vivo* biocompatibility of bacterial cellulose. *J. Biomed. Mater. Res., Part A* **2006**, *76A* (2), 431–438.

(74) Martínez Ávila, H.; Schwartz, S.; Feldmann, E.-M.; Mantas, A.; vom Bomhard, A.; Gatenholm, P.; Rotter, N. Biocompatibility evaluation of densified bacterial nanocellulose hydrogel as an implant material for auricular cartilage regeneration. *Appl. Microbiol. Biotechnol.* **2014**, *98* (17), 7423–7435.

(75) Andrade, F. K.; Alexandre, N.; Amorim, I.; Gartner, F.; Mauricio, A. C.; Luis, A. L.; Gama, M. Studies on the biocompatibility of bacterial cellulose. *J. Bioact. Compat. Polym.* **2013**, *28* (1), 97–112.

(76) Veiseh, O.; Doloff, J.; Ma, M.; Vegas, A. J.; Tam, H. H.; Bader, A. R.; Li, J.; Langan, E.; Wyckoff, J.; Loo, W. S.; Jhunjunwala, S.; Chiu, A.; Siebert, S.; Tang, K.; Hollister-Lock, J.; Aresta-Dasilva, S.; Bochenek, M.; Mendoza-Elias, J.; Wang, Y.; Qi, M.; Lavin, D. M.; Chen, M.; Dholakia, N.; Thakrar, R.; Lacík, I.; Weir, G. C.; Oberholzer, J.; Greiner, D. L.; Langer, R.; Anderson, D. G. Size- and shape-dependent foreign body immune response to materials implanted in rodents and non-human primates. *Nat. Mater.* **2015**, *14* (6), 643–651.

(77) Sussman, E. M.; Halpin, M. C.; Muster, J.; Moon, R. T.; Ratner, B. D. Porous implants modulate healing and induce shifts in local macrophage polarization in the foreign body reaction. *Ann. Biomed. Eng.* **2014**, *42* (7), 1508–1516.

(78) Gardner, A. B.; Lee, S. K. C.; Woods, E. C.; Acharya, A. P. Biomaterials-based modulation of the immune system. *Biomed. Res., Int.* **2013**, *2013*, No. 732182.

(79) Brodbeck, W. G.; Anderson, J. M. Giant cell formation and function. *Curr. Opin. Hematol.* **2009**, *16* (1), 53–57.

(80) Solhi, L.; Guccini, V.; Heise, K.; Solala, I.; Niinivaara, E.; Xu, W.; Mihhels, K.; Kröger, M.; Meng, Z.; Wohler, J.; Tao, H.; Cranston, E. D.; Kontturi, E. Understanding Nanocellulose–Water Interactions: Turning a Detriment into an Asset. *Chem. Rev.* **2023**, *123* (5), 1925–2015.

(81) Baniasadi, H.; Ajdary, R.; Trifol, J.; Rojas, O. R.; Seppälä, J. Direct ink writing of aloe vera/cellulose nanofibrils bio-hydrogels. *Carbohydr. Polym.* **2021**, *266*, No. 118114.