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Solubility of Mg-containing β-tricalcium phosphate at 25 °C

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Abstract

The equilibrium solubility of Mg-containing β-tricalcium phosphate (βMgTCP) with various magnesium contents was determined by immersing β MgTCP powder for 27 months in a CH₃COOH– $CH₃COONa$ buffer solution at 25 °C under a nitrogen gas atmosphere. The negative logarithm of the solubility product (pK_{SD}) of β MgTCP was expressed as $pK_{\text{SD}} = 28.87432 + 1.40348C$ − $0.3163C^2 + 0.04218C^3 - 0.00275C^4 + 0.0000681659C^5$, where *C* is the magnesium content in βMgTCP (mol.%). The solubility of βMgTCP decreased with increasing magnesium content owing to the increased structural stability and possible formation of a whitlockite-type phase on the surface. As a result, βMgTCP with 10.1 mol.% magnesium had a lower solubility than that of hydroxyapatite below pH 6.0. βMgTCP was found to be more soluble than zinc-containing β-tricalcium phosphate given the same molar content of zinc or magnesium. The solubility of βMgTCP and release rate of magnesium from βMgTCP can be controlled by adjusting the Mg content by selecting the appropriate p*K*sp.

Keywords

Magnesium-containing tricalcium phosphate; Solubility product; Dissolution

1. Introduction

Although magnesium (Mg) is the fourth major cation in the human body and the second most prevalent intracellular cation, approximately 50–65% of the total amount of Mg is in bone tissue and only 1.0% is in the extracellular fluid [1,2]. Mg is essential for the normal function of the parathyroid glands and vitamin D metabolism [3] and Mg depletion results in significantly lower serum parathyroid hormone (PTH) and $1,25(OH)_2$ -vitamin D levels [4]. Mg can directly stimulate osteoblast proliferation [5] and Mg depletion causes cellular growth inhibition because of the resultant reduction in DNA, RNA and protein synthesis [6,7]. Mg binds to the surface of apatite crystals and inhibits their formation and growth [8,9], and decreased bone Mg content and high crystallinity were found to be associated with senile osteoporosis [10]. These Mg-deficiency-associated adverse effects together could impair bone growth and mineralization, reduce bone quality, strength and density, and increase bone fragility [11,12]. Mg deficiency has been suggested to be a potential risk factor for osteoporosis

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[13–15]. Mg therapy was reported to increase the bone mass in patients suffering from postmenopausal osteoporosis [13,16,17]. Dietary Mg is absorbed in the intestine through both active and passive transport systems [18,19].

Mg-containing biomaterials have been fabricated, including Mg-containing calcium phosphates [9,20–26], Mg-containing bioactive glasses [27], akermanite $(Ca_2MgSi_2O_7)$ [28] and bredigite $(Ca_7MgSi_4O_{16})$ [29]. Among them, β-tricalcium phosphates in ceramic form are highly biocompatible and resorbable in bone tissue. Therefore, Mg-containing β-tricalcium phosphates (βMgTCP) can be used as an Mg carrier [30].

In vivo resorption of βTCP-based material is controlled by two factors: self-dissolution and cell-mediated dissolution. The evaluation of self-dissolution behavior aimed to control the release rate of Mg, which will help to study the effects of Mg release from βMgTCP on the cell-mediated dissolution or cell function. The self-dissolution rate $J = f(K_{\text{SD}})$ is a function of the solubility product $K_{\rm SD}$, which is the equilibrium ion activity product. However, the available data on the equilibrium solubility of βMgTCP is limited to only a narrow Mg content (7.7–13 mol.% Mg) [31]. In this study, the equilibrium solubility of βMgTCP was determined over a wider range of Mg contents: from 0 to 10 mol.%.

2. Materials and methods

βMgTCP powders were synthesized from pure βTCP (Advance Co. Ltd., Tokyo, Japan) and 10 mol.% βMgTCP (Ca_{2.7}Mg_{0.3}(PO₄)₂) prepared by the solution method described elsewhere [32]. Briefly, the starting materials were high-purity calcium carbonate (99.99 wt.%, Ube Materials, Ube, Japan), reagent-grade phosphoric acid (85 wt.%, Nakalai Tesque, Kyoto, Japan) and reagent-grade magnesium nitrate hexahydrate (99 wt.%, Wako Pure Chemical Industries, Ltd Co., Tokyo, Japan). A suspension of 0.6 M Ca(OH)₂ was prepared using calcium oxide obtained by heating calcium carbonate at 1000 °C for 3 h. The suspension wa s mixed with 2.6 M $Mg(NO_3)$ ²·6H₂O and a phosphoric acid solution, followed by filtration and heat treatment at 850 °C for 1 h to obtain the 10 mol.% βMgTCP powder. The mixture of 10 mol. % βMgTCP and various amounts of pure βTCP powders were ground for 20 min in an alumina mortar and heated at 1000°C for 5 h. βMgTCP powders with magnesium contents of 0.0, 2.3, 4.9, 7.3 and 10.1 mol.% were obtained (Table 1).

The βMgTCP powders at 15 mg each were immersed in 15 ml of 0.08 M CH₃COOH– CH₃COONa buffer solution of pH 5.5 at 25 ± 3 °C in poly ethylene bottles for 1.5, 3 and 27 months. The ratio of powders surface area to the solution volume was 0.00035 ± 0.00015 m² ml⁻¹, which was estimated from the grain size of the powders. TCP and MgTCP can convert into dicalcium phosphate dihydrate (DCPD), octacalcium phosphate (OCP) and hydroxyapatite (HAP) in an aqueous solution, depending on pH, which can lead to serious errors in measuring solubility. To minimize the error, we carried out the determination of the equilibrium solubility of TCP and MgTCP in a solution at pH 5.5, at which point the solubility of TCP was nearly equivalent to that of DCPD and OCP. The 1.5- and 3-month immersions were performed under air with slow shaking by bottle rotation. The 27-month immersion was performed without shaking in a nitrogen gas atmosphere to avoid contamination with atmospheric $CO₂$. The suspensions were then filtered using a Millipore filter of 0.22 μ m pore size.

The immersion solutions were analyzed for calcium, phosphorus and magnesium using the model SPS 7800 inductively coupled plasma (ICP) spectrometer (Seiko Instruments, Tokyo, Japan), within an error of $\pm 1.0\%$. The uncertainty of pH measurements was ± 0.02 pH units. X-ray diffraction (XRD) patterns of the βMgTCP powders were measured before and after the immersion using an X-ray diffractometer (RINT-2500, Rigaku, Japan). The βMgTCP powders

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before and after the immersion were characterized using a scanning electron microscope (SEM; JSM-5400, JEOL, Japan) and Fourier transform infrared spectroscopy (FTIR; JASCO FT/ IR-350, Japan).

The solubility product (K_{sp}) of β MgTCP was calculated from the equation:

$$
K_{sp} = (Ca^{2+})^{3-x} (Mg^{2+})^{x} (PQ_4^{3-})^{2}
$$

= {[Ca - T]f_{ca}}^{3-x} {[Mg - U]f_{Mg}}^{x} {K_{p3} [P - T - U]}² / {J(H⁺)}² (1)

and the Debye–Hückel limiting law,

$$
-log\ \mathbf{fi} = A\ Z_i^2 \mathbf{m}^{1/2} / (1 + B\ \mathbf{ai}\ \mathbf{m}^{1/2})
$$
 (2)

The definitions of all the abbreviations used in the equations and the equilibrium constants used in the calculations [33–38] are listed in Table 2.

The validities of the present computation algorithm and program are shown in Table 3. The present algorithm and program accurately reproduced published K_{sp} values for βTCP and βMgTCP [31,39] when the corresponding reported raw data on calcium, phosphorous and magnesium concentrations, and pH were used for the calculation.

The ion products of DCPD, OCP and HAP, which are expressed as $(Ca^{2+})(H^+)(PO_4^{3-})$, $(Ca^{2+})^4(H^+)(PO_4^{3-})^3$ and $(Ca^{2+})^{10}(PO_4^{3-})^6(OH^-)^2$, respectively, were also calculated from the measured concentrations and pH. These values were compared with the reported solubility products of DCPD, OCP and HAP in order to determine whether the solutions were saturated with respect to these phases as well after the equilibration.

3. Results

Figs. 1 and 2 show the XRD patterns of βMgTCP before and after the 27-month immersion, respectively. All the synthesized βMgTCP powders were proven to be single phase. Substitution of Mg²⁺ for Ca²⁺ in the β TCP structure was confirmed by the linear decrease in lattice constants with increasing Mg content (Table 1). After the 27-month immersion, the XRD patterns showed single-phase βMgTCP or whitlockite. Although slight increases of 0.03– 0.05% in the *a*-parameter were observed after the 27-month immersion compared with those before the immersion, the increases were not statistically significant.

Fig. 3 shows the SEM images of the βTCP and βMgTCP powders before and after the 27 month immersion. Before immersion, the βTCP and βMgTCP powder consisted of round grains, 2–5 μm in size, sintered partially by the heat treatment at 1000°C. After the 27-month immersion, the grains beca me polyhedral in shape due to dissolution on the grain surfaces and boundaries. βTCP grains after the immersion showed two distinct surfaces: a smooth surface with no precipitate and a rough surface with precipitate. βMgTCP grains after the immersion showed smooth surfaces only.

Fig. 4 shows the FTIR spectra of βTCP and βMgTCP before and after the 27-month immersion. βTCP showed its characteristic absorption bands at 543, 551, 570, 578, 589, 605 and 612 cm−¹ in the γ4 PO4 domain (a of Fig. 4A) and those at 970, 998, 1023, 1042, 1066,1080, 1097 and 1120 cm⁻¹ in the γ₃ PO₄ domain (a of Fig. 4B). βMgTCP with 10.1 mol.% Mg showed absorption bands at 548, 558, 557, 582, 594, 612 and 619 $\rm cm^{-1}$ in the γ_4 PO $_4$ domain (e of Fig.

4A). βMgTCP with 7.3 and 10.1 mol.% Mg showed an additional band at 986–990 cm⁻¹ that was absent in βMgTCP with Mg lower than 7.3 mol.% (Fig. 4 B). After the 27-month immersion, the absorption bands at 530–570 cm⁻¹ in the γ_4 PO₄ domain became sharp, and showed an increase in the ratio of band intensity at $551-558$ cm⁻¹ to that at $543-548$ cm⁻¹ (Fig. 4C). The increase in band intensity ratio indicated formation of whitlockite [40]. Whitlockite-specific bands at 986–990 and 1150 cm⁻¹ became strong and sharp in β MgTCP with 4.9, 7.3 and 10.1 mol.% Mg [41].

Tables 4–6 and Fig. 5 show the composition and pH values of the equilibrium solutions, and the negative logarithms of the solubility products (p*K*sp) of βTCP and βMgTCP. The higher the magnesium content of βMgTCP, the larger the amount of magnesium released (Tables 4– 6). However, the solubility of the βMgTCP powder as a whole decreased with increasing magnesium content.

The pK_{SD} of β TCP and β MgTCP as functions of magnesium content after different immersion times are shown in Fig. 5. The 1.5-month immersion under ambient air did not reach equilibrium, as evidenced by the pK_{sp} of βTCP (29.30 ± 0.04), which was significantly higher $(p<1 \times 10^{-5})$ than that (29.05 ± 0.11) obtained in the 3-month immersion. The p K_{sp} of β TCP after the 27-month immersion under nitrogen gas (28.87 ± 0.04) was significantly lower than that after the 3-month immersion under ambient air (29.05±0.11). It should be noted that the p*K*sp of βMgTCP after the 27-month immersion were significantly higher than those after the 3-month immersion under air ambient.

The relationship between the pK_{sp} and the magnesium contents of βTCP and $\beta MgTCP$ after the immersion for 1.5, 3 and 27 months was determined by regression analysis as Eqs. (3) , (4) and (5), respectively, using the fifth-degree equation:

$$
pK_{sp} = 29.29896 + 1.00779C - 0.2112C^{2} + 0.02983C^{3} - 0.00204C^{4} + 0.0000526763C^{5}
$$
\n(3)

$$
pK_{sp} = 29.0502 + 1.0016C - 0.24107C^{2} + 0.041C^{3} - 0.00342C^{4} + 0.000105953C^{5}
$$
\n⁽⁴⁾

$$
pK_{sp} = 28.87432 + 1.40348C - 0.3163C^{2} + 0.04218C^{3} - 0.00275C^{4} + 0.0000681659C^{5}
$$
\n
$$
\tag{5}
$$

where *C* is the magnesium content of β MgTCP in mol.%. Quadratic, cubic and quartic regression equations fit less precisely the plots of pK_{sp} as a function of the magnesium content than the fifth-degree equations. From the above results, it can be observed that, with increasing magnesium content, the pK_{sp} increased and the solubility of β MgTCP decreased under all three conditions.

Fig. 6 shows the index of incongruent solubility, $Δ[Mg/(Ca+Mg)]$, of the $βMgTCP$ defined as:

$$
\Delta [Mg/(Ca+Mg)] = (R_s - R_p)/R_p
$$

where R_s and R_p represent the Mg/(Ca+Mg) molar ratios of the equilibrium solution and immersed powder, respectively. Although the Mg/(Ca+Mg) molar ratios of the βMgTCP powders were 0.023, 0.049, 0.073 and 0.101, the equilibrium solutions showed the Mg/(Ca +Mg) molar ratios to be 0.019, 0.043, 0.072 and 0.108, respectively, after the 27-month

immersion. These changes in the Mg/(Ca+Mg) molar ratios gave Δ [Mg/(Ca+Mg)] values of −17.39, −14, −1.37 and 9.09% for βMgTCP with 2.3, 4.9, 7.3 and 10.1 mol.% Mg, respectively.

The negative logarithms of the ion products of HAP, OCP and DCPD showed that all the solutions were undersaturated with respect to DCPD and saturated or supersaturated with respect to HAP after the immersion (Table 7). In addition, the immersion solutions for βTCP were also saturated with respect to OCP while those for βMgTCP were undersaturated.

By using the p*K*sp, the solubility of βMgTCP was calculated as a function of pH in the range from 5.0 to 7.5, as shown in Fig. 7. The Ca+Mg concentration in the solution, reflecting the solubility of βMgTCP, decreased at higher pH. βMgTCP with 10.1 mol.% magnesium exhibited solubility lower than that of HAP below pH 6.0 and higher than that of HAP above pH 6.0.

4. Discussion

The equilibrium solubility of βMgTCP decreased with increasing magnesium content of βMgTCP up to 10 mol.%. From the SEM images of βTCP and βMgTCP, it can be seen that all the powders with different magnesium content showed a similar particle size of about 2– 5 μm. Base on the density of β TCP (3.067 g cm⁻³) and the particle size, we can estimate the specific surfaces of β TCP and β MgTCP to be about 0.20–0.50 cm² g⁻¹ and infer that the specific surfaces would not have obvious effects on the solubility. Moreover, in this study, 27 months were more than long enough to achieve the solubility equilibrium in the buffer solution. The decrease in solubility can be attributed to the increased stability of the β TCP structure caused by the addition of magnesium ions. Because the ionic radius of magnesium is 0.072 nm for six coordinations, magnesium ions reside in the Ca(4) and Ca(5) sites in the β TCP structure, which can accommodate divalent cations with ionic radii from 0.060 to 0.080 nm [42]. The increase in the stability of the βTCP structure has been demonstrated by an increase in the β-to- α transition temperature of TCP with the incorporation of magnesium [43]. TCP has three polymorphs: βTCP below 1180 °C, αTCP between 1180 and 1430 °C, and α[']TCP above 1430 °C. Even 1 mol.% Mg incorporated into the βTCP structure increases the β-to- α transition temperature up to 1230 \degree C [43]. An increase in the stability and, hence, a decrease in the standard free energy of formation of βTCP inevitably leads to an increase in the standard free energy of β TCP solultion ($\Delta_{\rm s}G^{\rm o}$), which is the sum of the standard free energies of solution of Ca²⁺, Mg^{2+} and PO₄³⁻ ions minus the standard free energy of formation of β TCP. The increase in $\Delta_s G^{\circ}$ leads to a decrease in the solubility constant (K_s) since K_s is given by $\Delta_s G^{\circ}$ = $-RT\ln K_s$, where *R* is the gas constant (8.314 J K⁻¹ mol⁻¹) and *T* is the absolute temperature [44]. A similar decrease in the solubility of βTCP was reported for zinc-containing βTCP (βZnTCP) [45].

Another possible reason for the solubility decrease could be the formation of a whitlockitetype phase on the surface of βMgTCP. In the case of βMgTCP, magnesium whitlockite was suggested to be the equilibrating solid present on the surface of βMgTCP and was responsible for the decrease in the solubility [31]. Although it is difficult to distinguish between magnesium whitlockite and βMgTCP by XRD, as shown in Fig 2, slight but nonsignificant increases in a lattice parameter were observed, indicating the presence of magnesium whitlockite [46]. FTIR spectra also supported the formation of a whitlockite-type phase. It is reported that whilockite has higher intensity ratio of the absorption band at $551-558$ cm⁻¹ to that at $543-548$ cm⁻¹ than βTCP has, and that the characteristic bands of whilockite appear at 990 and 1150 cm⁻¹ [40, 41]. In the present study, FTIR spectra of βMgTCP with 4.9, 7.3 and 10.1 mol.% Mg after the immersion showed an increase in the ratio of the absorption band intensity at $551-558$ cm⁻¹ compared to that at 543–548 cm⁻¹, and an increase in the absorption band intensity at around 986–990 and 1150 cm⁻¹. However, HPO₄^{2–} bands were not clearly observed in the βMgTCP

after the immersion, although the $HPO₄²$ bands appeared in the whitlockite prepared by the precipitation method or hydrothermal method using Ca^{2+} , Mg^{2+} and phosphate-containing solution [40]. From these results, it is possible that, taking advantage of structural similarity between βMgTCP and whitlockite, a whitlockite-type phase was formed only in a very thin surface layer of β MgTCP, which made the absorption band of HPO₄^{2–} difficult to detect by FTIR. SEM observation supports the above assumption: no appreciable precipitate with a rhombohedral morphology characteristic of whitlockite was observed by SEM among the βMgTCP particles and on the smooth surface of βMgTCP after the immersion.

The index of incongruent solubility, $\Delta[Mg/(Ca+Mg)]$, which reflects the difference in the Mg/ $(Ca+Mg)$ molar ratios between the initial $\beta MgTCP$ powder and equilibrium solutions, also indicates the presence of the whitlockite-type phase. The indexes of incongruent solubility of βMgTCPs with 2.3, 4.9, 7.3 and 10.1 mol.% magnesium were −17.39, −14, −1.37 and 9.09%, respectively, which indicate increases in the magnesium content in the former three βMgTCP powders and a decrease in the magnesium content in the last βMgTCP powder after the 27 month immersion. As a result, the magnesium contents of the βMgTCP powder changed to 2.5, 5.1, 7.4 and 10.0 mol.%, respectively, after the 27-month immersion. The increases and the decrease in magnesium content were statistically significant (*p*=5×10−4−1×10−10) except for the change from 7.3 to 7.4 mol.%. Note that the magnesium content is 10 mol.% $(C_{18}Mg_2H_2(PO_4)_{14})$ for synthetic magnesium whitlockite [47,48]. In addition, acidic conditions (pH 5 to 6) favor the crystallization of whitlockite [24,44,46,48], which is related to the fact that HPO_4^{2-} ions must be present in order to be incorporated into the whitlockite lattice. On the other hand, the ionic product corresponding to whitlockite, defined as

$$
IPw{=}(Ca^{2+})^9(Mg^{2+})(H^+)(PO_4{}^{3-})^7
$$

yielded a value of 106.2 ± 0.3 on the basis of the concentration and pH data of the 27-month immersion of βMgTCP powder with 10.1 mol.% magnesium. It was reported that whitlockite crystallized on whitlockite seeds in metastable supersaturated solutions with an IPw of 106.3 at 37 °C [48]. The solubility after the 3-month immersion under ambient air is significantly higher than that after the 27-month immersion under nitrogen, which could indicate the dissolution of βMgTCP and the precipitation of the whitlockite-type phase. All these observations suggest the formation of a whitlockite-type phase on the surface of βMgTCP, although further clarification is required.

The pK_{sp} reported here for β MgTCP are significantly lower than those reported previously (Fig. 5). p*K*sp for βMgTCP with magnesium contents of 0.0, 7.7, 11.6 and 13.3 mol.% were previously reported as 29.9 ± 0.1 , 32.9 ± 0.1 , 33.4 ± 0.2 and 33.5 ± 0.1 , respectively, which were obtained by 2-week equilibration with stirring [31]. However, the p $K_{\rm SD}$ reported here for βMgTCP with magnesium contents of 0.0, 7.7, 11.6 and 13.3 mol.% were calculated as 28.87, 32.36, 32.96 and 33.14, respectively, which were significantly lower than the above values. The discrepancy in pK_{sp} value is not explained by the differences in values used for the association constants of $CaH_2PO_4^+$ and $CaHPO_4^0$: the difference in the association constants leads only to a 0.03% difference in pK_{sp} at most. The pK_{sp} reported here for pure TCP (28.87) \pm 0.04) coincided with that reported by Gregory (28.90 \pm 0.07), with no statistically significant difference [39]. Therefore, the present 27-month immersion can be more completely equilibrated than that in the previous study.

βMgTCP was found to be more soluble than βZnTCP. The solubility of βZnTCP also decreased with increasing zinc content up to 10 mol.%, which was nearly the limit of the solid solution of zinc in TCP. The relationship between pK_{sp} and zinc content *C* is reported as follows:

 $pK_{sp} = 28.665 + 1.7308C - 0.40326C^{2} + 0.060659C^{3} - 0.0049308C^{4} + 0.00015974C^{5}$

βZnTCP exhibits a higher p*K*sp than MgTCP given the same molar ratio of zinc or magnesium, which shows that MgTCP is more soluble than ZnTCP. At a zinc content of 6 mol.%, the solubility of βZnTCP decreased to a level corresponding to that of HAP, while a magnesium content of 10.1 mol.% is required to reduce the solubility of βMgTCP to the same level.

Finally, the relationship between the solubility and magnesium content of MgTCP has been confirmed. Thus, the solubility and release rate of Mg can be controlled by adjusting the Mg content in MgTCP by selecting the appropriate pK_{sp} , if other factors such as particle size and surface area are unchanged.

Conclusion

The equilibrium solubility of βMgTCP was determined by immersing βMgTCP powders in the CH₃COOH–CH₃COONa buffer solution of pH 5.5 at 25 \pm 3 °C for 27 months under a nitrogen gas atmosphere. The negative logarithm of the solubility product (K_{sp}) of βMgTCP was expressed as p*K*sp = 28.87432 + 1.40348*C* − 0.3163*C*² + 0.04218*C*³ − 0.00275*C*⁴ + 0.0000681659 C^5 , where *C* is the magnesium content in β MgTCP in mol.%. The p K_{sp} for βMgTCP were significantly lower than those reported previously. The solubility of βMgTCP decreased with increasing magnesium content. This decrease in solubility results from an increase in the stability of βMgTCP upon magnesium incorporation. The decrease in solubility could also result from the formation of a whitlockite-type phase on the surface of βMgTCP during equilibration, which requires further clarification.

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The XRD patterns of βTCP and βMgTCP with different magnesium contents before immersion: (a) 0 mol.%; (b) 2.3 mol.%; (c) 4.9 mol.%; (d) 7.3 mol.%; (e) 10.1 mol.%.

Fig. 2.

The X-ray diffraction patterns of βTCP and βMgTCP with different magnesium contents after immersion for 27 months: (a) 0 mol.%; (b) 2.3 mol.%; (c) 4.9 mol.%; (d) 7.3 mol.%; (e) 10.1 mol.%.

Fig. 3.

SEM images for the powders of $βTCP(A)$ and $βMgTCP$ with 10.1 mol.% Mg (B) before immersion; and βTCP(C), β MgTCP with 2.3 mol.% Mg (D), 4.9 mol.% Mg (E), 7.3 mol.% Mg (F) and 10.1 mol.% Mg (G) after immersion for 27 months.

Fig. 4.

FTIR spectra of βTCP and βMgTCP before (A, B) and after (C, D) immersion for 27 months. The Mg contents are 0 mol.% (a), 2.3 mol.% (b), 4.9 mol.% (c), 7.3 mol.% (d) and 10.1 mol. % (e). The arrows indicate bands characteristic of whitlockite.

Fig. 5.

Negative logarithms of the solubility product of βTCP and βMgTCP as a function of the magnesium content after immersion for 1.5, 3 and 27 months; comparison to the previous p*K*sp data of βTCP (published by Gregory et al. [33]) and βMgTCP (published by Mcdowell et al. [31]).

The magnesium exchange percentage between the initial powder and the solution after the 1.5-, 3- and 27-month immersions of βMgTCP with different Mg contents.

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*** The values were calculated by using the Ca, Mg and P concentrations and pH values published in the literature.

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Table 4

Solubility of βTCP and βMgTCP after immersion for 1.5 months at 25

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Table 7
Negative logarithms of ion products for various phases Negative logarithms of ion products for various phases

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g and §§, indicate saturation and supersaturation with respect to the corresponding phase. Values of 116.65–118.65, 46.9–49.6 and 6.55 have been used for the negative logarithms of solubility products for the negative loga *§ and §§*indicate saturation and supersaturation with respect to the corresponding phase. Values of 116.65–118.65, 46.9–49.6 and 6.55 have been used for the negative logarithms of solubility products for HAP, OCP and DCPD, respectively, to define saturation and supersaturation. for HAP, OCP and DCPD, respectively, to define saturation and supersaturation.