

REVIEW

# Bone tissue engineering therapeutics: controlled drug delivery in three-dimensional scaffolds

Viviana Mouriño and Aldo R. Boccaccini\*

*Department of Materials, Imperial College London, Prince Consort Road,  
London SW7 2BP, UK*

This paper provides an extensive overview of published studies on the development and applications of three-dimensional bone tissue engineering (TE) scaffolds with potential capability for the controlled delivery of therapeutic drugs. Typical drugs considered include gentamicin and other antibiotics generally used to combat osteomyelitis, as well as anti-inflammatory drugs and bisphosphonates, but delivery of growth factors is not covered in this review. In each case reviewed, special attention has been given to the technology used for controlling the release of the loaded drugs. The possibility of designing multifunctional three-dimensional bone TE scaffolds for the emerging field of bone TE therapeutics is discussed. A detailed summary of drugs included in three-dimensional scaffolds and the several approaches developed to combine bioceramics with various polymeric biomaterials in composites for drug-delivery systems is included. The main results presented in the literature are discussed and the remaining challenges in the field are summarized with suggestions for future research directions.

**Keywords:** bone tissue engineering; drug release; drug-delivery system; scaffolds; biodegradable materials; bioactive glass

## 1. INTRODUCTION

Even though osseous tissue has the unique internal repair capacity to heal and remodel without scarring, there are several conditions, both congenital and acquired, where bone replacement is needed (Buckwalter *et al.* 1996*a,b*). Furthermore, the clinical need to effectively treat bone defects is expected to increase as population ageing continues to grow. Autologous grafting is the therapeutically obvious gold standard in reconstructive surgery owing to high immunocompatibility (Reichert *et al.* 2009). However, this concept is bound by several constraints (e.g. requirement of secondary surgery, limited amount of tissue that can be harvested, increased risk of infection or recurrent pain) (Younger & Chapman 1989; Arosarena 2004). Allograft or xenograft graftings are optional treatments, but each process has its own disadvantages including the possibility of graft rejection by the immune system, risk of infections and transmission of donor pathogens (Kumar 2006). To overcome these problems, tissue engineering approaches are emerging as convenient alternatives to promote the regenerative ability of the host body (Goessler *et al.* 2007; Lee & Shin 2007; Bran *et al.* 2008; Kanczler & Oreffo 2008). One of the most important stages of bone tissue engineering (TE) is the design and processing of a porous,

biodegradable three-dimensional structure called a ‘scaffold’, exhibiting high porosity, high pore interconnectivity and uniform pore distribution (Hutmacher 2000). These scaffolds should provide structural support for cells and the new tissue being formed, acting as a temporary extracellular matrix inducing the natural processes of tissue regeneration and development (Lee & Shin 2007). Thus, the improvement of the biological activity and performance of bone-substitute materials and scaffolds is one of the main concerns in bone regeneration (Le Bolay *et al.* 2009).

Further, bone tissue requires the action of growth factors that provide signals at local injury sites allowing progenitors and inflammatory cells to migrate and trigger the healing process (Furth *et al.* 2007; Lee & Shin 2007). Several attempts have been made to include growth factors and morphogens within bioactive scaffolds to stimulate cellular adhesion, proliferation and differentiation, in order to promote bone regeneration (Habraken *et al.* 2007; Lee & Shin 2007). In addition, enhancing further the functionality of these already complex matrices by loading drugs into them to treat bone disorders or to act on the surrounding tissues with an adequate therapeutic concentration level and for a desired time frame is recognized as being highly beneficial, hence the increasing interest in incorporating a drug-delivery function in TE applications

\*Author for correspondence (a.boccaccini@imperial.ac.uk).

(Gomes & Reis 2004; Duarte *et al.* 2009a). The increasing volume of work dealing with this approach is leading to the establishment of an emerging field which has been termed TE therapeutics (Baroli 2009).

Three-dimensional bioactive bone scaffolds can be fabricated from bioceramics, biodegradable polymers and their composites (Rezwan *et al.* 2006). Bioactive ceramic scaffolds alone used in bone TE can serve as a delivery vehicle of drugs but the drug release patterns are difficult to control (Habraken *et al.* 2007). On the other hand, biodegradable polymeric materials such as poly(lactic-co-glycolic acid) (PLGA) (Garvin & Feschuk 2005) and poly(propylene glycol-fumerate)/methylmethacrylate (Gerhart *et al.* 1993) can be used to control the local delivery of drugs; however, they show impaired osteoconduction and sometimes they provoke an adverse tissue response owing to inflammation as a consequence of acidic degradation (Böstman & Pihlajamäki 2000). Thus, the smart combination of bioceramics (including calcium phosphates (CaP), hydroxyapatite (HA) and silicate bioactive glasses (BGs)) and biodegradable polymers can not only improve the degradability of the inorganic material and alter its mechanical/physical properties (Rezwan *et al.* 2006), but also drug-release profiles can be controlled to a greater extent than on pure ceramics (Habraken *et al.* 2007). There is a wide range of different polymers that can be selected for this application, which show different degradation rates and mechanisms (Nair & Laurencin 2007), and a wide range of bioceramic/biopolymer composite scaffolds is available for bone TE (Rezwan *et al.* 2006; Guarino *et al.* 2007; Yunos *et al.* 2008).

There are different possible routes of administration for most drugs, which can be broadly divided into two categories: local and systemic (Somayaji *et al.* 1998). Drugs delivered systemically are absorbed into the blood stream and distributed throughout the body via the circulatory system, which can result in systemic toxicity with associated renal and liver complications, poor penetration into the targeted tissue and the need for hospitalized monitoring (Price *et al.* 1996; Somayaji *et al.* 1998; Ruszczak & Friess 2003). When a drug is delivered locally, the side effects and risk of overdose of systemic administration can be limited and there is a higher concentration of medication effectively reaching the targeted site (Somayaji *et al.* 1998). Moreover, controlled release systems might be helpful to improve drug bioavailability because they could be designed to deliver a drug molecule at a specific rate and for a specific period of time at the desired location (Roman & Madrugá 1989). The rate and manner in which a particular drug is released are determined by the matrix into which the drug is loaded, the type of drug and its clearance rate (Wu & Grainger 2006; Zilberman & Elsner 2008; Baroli 2009). Different strategies followed to deliver specific growth factors (e.g. cytokines and hormones); morphogens and proteins using TE scaffolds to stimulate cellular adhesion, proliferation and differentiation, thus promoting bone regeneration, have been proposed and these have been reviewed in the literature (Saltzman & Olbricht 2002; Chung & Park 2007; Habraken *et al.* 2007; Lee &

Shin 2007; Baroli 2009). However, there has been no previous review, to the authors' knowledge, focusing exclusively on TE approaches to deliver therapeutic (synthetic) drugs. The review, thus, focuses specifically on the rationale to load different therapeutic drugs relevant for treating pathologies associated with the bone repair process in three-dimensional scaffolds used in bone TE. Specific attention has been given to the most recent developments related to controlling the release rate of the relevant therapeutic drugs, which is the marker that defines the degree of success of a given strategy. Approaches followed to deliver growth factors and morphogens will not be covered in this review. Moreover, the review focuses on TE therapeutic strategies based only on solid, three-dimensional scaffolds, e.g. injectable systems are not included. The paper is organized in the following manner: §2 briefly discusses the key biomaterials used for bone tissue scaffolds, presenting their relative advantages and disadvantages. This section also covers general approaches used for the appropriate formulation of scaffolds with a focus on processing methods of relevance to scaffolds as drug-delivery carriers. Sections 3 and 4 present the specific strategies adopted to develop TE scaffolds with drug-delivery capability for treating relevant pathologies associated with bone, focusing on antimicrobial agents (§3) and on other relevant used drugs such as anti-inflammatory and antiresorptive pharmacotherapeutics (§4). Subsequently, §5 focuses on the key variables needed to be taken into account to investigate *in vitro* drug release from particular scaffolds used. Finally, remaining challenges in the field are summarized in §6, where directions for future research efforts are also highlighted.

## 2. BIOACTIVE BONE SCAFFOLDS AS THERAPEUTIC DRUG CARRIERS: THE ADDED VALUE

### 2.1. Scaffold materials

An optimal scaffold for bone TE should be osteoconductive and angiogenic and it should serve as a three-dimensional template to provide structural support to the newly formed bone through an interpenetrating network of pores (at least 100  $\mu\text{m}$  wide) to allow cell migration, tissue in-growth and vascularization (Freyman *et al.* 2001; Hutmacher 2001; Guarino *et al.* 2007; Moroni *et al.* 2008). Analysis of the literature demonstrates that a wide range of three-dimensional bioactive scaffolds has been developed which can be potentially used as delivery systems for therapeutic drugs relevant for bone repair processes. Basically, three major scaffold materials are used: ceramics, polymers and composites and their main characteristics are described briefly below. Comprehensive descriptions of scaffold fabrication techniques are available in the specialized literature (Hutmacher 2000; Guarino *et al.* 2007; Chen *et al.* 2008; Moroni *et al.* 2008). Table 1 (Chen *et al.* 2008) summarizes the advantages and disadvantages of the scaffold materials used for bone TE, while the techniques already employed to fabricate bone scaffolds with

Table 1. Advantages and disadvantages of scaffold materials used in three-dimensional bone TE: ceramics, polymers and composites (adapted from Chen *et al.* 2008). ++, excellent; +, good; HA, hydroxyapatite;  $\beta$ -TCP,  $\beta$ -tricalcium phosphate.

material	advantages	disadvantages
inorganic		
calcium phosphates (HA, $\beta$ -TCP)	(i) biocompatibility (++) (ii) osteoconductivity (+) (iii) adequate mechanical strength	(i) slowly degradable (crystalline phase) (ii) brittle
bioactive glasses and glass ceramics (bioglass, phosphate glasses)	(i) biocompatibility (++) (ii) osteoconductivity (++) (iii) tailorable resorption (+) (iv) angiogenetic (+) (v) upregulation of gene expression in osteoblasts (++) (vi) adequate mechanical strength	(i) slowly degradable (in case of partially crystalline structures) (ii) brittle (amorphous structure)
organic		
bulk biodegradable polymers (poly(lactic acid), poly(glycolic acid), poly(lactic-co-glycolic acid), poly(propylene fumarate))	(i) biocompatibility (++) (ii) wide range of biodegradation rate (++) (iii) bioresorbable (+)	(i) inflammatory response possible (acid degradation products) (ii) poor stiffness and compression strength
surface bioerodible polymers (poly(ortho esters), poly(anhydrides), poly(phosphazene))	(i) biocompatibility (++) (ii) retention of mechanical integrity possible	(i) they cannot be completely replaced by new bone tissue
composites		
	(i) biocompatibility (++) (ii) osteoconductivity (+) (iii) tailorable degradation rate (iv) improved mechanical properties	(i) fabrication techniques can be complex

potential therapeutic drug-release capabilities are listed in table 2. Figure 1 presents images showing the pore structure of selected ceramic and polymer scaffolds fabricated using different techniques, which are finding successful applications in bone TE developments (Hutmacher 2000; Hollister 2005; Chen, Q. Z. *et al.* 2006). Some of these techniques have already been considered for fabricating scaffolds with therapeutic drug-delivery function (summarized in table 2 and discussed in detail in sections below). These scaffolds can be developed with the aim of providing not only the physico-chemical environment and the structural integrity required for bone regeneration (which is the main scaffold function) but also with the added value of acting as a local regulator to control the dose and kinetics of therapeutic drug release without the need to use an additional drug carrier (Berger *et al.* 1997). Several comprehensive reports covering the state-of-the-art techniques developed to fabricate bone scaffolds are available and the field continues to expand (Hollister 2005; Guarino *et al.* 2007; Hutmacher *et al.* 2007; Lickorish *et al.* 2007; Moroni *et al.* 2008). However, while several of the proposed methods have been conveniently adapted for incorporating growth factors (e.g. Baroli 2009), only a limited number of techniques have been specifically used for fabricating scaffolds with therapeutic drug-delivery capability (table 2). In the following sections we summarize the progress in the field of bone TE scaffold development with potential application as therapeutic drug-delivery vehicles.

The sections are organized by group of scaffold materials, e.g. inorganic, polymeric and composite scaffolds, with the discussion of processing techniques being limited to those methods already applied for therapeutic drug loading.

## 2.2. Inorganic scaffolds

### 2.2.1. Characteristics of bioactive ceramics and glasses.

Synthetic bioactive ceramics such as some forms of CaP (HA,  $\alpha$ -tricalcium phosphate (TCP)) and combinations of them have been widely studied as scaffold materials owing to their structural and chemical similarity with the inorganic component of bone (Yuan *et al.* 1999; Hutmacher 2000; Dong *et al.* 2001; Zhang *et al.* 2001; Ramay & Zhang 2003; Karageorgiou & Kaplan 2006). However, some of these ceramics degrade very slowly, with crystalline HA being a very stable CaP in the biological environment (table 1) (Maccacci *et al.* 2007; Chen *et al.* 2008). More recently, bioactive silicate glasses and glass-ceramics have been investigated in their application as scaffolds (Livingston *et al.* 2002; Di Nunzio *et al.* 2004; Vitale-Brovarone *et al.* 2004; Chen, Q. Z. *et al.* 2006; Fu *et al.* 2007; Vitale-Brovarone *et al.* 2008). This is a group of inorganic materials with several unique properties favourable for bone TE. Bioactive glass is a class A bioactive material exhibiting the capability to bond to both bone and soft tissue and to stimulate bone growth (Hench & Polak 2002). *In vitro*, BGs are able to form a surface layer of

Table 2. Techniques developed to elaborate porous three-dimensional scaffolds for bone TE already applied to fabricate scaffolds with therapeutic drug-delivery capability.

technique	details of the process	references
melt moulding	<ul style="list-style-type: none"> <li>(i) scaffolds are prepared by melting polymers/ceramics in the presence of porogens (such as sodium chloride, sugar crystals)</li> <li>(ii) once the mixture is cooled, porosity is achieved by dissolving the porogens in water</li> <li>(iii) finally, the porous scaffolds are usually lyophilized</li> </ul>	Di Nunzio & Verné (2005)
solvent-casting	<ul style="list-style-type: none"> <li>(i) scaffolds are prepared by dissolving/suspending polymers/ceramics in the presence of porogens (such as sodium chloride or sugar crystals)</li> <li>(ii) porosity is achieved by dissolving the porogens in water</li> <li>(iii) after pouring the mixture into a mould, solvents are removed by either evaporation or vacuum/freeze-drying</li> </ul>	Thomson <i>et al.</i> (1998)
freeze-drying	<ul style="list-style-type: none"> <li>(i) scaffolds are prepared by dissolving/suspending polymers/ceramics in water or in an organic solvent followed by emulsification with a water phase</li> <li>(ii) after pouring the mixture into a mould, solvents are removed by freeze-drying and porous structures are obtained</li> </ul>	Whang <i>et al.</i> (1999); Cabañas <i>et al.</i> (2009)
liquid/liquid thermally induced separation technique	<ul style="list-style-type: none"> <li>(i) scaffolds are prepared by dissolving/suspending polymers/ceramics in a solvent which freezes below the phase separation temperature of the polymer solution</li> <li>(ii) porous structure is obtained by successively freeze-drying</li> </ul>	Zhang & Zhang (2002)
foaming	<ul style="list-style-type: none"> <li>(i) effervescent salts (ammonium bicarbonate) are used as porogens and mixed with an organic viscous solution/suspension of polymer/ceramic</li> <li>(ii) after solvent evaporation, porosity is achieved by placing scaffolds into hot water or an aqueous solution of citric acid to dissolve the salts</li> <li>(iii) an alternative is CO<sub>2</sub>-based gas</li> </ul>	Mooney <i>et al.</i> (1996); Harris <i>et al.</i> (1998)
template technique	<ul style="list-style-type: none"> <li>(i) scaffolds are prepared by dipping a polyurethane sponge into a slurry of proper viscosity containing ceramic particles</li> <li>(ii) the impregnation step and the removal of the surplus slurry should be adjusted to obtain, after the sponge removal, a defect-free porous three-dimensional scaffold</li> <li>(iii) sometimes, in order to obtain mesoporous structures, surfactants may be added</li> </ul>	Vitale-Brovarone <i>et al.</i> (2007); Chen, Q. Z. <i>et al.</i> (2006)
sol-gel	<ul style="list-style-type: none"> <li>(i) scaffolds are prepared by dissolving inorganic metal salts or metal organic compounds in a solvent where a series of hydrolysis and polymerization reactions allow the formation of a colloidal suspension ('sol')</li> <li>(ii) after casting the 'sol' into a mould, a wet 'gel' is formed</li> <li>(iii) with further drying and heat treatment, the 'gel' is converted into dense ceramic or glass articles</li> </ul>	Domingues <i>et al.</i> (2004)
powder compaction	<ul style="list-style-type: none"> <li>(i) scaffolds are prepared by compressing the polymers/ceramics using projectiles or punch and dies</li> <li>(ii) the velocity of compaction of the projectile or punch and dies is adjusted to achieve powder consolidation and the desired porosity</li> <li>(iii) the process can include sintering</li> <li>(iv) an alternative is to use uniaxial or isostatic pressing</li> </ul>	Kimakhe <i>et al.</i> (1999); Vallet-Regí <i>et al.</i> (2001); Castro <i>et al.</i> (2005); Miyai <i>et al.</i> (2008)

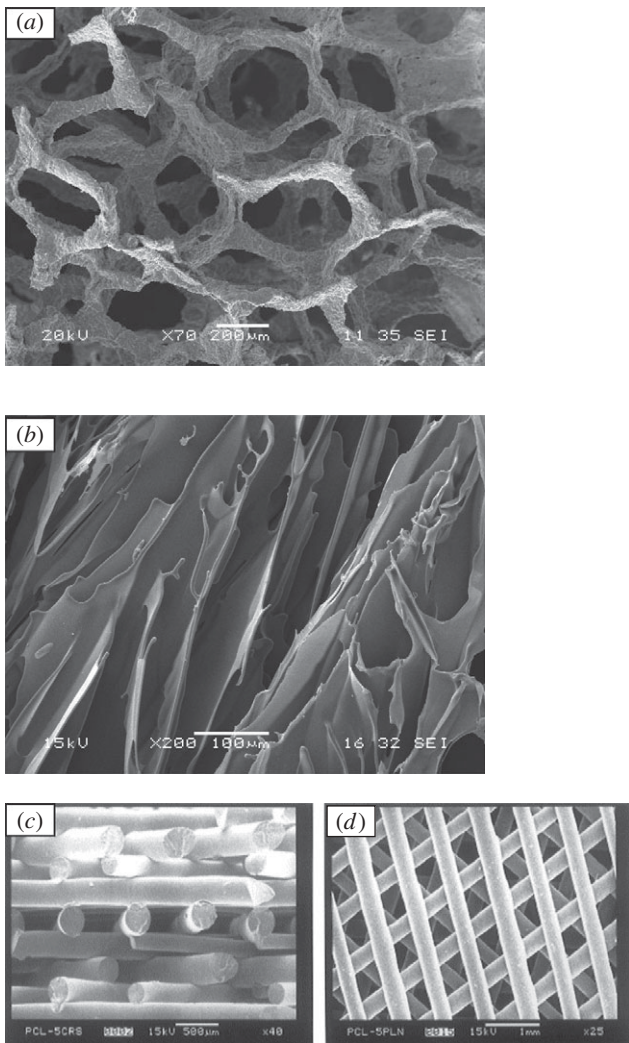


Figure 1. Scanning electron microscopy images showing typical pore structures of scaffolds successfully developed for bone TE: (a) bioactive glass-ceramic scaffolds fabricated by the foam replica method (Chen, Q. Z. *et al.* 2006); (b) PDLLA foam fabricated by thermally induced phase separation (Maquet & Jerome 1997); (c) polymer scaffold architectures developed by FDM (Hutmacher 2000). (Part (c) reproduced from Hutmacher (2000) with permission of Elsevier.)

microcrystalline HA when in contact with simulated body fluid (SBF) (Kim *et al.* 1995; Hench 1998; Fujibayashi *et al.* 2003). More recently, it has been discovered that ionic dissolution products released from silica-based BGs upregulate seven families of genes found in osteoblasts (Xynos *et al.* 2001). Dissolution products of BGs have also been shown to promote angiogenesis (Day *et al.* 2004). Additionally, research efforts on mesoporous inorganic materials have led to the development of ceramic matrices with both drug delivery and bone tissue regeneration capabilities (Vallet-Regí *et al.* 2006), which has promoted these biomaterials to novel uses in the biomedical field (Izquierdo-Barba *et al.* 2008).

**2.2.2. Processing techniques.** Numerous techniques have been developed to fabricate porous ceramic and glass-ceramic scaffolds, including starch consolidation, incorporation of volatile organic particles, sol-gel,

foaming, solid-free form fabrication methods and replication of polymeric sponges (figure 1a) (Livingston *et al.* 2002; Vitale-Brovarone *et al.* 2004, 2006, 2007; Chen, Q. Z. *et al.* 2006; Ahmad *et al.* 2008; Yun *et al.* 2008; Yunos *et al.* 2008). In order to maximize the mechanical strength of the scaffold, and depending on the fabrication technique used, different parameters such as solid loading of the ceramic slurry, type and amount of additives (binders, dispersants, etc.) should be optimized. However, the condition for preserving the activity of a drug is that the process of drug incorporation should not degrade it. Traditionally, fabrication of bioceramic scaffolds involves thermal treatment at high temperature ( $>800^{\circ}\text{C}$ ), hence this process cannot be used if a drug will be incorporated because most of it would be degraded (Vallet-Regí *et al.* 2008). Therefore, different methods have been proposed to tackle this problem. One of the simplest strategies to load drugs within ceramic scaffolds is conventional impregnation, which consists of immersing the scaffold in a drug containing phosphate-buffered saline (PBS) (Habraken *et al.* 2007). In this simple case, however, the drug release rate usually cannot be adequately controlled (Habraken *et al.* 2007). Sol-gel technology allows the introduction of drugs into glass-like materials at room temperature where pore formation is due to byproducts (water and alcohols) of the sol-gel synthesis (Nicoll *et al.* 1997; Santos *et al.* 1999). These byproducts are volatile and evaporate during the ageing and drying of the sol-gel materials, thus leaving pores (Nicoll *et al.* 1997; Santos *et al.* 1999). However, materials with optimal scaffold structure (exhibiting high porosity and large interconnected macropores) are in general unable to hold a large reservoir of drugs and they are not capable of retaining the drug at the site for a prolonged period of time (Malafaya *et al.* 2002; Ziegler *et al.* 2002). Further, solvents used in sol-gel methods have to be removed with the disadvantage of adding a drying step to the process. This drying step usually includes heating, which, even if at moderate temperatures, can degrade thermolabile substances. One of the first successful attempts to solve this difficulty was the use of uniaxial and isostatic pressure at room temperature to obtain gentamicin-loaded bioceramics (Vallet-Regí *et al.* 2001).

### 2.3. Polymer scaffolds

**2.3.1. Characteristics of biodegradable polymers.** An alternative approach to obtain multifunctional three-dimensional bioactive bone scaffolds includes their fabrication from biodegradable polymers (Maquet & Jerome 1997; Hutmacher 2000; Hollister 2005; Rezwani *et al.* 2006; Guarino *et al.* 2007). In this case, it is considered that biodegradable polymers can control the release of drugs since polymer degradation properties can be tailored for each specific application (Hutmacher 2000). Thus, a range of processes including cell growth, tissue regeneration and host response can be influenced by choosing suitable biopolymers (Lu *et al.* 1999; Hutmacher 2000). It is known that cells usually 'prefer' hydrophilic surfaces; however, hydrophobic

materials have typically longer residence times *in vivo* (Streubel *et al.* 2000; Malafaya *et al.* 2002), meaning that the selection of the optimal biodegradable polymer for the drug-delivery function in TE scaffolds should be aimed at balancing both aspects. Synthetic biodegradable polymers that have been reported for various drug-eluting devices include PLGA copolymers (Garvin *et al.* 1994; Nie *et al.* 1995; Overbeck *et al.* 1995; Ambrose *et al.* 2003), polycaprolactone (Hendricks *et al.* 2001; Rutledge *et al.* 2003), polyanhydrides (Jacob *et al.* 1991; Nelson *et al.* 1997; Kanellakopoulou & Giamarellos-Bourboulis 2000; Li, L. C. *et al.* 2002; Li, W. 2002), polyhydroxybutyrate-co-hydroxyvalerate (PHBV) (Yagmurly *et al.* 1999; Rossi *et al.* 2004) and other polyhydroxyalkanoates (Turesin *et al.* 2001). Natural polymers including proteins such as collagen (Lee *et al.* 2002; Park *et al.* 2004; Sripriya *et al.* 2004; Prabu *et al.* 2006; Shanmugasundaram *et al.* 2006) and polysaccharides such as alginate, hyaluronic acid and chitosan (Muzzarelli *et al.* 1990; Chung *et al.* 1994; Mi *et al.* 2002; Aoyagi *et al.* 2007; Rossi *et al.* 2007) are also attractive, since they exhibit superior biocompatibility and can facilitate cell growth. Many of these polymers are inexpensive and readily available. Moreover, ‘surface-eroding polymers’, which have the feature of being biodegraded only at their surfaces, are also candidates for this application. Poly(anhydrides), poly(orthoesters) and polyphosphazene show this property (‘surface eroding’) as opposed to ‘bulk degradation’ (Rezwan *et al.* 2006). Surface-eroding polymers offer advantages over bulk degradation polymers when used as multifunctional scaffold materials: (i) they are capable of releasing entrapped drugs by zero-order kinetics (when drug release can be limited to regions undergoing degradation) and (ii) the release rate is proportional to the surface area of the matrix (Heller 1985; Rezwan *et al.* 2006).

**2.3.2. Processing techniques.** Porous polymer scaffolds can be produced through thermally induced phase-separation (TIPS) (figure 1b), evaporation, freeze-drying, solid-free form fabrication (e.g. fused deposition modelling (FDM), three-dimensional printing, selective laser sintering) (figure 1c), solvent-casting and foam-coating among others techniques (Maquet & Jerome 1997; Thomson *et al.* 1998; Whang *et al.* 1999; Zhang & Zhang 2002; Boccaccini & Maquet 2003; Guan & Davies 2004; Cabañas *et al.* 2009). While a range of these methods have already been considered to incorporate therapeutic drugs, other suitable scaffold manufacturing methods, e.g. solid-free form fabrication methods (Hutmacher 2000; Hollister 2005) (figure 1c), have not yet been investigated with this purpose, thus remaining an area for future research.

Recently, impregnation of three-dimensional porous scaffolds using carbon dioxide (scCO<sub>2</sub>) has been used to develop alternative clean processes for the preparation of drug-loaded polymeric matrices when the drug is soluble in scCO<sub>2</sub> and the polymer chosen can be swollen by the supercritical fluid (Mooney *et al.* 1996; Harris *et al.* 1998; Kikic & Sist 2000; Duarte *et al.* 2007; Heyde *et al.* 2007). Impregnation using

supercritical fluid technologies has the advantage of using a supercritical fluid with high diffusivity in the polymer chosen in addition to its high solubility and plasticizing capability (Berens *et al.* 1992; Kazarian 2000; Duarte *et al.* 2007). Electrospinning is another technique with a potential to develop bioactive three-dimensional bone scaffolds with drug-release capabilities (Li, W. *et al.* 2002; Matthews *et al.* 2002; Ma *et al.* 2005; Gupta *et al.* 2007). However, this technique seems to have its limitations because some organic solvents used to prepare the polymer solutions may degrade the drugs (Han & Gouma 2006; Gupta *et al.* 2007; Ahmad *et al.* 2008; Baroli 2009). Finally, solid-free form fabrication technologies such as three-dimensional printing and FDM appear promising for further developments in this field, as mentioned above (Hollister 2005; Hutmacher & Cool 2007; Hutmacher *et al.* 2007; Pang *et al.* 2007; Wang *et al.* 2007).

## 2.4. Composite scaffolds

**2.4.1. Characteristics of biodegradable and bioactive composites.** In this approach, usually the matrix is prepared by using biodegradable polymers and inclusions in the form of particles of fibres of BG, CaP or HA are added to improve the mechanical strength and bioactivity (Roether *et al.* 2002; Rezwan *et al.* 2006; Guarino *et al.* 2007; Vallet-Regí *et al.* 2008). Polymers combined with ceramic particles, such as HA, can also be applied as coating on porous bioceramic scaffolds, fabricated using one of the techniques mentioned in §2.2.2, in order to tailor the controlled release of a drug (Kim *et al.* 2004a,b; Yunos *et al.* 2008). Composite scaffolds represent a convenient alternative as they combine the advantages of both biodegradable polymers and bioactive ceramics for bone engineering scaffolds. This combination leads to composites with improved mechanical properties because of the inherent higher stiffness and strength of the inorganic material. In addition, bioactive inorganic particles such as HA, bioglass or tricalcium phosphate induce the effective interaction of the scaffold with the surrounding bone tissue by forming a tenacious bond via the growth of a carbonate HA layer, as mentioned above (Rezwan *et al.* 2006). Moreover, addition of inorganic materials to bioresorbable polymers can change the polymer degradation behaviour by buffering the pH of the nearby solution, thus preventing the autocatalytic effect of the acidic end groups resulting from hydrolysis of polymer chains, e.g. in polylactic acid. It is well known that incorporation of bioactive inorganic phases in biodegradable polymers can enhance water ingress owing to the internal interfaces formed between the polymer and the more hydrophilic bioactive inclusions, hence enabling control of the degradation kinetics of scaffolds (Boccaccini & Maquet 2003).

**2.4.2. Processing techniques.** As mentioned above, one of the simplest strategies is the application of biodegradable polymer coatings loaded with the relevant drugs onto the three-dimensional structure of bioceramic scaffolds (Kim *et al.* 2004b). Other techniques reported are: solvent-casting, TIPS, evaporation,

freeze-drying and foam-coating (table 2—Thomson *et al.* 1998; Whang *et al.* 1999; Zhang & Zhang 2002; Boccaccini & Maquet 2003; Cabañas *et al.* 2009). An interesting approach to drug delivery in TE is to combine drug-loaded microspheres within the scaffold macroporous structure or even to use the loaded microspheres to construct an open-pore three-dimensional scaffold structure (Malafaya *et al.* 2002; Vallet-Regí *et al.* 2008; Zhu & Kaskel 2009). The approach of combining microspheres and a matrix (scaffold) could be very useful in cases where the scaffold material is not suitable for the regulated release of specific drugs, such as low-molecular weight or water-soluble drugs (Malafaya *et al.* 2002). When microspheres are added to an inorganic scaffold, they can interact with the surface of the scaffold by electrostatic charges depending on the nature of the biodegradable polymer and the bioceramic used (Xu & Czernuszka 2008). Moreover, by using microspheres, drug-release profiles can be altered and tuned depending on the polymer selected, as mentioned above.

A summary of experimental research carried out on the development of three-dimensional scaffolds for bone TE with controlled-release capability is presented in table 3 including both *in vitro* and *in vivo* studies. From this table it appears that porous matrices based mainly on well-characterized biocompatible polymeric scaffolds or, in some cases, composites comprising polymeric matrices and added inorganic particles represent the best systems to date to incorporate therapeutic drug delivery in bone TE approaches. A schematic diagram summarizing the different strategies proposed so far is shown in figure 2. Although several novel techniques have been developed to introduce therapeutic drugs within scaffolds, in most cases the strategy followed has been the direct incorporation of the drug into the scaffold by immersion of the scaffold in a drug-containing buffer aqueous solution. Nevertheless, thermo-labile drugs can also be loaded within three-dimensional scaffolds in a one-step process using room temperature compaction of the powder mix, this being a solvent-free process, which avoids the use of toxic solvents (Kimakhe *et al.* 1999; Vallet-Regí *et al.* 2001; Castro *et al.* 2005). Further details of relevant studies are discussed in the following sections, highlighting the strengths and weaknesses of the different approaches.

### 3. ANTIMICROBIAL AGENTS

#### 3.1. Controlled release of antibiotics

Infection is defined as a homeostatic imbalance between host tissue and the presence of micro-organisms at a concentration that exceeds  $10^5$  organisms per gram of tissue or the presence of beta-haemolytic streptococci (Sussman & Bates-Jensen 2001; Zilberman & Elsner 2008). The principal aim of treating wound infections is to reduce the bacterial load in the wound to a level at which wound-healing processes can take place. If the drug is released too quickly, the entire drug amount could be released before the infection is stopped. On the other hand, if the release of the drug is delayed, infection may set in further, thus making it

difficult to manage the healing of the wound (Gold & Moellering 1996; Gransden 1997; Zilberman & Elsner 2008). Bacterial infection remains a major problem affecting the service life of medical implants and scaffolds (Segreti 2000; Virk & Osmon 2001). Generally, sources for contamination include the air of the operating room and resident bacteria on the patient's skin and bacteria already present in the body (An & Friedman 1996; Zilberman & Elsner 2008). In addition to human pain and suffering, direct medical costs associated with such infections are often extremely high (Hetrick & Schoenfisch 2006). Similar complexities encountered with conventional orthopaedic implants can be expected in TE approaches based on implantation of engineered biomaterial scaffolds. Device-associated infections are the result of bacterial adhesion and subsequent bio-film formation at the implantation site. Inhibiting bacterial adhesion is often regarded as the most critical step in preventing infection. Upon implantation, there is a competition between the integration of the material into the surrounding tissue and adhesion of bacteria to the implant surface. Therefore, implantation will be successful only if tissue integration occurs prior to considerable bacterial adhesion, thus preventing colonization of the implant (Gristina 1987). Furthermore, certain bacterial species are able to attach to the implant surfaces and form a protective bio-film layer, which is extremely resistant to both the immune system and antibiotics. These bio-films are considered the primary cause of implant-associated infection (Hetrick & Schoenfisch 2006). Local antibiotic release profiles should exhibit a high initial release rate in order to respond to the elevated risk of infection from bacteria introduced during the initial shock, followed by a sustained release at an effective level for inhibiting the occurrence of latent infection (Zilberman & Elsner 2008). In the case of orthopaedic-related devices, including TE scaffolds, it is important to combat bacteria possibly introduced during implantation and also those introduced systemically afterwards. Therefore, sustained drug release is necessary (Zilberman & Elsner 2008).

The most common antibiotic carrier to treat osteomyelitis after debridement surgery to remove necrotic bone tissue is poly(methylmethacrylate) (PMMA) beads (Vallo *et al.* 2004; Habraken *et al.* 2007, 2008; Shi *et al.* 2009). However, PMMA is not a resorbable polymer and must be removed in a second surgical procedure (Vallet-Regí *et al.* 2007; Shi *et al.* 2009). In this context, it should be pointed out that most previous work on loading biomaterials with antibiotics for orthopaedic applications has been carried out on bone-filler materials and bone cements (Takechi *et al.* 1998; Armstrong *et al.* 2002; Diez-Peña *et al.* 2002; Gbureck *et al.* 2002; Joseph *et al.* 2003; Hanssen 2004; Joosten *et al.* 2004; Webb *et al.* 2005; Schnieders *et al.* 2006; Krasko *et al.* 2007; Zilberman & Elsner 2008). In one of the first attempts to develop multifunctional re-absorbable implants, Queiroz *et al.* (2001) prepared sodium ampicillin which was adsorbed onto HA and glass-reinforced HA composites as a potential pharmaceutical formulation for periodontitis. Ampicillin-loaded methylpyrrolidinone chitosan microparticles

Table 3. Selected experimental trials carried out for three-dimensional bone scaffolds, both *in vivo* and *in vitro*, with a combination of controlled drug release.  $\beta$ -TCP,  $\beta$ -tricalcium phosphate; CP, calcium phosphate invert glasses; GS, gentamicin; HA, hydroxyapatite; HMS, mesoporous silica; PLA, poly(L-lactic acid); PLGA, poly(lactide-co-glycolide); PCL, poly- $\epsilon$ -caprolactone; PBS, phosphate-buffered saline; CPA, calcium phosphate-deficient apatite.

drug	composition	end-product shape	methodology	experimental trial	references
gentamicin	$\beta$ -TCP/CP/chitosan	porous matrix	thermally induced phase-separation technique + immersion in drug-containing PBS solution	<i>in vitro</i>	Zhang & Zhang (2002)
gentamicin	bioactive glass	bioactive glass pieces	uniaxial and isostatic compression at room temperature	<i>in vivo</i>	Vallet-Regí <i>et al.</i> (2001)
gentamicin	bioactive glass	mesoporous bioactive glass/bioactive glass	polyurethane sponge technique + immersion in drug-containing PBS solution	<i>in vitro</i>	Zhu & Kaskel (2009)
gentamicin	HMS-HA/PLGA microspheres	porous matrix	tensioactive template to obtain HMS-HA + double-emulsion evaporation technique to obtain GS-loaded PLGA microspheres + sinterization at 70°C	<i>in vitro</i>	Shi <i>et al.</i> (2009)
tetracycline	bioactive glass/ $\beta$ -cyclodextrin	bioactive glass pieces	sol-gel	<i>in vivo</i>	Domingues <i>et al.</i> (2004)
vancomycin	$\beta$ -TCP/agarose	porous matrix	freeze-drying and heat desiccation at 37°C	<i>in vitro</i>	Cabañas <i>et al.</i> (2009)
polymyxin B	calcium phosphate	ceramic pieces	compaction	<i>in vitro</i>	Kimakhe <i>et al.</i> (1999)
gatifloxacin	$\beta$ -TCP/PCL	porous matrix	compaction + sintering + immersion in drug-loaded PCL slurry	<i>in vitro</i> and <i>in vivo</i>	Miyai <i>et al.</i> (2008)
ciprofloxacin	HA/ $\beta$ -TCP/PLA	porous matrix	compression	<i>in vitro</i> / <i>in vivo</i>	Castro <i>et al.</i> (2005)
silver	bioactive glass	bioactive glass pieces	melting and sintering at high temperature + introduction of drug by ion-exchange process	<i>in vitro</i>	Di Nunzio & Verné (2005)
dexamethasone	starch/PLA	porous matrix	supercritical phase-inversion technique	<i>in vitro</i>	Duarte <i>et al.</i> (2009a)
dexamethasone	chitosan	porous matrix	freeze-drying + drug impregnation by supercritical fluid technology	<i>in vitro</i>	Duarte <i>et al.</i> (2009b)
ibuprofen	bioactive glass/MCM-41	porous matrix	MCM-41 spheres + melting and sintering at high temperature to obtain bioactive glass scaffold + drug impregnation	<i>in vitro</i>	Mortera <i>et al.</i> (2007)
alendronate	silica (SBA-15)	mesoporous silica matrix	triblock copolymers technique + functionalization + immersion in drug-containing buffer aqueous solution	<i>in vitro</i>	Nieto <i>et al.</i> (2008)
zoledronate	CPA	pellets	suspension of CDA in drug-containing water solution	<i>in vitro</i>	Fauchex <i>et al.</i> (2009)

have been described for a similar purpose (Giunchedi *et al.* 1998). Moreover, calcium phosphate cement (CPC)-chitosan composites have been applied for drug release of the cephalosporin antibiotic flomoxef, which is active against methicillin-resistant *Streptococcus aureus*. For example, Takechi *et al.* (2002) added flomoxef sodium to the liquid phase of tetracalcium phosphate (TTCP) cement/chitosan composites and measured the drug release from preset discs for 3 days. Results showed

a release pattern that was characterized by an initial burst, followed by a more sustained release. The total per cent of drug released in 24 h was 24–35% and it was found that the addition of chitosan in different amounts did not influence the total drug release after 72 h. Moreover, the release from these chitosan-enriched cements did not differ significantly from that of normal TTCP cement, though the maximum amount of chitosan used was 1.0 per cent w/v. Analysis of the literature



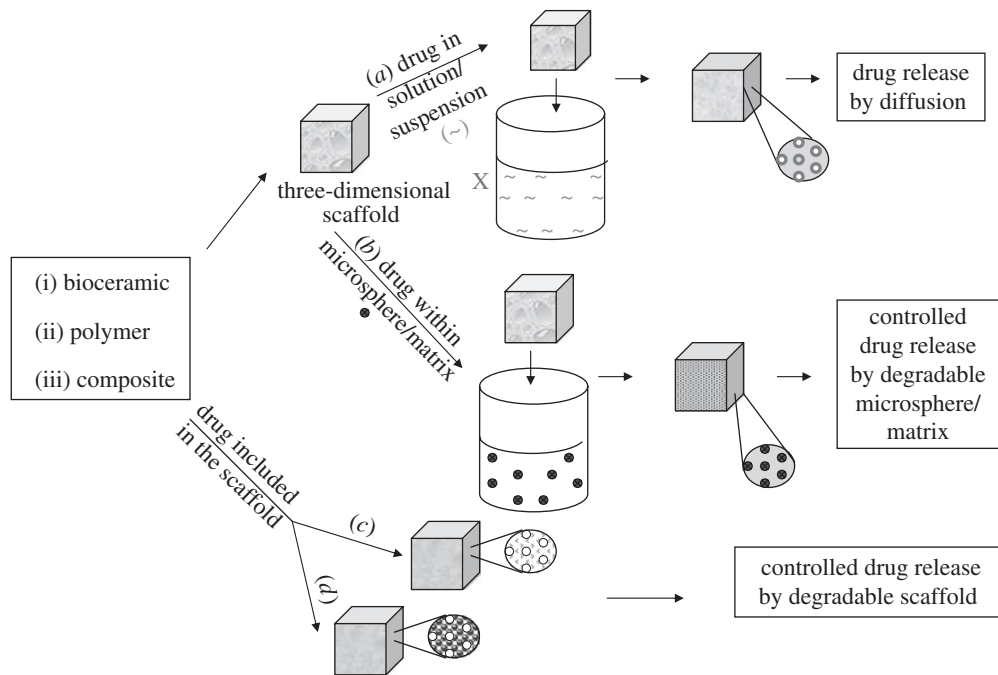


Figure 2. Schematic representation of the most common strategies to deliver drugs from three-dimensional scaffolds in bone TE. Drugs may be adsorbed onto the pore surface of the scaffolds in either their unprotected (*a*) or their protected (microsphere/matrix) (*b*) forms. Alternatively, drugs may be entrapped in the scaffold structure in either their unprotected (*c*) or their protected (microsphere/matrix) (*d*) forms.

reveals that new multifunctional composite materials, produced by different techniques, are being developed for specific application in bone TE and are detailed below, with a focus on specific antibiotics used in three-dimensional TE scaffolds.

### 3.2. Gentamicin

Gentamicin is a commonly employed antibiotic in trauma, widely used for the treatment of osteomyelitis because of its broad-spectrum characteristics (Li & Hu 2001). Most of the major bacteria causing chronic osteomyelitis are sensitive to gentamicin. This drug has been loaded in several scaffolds in order to evaluate their ability as controlled-release carriers (Li & Hu 2001). Zhang & Zhang (2002) prepared macroporous chitosan scaffolds reinforced by CaP particles such as  $\beta$ -TCP and CaP invert glasses using a thermally induced phase-separation technique. These porous composite materials were loaded with gentamicin sulphate (GS) by immersing them in drug-containing PBS solutions. *In vitro* tests showed that, in comparison with GS-loaded pure chitosan scaffolds, the initial high burst release of GS was decreased through incorporating CaP crystals and glass particles into the scaffolds, and a sustained release for more than three weeks was achieved (Zhang & Zhang 2002). The highest sustained release was observed from the particle-containing composite, which was suggested to occur owing to a higher extent of chitosan cross-linking. Scanning electron microscopy micrographs showed no apparent morphological differences for osteoblastic cells grown on the pure chitosan scaffolds and those grown on composite scaffolds. The cells attached and migrated on these scaffolds, suggesting a good cellular compatibility (Zhang & Zhang 2002; Habraken *et al.* 2007).

Vallet-Regí and co-workers used uniaxial and isostatic pressure at room temperature to obtain GS-loaded bioceramic tablets (Vallet-Regí *et al.* 2001). These tablets were tested *in vivo* in New Zealand rabbit femurs for one, four, eight and 12 weeks to study their biological response. The bone response to the implant was of perfect osseo-integration. The local GS levels detected in bone tissue were above the minimal inhibitory concentration (MIC) and they were effective because they were toxic for the majority of the resident micro-organisms. In addition, there was progressive decrease of GS levels in bone tissue with time, but the levels were always above the MIC until the end of the assay. In related studies, Zhu & Kaskel (2009) compared *in vitro* the local drug-release behaviour of two bioactive silicate glass scaffolds: three-dimensional mesoporous ( $\text{SiO}_2\text{-CaO-P}_2\text{O}_5$ ) (MBG) and three-dimensional BG scaffolds. Both scaffolds were prepared by using the polyurethane sponge technique, which involves the burning-out of a sacrificial sponge to create the pore structure (figure 1*a*) (Chen, Q. Z. *et al.* 2006); but in case of MBG scaffold a surfactant was added. Afterwards, the scaffolds were immersed in a gentamicin solution for loading the drug. The results indicated that the mesoporous structure played an important role in the drug-loading capability and its release rate. The drug-uptake capacity of the MBG scaffold was over twofold higher than that of the BG scaffold. During the whole release period in SBF, gentamicin was released from the MBG scaffold at a much lower release rate than from the BG scaffold. A novel design of a TE scaffold with controlled drug-delivery capability has been developed by Shi *et al.* (2009). The scaffold is based on mesoporous silica-HA (HMS-HA) composite particles used as fillers in PLGA microspheres. HMS-HA particles were produced using dodecylamine as a

template and GS-loaded PLGA microspheres were prepared using a double emulsion solvent evaporation technique (water/oil/water). PLGA/HMS–HA–GS composite microspheres were prepared using a single emulsion solvent evaporation method. Afterwards, PLGA or PLGA/HMS–HA–GS microsphere sintered scaffolds were fabricated by pouring PLGA or PLGA/HMS–HA–GS microspheres into cylindrical moulds, and subsequently sintering at 70°C for 2 h. The results showed that the presence of HA in PLGA/HMS–HA scaffolds could balance the decreased pH values caused by the acidic degradation product of PLGA. Moreover, HMS–HA improved the cytocompatibility and bioactivity of PLGA. It was also claimed that the compressive strength and elastic modulus of PLGA/HMS–HA scaffolds were higher than those of pure PLGA scaffolds, showing similar mechanical properties to human cancellous bone (Shi *et al.* 2009). *In vitro* drug-delivery testing in SBF of the PLGA/HMS–HA scaffolds showed that PLGA reduced the GS release from HMS–HA particles, and the release lasted for nearly one month (Shi *et al.* 2009).

### 3.3. Other antibiotics

A series of antibiotics has been further considered in combination with scaffolds for bone TE. Tetracycline was incorporated in a hybrid coating consisting of polycaprolactone (PCL) and HA powder, which was applied on the surface of a HA porous bone scaffold through dip-coating and solvent-casting method (Kim *et al.* 2004a). *In vitro* drug-delivery testing in PBS revealed that the release amount was controlled via the coating cycle and initial drug loading and a sustained release preceded by a burst release during the first 2 h was achieved (Kim *et al.* 2004a). Domingues *et al.* (2004) introduced tetracycline and an inclusion complex formed by tetracycline and  $\beta$ -cyclodextrin, at 1:1 molar ratio tetracycline/ $\beta$ -cyclodextrin, into a BG prepared by the sol–gel technique. The inclusion complex tetracycline/ $\beta$ -cyclodextrin was prepared by freeze-drying and either tetracycline hydrochloride or tetracycline in  $\beta$ -cyclodextrin inclusion compound was loaded into sol–gel solutions to prepare drug-loaded silicate BGs. An initial burst of 12 per cent was observed *in vitro* (in SBF), followed by a sustained release over 80 days achieving a total release of 22–25%. The *in vivo* test was carried out with three groups of female mice treated with BG without drugs, or associated with tetracycline (BT), or with tetracycline/ $\beta$ -cyclodextrin (BTC) by subcutaneous implantation. A considerable bacteriostatic activity was found with BT and BTC-loaded glasses, when compared with plain glass. The presence of cyclodextrin was important to slow down the release of tetracycline for a long period of time and it was verified that the presence of tetracycline or its inclusion complex, tetracycline/BTC, did not affect the bioactivity of the glass. Recently, Cabañas *et al.* (2009) loaded vancomycin (VAN), a drug susceptible to heat degrading, into  $\beta$ -TCP/agarose scaffolds by two different methods: freeze-drying and desiccation at 37°C. Poly(ethylene glycol) (PEG) was included in the formulation to tailor the release of VAN (Cabañas *et al.* 2009). The freeze-dried samples had a higher

porosity structure than samples dried at 37°C. The pieces obtained showed a microstructure similar to that of human cancellous bone. In addition, the different pore architectures and the formation of an agarose–PEG–VAN complex yielded different drug-release patterns (Cabañas *et al.* 2009). Polymyxin B (PMB) is a polypeptidic antibiotic that undergoes thermolysis above 60°C. With this limitation to be taken into account, Kimakhe *et al.* (1999) made one of the first attempts to obtain a multifunctional inorganic matrix for a bioactive drug-delivery system (DDS) in which the effect of a released therapeutic agent is favoured by the biocompatibility, osteoconductivity and bioresorption of the ceramic material. PMB was loaded on CaP powders using a dynamic compaction method at high velocities (25 and 50 m s<sup>-1</sup>) without external heating. The compaction procedure did not cause any loss in PMB integrity and biological activity. PMB release *in vitro* began after 2–3 days of incubation for blocks compacted at 25 m s<sup>-1</sup> velocity and on day 5 for those compacted at 50 m s<sup>-1</sup> (Kimakhe *et al.* 1999).

Some important results have shown the potential effectiveness of quinolones in local DDS designed to treat bone infection using carriers such as HA–anionic collagen composite (Martins & Goissis 2000), biodegradable polymers (Overbeck *et al.* 1995; Nicolau *et al.* 1998; Nie *et al.* 1998; Ramchandani & Robinson 1998; Kanellakopoulou & Giamarellos-Bourboulis 2000; Désévaux *et al.* 2002) and mixtures of calcium phosphates and biodegradable polymers (Castro *et al.* 2003). A composite of gatifloxacin (GFLX)-loaded poly-x-caprolactone combined with x-TCP porous ceramic was obtained by compression moulding, followed by sintering (Miyai *et al.* 2008). This process had the advantage of being a solvent-free process. GFLX mostly retained its bactericidal property after processing. *In vitro* testing in Hanks' balanced solution showed that the composite of GFLX-loaded PCL/ $\beta$ TCP ceramic released GFLX for four weeks and had sustained bactericidal activity against *S. milleri* and *Bacteroides fragilis* for at least one week (Miyai *et al.* 2008). *In vivo* tests in rabbits with osteomyelitis lesions induced by *S. milleri* and *B. fragilis* in the rabbit mandible showed that the composite of GFLX-loaded PCL/xTCP was effective in controlling infection at the bone defect formed by debridement. Moreover, after four weeks, new bone formation was observed on the surface of the composite. Further, after 50 weeks, ingrowing bone tissue with vascular vessels was observed along the PCL and  $\beta$ -TCP interface. Additionally, GFLX concentrations in the serum and soft tissues were very low suggesting a low risk of systemic toxicity (Miyai *et al.* 2008). Ciprofloxacin (CFX) is the most widely used fluoroquinolone for bacterial bone infection with a low MIC (0.25–2  $\mu$ g ml<sup>-1</sup>). CFX has been considered in combination with bioceramics and polymers. For example CFX was loaded into a composite comprising HA, TCP and poly(DL-lactide) (PLA) to treat complicated multi-organism bone infections that required high antibiotic concentrations (40%) in bone for long periods (Castro *et al.* 2005). The composites were prepared by mixing the ingredients and by further uniaxial compression using a hydraulic press. *In vitro*

results showed that the release rate decreased proportionally to the PLA/CaP ratio, drug-loading and compaction pressure as a more tightly packed matrix was obtained. *In vivo* tests in rabbits showed that CFX concentrations along the bond remained higher than the MIC against the most common pathogens causing osteomyelitis. A good *in vivo/in vitro* correlation was found; however, the release of the last 20 per cent CFX remaining within the matrix was faster *in vivo* than *in vitro*, as the release was enhanced *in vivo* by neovascular vessels formed inside the matrix (Castro *et al.* 2005).

### 3.4. Inorganic ions

Silver, in its oxidized form ( $\text{Ag}^+$ ), is an antibacterial agent and it has been proposed as an additive in bone TE scaffolds (Di Nunzio *et al.* 2004). Although the antibacterial activity of silver ions has been demonstrated in several works (Matsuura *et al.* 1997; Gatter *et al.* 1998; Kim *et al.* 1998; Adams *et al.* 1999; Kawashita *et al.* 2000; Bellantone *et al.* 2002; Blaker *et al.* 2004; Di Nunzio *et al.* 2004; Verné *et al.* 2005), the silver ion antimicrobial mechanism is not fully understood. It is well known that silver ions can interact with bacterial cells in different ways: they can bind to microbial DNA preventing bacterial replication or to sulphhydryl groups of bacterial enzymes, inhibiting cell respiration and binding transport of important substances across the cell membrane and within the cells (Chen, W. *et al.* 2006). These different ways of interaction are the origin of low bacterial resistance to silver (Chen, W. *et al.* 2006; Hetrick & Schoenfish 2006). The bactericidal activity of  $\text{Ag}^+$  is effective against a broad range of bacteria potentially found at the sites of scaffolds or implants including *S. aureus* (Hetrick & Schoenfish 2006).

The incorporation of silver into bone TE scaffolds is a novel approach. In one of the earliest studies,  $\text{Ag}^+$  ions were introduced into three-dimensional bioactive glass–ceramic scaffold surfaces through a patented ion-exchange process (Di Nunzio & Verné 2005). The control of  $\text{Ag}^+$  content on the surface, as well as its diffusion profile throughout the ion-exchanged layer, was achieved by a careful choice of the ion-exchange parameters (temperature, time and silver concentration in the molten bath). According to the authors of the original investigation (Di Nunzio & Verné 2005), this technique allows, by tuning the process parameters, a controlled silver ion incorporation into the superficial layers of the scaffold, maintaining the scaffold structure and its characteristics unchanged (Di Nunzio *et al.* 2004; Vitale-Brovarone *et al.* 2008). Moreover, the possibility of using  $\text{Ag}^+$ -doped BG as coating of polymeric fibres (sutures) for fabricating textile scaffolds was investigated by Blaker *et al.* (2004). The authors indicated that the Ag-loaded BG coating of textile structures can represent a convenient alternative to fabricate multifunctional biodegradable scaffolds (Blaker *et al.* 2004). In a related work, Pratten *et al.* (2004) investigated *in vitro* the ability of a silver-doped BG (AgBG) coating, which was elaborated using a slurry-dipping process, to prevent *Staphylococcus epidermidis* colonization on surgical sutures. The *in vitro* studies were carried out under both batch and flow conditions to quantify the number of

viable cells adhered to the surface and to determine the attachment and detachment over time, respectively. The authors showed that AgBG coating had a significant effect on preventing *S. epidermidis* attachment when compared with coatings of standard 45S5 Bioglass.

## 4. OTHER USED DRUGS

### 4.1. Anti-inflammatory drugs

Implantation of engineered biomaterials might cause local inflammation owing to the host immunoresponse (Corry & Moran 1998; Mendez *et al.* 2004; Gonzalez-Corchon *et al.* 2006; Chevalier *et al.* 2009), which therefore requires the use of anti-inflammatory agents, either steroids (glucocorticoids) or non-steroids. In particular, glucocorticoids have been shown to have strong inhibitory effects on cytokine-related inflammation by downregulating transcription of interleukin (IL)-1, tumour necrosis factor (TNF)- $\alpha$ , granulocyte macrophage–colony-stimulating factor (GM-CSF), IL-3, IL-4, IL-5, IL-6 and IL-8 (Norton *et al.* 2005; Novak *et al.* 2009). In particular, dexamethasone (DEX) is also used in TE for the preparation of osteogenic medium (Duarte *et al.* 2009a,b). In this context, Duarte *et al.* (2009b) impregnated DEX using supercritical fluid technology in chitosan porous scaffolds prepared by a freeze-drying process. The highest loading was obtained at low pressures and temperatures (8.0 MPa and 35°C), corresponding to a lower solubility of DEX in the supercritical fluid (carbon dioxide) and a low swelling of the polymeric matrix. Results from *in vitro* drug-release studies showed that the release of DEX from chitosan scaffolds exhibited a sustained profile, which is suitable for application in TE. Supercritical fluid technology can be adapted to prepare porous structures using a variety of polymeric systems or composites and other bioactive compounds in order to develop multifunctional scaffolds with improved mechanical properties and biocompatibility (Quirk *et al.* 2004). In this context, Duarte *et al.* (2009a) reported the feasibility of using supercritical fluid methods to process in one step a porous matrix loaded with a pharmaceutical agent for TE purposes. In order to investigate the scaffold's performance, the release of DEX loaded in a three-dimensional starch-based porous matrix was tested. A supercritical phase-inversion technique, using carbon dioxide as the supercritical fluid, was employed to prepare the composite scaffolds of DEX and a polymeric blend of starch and poly(L-lactic acid). *In vitro* drug-release studies were carried out in PBS and results showed that a sustained release of DEX was achieved over 21 days. The fitting of a power law to the experimental data demonstrated that drug release is governed by an anomalous transport, i.e. it was proposed that both the drug diffusion and the swelling of the matrix influence the release of DEX out of the scaffold (Duarte *et al.* 2009a).

Within non-steroids, ibuprofen (IBU) has been largely used orally, intravenously and even topically. Recently, Mortera *et al.* (2007) used silica-ordered mesophase (MCM-41) submicron spheres incorporated inside glass–ceramic ( $\text{SiO}_2\text{--CaO--K}_2\text{O}$ ) bioactive scaffolds for the controlled delivery of IBU in TE. The

scaffolds were fabricated by dipping the glass–ceramic scaffold into the MCM-41 synthesis solution and after that samples were dipped into an IBU solution for drug adsorption over 3 days. The *in vitro* drug-release study performed in SFB showed a fast release of IBU during the first 8 h followed by a slower release between 8 and 120 h (Mortera *et al.* 2007).

#### 4.2. Bisphosphonates

Bisphosphonates (BPs) have been in widespread use since the 1970s for the treatment of a variety of bone diseases characterized by osteoclast-mediated bone resorption such as Paget's disease, tumour-induced hypercalcaemia, metastatic bone diseases and osteoporosis (Siris *et al.* 1996; Colucci *et al.* 1998; Major *et al.* 2000; Rodan & Martin 2000; Van Beek *et al.* 2003; Miller 2005; Nieto *et al.* 2008; Panzavolta *et al.* 2009). BPs act by inhibiting the osteoclastic resorption of bone tissue (Colucci *et al.* 1998; Van Beek *et al.* 2003), they bind strongly to HA crystals and are retained for a long time in bone, being excreted unmetabolized in urine (Papapoulos 2006; Panzavolta *et al.* 2009). Chemically, BPs are analogous to pyrophosphates, which are natural modulators of bone metabolism (Nancollas *et al.* 2006; Nieto *et al.* 2008). However, in BPs, the oxygen atom that binds the two phosphate groups of pyrophosphate (P–O–P) is substituted by a carbon atom (P–C–P) making the molecule less resistant to hydrolysis than pyrophosphates (Nancollas *et al.* 2006; Papapoulos 2006; Nieto *et al.* 2008). Recently, the positive effect of adjunct treatment with injectable antiresorptive zoledronic acid on the biological process induced by BG incorporation in bone defects has been proved (Välimäki *et al.* 2006). A member of the family of BPs, sodium alendronate (SA) was used by Nieto and collaborators (Nieto *et al.* 2008) as a drug model to test how the organic modification of the surface of silica-based ordered mesoporous scaffolds can control the drug dosage. The mesoporous silica structure (SBA-16) was synthesized using triblock copolymers followed by functionalization by acid catalysis. Finally, SA was adsorbed by immersing the specimens in a drug-containing buffered aqueous solution. The adsorption rates of SA, and consequently the SA release rates, were tuned by using a range of amine-functionalization degrees (Nieto *et al.* 2008). Recently, Fauchex *et al.* (2009) used calcium-deficient apatite as a carrier for delivery of zoledronate acid (ZA) in bone tissue. According to the *in vitro* tests carried out, the released ZA inhibited osteoclastic resorption without affecting osteoblasts. Future studies are planned to develop three-dimensional TE scaffolds loaded with BPs and the use of BPs in TE approaches is bound to increase as novel methodologies are developed for the efficient loading of BPs in porous three-dimensional matrices.

### 5. EVALUATING THE EFFICACY OF BIOACTIVE BONE SCAFFOLDS AS DRUG-DELIVERY SYSTEMS

It is important to highlight that even though several *in vitro* tests have been performed in order to investigate the suitability of bone scaffolds as drug carriers, there is

no *in vitro* standard technique available for this purpose yet. Indeed, several methods described in the literature try to simulate *in vivo* conditions and to avoid non-physiological turbulence. In most cases, loaded scaffolds are soaked in a beaker, a bottle or a sampling tube and/or in a culture chamber with or without controlled agitation (Guicheux *et al.* 1997; Paul & Sharma 1999; Krajewski *et al.* 2000; Perry *et al.* 2002; Hasegawa *et al.* 2004; Murugan & Ramakrishna 2004; Palazzo *et al.* 2005; Medvecky *et al.* 2007; Melville *et al.* 2008; Victor & Kumar 2008; Chevalier *et al.* 2009). Released tests are usually performed considering only low volumes (from 3 up to 150 ml) of various dissolution media (phosphate buffer, saline physiological solution, SBF, etc.). For bone TE, SBF is frequently the dissolution medium of choice as it allows the study of how the release of a drug can be affected by the formation of a surface layer of microcrystalline HA on the bioactive scaffold, which is the expected behaviour *in vivo*. There are several sampling methods reported for studying the release of drug substances from biomaterials (replacing the entire volume with fresh media after taking a sample, replacing the same volume withdrawn after taking a sample); sometimes ensuring sink conditions (ensuring that sufficient medium volume is available during the test to dissolve the total drug quantity from the loaded scaffold) and sometimes using continuous flow through chambers (flow rate from 0.02 to 2.5 ml min<sup>-1</sup>) (Perry *et al.* 2002; Hall *et al.* 2004; Chevalier *et al.* 2009). Moreover, it could be beneficial to mimic the blood/plasma flow speed observed at the site of the scaffold's implantation (Baroli 2009). Further, and surprisingly, there is very limited reported information regarding the stability of the drugs involved in the studies, in particular the thermal stability. If a drug is included within a scaffold in order to be released over a month or longer period, the normal stability tests usually carried out by suppliers will not be sufficient; further tests should be done in order to evaluate the activity and drug degradation rate at body temperature for a prolonged period of time. In addition, when *in vivo* tests were performed, little work reported on the drug clearance from the zone and how the new vascularization induced by the presence of the scaffold would affect the residence time of the drug, especially when the matrix is expected to sustain the release of the drug for a prolonged period of time (Castro *et al.* 2003). In this regard, also very little work has been carried out on the correlation of *in vitro* and *in vivo* results of drug release to evaluate the significance of establishing novel *in vitro* models, which would represent realistic studies, prior to *in vivo* tests on host response after scaffold implantation (Malafaya *et al.* 2002).

### 6. CONCLUSIONS AND PERSPECTIVES

In reviewing the published studies on bioactive bone TE scaffolds with additional drug-delivery capability, it becomes clear that there have been continued advances, particularly in recent years, towards the further development of the field. New materials and combinations of materials, as well as improved scaffold designs based on novel processing techniques, are being continuously proposed to advance the drug-delivery

capability of bone tissue scaffolds. However, a large amount of biological information is still needed to fully understand the *in vitro* and *in vivo* performance of such scaffolds in specific applications, which once obtained will provide a rational scaffold design and optimization of their drug-delivery function. For example, analysis of the literature has revealed that there is a lack of biological data regarding the specific concentration in which a given drug is needed in relation to the particular local microenvironment and how this is affected by the interaction between the scaffold and new tissue, considering also the effect on vascularization. In addition, during the design and development of multifunctional scaffolds and in order to control the drug-release patterns, the effect of processing parameters on scaffold microstructure (i.e. porosity) and on the resulting drug-release profiles, stability, biodegradation behaviour, as well as the scaffold's mechanical and physical properties, must be further investigated. Despite significant efforts in this direction, several challenges have yet to be resolved. These include understanding the link between the key variables determining the scaffold-processing conditions and the physico-chemical properties of the novel three-dimensional delivery systems with special consideration given to the stability of the incorporated drug. These challenges also include achieving accurate control over time and space of specific quantities of drug for determined applications and engineering the release patterns, which are related to the interaction between the drug and the scaffold (matrix degradation and erosion and drug diffusion through the matrix), drug solubility and the amount of drug loaded available to be delivered. Further, only limited work has been reported on how processing parameters and sterilization can affect the homogeneous loading, the stability and the release kinetics of the drug incorporated. In addition, there is a lack of focus in several works in terms of the pathology they would like to address. It is clear that the combination of drug delivery and TE using advanced three-dimensional scaffold concepts is in its early years and it will take time to achieve clinical results of relevance. It is therefore expected that this field will keep growing within the next few years and related research outputs could lead to results with the potential to transform the clinical approach to bone-related pathologies. Specifically, in order to achieve these goals, it would be important to intensify the working collaboration of researchers and technologists from different relevant communities. The interdisciplinary character of TE, allowing the confluence of different scientific fields, backgrounds and knowledge, is mandatory for the advance of bone TE therapeutics.

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