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Potential of 3-D tissue constructs engineered from bovine chondrocytes / silk fibroin-chitosan for *in vitro* cartilage tissue engineering

Nandana Bhardwaj^{#,a}, Quynhhoa T Nguyen^{#,b}, Albert C Chen^b, David L. Kaplan^c, Robert L Sah^{*,b}, and Subhas C Kundu^{*,a}

^aDepartment of Biotechnology, Indian Institute of Technology, Kharagpur-721302, India

^bDepartment of Bioengineering, University of California-San Diego, La Jolla, CA 92093, USA

^cDepartment of Biomedical Engineering, Tufts University, Medford 02155, USA

Abstract

The use of cell-scaffold constructs is a promising tissue engineering approach to repair cartilage defects and to study cartilaginous tissue formation. In this study, silk fibroin/chitosan blended scaffolds were fabricated and studied for cartilage tissue engineering. Silk fibroin served as a substrate for cell adhesion and proliferation while chitosan has a structure similar to that of glycosaminoglycans, and shows promise for cartilage repair. We compared the formation of cartilaginous tissue in silk fibroin/chitosan blended scaffolds seeded with bovine chondrocytes and cultured in vitro for 2 weeks. The constructs were analyzed for cell viability, histology, extracellular matrix components glycosaminoglycan and collagen types I and II, and biomechanical properties. Silk fibroin/chitosan scaffolds supported cell attachment and growth, and chondrogenic phenotype as indicated by Alcian Blue histochemistry and relative expression of type II versus type I collagen. Glycosaminoglycan and collagen accumulated in all the scaffolds and was highest in the silk fibroin/chitosan (1:1) blended scaffolds. Static and dynamic stiffness at high frequencies was higher in cell-seeded constructs than non-seeded controls. The results suggest that silk/chitosan scaffolds may be a useful alternative to synthetic cell scaffolds for cartilage tissue engineering.

Keywords

Silk fibroin; Chitosan; Scaffolds; Chondrocytes; Extracellular matrix; Cartilage; Tissue engineering

1. Introduction

Damaged adult articular cartilage exhibits a limited propensity for self-repair. Normal cartilage is composed of chondrocytes in a hydrated extracellular matrix (ECM). The

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^{*}Corresponding authors, **Professor S. C. Kundu**, Tel: +91-3222-283764, Fax: +91-3222-278433, kundu@hijli.iitkgp.ernet.in, **Professor Robert L. Sah**, Tel: (858) 534-0821, Fax: (858) 822-1614, rsah@ucsd.edu. [#]Equal contribution

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chondrocytes synthesize sulfated glycosaminoglycan (GAG) and type II collagen, which provide the tissue with load-bearing function [1, 2]. Current methods of cartilage repair, including microfracture, osteochondral grafts, and prosthetic joint replacement do not provide for long-lasting and complete recovery [3–7]. Thus, engineered tissue implants are attractive and offer a promising approach to cartilage restoration.

Formation of cartilaginous constructs *in vitro* involves manipulation of four parameters: scaffold, cells, soluble factors, and the physical environment [8]. A variety of biomaterials, both natural and synthetic, have been analyzed to form scaffolds and tested for cartilage tissue engineering (Table 1). Physiologic biomaterials include fibrin, hyaluronic acid, and various forms of collagen. Natural materials include alginate, chitosan, and silk fibroin. Synthetic materials include poly (glycolic acid) (PGA) and poly (lactic acid) (PLA). A number of natural and synthetic materials undergo relatively rapid degradation, during which their size, shape, and function changes [9]. The effects of this degradation on the formed construct may include physical and chemical, such as due to formation of acid by-products [10]. In contrast, more stable scaffold or hydrogel materials, such as agarose [11], allow for analysis of the contribution of cells and matrix deposited in the material.

Silk fibroin (SF) is an attractive natural fibrous protein for biomedical applications and studies due to a number of biological, chemical, and physical properties. SF facilitates cell adhesion and growth, and has relatively low thrombogenicity, low inflammatory response, and low protease susceptibility when highly crystallized [12–15]. Furthermore, silk can be subjected to aqueous or organic solvent processing and can be chemically modified to address a wide range of applications. SF scaffolds can provide high permeability to oxygen and water as well as robust mechanical properties. Recent cartilage tissue engineering studies performed using silk scaffolds resulted in adhesion and proliferation of chondrocytes and mesenchymal stem cells, and production of cartilaginous matrix *in vitro* [16–17]. Thus, SF is attractive for studies of cartilage tissue engineering, and because of its slow degradation, SF may be blended with other materials to form suitable scaffolds.

Chitosan (CS) is a biomaterial that mimics the glycosaminoglycan (GAG) components of cartilage. CS is a partially deacetylated derivative of chitin found in arthropod exoskeletons. It consists primarily of repeating units of β (1–4) linked glucosamine and N-acetyl glucosamine. It is formed through the N-deacetylation of chitin and structurally similar to GAGs. Chitosan supports chondrogenic activities [18–25] and is being evaluated in cartilage tissue engineering applications. Chitosan has also roles in wound healing, is non-toxic, and generates a minimal foreign body response with accelerated angiogenesis [26]. The properties of porous chitosan matrices such as microstructure, crystallinity, and mechanical strength can be varied by altering chitosan concentration, freezing rate, the molecular weight and percent deacetylation [9, 27–29]. Despite the growing interest for chitosan as a biomaterial for tissue engineering, most studies on pure chitosan scaffolds have focused on sponges [8, 30, 31, 32–36] or hydrogels [20, 37]. Porous scaffolds allow seeding of cells with desirable and tunable characteristics such as biocompatibility, mechanical properties and biodegradability [12–15, 38, 39].

Silk-chitosan blend hybrid material may have beneficial properties, as shown for the culture of HepG2 hepatocyte and fibroblast cells [40, 41, 42]. Although silk fibroin and chitosan have been studied separately for *in vitro* chondrogenesis [31, 17, 43, 44], the influence of silk fibroin/chitosan composite scaffolds on chondrocyte morphology, differentiation, and function has not been studied yet and no study of this type has been performed earlier on chondrocytes. Earlier, we fabricated and characterized the polyelectrolyte complex porous scaffolds of silk fibroin/chitosan and investigated their suitability for tissue engineering applications [42]. Silk fibroin and silk fibroin/chitosan blended scaffolds of different ratios

(1:1 and 2:1) appeared promising based on cell viability and attachment. Thus, these scaffolds are used in the present study to evaluate the silk fibroin/chitosan blended scaffolds as matrices using bovine chondrocytes to analyze the cellular activity, viability, biochemical and biomechanical properties for cartilage tissue engineering.

2. Materials and methods

2.1. Materials

For scaffolds, CS derived from crab shells with a deacetylation degree of >85% was purchased from Sigma Aldrich (St. Louis, MO USA), and silk cocoons were kindly provided by Debra silkworm farm (West Bengal, India). For chondrocyte isolation and culture, biochemical, and immunochemical analyses, reagents were obtained as described previously [45, 46].

2.2. Experimental Design

The study design is summarized in Fig. 1. Porous scaffolds of (1) SF alone, and SF blended with CS at two ratios (2) SF/CS (1:1), and (3) SF/CS (2:1) were compared for their ability to support the formation of cartilaginous tissues. Scaffolds were either analyzed after preparation (a) directly (without seeded cells) or (b) with seeded chondrocytes and two weeks of incubation. Some samples were analyzed for cartilaginous matrix components, sulfated glycosaminoglycan (GAG) and collagen (COL) while others were analyzed for viable vs. non-viable cells by staining with fluorescent indicators, for the location of GAG by histochemical staining, for the location of types I and II collagen by immunohistochemistry, and for compressive load-bearing properties by static and dynamic compression testing.

2.3. Preparation and characterization of scaffolds

Porous scaffolds of silk fibroin (SF) alone and with blends of chitosan (CS) were fabricated as described previously [42, 53]. Briefly, CS was dissolved in 2% acetic acid and clarified by centrifugation. The final concentration of chitosan was 2%. SF solution was prepared following the method of Sofia et al., [53] with slight modification. Briefly, silk cocoons were cut into pieces, degummed (to remove sericin) with a boiling 0.02 M Na₂CO₃ solution for 30 min, and the fibers were washed with elix (deionized) water and then kept at 37°C overnight to dry. Purified fibers were dissolved in 9.3 M LiBr and then dialyzed against water using a 12 kDa molecular weight cutoff cellulose dialysis membrane. Dialysis was carried out to remove LiBr from the silk fibroin solution. The final concentration of silk fibroin solution used was 2% and was determined gravimetrically by drying the solution. SF, SF/CS (1:1) blend, and SF/CS (2:1) blends were prepared and used to fabricate scaffolds by freezing the solution at -20° C and then lyophilizing for 36 hr. Scaffolds were then treated with a gradation of ethanol (100% ethanol for 1 hr, 70% for 30 min, and 50% ethanol for 30 min) to neutralize and sterilize the scaffolds [42, 54]. Some scaffolds were characterized for pore structure by sputter-coating with gold, viewing by scanning electron microscopy (SEM) with a JEOL-JSM 5800 SEM, and recording images. Pore size was then determined by measuring pore diameter in >30 pores using Image J software (Wayne Rasband, National Institute of Health, USA). Before use in tissue engineering experiments, scaffolds were cut to ~2 mm thickness, punched to 6.4 mm diameter, and rinsed with culture medium as described below.

2.4. Isolation of bovine chondrocytes

Chondrocytes were isolated from the knees of immature bovine calves by sequential enzymatic digestion [44, 55]. Two preparations of cells were generated, each preparation

from both knee joints of an individual animal. Briefly, joints were exposed under aseptic conditions, and the cartilage from the superficial ~2 mm was collected, minced into fragments, digested with 0.2% protease type XIV to remove glycosaminoglycans, and then 0.02% collagenase to release chondrocytes from remaining matrix. The cell solution was clarified by filtering with 70 μ m and then 40 μ m filters. Cell viability was confirmed and cell counts were determined using Trypan Blue and a hemacytometer. Cells were then resuspended in medium (Dulbecco's modified Eagle medium, 10 mM HEPES, 0.1 mM non-essential amino acids, 0.4 mM L-proline, 2 mM L-glutamine, 100 U/ml penicillin, 100 μ g/ml streptomycin, and 0.25 μ g/ml amphotericin B) with 10% FBS and then plated at high-density (250,000 cells/cm²) and maintained overnight to allow recovery from isolation. All cultures were maintained in a humidified incubator (5% CO₂, 95% air) at 37°C.

2.5. Scaffold seeding and construct culture

The three types of scaffold were seeded with chondrocytes or used as non-seeded controls, and incubated for two weeks. For constructs containing cells, chondrocytes were seeded to achieve a density of 50 million/ml. To do so, scaffolds were placed in 6.5 mm diameter transwells and pre-wet with media. Chondrocytes were released with trypsin, and resuspended at 2–3.2 million cells/ml in medium including 25 μ g/ml L-ascorbic acid and 10% FBS. Then, 3.2 million cells (in 10–15 μ l seeding volume) were seeded onto each scaffold. Media (50 μ l) was added to the seeded scaffolds every 30 minutes, and after 3–4 hours, cell-seeded scaffolds were transferred to 6 well non tissue culture treated plates with fresh media. Constructs were incubated for 2 weeks with media changes (5 ml) every other day. Spent medium was collected for biochemical analysis.

2.6. Biochemical analysis of construct composition

Some constructs were analyzed for sulfated glycosaminoglycan (GAG) and collagen content. Constructs were weighed wet at the end of culture and then solubilized by incubation with proteinase K (0.5 mg/ml in 0.1 M sodium phosphate, 10 mM Na₂EDTA, pH 6.5), using a volume ~25 times that of the scaffolds, at 60°C overnight. Solubilized tissue was analyzed for GAG using dimethylmethylene blue [56] and shark chondroitin sulfate as a standard, as well as for collagen with hydroxyproline as an index [57] using a conversion factor of 7.25 μ g collagen/ μ g hydroxyproline [58]. Scaffolds that were not seeded with chondrocytes did not have measurable GAG or COL, and also did not interfere with GAG or COL assays (data not shown).

2.7. Cell viability within constructs

The viability of chondrocytes within some constructs was assessed using the Live-Dead assay. Briefly, scaffolds were rinsed with PBS, incubated in 4 mM calcein AM (staining live cells) and 2 mM ethidium homodimer (staining dead cells) in DMEM for 15–20 minutes at 37°C, washed with PBS and then visualized with a fluorescence microscope (Nikon Eclipse TE 300, AG Heinze, Irvine, CA). Digital images were captured using SPOT RT (Diagnostic Instruments, Sterling Heights, MI). Viable cells are indicated by the presence of intracellular esterase activity which converts calcein AM to calcein which fluoresces green. Dead cells are simultaneously recognized by ethidium homodimer 1 the entering into cells through damaged membrane, binding with deoxyribonucleic acid, and producing red fluorescence.

2.8. Histology

To localize sulfated GAG within constructs, histochemical analysis was performed using Alcian Blue. Some constructs were fixed with 4% paraformaldehyde in PBS, embedded in optimal cutting temperature (OCT), and snap frozen. Constructs were then cryo-sectioned at 20 µm and stained with 0.1% Alcian Blue in 0.4 M MgCl₂, 0.025 M sodium acetate, pH 5.6

To qualitatively localize and evaluate the extent of types I and II collagen deposition, some constructs were analyzed by immuhistochemistry using mouse monoclonal antibody against type I or type II [60]. Positive reactivity of staining for Col I and Col II was documented by photomicroscopy using brightfield illumination.

2.9. Biomechanical analysis of compressive properties

Some constructs were subjected to static and dynamic unconfined compression stress relaxation tests. Briefly, samples were taken out from media after 14 days of culture and tested for biomechanical properties while immersed in phosphate buffered saline (PBS). The samples were compressed 40% and allowed to relax to equilibrium (2400s or 0.002 MPa/ 180s). The samples were then subjected to a dynamic displacement of 5% amplitude relative to the compressed thickness, at frequencies of f = 0.005, 0.015, 0.05, 0.15, 0.5, and 1.5 Hz. From the static equilibrium data, the Young's modulus was determined. From the dynamic data, dynamic stiffness amplitude and phase were determined [61–63]. Correlation of biomechanical properties and biochemical properties was assessed by univariate regression.

2.10. Statistical analysis

All data are reported as mean \pm standard deviation (SD). One-way analysis of variance (ANOVA) was used to determine the effects of scaffold type on pore size, and biochemical composition, with the post-hoc Tukey test to compare between groups when significant variation was found. Thickness and biomechanical data were analyzed by repeated measures two-way ANOVA. p<0.05 was considered significant. All statistical analyses were performed with Systat version 10.2 (Systat, Richmond, CA).

3. Results

3.1. SEM of scaffold structure

All of the scaffolds exhibited porous structures as revealed by SEM (Fig. 2). The pore sizes of SF scaffolds was $80 \pm 15 \,\mu\text{m}$ (Fig. 2A), whereas that of SF/CS (1:1) (Fig. 2B) and SF/CS (2:1) (Fig. 2C) scaffolds were slightly larger, at $100 \pm 11 \,\mu\text{m}$ and $116 \pm 16 \,\mu\text{m}$, respectively.

3.2 Viability of cells in constructs

Live-dead assay demonstrated the cellularity and localization of live versus dead bovine chondrocytes in scaffolds after day 14 (Fig. 3A, 3B and 3C). A large number and high percentage of viable cells were evident and distributed relatively homogeneously within all three types of scaffold.

3.3 Localization of sulfated GAG and immunolocalization of Collagen I and II within tissue constructs

Histochemical analysis for sulfated glycosaminoglycan indicated differences in the matrix composition depending on the type of scaffolds (Fig. 3D, 3E, and 3F). SF/CS (1:1) scaffolds showed more evenly distributed staining than SF/CS (2:1) blended scaffolds and pure silk fibroin scaffolds. SF/CS (1:1) and silk constructs consisted primarily of individual or a few cells surrounded by matrix, interspersed between short fragments of scaffolds (Fig. 3D and 3F). Pure silk fibroin and SF/CS (2:1) constructs showed relatively less amounts of matrix as compared to SF/CS (1:1) (Fig. 3E).

Immunohistochemical staining indicated a general absence of staining for collagen type I (Fig. 3G, 3H and 3I) and positive staining for collagen type II (Fig. 3J, 3K and 3L). Nonseeded scaffolds showed less or minimum staining for sulphated GAG and collagen as shown in Fig. 4.

3.4 Biochemical content of tissue constructs

GAG deposition overall and within scaffolds varied with scaffold type (p<0.001). Sulfated GAG deposited into SF/CS (1:1) constructs were higher than those in SF/CS (2:1) constructs and SF constructs (each, p<0.05). Total GAG varied significantly with scaffold type (p<0.05), and was higher in SF/CS (1:1) constructs than SF constructs (by 63%) and SF/CS (2:1) constructs (by 49%) (Fig. 5A). In contrast, the content of sulfated GAG released in media did not vary between constructs (Fig. 5A, p=0.96). When normalized to scaffold wet weight, the GAG contents overall, in the medium, and remaining with the construct exhibited similar trends (p<0.05, Fig. 5B). For SF/CS (1:1), the overall GAG in the culture system over the two week period averaged 265 µg. For all groups, the GAG content in the constructs after the two week culture period averaged 5.0 mg GAG/gm wet weight of scaffolds.

Collagen within the construct (Fig. 5C), and also construct collagen normalized to wet weight (Fig. 5D), exhibited trends similar to that of overall GAG, varying significantly with scaffold type (p<0.05). Collagen content was higher in SF/CS (1:1) constructs than SF constructs (by 108%) and SF/CS (2:1) constructs (by 116%). Collagen normalized to construct wet weight showed similar trends (p<0.05). SF/CS (1:1) had a collagen content averaging ~37 μ g per construct and 0.9 mg collagen/gm wet weight.

3.5 Thickness of tissue constructs

After 14 days of culture the geometry and mechanical properties of scaffolds were determined. The thickness of the scaffolds increased with culture duration (p<0.05, Fig. 6), and with scaffold type (p<0.001), but without interaction (p=0.18). At the end the culture period, the construct thickness of SF, SF/CS (2:1), and SF/CS (1:1) were 2.3, 2.1, and 1.7 mm, respectively, corresponding to increases of 36%, 17%, and 21%.

3.6 Static and dynamic compressive properties of tissue construct

The equilibrium modulus, E, varied between the non-seeded scaffolds and the seeded constructs (p<0.05, Fig. 7), but not within scaffold type (p=0.15), and without an interactive effect of scaffold type and cell seeding (p=0.59). After 14 days of culture, the equilibrium modulus of SF, SF/CS (2:1), and SF/CS (1:1) constructs were 2.8, 2.3 and 1.9 kPa, respectively, corresponding to increases of 62%, 105% and 12%.

The dynamic unconfined compression stiffness (Fig. 8A) varied between the non-seeded scaffolds and seeded constructs (p<0.05), and with frequency (p<0.001) with an interaction effect (p<0.005). At a low frequency of 0.005 Hz, the non-seeded SF/CS (1:1) constructs were softest followed by SF/CS (2:1), and then SF, averaging 3.9 kPa, 6.8 kPa, and 8.6 kPa, respectively. After seeding and 14 days of culture, these same constructs increased in stiffness, averaging 4.9, 6.0, and 13.4 kPa, respectively. At a higher frequency (f=1.5 Hz), the non-seeded SF/CS (1:1) constructs were softest followed by SF/CS (2:1), and then SF, averaging 5.5 kPa, 9.3 kPa, and 11.6 kPa, respectively. After seeding and 14 days of culture, these same constructs increased in stiffness, averaging 12.2, 15.2, and 24.7 kPa, respectively.

The phase angle of dynamic stiffness also indicated distinct differences in dynamic behaviour (Fig. 8B), with differences between the non-seeded scaffolds and seeded

constructs (p<0.05) and with frequency (p<0.001) without an interaction effect (p=0.86). For both non-seeded scaffolds and 14 day constructs, similar trends of generally increasing phase shift with increasing test frequency from 0.005 Hz to 1.5 Hz were observed, ranging from 6–22 degrees for non-seeded scaffolds, and 10–35 degrees for 14 day constructs.

Certain mechanical and biochemical properties were correlated. The compressive modulus of SF constructs was positively correlated with collagen content ($r^2=0.60$, p<0.05). The compressive modulus of SF/CS (2:1) scaffolds was positively associated with GAG content ($r^2=0.11$, p<0.05) and collagen content ($r^2=0.68$, p<0.05). The compressive modulus of SF/CS (1:1) was positively correlated with GAG ($r^2=0.60$, p<0.05).

4. Discussion

Successful, cell-based tissue engineering of cartilage requires the maintenance of the chondrocytes in highly differentiated phenotype and their ability to produce cartilage-specific matrix. The use of glycosaminoglycans analogues, ECM of cartilage is gaining much popularity for cartilage repair. Earlier both silk fibroin and chitosan scaffolds have been used separately for cartilage tissue engineering but the influence of silk fibroin/ chitosan blended scaffolds have not yet been analyzed. In the present study, we evaluated the formation of cartilaginous tissue by bovine chondrocytes cultured in porous silk fibroin/ chitosan three-dimensional polyelectrolyte complex porous scaffolds. We employed silk fibroin/chitosan blended scaffolds of different ratios (2:1) and (1:1) and silk fibroin scaffolds characterized earlier in our previous study [42]. These blends and pure silk fibroin showed promising results and suitability in terms of porosity, pore size, degradation, cell viability and other properties. The results of this study demonstrate the ability of bovine chondrocytes to proliferate and deposit a mechanically functional matrix with geometry maintained after 2 weeks of culture.

The development of biomimetic scaffolds to enhance the differentiation of chondrocytes has largely focused on altering the chemical composition of scaffolds. Following the same rationale, chitosan has been combined with other biomaterials because of its similarity with the glycosaminoglycans normally present in native cartilage and the potential to promote chondrogenesis [18]. The resulting hybrid scaffolds enhanced cartilage tissue formation [64, 65]. Many factors such as number of attached cells and fluid flow to enhance nutrient availability and waste product removal has also impact on ECM production [66]. In the present study, matrix deposition was also modulated by the type of scaffold, being highest with macroporous silk/chitosan (1:1) compared to silk/chitosan (2:1) and pure silk scaffolds (Fig. 5). Immunohistochemical staining showed more col II (Fig. 3J, 3K and 3L) and less col I (Fig. 3G, 3H and 3I) in all scaffolds. Alcian Blue staining indicated the presence of proteoglycan in these scaffolds (Fig. 3D, 3E and 3F). Alcian Blue and immunohistochemical staining showed that GAG and type II collagen synthesis were most active in the SF/CS (1:1) scaffolds. These results indicate a difference in the activity of chondrocytes in the different types of scaffold. The reason for varied production of extracellular matrices in the different scaffolds may be due to the differences in chemical compositions of the matrices. The enhanced deposition of glycosaminoglycan and collagen promoted by chitosan is consistent with that observed in pure chitosan scaffolds and collagen/chitosan/ glycosaminoglycans composite scaffolds [31, 67] and may reflect the biological similarity of chitosan to native GAG [30]. .

An optimum scaffold should meet certain criteria, such as suitable 3-D structure for cell growth and nutrient transport and optimal pore size to prevent cell loss from the scaffolds [67–69]. Although the SEM results showed three dimensional structures with interconnecting pores, slight differences were observed in the pore size and morphology of

the blended scaffolds (100– 130 μ m) from the pure silk fibroin scaffolds with 80 μ m (p<0.001) pores in the present study (Fig. 2). The SF/CS (1:1) scaffold showed the highest level of chondrogenesis and enhanced matrix production after 2 weeks and may be due to different pore size and morphology of the different scaffolds. The difference in pore size and morphology can be explained in terms of the crystallization of ice [14, 67]. The pore size of the scaffolds depends mainly on the water content of the initial silk fibroin and chitosan blending solution. Considering that the pore characteristics depend on the properties of this initial solution (2% chitosan solution and 2% silk fibroin solution), an increase in the chitosan content induced a larger pore size, which is attributed to the higher water content of the initial silk fibroin/chitosan blending solution. The effect of pore size on chondrogenesis was in agreement with the earlier report of chitosan and collagen/chitosan/ glycosaminoglycans composite scaffolds where larger pores facilitated the chondrogenesis [30, 67, 70]. This result suggests that porous structures and pore size can be controlled by varying the blending ratio of the silk fibroin and chitosan solutions and these differences may play a role in enhancing chondrogenesis.

Cell-based tissue engineering of cartilaginous tissue is promoted by cells that maintain the differentiated chondrocyte phenotype and that deposit cartilage-specific matrix with mechanical function. Live-dead assay indicated attachment of articular chondrocytes to the scaffolds (Fig. 3A, 3B and 3C). Chondrocytes maintained their spherical morphology after 2 week of culture and distribution was uniform in all the scaffolds. This suggests the suitability of silk fibroin/chitosan blend scaffolds for chondrocyte growth. Histological and immunohistochemical observations of cryo-sectioned scaffolds showed the presence of matrix in each of the scaffolds. Chondrocytes grown on silk fibroin/chitosan blended scaffolds synthesized an extracellular matrix containing proteoglycans and type II collagen, demonstrating the ability of these blended SF/CS scaffolds to support chondrocyte attachment and cartilaginous matrix biosynthesis.

To serve its function as a biomechanical structure articular cartilage is exposed to a variety of forces such as hydrostatic pressure, compression and shear forces [71]. In this study we examined the biomechanical properties of engineered constructs by unconfined dynamic compression. In our study, the deposition of matrix within constructs was related to the development of frequency-dependent dynamic compressive behavior indicative of viscoelastic properties that are important for the biomechanical function of articular cartilage. The increase in compressive modulus with cell seeding indicates functional deposition of matrix (Fig. 7). The increase in phase shift with dynamic test frequency indicates time-dependent behavior that is not present in scaffolds alone (Fig. 8 A and B). Such a difference in stiffness amplitude and phase suggests that hydrostatic pressure is developed within the cartilaginous matrix and contributes to load support by the constructs. While the magnitude of compressive properties of constructs and cartilage are quite different, cultures for more prolonged durations, or with higher concentrations of cells, may accentuate the development of mechanical properties in the SF/CS constructs.

5. Conclusions

Silk fibroin based chitosan hybrid polymer scaffolds promote favorable biological responses of seeded chondrocytes for cartilage tissue engineering. Silk fibroin is reported to be an effective tissue engineered scaffold for facilitating chondrogenesis. Addition of chitosan in the silk fibroin scaffold resulted in increased GAG and collagen synthesis indicating an effect of different chemical compositions on ECM production, with silk fibroin and chitosan (1:1) scaffolds having the most favorable chondrogenic properties. The resulting constructs exhibited viscoelastic properties that are important for cartilage biomechanical function. These findings demonstrate that silk fibroin/chitosan scaffolds are suitable substrates for

tissue engineering of articular cartilage. Based on the observations of two weeks culture, future cartilage tissue engineering investigations could study whether such properties are maintained and enhanced in longer term studies, or the culture of different type of cells, such as mesenchymal stem cells on the silk fibroin and chitosan blended scaffolds would yield similar outcomes.

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Figure 1.

Schematic of experimental design. Polymer solutions of 2% silk fibroin (SF) and 2% chitosan (CS) were used to generate scaffolds of SF, SF/CS (2:1) blend, and SF/CS (1:1) blend. Bovine chondrocytes were seeded into scaffolds and incubated for 2 weeks. Tissue-engineered constructs were then analyzed for composition (biochemistry), structure (histochemistry and immunohistochemistry), and function (biomechanics).



Figure 2.

SEM micrographs showing morphology of scaffolds. (A) Pure silk fibroin (SF) scaffolds (2% w/v); (B) SF/ CS (2:1); (C) SF/ CS (1:1). Scale bar = $100 \mu m$.

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Figure 3.

Micrographs of cells and matrix in SF, SF/CS (2:1), and SF/CS (1:1) construct. Live/Dead (A–C), Alcian Blue (D–F) histochemistry, collagen I immunohistochemistry (G–I), and collagen II immunohistochemistry (J–L) of SF (A, D, G, J), SF/CS (2:1) (B, E, H, K), and SF/CS (1:1) (C, F, I, L) constructs. In live-dead assay, green color indicates live cells and red color indicates dead cells (nuclei). In Alcian Blue staining, light blue colour shows sulphated glycosaminoglycans deposition and purplish blue colour in immunohistochemical staining shows localization of Col I and Col II in cartilage matrix. Scale bar = $50 \,\mu$ m.



Figure 4.

Pictograph shows (A) Alcian Blue and (B) immunohistochemical staining of non-seeded scaffolds. Scale bar = $50 \ \mu m$.



Figure 5.

GAG and collagen deposition by chondrocytes in SF, SF/CS (2:1), and SF/CS (1:1) constructs. Total GAG present in media and scaffolds and normalized GAG (A and B). Total collagen and normalized collagen in blended and silk scaffolds (C and D). Normalization of GAG and collagen was done by wet weight of the constructs after 2 weeks of culture. Each point represents the mean \pm SD (n=6). * Significant differences between groups at p<0.05.



Figure 6.

Thickness of the non-seeded scaffolds (open) and constructs after 14 days of culture (filled).

* Significant differences between groups at p<0.05.

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Compressive elastic modulus of the non-seeded scaffolds (open) and constructs after 14 days of culture (filled). * Significant differences between groups at p<0.05.



Figure 8.

Dynamic compressive properties of non-seeded scaffolds and seeded constructs after 14 days of culture. (A) Dynamic stiffness amplitude and (B) phase. Error bars are averages over all test frequencies.

Table 1

List of some representative biomaterials (natural and synthetic) used for different chondrocytes based cartilage tissue engineering.

Materials	Cell types	Methods	Findings	References
Silk fibroin	Rabbit chondrocytes	SEM, Safranin O staining, immuno histochemical staining	Comparison of salt leached and solvent based scaffolds	43
Silk fibroin	Human chondrocytes	Histology, RTPCR, Alcian Blue staining	Comparison of 2-D and 3-D matrices	17
Chitosan	Porcine chondrocytes	Quantitative analysis, SEM	Effect of varying pore size	31
Chitosan	Sheep chondrocytes	Biochemical and biomechanical analysis, histology, <i>in vivo</i> studies	In vivo cartilage repair in sheep	44
Chitosan/polyester	Bovine chondrocytes	Microcomputed tomography, histology	Effect of pore size and geometry of pore	7
Silk fibroin/chitosan	Bovine chondrocytes	Biochemical and biomechanical analysis, Histology,SEM, Immunohistochemistry	Effect of different chemical composition	This study
Collagen	Porcine chondrocytes	SEM, cell viability, Histology, RTPCR	Effect of different matrices	47
Gelatin	Rat chondrocytes	Massons stain, SEM	Effect of novel gel cross-liking method	48
Gelatin– chondroitin– hyaluronan	Porcine chondrocytes	SEM, histology	Comparison of static and dynamic culture conditions	49
Chondroitin-6- sulfate/dermatan sulfate/chitosan	Rat chondrocytes	RTPCR, immuno histochemical staining	Response surface methodology used for scaffolds preparation	50
Poly (glycolic acid)	Porcine chondrocytes	Cryopreservation, DNA content determination	Effect of cryopreservation	51
Poly(€caprolactone)	Porcine chondrocytes	SEM, histology, qtPCR	Effect of material design on chondrogenesis	52