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Single molecule detection from a large-scale SERS-active Au₇₉Ag₂₁ substrate

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Detecting and identifying single molecules are the ultimate goal of analytic sensitivity. Single molecule detection by surface-enhanced Raman scattering (SM-SERS) depends predominantly on SERS-active metal substrates that are usually colloidal silver fractal clusters. However, the high chemical reactivity of silver and the low reproducibility of its complicated synthesis with fractal clusters have been serious obstacles to practical applications of SERS, particularly for probing single biomolecules in extensive physiological environments. Here we report a large-scale, free standing and chemically stable SERS substrate for both resonant and nonresonant single molecule detection. Our robust substrate is made from wrinkled nanoporous Au₇₉Ag₂₁ films that contain a high number of electromagnetic "hot spots" with a local SERS enhancement larger than 109 . This biocompatible gold-based SERS substrate with superior reproducibility, excellent chemical stability and facile synthesis promises to be an ideal candidate for a wide range of applications in life science and environment protection.

For the few techniques^{1–4} that are capable to reach analytical
limit. SERS-based single molecule detection $(SM-SERS)^{1,5-12}$ is mostly achieved from silver colloidal
nanoparticles that are randomly deposited on glass or limit. SERS-based single molecule detection (SM-SERS)^{1,5-12} is mostly achieved from silver colloidal nanoparticles that are randomly deposited on glass or silicon substrates. Very low molecule concentrations $(< 10^{-8}$ M) are usually chosen to ensure that statistically there is no more than one molecule per colloid, and that the Raman signal originating from the sample can be considered from a single molecule. Large electromagnetic fields for SM-SERS are typically generated at the junction between nanoparticles (namely ''hot spots''). Extensive studies have recently focused on creating ''hot spots'' or boosting SERS-active sites on colloidal nanoparticles^{7-9,13,14}, such as in nanogaps⁷, or in the nanofabrication of silver heterodimers⁸, star-like nanoparticles⁹, and oxide shell-isolated nanoparticles¹⁴. The main problems with colloidal particles are their poor structural reproducibility and the easy oxidation and sulfuration of silver nanoparticles. To control and immobilize SERS-active sites, advanced lithographic techniques have succeeded in optimizing well-ordered arrays of silver-dimer¹⁵ and bowtie-nanoantenna¹⁶. These devices show a large Raman enhancement ability but not SM-SERS. One possible reason is that the maximum enhancement of the local field at the minimum scale with these ''top-to-down'' approaches is still insufficient for single-molecule detection. Other attempts for developing highperformance SERS substrates include the use of poly(vinyl alcohol) fibers¹⁷ or three-dimensional (3D) nanoporous alumina membranes18 to carry SERS-active nanoaggregates. The shortcoming of these techniques is that the fabrication procedure is very complicated and the SERS performance relies on the quality of both the membranes and the nanoparticles. To the best of our knowledge, there is no report on large scale and free-standing SERSactive films with single molecule Raman resolution.

It has been known for decades that nanoporous metal films with a bicontinuous network structure can be synthesized by a dealloying method^{19,20}, but their superior optical properties in SERS have been demonstrated only very recently^{21,22}. Strong electromagnetic fields appear in the vicinity of metallic ligaments due to the large curvatures of the nano-sized ligaments and nanopores, as well as the electromagnetic coupling between the neighbour ligaments²³. Easy synthesis and large film size promote dealloyed nanoporous metal films as effective SERS-active substrates. However, their in-plane structural features and smooth ligament surfaces resulted from chemical etching constrain the local electromagnetic field strength below the limit for SERS-based single molecule

Figure 1 | Wrinkled nanoporous substrate. (a) Schematic diagram of the preparation of wrinkled nanoporous films. (b) Photograph of a plasmonic substrate consisting in a wrinkled nanoporous film with dimensions 30 mm \times 15 mm supported on a polymer substrate. (c) Photograph of the substrate used in this work with dimensions of 8 mm \times 8 mm.

detection²¹⁻²³. The SERS performances of nanoporous metal films can be further improved by introducing 3D quasi-periodic wrinkles through thermal contraction of pre-strained polymer substrates (Fig. 1a) 24–26. Using this procedure detailed in the Method section, rich SERS-active nanostructures at the ridges of the wrinkles, such as nanogaps and nanotips, are produced by deformation and failure of metal ligaments during film shrinking. The dimension of the wrinkled substrates can be discretionarily changed from several micrometres to tens of centimetres ready for direct device applications (Fig. 1b and c). In this study, we demonstrate that SM-SERS for both resonant and nonresonant molecules can be readily achieved from a robust, chemically stable and large-scale SERS-active wrinkled nanoporous $Au_{79}Ag_{21}$ alloy film substrate.

Results

The microstructure of the as-prepared nanoporous film used in this study is illustrated in Fig. 2a. As shown in the low-magnification scanning electron microscopy (SEM) image, the as-prepared film is flat before wrinkling treatment. The dealloyed film possesses bicontinuous nanoporosity and free etching leads to nanopores and ligaments nearly identical in size, geometry and topology (Fig. 2a inset). Quasi-periodic wrinkles formed by annealing distribute uniformly across the entire film (Fig. 2b). The biaxial wrinkling gives rise to rose-petal-shape nanostructures, as shown in the inset of Fig. 2b. According to chemical composition analysis, the nanoporous film contains more gold than silver after dealloying, in a proportion of 71:29 (at. %) (Fig. 2c). It is worth noting that the wrinkling treatment

Figure 2 | Microstructure characterization of wrinkled nanoporous films. (a) SEM micrographs of the flat nanoporous film with a characteristic length of 20 \sim 25 nm. (b) Microstructure of a wrinkled nanoporous film with a quasi-periodic wavelength of 10 \sim 15 µm. (c) Chemical composition of a nanoporous film measured by energy dispersive X-ray spectroscopy. (d) Microstructure of wrinkle ridges showing nanogaps, interleaving broken ligaments and linear chains of self-similar nanocavities.

does not cause any obvious change in the nanoporosity of the wrinkled nanoporous film except for a slight coarsening of the nanopores and the gold ligaments. Owing to the large deformation caused by wrinkling, cracks are frequently observed along the ridges of wrinkles (Fig. 2d), which form nanogaps by interleaving fractured gold ligaments with various spaces from smaller than one nanometre to tens of nanometres.

We first demonstrate the SERS efficiency of the wrinkled nanoporous film by comparing average Raman intensities of active fluorescent molecules of R6G deposited on the flat nanoporous film and the wrinkled one. As illustrated in Fig. 3a, Raman signals are undetectable for R6G with 1×10^{-10} M concentration on the flat film. However, intense characteristic Raman bands of R6G at the same concentration emerge on the wrinkled film. Even when the R6G solution is further diluted to 1×10^{-12} M, the characteristic bands of R6G are still clearly visible. The Raman mapping image shows wrinkle patterns due to the enhancement of the Raman signals along with the fluorescence background from bright ridges (Fig. 3b). The overlay (Fig. 3c) of the Raman mapping with the corresponding optical microscopic image indicates that the strongest enhanced signals are mostly from broken wrinkle ridges. Since Raman scattering is an extremely inefficient process with a cross section $(-10^{-30} \text{ cm}^2 \text{ per molecule})$ -14 orders of magnitudes smaller than that of florescence from the chromospheres with large quantum yields (\sim 10⁻¹⁶ cm² per molecule)¹, the SERS signal is comparable with normal fluorescent signal only at ultra high SERS enhancement. Unlike fluorescence signals, SERS is rich of molecular information provided from light scattered from very local and confined fields ("hot spots" which are about 1 μ m² size in this study and identified by Raman fingerprints). Even at 10^{-10} M, a typical SERS spectrum

with plenty of peaks recorded from a single hot spot (Fig. 3d) suggests ensemble molecules in the probe region²⁷.

Typical single-molecule SERS spectra from hot spots are shown in Fig. 3d for 10^{-12} M R6G (~0.25 molecules per μ m², see the Method Section). In contrast to 10^{-10} M R6G, only selective peaks with narrower linewidth are visible in each spectrum. While these peaks differ from site to site in frequencies, relative intensities and linewidth, they can all be assigned to vibrational bands of R6G (see the Supplementary Information), demonstrating a common signature of single molecule detection^{1,6,27}. In the case of a molecule ensemble, the characteristic SERS spectra are nearly identical from site to site since a large number of molecules with various orientations contribute to the accumulated SERS spectra.

The intensity of 28 hot spots for 10^{-12} M R6G molecules is reported in the inset of Fig. 3d. Most hot spots have a similar enhancement factor, but about 1/10 hot spots show an enhancement higher by more than one order of magnitude. The average enhancement factor of the wrinkled nanoporous $Au_{79}Ag_{21}$ film is estimated to be \sim 3 \times 10⁸ (see the Supplementary Information), which is about 4 times higher than that of wrinkled nanoporous pure gold²⁶. In our Raman imaging measurements, the intensity of a hottest spot is \sim 500 times stronger than the average, indicating that the maximum enhancement factor of local fields on the wrinkled nanoporous Au₇₉Ag₂₁ film is as high as \sim 10¹⁰–10¹¹. The ultrahigh SERS enhancement from the wrinkled substrate is most likely from the superposition of the high electric field strength at the molecule (so-called electromagnetic effect) and the molecular polarizability (so-called chemical enhancement). The intensity fluctuation of scattering modes in each Raman spectrum for 10^{-12} M R6G (Fig. 3d) is related to the specific orientation of the molecule with respect to the laser

Figure 3 | Single molecule detection of R6G molecules. (a) Raman spectra of R6G from a flat nanoporous $Au_{79}Ag_{21}$ film and wrinkled one. (b) Typical Raman map of R6G (10^{-12} M) on the wrinkled Au₇₉Ag₂₁ film. The pixel size in the Raman map is 250 \times 250 nm². (c) Overlay of the image in **b** and the corresponding optical microscopic image. The hot spots can be identified along the broken gaps of the wrinkled nanoporous film. (d) Typical SERS spectrum (blue) taken from a hot spot for 10^{-10} M R6G. Single molecule SERS spectra (A) (B) (C) for 10^{-12} M R6G taken from the three hot spots in the same colour squares in b. The inset in d gives the statistical distribution of Raman intensity of 28 hot spots in a Raman map of 10^{-12} M R6G. The laser excitation is 532 nm.

beam polarization²⁸, and the predominant peaks are associated with the individual adsorption status of the detected single molecules.

To further confirm the suitability of the wrinkled nanoporous film for single molecule SERS detection, we investigate an optically nonresonant bio-molecule: DNA nucleobase adenine. Due to their electrical neutrality in solution, a much smaller number of adenine molecules adsorb onto the nanoporous $Au_{79}Ag_{21}$ surface compared to R6G. In this study we chose a 785 nm excitation laser with which adenine molecules are not resonant. A typical Raman map of 10^{-9} M adenine is shown in Fig. 4a. The overlay of the Raman map and the optical microscopic image shows more clearly that the most active sites locate at the narrow broken ridges of the wrinkled nanoporous film. We believe that these ''hottest spots'' are associated with the coupled tips of fractured ligaments at the nanogaps, as shown in Fig. 2d. For a pair of tip-to-tip nanostructure with a small gap, the field enhancement due to the coupling effect can be 1000 times larger than that of an isolated one²⁹. Computer simulations suggest that a linear chain of several metal particles or self-similar nanocavities creates spectrally and spatially configurable ''superlenses'' as resonant amplifiers and nanoconcentrators³⁰. Interestingly, the "superlenses'' can be frequently observed along the ridges of the wrinkled film (Fig. 2d). They can also act as the ''hottest spots'' responsible for the giant Raman scattering enhancement illustrated in the inset of Fig. 4b.

Similarly to the result for R6G, several characteristic Raman modes of adenine are visible for a concentration of 10^{-9} M (Fig. 4c), while only a few selective Raman bands appear when the concentration decreases to 10^{-12} M (Fig. 4d) (see the Supplementary Information). We rinsed the sample by large amount of water to remove the adenine molecules close to zero in any record pixel. Yet, Raman signals of adenine are still observed at few broken ridges of the wrinkles though the detectable hot spot number decreases dramatically. The extremely dilute-molecule detection at the single-molecule level appears to result from dominant local plasmon fields. In this case, the concentration of molecules is not the most important factor for detection. What really matters is that molecules reside at ''hot spots''.

Discussion

The density of hottest spots for SM-SERS detection in the wrinkled nanoporous film is estimated to be \sim 0.032 per μ m² for 10⁻¹² M of R6G in a Raman map. This value could be underestimated because molecules may not cover all SERS-active sites with the 10^{-12} M R6G concentration. Using colloidal particles, a quantitative measurement shows that the hot SERS-active sites (enhancement factor $>10^9$) account for only 63 sites over 1 million of the total sites⁵. According to our conservative estimation, the density of the ''hot spots" (enhancement factor $>10^9$) in the wrinkled film is thus more than 2 orders of magnitudes higher than that of colloidal particles. Regarding the chemical stability, detectable degradation in the SERS enhancement has not been seen after keeping the wrinkled nanoporous $Au_{79}Ag_{21}$ substrate in ambient environment for more than one month.

In conclusion, we developed a novel nanoporous $Au_{79}Ag_{21}$ substrate with quasi-periodic wrinkles. We observe super enhanced SM-SERS signals from the substrate whatever the probed molecules are resonant or nonresonant with the excitation laser. The enhancement factor for most of the "hot spots" typically reaches up to 10⁹, and can be as high as 10^{10} – 10^{11} for \sim 10% of them. The excellent SERS performance of the wrinkled nanoporous $Au_{79}Ag_{21}$ film comes from its heterogeneous nanostructures containing nano-pores, nano-tips, nanogaps and even superlenses, which give the substrate a broadspectrum of plasmon frequencies for a wide range of molecule detection. The large-scale gold-based SM-SERS substrate with superior

Figure 4 | Single molecule detection of DNA adenine molecules. (a) Typical Raman map of adenine $(10^{-9}$ M) (left), corresponding optical microscopic image (middle), and their overlay image (right). The hot spots can be directly observed at the narrower gaps of the wrinkled nanoporous film. (b) Fluctuation field intensity with 3 chained hot spots (adenine 10⁻⁹ M). The configuration is analogous to the "superlenses" with a linear chain of nanocavities suggested by computer simulations 30 . Raman map size: 4.0 \times 5.5 µm². The spectral interval is 250 nm. (c) SERS spectrum of 10 $^{-9}$ M adenine taken from a hot spot. (d) Single molecule SERS spectrum of 10^{-12} M adenine with selective Raman bands. The laser excitation is 785 nm.

reproducibly, facile synthesis and excellent stability may open new avenues for a wide range of applications in life science and environment protection where single molecule detection and identification are critical. Considering the chemical inertness and biocompatibility of the gold-rich alloy, the single-molecule SERS substrate may allow direct visualization of single biomolecules and their assemblies under native physiological conditions for improving our understanding of the behaviour and interaction of individual biological molecules.

Methods

Nanoporous films were prepared by selective dissolution of silver from 100 nm thick Au₂₅Ag₇₅ (at. %) leaves in 69% nitric acid²⁰. The chemical composition of the dealloyed films was optimized by controlling etching time³¹. The carefully washed nanoporous films were physically attached on pre-strained amorphous polystyrene sheets (PS) (KSF50-C, Grafix). A nanoporous film veneered PS substrate was first baked at 80 $^{\circ}$ C and subsequently annealed at 160 $^{\circ}$ C, ²⁶ leading to more than half volume shrinking of the PS. The wrinkled substrate was put in an aqueous solution of the probed molecules and dried in air. We estimate the average surface density of the probed molecules to be \sim 0.25 per μ m² of the geometric area of the surface for 1×10^{-12} M solutions (see the Supplementary Methods for more details). Rhodamine 6G (R6G, Aldrich) in aqueous solution and adenine (A, Aldrich) in 0.1 mM NaCl aqueous solution were used as probed molecules for SERS measurements. The structural features of as-prepared and wrinkled $Au_{79}Ag_{21}$ films were investigated by scanning electron microscope (SEM, JIB-4600F). SERS spectra and images were measured using a Nanophoton laser Raman microscope (RAMAN-11) (100×, $NA = 0.9$). The laser power was set at as low as ~ 0.1 mW to avoid possible damage from laser irritation.

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Author contributions

H.W.L., L.Z., Q.K.X. and M.W.C. designed the experiments. H.W.L. and L.Z. performed the experiments and analyzed the data. X.Y.L. contributed in the development of nanoporous metals. Y.Y., H.I. and Y.I. assisted in Raman experiments. H.W.L., Q.K.X. and M.W.C. co-wrote the paper. All authors discussed the results and commented on the manuscript.

Additional information

Supplementary information accompanies this paper at [http://www.nature.com/](http://www.nature.com/scientificreports) [scientificreports](http://www.nature.com/scientificreports)

Competing financial interests: The authors declare no competing financial interests.

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