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Fibronectin and vitronectin promote human fetal osteoblast cell attachment and proliferation on nanoporous titanium surfaces

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Abstract

Improvements in osteoconduction of implant biomaterials require focusing on the bone-implant interface, which is a complex multifactorial system. Surface topography of implants plays a crucial role at this interface. Nanostructured surfaces have been shown to promote serum protein adsorption and osteoblast adhesion when compared to microstructured surfaces for bone-implant materials. We studied the influence of the serum proteins fibronectin and vitronectin on the attachment and proliferation of osteoblasts onto nanostructured titania surfaces. Human fetal osteoblastic cells hFOB 1.19 were used as model osteoblasts and were grown on nanoporous $TiO₂$ templates, using Ti6Al4V and commercially pure Ti substrates as controls. Results show a significant increase in cell proliferation on nanoporous $TiO₂$ over flat substrates. Initial cell attachment data exhibited a significant effect by either fibronectin or vitronectin on cell adhesion at the surface of any of the tested materials. In addition, the extent of cell adhesion was significantly different between the nanoporous $TiO₂$ and both Ti6Al4V and commercially pure Ti substrates, with the first showing the highest surface coverage. There was no significant difference on osteoblast attachment or proliferation between the presence of fibronectin or vitronectin using any of the material substrates. Taken together, these results suggest that the increase in osteoblast attachment and proliferation shown on the nanoporous $TiO₂$ is due to an increase in the adsorption of fibronectin and vitronectin because of the higher surface area and to an enhanced protein unfolding, which allows access to osteoblast binding motifs within these proteins.

Keywords

nanoporous TiO2; hFOB 1.19; protein adsorption; vitronectin; fibronectin

1. INTRODUCTION

Osteoconduction is the growth of bony tissue into the structure of an implant biomaterial.¹ Improvements in the osteoconductive capacity of a biomaterial requires focusing efforts in the bone-implant interface, which is complex and involves numerous factors, e.g., mechanical loading, implant material-related factors, such as composition, topography, and surface energy, and patient variables, such as bone quantity and quality. 2 Since osteoconduction involves the direct interaction between bone cells and the biomaterial, the

understanding of the adhesion of osteoblasts forming the bone-implant interface will pave the way to the design of new and efficient implant biomaterials.

Commercially pure titanium (cpTi) and its alloys (e.g., Ti6Al4V and Ti6Al7Nb) are widely used for the fabrication of prosthetics with the former being commonly used for load bearing devices such as artificial hip joint stems. Their use as biomaterials offer attractive features which include excellent corrosion resistance, biocompatibility and good strength-to-weight ratio.³ The induction of corrosion resistance and biocompatibility are related to the presence of an oxide (passive) layer, which prevents the formation of fibrous tissue around the implant, and creates direct contact to the bone tissue. In Ti6Al4V the mixed oxide layer consists of mainly TiO₂ with 6–8 at% Al and less than 0.1 at% V in the outermost atomic layers of the oxide.^{4,5} The percentage of area coverage by each of these oxides is strongly dependent on the surface composition and processing conditions. This oxide layer also has a significant effect on the Ca/P ratio in the calcium phosphate film deposited when compared to pure titanium alloy during mineralization.^{6,7} In comparison with cpTi, the calcium phosphates formed on Ti6Al4V had a Ca/P ratio less similar to that of naturally-occurring hydroxyapatite, which can be attributed to the presence of regions consisting of aluminum $oxide.^{6,7,8}$

An important part of the osseointegration is the osteoblast adhesion process which involves protein adsorption, cell interaction with the adsorbed proteins, cell attachment and spreading on the implant surface.⁹ In cell culture, protein adsorption on flat surfaces (e.g., tissue culture polystyrene) occurs nearly instantaneously forming a 2–5 nm layer through molecular-scale interactions with the substrate. Such proteins are part of the serum which is used to supplement the cell culture media and include extracellular matrix (ECM) proteins such as collagen, thrombospondin, fibronectin, vitronectin, and osteopontin. Among these, fibronectin and vitronectin are of particular interest for osseointegration since they induce the reorganization of actin microfilaments promoting cell adhesion and spreading, which in turn affects cell morphology and migration. $10-13$

When cells come in contact with this adsorbed protein layer (i.e., \sim 5 min after seeding), they attach through non-specific physicochemical interactions, such as ionic and van der Walls forces. This is followed by recognition of cell binding motifs on these proteins which is mediated by integrins, a widely expressed family of transmembrane adhesion receptors, consisting of α and β heterodimers, which bind to specific amino acid sequences, such as the RGD cell binding domain.^{14,15} Upon binding to their ligand, integrins rapidly associate with the actin network of the cytoskeleton and cluster together to form focal adhesions, which are discrete complexes that contain structural and signaling molecules ^{16,17} and function as structural links between the cytoskeleton and ECM to mediate cell adhesion and migration. In conjunction with growth-factor receptors, focal adhesions activate signaling pathways such as MAPK and JNK which regulate transcription factor activity and direct cell proliferation and other functions.^{18–20} At later times, cell attachment is strengthened by synthesis and deposition of additional ECM proteins promoting a stronger binding.⁹ The layer of adsorbed proteins mediate subsequent interactions with cells from the surrounding tissues, promoting cell functions pertinent to new tissue formation, which lead to integration and stabilization of the implant. $10,21$

The chemical and physical characteristics of the material surface affect the amount, distribution, density, conformation and orientation of adsorbed proteins.^{22–24} Although all aspects and the underlying mechanisms of surface-protein interactions are not well understood, it is known that the material surface chemistry is a determining factor. In addition to surface composition (i.e., biomaterial chemical composition and proteins adsorbed), surface topography also plays an important role in osseointegration. For instance,

an increasing number of studies demonstrate that nanoscale topographies significantly enhance cell functions leading to improved osseointegration compared to conventional micron-structured surfaces.^{25–28} Recent interest in nanoporous $TiO₂$ templates as alternatives to improve on Ti-based implants have shown promise by improving osteoblast attachment and proliferation.^{29–32} These nanoporous templates can be used to modify implants by RF sputtering followed by anodization.³³ Despite all the work focused on the biomaterial-tissue interface, an understanding on how biomaterial surface properties and the adsorbed protein layer affect cell behavior is still lacking. While investigations have found enhanced protein adsorption on nanostructured surfaces $34-36$ very few studies have focused on its effect on cell function.³⁷ In this work, we have fabricated nanostructured TiO₂ substrates using a simple anodization process, $30,38$ to study the influence of the serum proteins fibronectin (FN) and vitronectin (VN), and nanotopography on osteoblast cell attachment and proliferation using the human fetal osteoblast cell line hFOB 1.19. Simple methods to improve the osteoconductive capacity of biomaterials will have a strong impact on the orthopedic implant industry.

2. MATERIALS AND METHODS

2.1. Nanoporous TiO2 Fabrication

Nanoporous $TiO₂$ templates were fabricated by anodization of titanium foil (Alfa Aesar) under a constant voltage of 60 V and in potassium fluoride (0.05M in 10% deionized (DI) water/90% glycerol) solution. After 15 h of anodization, nanotubes 125 nm in diameter and 2.5mm in length were achieved as previously reported.³⁸

2.2.Characterization

Scanning electron microscopy (SEM) was used to characterize the surface morphology of the nanoporous $TiO₂$ templates and the Ti6Al4V and cpTi substrates using a Field Emission Scanning Electron Microscope Supra 35 (Zeiss). The discs were analyzed in the "asreceived" condition and without the use of coatings. The equipment allows analysis of features in the nanometer range using low accelerating voltage (3kV).

2.3. Cell culture

Human osteoblast cell line hFOB 1.19 (CRL-11372, ATCC, Manassas, VA,USA) was cultured in 90% Dulbecco's Modified Eagle's Medium Nutrient Mixture F-12 Ham, 1:1 (DMEM) (Sigma-Aldrich, St. Louis, MO) with 2.5mM L-Glutamine and 15mM Hepes, without phenol red, supplemented with 0.3 mg/ml G418 (Sigma-Aldrich, St. Louis, MO) and 10% Fetal Bovine Serum (FBS) (Sigma-Aldrich, St. Louis, MO). Cells were cultured at 33.5° C and 5% CO₂. At confluence, cells were trypsinated and seeded at a final concentration of $1x10^4$ cells/cm² onto the substrates.

2.4. Substrate preparation for cell culture experiments

Metal rods 1cm in diameter by 30 cm in length of Ti6Al4V and cpTi were obtained from a commercial source, which were cut into disks 0.1 cm in thickness. Silicon carbide paper of grits 240, 320, 400 and 600 were used sequentially to grind the Ti6Al4V and cpTi metallic surfaces. After grinding, a solution of A_2O_3 (alumina particles 5 μ m and 3 μ m in size) was used as the polishing agent. The substrates were further polished by a polycrystalline diamond suspension with particle size of 3μ m. Nanoporous TiO₂ templates used were 1cmby-1cm in size and, Ti6Al4V and cpTi substrates were 1.0 cm in diameter and 0.1 cm in thickness (0.5 cm² area). All substrates were incubated for 30 min in 0.53 mM EDTA/D-PBSA, washed with distilled DI water, scrapped using a plastic cell scraper, washed with absolute alcohol and washed again with distilled DI water. Thereafter the discs and controls

were sterilized by autoclaving at 15 psi and 121°C and placed in sterile 6-well plates (Corning, NY).

The nanoporous $TiO₂$ templates and the Ti6Al4V and cpTi substrates were incubated with either 10% FBS, or serum levels of FN (20 μ g/mL) of VN (2 μ g/mL) for 24 h at 37°C. After this time period, each substrate was gently rinsed with PBS at room temperature just before seeding with cells.

2.5. Cell adhesion experiments and imaging

The hFOB 1.19 cells were cultured on substrates at a density of 10⁴ cells/ml and incubated for 6 hours, at 33.5° C and 5% CO₂. Substrates were washed twice with serum-free medium and once with D-PBSA. Cells were fixed using methanol at 4°C for 10 min and washed three times with D-PBSA. Non-specific staining was blocked by incubation in blocking solution (1% BSA in D-PBSA) for 1h at room temperature. The blocking solution was removed and the cells were permeabilized with 0.5% triton X-100 at 4oC for 5 min, washed with D-PBSA and incubated with Phalloidin-TRITC (Sigma-Aldrich, St. Louis, MO) at 10 μg/ml. After 1h at room temperature of incubation, substrates were washed with D-PBSA and followed by the addition of 0.8 mg/ml DAPI (Research organics, Cleveland, OH), for 15min at 37oC. After washing again the substrates with D-PBSA, they were fixed with mounting media and stored at 4°C in the dark. Images were captured using an inverted confocal laser scanning microscope (CLSM) (FluoView 300, Olympus, USA) equipped with 405nm, 543nm Green HeNe and 633nm Red HeNe lasers. Phalloidin-TRITC has a peak excitation at 540–545nm and a peak emission at 570–573nm, while DAPI has a peak excitation at 345–350nm and a peak emission at 455–470nm.

2.6. Cell Viability

Viability of the cells cultured on the substrates were assessed at 6 hours of culture using the commercially available LIVE/DEAD viability/cytotoxicity assay (Invitrogen) and following manufacturer's instructions. Stained cells were visualized in situ using fluorescent microscopy. Live cells appear green (excitation: 494 nm; emission: 517 nm), while dead cells appear red (excitation: 528 nm; emission: 617 nm). Micrographs (taken at twenty five random fields of view) were obtained and used to quantify the number of live and dead cells per substrate tested; percent cell viability on each substrate was thus obtained. These experiments were conducted in triplicate.

2.7. Proliferation rate of hFOB 1.19 cells

Cell proliferation was measured at 1, 3 and 7 days on the nanoporous $TiO₂$ templates and the Ti6Al4V and cpTi substrates. Cells were cultured on substrates at a density of 10^4 cells/cm² and incubated at 33.5° C and 5% CO₂, following labeling with Phalloidin-TRICT (Sigma-Aldrich, St. Louis, MO) at $10 \mu g/ml$. Cells were enzymatically released (1 mM EDTA, 1.3 mg/mL collagenase and 0.25% trypsin; Gibco) and counted using a hemocytometer.

2.8. Statistical analysis

The Student's t-test (two-tailed) was used to evaluate the significance of changes in cell proliferation between control and experimental groups. A probability value of $p<0.05$ was considered statistically significant. All data were analyzed using GraphPad Prism 5.

3. RESULTS AND DISCUSSION

Improvements in the osteoconductive capacity of implant biomaterials require the study of how important implant biomaterial-related factors, such as topography and composition promote osteoblast attachment. Nanostructured materials have shown increased cell

attachment over microstructured or smooth surfaces.^{39,40} An alternative for fabricating nanostructures at the surface of implant materials is the use of an anodization process. ^{29–32} This is an attractive alternative since it can be easily applied to existing implants without affecting current fabrication processes. 41 In addition to nanotopography, biomaterial surfaces modified with ECM proteins or cell-binding motifs have also shown an increase in cell attachment.^{42,43} To study the effect of these parameters we have fabricated nanoporous $TiO₂$ substrates and measured the influence of the FN and VN on osteoblast cell attachment and proliferation on these nanostructured substrates using the human fetal osteoblastic cell line hFOB 1.19.

Figure 1 shows SEM images of cpTi, Ti6Al4V substrates and nanoporous $TiO₂$ templates. The nanoporous TiO₂ with pore diameter of approximately 125 nm and length of 200 μ m were prepared using an anodization voltage of 60V for 15 h. It can be seen that the nonopore distribution is uniform over the substrate. In addition, anodizing conditions can be tuned (e.g., voltage, electrolyte, and pH) to optimize parameters such as pore diameter and length to maximize osteoinduction.⁴⁴

The effect of adsorbed FN or VN on the adhesion of hFOB 1.19 cells on fabricated nanoporous $TiO₂$ templates, Ti6Al4V and cpTi substrates after 6 h of seeding is shown in Figure 2. During this time, cells reach an equilibrium adherence level where they spread on the surface and depend on receptor-mediated binding.⁴⁵ For all three substrates the presence of proteins (i.e., FN, VN or FBS) had a significant (p < 0.05) effect on adhesion. In addition, difference in cell adhesion between FN or VN and FBS were significant ($p < 0.05$) in the case of the Ti6Al4V and cpTi substrates, but not for the nanoporous $TiO₂$ templates. In both the Ti6Al4V and cpTi substrates, FBS decreased the number of cells attached to the substrate, which could be explained by competitive adsorption of other proteins in the serum that are either in larger quantities or with a higher affinity for the substrate.^{46,47} Also, these two substrates showed no significant difference in cell adhesion between them in the presence of FBS or when proteins were not present. Different from these two substrates, nanoporous $TiO₂$ templates had no significant difference between FN, VN or FBS. It has been shown that nanostructured materials allow for an increase in protein unfolding^{39,40} and adsorbed protein density.48 For instance, both FN and VN must unfold on the nanoporous surface to allow access to their otherwise hidden cell-binding RGD tripeptide sites.^{39,49} Thus, nanoporous $TiO₂$ templates could allow higher cell-binding by providing a higher density of cell-binding motifs over the other two substrates. This was shown by the significantly enhanced cell adhesion on the nanoporous $TiO₂$ templates over both the Ti6Al4V and cpTi substrates. In addition to improving protein adsorption and unfolding to allow cells access to the binding motifs, nanopores can also be used as reservoirs of growth factors which affect osteoprogenitor cell proliferation and differentiation such as insulin-like growth factors (IGFs), transforming growth factor-βs (TGFβs), and bone morphogenetic proteins (BMPs).⁵⁰ In a recent study by our group we presented an alumina nanopore template for the sustained release of the model drug doxorubicin over several days.³⁸ It was observed that after 6 h of culture cell viability ranged between 93% and 100% on all the substrates (Data not shown).

To assess cell morphology, hFOB 1.19 cells were analyzed by fluorescence microscopy after 3 days in culture on Ti6Al4V, cpTi, and nanoporous $TiO₂$ that were pre-treated with or without VN (Figure 3). Images similar to the VN-treated surfaces (figures 3a and 3b) were obtained when these were treated with FN and are not shown. In addition, results using cpTi were not shown because of the low cell coverage obtained when compared to the Ti6Al4V substrates. Figure 3 clearly shows that the presence of proteins increases cell attachment on either substrate. Furthermore, Figure 3d shows the formation of cell clusters, which result from cell spreading and interaction and eventually drives their differentiation. This is

expected since after three days hFOB 1.19 cells are in full exponential growth at which time cells start to cluster in a manner proportional to proliferation.⁵¹ Thus, nanoporous TiO₂ templates in the presence of cell-binding proteins promote a more proliferative surface coverage over either Ti6Al4V or cpTi substrates.

To quantify the results shown in Figure 3, hFOB cell proliferation was measured at 1, 3 and 7 days of culture in the presence of 10% FBS. Figure 4 shows a much faster hFOB cell proliferation rate using the nanoporous $TiO₂$ template over the Ti6Al4V or cpTi substrates, confirming the previous results shown in Figure 3. The results for each of the biomaterials were significantly different from each other ($p < 0.05$).

4. CONCLUSION

Simple methods to improve the osteoconductive capacity of biomaterials will have a strong impact on the orthopedic implant industry by providing implants which can provide better osseointegration. In this work, we have shown a strong effect on ostreoblast proliferation by the ECM proteins VN an FN onto nanoporous $TiO₂$ templates over the standard Ti6Al4V and cpTi. These nanopores can me generated on Ti-based implant biomaterials using simple anodization, which can be implemented on existing implants. Because of the increased surface area, nanoporous $TiO₂$ templates are able to adsorb more protein than flat microstructured materials such as the Ti6Al4V or cpTi allowing improved cell attachment and proliferative capacity. In addition, these nanostructured templates seem to allow osteoblast-binding ECM proteins to unfold and permit access to their cell-binding motifs, such as the RGD tripeptide. Further studies are under way to study the conformational changes that occur upon FN and VN adsorption to the nanoporous $TiO₂$ templates.

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Figure 1.

SEM images of (a) cpTi substrates, (b) Ti6Al4V substrates and c) nanoporous TiO₂ templates. Bars (a) and (b) $10 \mu m$, and (c) 300 nm .

Figure 2.

Effect of adsorbed protein on hFOB 1.19 adhesion after 6 h of being seeded onto nanoporous TiO2 templates, Ti6Al4V and cpTi substrates. Data is normalized to the highest number of cells on a template. Bars = mean + SEM with $n = 25$.

Figure 3.

Fluorescence microscopy images (10x) of hFOB 1.19 cells on Ti6Al4V substrates without and with VN treatment ((a) and (c), respectively) and nanoporous $TiO₂$ templates without and with VN treatment ((b) and (d), respectively).Cells were stained with phalloidin-TRITC (red) and DAPI (blue) to see the F-actin cytoskeleton and nuclei, respectively. Bar = 200 μm.

Figure 4.

hFOB 1.19 cell proliferation on nanoporous TiO₂ templates, Ti6Al4V and cpTi substrates for up to 7 days of culture in the presence of 10% FBS. Nanoporous templates show about 40% higher cell proliferation after 7 days of culture than cpTi substrates ($p < 0.05$). Bars = mean $+$ SEM with $n = 25$.