

Molecule-hugging graphene nanopores

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It has recently been recognized that solid-state nanopores in single-atomic-layer graphene membranes can be used to electronically detect and characterize single long charged polymer molecules. We have now fabricated nanopores in single-layer graphene that are closely matched to the diameter of a double-stranded DNA molecule. Ionic current signals during electrophoretically driven translocation of DNA through these nanopores were experimentally explored and theoretically modeled. Our experiments show that these nanopores have unusually high sensitivity (0.65 nA/Å) to extremely small changes in the translocating molecule's outer diameter. Such atomically short graphene nanopores can also resolve nanoscale-spaced molecular structures along the length of a polymer, but do so with greatest sensitivity only when the pore and molecule diameters are closely matched. Modeling confirms that our most closely matched pores have an inherent resolution of ≤ 0.6 nm along the length of the molecule.

Recent work has shown that single DNA molecules in solution can be detected and characterized with graphene nanopores (1–3). The potential advantages of graphene nanopores include high sensitivity and subnanometer-scale resolution along a translocating polymer. We have previously reported that nanopores in single-layer graphene have a length of 0.6 nm, which is the ion-insulating thickness of a graphene layer (1). Here we report on the sensitivity and resolution of graphene nanopores that closely match the diameter of translocating double-stranded (ds)DNA molecules. We conclude that such nanopores should be able to detect and resolve components along the length of a polymer that are as closely spaced as ~ 0.5 nm. This is better resolution than can be obtained from the shortest nanopores previously fabricated in other solid-state substrates such as SiN_x (4, 5). [Nanopores shorter than 1 nm have yet to be fabricated in other solid-state substrates (6, 7).]

In a nanopore sensor, a charged polymer in solution is electrophoretically translocated through a nanometer-scale pore in an insulating membrane separating two voltage-biased reservoirs filled with aqueous ionic solution (Fig. 1A) (8). Here we experimentally explore and theoretically model the properties of nanopores in single-layer graphene. We demonstrate that they achieve their best resolution and sensitivity when their diameter is very close to that of the translocating polymer. Our modeling confirms that subnanometer-scale resolution and unusually high sensitivity can be achieved with a tightly dimensioned pore because most of the ionic current in our atomically short pores is localized near their sharp perimeters.

Most of the work reported here uses dsDNA as the charged polymer. When the negatively charged molecule is translocated through a voltage-biased nanopore, it blocks the flow of ions through the nanopore, leading to a transient drop of the recorded ionic current. This transient drop is called a “blockade”. Its magnitude, ΔI_B , is defined as the difference between the ionic current through the open pore and the current through the pore during polymer translocation (Fig. S1). We show molecule translocation through graphene nanopores for both dsDNA and single-stranded DNA (ssDNA), but the diameters of the nanopores reported here are small enough to provide high sensitivity and resolution only for dsDNA. Recently reported fabrication methods (9) should soon provide tightly fitting nanopores (≤ 2 nm diameter) for ssDNA.

Experimental Results

We fabricated a single nanopore in a series of single-layer graphene membranes, using an electron microscope (*Materials and Methods*). Each graphene membrane was suspended over a 200-nm \times 200-nm aperture in a thin, free-standing SiN_x film supported on a silicon frame. The membrane separated an ionic solution-filled flow cell into two chambers. The two solution-filled chambers, each in contact with a Ag/AgCl electrode, were in fluid contact with each other only through the nanopore in the graphene membrane (Fig. 1A and *Materials and Methods*).

Ionic conductivity measurements through a series of different diameter graphene nanopores show that the open pore currents (I_o) are linearly proportional to the pore diameter even for nanopores with diameters as small as $D \sim 3$ nm (Fig. 1B). The linear dependence of I_o on the pore's diameter (3.85 nA/nm), as opposed to the pore's cross-sectional area, is expected for atomically short nanopores whose current density is sharply peaked near the pore's perimeter and whose resistance is dominated by its access resistance (10). This linear dependence is also consistent with molecular dynamic calculations (11) and with our numerical solutions of the Laplace equation for the scalar potential of the ionic current density distribution in an atomically short pore (*Materials and Methods*). These solutions show that when the nanopore length, L , is much less than the pore diameter D (i.e., $L < D$), the ionic current density is strongly peaked near the nanopore perimeter (Fig. 1C, *Left*). In contrast, when $L \geq D$, the ionic current density is quite uniformly distributed within the nanopore (Fig. 1C, *Right*). Recognizing these fundamentals of ionic current distribution in atomically short pores makes it all the more important to consider, as we do below (see Fig. 3), how these current distributions are affected when the pore is used to sense a translocating molecule.

Fig. 1D shows DNA blockades observed as 10-kb dsDNA molecules are electrophoretically translocated by a 160-mV bias through nanopores of various diameters. To maximize the current through the nanopore, and hence the current blockages attributable to the presence of the translocating molecule, we tested our graphene membranes' abilities to withstand the largest possible biases. Although some graphene membranes maintained their integrity and high ionic current resistance to bias voltages as high as 500 mV, many exhibited a gradual loss of resistance at this bias. Because all tested membranes maintained their resistances when subjected to a 160-mV bias, we used this 160-mV bias in all of the experiments reported here.

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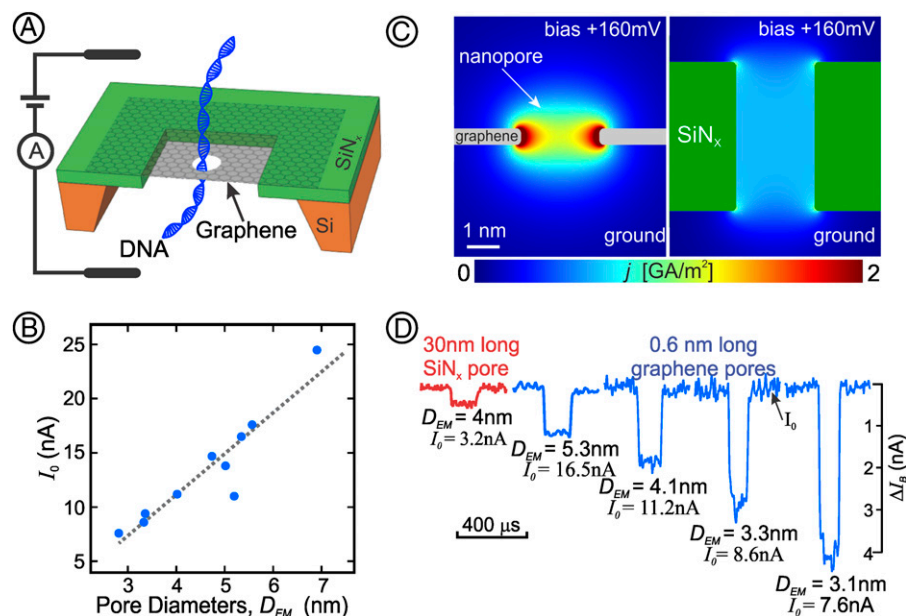


Fig. 1. (A) The experimental device with dsDNA translocating through a nanopore in graphene mounted on a SiN_x frame. (B) Open pore currents through a series of different EM-measured graphene nanopore diameters before addition of DNA. SE bars for I_o are encompassed within the area of the data points; estimated errors for D_{EM} are discussed in *Materials and Methods*. (C) Solution of the Laplace equation graphically displayed for the ionic current density, j (in units of A/m²), mapped through a cross-section of (Left) a short nanopore ($L = 0.6$ nm, $D = 2.5$ nm) and (Right) a longer nanopore with the same diameter ($L = 5$ nm, $D = 2.5$ nm). Note the current density is strongly localized close to the edge of the short nanopore but quite uniformly distributed throughout the entire diameter within the longer nanopore. Because of its greater electrical resistance and lesser current, nowhere in the longer nanopore will the current density reach that seen (yellow and red colors) near the perimeter of the atomically short nanopore in graphene. Scale bar and bias voltage are identical Left and Right. (D) Typical blockades as dsDNA translocates through a SiN_x nanopore and through graphene nanopores of different diameters, D_{EM} . The open pore current (I_o) portion of each trace is shown on the same horizontal line even though the open pore currents varied with the pore diameters; the actual value of each pore's I_o is stated below each pore's EM-measured diameter (D_{EM}). All of the blockade traces are drawn to the same scale (indicated on the right).

Unlike those in the many solid-state nanopores in SiN_x we and others have investigated, the blockade magnitudes in atomically short graphene nanopores always exhibit strong inverse nanopore diameter dependence (Fig. 1D and Fig. S1). Especially for the smaller nanopores whose diameters approach the diameter of the translocating polymer, the observed nanoampere blockade magnitudes are significantly larger than those of the picoampere blockades obtained in other nanopores, exceeding even those of the large blockades observed in *Mycobacterium smegmatis* porin A nanopores (12, 13).

The large blockade magnitudes observed in the closely fitting graphene nanopores (Fig. 1D) are a direct consequence of the molecule blocking the ionic current except in the small region between the molecule's surface and the pore's perimeter. Because of the high current density and large current density variations in this region, small variations in the polymer's geometry or surrounding fluid properties will be significantly reflected in the resulting blockade.

To evaluate the geometrical sensitivity of graphene nanopores to the molecule's blocking diameter, we investigated and analyzed many DNA translocation events for pores of different diameters. Fig. 2 shows translocation event distributions for two pores of different diameter. We characterize each DNA translocation event by two parameters: the event duration, which indicates the time it takes for the molecule to fully translocate through the nanopore, and the average blockade magnitude, which reflects the extent to which the molecule obstructs or blocks the flow of ionic current through the nanopore.

In nanopores with diameters greater than ~ 4 nm, dsDNA molecules translocate through the pore either as an extended linear molecule (leading to a single-dip event; Fig. 2A and B, Left) or as a folded molecule (typically more than doubling the blockade magnitude during traversal of the folded part of the molecule;

Fig. 2B, Right) (14). In nanopores with diameters $< \sim 4$ nm (Fig. 2C), only unfolded molecules were observed (single-dip events; Fig. 2D, Left), consistent with the fact that these nanopores are too small to easily admit folded dsDNA molecules.

From the translocation time distributions observed we conclude that the dsDNA molecules are minimally retarded by strong stochastic or nonspecific chemical interactions with the graphene membrane or the nanopore edge—i.e., they translocate freely through the graphene nanopore under the high-salt, high-pH conditions of our experiment (*Materials and Methods*). This is evident because (a) the mean translocation time ($T_m \sim 100$ μ s) is short, consistent with free polymer translocation theory (11, 15); (b) the translocation-time distribution is narrow, with the full width at half maximum (FWHM) of $\sim 33\%$ of the mean translocation time (Fig. S2) as expected from the ensemble of initial polymer conformations in solution (15); and (c) the distribution of blockade magnitudes from single-file DNA translocation events is narrow, with the FWHM of 10%, indicating well-defined pore-blocking geometry. Strong stochastic interactions are not conducive to polymer analysis because the blockade magnitude values in such cases can be distorted by trapping (16) and coiling up of the molecule at the pore entrance, leading to much larger blockade signals than theoretically expected for fully extended polymers.

The data in Figs. 1C and 2 and Fig. S2 show a trend toward shorter translocation times for smaller pores. This effect may be due to a decreased electroosmotic drag on the translocating DNA in smaller pores. As noted by van Dorp et al. (17), electroosmotic drag was minimized in their smallest nanopores. But, because van Dorp and colleagues performed their experiments with relatively large SiN_x nanopores under different conditions than those used here with graphene nanopores, an explanation for the shorter translocation times with smaller diameter graphene pores will require further investigation. Nevertheless, in contrast to the strong

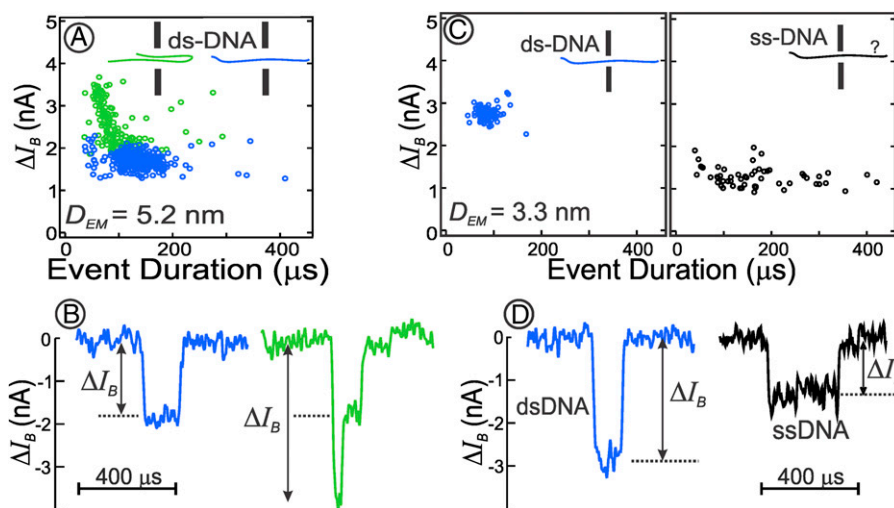


Fig. 2. (A and C) Event plots for different diameter nanopores. Each circle corresponds to a DNA translocation event. Each of the 309 blue circles represents the translocation of a single extended, unfolded dsDNA molecule (pH 10); each of the 170 green circles represents partially or fully folded dsDNA translocations (pH 10); and black circles represent ssDNA translocations (pH 12.5). (B and D) Single typical event current traces corresponding to one of the events from the event plots in A and C for unfolded dsDNA (blue), partially folded dsDNA (green), and ssDNA (black) whose folding is unknown. Both of the event plots and current traces in C and D were from an experiment performed with the same nanopore.

interactions that lead to the long translocation times we and others have observed in small diameter ($\sim 3\text{--}4$ nm) SiN_x nanopores at pH 8.5 or 8.0 with 1.0 or 1.6 M KCl (14, 18–20), our pH 10 with 3 M KCl conditions minimized DNA–graphene interactions (1). These results lead us to conclude that in graphene pores under our conditions, stochastic interactions between the translocating polymer and the nanopore perimeter are not the primary factor that determines the rate of polymer translocation.

Raising the pH of the salt solution from 10 to 12.5 denatures the dsDNA into ssDNA molecules (18). Unfolded ssDNA translocation events (Fig. 2 C and D, Right) show blockade magnitudes less than half those of dsDNA (Fig. 2D, Left), a reflection of the reduced diameter of ssDNA. To our knowledge this is a unique observation of ssDNA translocating through a graphene nanopore. Many ssDNA event durations are widely spread, extending from less than 100 μs to over 400 μs (Fig. 2C). These extended translocation times strongly suggest that many of the ssDNA molecules stick to, or interact with, graphene. Because it is known that nucleobases can form π - π stacking or van der Waals interactions with a graphene surface (21), such interactions are not surprising.

The average blockade magnitude for unfolded dsDNA translocations in graphene nanopores was measured as a function of nanopore diameter for a series of graphene nanopores with different diameters (Fig. 3A, solid blue circles). A steep increase in blockade magnitude with decreasing nanopore diameter is observed. This indicates a continuing increase in sensitivity to the difference between the polymer's diameter and pore diameter as the fit between the two becomes closer. For comparison, we also measured dsDNA blockades for each of two electron beam drilled nanopores with electron microscope-determined diameters (D_{EM}) of 7 nm and 4 nm in 30-nm-thick SiN_x (Fig. 3A, black open circles). Note that because of their more evenly distributed ionic current density even when occupied by dsDNA (Fig. 3D), neither the measured (Fig. 3A, open circles) nor the numerically calculated blockades in longer pores ($L = 30$ nm, 10 nm, or 5 nm; Fig. 3A, solid black and solid colored lines) show very pronounced increases in sensitivity or resolution as the nanopore diameter is decreased.

Model

To understand the experimental data in Fig. 3A and ascertain whether a simple model can explain the observed results, we

numerically solved the Laplace equation to determine current flow through an open graphene pore (I_o) vs. the pore diameter (Fig. S1, dashed line) and through pores of the same diameters containing a dsDNA molecule through its center (depicted in Fig. 3B), to obtain the blockade magnitude, ΔI_B . As fitting parameters we used D_C , the ion-conducting diameter of the pore; L , the length of the nanopore [essentially the ion-insulating thickness of the graphene layer (1)]; and d_{poly} , the diameter of the insulating DNA molecule at the center of the pore. The best fit to the data is given by the black dashed line in Fig. 3A for which the diameter, D_C , is smaller than the D_{EM} by 0.5 ± 0.3 nm, $L = 0.6 \pm 0.4$ nm, and $d_{poly} = 2.1 \pm 0.2$ nm. The 0.5-nm correction to each nanopore diameter may account for solution ions that do not approach the hydrophobic EM-determined pore perimeter due to the energy cost of an ion shedding its hydration shell within the nanopore (22, 23) or the consequences of chemical groups bound to the graphene nanopore's edge (22). Although we cannot rule out error in our method of calculating D_{EM} (Materials and Methods), we note that similar considerations concerning ions approaching each of the two surfaces of the membrane may account for the value of L , 0.6 ± 0.4 nm, which agrees with our previously reported value of $L = 0.6 [+0.9, -0.6]$ nm (1) as well as with theoretical predictions for the similar interaction distance of water on the outside of a single-walled carbon nanotube (0.31–0.34 nm) (24). Finally, we note that the best-fit 2.1 ± 0.2 -nm diameter value for d_{poly} is the generally accepted value for the diameter of B-DNA (25).

Using the parameters given above, the modeled dependence of the blockade magnitude vs. pore diameter (Fig. 3A, dashed black curve) is seen to fit the data (Fig. 3A, solid circles) rather well and leads us to conclude that the calculation and simple model of dsDNA as a cylinder with a diameter of 2.1 nm capture the essential physics with the parameters L and D_C absorbing all of the molecular interaction effects (26).

We define the nanopore's sensitivity (S) as the change in blockade magnitude caused by a change in the effective blocking diameter of the translocating polymer,

$$S(D_C, d_{poly}) = -\frac{\partial(\Delta I_B)}{\partial d_{poly}} \approx -\frac{\partial(\Delta I_B)}{\partial D_C} + \frac{dI_o}{dD_C} \quad [1]$$

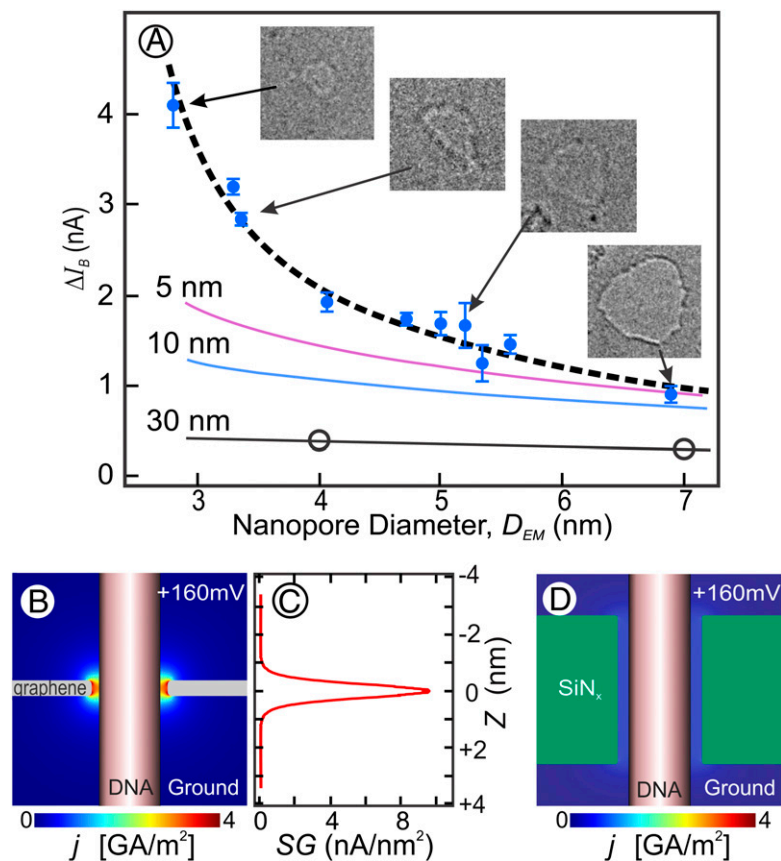


Fig. 3. (A) Average blockade magnitude for dsDNA translocations vs. nanopore diameter. Solid blue circles are the experimentally determined mean ΔI_B signals ($\pm 5\text{E}$ bars) from nanopores in single-layer graphene; open circles are measured ΔI_B signals from a 30-nm-long nanopore in SiN_x . Errors for the D_{EM} values are discussed in *Materials and Methods*. Solid black and colored lines are numerically simulated values for ideal circular nanopores through membranes of the indicated thicknesses. The black dashed curve is the numerically simulated ΔI_B using the dimensions of the boundary conditions D_C , L , and d_{poly} explained in the text. *Insets* are transmission electron micrographs of four sample pores to which the arrows point. The full width of each *Inset* image = 10 nm of the graphene sample. (B) Graphical representation of the calculated current density in a 2.5-nm diameter graphene nanopore ($L = 0.6$ nm) traversed by dsDNA modeled as a 2.1-nm diameter insulating cylinder. (C) Calculated sensitivity gradient (SG) (main text) along the length of a DNA molecule in a graphene nanopore as in B. (D) Graphical representation of the calculated current density in a 2.5-nm diameter SiN_x nanopore ($L = 5$ nm) traversed by dsDNA modeled as a 2.1-nm diameter insulating cylinder. Note that in contrast to the atomically short graphene nanopore in B, the ionic current density in a long pore remains relatively low and quite uniformly distributed throughout the small space between the nanopore's diameter and the dsDNA's diameter, even when the nanopore is occupied by dsDNA.

Eq. 1 is obtained by equating the Taylor expansions of $\Delta I_B(D_C, d_{poly})$ in terms of small changes in d_{poly} to those in terms of small changes in D_C (*SI Text, Derivation of Eq. 1*). S is therefore the sum of the negative slope of the dashed line in Fig. 3A and the slope of the dotted line in Fig. 1B as a function of nanopore diameter. As D_{EM} falls below ~ 3.5 nm ($D_C < 2.8$ nm), the value of the negative slope of the dashed line in Fig. 3A exceeds 0.3 nA/Å in our 3-M KCl medium with a bias of 160 mV. This shows that for each 0.1 -Å change in the graphene nanopore diameter of these tightly fitting nanopores there is a 30-pA change in the ionic current through the DNA-filled nanopore. On the other hand, Eq. 1 shows that the sensitivity to molecule diameter change is augmented by the slope in Fig. 1B (which exceeds 0.38 nA/Å), yielding an S that exceeds 0.65 nA/Å. This shows that for each 0.1 -Å change in the polymer's diameter in a tightly fitting nanopore there would be, remarkably, a >65 -pA change in the ionic current through the polymer-filled nanopore. Similar results should be obtained for ssDNA ($d \sim 1.4$ nm) when similarly tight-fitting nanopores become available.

If a graphene nanopore is used to resolve differences in the geometrical and conductive properties of the fluid along the length of a translocating DNA molecule, one must also be concerned with how the pore's sensitivity falls off as a function of the distance,

z , from the symmetry plane of the membrane. This longitudinal sensitivity gradient (SG) (Fig. 3C) is the resolution function of the nanopore molecule system that identifies where the contributions to S come from along the molecule's length. The nanopore sensitivity must be the integral of this sensitivity gradient. It follows that the resolution function is given by

$$SG = \lim_{\delta d \rightarrow 0} \frac{1}{\delta d R} \left(\frac{dV_{d_{poly} + \delta d}}{dz} - \frac{dV_{d_{poly}}}{dz} \right), \quad [2]$$

where R is the pore resistance when occupied by a translocating molecule, and $V_{d_{poly}}(z)$ is the electrostatic potential along the surface and length of a molecule of diameter, d_{poly} , as a function of distance from the pore, z . The sensitivity gradient full width at half maximum of ≤ 0.6 nm for the $D_{EM} = 3.2$ -nm pore (Fig. 3C) agrees with our prior estimate of a graphene nanopore's resolution (1). This potential for both excellent resolution and high sensitivity is a consequence of the sharply peaked current density near the tightly fitted molecule's surface in an atomically thin nanopore (Fig. 3B).

The high sensitivity and resolution suggest that a graphene nanopore may be an ideal sensor for nanopore sequencing because of the large absolute values of the ionic current differences

that would discriminate among nucleobases along a strand of DNA. It will therefore be important to test this potential by demonstrating that the results shown here with dsDNA are in fact also found when testing ssDNA with precisely crafted graphene nanopores that closely fit the smaller 1.2- to 1.4-nm diameter of a single DNA strand. We have not yet been able to reliably produce such small pores in graphene using electron beam drilling but are optimistic that the recent proof-of-principle demonstration of an atom-by-atom method to grow graphene nanopores (9) will allow us to do so in the near future.

Reaping the benefits of graphene for achieving improved polymer characterization will also require solutions to several of the same problems that have been addressed by others, using protein nanopores (13, 27): (a) the need for a ratcheting method to control the rate of polymer translocation (28) and suppress Brownian molecular motion (15) and (b) the suppression of large $1/f$ ionic current noise that can severely degrade the ionic current signal-to-noise ratio. It remains to be seen whether new approaches will be required to control polymer translocation through graphene pores (29) or whether the approaches already used with biological pores (27) can also be applied to graphene pores. Likewise, further research will be required to determine how to optimally suppress the large $1/f$ ionic current noise frequently observed in graphene nanopores (Fig. S3). This noise can be reduced by applying various coatings (2, 3), but because any coatings will increase the length of the nanopore, they are not compatible with the high-resolution, high-sensitivity detector strategy reported here. We have consistently achieved noise reduction by simply reducing the free standing area of the graphene membrane surrounding the nanopore (Fig. S3).

The possibility of subnanometer resolution in graphene nanopores (1) together with the high resolution and geometrical sensitivity shown here bodes well for extending the molecular sensing capabilities of solid-state nanopores. In particular we envision the eventual use of graphene nanopores as a robust solid-state alternative to protein pores that have been shown capable of nanopore sequencing (13).

Materials and Methods

The graphene was grown via chemical vapor deposition (CVD) on a copper foil (30), subsequently transferred (31) over the aperture on a free-standing SiNx film covering a silicon chip, and carefully processed to remove any residual hydrocarbon impurities from the polymer support film used during

the transfer process (1). Individual graphene nanopores were drilled with a well-focused electron beam, using a JEOL 2010F EM operating at 200 kV as previously described (1). Micro-Raman measurements (model WITec CRM 200) and EM imaging with atomic resolution (aberration-corrected Zeiss Libra 200, operating at 80 keV electron energies) demonstrated that the transferred graphene film was single layer and contained large areas free of hydrocarbon contaminants. The graphene on its silicon chip was sealed in a fluidic cell to separate two chambers that were subsequently filled with 3 M potassium-chloride solutions, pH 10 or pH 12.5. The graphene nanopore was the only path through which ions and DNA molecules could pass between the two chambers. Ten kilobase-pair restriction fragments of λ -DNA were used for all experiments. Applying a constant 160-mV bias voltage between the chambers with Ag/AgCl electrodes, we measured the ionic current during DNA translocation through the nanopore, using standard electrophysiological methods. All translocation experiments were investigated under the high-salt, high-pH conditions (3 M KCl, pH 10, conductivity = 27.5 S/m), which we have found suppresses dsDNA adsorption on the graphene surface.

To estimate the EM-determined diameter of a nanopore, many of which are not perfectly circular (Fig. 3A), the outline of each nanopore's electron microscopic image was manually traced (as shown, for example, by the curly white line in Fig. S4). The nanopore's area was then calculated by summing the known sample areas represented by each of the pixels enclosed within the manually traced nanopore's perimeter. Finally, D_{EM} was equated to twice the radius of a perfect circle with that determined area (orange circle in Fig. S4). We estimate that because contrast in images of our clean, single-layer graphene is very low, the manual tracing step may have introduced as much as $\pm 5\%$ error in our estimate of nanopore diameter whereas equating the radius with that of a perfectly circular area could have introduced no more than an additional 5% systematic error.

Numerical solutions of the Laplace equation using the measured electrolyte conductivity (27.5 S/m) were performed using the COMSOL Multiphysics finite-element solver in 3D geometry, cylindrically symmetric along the axis of the nanopore. The total ionic current was calculated by integrating current density across the diameter of the nanopore. We also solved the full set of Poisson–Nernst–Planck (PNP) equations, with the boundary conditions at the graphene corresponding to idealized, uncharged membrane impermeable to ions. In our experimental regime with 3 M KCl concentration and 160 mV applied bias, the PNP solution was found to differ by $<3\%$ from the numerical solutions to the Laplace equation. A DNA molecule was modeled as a long stiff insulating rod threading the nanopore along its axis.

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