



OPEN

# From nanocorals to nanorods to nanoflowers nanoarchitecture for efficient dye-sensitized solar cells at relatively low film thickness: All Hydrothermal Process

SUBJECT AREAS:

TWO-DIMENSIONAL  
MATERIALS

SYNTHESIS AND PROCESSING

Received  
23 April 2014Accepted  
5 June 2014Published  
30 June 2014

Correspondence and  
requests for materials  
should be addressed to  
P.S.P. (psp\_phy@  
unishivaji.ac.in) or  
C.K.H. (hongck@  
chonnam.ac.kr)

Sawanta S. Mali<sup>1,4</sup>, Chirayath A. Betty<sup>2</sup>, Popatrao N. Bhosale<sup>3</sup>, Pramod S. Patil<sup>1</sup> & Chang Kook Hong<sup>4</sup>

<sup>1</sup>Thin Film Materials Laboratory, Department of Physics, Shivaji University, Kolhapur, India -416 004, <sup>2</sup>Chemistry Division, Bhabha Atomic Research Center (BARC), Trombay-Mumbai, India -400085, <sup>3</sup>Materials Research Laboratory, Department of Chemistry, Shivaji University, Kolhapur M. S. India -416004, <sup>4</sup>School of Applied Chemical Engineering, Chonnam National University, Gwangju, 500-757 (South Korea).

Simple and low temperature hydrothermal process is employed to synthesize exotic nanostructures of TiO<sub>2</sub>. The nanostructures are obtained merely by changing the nature of the precursors and processing parameters. The chloride and isopropoxide salts of titanium are used to grow high quality thin films comprising anatase nanocorals, rutile nanorods and rutile nanoflowers respectively. A novel route of addition of room temperature ionic liquid (RTIL) is used to synthesize hitherto unexplored nano-morphologies. The Bronsted Acidic Ionic Liquid [BAIL] 0.01 M, 1: 3-ethoxycarbonylethyl-1-methyl-imidazolium chloride [CMIM][HSO<sub>4</sub>] RTIL directed growth of TiO<sub>2</sub> flowers with bunch of aligned nanorods are obtained. The structural, optical and morphological properties of hydrothermally grown TiO<sub>2</sub> samples are studied with the different characterization techniques. The influence of these exotic nano-morphologies on the performance of dye sensitized solar cells (DSSCs) is investigated in detail. It is found that [CMIM][HSO<sub>4</sub>] can facilitate the formation of novel nanoflower morphology with uniform, dense, and collectively aligned in regular petal like oriented TiO<sub>2</sub> nanorods and hence improves the dye adsorption and the photovoltaic performance of DSSCs, typically in short-circuit photocurrent and power conversion efficiency. A best power conversion efficiency of 6.63% has been achieved on a DSSC based on nanoflowers (TNF) film obtained from a [CMIM][HSO<sub>4</sub>] solution.

Dye Sensitized Solar Cells (DSSCs), a molecular approach to photovoltaic solar energy conversion, is one of the emerging solar technologies that offer the potential to reduce the cost of photovoltaic electricity production. During the past two decades, nanoporous polycrystalline titania has been extensively used in (DSSCs), which have been demonstrated to be a promising alternative of silicon based solar cells due to their relatively high solar-to-electric power conversion efficiency and low cost<sup>1</sup>. In the DSSC, the mesoscopic structure of the titanium dioxide (TiO<sub>2</sub>) electrode plays a significant role in increasing cell efficiency by providing the photosensitizer dye with much greater surface area for light harvesting<sup>2</sup>. The mesoporous TiO<sub>2</sub> photoanodes that have the percolated links of the nanoparticles produce very large photocurrent due to their high surface area suitable for dye adsorption and electron energy level matching allowing for the injection of carriers from dye molecules to nanostructured TiO<sub>2</sub> films. Despite this advantage of the nanoparticle-based photoanodes, their numerous interparticle boundaries easily trap the charge carriers, which result in the decrease in the carrier mobility and the carrier lifetime. In the nanoparticle-based DSSCs, therefore, the carrier transport is a trap-limited diffusion process and the electron diffusion coefficient is several orders smaller than the expected value that is deduced from the physical properties of the single crystalline bulk TiO<sub>2</sub><sup>3</sup>. Nanowire/nanorods have demonstrated a great potential to achieve high diffusion coefficient of carriers in electric devices due to their unique one-dimensional (1D) structure<sup>4</sup>. Many attempts have been made for controlling the 1D nanostructured TiO<sub>2</sub> using different techniques which includes electrodeposition<sup>5</sup>, hydrothermal and Solvothermal<sup>6,7</sup>, anodiza-



tion<sup>8</sup>, template based growth sol-gel electrophoresis<sup>9</sup>, and by using different surfactants<sup>10</sup>. Recently, single crystalline TiO<sub>2</sub> nanorods with a rutile phase have been successfully grown on the top of fluorine-doped tin oxide (FTO) films coated glass, by a hydrothermal method<sup>11</sup>. Generally, spin-coated or screen printed Pt/FTO is used as counter electrodes for DSSC application. Recently, Y. H. Hu *et al.* had fabricated variety of counter electrodes like PEDOT<sup>12</sup>, NiO<sup>13</sup>, Graphene<sup>14–16</sup>, ZrN<sup>17</sup>, ZnO<sup>18</sup>.

In recent years, many inorganic nanostructures have been fabricated via various room temperature ionic liquid (RTIL)-involved processes, including electrodeposition, chemoreduction, sol-gel and solvothermal route. RTIL have unique properties such as extremely low volatility, wide liquid temperature range, good thermal stability, good dissolving ability, excellent microwave (MW) absorbing ability, designable structures, high ionic conductivity, and wide electrochemical window, etc<sup>12</sup>. Moreover, RTIL is the excellent surfactant for the growth of nanostructured material. Very recently, RTILs have been used as solvents, reactants, or templates for the synthesis of inorganic nanomaterials with novel morphologies and improved properties<sup>13–15</sup>. Further, recently IL based on imidazolium salts have been widely used as solvents for DSSCs<sup>16</sup>. Many inorganic nanostructures<sup>17</sup>, including titanium dioxide<sup>18–23</sup>, have been fabricated via various ILs-involved processes. For titania nanomaterials, however, to the best of our knowledge, few works about the synthesis of rutile nanostructures have been reported in ILs solution<sup>24–32</sup>. Recently, Kunlun Ding *et al.* successfully developed route for the synthesis of high quality TiO<sub>2</sub> nanocrystals in ionic liquid via a Microwave-Assisted Process<sup>33</sup>.

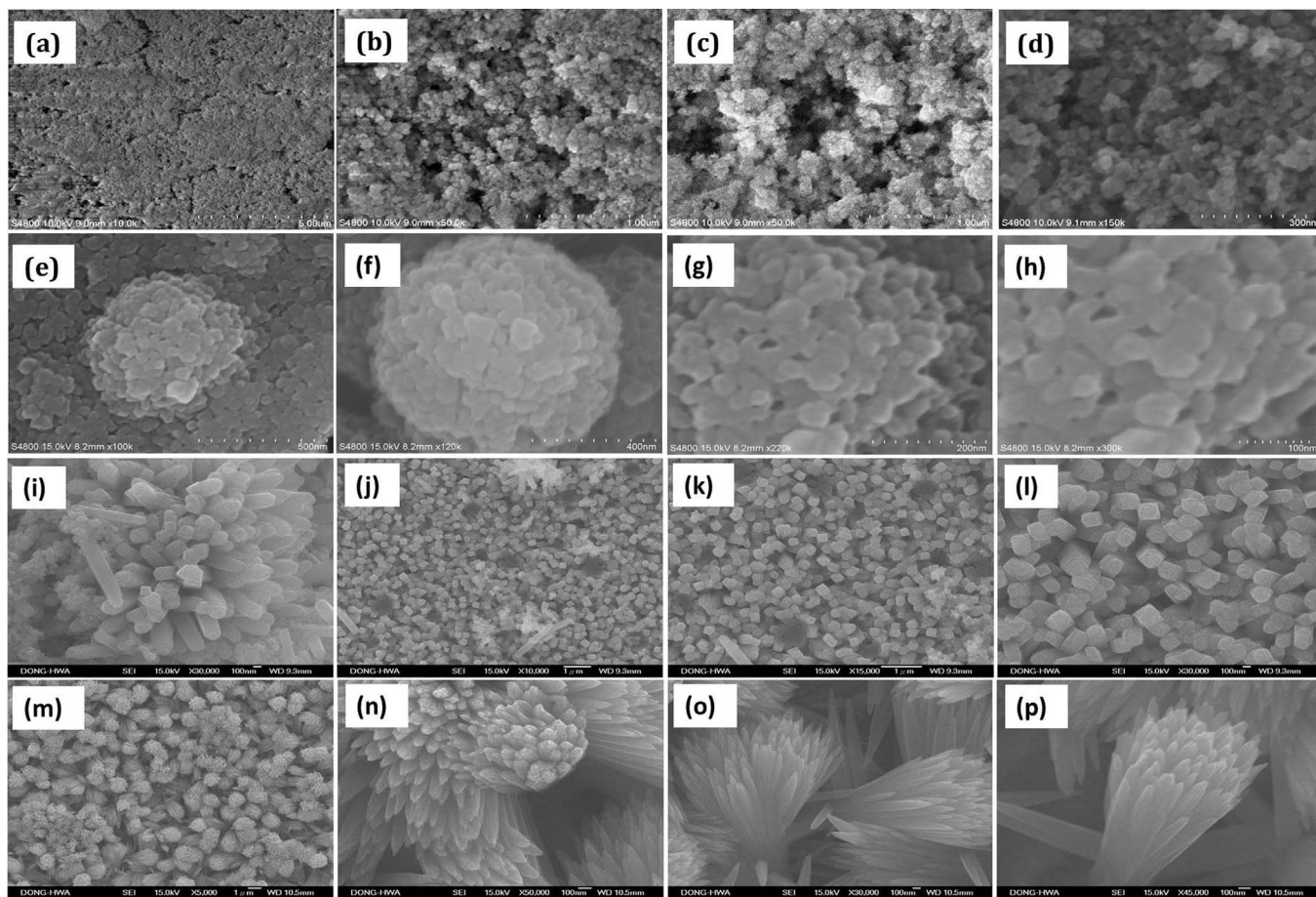
In this article, we have studied effect of different TiO<sub>2</sub> morphology from nanocorals (TNC)<sup>34,35</sup>, nanorods (TNR) to nanoflowers (TNF)<sup>36</sup> thin films on FTO substrate that were prepared by facile hydrothermal route. Herein, TNF film was prepared from 0.001 M of 1:3-ethoxy-carbonylethyl-1-methyl-imidazolium chloride [CMIM][HSO<sub>4</sub>] RTIL mixed with Titanium tetraisopropoxide (TTIP):Hydrochloric Acid (HCl, 38%) (1:1 Volume) solution, and uniform, compressed and nanocrystalline TiO<sub>2</sub> nanoflowers were obtained. These nanostructures of TiO<sub>2</sub> films were further effectively loaded with N719-dye and studied its DSSCs properties. As compared to the TNC, TNR and P25, the TNF prepared from a solution containing the [CMIM][HSO<sub>4</sub>] acids exhibited higher dye loading, resulting in higher short-circuit photocurrent and power conversion efficiency. Our results showed that the TNF sample exhibits higher short-circuit photocurrent density (13.15 mAcm<sup>-2</sup>) and power conversion efficiency (6.63%).

## Results

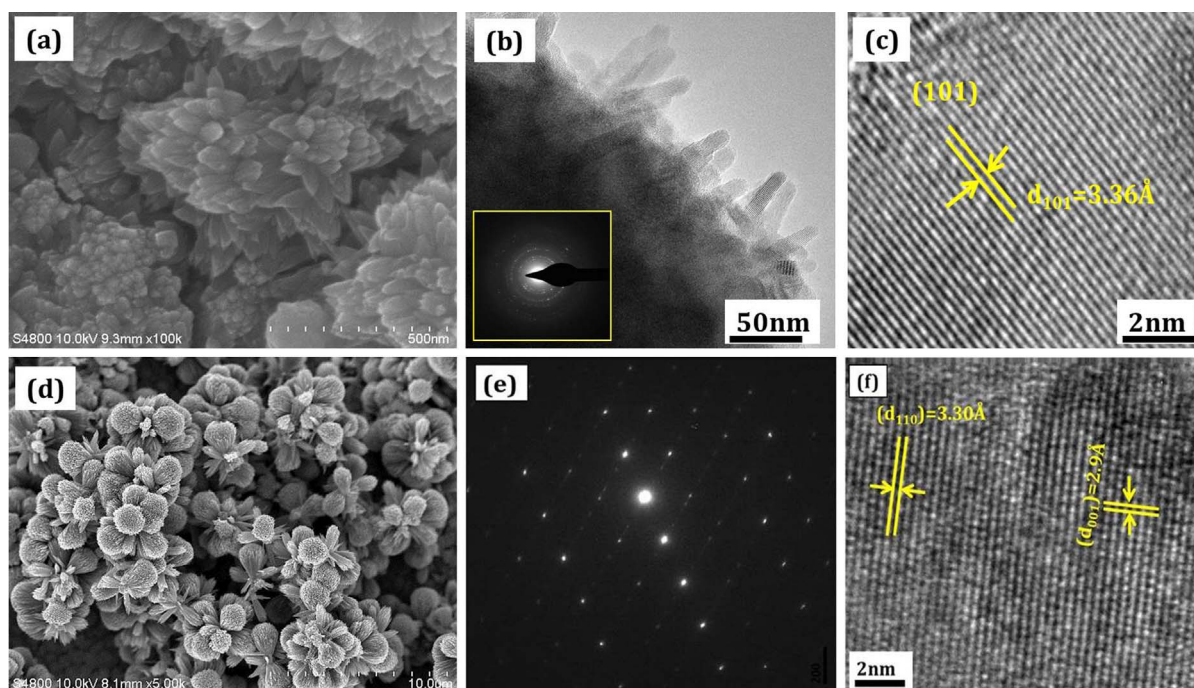
The morphological features of the TiO<sub>2</sub> samples are investigated by Field Emission scanning electron microscopy (FESEM). Fig. 1 (a–d) shows the P25 TiO<sub>2</sub> nanoparticles that were deposited by doctor blade technique. FESEM images show the compactly arranged and uniformly deposited TiO<sub>2</sub> nanoparticles over the substrate. Fig. 1 (e–h) shows FESEM image of the TiO<sub>2</sub> nanocoral (TNC) sample that exhibits a novel coral-like morphology with nanopolyps. The diameter of the circular corals is about 400–500 nm, on top of which nanopolyps are nucleated and their average size is about 20 nm. This peculiar morphological feature is beneficial as it provides larger active surface area. Detailed growth mechanism is explained in our previous work<sup>34–36</sup>. FESEM images of the TiO<sub>2</sub> nanorods (TNR) are shown in fig. 1 (i–l). Most of TNRs displayed microspheres containing nanorods of nearly 250 nm diameter. Fig. 1 (m–p) shows that the TiO<sub>2</sub> nanoflowers (TNF) bunch of aligned nanorods containing petals prepared with RTIL retain the rod like array geometry but the nanorod diameter drastically decreased to nearly 62 nm. It is also to be noted that the nanorods are tapered at the tip and separated to each other.

Fig. 2 (a–c) shows the FESEM, selected area electron diffraction (SAED), TEM and HRTEM pattern of TNC sample. The formation of the ring pattern within the dotted line indicates a highly crystalline sample. The morphology of the as-obtained anatase TiO<sub>2</sub> nanocorals were characterized by TEM, as shown in Fig. 2 (b). The size of each polyps is around 20–30 nm with high crystallinity. Fig. 2(c) shows an HRTEM image of the respective sample. The lattice planes are clearly indicates that the synthesized nanocorals are highly crystalline in nature. The regular periodic arrangement of the atoms to form atomic plane with an interplanar distance of 3.36 Å affirms the growth of anatase TiO<sub>2</sub> phase. Fig. 2 (d–f) shows FESEM, SAED and HRTEM micrographs of TNF sample. The spotted SAED indicates the single-crystalline nature of the rutile TiO<sub>2</sub> nanoflowers. The formation of the dark spot pattern indicates a single crystalline nature of the sample. It is believed that the synthesized films are composed of single-crystalline rutile TiO<sub>2</sub> nanorods that grow in the [001] direction. The clear lattice fringe of the single nanorod of TNF sample is single crystalline along their entire length (fig. 2 (e)). Interplanar spacings,  $d_{110}=3.3\text{\AA}$  between the adjacent lattice fringes perpendicular to the rod axis can be assigned to the interplanar distance of rutile TiO<sub>2</sub> (110) and  $d_{001}=2.9\text{\AA}$  is interplanar distance for the (001) planes along the rod axis that are obtained from the HRTEM lattice fringes of TNF. This indicates the formation of single crystalline rutile phase, in agreement with the XRD results (*Supporting information Figure S1 and S2*).

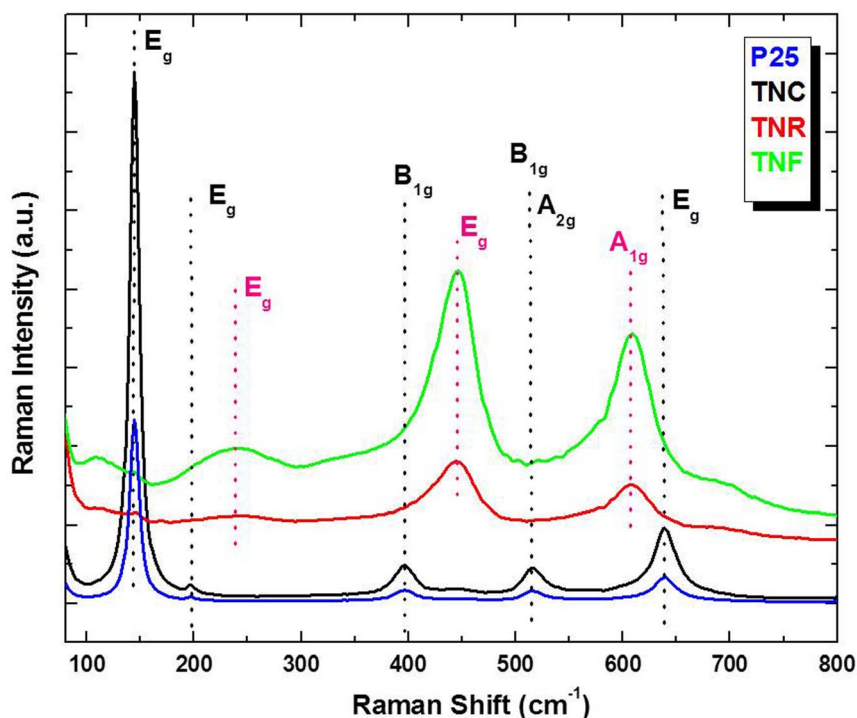
The Fourier transform Raman (FT-Raman) spectra of the films were recorded in the spectral range of 36–2000 cm<sup>-1</sup> using FT-Raman spectrometer (Bruker MultiRAM, Germany) that employs a Nd:YAG laser source with an excitation wavelength 1064 nm and resolution 4 cm<sup>-1</sup> at 336 mW laser power. The FT-Raman of the P25, TNC, TNR and TNF films are shown in Fig. 3. The ‘tetragonal anatase structured’ TiO<sub>2</sub> belongs to D194h (I41/amd) space group (SG) and following normal lattice  $A_{1g}+B_{1g}+B_{2g}+E_g$ . The first  $E_g$  peak at ~144 cm<sup>-1</sup>, a characteristic of anatase TiO<sub>2</sub> was formed in the P25 as well as TNC samples. The peaks at 516 (corresponding to  $B_{1g}, A_{2g}$ ) and 635 cm<sup>-1</sup> (correspond to  $E_g$ ) modes of anatase TiO<sub>2</sub> are observed. The rutile phase of TiO<sub>2</sub> is tetragonal and exhibits symmetry characters of the space group with two TiO<sub>2</sub> molecules per unit cell. The TNR and TNF samples show four Raman active fundamental modes that reveals rutile TiO<sub>2</sub> at 143 cm<sup>-1</sup> ( $B_{1g}$ ), 447 cm<sup>-1</sup> ( $E_g$ ), 612 cm<sup>-1</sup> ( $A_{1g}$ ), and 826 cm<sup>-1</sup> ( $B_{2g}$ ) expressed as  $A_{1g}+B_{1g}+B_{2g}+E_g$ . For the rutile phase two prominent maxima at 445 cm<sup>-1</sup> ( $E_g$ ) and 609 cm<sup>-1</sup> ( $A_{1g}$ ), are comparable with that found in the rutile TiO<sub>2</sub> single crystal. In addition, there are second-order scattering features, the most prominent one at ~237 cm<sup>-1</sup> ( $E_g$ ) peak due to the multiple-phonon scattering processes, which is also considered as a characteristic Raman peak of rutile type TiO<sub>2</sub><sup>27</sup>. In the FT-Raman spectra of our samples, the  $E_g$  and  $A_{1g}$  modes, as well as the second-order effect at ~237 cm<sup>-1</sup>, are the major features; the  $B_{1g}$  and  $B_{2g}$  modes are extremely weak or absent (fig 3). With conversion from TNRs to TNF, the wave number of all peaks decreased relevantly; in particular, the  $E_g$  (612 cm<sup>-1</sup>) mode decreased by about 6 cm<sup>-1</sup>, while the intensity of second-order  $E_g$  mode (~237 cm<sup>-1</sup>) is drastically increased. The second order  $E_g$  at 237 cm<sup>-1</sup>, a characteristic of rutile TiO<sub>2</sub>, is more intense and slightly broader and shifted with respect to that of a TNRs sample. This nature is due to phonon confinement effect in which the decrease in the particle dimensions to the nanometer scale can cause a wave number shift and broadening of Raman peaks as a result of phonon confinement. The non-linear behavior of the  $E_g$  peak intensity suggests strong light scattering differences among samples, potentially related to changes on the morphological properties of the nanorods and nanoflowers. Figure S4 shows the N2 adsorption-desorption isotherm of the TiO<sub>2</sub> samples. The isotherm displays a typical type IV curve while the hysteresis loop is associated with narrow slit-like pores in the sample. The BET specific surface area of the P25, TNC, TNR and TNF sample



**Figure 1** | FESEM images of hierarchical  $\text{TiO}_2$  nanostructures: (a–d) Degussa P25 nanoparticles deposited by doctor blade, (sample-1). (e–h)  $\text{TiO}_2$  nanocorals (TNC) deposited by hydrothermal process from saturated NaCl solvent having ( $\text{TiCl}_4$ ) at  $120^\circ\text{C}$ , solutions (sample-2), (i–l)  $\text{TiO}_2$  nanorods (TNR) deposited by hydrothermal process from TTIP in  $\text{HCl}:\text{H}_2\text{O}$  (1:1v) at  $180^\circ\text{C}$  (sample-3) and (m–p)  $\text{TiO}_2$  nanofloweres (TNF) obtained by hydrothermal process in  $\text{TTIP}:\text{HCl}:\text{H}_2\text{O}:[\text{CMIM}][\text{HSO}_4]$  ionic liquid (sample-4).



**Figure 2** | High-magnification FESEM images, SAED and HRTEM characterizations of the of the different  $\text{TiO}_2$  nanostructures in the form of (a–c) TNC, (d–f) TNF.



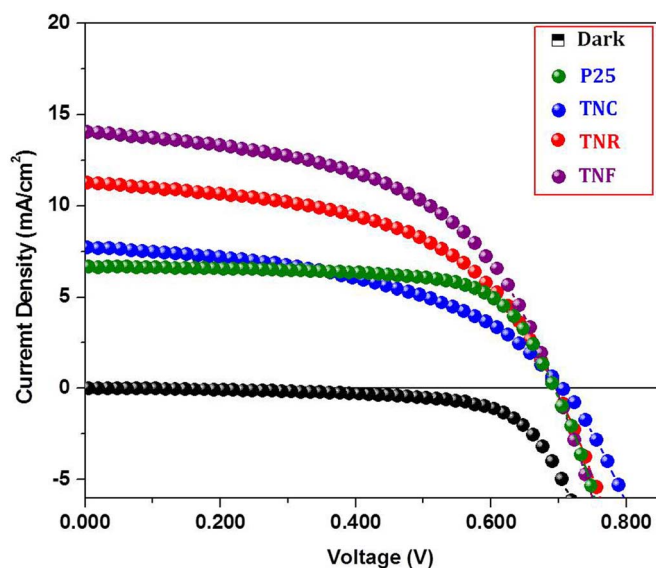
**Figure 3** | FT-Raman spectra of synthesized TNC, TNR and TNF recorded in the wave number range of 35–800  $\text{cm}^{-1}$  with 1064 nm Nd:YAG laser excitation wavelength.

are  $54.7 \text{ m}^2\text{g}^{-1}$ ,  $73.5 \text{ m}^2\text{g}^{-1}$ ,  $89.4 \text{ m}^2\text{g}^{-1}$  and  $106.1 \text{ m}^2\text{g}^{-1}$  respectively. From these data it is clear that the TNF sample has much higher surface area than other  $\text{TiO}_2$  photoelectrodes. These results clearly reveal that  $\text{TiO}_2$  nanoflowers facilitates larger amount of dye adsorption as well as effective light harvesting.

The DSSC devices assembled using the above films were tested under simulated AM1.5G solar illumination ( $100 \text{ mWcm}^{-2}$ ). The 0.3 mM ethanolic solution of N-719 (Di-tetrabutylammonium cis-bis(isothiocyanato)bis(2,2'-bipyridyl-4,4'-dicarboxylato)ruthenium (II), Sigma Aldrich) dye solution was used for dye loading. The standard iodide-based electrolyte (TEL-294H, EL-HPE, Dyesol) was used as the redox electrolyte. Figure 4 shows the J-V plots of different  $\text{TiO}_2$  morphologies denoted as Dark (●) P25 (●), TNC (●), TNR (●) and TNF (●). The average photovoltaic data were summarized in Table 1. The DSSC based on P25 produced conversion efficiency ( $\eta$ ) of 3.18% ( $J_{\text{sc}} = 6.68 \text{ mAcm}^{-2}$ ,  $V_{\text{oc}} = 0.692 \text{ V}$ ,  $\text{FF} = 0.69$ ). The TNC sample shows  $\eta$  of 3.53% ( $J_{\text{sc}} = 8.63 \text{ mAcm}^{-2}$ ,  $V_{\text{oc}} = 0.706 \text{ V}$ ,  $\text{FF} = 0.58$ ). In case of  $\text{TiO}_2$  nanorods (TNR), it shows  $J_{\text{sc}} = 11.24 \text{ mAcm}^{-2}$ ,  $V_{\text{oc}} = 0.699 \text{ V}$ ,  $\text{FF} = 0.62$ . The conversion efficiency of TNR comes to be 4.87%. While, the nanoflowers grown from RTIL [CMIM][ $\text{HSO}_4$ ] resulted in a big enhancement of  $J_{\text{sc}}$  by twice compared to P25. The TNF sample shows  $\eta$  of 6.63% ( $J_{\text{sc}} = 13.15 \text{ mAcm}^{-2}$ ,  $V_{\text{oc}} = 0.699 \text{ V}$ ,  $\text{FF} = 0.67$ ). As a result, power conversion efficiency improved from 3.18% for P25 sample to 6.63% for TNF.

Using the same dye and redox electrolyte, a DSSC made of a film comprising 20 nm  $\text{TiO}_2$  anatase nanoparticles produced 3.18% conversion efficiency ( $J_{\text{sc}} = 6.68 \text{ mAcm}^{-2}$ ,  $V_{\text{oc}} = 0.692 \text{ V}$ ,  $\text{FF} = 0.69$ ). From Table 1 it is obvious that the nanoflower rutile film is superior to the P25 nanoparticulate anatase film in the generation of  $J_{\text{sc}}$  and  $\eta\%$  even though anatase is known to show better efficiency. In order to verify this, the adsorbed amounts of dye were determined by measuring the eluted dye concentration from the porous  $\text{TiO}_2$  structure. With UV-vis (UV-1800, Shimadzu) absorption spectroscopy P25 sample shows  $89.1 \mu\text{mol}\cdot\text{cm}^{-2}$  which is less as compared to TNC, TNR and TNF samples. The amount of dye loaded by TNF

is higher ( $129.9 \mu\text{mol}\cdot\text{cm}^{-2}$ ) compared to both TNC ( $97.5 \mu\text{mol}\cdot\text{cm}^{-2}$ ) and TNR ( $115.3 \mu\text{mol}\cdot\text{cm}^{-2}$ ) sample. From the above discussion it is clear that the increment in dye adsorption was simply proportional to the surface area of the  $\text{TiO}_2$  morphology. Therefore TNF sample provides large surface area for dye adsorption which results in increased efficiency. From Table 1 it is clear that all samples shows FF less than 0.70, which may be due to high series resistance ( $R_s$ ). In our experiment we used FTO with sheet resistance  $15 \Omega\cdot\text{cm}^{-2}$  which will cause the increment in the series resistance.



**Figure 4** | J-V curves of the DSSCs assembled respectively from P25, TNC, TNR and TNF under simulated AM 1.5G solar light ( $100 \text{ mW cm}^{-2}$ ). P25 film obtained from commercial Degussa powder, TNC obtained from the  $\text{TiCl}_4$  and sat. NaCl; TNR obtained from the TTIP:HCl:  $\text{H}_2\text{O}$ ; TNF obtained from the TTIP:HCl: $\text{H}_2\text{O}$ : [CMIM][ $\text{HSO}_4$ ].

Table 1 | Average photovoltaic performance parameters for DSSCs based on the different TiO<sub>2</sub> morphologies

Sample	Voc (V)	Jsc (mAcm <sup>-2</sup> ) <sup>[a]</sup>	R	R	FF (%)	(η) (%)	Adsorbed Dye (μmol cm <sup>-2</sup> )	Surface Area m <sup>2</sup> g <sup>-1</sup>
			at Voc (Ω·cm <sup>2</sup> )	at Isc (Ω·cm <sup>2</sup> )				
<b>P25</b>	0.692	6.68	183	3152.9	69	3.18	89.1	54.7
<b>TNC</b>	0.706	8.63	78.1	2018.3	58	3.53	97.5	73.5
<b>TNR</b>	0.699	11.24	74.2	2129.5	62	4.87	115.3	89.4
<b>TNF</b>	0.699	13.15	45.81	809.7	67	6.63	129.9	106.1

[a] Device measured @ 0.5 × 0.5 cm active area illumination.

Open-circuit voltage-decay (OCVD) measurements were conducted to investigate the recombination kinetics of these nanostructures. Following the technique as reported by Zaban et al.<sup>37</sup> Open-circuit voltage-decay measurements were performed by monitoring the Voc transient during relaxation from an illuminated quasi equilibrium state to the dark equilibrium (Figure 5 (a)). When the visible light illumination on a DSSC at open circuit is interrupted, the excess electrons are removed due to recombination, with the photo-voltage decay rate directly related to the electron lifetime.

Appropriate use of OCVD equation, assumes that the recombination is linear with a first-order dependence on electron concentration and that electron recombination occurs only with the electrolyte. Figure 5 shows the OCVD curves measured for various DSSCs, and it appears that the TNF has the slower voltage decay rate as compared to other samples. Figure 5 (b) is the plot of the response time obtained by applying equation to the data in Figure 5 (a). Recombination in all samples is very slow and cells can sustain significant voltages many seconds after light has been turned off. In comparison to P25, TNC, TNR and TNF OCVD measurements, the TNF based DSSCs exhibit the much larger  $\tau_n$  values compared to all other samples. The derived electron lifetimes from transient equation, also demonstrate that the TNF electrode has larger  $\tau_n$  values compared with those for the other DSSCs electrodes. This result indicates that the recombination rate of photogenerated electrons with electrolyte-oxidized species is the slowest in the DSSCs fabricated based on TNF morphology. This effect is also responsible for the higher performance of the DSSCs electrode.

In order to understand the reasons behind such results, we decided to perform electrochemical impedance spectroscopy (EIS) on the assembled cells in the dark at -0.70 V applied potential. Analyzing EIS results using the appropriate transmission line model, gives insights into the interfacial charge transfer processes that take place at the working and counter electrodes/electrolyte interfaces. Typical EIS Nyquist plots of TiO<sub>2</sub> DSSCs devices (P25, TNC, TNR and TNF) and their corresponding equivalent circuit fitted plots are shown in Figure 6 (a–d). Inset shows its equivalent circuit used to analyze the impedance spectra. In general, under an applied potential of -0.70 V three phase angle peaks are present. The peak seen in the high frequency region (20 kHz–100 Hz) results from the charge transfer resistance ( $R_{CE}$ ) and the double layer capacitance ( $C_{CE}$ ) at the electrolyte/counter electrode interface, whereas the one observed at mid frequencies (1–100 Hz) is due to the charge transfer resistance ( $R_{CT}$ ) of the electron recombination and the chemical capacitance ( $C_1$ ) at the TiO<sub>2</sub>/N719/electrolyte interface.

The third arc at low frequencies (0.1–1 Hz) is characteristic of the Warburg impedance ( $W_1$ ) corresponding to the electrolyte's ions diffusion in a DSSC as seen in the case of TNC sample. The fitted impedance spectra for TNC sample consists of ( $R_3$ ,  $Q_2$ ) which represents the electrode/electrolyte interface and  $R_1$  is the shunt resistance in parallel with the TiO<sub>2</sub>-Dye structure represented by  $W_1$ ,  $R_2$ ,  $C_1$ .  $R_3$  is the charge transfer resistance,  $R_{CE}$  and  $Q_2$  is the double layer capacitance  $C_{CE}$ . While  $W_1$  and  $R_2$  (or  $R_{CT}$ ) represent the diffusion resistance and series resistance of the TiO<sub>2</sub>-Dye structure,  $C_1$  repre-

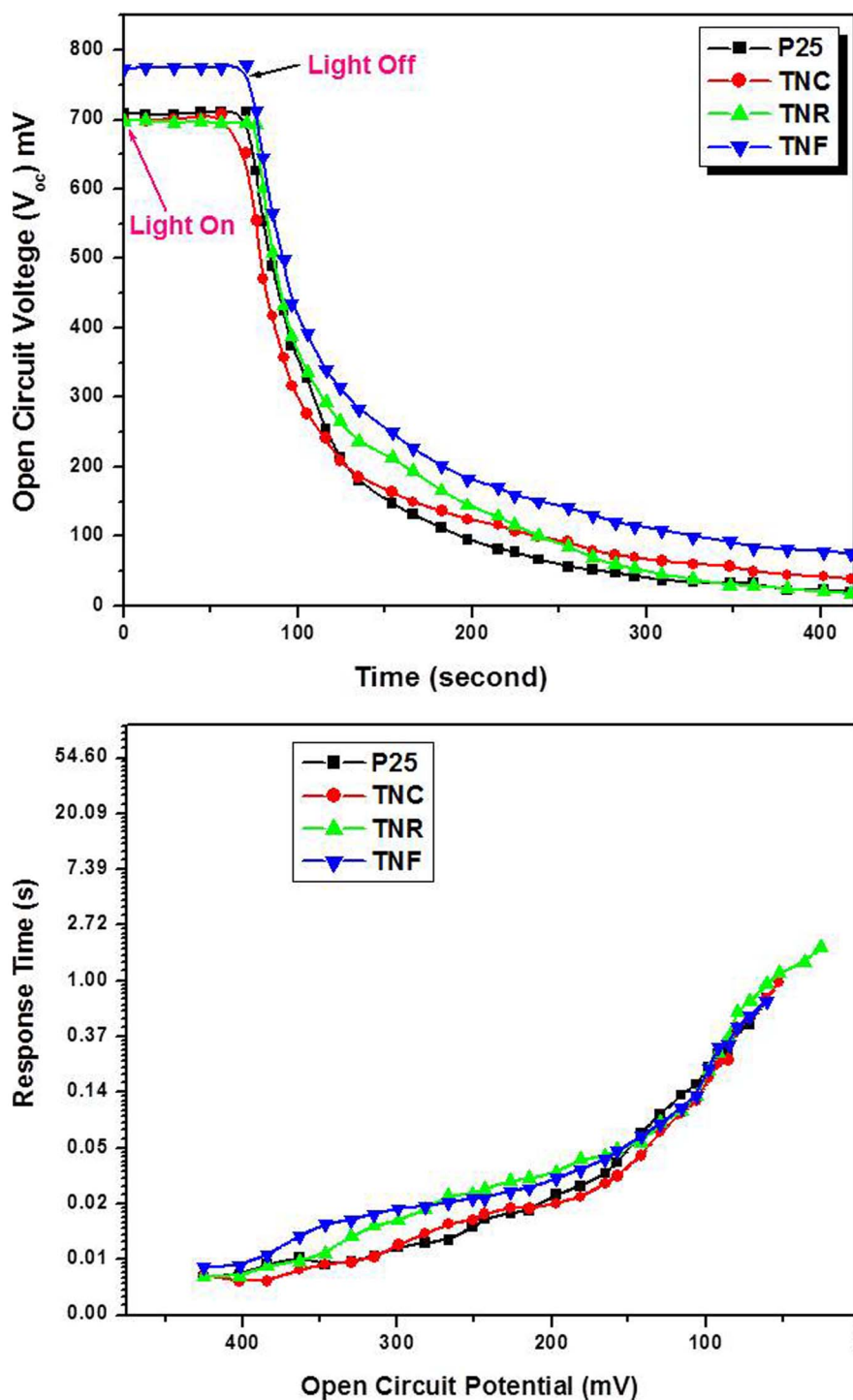
sents the TiO<sub>2</sub>-Dye junction capacitance. In case of nanorods, the  $W_1$  gets replaced by the constant phase element  $Q_0$ , while the remaining circuit elements are the same. For P25 and TNF the equivalent circuits are similar as shown in the inset. The product of the charge-transfer resistance of the electron recombination ( $R_{CT}$  or  $R_2$  in the case of TNC and TNR while it is  $R_1$  in the case of P25 and TNF) and the chemical capacitance ( $Q_1$  or  $C_1$ ) at the TiO<sub>2</sub>/N719 interface corresponds to the electron lifetime in the TiO<sub>2</sub> film,  $\tau_e = R_{CT} \times Q_1$ .

The series and shunt resistances obtained after fitting the impedance data are shown in Table 2 along with the carrier lifetime estimates. From the Table-2 it is clear that, there is a significant increase in the carrier lifetime for all TiO<sub>2</sub> nanostructures prepared by hydrothermal method when compared to P25. The shunt resistance  $R_1$  of device P25 and TNR is very high than that of  $R_2=2531$  and  $23 \Omega \cdot \text{cm}^{-2}$  for devices TNC and TNF measured at -0.70 V, respectively. Though the series resistance ( $R_2$ ) of TNF is very less, the shunt resistance also is very small decreasing the efficiency. So better optimization of the TiO<sub>2</sub>-Dye interface promises further increase in efficiency. Higher  $R_{CT}$  for TNC electrode results in lower fill factor, thereby lowering efficiency when compared to TNF based device.

For TNF sample increased Jsc may be due to typical morphology of TiO<sub>2</sub> sample which will facilitate for effective dye adsorption and the corresponding increase in current. Such morphology will also provide effective light harvesting accessibility. Further, Jsc increased significantly because the size and width of nanoflowers are thinner about (~61 nm) compared to TNR (~250 nm) sample, denser and more aligned, which yielded more dye loading and faster electron diffusion. Among the above-mentioned morphologies of nanoflowers (TNF), those grown from the TTIP:HCl:[CMIM][HSO<sub>4</sub>] solution produced the highest short-circuit photocurrent (13.15 mAcm<sup>-2</sup>) and power conversion efficiency (6.63%). Therefore, we suggest that TNF are the best candidate for DSSCs application.

## Discussion

In conclusion, we have presented first report on a synthetic strategy to deposit different nanostructured TiO<sub>2</sub> thin films by simple and cost effective hydrothermal route for dye sensitized solar cells (DSSCs) application. We have also demonstrated a facile way to synthesize novel nanoflower morphology of TiO<sub>2</sub> using [CMIM][HSO<sub>4</sub>] room temperature ionic liquid (RTIL) which is non-toxic and are useful in developing green chemistry approach for efficient DSSCs applications. Furthermore, [CMIM][HSO<sub>4</sub>] helped to achieve a high degree of crystallization even at mild annealing temperatures, such as 180°C. Among the above-mentioned morphologies TiO<sub>2</sub> nanoflowers (TNF), those grown from the TTIP:HCl:[CMIM][HSO<sub>4</sub>] solution produced the highest short-circuit photocurrent (13.15 mAcm<sup>-2</sup>) and power conversion efficiency (6.63%). The TNFs are superior to those grown from the single step hydrothermal system in solar cell performance, typically with current density Jsc (13.15 mAcm<sup>-2</sup>). Even with a 1.8~μm long nanorod array film, a solar-to-electric power conversion efficiency of 6.63% has been achieved, which is more efficient than a DSSC based on a 1.8 μm thick nanoparticulate film. This method will open a new



**Figure 5** | (a) Photovoltage-decay measurement of  $\text{TiO}_2$  DSSCs electrodes to illumination excitation (AM 1.5G solar light ( $100 \text{ mWcm}^{-2}$ )), (b) Response time determined by open circuit photovoltage decay for different DSSCs samples.

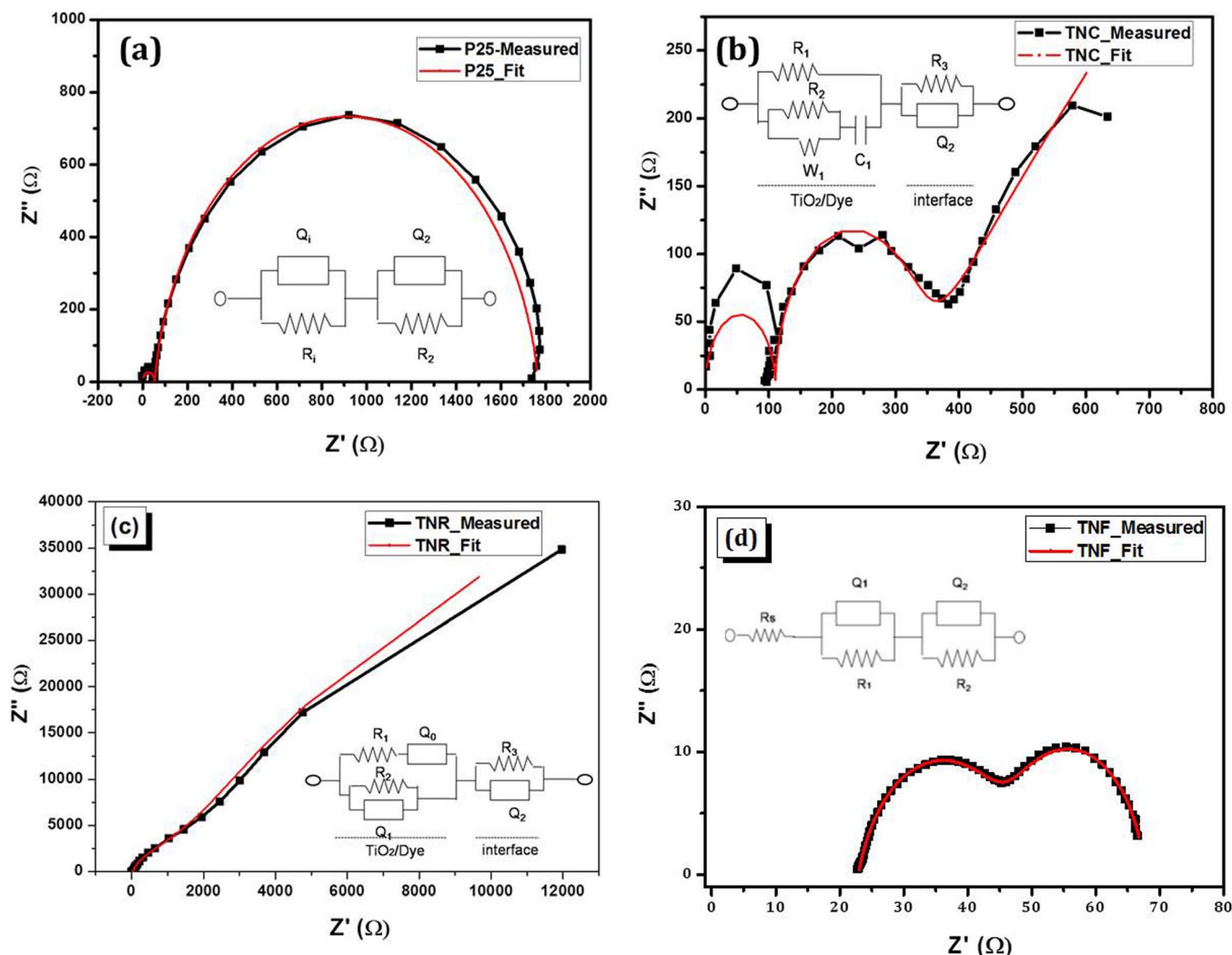
approach to develop hierarchical functional nanomaterials using ionic liquid and subsequent applications.

## Methods

**Materials.** Titanium tetrachloride ( $\text{TiCl}_4$ ) (99.98% Spectrochem, India), and Titanium tetraisopropoxide (TTIP) were purchased from Alfa Aesar. Hydrochloric acid (HCl, 35.46%), sulfuric acid ( $\text{H}_2\text{SO}_4$ , 99.9%), nitric acid ( $\text{HNO}_3$ , 38%), and sodium chloride (NaCl) were obtained from Thomas Baker. Absolute ethanol (99.9%) AR-grade was obtained from Changshu Yangyuan Chemicals China. Conducting FTO glass (fluorine-doped  $\text{SnO}_2$ , resistance  $15 \Omega \text{ cm}^{-2}$ , transmittance 90%, Solaronix) was received from Solaronix, Switzerland.

Iodolyte AN-50 (Solaronix) redox electrolyte were used in the DSSCs. The sensitizer used in this work is an organic N719 dye (Di-Tetrabutylammonium cis-bis(isothiocyanato)bis(2,2-bipyridyl-4,4'-dicarboxylato)-ruthenium(II), 95% (NMR)) purchased from Dysol.

**Hydrothermal Synthesis of  $\text{TiO}_2$  Nanocorals (TNC).**  $\text{TiO}_2$  nanocorals (TNC) were synthesized by controlled hydrothermal process<sup>34,35</sup>. In a typical experiment, hydrothermal inorganic precursor solution was prepared by mixing 0.05 M  $\text{TiCl}_4$  carefully with absolute ethanol in ice-cold bath forming yellow colored titanium chloroalkoxide [ $\text{TiCl}_2(\text{OEt})_2$ ] solution. A small amount of 0.001 M concentrated hydrochloric acid (HCl, 36%, Thomas Baker) was added as a catalyst and pH of solution was adjusted to 2.1 using nitric acid ( $\text{HNO}_3$ , 38%, Thomas Beaker) as a



**Figure 6** | (a–d) Impedance spectra of  $\text{TiO}_2$  DSSCs electrodes measured in the dark under 20.70 V applied bias. Inset shows the respective equivalent circuits used to analyze the impedance data.

peptizer. The transparent solution thus prepared was further mixed with saturated solution of 6.8 M NaCl (Extra pure, Thomas Baker). The addition of water causes the hydrolysis of the inorganic moieties, producing in situ ethanol and HCl, the latter being the reason for the high stability of the solution. The final solutions are very acidic and stable for several months at ambient temperature (20–25 °C) or 1–2 weeks at 30 °C. The solution was stirred for 30 min using magnetic stirrer and transferred into a polytetrafluoroethylene (Teflon)-coated 25 ml vessel. The cleaned glass or TCO substrate was immersed into this solution vertically. It was then sealed and maintained at 120 °C for 3 h (at 1.5 kg/cm<sup>2</sup> pressure). Upon completion of the reaction, the autoclave was allowed to cool at room temperature and the deposited film was rinsed in double distilled water and finally dried at room temperature. The  $\text{TiO}_2$  film was whitish, uniform, crack-free and adherent to the substrates, whose adherence was checked by a scotch tape test. The deposited film is designated as TNC.

**Hydrothermal Synthesis of  $\text{TiO}_2$  Nanorods (TNR).** A  $\text{TiO}_2$  TNR on conducting glass electrodes was prepared by the hydrothermal method. In a typical synthesis, titanium (IV) isopropoxide (0.5 mL) was added to equal volume of distilled water and concentrated HCl. The resulting mixture was stirred for 30 min. The clear transparent solution was then transferred to a Teflon-lined stainless steel autoclave (25 mL), and a piece of F-doped  $\text{SnO}_2$  (FTO) glass (Solaronix, 15  $\Omega\cdot\text{cm}^{-2}$ ) was immersed in the solution parallel along the wall. The autoclave was sealed and placed in an oven at 180 °C for 3 h. After cooling the autoclave to room temperature, the  $\text{TiO}_2$  films were washed several times with distilled water and dried in oven at 70 °C for 1 h designated as TNR.

**Hydrothermal Synthesis of  $\text{TiO}_2$  Nanoflowers (TNF).**  $\text{TiO}_2$  nanoflowers (TNF) on conducting glass electrodes were prepared by the hydrothermal method as per our previous method<sup>36</sup>. The  $\text{TiO}_2$  nanoflowers (TNF) were synthesized by the addition of 0.001 M, 1: 3-ethoxycarbonyl-ethyl-1-methyl-imidazolium chloride [CMIM][HSO<sub>4</sub>] RTIL in above precursor solution and carried out the same reaction at same 180 °C temperature for 3 h and the sample was designated as TNF. For comparison we used P25 Degussa  $\text{TiO}_2$  nanoparticles and 1.8  $\mu\text{m}$  thick film was fabricated by using well known doctor blade technique. Briefly, 1 g  $\text{TiO}_2$  nanoparticles paste was prepared using available literature and film was deposited on FTO coated glass substrate and annealed at 450 °C for 5 min.

**Device fabrication and dye sensitized solar cell tests.** The DSSCs properties were measured by Sol2A Oriel New Port Corporation USA, with Keithley-2420 source meter under 1.5 AM illumination. A compact and sealed dye sensitized solar cell (DSSCs) device was fabricated using a standard two-electrode configuration, comprising dye loaded Glass/FTO/P25 (with an active surface area of 0.5 cm<sup>2</sup>) as photoanode, platinum coated FTO, as a counter electrode which is sealed with the working electrode using a spacer (1 mm) of polyacralamyde. The platinum coated FTO counter electrode, was prepared by dropping a  $\text{H}_2\text{PtCl}_6$  solution (0.7 mM) on an FTO glass followed by heating at 400 °C for 20 min in air. The dye-adsorbed

**Table 2** | Fitted impedance parameters of the DSSC's for different devices

Device	$Q_1$ (F)	$R_2$ $\Omega\cdot\text{cm}^{2[a]}$	$R_1$ $\Omega\cdot\text{cm}^2$	$\tau_e \pm 17\%$ (ms)
P25	(1.25e-8) <sup>1</sup>	53	150000	0.00066
TNC	(1.033e-5) <sup>1</sup>	238	2246	0.245
TNR	(1.495e-5) <sup>0.6562</sup>	2531	228900	0.112
TNF	(5.753e-5) <sup>0.8007</sup>	23	23.8	0.116

[a] Measured under dark conditions with an applied potential = -0.70 V and a cell area = 0.5 × 0.51 cm<sup>2</sup>.



nanostructured TiO<sub>2</sub> electrodes and the Pt counter electrode was sealed with 25 μm-thick Surlyn (SX1170–25, Solaronix) thermoplastic at a pressure of ~210 Kpacm<sup>-2</sup> and a temperature of about ~110°C. The Iodolyte AN-50 (Solaronix) electrolyte was inserted through predrilled hole from Pt/FTO counter electrode. Finally device was sealed using thermoplastic. All the measurements were carried out at room temperature in air.

**Dye Adsorption Measurement.** To measure the adsorbed dye concentration on the TiO<sub>2</sub> electrode surface, the dye loaded TiO<sub>2</sub> samples were desorbed by a 0.1 M NaOH solution in water and ethanol (50 : 50, v/v) for 10 min. The dimension of the TiO<sub>2</sub> electrode was 1 cm<sup>2</sup>. The adsorbed dye was quantitatively determined from the absorbance at 510 nm measured by UV-1800, Shimadzu UV-vis spectrophotometer (Resolution 0.01 nm).

For more details of effective dye loading, characterization techniques used for TiO<sub>2</sub> samples, device fabrication and dye sensitized solar cell tests, XRD spectra of TiO<sub>2</sub> samples refer Fig. S1 and Fig S2, for FT-IR spectra of TiO<sub>2</sub> samples refer Fig. S3, and for Nitrogen adsorption-desorption isotherms plots of TiO<sub>2</sub> samples refer to Fig. S4.

- Grätzel, M. Photoelectrochemical cells. *Nature* **414**, 338–344 (2001).
- Yum, J. H. *et al.* Phosphorescent energy relay dye for improved light harvesting response in liquid dye-sensitized solar cells. *Energy Environ. Sci.* **3**, 434–437 (2010).
- Wang, H. E. *et al.* Hydrothermal synthesis of ordered single-crystalline rutile TiO<sub>2</sub> nanorod arrays on different substrates. *Appl. Phys. Lett.* **96**, 263104–263107 (2010).
- Law, M., Greene, L. E., Johnson, J. C., Saykally, R. & Yang, P. D. Nanowire dye-sensitized solar cells. *Nat. Mater.* **4**, 455–459 (2005).
- Kang, S. H., Kim, J. Y., Kim, Y., Kim, H. S. & Sung, Y. E. Surface Modification of Stretched TiO<sub>2</sub> Nanotubes for Solid-State Dye-Sensitized Solar Cells. *J. Phys. Chem. C* **111**, 9614–9623 (2007).
- Kasuga, T., Hiramatsu, M., Hoson, A. & Niihara, T. S. K. Titania Nanotubes Prepared by Chemical Processing. *Adv. Mater.* **11**, 1307–1311 (1995).
- Mali, S. S., Betty, C. A., Bhosale, P. N. & Patil, P. S. Hydrothermal synthesis of rutile TiO<sub>2</sub> with hierarchical microspheres and their characterization. *Cryst. Engg. Comm.* **13**, 6349–6351 (2011).
- Gong, D. *et al.* Titanium oxide nanotube arrays prepared by anodic oxidation. *J. Mater. Res.* **16**, 3331–3334 (2001).
- Ren, X. *et al.* The selective fabrication of large-area highly ordered TiO<sub>2</sub> nanorod and nanotube arrays on conductive transparent substrates via sol-gel electrophoresis. *Nanotechnology* **20**, 365604–365608 (2009).
- Chu, S. Z., Wada, K., Inoue, S., Hishita, S. I. & Kura-shima, K. Fabrication and Structural Characteristics of Ordered TiO<sub>2</sub>-Ru(-RuO<sub>2</sub>) Nanorods in Porous Anodic Alumina Films on ITO/Glass Substrate. *J. Phys. Chem. B* **107**, 10180–10184 (2003).
- Liu, B. & Aydil, E. S. Growth of Oriented Single-Crystalline Rutile TiO<sub>2</sub> Nanorods on Transparent Conducting Substrates for Dye-Sensitized Solar Cells. *J. Am. Chem. Soc.* **131**, 3985–3990 (2009).
- Wei, W., Wang, H. & Hu, Y. H. A review on PEDOT-based counter electrodes for dye-sensitized solar cells. *Int. J. Energy Res. DOI:10.1002/er.3178* (2014).
- Wang, H., Wei, W. & Hu, Y. H. NiO as an Efficient Counter Electrode Catalyst for Dye-Sensitized Solar Cells. *Top Catal.* **57**, 607–611 (2014).
- Wang, H., Sun, K., Tao, F., Stacchiola, D. J. & Hu, Y. H. 3D Honeycomb-Like Structured Graphene and Its High Efficiency as a Counter-Electrode Catalyst for Dye-Sensitized Solar Cells. *Angew. Chem. Int. Ed.* **52**, 9210–9214 (2013).
- Wang, H. & Hu, Y. H. Graphene as a counter electrode material for dye-sensitized solar cells. *Energy Environ. Sci.* **5**, 8182–8188 (2012).
- Wang, H., Leonard, S. L. & Hu, Y. H. Promoting Effect of Graphene on Dye-Sensitized Solar Cells. *Ind. Eng. Chem. Res.* **51**, 10613–10620 (2012).
- Wei, W., Wang, H. & Hu, Y. H. Unusual particle-size-induced promoter-to-poison transition of ZrN in counter electrodes for dye-sensitized solar cells. *J. Mater. Chem. A* **1**, 14350–14357 (2013).
- Wang, H., Wei, W. & Hu, Y. H. Efficient ZnO-based counter electrodes for dye-sensitized solar cells. *J. Mater. Chem. A* **1**, 6622–6628 (2013).
- Duan, X., Lian, J., Ma, J., Kim, T. & Zheng, W. Shape-Controlled Synthesis of Metal Carbonate Nanostructure via Ionic Liquid-Assisted Hydrothermal Route: The Case of Manganese Carbonate. *Cry. Growth Design* **10**, 4449–4455 (2010).
- Parnham, E. R. & Morris, R. E. Ionothermal synthesis using a hydrophobic ionic liquid as solvent in the preparation of a novel aluminophosphate chain structure. *J. Mater. Chem.* **16**, 3682–3684 (2006).
- Taubert, A. CuCl Nanoplatelets from an Ionic Liquid-Crystal Precursor. *Angew. Chem., Int. Ed.* **43**, 5380–5382 (2009).
- Parnham, E. R. & Morris, R. E. Ionothermal Synthesis of Zeolites, Metal-Organic Frameworks, and Inorganic-Organic Hybrids. *Acc. Chem. Res.* **40**, 1005–1013 (2007).
- Yoo, K., Choi, H. & Dionysiou, D. D. Ionic liquid assisted preparation of nanostructured TiO<sub>2</sub> particles. *Chem. Commun.* 2000–2001 (2004).
- Cooper, E. R. *et al.* Ionic liquids and eutectic mixtures as solvent and template in synthesis of zeolite analogues. *Nature* **430**, 1012–1016 (2004).
- Zhou, Y. & Antonietti, M. Synthesis of Very Small TiO<sub>2</sub> Nanocrystals in a Room-Temperature Ionic Liquid and Their Self-Assembly toward Mesoporous Spherical Aggregates. *J. Am. Chem. Soc.* **125**, 14960–14961 (2003).
- Ding, K. L. *et al.* Facile Synthesis of High Quality TiO<sub>2</sub> Nanocrystals in Ionic Liquid via a Microwave-Assisted Process. *J. Am. Chem. Soc.* **129**, 6362–6363 (2007).
- Yoo, K. S., Choi, H. & Dionysiou, D. D. Synthesis of anatase nanostructured TiO<sub>2</sub> particles at low temperature using ionic liquid for photocatalysis. *Catal. Commun.* **6**, 259–262 (2005).
- Liu, Y. *et al.* Preparation and Properties of Nanostructure Anatase TiO<sub>2</sub> Monoliths Using 1-Butyl-3-methylimidazolium Tetrafluoroborate Room-Temperature Ionic Liquids as Template Solvents. *Cryst. Growth Des.* **5**, 1643–1649 (2005).
- Nakashima, T. & Kimizuka, N. Interfacial Synthesis of Hollow TiO<sub>2</sub> Microspheres in Ionic Liquids. *J. Am. Chem. Soc.* **125**, 6386–6387 (2003).
- Welton, T. Room-Temperature Ionic Liquids. Solvents for Synthesis and Catalysis. *Chem. Rev.* **99**, 2071–2084 (1999).
- Yu, N., Gong, L., Song, H., Liu, Y. & Yin, D. Ionic liquid of [Bmim]<sup>+</sup> Cl<sup>-</sup> for the preparation of hierarchical nanostructured rutile titania. *J. Solid State Chem.* **180**, 799–803 (2007).
- Kaper, H. *et al.* Direct Low-Temperature Synthesis of Rutile Nanostructures in Ionic Liquids. *Small* **3**, 1753–1763 (2007).
- Jia, X. *et al.* Microwave-assisted synthesis of anatase TiO<sub>2</sub> nanorods with mesopores. *Nanotechnology* **18**, 075602–075607 (2007).
- Mali, S. S. *et al.* Nanocoral architecture of TiO<sub>2</sub> by hydrothermal process: Synthesis and Characterization. *Appl. Surf. Sci.* **257**, 9737–9746 (2011).
- Mali, S. S. *et al.* CdS-sensitized TiO<sub>2</sub> nanocorals: hydrothermal synthesis, characterization, Application. *Photochem. Photobiol. Sci.* **10**, 1652–1658 (2011).
- Mali, S. S. *et al.* Hydrothermal synthesis of rutile TiO<sub>2</sub> nanoflowers using Brønsted Acidic Ionic Liquid [BAIL]: Synthesis, characterization and growth mechanism. *CrystEngComm* **14**, 1920–1924 (2012).
- Zaban, A., Greenshtein, M. & Bisquert, J. Determination of the Electron Lifetime in Nanocrystalline Dye Solar Cells by Open-Circuit Voltage Decay Measurements. *ChemPhysChem* **4**, 859–864 (2003).

## Acknowledgments

One of the authors (SSM) wish to acknowledge DAE-BRNS Mumbai for financial support through the DAE-BRNS project no. 2008/37/8/BRNS/1489 for the 2008–2012. This work was also supported by the Priority Research Centers Program through the National Research Foundation of Korea (NRF) funded by the Ministry of Education, Science and Technology (2009–0094055).

## Author contributions

S.S.M., C.K.H. and P.S.P. contributed to the conception and design of the experiments, analysis of the data and writing the paper. S.S.M. carried out all experiments and wrote the paper. S.S.M. and C.A.B. performed EIS measurements, calculations and analyzed data. C.A.B., P.S.P. and P.N.B. participated in the scientific discussion and valuable suggestions during the course of this manuscript. All authors discussed the results and reviewed the manuscript.

## Additional information

**Supplementary information** accompanies this paper at <http://www.nature.com/scientificreports>

**Competing financial interests:** The authors declare no competing financial interests.

**How to cite this article:** Mali, S.S., Betty, C.A., Bhosale, P.N., Patil, P.S. & Hong, C.K. From nanocorals to nanorods to nanoflowers nanoarchitecture for efficient dye-sensitized solar cells at relatively low film thickness: All Hydrothermal Process. *Sci. Rep.* **4**, 5451; DOI:10.1038/srep05451 (2014).



This work is licensed under a Creative Commons Attribution-NonCommercial-NoDerivs 4.0 International License. The images or other third party material in this article are included in the article's Creative Commons license, unless indicated otherwise in the credit line; if the material is not included under the Creative Commons license, users will need to obtain permission from the license holder in order to reproduce the material. To view a copy of this license, visit <http://creativecommons.org/licenses/by-nc-nd/4.0/>