

## A seven-crystal Johann-type hard x-ray spectrometer at the Stanford Synchrotron Radiation Lightsource

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(Received 8 March 2013; accepted 18 April 2013; published online 7 May 2013)

We present a multicrystal Johann-type hard x-ray spectrometer ( $\sim 5$ – $18$  keV) recently developed, installed, and operated at the Stanford Synchrotron Radiation Lightsource. The instrument is set at the wiggler beamline 6-2 equipped with two liquid nitrogen cooled monochromators – Si(111) and Si(311) – as well as collimating and focusing optics. The spectrometer consists of seven spherically bent crystal analyzers placed on intersecting vertical Rowland circles of 1 m of diameter. The spectrometer is scanned vertically capturing an extended backscattering Bragg angular range ( $88^\circ$ – $74^\circ$ ) while maintaining all crystals on the Rowland circle trace. The instrument operates in atmospheric pressure by means of a helium bag and when all the seven crystals are used (100 mm of projected diameter each), has a solid angle of about 0.45% of  $4\pi$  sr. The typical resolving power is in the order of  $\frac{E}{\Delta E} \sim 10\,000$ . The spectrometer's high detection efficiency combined with the beamline 6-2 characteristics permits routine studies of x-ray emission, high energy resolution fluorescence detected x-ray absorption and resonant inelastic x-ray scattering of very diluted samples as well as implementation of demanding *in situ* environments. © 2013 AIP Publishing LLC. [<http://dx.doi.org/10.1063/1.4803669>]

### I. INTRODUCTION

During the last decade, the availability of high brilliance synchrotron radiation facilities boosted the development and the applicability of high-resolution x-ray spectroscopic techniques. X-ray emission (XES), resonant inelastic x-ray scattering (RIXS), and high energy resolution fluorescence detected x-ray absorption (HERFD-XAS) have been well-established as advanced characterization tools. Their element-specificity provides insights into the electronic structure and local chemical environment<sup>1–4</sup> of many materials. In the soft x-ray regime ( $< 2$  keV), and through grazing incidence angle grating spectrometers arranged on Rowland-based geometries,<sup>5,6</sup> XES and RIXS have provided unique insights in relevance to the energetics, momentum, and spin of superconductors and transition metal oxides, as well as surface chemistry of adsorbates.<sup>1,7–10</sup> On the other hand, hard x-rays provide some advantages related to the implementation of demanding sample environments and thus expand the applicability of these techniques to a broader field of scientific applications.<sup>2,3</sup> For example, electrochemical studies under *in situ* conditions, measurements on enzymes, proteins, catalytic systems under ambient conditions, or transition metal complexes have been studied routinely in the past few years.<sup>11–20</sup> Further on, XES has also been used to study the dynamics of time dependent systems with a temporal resolution of few picoseconds,<sup>21,22</sup> as limited by the storage ring technology. The recent availability of femtosecond hard x-ray sources, such as the Linear Coherent Light Source, provides some unique opportunities for the study of ultra-fast elec-

tronic structure dynamics in various phenomena,<sup>24</sup> such as electron transfer processes, transient molecular states, molecular dissociation, etc.

Typically, hard x-rays are resolved with high-energy-resolution by employing Bragg type perfect-crystal x-ray optics. The efficient collection of x-rays emitted from a divergent x-ray point source can be achieved with high-energy-resolution when single or multiple bent crystals are arranged in the so-called Johann,<sup>25</sup> Johansson,<sup>26</sup> or Von Hamos<sup>27</sup> geometries. These geometries can be separated by the way a spectrum is collected, i.e., point-to-point focusing where the collection of different energy points requires scanning elements, or energy dispersive geometries, where the full spectrum is collected simultaneously without the need of any moving parts.

In the Von Hamos geometry, a cylindrically bent crystal disperses the radiation along its flat surface's plane and focuses it along its axis of curvature onto a line like feature. The spatially distributed signal is recorded with a position sensitive detector at the crystal's focusing axis providing the overall spectrum. Alternative wavelength dispersive concepts have been proposed and implemented based on Johansson geometry having the source positioned inside the Rowland circle.<sup>28</sup> Various setups based on these aforementioned conceptual principles have been built up to the present moment.<sup>23,28–33</sup> Dispersive setups have certain important practical advantages since they do not incorporate movable parts and record the overall spectrum simultaneously. The latter point becomes especially relevant for X-ray Free Electron Laser (XFEL) type of experiments; a multicrystal Von Hamos spectrometer was recently build for this purpose.<sup>23</sup> On the other hand, the two main disadvantages of

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the dispersive setups are: (1) the much larger exposed detector area, which unavoidably increases the background to signal ratio and (2) the fact that the effective solid angle depends on the exact spectral range which might vary depending on the study. This is particularly limiting for HERFD-XAS studies where often a very narrow spectral range is required, which in turn results in a very small solid angle.

Point-to-point scanning instruments in Rowland circle geometries can provide orders of magnitude higher signal to background ratio and often a larger solid angle. A prototype single element Johann spectrometer was built in early 1990s at NSLS.<sup>37</sup> Since then, multi-crystal spectrometers based on intersecting Rowland circles have been developed, enhancing the solid angle and therefore sensitivity as well as collection efficiency. The first of these multi-crystal instruments was developed by Wang *et al.*;<sup>38</sup> an updated design based on an 8-crystal analyzer spectrometer using a simpler scanning procedure was built by Bergmann and Cramer.<sup>39</sup> Currently, various multi-crystal instruments are in operation at several synchrotron radiation facilities;<sup>34–42,45–48</sup> until recently a 14-crystal spectrometer operated with a simplified energy scanning scheme at the Stanford Synchrotron Radiation Light-source (SSRL).<sup>3</sup>

Here, we present the design and discuss the performance of a new dedicated 7-crystal Johann-type hard x-ray spectrometer developed at SSRL. The instrument is used for hard x-ray emission and resonant inelastic x-ray scattering, as well as high energy resolved fluorescence detection x-ray absorption and extended x-ray absorption fine structure applications. Some examples demonstrating the performance of this now operational end-station will be discussed.

## II. TECHNICAL DESCRIPTION

The x-ray spectrometer is stationed at the 57-pole 0.9T wiggler beamline 6-2 at SSRL. Beamline 6-2 is equipped with two liquid nitrogen cooled double crystal monochromators, a Si(111) ( $\phi = 0$ ) and a Si(311) ( $\phi = 0$ ) device that can be alternately operated depending on the required energy resolution. A collimating and a focusing Rh-coated mirrors are positioned before and after the monochromator, respectively. The beamline delivers monochromatic x-rays over an energy range of 4–18 keV. Under the current SPEAR3 storage ring operational conditions (electron energy of 3 GeV, current 500 mA in top-off mode) and through the Si(311) (Si(111)) monochromator, an incident beam flux of  $\sim 4 \times 10^{12}$  photons/s ( $2 \times 10^{13}$  photons/s) at 6.5 keV with an energy resolution of about 180 meV (850 meV) and a beam size of about  $140 \times 400 \mu\text{m}^2$  ( $v \times h$ ) is delivered at the sample position ( $\sim 28$  m from the source).

The seven analyzer crystals are placed on intersecting Rowland circles of 1 m of diameter (Fig. 1). Following the Johann geometry, the analyzer crystals, with a projected diameter of 100 mm, are spherically bent to a radius of 1 m. The Rowland circle is set on the vertical plane since the vertical beam size is smaller than the horizontal one and thus the contribution of the geometric effects within the energy resolution is minimized.<sup>39</sup> In order to minimize unwanted x-ray scattering the seven crystals are placed around  $\sim 90^\circ$  with respect to

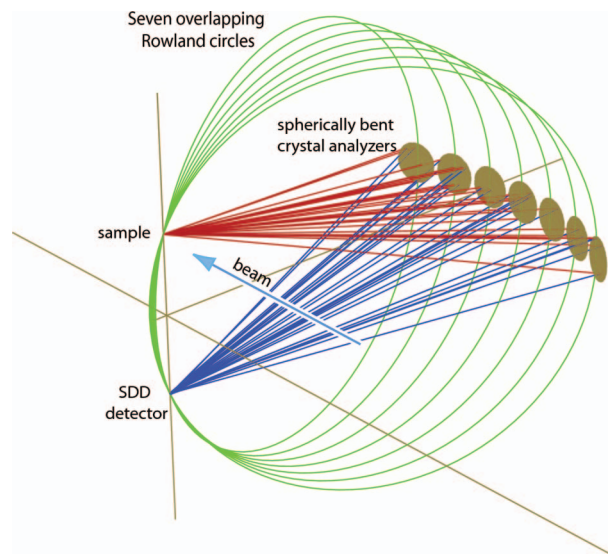


FIG. 1. Schematic representation of the overlapping Rowland circles concept. Seven spherically bent crystal analyzers are aligned on the same sample-detector axis.

the incident beam direction along the horizontal plane, covering an angular range of  $\sim 75^\circ$ – $120^\circ$ . The solid angle of the instrument, when all seven crystals are used, is 0.45% of  $4\pi$  sr. All crystal modules are positioned on a common rigid hosting plate that moves vertically with a stepper motor (Fig. 2). Each crystal module can move along its normal direction through its respective linear translation stage; this motion allows the crystals to follow the exact Rowland circle as the spectrometer is scanned. The analyzed signal from all crystals

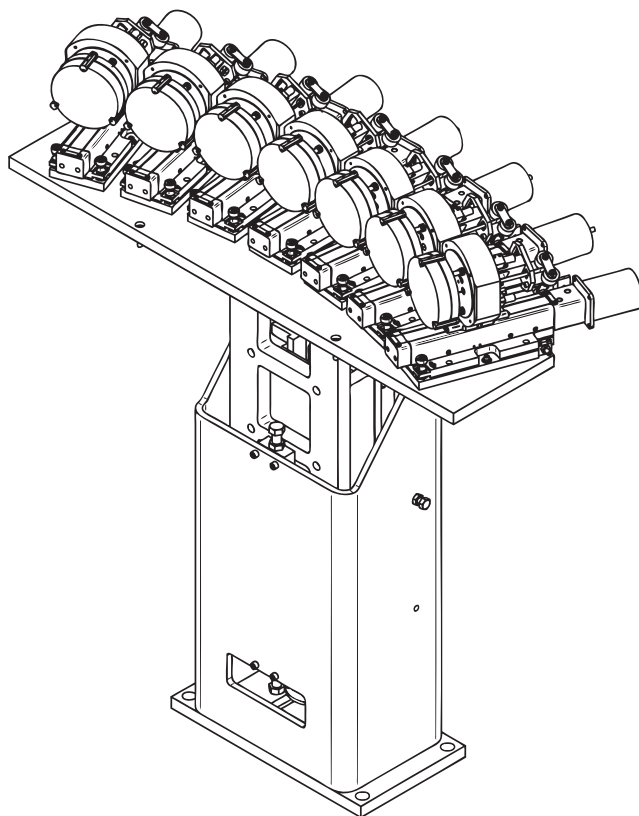


FIG. 2. Spectrometer's mechanical drawing.

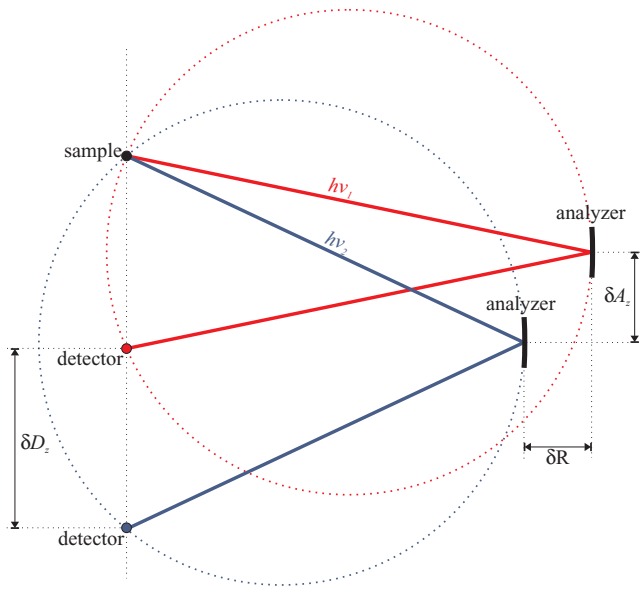


FIG. 3. Schematic representation of the relative translations ( $\delta R$ ,  $\delta A_z$ ,  $\delta D_z$ ) of motors  $R$ ,  $A_z$ ,  $D_z$  for two different photon energies  $h\nu_1$  and  $h\nu_2$ ; the analyzer is maintained on the exact Rowland circle (shown with dotted lines).

is recorded simultaneously with a detector placed on a vertical motorized stage. The detector's overall range of motion allows the spectrometer to capture Bragg angles between  $74^\circ$  and  $88^\circ$  (Bragg angles even closer to backscattering,  $>88^\circ$ , can be reached when the sample environment permits it). The detector routinely used is a silicon drift detector (energy resolution full width at half maximum (FWHM) of about 150 eV at 5.9 keV). Such an energy resolving detector further separates the wanted analyzed signal from the unwanted diffuse scattering or other fluorescence radiation that may reach the detector. This can improve the signal to background ratio significantly and is of particular importance when one needs to analyze very weak signals (e.g., valence electron transitions in diluted samples). For high-count rate applications, or  $\sim$ ps time-resolved studies, the detector can be replaced with an avalanche photodiode. Since the overall experimental station operates in atmospheric pressure, a polypropylene bag filled with helium is used to minimize both the attenuation and the diffuse scattering of the x-rays as they propagate from the sample to the analyzers and then toward the detector.

The spectrometer's energy scanning scheme is based on the Bragg angle equation and trigonometric expressions that represent the Rowland circle geometric conditions (Fig. 3). In this way, the positions for all spectrometer's motors for a given analyzed photon energy  $h\nu$  (in eV) are calculated through the following expressions.

$$\theta = \arcsin \frac{12398.5}{2d_{hkl} h\nu}, \quad (1)$$

$$R = 1000 \sin \theta, \quad (2)$$

$$A_z = R \cos \theta, \quad (3)$$

$$D_z = 2 A_z, \quad (4)$$

where  $\theta$  is the Bragg angle,  $d_{hkl}$  the d-spacing for the given crystal orientation/type used,  $R$  the distance of the crystals

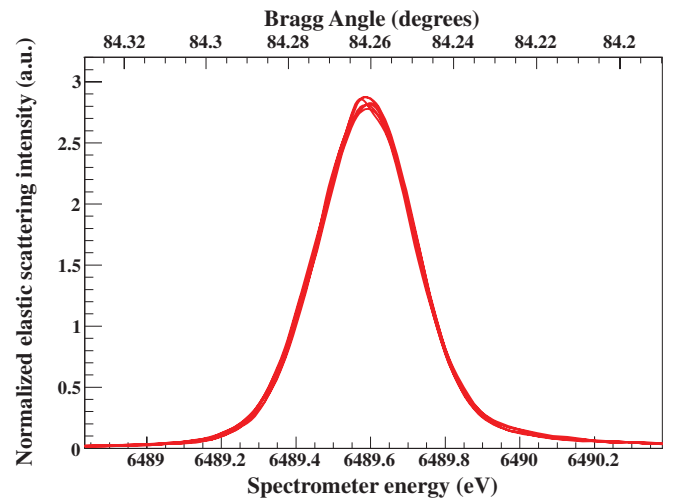


FIG. 4. Seven elastic scattering scans at 6489.58 eV (Bragg angle  $\sim 84.26^\circ$ ) for each of spectrometer's crystal positions. The response is identical. (Note: to compensate the polarization of the incident radiation and thus the strong angular dependence of the scattering intensity, the elastic scans for the various spectrometer positions have been normalized to their integrated area for comparison purposes.)

from the sample-detector axis,  $A_z$  the distance of the crystal from the sample/incident x-ray beam plane, and  $D_z$  the detector's distance from the incident x-ray beam plane (all distances in mm). The actual positions of all motors ( $R$ ,  $D_z$ ,  $A_z$ ) have been measured with an accuracy of less than 1 mm, whereas the relative motions of the motorized stages have been calibrated within  $<0.5\%$ .

Each analyzer crystal is installed on a motorized module providing the yaw and pitch angular motions required for its alignment. The adopted modules have been described in details elsewhere.<sup>42</sup> For a given set of crystals these angular positions are initially aligned by using a laser source and thus according to their optical surface. The fine alignment of their angular positions and the correction for slight miscuts (if the Bragg plane is not parallel to the crystal surface) is accomplished with x-rays (most frequently with elastic scattering scans). Iterative scripts have been developed in order to facilitate and optimize the overall x-ray based alignment procedure of the spectrometer. Figure 4 shows seven elastic scattering peaks for each of the seven spectrometer's crystal positions using a Si(440) diced-and-bent crystal cut analyzer<sup>42,76</sup> and the Si(311) monochromator. All elastic peaks have been normalized to their integrated area for comparison purposes; the response is identical. In order to calibrate the spectrometer's nominal energy values (calculated solely through the measured distances) with respect to the monochromator, the elastic peaks are measured along the overall energy range required for a given study. For instance, Fig. 5 shows elastic scattering peaks along an energy range of 6462–6572 eV.

As the Bragg angle moves away from the  $90^\circ$  backscattering geometry, the energy resolution of the spectrometer is expected to gradually degrade. This is due to geometrical effects inherent to the Johann geometry ( $\cot \theta$ ) and has been extensively reported in previous works.<sup>39</sup> Figure 6 shows the evolution of the FWHM along the whole range of the recorded scattering peaks presented in Fig. 5 (note that the FWHM

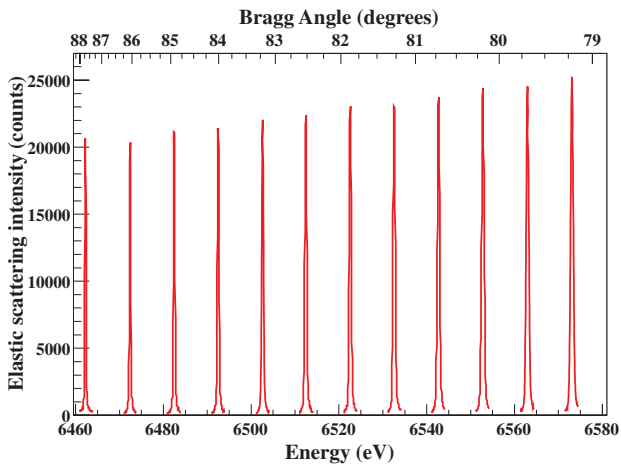


FIG. 5. Elastic peaks along an extended energy range of the spectrometer covered using diced-and-bent Si(440) cut.

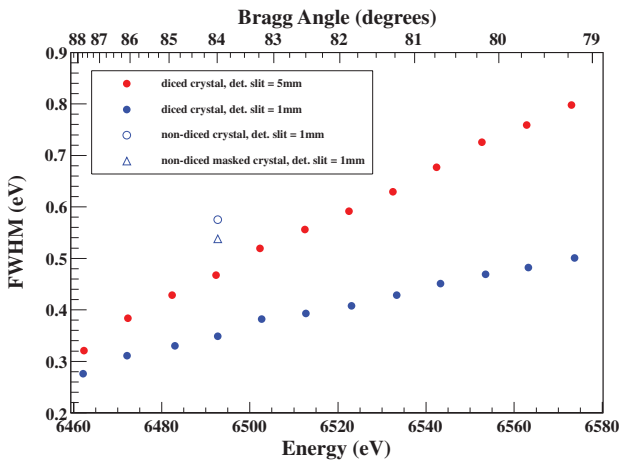


FIG. 6. Experimental full width at half maximum (FWHM) of the elastic scattering peaks with the diced-and-bent Si(440). Data for 5 mm and 1 mm slits in front of the SDD detector are shown. Also, a representative value for a conventional bent analyzer (non-diced) of 100 mm diameter is given with and without masking its outermost 25 mm along the dispersive direction.

corresponds to the cumulative contribution of the Si(311) monochromator employed here and that of Si(440) crystal analyzers). We compare the evolution of the FWHM when using two sets of vertical slits of different size, 5 mm (practically uncollimated) and 1 mm, in front of the detector. A dramatic improvement of the energy resolution is achieved when using the tighter set of slits, especially as the diffraction angle moves away from the backscattering, i.e., where the geometry effects have more prominent contribution (for the present case when moving from 5 mm detector slit size to 1 mm, the overall intensity is reduced by about a factor of 2). In the same Fig. 6, some representative FWHM values are given for conventional (non-diced) spherical Si(440) analyzers of 100 mm of diameter with and without masking the crystal along its dispersive direction (in both cases a 1 mm vertical detector slit is used). It is worth noting that under identical conditions, the maxima of the elastic peak for the conventional analyzer found to be 10% higher compare to the diced-and-bent one (we should keep in mind that the diced-and-bent crystal has effectively 20% smaller active area due to the cuts<sup>42</sup>).

For capturing as many fluorescence lines as possible within the 5–18 keV energy range, and for the available spectrometer’s angular scanning range (74°–88°), we are gradually obtaining an extended collection of crystal analyzers. Up to the present moment, the available crystal cuts are the followings: Si(110) (diced-and-bent and conventional), Si(111), Si(211), Si(620), Si(551), Si(1020), Si(553), Ge(110), Ge(111), Ge(211), and Ge(620). In Fig. 7, the energy ranges captured with the available crystals (accounting their various allowed high order reflections) are displayed with horizontal bars; some exemplary x-ray emission lines and absorption edges are also shown. The energy resolution for these conventional spherically bent crystals is typically in the order of  $\frac{E}{\Delta E} \sim 10\,000$  (for diced-and-bent ones this factor can be increased by a factor of  $\sim 2$ ). As shown in Fig. 6, masking the crystals’ outermost areas can result in improvements on the energy resolution up to 20%; however, the magnitude depends also on the quality of the analyzers.

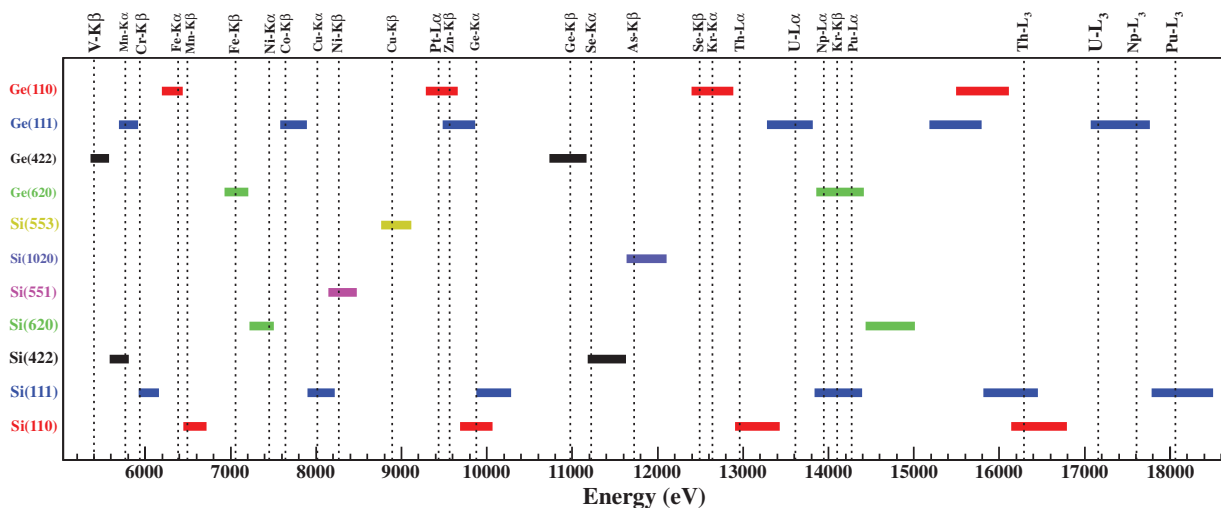


FIG. 7. Energy ranges for the current SSRL’s crystal collection ( Si(110), Si(111), Si(211), Si(620), Si(551), Si(1020), Si(553), Ge(110), Ge(111), Ge(211), and Ge(620)) accounting their various high order allowed reflections. These energies can be analyzed within the 74°–88° scanning angular range of the spectrometer. Some exemplary emission lines and absorption edges are also indicated.



### III. EXAMPLES

In Fig. 8 we present non-resonant  $K\beta$  x-ray emission spectra from three polycrystalline manganese compounds ( $\text{MnO}$ ,  $\text{Mn}_2\text{O}_3$ ,  $\text{MnO}_2$ ). These measurements were collected using the conventional  $\text{Si}(440)$  crystal analyzers while the spectrometer was scanned within the Bragg angular range of  $88^\circ$ – $80^\circ$ ; the incident energy was set at 7000 eV. The spectrometer's high throughput typically provides the peak intensity of the  $3d$  metals'  $K\alpha_1$  fluorescence line to be of about several  $10^6$  photons per second per crystal analyzer when  $\sim 10^{13}$  incident photons per second are tuned right above the absorption edge and when thick pure samples are used. In the same way, for the peak intensity of the  $K\beta$  lines the spectrometer provides from several  $10^5$  to about  $10^6$  counts per second per analyzer crystal.  $K\beta$  non-resonant x-ray emission spectroscopy on  $3d$  metals is a routine analytical methodology and its chemical sensitivity has been discussed in the literature.<sup>2</sup> In such spectra two regions of interest can be distinguished. The prominent  $K\beta$  main line features are at the lower energies, and the significantly weaker K-valence features lie at the higher energies close to the absorption edge (expanded within the inset figure of Fig. 8). For both cases the obtained spectra are broadened due to the lifetime of the initial and final states involved (normally the initial state ( $1s$ ) has the dominant contribution, about 1–2 eV for the  $3d$  metals<sup>43</sup>), as well as to the spectrometer's response function. Since the experimental energy resolution is smaller than the lifetime of the involved states ( $\frac{E}{\Delta E} \sim 10\,000$ ), the instrument's contribution to the total energy broadening is negligible. In non-resonant emission experiments, the incident beam energy resolution does not have any influence neither. However, it should be noted that as the incident photon energy gets gradually above the ionization threshold, multi-electron shake-off/ups processes are slowing opening and this may result on satellite spectral fingerprints within the emission spectrum.<sup>44</sup>

The  $K\beta$  main lines ( $K\beta_{1,3}$  and  $K\beta'$ ) correspond to the  $3p \rightarrow 1s$  core electron transitions; the chemical sensitivity comes indirectly from the electron-electron interactions and

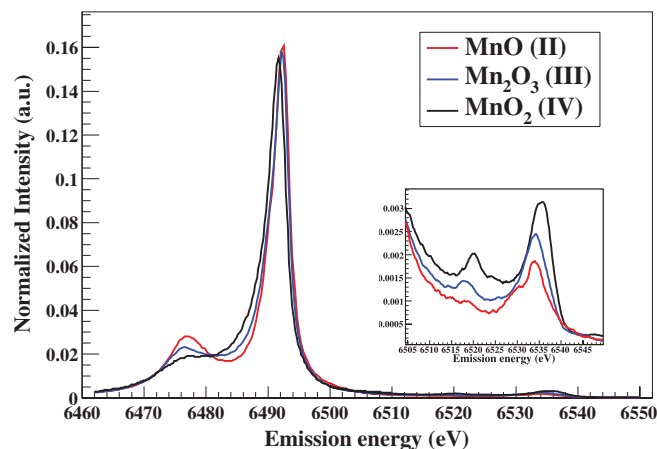


FIG. 8.  $K\beta$  x-ray emission from various polycrystalline manganese compounds ( $\text{MnO}$ ,  $\text{Mn}_2\text{O}_3$ ,  $\text{MnO}_2$ ). The inset figure shows the K-valence emission pattern. Spectra have been normalized to the integrated area of the  $K\beta$  main lines.

core screening effect of the valence electrons. For  $3d$  transition metals such as Mn, the  $K\beta_{1,3}$  and  $K\beta'$  splitting is mostly due to the  $3p$ - $3d$  exchange interaction consisting a fingerprint for the metal's spin and oxidation state (and indirectly to the local geometrical structure). Various data analysis schemes have been proposed and adopted for  $K\beta$  main lines depending on the application, e.g., the maxima position, the first moment position or the integrated absolute difference with respect to a reference spectrum.<sup>2,49–57</sup> However, the complete theoretical reconstruction of  $K\beta$  main lines for molecular and condensed matter systems through *ab initio* approaches remains yet unresolved and theoretical efforts beyond atomic multiplet theory are undergoing. This is a topic of paramount interest for spectroscopists since it would provide a thorough understanding of the emission pattern and potentially enhance the electronic structure insights that  $K\beta$ -based x-ray emission spectroscopy provides.

On the other hand, the valence to core transitions, although exhibiting about two orders of magnitude weaker transition probabilities, have a direct sensitivity on the local chemical environment/structure through the occupied valence orbitals (for the valence electrons involved in the symmetry allowed electronic transitions). It has been shown that for various systems simple one electron transition calculations, based on ground state wavefunctions, could address all the spectral fingerprints within the K valence x-ray emission spectra.<sup>13,58–63</sup> In this way, Density Functional Theory (DFT) approaches provide a direct intuitive insight about the spectral patterns and the electronic structure changes due to the ligand environment modifications. Most notably, the so-called crossover K valence fingerprint has been shown to sensitively discriminate ligands with similar atomic number (e.g., C, N, O) within metal's first coordination shell, overcoming in this way a limitation of EXAFS.<sup>13,14,62–65</sup> The weak nature of those transition stimulates instrument developments such as the one presented here.

Recording of x-ray emission while tuning the incident radiation along the absorption edge allows resonant x-ray emission (RXES) studies, referred also as RIXS. In Fig. 9, we present a resonant study on a polycrystalline  $\text{Fe}_2\text{O}_3$  measured with  $\text{Ge}(620)$  analyzers and using a  $\text{Si}(311)$  monochromator (the combined monochromator and analyzers energy resolution was 0.7 eV). A RIXS contour map is shown in the left, and a HERFD-XAS on the right (compared with a conventional XAS spectrum obtained using the  $\text{Fe-K}\alpha$  fluorescence discriminated by a solid state silicon detector). Several reports in the literature discuss about both RIXS spectroscopy and HERFD-XAS characteristics as well as the relevant emerging applications.<sup>2,3,66–68</sup> The HERFD-XAS is practically a diagonal cut along the RIXS contour map (when plotted as incident energy versus energy transfer), i.e., when the x-ray emission energy is fixed while scanning the incident radiation energy. The main advantages of HERFD-XAS are two-fold; the signal-to-background is significantly improved since crystal analyzers resolve any overlapping fluorescence peaks and/or scattering radiation within the detection channel, while it sharpens the XAS spectral features (this becomes evident for instance for the  $t_{2g}$  and  $e_g$  pre-edge peaks shown within the inset of Fig. 9). However, the interpretation of

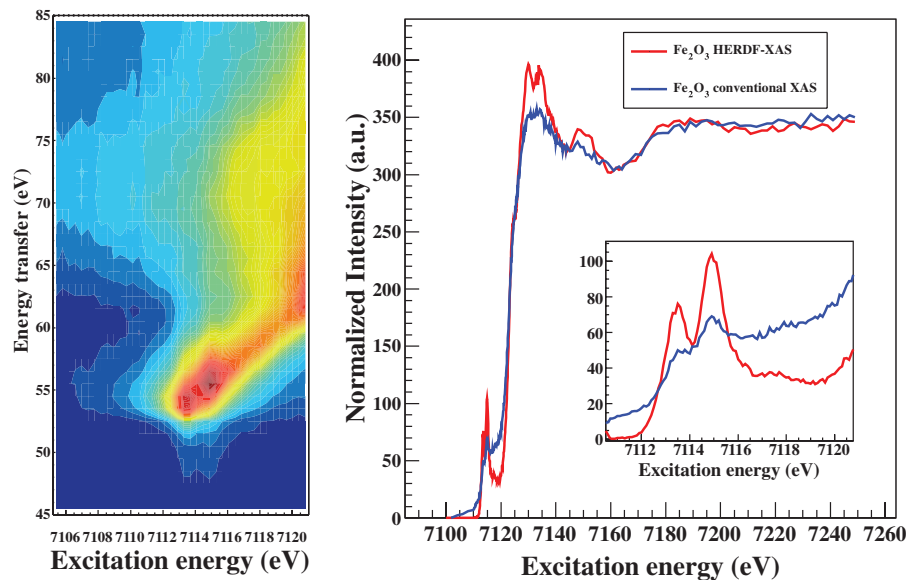


FIG. 9. A RIXS contour map of Fe<sub>2</sub>O<sub>3</sub> along the pre-edge of Fe (left). Comparison of a HERFD-XAS and conventional fluorescence XAS spectrum using a solid state detector integrating the Fe-K $\alpha$  line (right).

HERFD-XAS through conventional XAS calculations (i.e., the recorded spectrum to represent the linear absorption coefficient) should be handled with care since this can be reliably attempted only when final-state effects are absent. For instance, for *5d* metals it has been demonstrated that this assumption stands very well and XAS calculations can be used for interpretation purposes.<sup>17,69,70,73</sup> On other hand, for the *3d* transition metals final state effects may be prominent and the full RIXS map would then be required for identifying them. For the exemplary case presented in Fig. 9 it is shown that HERFD-XAS along the contour's diagonal represents a reliable approximation of the XAS.

Since the contribution of the core hole broadening in resonant spectroscopies is not as prominent as in the non-resonant experiments, the energy resolution of the spectrometer/incident beam may not have a negligible influence anymore. Thus in certain case studies having an *1s* lifetime comparable energy resolution may become a limitation. For such type of applications diced crystals can be incorporated enhancing the energy resolution of this spectrometer to few hundreds of meV (in the present case further improvement of the energy resolution would not improve the overall performance due to the Si(311) monochromator energy resolution limitation). The implementation of four bounces monochromators with high order cuts combined with diced analyzers and positions sensitive detectors has been shown to exhibit significant improvement on the energy resolution ( $\frac{\Delta E}{E} \sim 10^5$ ),<sup>35,71,72</sup> however that comes at the expense of the throughput.

#### IV. CONCLUSION

Described here is a multicrystal Johann-type hard x-ray spectrometer recently developed, installed and now operating at SSRL. The setup consists of 7 analyzer crystals arranged on overlapping vertical Rowland circles. The spectrometer can

be scanned vertically, within the Bragg angular range of 74°–88°, maintaining the exact Rowland circle condition for all the analyzers. Using SSRL's wide collection of Si and Ge crystal cuts the spectrometer can efficiently analyze x-ray energies within the energy range of about 5–18 keV. The large solid angle acceptance of the instrument combined with point-to-point focusing geometry and the energy dispersive silicon drift detector, minimizes the background signal and ensure an optimized detection sensitivity for weak signals and/or dilute samples.

X-ray emission, high-energy resolution x-ray absorption spectroscopy and resonant inelastic x-ray scattering techniques exhibit a growing interest within the scientific community. The emerging applicability of these powerful spectroscopic techniques on cross-disciplinary scientific fields requires advanced instruments coupled to high-brilliance x-ray sources in order to achieve the required throughput and energy-resolution as well as the flexibility to accommodate demanding sample environments. *In situ* catalysis on surfaces, matter under extreme condition, radiation sensitive metallo-proteins and protein samples includes only some examples of cases that need to be routinely addressed on synchrotron radiation facilities worldwide. Up to the present moment a wide scientific research program is supported with the presented advanced instrumentation at SSRL.<sup>17,23,24,74,75</sup> The projected new SSRL's undulator beamline which will be dedicated to high-resolution x-ray spectroscopy purposes, is expected to boost even further the capabilities of the instrument towards a wider range of applications, especially related to the more efficient implementation of high pressure studies (e.g., diamond anvil cells), as well as the laser-pump x-ray probe scheme for  $\sim$ ps time-resolved applications.

#### ACKNOWLEDGMENTS

Support from SLAC staff is thankfully acknowledged. The spectrometer was funded by the America Recovers

and Reinvests Act Project No. 2005045. The Stanford Synchrotron Radiation Lightsource is a National User Facility operated by Stanford University on behalf of the U.S. Department of Energy, Office of Basic Energy Sciences.

- <sup>1</sup>A. Kotani and S. Shin, *Rev. Mod. Phys.* **73**, 203 (2001).
- <sup>2</sup>P. Glatzel and U. Bergmann, *Coord. Chem. Rev.* **249**(1–2), 65–95 (2005).
- <sup>3</sup>U. Bergmann and P. Glatzel, *Photosynth. Res.* **102**(2–3), 255–266 (2009).
- <sup>4</sup>F. M. F. de Groot and A. Kotani, *Core Level Spectroscopy of Solids* (Taylor and Francis, New York, 2008).
- <sup>5</sup>J. Nordgren, G. Bray, S. Cramm, R. Nyholm, J. E. Rubensson, and N. Wassdahl, *Rev. Sci. Instrum.* **60**(7), 1690–1696 (1989).
- <sup>6</sup>G. Ghiringhelli, A. Piazzalunga, C. Dallera, G. Trezzi, L. Braicovich, T. Schmitt, V. N. Strocov, R. Betemps, L. Patthey, X. Wang, and M. Griioni, *Rev. Sci. Instrum.* **77**, 113108 (2006).
- <sup>7</sup>L. J. P. Ament, M. van Veenendaal, T. P. Devereaux, J. P. Hill, and J. van den Brink, *Rev. Mod. Phys.* **83**(2), 705–767 (2011).
- <sup>8</sup>A. Nilsson and L. G. M. Pettersson, *Surf. Sci. Rep.* **55**(2–5), 49–167 (2004).
- <sup>9</sup>T. Schiros, S. Haq, H. Ogasawara, O. Takahashi, H. Öström, K. Andersson, L.G.M. Pettersson, A. Hodgson, and A. Nilsson, *Chem. Phys. Lett.* **429**(4–6), 415–419 (2006).
- <sup>10</sup>D. J. Miller, H. Öberg, S. Kaya, H. Sanchez Casalongue, D. Friebe, T. Anniyev, H. Ogasawara, H. Bluhm, L. G. M. Pettersson, and A. Nilsson, *Phys. Rev. Lett.* **107**, 195502 (2011).
- <sup>11</sup>U. Bergmann, P. Glatzel, and S. P. Cramer, *Microchem. J.* **71**(2–3), 221–230 (2002).
- <sup>12</sup>U. Bergmann, P. Glatzel, F. deGroot, and S. P. Cramer, *J. Am. Chem. Soc.* **121**, 4926–4927 (1999).
- <sup>13</sup>U. Bergmann, C. R. Horne, T. J. Collins, J. M. Workman, and S. P. Cramer, *Chem. Phys. Lett.* **302**(1–2), 119–124 (1999).
- <sup>14</sup>K. M. Lancaster, M. Roemelt, P. Ettenhuber, Y. L. Hu, M. W. Ribbe, F. Neese, U. Bergmann, and S. DeBeer, *Science* **334**(6058), 974–977 (2011).
- <sup>15</sup>R. Alonso-Mori, E. Paris, G. Giuli, S. G. Eeckhout, M. Kavcic, M. Zitnik, K. Bucar, L. G. M. Pettersson, and P. Glatzel, *Inorg. Chem.* **49**(14), 6468–6473 (2010).
- <sup>16</sup>R. Alonso-Mori, E. Paris, G. Giuli, S. G. Eeckhout, M. Kavcic, M. Zitnik, K. Bucar, L. G. M. Pettersson, and P. Glatzel, *Anal. Chem.* **81**(15), 6516–6525 (2009).
- <sup>17</sup>L. R. Merte, F. Behafarid, D. J. Miller, D. Friebe, S. Cho, F. Mbuga, D. Sokaras, R. Alonso-Mori, T.-C. Weng, D. Nordlund, A. Nilsson, and B. R. Cuenya, *ACS Catal.* **2**(11), 2371 (2012).
- <sup>18</sup>Y. Pushkar, X. Long, P. Glatzel, G. W. Brudvig, G. C. Dismukes, T. J. Collins, V. K. Yachandra, J. Yano, and U. Bergmann, *Angew. Chem., Int. Ed.* **49**(4), 800–803 (2010).
- <sup>19</sup>P. Glatzel, J. Singh, K. O. Kvashnina, and J. A. van Bokhoven, *J. Am. Chem. Soc.* **132**(8), 2555 (2010).
- <sup>20</sup>G. Smolentsev, A. V. Soldatov, J. Messinger, K. Merz, T. Weyhermuller, U. Bergmann, Y. Pushkar, J. Yano, V. K. Yachandra, and P. Glatzel, *J. Am. Chem. Soc.* **131**(36), 13161–13167 (2009).
- <sup>21</sup>G. Vanko, P. Glatzel, V. T. Pham, R. Abela, D. Grolimund, C. N. Borca, S. L. Johnson, C. J. Milne, and C. Bressler, *Angew. Chem., Int. Ed.* **49**(34), 5910–5912 (2010).
- <sup>22</sup>K. Haldrup, G. Vankó, W. Gawelda, A. Galler, G. Doumy, A. M. March, E. P. Kanter, A. Bordage, A. Dohn, T. B. van Driel, K. S. Kjaer, H. T. Lemke, S. E. Canton, J. Uhlig, V. Sundström, L. Young, S. H. Southworth, M. M. Nielsen, and C. Bressler, *J. Phys. Chem. A* **116**(40), 9878 (2012).
- <sup>23</sup>R. Alonso-Mori, J. Kern, D. Sokaras, T.-C. Weng, D. Nordlund, R. Tran, P. Montanez, J. Delor, V. K. Yachandra, J. Yano, and U. Bergmann, *Rev. Sci. Instrum.* **83**, 073114 (2012).
- <sup>24</sup>R. Alonso-Mori, J. Kern, R. J. Gildea, D. Sokaras, T.-C. Weng, B. Lassalle-Kaiser, R. Tran, J. Hattne, H. Laksmono, J. Hellmich, C. Glöckner, N. Echols, R. G. Sierra, D. W. Schafer, J. Sellberg, C. Kenney, R. Herbst, J. Pines, P. Hart, S. Herrmann, R. W. Grosse-Kunstleve, M. J. Latimer, A. R. Fry, M. M. Messerschmidt, A. Miahnahri, M. M. Seibert, P. H. Zwart, W. E. White, P. D. Adams, M. J. Bogan, S. Boutet, G. J. Williams, A. Zouni, J. Messinger, P. Glatzel, N. K. Sauter, V. K. Yachandra, J. Yano, and U. Bergmann, *Proc. Natl. Acad. Sci. U.S.A.* **109**(47), 19103 (2012).
- <sup>25</sup>H. H. Johann, *Z. Phys.* **69**, 185 (1931).
- <sup>26</sup>T. Johansson, *Z. Phys.* **82**, 507 (1933).
- <sup>27</sup>L. v. Hamos, *Naturwiss.* **20**, 705–706 (1932).
- <sup>28</sup>M. Kavčič, M. Budnar, A. Mühleisen, F. Gasser, M. Žitnik, K. Bučar, and R. Bohinc, *Rev. Sci. Instrum.* **83**, 033113 (2012).
- <sup>29</sup>J. Hozszowska and J. C. Dousse, *J. Electron Spectrosc. Relat. Phenom.* **137–140**, 687–690 (2004).
- <sup>30</sup>H. Hayashi, M. Kawata, R. Takeda, Y. Udagawa, Y. Watanabe, T. Takano, S. Nanao, and N. Kawamura, *J. Electron Spectrosc. Relat. Phenom.* **136**(1–2), 191–197 (2004).
- <sup>31</sup>B. Dickinson, G. T. Seidler, Z. W. Webb, J. A. Bradley, K. P. Nagle, S. M. Heald, R. A. Gordon, and I. M. Chou, *Rev. Sci. Instrum.* **79**(12), 123112 (2008).
- <sup>32</sup>B. A. Mattern, G. T. Seidler, M. Haave, J. I. Pacold, R. A. Gordon, J. Planillo, J. Quintana, and B. Rusthoven, *Rev. Sci. Instrum.* **83**, 023901 (2012).
- <sup>33</sup>J. Szlachetko, M. Nachtegaal, E. de Boni, M. Willmann, O. Safonova, J. Sa, G. Smolentsev, M. Szlachetko, J. A. van Bokhoven, J.-C. Dousse, J. Hozszowska, Y. Kayser, P. Jagodzinski, A. Bergamaschi, B. Schmitt, C. David, and A. Lücke, *Rev. Sci. Instrum.* **83**, 103105 (2012).
- <sup>34</sup>C. C. Kao, K. Hamalainen, M. Krisch, D. P. Siddons, and T. Oversluisen, *Rev. Sci. Instrum.* **66**, 1699 (1995).
- <sup>35</sup>T. Gog, G. T. Seidler, D. M. Casa, M. H. Upton, J. Kim, S. Stoupin, K. P. Nagle, M. Balasubramanian, R. A. Gordon, T. T. Fister, S. M. Heald, T. Toellner, J. P. Hill, D. S. Coburn, Young-June Kim, A. H. Said, E. E. Alp, W. Sturhahn, H. Yavas, C. A. Burns, and H. Sinn, *Synchrotron Radiat. News* **22**, 12 (2009).
- <sup>36</sup>J. P. Hill, D. S. Coburn, Y.-J. Kim, T. Gog, D. M. Casa, C. N. Koditwakk, and H. Sinn, *J. Synchrotron Radiat.* **14**, 361 (2007).
- <sup>37</sup>V. Stojanoff, K. Hamalainen, D. P. Siddons, J. B. Hastings, L. E. Berman, S. Cramer, and G. Smith, *Rev. Sci. Instrum.* **63**(1), 1125–1127 (1992).
- <sup>38</sup>X. Wang, M. M. Grush, A. G. Froeschner, and S. P. Cramer, *J. Synchrotron Radiat.* **4**, 236–242 (1997).
- <sup>39</sup>U. Bergmann and S. P. Cramer, *Proc. SPIE* **3448**, 198–209 (1998).
- <sup>40</sup>P. Glatzel, F. M. F. de Groot, and U. Bergmann, *Synchrotron Radiat. News* **22**(2), 12–16 (2009).
- <sup>41</sup>I. Llorens, E. Lahera, W. Delnet, O. Proux, A. Brailard, J. L. Hazemann, A. Prat, D. Testemale, F. Gelebart, M. Morand, A. Shukla, N. Bardou, O. Ulrich, S. Arnaud, J. F. Berar, N. Boudet, B. Caillot, P. Chaurand, J. Rose, E. Doelsch, P. Martin, and P. L. Solari, *Rev. Sci. Instrum.* **83**, 063104 (2012).
- <sup>42</sup>D. Sokaras, D. Nordlund, T. C. Weng, R. Alonso-Mori, P. Velikov, D. Wagner, A. Garachtchenko, M. George, V. Borzenets, B. Johnson, T. Rabedeau, and U. Bergmann, *Rev. Sci. Instrum.* **83**, 043112 (2012).
- <sup>43</sup>M. O. Krause and J. H. Oliver, *J. Phys. Chem. Ref. Data* **8**, 329 (1979).
- <sup>44</sup>M. Kavčič, M. Žitnik, K. Bučar, A. Mihelič, M. Štuhec, J. Szlachetko, W. Cao, R. Alonso-Mori, and P. Glatzel, *Phys. Rev. Lett.* **102**, 143001 (2009).
- <sup>45</sup>E. Kleymenov, J. A. van Bokhoven, C. David, P. Glatzel, M. Janousch, R. Alonso-Mori, M. Studer, M. Willmann, A. Bergamaschi, B. Henrich, and M. Nachtegaal, *Rev. Sci. Instrum.* **82**, 065107 (2011).
- <sup>46</sup>Q. Qian, T. A. Tyson, W. A. Caliebe, and C. C. Kao, *J. Phys. Chem. Solids* **66**, 2295–2298 (2005).
- <sup>47</sup>R. Verbeni, M. Kocsis, S. Huotari, M. Krisch, G. Monaco, F. Sette, and G. Vanko, *J. Phys. Chem. Solids* **66**, 2299–2305 (2005).
- <sup>48</sup>L. Journal, L. El Khoury, T. Marin, R. Guillemin, S. Carniato, A. Avila, R. Delaunay, C. F. Hague, and M. Simon, *Rev. Sci. Instrum.* **80**, 093105 (2009).
- <sup>49</sup>U. Bergmann, M. M. Grush, C. R. Horne, P. DeMarois, J. E. Penner-Hahn, C. F. Yocum, D. W. Wright, C. E. Dube, W. H. Armstrong, G. Christou, H. J. Eppley, and S. P. Cramer, *J. Phys. Chem. B* **102**, 8350 (1998).
- <sup>50</sup>H. Visser, E. Anxolabehere-Mallart, U. Bergmann, P. Glatzel, J. H. Robblee, S. P. Cramer, J.-J. Girerd, K. Sauer, M. P. Klein, and V. K. Yachandra, *J. Am. Chem. Soc.* **123**, 7031 (2001).
- <sup>51</sup>S. A. Pizarro, P. Glatzel, H. Visser, J. H. Robblee, G. Christou, U. Bergmann, V. K. Yachandra, *Phys. Chem. Chem. Phys.* **6**, 4864–4870 (2004).
- <sup>52</sup>J. R. Bargar, B. M. Tebo, U. Bergmann, S. M. Webb, P. Glatzel, V. Q. Chiu, and M. Villalobos, *Am. Mineral* **90**, 143–154 (2005).
- <sup>53</sup>C. J. Doonan, L. Zhang, C. G. Young, S. J. George, A. Deb, U. Bergmann, G. N. George, and S. P. Cramer, *Inorg. Chem.* **44**, 2579 (2005).
- <sup>54</sup>H. Gretarsson, A. Lupascu, J. Kim, D. Casa, T. Gog, W. Wu, S. R. Julian, Z. J. Xu, J. S. Wen, G. D. Gu, R. H. Yuan, Z. G. Chen, N.-L. Wang, S. Khim, K. H. Kim, M. Ishikado, I. Jarrige, S. Shamoto, J.-H. Chu, I. R. Fisher, and Y.-J. Kim, *Phys. Rev. B* **84**, 100509(R) (2011).
- <sup>55</sup>J. M. Chen, S. C. Haw, J. M. Lee, T. L. Chou, S. A. Chen, K. T. Lu, Y. C. Liang, Y. C. Lee, N. Hiraoka, H. Ishii, K. D. Tsuei, E. Huang, and T. J. Yang, *Phys. Rev. B* **84**, 125117 (2011).
- <sup>56</sup>L. Simonelli, N. L. Saini, M. Moretti Sala, Y. Mizuguchi, Y. Takano, H. Takeya, T. Mizokawa, and G. Monaco, *Phys. Rev. B* **85**, 224510 (2012).

- <sup>57</sup>J. Messinger, J. H. Robblee, U. Bergmann, C. Fernandez, P. Glatzel, H. Visser, R. M. Cinco, K. L. McFarlane, E. Bellacchio, S. A. Pizarro, S. P. Cramer, K. Sauer, M. P. Klein, and V. K. Yachandra, *J. Am. Chem. Soc.* **123**(32), 7804 (2001).
- <sup>58</sup>R. C. C. Perera and R. E. LaVilla, *J. Chem. Phys.* **81**, 3375 (1984).
- <sup>59</sup>K. E. Miyano, U. Arp, S. H. Southworth, T. E. Meehan, T. R. Walsh, and F. P. Larkins, *Phys. Rev. A* **57**, 2430 (1998).
- <sup>60</sup>U. Bergmann, J. Bendix, P. Glatzel, H. B. Gray, and S. P. Cramer, *J. Chem. Phys.* **116**, 2011 (2002).
- <sup>61</sup>M. U. Delgado-Jaime, B. R. Dible, K. P. Chiang, W. W. Brennessel, U. Bergmann, P. L. Holland, and S. DeBeer, *Inorg. Chem.* **50**(21), 10709 (2011).
- <sup>62</sup>N. Lee, T. Petrenko, U. Bergmann, F. Neese, and S. DeBeer, *J. Am. Chem. Soc.* **132**(28), 9715 (2010).
- <sup>63</sup>V. A. Safonov, L. N. Vykhodtseva, Y. M. Polukarov, O. V. Safonova, G. Smolentsev, M. Sikora, S. G. Eeckhout, and P. Glatzel, *J. Phys. Chem. B* **110**(46), 23192 (2006).
- <sup>64</sup>M. A. Beckwith, M. Roemelt, M.-N. Collomb, C. DuBoc, T.-C. Weng, U. Bergmann, P. Glatzel, F. Neese, and S. DeBeer, *Inorg. Chem.* **50**, 8397 (2011).
- <sup>65</sup>K. M. Lancaster, Y. Hu, U. Bergmann, M. W. Ribbe, and S. DeBeer, *J. Am. Chem. Soc.* **135**, 610 (2013).
- <sup>66</sup>W. M. Heijboer, P. Glatzel, K. R. Sawant, R. F. Lobo, U. Bergmann, R. A. Barrea, D. C. Koningsberger, B. M. Weckhuysen, and F. M. F. de Groot, *J. Phys. Chem. B* **108**, 10002 (2004).
- <sup>67</sup>P. Glatzel, T.-C. Weng, K. Kvashnina, J. Swarbrick, M. Sikora, E. Gallo, N. Smolentsev, and R. Alonso Mori, "Reflections on hard X-ray photon-in/photon-out spectroscopy for electronic structure studies," *J. Electron Spectrosc. Relat. Phenom.* (in press).
- <sup>68</sup>P. Glatzel, M. Sikora, G. Smolentsev, and M. Fernández-García, *Catal. Today* **145**, 294 (2009).
- <sup>69</sup>O. V. Safonova, M. Tromp, J. A. van Bokhoven, F. M. F. de Groot, J. Evans, and P. Glatzel, *J. Phys. Chem. B* **110**(33), 16162 (2006).
- <sup>70</sup>F. M. F. de Groot, M. H. Krisch, and J. Vogel, *Phys. Rev. B* **66**, 195112 (2002).
- <sup>71</sup>S. Huotari, F. Albergamo, G. Vanko, R. Verbeni, and G. Monaco, *Rev. Sci. Instrum.* **77**, 053102 (2006).
- <sup>72</sup>T. Gog, D. M. Casa, A. H. Said, M. H. Upton, J. Kim, I. Kuzmenko, X. Huang, and R. Khachatryan, *J. Synchrotron Radiat.* **20**, 74 (2013).
- <sup>73</sup>D. Friebe, D. J. Miller, D. Nordlund, H. Ogasawara, and A. Nilsson, *Angew. Chem., Int. Ed.* **123**(43), 10372 (2011).
- <sup>74</sup>C. H. Booth, Y. Jiang, D. L. Wang, J. N. Mitchell, P. H. Tobash, E. D. Bauer, M. A. Wall, P. G. Allen, D. Sokaras, T.-C. Weng, D. Nordlund, M. A. Torrez, and J. Sarra, *Proc. Natl. Acad. Sci. U.S.A.* **109**(26), 10205 (2012).
- <sup>75</sup>F. Lin, D. Nordlund, T.-C. Weng, D. Sokaras, K. M. Jones, R. B. Reed, D. T. Gillaspie, D. G. J. Weir, R. G. Moore, A. C. Dillon, R. M. Richards, and C. Engtrakul, "Origin of Electrochromism in High-Performing Nanocomposite Nickel Oxide," *ACS Appl. Mater. Interfaces* (in press).
- <sup>76</sup>We use the "diced-and-bent" crystal notation here in order to emphasize that every single dice of the crystal analyzer is spherically bent, in contrary to the diced and flat analyzers where each dice is flat lying along a spherically bent substrate.