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Nano Lett. Author manuscript; available in PMC 2014 August 06.

## Published in final edited form as: *Nano Lett.* 2013 February 13; 13(2): 637–642. doi:10.1021/nl3043823.

# Tunable localized surface plasmon-enabled broadband light harvesting enhancement for high-efficiency panchromatic dyesensitized solar cells

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# Abstract

In photovoltaic devices, light harvesting (LH) and carrier collection have opposite relations with the thickness of the photoactive layer, which imposes a fundamental compromise for the power conversion efficiency (PCE). Unbalanced LH at different wavelengths further reduces the achievable PCE. Here, we report a novel approach to broadband balanced LH and panchromatic solar energy conversion using multiple-core-shell structured oxide-metal-oxide plasmonic nanoparticles. These nanoparticles feature tunable localized surface plasmon resonance frequencies and the required thermal stability during device fabrication. By simply blending the plasmonic nanoparticles with available photoactive materials, the broadband LH of practical photovoltaic devices can be significantly enhanced. We demonstrate a panchromatic dyesensitized solar cell with an increased PCE from 8.3% to 10.8%, mainly through plasmon-enhanced photo-absorption in the otherwise less harvested region of solar spectrum. This general and simple strategy also highlights easy fabrication, and may benefit solar cells using other photo-absorbers or other types of solar-harvesting devices.

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Author contributions

X.D., J.Q., P.T.H., and A.M.B. conceived the idea and designed the experiments. X.D. and J.Q. performed the synthesis. X.D., J.Q., and D.S.Y. performed the TEM imaging. M.T.K. and N.X.F. performed the simulation. X.D., M.T.K., and N.X.F. analyzed the simulation results. X.D., J.Q., and P.-Y.C. performed the fabrication and characterization of the solar cells. X.D., J.Q., P.T.H., and A.M.B. co-wrote the paper and all authors discussed the results and commented on the manuscript.

Supporting Information. Supplementary discussion, supplementary materials and methods, supplementary figures, supplementary tables, and supplementary references. This material is available free of charge via the Internet at http://pubs.acs.org.

#### Keywords

Tunable localized surface plasmon; light harvesting enhancement; panchromatic dye-sensitized solar cells; multiple-core-shell nanoparticles

Developing efficient and inexpensive solar cells is of paramount importance for preserving non-renewable energy and lowering carbon dioxide emissions. The power conversion efficiency (PCE) of the solar cells is generally governed by the tradeoff between light harvesting (LH) and carrier collection. To break this compromise, strategies of improving either LH or carrier collection while maintaining the other have been successful<sup>1, 2</sup>. However, most photo-absorbing materials possess unbalanced LH in high- and lowabsorption wavelength regions ( $\lambda_{Hi}$  and  $\lambda_{Lo}$ , figure 1a, b), which results in another tradeoff. At  $\lambda_{Lo}$ , long absorption lengths are required, which reduce carrier collection; at  $\lambda_{Hi}$ , solar energy is fully-absorbed by thin photoactive layers, which inefficiently exploit energy at  $\lambda_{Lo}$ (usually the red to near-infrared (NIR) region, figure 1c). Consequently, the ability to achieve broadband balanced LH is desirable for high-efficiency solar cells.

Achieving such broadband balanced LH would lead to panchromatic solar energy conversion, an ideal strategy for maximizing overall LH. As a promising solution-processed photovoltaic technology, dye-sensitized solar cells (DSSCs) have demonstrated PCE exceeding  $12\%^{3-8}$ ; to further improve the PCE, several approaches toward panchromatic DSSCs<sup>9</sup> have been attempted, including the development of panchromatic dyes<sup>10</sup>, co-adsorbing dyes<sup>11</sup>, and energy transfer systems<sup>12</sup>. However, these methods usually require extensive synthesis and optimization for various parameters (*e.g.*, spectral overlapping and spatial arrangement) and are limited to small ranges of materials. Therefore, a general and simple approach for broadband balanced LH and panchromatic DSSCs is required.

Localized surface plasmons (LSPs) are the elementary excitation states in noble metal nanoparticles (NPs), and can improve LH of photo-absorbers<sup>2, 13–18</sup>. Here, we introduce a general and simple strategy for broadband LH enhancement and panchromatic DSSCs by matching LSP resonance (LSPR) wavelength ( $\lambda_{LSPR}$ ) of plasmonic NPs with  $\lambda_{Lo}$  of existing photo-absorbers. Demonstrated by both simulations and experiments, matching  $\lambda_{LSPR}$  with  $\lambda_{Hi}$  or  $\lambda_{Lo}$  affects LH differently (figure 1d). Although matching  $\lambda_{LSPR}$  with  $\lambda_{Hi}$  readily improves LH and PCE for optically-thin photo-absorbing layers, matching  $\lambda_{LSPR}$  with  $\lambda_{Lo}$ maximally increases LH and PCE for practical photovoltaic devices with optically-thick layers, through enhancing photo-absorption in the weakly-absorbing region. The multiplecore-shell oxide-metal-oxide plasmonic NPs developed and utilized here are advantageous over other geometries, by featuring: adjustable  $\lambda_{LSPR}$  of 600–1,000 nm, substantially enhanced near-field electromagnetic (EM) intensity, and preservable plasmonic properties during device fabrication. By matching  $\lambda_{LSPR}$  with  $\lambda_{Lo}$  of a common dye (N719), a panchromatic DSSC with broadband balanced LH and a PCE of 10.8% is achieved (30% increase comparing to DSSCs without plasmonic NPs).

Synthesis and optical characterization are performed for the core-shell ( $d_{core}$ ~15 nm,  $d_{T}$ ~2 nm) Ag@TiO<sub>2</sub> (AgT) and Au@TiO<sub>2</sub> (AuT) and multiple-core-shell ( $d_{core}$ ~15 nm,  $d_{Au}$ ~0–4 nm, and  $d_{T}$ ~2 nm) TiO<sub>2</sub>-Au-TiO<sub>2</sub> (TAuT) NPs (figure 2a, S3–S6).  $d_{core}$ ,  $d_{Au}$ , and  $d_{T}$ 

represent core diameter, gold-shell thickness, and TiO<sub>2</sub>-shell thickness, respectively. When synthesizing the gold-shell of TAuT NPs, Au seeds gradually cover the core and coalesce to form a continuous layer, which is stable at 500°C (annealing condition for DSSCs). The outer TiO<sub>2</sub>-shells impede the metals from promoting recombination in the DSSCs<sup>2, 14</sup>. The  $\lambda_{\text{LSPR}}$  of AgT and AuT NPs are 420 and 550 nm respectively, and the  $\lambda_{\text{LSPR}}$  of TAuT NPs is continuously tunable from 1,000 to 600 nm by increasing  $d_{\text{Au}}$  to 4 nm (figure 2b). Tunable  $\lambda_{\text{LSPR}}$  across visible-NIR is thus achieved.

The interaction of the NPs with light was investigated using the finite-difference timedomain (FDTD) method. All plasmonic NPs strongly amplify the near-field EM intensity (figure 2c). Therefore, surrounding dye-molecules would experience a significantly increased light intensity near the LSPR frequency<sup>16, 19</sup> and a higher photon flux ( $\varphi_{\rm ph}$ ), which increases electron-hole pair generation. The enhancement factors (derived by integrating over near-field space, figure S7) of plasmonic NPs as functions of wavelength (figure 2d) show similar results of  $\lambda_{\rm LSPR}$  to those obtained from absorption spectra. The characteristic absorption peak of the TAuT NPs originates from the LSPs arising in the goldshell structure<sup>20–23</sup>. By tuning  $d_{Au}$  and  $d_T$  of TAuT NPs in 2–6 nm,  $\lambda_{LSPR}$  of 600–800 nm is achieved; as expected, thinner gold- and thicker TiO<sub>2</sub>-shells result in longer  $\lambda_{LSPR}$  (figure 2e, S8). Furthermore, an optimum  $d_{Au}$ ~3 nm maximizes the spectrum-integrated enhancement factor and thicker TiO2-shells tend to weaken the enhancement (figure 2f, the enhancement factor at  $\lambda_{LSPR}$  is shown in figure S9). The  $\lambda_{LSPR}$  and enhancement factors are also obtained for AgT and AuT NPs, which extend the LSP-enhanced spectral range to vis-NIR. These results represent the intrinsic near-field enhancement of plasmonic NPs and do not take the spectral response of dye-molecules into account.

The impact on the photo-absorption of surrounding dye-molecules is investigated by comparing the enhancement factors at different wavelengths (figure 2g, S10). At  $\lambda_{Lo}$ =600–800 nm (for N719), the greatest enhancement occurs by using TAuT NPs (*e.g.*,  $d_{Au}$ ~3 nm and  $d_{T}$ ~2 nm for 700 nm), where  $\lambda_{LSPR}$  matches  $\lambda_{Lo}$ ; at  $\lambda_{Hi}$ =400–600 nm, the maximum enhancement occurs by using AuT and AgT NPs, where  $\lambda_{LSPR}$  matches  $\lambda_{Hi}$ . The overall dye absorption is also maximally improved by AgT and AuT NPs (bottom part of figure 2g). These results and the simulated LSP-enhanced dye absorption ( $\lambda_{LSPR}$ =400–700 nm, figure S11) suggest that, on the level of individual plasmonic NP, matching  $\lambda_{LSPR}$  with  $\lambda_{Hi}$  enhances overall LH of dye-molecules more efficiently, whereas matching  $\lambda_{LSPR}$  with  $\lambda_{Lo}$  balances LH in a broad spectral range.

Moreover, the LSP-enhanced LH of N719-sensitized mesoporous TiO<sub>2</sub> films (figure 2h) agrees with the simulation results. AgT and AuT NPs improve LH mostly at  $\lambda_{Hi}$ , whereas TAuT NPs (TAuT-700 NPs with  $\lambda_{LSPR}$ =700 nm are used for all thin film LH and device performance characterizations) increase LH at  $\lambda_{Lo}$ , resulting in balanced spectra. The similar enhancements of simulated EM intensity and LH of thin films around  $\lambda_{LSPR}$  (supplementary discussion) confirm that the LH improvement stems from the interaction between dyemolecular dipoles and the LSP-enhanced near field and scattering cross-section of the NPs.

The multiple-core-shell oxide-metal-oxide plasmonic NPs are advantageous over other geometries (*e.g.* nanorods, nanodisks, and hollow nanoshells) which also possess tunable

 $\lambda_{\text{LSPR}}$  in vis-NIR<sup>20, 24</sup>. Under thermal treatment, the TAuT NPs maintain geometric and plasmonic properties with the rationally-designed templated metal-shell structure, while other geometries melt to spheres or spheroids and the characteristic  $\lambda_{\text{LSPR}}$  blue-shifts (figure 2i, S12, S13), severely reducing the ability to enhance LH at  $\lambda_{\text{Lo}}$ . In addition, little change in synthesis is required to tune  $\lambda_{\text{LSPR}}$  of TAuT NPs due to their high sensitivity to geometry (supplementary discussion). Moreover, metal-shell structures (*e.g.* TAuT, hollow gold-shell (figure S15, S16), and TAgT (figure S17, S18)) possess larger EM field enhancement than spherical solid metal structure (*e.g.* AuT and AgT). Therefore, the multiple-core-shell plasmonic NPs can benefit many LSP-enhanced applications which require tunable spectral response and high temperature fabrications. In addition, unlike the method using the interparticle coupling<sup>25</sup> to tune  $\lambda_{\text{LSPR}}$ , the current approach does not require sophisticated chemistry to control particle aggregation and the enhanced EM field uniformly surrounds the plasmonic NPs.

To study the LH of tunable LSP-enhanced DSSCs, different plasmonic NPs-incorporated TiO<sub>2</sub> photoanodes (thickness of 1–20 µm and plasmonic NPs-TiO<sub>2</sub> ratio of 0.01–3.2 wt%) are assembled into DSSCs<sup>1, 2</sup>. In 1.5-µm-thick optically-thin photoanodes, AgT, AuT, and TAuT NPs increase the incident photon-to-current conversion efficiency (IPCE) at  $\lambda_{max}$  (maximum absorption wavelength~530 nm) by 45%, 60%, and 50%, respectively (figure 3a-left). Enhancement is maximized when  $\lambda_{LSPR}$  is closest to  $\lambda_{max}$ . Similarly, at 700 nm, TAuT NPs increase IPCE the most by 80%, whereas AgT and AuT NPs enhance IPCE only by 21% and 22% due to large mismatch between  $\lambda_{LSPR}$  and  $\lambda_{Lo}$ . By correlating the absorbance, LH efficiency (LHE), and IPCE (supplementary discussion), we find that the experimental results of absorbance and IPCE enhancement are similar, which indicate that the IPCE improvement mainly arises from LSP-enhanced LH.

While IPCE (specifically LHE) of thin photoanodes is readily increased by LSPs, opticallythick photoanodes (10~15 µm) for practical devices are required to ensure balance between LH and carrier collection for maximized PCE<sup>26</sup>, where the LHE at  $\lambda_{\text{Hi}}$  and internal quantum efficiency are close to unity. At  $\lambda_{\text{max}}$ , less than 5% enhancement of IPCE for all plasmonic NPs is observed (figure 3a-middle, supplementary discussion). In contrast, at  $\lambda_{\text{Lo}}$ , the TAuT NPs increase IPCE by up to 100%, whereas AgT and AuT NPs increase IPCE slightly. Additionally, the IPCE increases monotonically with the concentration of TAuT NPs (figure 3a-right).

Therefore, both experiments and simulations have demonstrated that matching  $\lambda_{LSPR}$  with  $\lambda_{Hi}$  or  $\lambda_{Lo}$  impacts the spectral response of LH differently. Matching  $\lambda_{LSPR}$  with  $\lambda_{Hi}$  enhances LH in already strongly-absorbing region, which benefits optically-thin photoanodes. In contrast, matching  $\lambda_{LSPR}$  with  $\lambda_{Lo}$  improves the weakly-absorbed part of solar spectrum, which is of great importance for broadband balanced LH in practical panchromatic solar cells.

The PCEs of tunable LSP-enhanced DSSCs are measured (figure S19, table S1) (PCE= $J_{SC}V_{OC}FF/P_{in}$ ,  $J_{SC}$ ,  $V_{OC}$ , FF, and  $P_{in}$  represent short-circuit current density, opencircuit voltage, fill factor, and incident power density, separately).  $J_{SC}$  (= $q \int IPCE(\lambda) \varphi_{ph}(\lambda) d\lambda$ ) is improved with increasing concentrations of plasmonic NPs (within the

concentration range studied); the largest  $J_{SC}$  is achieved by incorporating the highest concentration of TAuT NPs (figure 3b). Besides the improved LH, thinner photoanodes for optimized PCEs (the optimized thicknesses for TiO<sub>2</sub>, AgT, AuT, and TAuT DSSCs are 13.2, 10.9, 10.1, and 11.0 µm) are also responsible for the improved IPCE and  $J_{SC}$ , by improving the carrier collection (electron diffusion lengths are not changed significantly, figure S22).

Additionally,  $V_{OC}$  is increased by introducing plasmonic NPs (figure 3c). Generally,  $V_{OC}$  is limited by the material properties of electronic structure (*e.g.*, the quasi-Fermi level of TiO<sub>2</sub> and redox potential of electrolyte); enhancing  $V_{OC}$  usually requires exploiting new materials<sup>3</sup>. Two reasons could be responsible for the LSP-induced  $V_{OC}$  enhancement: thinner optimized photoanodes reduce the voltage loss from charge recombination; and the quasi-Fermi level is lifted due to the equilibrium between quasi-Fermi level of TiO<sub>2</sub> and LSP energy level of plasmonic NPs<sup>15, 27</sup>. Our result is also in agreement with a recent study on plasmoelectronics<sup>28</sup>. In addition, we observe that TAuT NPs increase  $V_{OC}$  less significantly than AgT and AuT NPs, which is likely due to lower LSP energy level (supplementary discussion). The different  $V_{OC}$  enhancements could assist in the elucidation of the origin of LSP-enhanced  $V_{OC}$ , and the ability of rationally increasing  $V_{OC}$  could further improve PCE. In addition, although we suggest that the  $V_{OC}$  increase is due to quasi-Fermi level equilibrium, we do not yet have direct evidence of charge transfer from metal to TiO<sub>2</sub>. Regarding this subject, we plan to perform further investigation.

Since LSPs have a larger effect on enhancing LH than changing  $V_{OC}$  and FF (no significant impact on FF is observed, figure S23), the PCEs are improved (figure 3d) with increasing concentrations of NPs (0–3.2 wt%, higher concentrations could reduce performance, supplementary discussion). Thus, maximum PCEs of 10.1%, 10.3%, and 10.8% are achieved for AgT, AuT, and TAuT NPs-incorporated DSSCs respectively, comparing to 8.3% without plasmonic NPs.

The optimized device performance helps distinguish the improvement from different plasmonic NPs. AgT and AuT NPs used here and similar materials in previous studies on LSP-enhanced DSSCs<sup>2, 14–16</sup> improve IPCE and PCE, mainly through improved LH at  $\lambda_{Hi}$ . Although the previous approach is responsible for achieving ultrathin solar cells<sup>29</sup>, it could not break the compromise imposed by unbalanced LH at  $\lambda_{\rm Hi}$  and  $\lambda_{\rm Lo}$ . In contrast, our general and simple strategy of matching  $\lambda_{LSPR}$  with  $\lambda_{Lo}$  balances and optimizes broadband LH, achieves panchromatic DSSCs with the existing dye, and further enhances LH,  $J_{SC}$ , and PCE. In fact, the distribution of core sizes and shell thicknesses result in the broad absorption spectrum of TAuT NPs with  $\lambda_{LSPR} \sim \lambda_{Lo}$  (figure 2b), which benefits panchromatic LH and carrier collection simultaneously. Our approach toward panchromatic DSSCs can apply to other photo-absorbers and other types of solar cells. For instance, a PCE of 12% was achieved by using a porphyrin dye in the DSSC<sup>3</sup>. We believe that the same strategy can be used to improve the overall LH by matching the  $\lambda_{\text{LSPR}}$  with the  $\lambda_{\text{Lo}}$  and a similar degree of improvement would be observed. Furthermore, since LH is the initial stage of a series of physical, chemical, or biochemical processes in many solar energy conversion devices, our strategy of matching  $\lambda_{LSPR}$  with  $\lambda_{Lo}$  may benefit many other devices<sup>30</sup> including artificial photosynthesis<sup>18</sup>, solar heating, solar thermal electricity, and solar fuels<sup>31</sup>.

# Supplementary Material

Refer to Web version on PubMed Central for supplementary material.

## Acknowledgments

This work was supported by Eni, S.p.A (Italy) through the MIT Energy Initiative Program. N.X.F. acknowledges support by NSF (ECCS Award No. 1028568) and by the Air Force Office of Scientific Research (AFOSR MURI Award No. FA9550-12-1-0488). M.T.K acknowledges support from the MIT Energy Initiative Eni-MIT Energy Fellowship.

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Figure 1. Spectral response of the LH of a solar cell and tunable LSP-enhanced broadband LH a, b, the absorption coefficient of bulk silicon (a) and extinction coefficient of N719 (b); the shaded areas are low LH regions, which require thicker photoactive layers to achieve efficient LH. c, a schematic of the spectral response of a solar cell. The solar energy is less utilized at  $\lambda_{Lo}$  (usually red-NIR). d, illustrations of enhancing EM intensity by AgT (yellow), AuT (purple), and TAuT (blue) plasmonic NPs.  $\lambda_{LSPR}$  of AgT and AuT overlaps with  $\lambda_{Hi}$  of N719, maximizing the effect of LSP-enhanced LH.  $\lambda_{LSPR}$  of TAuT matches  $\lambda_{Lo}$  of N719, balancing LH at different wavelengths.



Figure 2. Synthesis, optical characterization, and FDTD simulation of AgT, AuT, and TAuT plasmonic NPs

**a**, illustrations and transmission electron microscope (TEM) images of AgT, AuT, and TAuT NPs. The scale bars are 5 nm. **b**, photograph picture and absorption spectra of AgT, N719, AuT, TAuT-590 ( $\lambda_{LSPR}$ =590 nm), TAuT-700, and TAuT-810 in ethanol solutions. **c**, simulated EM intensity enhancement ( $|E|^2/|E_0|^2$ ) in near field at  $\lambda_{LSPR}$  for AgT, AuT, and TAuT NPs. The inner circles (white and yellow) represent different layers of NPs, and the outermost circles (cyan) represent the volume of integration.  $d_{core}$ =15 nm (used for all simulations unless specified),  $d_{Au}$ =3 nm, and  $d_T$ =2 nm. **d**, enhancement factor as a function of wavelength for AgT, AuT, and TAuT NPs with different  $d_{Au}$  ( $d_T$ =2 nm). **e**, **f**, simulated  $\lambda_{LSPR}$  (**e**) and spectrum-integrated enhancement factor over 300–900 nm (**f**) for AgT, AuT, and TAuT NPs as a function of  $d_{Au}$  (of TAuT) and  $d_T$  (of AgT, AuT, and TAuT). **g**, enhancement factor at different wavelengths (top) and enhancement of overall dye absorption (bottom) for AgT, AuT, and TAuT with different  $d_{Au}$  ( $d_T$ =2 nm). **h**, absorption of N719 in sensitized 3- $\varphi$ m-thick TiO<sub>2</sub> films improved by AgT (1.0 wt%), AuT (1.8 wt%), and TAuT (3.2 wt%) NPs. **i**, absorption spectra and photograph images of thin films of Au

nanocage @TiO<sub>2</sub> (AuNCT, left) and TAuT-700 (right) on glass substrates after 500°C annealing; TAuT NPs maintain optical properties, whereas  $\lambda_{LSPR}$  of AuNCT blue-shifts toward  $\lambda_{LSPR}$  of AuT NPs.

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#### Figure 3. Tunable LSP-enhanced LH and photovoltaic performance of DSSCs

**a**, IPCE spectra of DSSCs with AgT, AuT, and TAuT NPs-incorporated photoanodes of 1.5  $\varphi$ m thickness (left), of optimized thickness for maximum PCE (middle), and as a function of concentration of TAuT NPs in optimized TAuT-DSSCs (right). **b**, **c**, **d**,  $J_{SC}$  (**b**),  $V_{OC}$  (**c**), and PCE (**d**) of LSP-enhanced DSSCs with AgT (left), AuT (middle), and TAuT (right) NPs-incorporated photoanodes, as a function of the concentration of plasmonic NPs (0–3.2 wt%) and thickness of photoanodes (1–20  $\varphi$ m). The particle densities for all plasmonic NPs are

the same and in the range of  $8.4 \times 10^{12}$ – $8.4 \times 10^{14}$  cm<sup>-3</sup>, and the difference in weight percent is due to the different geometries of NPs (supplementary discussion).