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DNA Brick Crystals with Prescribed Depth

Yonggang Ke^{1,3,4,†,‡,*}, Luvena L. Ong^{1,5,†}, Wei Sun^{1,2,†}, Jie Song⁶, Mingdong Dong⁶, William M. Shih^{1,3,4}, and Peng Yin^{1,2,*}

¹Wyss Institute for Biologically Inspired Engineering, Harvard University, Boston, Massachusetts 02115, USA

²Department of Systems Biology, Harvard Medical School, Boston, Massachusetts 02115, USA

³Department of Cancer Biology, Dana-Farber Cancer Institute, Harvard Medical School, Boston, Massachusetts 02115, USA

⁴Department of Biological Chemistry and Molecular Pharmacology, Harvard Medical School, Boston, Massachusetts 02115, USA

⁵Harvard-MIT Division of Health Sciences and Technology, Massachusetts Institute of Technology, Cambridge, Massachusetts 02139, USA

⁶Center for DNA Nanotechnology at the Interdisciplinary Nanoscience Center (iNANO), Aarhus University, 8000 Aarhus C, Denmark

Abstract

We describe a general framework for constructing two-dimensional crystals with prescribed depth and sophisticated three-dimensional features. These crystals may serve as scaffolds for the precise spatial arrangements of functional materials for diverse applications. The crystals are self-assembled from single-stranded DNA components called DNA bricks. We demonstrate the experimental construction of DNA brick crystals that can grow to micron-size in the lateral dimensions with precisely controlled depth up to 80 nanometers. They can be designed to display user-specified sophisticated three-dimensional nanoscale features, such as continuous or discontinuous cavities and channels, and to pack DNA helices at parallel and perpendicular angles relative to the plane of the crystals.

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*py@hms.harvard.edu; yonggang.ke@emory.edu.

[†]Equal contribution authors: Ke, Ong, Sun

[‡]Present address: Wallace H. Coulter Dept. of Biomedical Engineering, Georgia Institute of Technology and Emory University, Atlanta, GA 30322

Author contributions

Y.K., L.L.O. and W.S. made equal contributions. Y.K. conceived the project, designed and performed the experiments, analyzed the data, and wrote the paper; L.L.O. and W.S. designed and performed the experiments, analyzed the data, and wrote the paper; J.S. and M.D. performed the cryo-EM and AFM experiments, analyzed the data, and wrote the paper; W.M.S. discussed the results and wrote the paper; P.Y. conceived, designed, and supervised the study, interpreted the data, and wrote the paper.

Additional information

Supplementary information is available in the online version of the paper. Reprints and permission information is available online at <http://www.nature.com/reprints>. Correspondence and requests for materials should be addressed to P.Y. and Y.K.

Competing financial interests

The authors declare competing financial interests: a provisional US patent application has been filed.

The production of two-dimensional materials, particularly crystals with prescribed depths and intricate three-dimensional (3D) features, provides an enabling platform for nanofabrication. For example, these two-dimensional (2D) crystals could be integrated with inorganic nanomaterials for developing complex nanoelectronics¹ and photonics systems.^{2,3} Although thin film structures have been created using either electron/ion beam lithography³ or self-assembly of block co-polymers,^{4,5} fabricating two-dimensional materials that simultaneously achieve precisely tunable thickness, and prescribed complex surface and internal features (e.g. channels or pores) with sub-5 nm resolution remains challenging.^{3,6–8}

A promising route to address this challenge is structural DNA nanotechnology.⁹ DNA has been used to create complex discrete shapes^{9–25} and extended periodic crystals,^{26–38} including ribbons,³³ tubes,^{27,32,33,35} two-dimensional crystals,^{18,26–32,36–38} and three-dimensional crystals.³⁴ DNA structures can serve as scaffolds for precise patterning of functional moieties (e.g. gold nanoparticles) for electronics and photonics applications.^{35,39,40} However, in contrast to current organic polymeric films,⁴¹ the two-dimensional DNA crystals are typically restricted to a single-layer of DNA helices with about 2 nanometer depth. A 3D crystal was previously reported but it grows in all three dimensions with no control in depth and uses a small triangular repeating unit.³⁴ One major categorical gap in constructing atomically precise DNA structures – and, more generally, synthetic molecular structures – is the lack of a general framework for making complex 2D crystals with precisely controlled depth and sophisticated three-dimensional features. Successful construction of such structures could enable a wide range of applications ranging from nanoelectronics and plasmonics to biophysics and molecular diagnosis.

Using single-stranded DNA bricks,^{21,22,33} we describe here a simple, robust, and general approach to engineer complex micron-sized two-dimensional crystals with prescribed depths and complex three-dimensional features with nanometer resolution. In previous reports,^{26–32,34–38} DNA crystals are typically formed via a two-stage hierarchical process: individual strands first assemble into a discrete building block (often known as a DNA tile) and individual tiles then assemble into crystals. In contrast, DNA brick crystals grow non-hierarchically: the growth of DNA crystals from short, floppy, single-stranded DNA bricks does not involve the assembly of pre-formed discrete multi-stranded building blocks with well-defined shapes. During the brick crystal growth, assembly and disassembly occur by relatively weak intermolecular interactions involving addition or subtraction of a single short strand at a time.

We constructed a total of 32 DNA brick crystals. These crystals can grow up to several microns in the lateral dimensions with a prescribed depth up to 80 nanometers, and display sophisticated user-specified nanometer scale three-dimensional features, including intricate cavities, channels, and tunnels (Supplementary Fig. S1). Additionally, the non-hierarchical nature of the assembly permits isothermal formation of the crystals. We illustrated the scaffolding utility of these crystals by functionalizing them with parallel arrays and layers of tightly-packed (1–2 nm spaced) gold nanoparticles.

Design and assembly of DNA-brick crystals

Crystal design is based on previous discrete three-dimensional DNA-brick structures.²² A DNA brick is a 32-nucleotide (nt) strand with four 8-nt binding domains, and can be modeled as a LEGO-like brick (Fig. 1a). In a one-step annealing reaction, DNA bricks – each with a distinct sequence – assemble into a prescribed structure by binding to their designated neighbors. Implementing “connecting” bricks between discrete structures yields DNA-brick crystals. The design strategy is illustrated using a 6H (helix) \times 6H (helix) \times 24B (basepair) cuboid structure that can be programmed to grow along three orthogonal axes (Fig. 1b). To achieve homo-multimerization along the Z-axis (i.e. parallel to helical axes), the first layer of domains are modified to be complementary to the last layer of domains. Homomultimerization along the X-axis or Y-axis is achieved by including bricks that each has two of its domains bound to one face of the cuboid and the other two domains to the opposing face. See Supplementary Fig. S2 for detailed strand connecting patterns. The crystals are designed to form via non-hierarchical growth: individual bricks (rather than pre-formed multi-brick blocks) are directly incorporated into the crystal (Fig. 1c).

We constructed four groups of crystals: (1) Z-crystals: one-dimensional “DNA-bundle” crystals extending along the Z-axis (Fig. 1d); (2) X-crystals: one-dimensional crystals extending along the X-axis; (3) ZX-crystals: two-dimensional “multilayer” crystals extending along the Z-axis and the X-axis (Fig. 1e); (4) XY-crystals: two-dimensional “DNA-forest” crystals extending along the X-axis and Y-axis (Fig. 1f). Using different designs of repeating units, DNA crystals with prescribed depths and features, such as pores, channels, and tunnels, can be made (Fig. 1g to 1i). Here we define a channel as a surface-exposed cavity extending across multiple repeating units, a pore as a hole across a single repeating unit, and a tunnel as a series of concatenated pores. A crystal is named as “[the growth direction(s)]-[the dimensions of the repeating-unit]-[the shape of the unit]”. For instance, an “XY6H \times 6H \times 24B-cuboid” crystal is a two-dimensional XY-crystal with a cuboid-shaped 6H \times 6H \times 24B repeating unit. Like discrete DNA-brick structures,²² the sequences for DNA-brick crystals were randomly generated. All crystals used 10.67 base-pair (bp)/turn reciprocal twist density which is slightly under-wound compared to the 10.5 bp/turn of natural B-form DNA.

Each crystal was assembled by mixing 100 nM of each unpurified DNA brick strand in 40 mM MgCl₂, without careful adjustment of strand stoichiometry. After 72-hour or 168-hour one-pot annealing, assembled crystals were imaged using transmission electron microscopy (TEM), cryo-electron microscopy (cryo-EM), or atomic force microscopy (AFM) without further purification. See Methods for details.

One-dimensional DNA-bundle crystals (Z-crystals)

Both solid Z-crystals (Fig. 2a–f) and Z-crystals with tunnels (Fig. 2g–i) were successfully constructed.

Solid Z-crystals with different cross-sectional shapes

We first constructed three solid Z-crystals with distinct square-shaped cross-sections (6H×6H×32B, 8H×8H×32B, and 10H×10H×32B; Fig. 2a–c). We then demonstrated crystals with more complex cross-sections: a Z-8H×8H×128B-spiral crystal with a surface helical channel along the Z-axis (Fig. 2d), a Z-43H×32B-triangle crystal (Fig. 2e), and a Z-44H×32B-hexagon crystal (Fig. 2f). The spiral channel was clearly visible in the TEM image of the Z-8H×8H×128B-spiral crystal. However, many broken structures were also observed for this spiral crystal (Supplementary Fig. S6).

Z-crystals with tunnels

Three Z-crystals with tunnels were tested (Fig. 2g–i). The cross-section of the Z-56H×32B-tunnel is an 8H×8H square with a 2H×4H rectangle removed from the center (Fig. 2g). The Z-108H×32B-tunnel has a 12H×12H square cross-section with a 6H×6H hole (Fig. 2h). The Z-60H×64B-tunnel crystal contains a 2H×2H tunnel along the Z-axis and 8H×2H×24B pores that intersect the 2H×2H tunnel every 64bp along the Z-axis (Fig. 2i). TEM images of the Z-60H×64B-tunnel showed many splintered structures containing only half of the designed DNA helices, likely reflecting the weakening effect of the periodic 8H×2H×24B pores on the connections between the top and bottom halves of the structures along the Y-axis.

All Z-crystals displayed a global right-handed twist, which likely resulted from the stress generated by the underwound design.^{17,42} Zoomed-out TEM images of Z-crystals are included in Supplementary Figs. S3 to S11.

One-dimensional X-crystals

We constructed two 1D-crystals that extended along the X-axis: an X-6H×6H×64B-cuboid crystal (Fig. 2j) and an X-32H×64B-pore crystal (Fig. 2k). Both appeared well-formed and grew up to a few hundred nanometers in length in TEM images. See Supplementary Fig. S12 for larger images.

Two-dimensional DNA-multilayer crystals (ZX-crystals)

Solid ZX-crystals (Fig. 3a–d), ZX-crystals with channels, pores, and tunnels (Fig. 3e–h), and an offset ZX-crystal (Fig. 3i) were successfully constructed.

Solid ZX-crystals

Four solid ZX-crystals were designed from 4H×32B units that contained 4, 6, 10, and 20 layers of helices, respectively (Fig. 3a–d). The thickness of each ZX-crystal was directly measured at the crease where the crystal folded over onto itself in TEM images. The thicknesses of the 4, 6, 10, and 20-layer ZX-crystals were measured to be approximately 10, 15, 25 and 50 nm, respectively, proving all layers of the crystals were completely formed and that the width of each DNA helix was about 2.5 nm in diameter.

ZX-crystals with channels, pores, and tunnels

We first designed three ZX-crystals from a 6H×6H×32B cuboid unit (Fig. 3e–g). Four helices were removed from the cuboid to generate ZX-32H×64B-channel (Fig. 3e). The

second design, ZX32H×64B-cross-channel, was obtained by further removing a perpendicular 2H×32B channel from ZX-32H×64B-channel (Fig. 3f). The third design, ZX-6H×6H×64B-pore, contained a 2H×4H×32B vertical pore along the Y-axis in each cuboid unit (Fig. 3g). This design yielded narrow and long crystals. The most complex ZX-crystal design is the ZX-96H×64B-crosstunnel crystal (Fig. 3h). Its repeating unit can be considered as a 10H×10H×64B cuboid with a 2H×2H×64B pore along the Z-axis and with a 10H×2H×24B pore along the X-axis. In this structure, the design contained two types of parallel tunnels separated by two layers of DNA. These tunnels appeared perpendicular when viewed from the ZX projection.

Offset 2D ZX-crystal

We constructed a ZX-crystal that extended 6H×6H×64B-cuboid repeating units along the Z-axis and X-axis using an “offset-register” scheme: the crystal’s Z-axis extension was shifted 4H along the X-axis; the crystal’s X-axis extension was shifted 32B along the Z-axis (Fig. 3i).

All ZX-crystals showed a small amount of right-handed twist, which also likely arose from the 10.67 bp/turn under-wound design. As a result, we observed that the crystals sometimes appeared twisted and folded on top of themselves in TEM images. We also observed that all ZX-crystals grew faster along the Z-axis than the X-axis, consistent with observations from previous crystal growth studies.²⁶ See Supplementary Figs. S13 to S21 for more TEM images.

Two-dimensional DNA-forest crystals (XY-crystals)

We constructed solid XY-crystals (Fig. 4a–d), XY-crystals with pores and surface channels (Fig. 4e–i), and XY-crystals that form a tube structure (Fig. 4j).

Solid XY-crystals

Solid XY-crystals of variable thicknesses constructed (Fig. 4a–d). Using a 4H×4H cuboid unit of various heights, we constructed four XY-crystals with 64B (21 nm), 128B (42 nm), 192B (63 nm), and 256B (84 nm) designed thickness.

XY-crystals with pores and channels

XY-32H×64B-pore and XY-32H×128B-pore crystals were constructed (Fig. 4e, h). Both designs contained periodic 2H×2H pores separated by 4H in each dimension. The two crystals resemble 21 and 42 nm porous membranes, respectively. Cryo-EM imaging was applied to these two crystals (Supplementary Fig. S22). Based on 3D reconstruction data, their depths were measured to be 26±2 nm (32H×64B-pore) and 45±3 nm (32H×128B-pore), respectively (Fig. 4g, h), in good agreement with the theoretical values (21 nm and 42 nm). The slight discrepancy is likely because theoretical estimations did not account for the single-stranded poly-T at the ends of the duplex. The depth of these two crystals were also measured by AFM to be approximately 16 nm and 36 nm, respectively (Supplementary Fig. S23). The smaller depths obtained from AFM likely reflects the compression of the crystal by the cantilever.

An XY-4H×8H×96B-channel crystal was constructed (Fig. 4g). It contained a solid 64B (42 nm) base and parallel channels. The channels are 4H (10 nm) in width and 32B (21 nm) in height, and are separated by 4 layers of helices.

A tube-shaped XY-crystal

An XY-4H×4H×32B-tube crystal (Fig. 4h) was designed using the same strategy as the other XY-crystals. However, when assembled in 40 mM MgCl₂, this thin 32B (10.6 nm) XY-crystal formed a tube (Supplementary Figs. S24 and S25) instead of a flat 2D crystal, likely due to the uneven distribution of connections between helices (Supplementary Fig. S24).

Annealing the XY-4H×4H×32B-tube at higher MgCl₂ concentration produced tubes with larger diameters, presumably due to a greater reduction in repulsion between negatively-charged DNA helices. At 60 mM MgCl₂, we observed many tubes with diameters between 140 and 300 nm (Supplementary Fig. S26). To further test our hypothesis that the tube formation is caused by the asymmetric distribution of crossovers, we designed an XY-4H×4H×32B-cuboid crystal in which the DNA bricks are arranged in an alternating fashion between layers.²² Connections between helices in this design were symmetrically distributed along both the X-axis and Y-axis (Supplementary Fig. S65). This alternating design produced only flat crystal structures (Supplementary Fig. S27). Additionally, the thicker 64, 128, 192, and 256B XY-crystals with non-alternating designs had 2, 4, 6, and 8 connections between each pair of neighboring helices, respectively. No visible curvature was observed for these designs in TEM images (Supplementary Figs. S28 to S31).

General observations

Unlike the ZX-crystals, the XY-crystals did not show global right-handed twist. The lack of global twist can be explained by the following analysis. For simplicity, assuming an XY-crystal forms a perfect cylinder containing n helices. The overall twist in radians of the cylinder is $\theta = TL/JG$, where T is the applied torque resulting from the underwound design, L is helix length, G is the modulus rigidity of a helix, and J is the torsion constant. The first three parameters can be considered as constants. The torsion constant J for a cylinder as a function of cross-sectional (XY-plane) radius can be approximated as $\pi^4/2 (r^4$ is proportional to n^2), where r is the circular cross-sectional radius. Thus, θ is inversely proportional to n^2 . As an XY-crystal grows to include a large number of helices, its global twist θ rapidly becomes negligible.

As both growth directions (X-axis and Y-axis) for an XY-crystal is perpendicular to the DNA helical axis, the XY-crystal grew in an isotropic fashion and did not exhibit any apparent directional preference. Because crystal growth along the X-axis or Y-axis is slower than growth along Z-axis (as discussed for ZX-crystals), the 72-hour annealing often produces either no XY-crystals or XY-crystals smaller than one hundred nanometers in the lateral dimensions. Therefore, we used a 168-hour annealing ramp for all XY-crystals.

XY-crystals provide a particularly attractive platform for DNA-directed guest molecule assembly: the surface of an XY-crystal can be considered as a “breadboard”¹³ on which

guest molecules can be conveniently attached to the ends of DNA helices at 2.5 nm resolution in the lateral plane.

See Supplementary Figs. S25 to S34 for more TEM images of XY-crystals.

Crystal growth mechanism

The non-hierarchical growth mechanism of DNA brick crystals is verified by the following experiments: (1) Analysis of the boundaries of DNA brick crystals based on high-resolution TEM images revealed that these crystals lack well defined, complete structural repeating units on their boundaries; (2) Annealing profile of the formation of the crystals revealed a single transition temperature (rather than two or more transition temperatures characteristic of hierarchically formed DNA crystals^{32,36}); (3) Time-lapse analysis of the TEM images and gel electrophoresis of the assembly process revealed the absence of repeating units of designed size.

Boundary analysis

We studied the edge of the XY-32H×64B-pore crystal by counting the number of units that match with that of the designed repeating unit (Supplementary Fig. S35). In hierarchically-assembled crystals, the designed repeating units are first formed before they are assembled into a crystal. As a result, the edges of the crystal often consist of the shape of the designed repeating unit (e.g. >90% of the edges of a previously reported hierarchically-assembled crystal³⁶ [Supplementary Fig. S36]). In contrast, only 2% of the edges of the XY-32H×64B-pore brick crystal match that of the designed shape (Supplementary Fig. S37), consistent with a non-hierarchical assembly mechanism. See Supplementary Sect. S6.1 for more details.

Annealing curve

To study the annealing profile of the brick crystals, we assembled the ZX-4H×20H×32B and XY-32H×64B-pore crystals in the presence of SYBR Green I (Supplementary Figs. S38 and S39). Annealing curves for both brick crystals depict a single, sharp transition peak, which occurred around 40°C for the ZX-crystal (Supplementary Fig. S38b, c) and 30°C for the XY-crystal (Supplementary Fig. S39b, c), consistent with a non-hierarchical assembly process where individual component strands are directly incorporated into the lattices. See Supplementary Sect. S6.2 for more details.

Time lapse analysis with gel electrophoresis and TEM

Time-points before, at, and after the transition temperature were sampled for both the above two ZX-and XY-crystals to confirm the lack of well-formed, discrete repeating units. Gel electrophoresis showed no intermediate band between the well and monomers (Supplementary Figs. S38d, S39d), as expected. Although TEM imaging revealed some small, randomly sized and shaped clusters of strands, no uniformly sized and shape-defined structures were observed, supporting the designed non-hierarchical assembly mechanism (Supplementary Figs. S38e, S39e). See Supplementary Sect. S6.2 for more details.

Isothermal assembly

The single transition temperature of the DNA-brick crystallization suggests that these crystals should be able to assemble isothermally. Thus, we assembled a ZX-6H×4H×96B-cuboid crystal isothermally at the observed annealing transition temperature of 33°C (Fig. 5a and Supplementary Fig. S40c). Gel electrophoresis showed no discrete monomer band during annealing (Supplementary Fig. S40d). Time-lapse TEM images further supported the non-hierarchical assembly mechanism: structures around 100 nm in size appeared after 4 hours and rapidly grew to micron-sized structures by 48 hours (Fig. 5c). See Supplementary Sect. S6.3 for more details.

Yield and defect analysis

We used crystal deposition density and strand depletion ratio to study the approximate yield of the brick crystals: after four-times dilution, an isothermally-assembled ZX-6H×4H×96B sample produced 0.23 structures per μm^2 on a TEM grid (Supplementary Fig. S41); Förster Resonance Energy Transfer analysis of dye-labeled ZX-6H×6H×64B sample suggests 80% strand depletion ratio (Supplementary Figs. S42). Analysis of the pore morphology of XY-32H×64B-pore TEM images indicates a 9% defect rate (Supplementary Fig. S43). See Supplementary Sect. S7 for more details.

Patterning gold nanoparticles

Gold nanoparticles have been previously arranged into discrete patterns^{39,40,43} and single-layer periodic patterns³⁵ using DNA structures as templates. However, it remains challenging to form close-packed periodic patterns, especially multi-layer patterns, of gold nanoparticles. This challenge is addressed here using DNA crystals. We constructed two close-packed gold nanoparticle superstructures on DNA-brick crystals: (1) parallel lines of gold particles arranged on a ZX-4H×6H×96channel crystal (Supplementary Fig. S44) spaced on average 2 nm apart (Fig. 6a, b), and (2) parallel gold-nanoparticle monolayers, with each particle spaced on average 1 to 2 nm apart, on an XY-4H×4H×64-cuboid (Fig. 6c–e). Aligning gold nanoparticles into micron-scale ordered arrays is required in diverse plasmonic applications. In particular, nanoparticle arrays with sub-2-nm face-to-face spacing are expected to exhibit strong plasmonic coupling.⁴⁴ See Supplementary Sect. S8 for more experimental details.

Discussion

Crystallization of increasingly large macromolecular complexes can be challenging. Traditionally the process is hierarchical, i.e. the complexes incorporate into the crystal as preformed monomeric units.^{26–32,34–38} However, the homogeneity of complex monomers is often difficult to ensure, and the addition of a defective monomer can compromise the growth of a well-ordered crystal. Furthermore, the kinetics of joining large monomers can be slow because of their size, especially in the case when they repel each other due to like charges. Finally, effective crystallization is thought to require error correction that involves near-equilibrium incorporation and dissociation of monomers to erase defects on the path to the lowest energy state. Therefore, if the strength of the interaction between monomers is

too strong, which is more likely for large structures, irreversible rather than reversible self-assembly may dominate, resulting in defective crystals.

In contrast, our DNA-brick framework explores a non-hierarchical route to crystallization that achieves reversible assembly of rapidly diffusing subcomponents without sacrificing the complexity of the larger repeating unit in the final crystals. Using a modular strategy that utilizes standardized components – such as our DNA bricks – a repeating unit can contain thousands of base pairs, enabling the implementation of designs bearing intricate features. Although we define a repeating unit for ease of design, there is no difference between bricks within each repeating unit and those that connect the repeating units, hence bricks can be added to or subtracted from the growing crystal one at a time. A corollary is that the boundary where one repeating unit begins and the next one ends is arbitrary.

Microscale two-dimensional DNA-brick crystals with prescribed depth and three-dimensional nanoscale features provide a new platform for developing diverse applications. In biophysics, host-guest two-dimensional DNA crystals could facilitate cryo-electron microscopic imaging and 3D reconstruction of protein structures.⁴⁵ Thicker two-dimensional crystals are more rigid and resistant to dynamic thermal fluctuations, providing better translational alignment of proteins; their 3D cavities could potentially hold the guest molecule in a specific conformation, providing better orientational alignment within each unit cell. In photovoltaics, bottom-up self-assembly of nanomaterials, such as nanowires,⁴⁶ provides a promising route to producing high efficiency photovoltaic devices. DNA-brick crystals may prove useful for self-assembling 3D nanomaterials with nanometer precision, thus achieving enhanced cooperation effects and energy conversion efficiency for photovoltaic devices. In nanofabrication, these two-dimensional DNA crystals may serve as etching masks for fabricating diverse inorganic materials based devices and circuits, providing better shape programmability than current block copolymer based masks.⁵

Methods

Design diagrams

DNA strand diagrams are included in Supplementary Sect. S9 (Supplementary Figs. S45 to S76).

Sample preparation

DNA strands were synthesized by Integrated DNA Technology, Inc. (www.idtdna.com). To assemble the structures, unpurified DNA strands were mixed in an equimolar stoichiometric ratio to the highest possible concentration from a 100 μ M stock in 0.5 \times TE buffer (5 mM Tris, pH 7.9, 1 mM EDTA) supplemented with 40 mM MgCl₂.

Annealing ramps and isothermal assembly

The strand mixture was then annealed in a PCR thermo cycler using a fast linear cooling step from 80°C to 60°C over 1 hour, then a 72-hour or 168-hour linear cooling ramp from 60°C to 24°C. The annealing ramps were named according to length of the second cooling step, as 72-hour annealing or 168-hour annealing. Most ZX-crystals were folded using the

72-hour annealing, except the ZX-96H×64B cross-tunnels which used the 168-hour annealing. All XY-crystals were folded with the 168-hour annealing. For isothermal assembly of DNA crystals, the sample solution was incubated at 33°C for up to 48 hours.

DNA modification of 10-nm gold nanoparticles

Conjugation of thiolated DNA onto 10-nm gold nanoparticles was achieved following previously reported protocol.⁴⁷ In a typical experiment, 20 μL 2.5 μM phosphine-coated 10-nm gold nanoparticle was mixed with 0.5 μL 2 M NaNO_3 and 0.65 μL 100 μM thiolated DNA in 0.25 \times TBE buffer. The reaction solution was incubated at room temperature for 36 hours in the dark. After that, the reaction solution was loaded into 1% agarose gel containing 0.5 \times TBE buffer. The electrophoresis was running at 95 V for 1 hour in a gel box on an ice-water bath. The purple band was recovered by pestle crushing, followed by centrifugation for 3 min at 10,000 rpm at room temperature using “Freeze 'N Squeeze” DNA Gel Extraction spin columns (Bio-Rad). Recovered DNA-modified gold nanoparticles were stored at 4°C in the dark for further use. The sequence for the thiolated DNA was: 5'-AAAAAAAAAA-/3ThioMC3-D/.

Gold nanoparticle decoration of DNA crystals

To 15 μL 400 mM NaCl solution, 0.8 μL (ZX-4H×6H×96B-channel crystal) or 0.6 μL (XY4H×4H×64B-cuboid crystal) DNA samples were added. Then 0.2 μL 95 nM 10 nm gold nanoparticles were introduced. After pipetting for 50 times, the reaction mixture was left at room temperature for three hours in the dark.

TEM imaging

For imaging, 2.5 μL of annealed sample were adsorbed for 2 minutes onto glow-discharged, carbon-coated TEM grids. The grids were then stained for 10 seconds using a 2% aqueous uranyl formate solution containing 25 mM NaOH. Imaging was performed using a JEOL JEM-1400 TEM operated at 80 kV.

Cryo-EM imaging

A 5 μL droplet of the crystal sample was added onto the positively-charged carbon grids, blotted 5.5 s, and plunge-frozen in liquid ethane using a Vitrobot (FEI). Grids were then loaded into a Titan Krios transmission electron microscope (FEI), which was operated at 300 kV with nominal $-3\text{-}\mu\text{m}$ defocus using a dose of 1.5 $\text{e}^\circ/\text{A}^2$. The specimen was tilted $\sim 2^\circ\text{C}$ increments over a total angular range of $\pm 60^\circ\text{C}$. All of the images were kept by using automated data acquisition software. 3D reconstructions from the above tilt series were performed with the weighted back-projection method and further analysis of tomograms was done using IMOD software.⁴⁸

Annealing and melting curves

500 nM of strands were folded following the annealing ramp protocol listed above in the presence of 0.3 \times SYBR Green I using the Eppendorf Master cycler realplex 4 PCR. Samples were read once during the first cooling step and four times during the second cooling step.

Following annealing, the same samples were melted at the rate of 0.2°C/min. 9 μ L samples were removed during the annealing or melting for TEM and gel electrophoresis analysis.

Agarose gel electrophoresis

Annealed samples were subjected to 1.5 or 2 percent native agarose gel electrophoresis for 2 hours (gel prepared in 0.5 \times TBE buffer supplemented with 10 mM MgCl₂ and 0.005% (v/v) EtBr) or 1 \times SYBR Safe in an ice water bath.

AFM imaging

AFM images were obtained using a MultiMode VIII SPM (Bruker) equipped with a liquid cell. 2 μ L of sample was applied onto the surface of highly ordered pyrolytic graphite (HOPG), which was pretreated with ultraviolet-ozone for 30 seconds. The 50 μ L of TAE buffer (12.5 mM Mg²⁺, 5 mM Ni⁺) was added into the liquid cell. And images were collected using silicon nitride cantilevers (Olympus).

Supplementary Material

Refer to Web version on PubMed Central for supplementary material.

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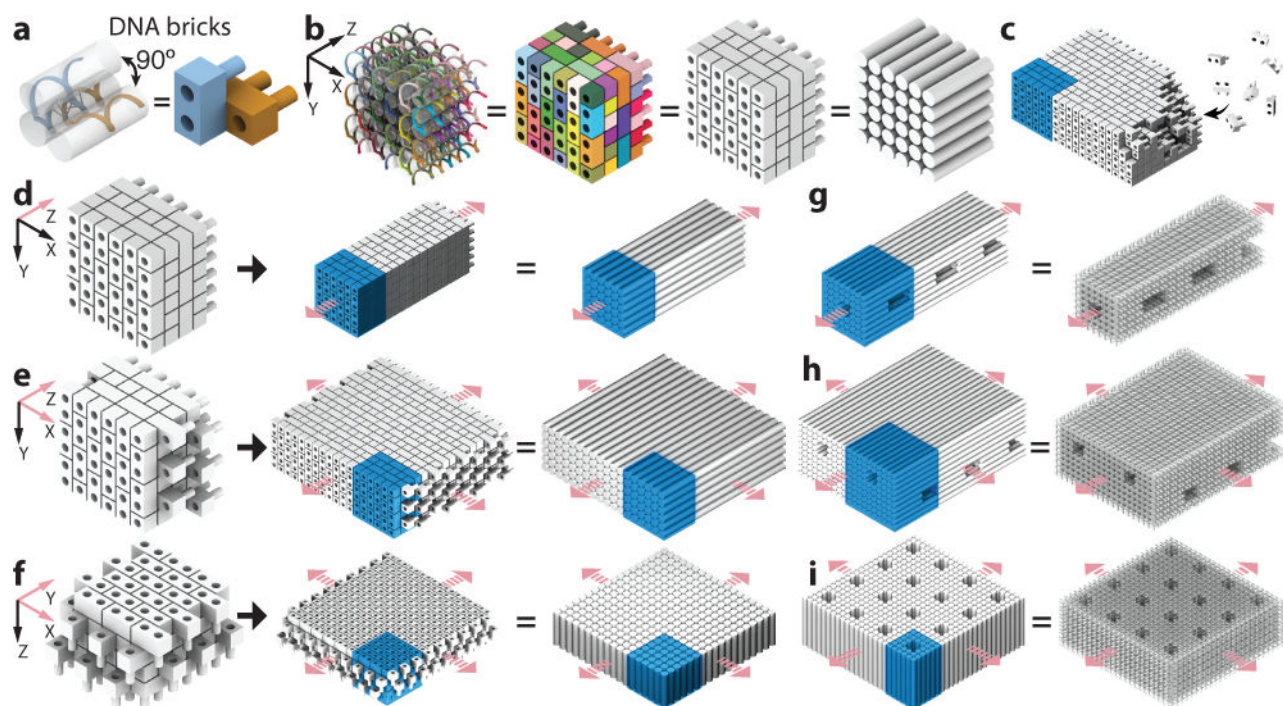


Fig. 1. Design of DNA-brick crystals

a, A strand (left) and a brick (right) model showing two 32-nt DNA bricks that form a 90° angle. **b**, Models of a $6H$ (helix) \times $6H$ (helix) \times $24B$ (basepair) cuboid with increasing levels of abstraction: a strand (leftmost) and a brick model where colors distinguish brick species, a brick model with all bricks colored grey, and a model where cylinders representing DNA double-helices. **c**, Individual DNA strands, rather than pre-assembled multi-brick blocks, are directly incorporated into the growing crystal. **d** to **f**, Brick and cylinder models of a 1D Z-crystal (**d**), a 2D ZX-crystal (**e**), and a 2D XY-crystal (**f**) designed from the $6H \times 6H \times 24B$ cuboid. **g** to **i**, Cylinder and DNA-helix models of crystals with pores and tunnels. **g**, A Z-crystal with a tunnel and periodic pores. **h**, A ZX-crystal with two groups of parallel tunnels. **i**, An XY-crystal with periodic pores. Repeating units of the crystals are denoted using blue-colored boxes. Pink arrows indicate the directions of crystal growth.

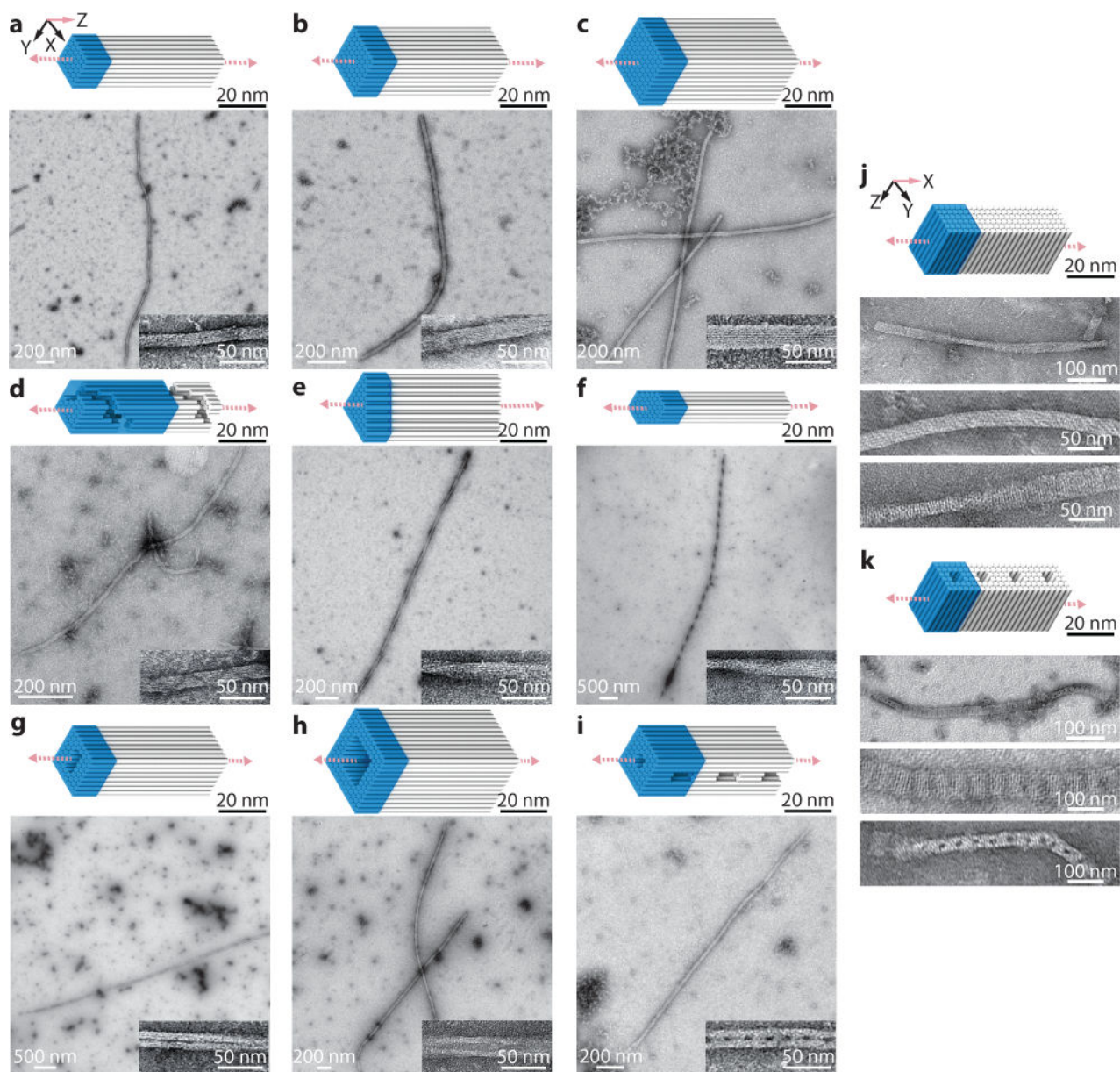


Fig. 2. One-dimensional DNA crystals

a to i, Z-crystals with cylinder models and TEM images. **a to c**, Z-crystals with solid cross-sections: $6H \times 6H$ (**a**), $8H \times 8H$ (**b**), and $10H \times 10H$ (**c**) Z-crystals. **d to f**, Z-crystals with different cross-sectional shapes: an $8H \times 8H$ Z-crystal with right-handed spiral channel (**d**), a $43H$ Z-crystal with triangle-shaped cross-section (**e**), and a $44H$ Z-crystal with hexagon-shaped cross-section (**f**). **g to i**, Z-crystals with porous cross-sections: an $8H \times 8H$ Z-crystal with a $2H \times 4H$ tunnel (**g**), a $12H \times 12H$ Z-crystal with a $6H \times 6H$ tunnel (**h**), and an $8H \times 8H$ Z-crystal with a $2H \times 2H$ tunnel and perpendicular $8H \times 2H \times 24B$ pores (**i**). **j** and **k**, cylinder models (top) and TEM images (bottom) of X-crystals. **j**, an X- $6H \times 6H \times 64B$ -cuboid crystal. **k**, a $6H \times 6H$ X-crystal with $2H \times 2H$ pores. Unit cells of crystals are denoted using blue-colored boxes. See Supplementary Figs. S3 to S12 for more TEM images.

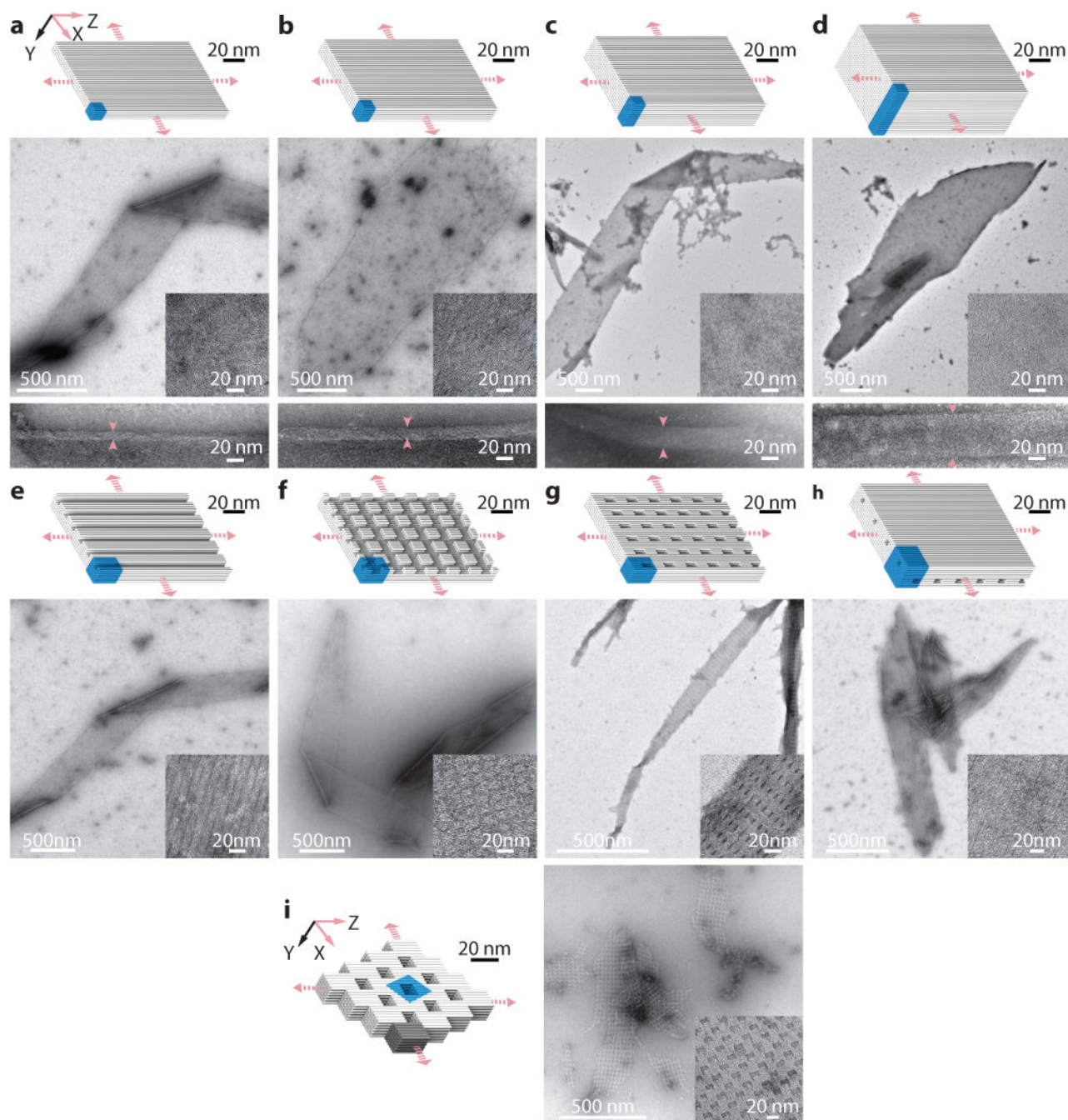


Fig. 3. Two-dimensional multilayer ZX-crystals

Cylinder models (top) and TEM images (bottom) are shown for each crystal. **a to d**, Solid ZX-crystals: 4-layer (**a**), 6-layer (**b**), 10-layer (**c**), and 20-layer (**d**), solid ZX-crystals.

Arrows indicate the positions for the thickness measurement of the crystals. **e to h**, ZX-crystals with channels, pores, and tunnels: 6-layer crystal with $2H \times 2H$ parallel channels (**e**), 6-layer crystal with two groups of crossing channels – $2H \times 2H$ channels that run parallel to the DNA helical axis and $2H \times 32B$ channels that run perpendicular to the DNA helical axis (**f**), 6 layer crystal with $2H \times 6H \times 32B$ pores (**g**), and 10 layer crystal with two groups of non-

contacting tunnels – $2H \times 2H$ tunnels parallel to the DNA helical axis and $2H \times 24B$ tunnels perpendicular to DNA helical axis (**h**). In **h**, the two groups of tunnels are separated by two layer of DNA helices. **i**, An offset-ZX- $6H \times 6H \times 64B$ -cuboid ZX-crystal. The dark grey part represents a $6H \times 6H \times 64B$ -cuboid repeating unit. Unit cells of crystals are denoted using blue-colored boxes. See Supplementary Figs. S13 to S21 for more TEM images.

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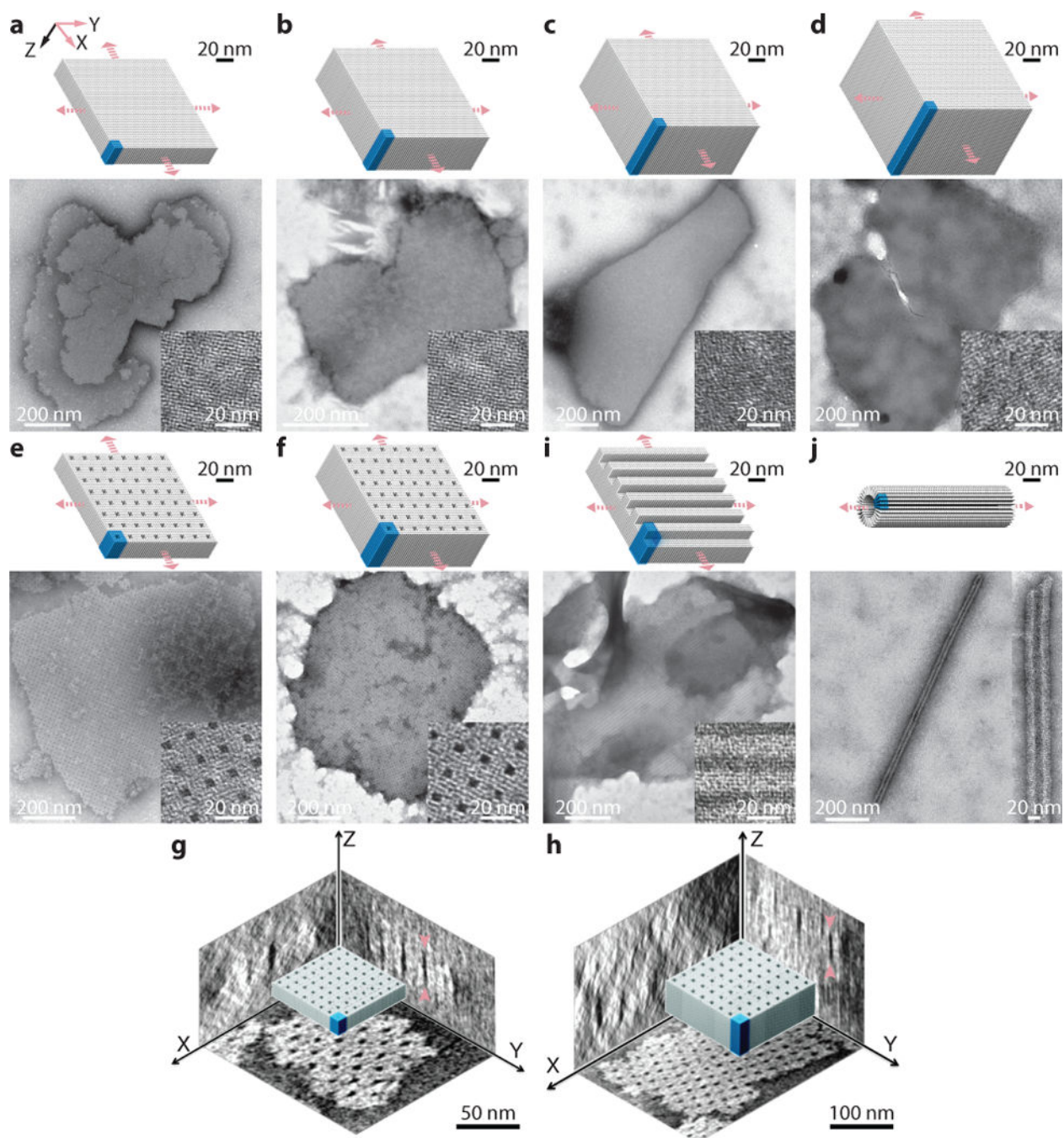


Fig. 4. Two-dimensional DNA-forest XY-crystals

Cylinder models (top) and TEM images (bottom) are shown for each crystal. **a to d**, Solid XY-crystals: 64B (**a**), 128B (**b**), 192B (**c**), and 256B (**d**) solid XY-crystals designed from a $4H \times 4H$ cuboid. **e** and **f**, XY-crystals with pores: a $32 \times 64B$ -pore XY-crystal with $2H \times 2H \times 64B$ parallel pores (**e**) and a $32H \times 128B$ -pore XY-crystal with $2H \times 2H \times 128B$ parallel pores (**f**). **g** and **h**, Cryo-EM 3D reconstruction images showing the three projections of the XY- $32H \times 64B$ -pore crystal (**g**) and the XY- $32H \times 128B$ -pore crystal (**h**). Arrows indicate the positions for thickness measurements. **i**, A 96B XY-crystal with $4H \times 32B$

parallel channels. **j**, A tube crystal formed by 32B helices with helical axis perpendicular to the tube axis. Unit cells of crystals are denoted using blue-colored boxes. See Supplementary Figs. S25 to S34 for more TEM images.

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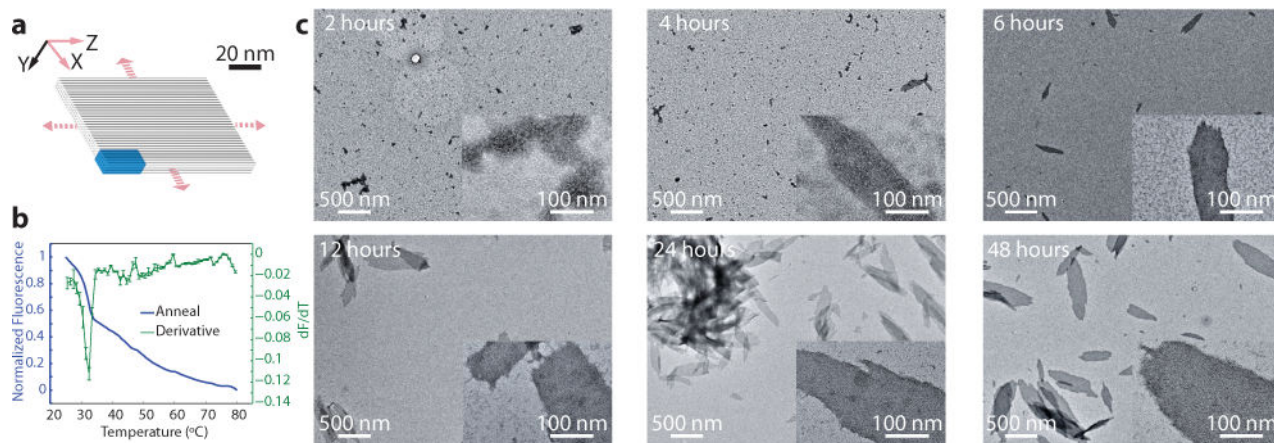


Fig. 5. Isothermal assembly of brick crystals

a, A cylinder model of a ZX-6H×4H×96B crystal. **b**, Annealing curve (blue) with the derivative of the fluorescence with respect to temperature (green) of the ZX-6H×4H×96B crystal. Annealing curves were obtained in the presence of 0.3× SYBR Green I at 500 nM of each strand. Error bars represent standard error from the mean acquired from 3 measurements. **c**, TEM images of the ZX-6H×4H×96B crystal incubated at 33°C for various durations. Samples were diluted four times before deposition on the TEM grid.

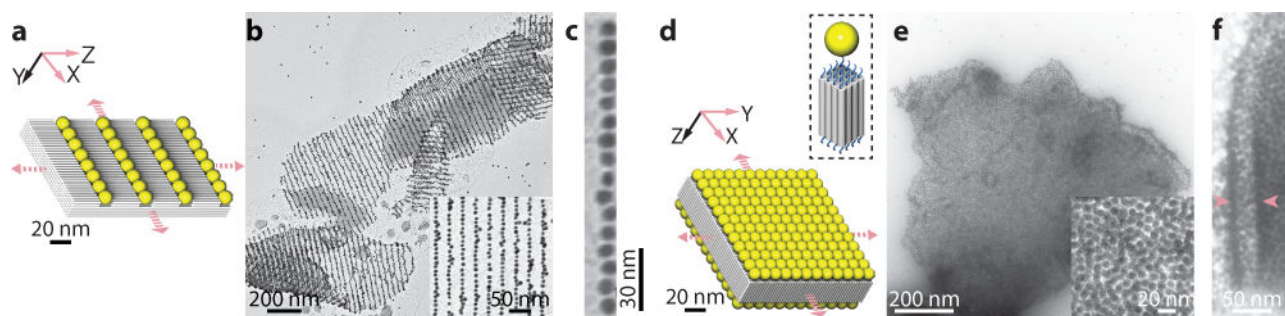


Fig. 6. Gold nanoparticles patterned using DNA-brick crystals

a to c, Model (**a**) and TEM images (**b**, **c**) of parallel lines of 10-nanometer gold nanoparticles closely packed on a ZX-4H×6H×96-channel crystal. (**c**), Zoomed-in TEM image of a single chain of gold nanoparticles. **d to e**, Model (**d**) and TEM images (**e**, **f**) of close-packed gold nanoparticle monolayers formed on the top and bottom surfaces of an XY-4H×4H×64B cuboid crystal. Inset shows the single-stranded poly-T extensions on each end of the helix and a 10 nm gold nanoparticle occupying a 4H×4H surface. **f**, A crystal displays curvatures on the edge. Pink arrows indicate the curved positions where the two gold nanoparticle monolayers can be seen.