# **NANO EXPRESS** Open Access and the set of the

# Conical islands of  $TiO<sub>2</sub>$  nanotube arrays in the photoelectrode of dye-sensitized solar cells

Woong-Rae Kim $^{1+}$ , Hun Park $^{2+}$  and Won-Youl Choi $^{1,3^*}$ 

# Abstract

Ti conical island structures were fabricated using photolithography and the reactive ion etching method. The resulting conical island structures were anodized in ethylene glycol solution containing 0.25 wt% NH4F and 2 vol% H<sub>2</sub>O, and conical islands composed of TiO<sub>2</sub> nanotubes were successfully formed on the Ti foils. The conical islands composed of TiO<sub>2</sub> nanotubes were employed in photoelectrodes for dye-sensitized solar cells (DSCs). DSC photoelectrodes based on planar Ti structures covered with  $TiO<sub>2</sub>$  nanotubes were also fabricated as a reference. The short-circuit current  $(J_{\rm sc})$  and efficiency of DSCs based on the conical island structures were higher than those of the reference samples. The efficiency of DSCs based on the conical island structures reached up to 1.866%. From electrochemical impedance spectroscopy and open-circuit voltage  $(V_{oc})$  decay measurements, DSCs based on the conical island structures exhibited a lower charge transfer resistance at the counter cathode and a longer electron lifetime at the interface of the photoelectrode and electrolyte compared to the reference samples. The conical island structure was very effective at improving performances of DSCs based on  $TiO<sub>2</sub>$  nanotubes.

Keywords: Dye-sensitized solar cells; TiO<sub>2</sub> nanotube; Conical islands; Anodic oxidation

# Background

Dye-sensitized solar cells (DSCs) have received great interest as promising alternatives to conventional silicon solar cells [\[1](#page-8-0)]. Their low cost and easy fabrication processes are powerful advantages. For example, DSCs can be fabricated without the high-cost vacuum equipment required for Si cells, such as plasma-enhanced chemical vapor deposition (PECVD) and atomic layer deposition (ALD). Wet etching processes, such as saw damage etching and texturing, which are widely used in the fabrication of Si solar cells, are not required in DSCs, which simplifies their fabrication.

Gratzel and O'Regan achieved remarkably high-efficiency DSCs by using mesoporous  $TiO<sub>2</sub>$  nanoparticle-based films [[1](#page-8-0)]. Recently, Burschka et al. reported high efficiency, up to 15%, by fabricating inorganic-organic hybrid solar cells containing perovskite compounds [\[2\]](#page-8-0). However, the efficiencies

Full list of author information is available at the end of the article



of DSCs are still too low for commercialization compared with conventional Si solar cells.

The main components which affect the performance of DSCs are the dyes [\[3-13](#page-8-0)], photoelectrodes [[14](#page-8-0)-[38](#page-8-0)], counter cathodes [\[39](#page-8-0)[-46\]](#page-9-0), and electrolytes [\[47-51\]](#page-9-0). Among these components, increasing the surface area of the photoelectrodes and reducing charge recombination between the photoelectrodes and electrolytes are critical factors in improving the efficiencies of DSCs.

TiO2 nanoparticle structures have been widely used for the photoelectrodes of DSCs. However, structural defects which exist at the contacts between the  $TiO<sub>2</sub>$ nanoparticles not only hinder electronic diffusion across  $TiO<sub>2</sub>$  nanocrystalline films but also act as charge recom-bination sites [\[20](#page-8-0)-[22,24](#page-8-0),[26](#page-8-0)]. To address this issue,  $TiO<sub>2</sub>$ nanostructures such as nanowires [\[31,33,34](#page-8-0)], nanotubes [[20-30,37,38](#page-8-0)[,52,53\]](#page-9-0), nanohemispheres [[35](#page-8-0),[36](#page-8-0)], and nanoforests [\[32](#page-8-0)] have been applied to the photoelectrodes of DSCs as alternatives to the conventional  $TiO<sub>2</sub>$  nanoparticle structure.

Of these nanostructures, the  $TiO<sub>2</sub>$  nanotube has received a great deal of attention due to its one-dimensional charge transport. Charge percolation in  $TiO<sub>2</sub>$  nanotube-

© 2015 Kim et al.; licensee Springer. This is an Open Access article distributed under the terms of the Creative Commons Attribution License [\(http://creativecommons.org/licenses/by/4.0\)](http://creativecommons.org/licenses/by/4.0), which permits unrestricted use, distribution, and reproduction in any medium, provided the original work is properly credited.

<sup>\*</sup> Correspondence: [cwy@gwnu.ac.kr](mailto:cwy@gwnu.ac.kr) †

Equal contributors

<sup>&</sup>lt;sup>1</sup>Department of Metal and Materials Engineering, Gangneung-wonju National University, Gangneung 210-720, South Korea

<sup>3</sup> Research Institute for Dental Engineering, Gangneung-Wonju National University, Gangneung 210-720, South Korea

based films is easier than that in  $TiO<sub>2</sub>$  nanoparticle-based films [[20-22\]](#page-8-0). It has been reported that electron lifetimes in  $TiO<sub>2</sub>$  nanotube-based films are longer than those in  $TiO<sub>2</sub>$  nanoparticle-based films [\[26\]](#page-8-0).

In this study, conical island shapes were patterned on a Ti substrate to increase the specific surface area for dye adsorption. Ti surfaces with these conical islands were anodized, and  $TiO<sub>2</sub>$  nanotube arrays were formed on the entire Ti surface. Conical islands composed of TiO2 nanotubes were successfully fabricated and applied as the photoelectrodes of DSCs. Planar Ti photoelectrodes were also prepared with  $TiO<sub>2</sub>$  nanotube arrays, and then the characteristics and performance of the conical island photoelectrode structure were compared with those of the planar photoelectrode structure. To our knowledge, this is the first trial to report the fabrication of conical islands composed of  $TiO<sub>2</sub>$  nanotubes and their application to DSCs.

# Methods

The experimental procedures for making conical islands composed of  $TiO<sub>2</sub>$  nanotubes are very similar to those found in our previous report [[30](#page-8-0)]. To make conical islands on the surface of 0.5-mm-thick Ti foils (99%, Alfa Aesar Co., Ward Hill, MA, USA), 5-μm-thick photoresists (PR; L-300, Dongjin Co., Hwaseong-si, South Korea) were coated on the surface of Ti foils using a spin-coater (Mark-8 Track, TEL Co., Tokyo, Japan). The spin-coated photoresists were softly baked at 120°C for 120 s and hardly baked at 110°C for 5 min. A dot-patterned photomask was used to make PR patterning by UV light exposure. The UV light, which had an energy of 14.5 mJ/s, was illuminated for 5 s, and the PR was developed. The PR in the UV-unexposed area was removed.

The PR-patterned Ti foil was dry-etched at 25°C for 40 min by using reactive ion etching (RIE) equipment (ICP380, Oxford Co., Abingdon, Oxfordshire, UK). BCl3 and  $Cl<sub>2</sub>$  gases were used in the RIE process with a top power of 800 W and a bottom power of 150 W. Photoresists on the UV-exposed area played a role in protecting the flat Ti surface during the RIE process. Only the Ti surface in the UV-unexposed area was etched out. The remaining photoresists after the RIE process were stripped at 250°C for 20 min using a photoresist stripper (TS-200, PSK Co., Hwaseong-si, South Korea).  $O_2$  and  $N_2$ gases were used for photoresist stripping with a power of 2,500 W.

Before the anodizing process, the Ti foils patterned with conical islands were successively ultrasonicated with acetone, ethanol, and deionized (DI) water to remove any residues on the surface of the Ti foils. Conical islands composed of  $TiO<sub>2</sub>$  nanotubes were fabricated by anodic oxidation of the patterned Ti foils in ethylene glycol solution. The schematic diagram of the anodic oxidation system is shown in Figure 1.

To obtain a proper microstructure of  $TiO<sub>2</sub>$  nanotube arrays, anodic oxidation voltages and times applied by a DC power supply were computer-controlled by a LabVIEW program. The ethylene glycol solution included 0.5 wt%  $NH_4F$  and 0.2 vol%  $H_2O$ . A constant potential of 60 V with a ramping speed of 1 V/s was applied between the anode and the cathode. Pt metal was used as a counter cathode. Anodizing time was varied to control the optimum thickness of  $TiO<sub>2</sub>$  nanotubes in the conical islands. As-anodized  $TiO<sub>2</sub>$  nanotubes were rinsed with DI water and annealed at 500°C for 1 h. The morphologies of the conical islands composed of  $TiO<sub>2</sub>$  nanotubes were studied by field emission scanning electron microscopy (FESEM;



Hitachi SU-70, Hitachi, Ltd., Chiyoda-ku, Japan) at Korea Basic Science Institute. As-anodized and annealed TiO<sub>2</sub> nanotubes were analyzed by X-ray diffraction (XRD; Rigaku D/MAX-RC, Shibuya-ku, Japan, Cu Kα radiation) to confirm their crystallization.

The Ti substrates with conical islands composed of  $TiO<sub>2</sub>$  nanotubes were then prepared for use as the photoelectrodes of DSCs. The  $TiO<sub>2</sub>$  photoelectrodes were immersed at room temperature for approximately 1 day in an ethanol solution containing  $3 \times 10^{-4}$  M cis-bis(isothiocyanato)bis(2,2′bipyridyl-4,4′dicarboxylato) ruthenium(II) bis-tetrabutylammonium (N719) dye. The dye-adsorbed photoelectrodes were rinsed with ethanol solution and were dried at room temperature. Pt-coated fluorine-doped tin oxide (FTO) glasses were prepared as counter electrodes by spin coating 0.7 mM  $H_2PtCl_6$  solution in 2-propanol at 500 rpm for 10 s and subsequently annealing at 380°C for 30 min. The dye-adsorbed photoelectrodes and Pt-coated FTO glasses were spaced by using 60-μm Surlyn film purchased from DuPont Co., Ltd. (Wilmington, DE, USA). The liquid electrolyte was prepared by dissolving 0.6 M 1 hexyl-2,3-dimethylimidazolium iodide (C6DMIm), 0.05 M iodine, 0.1 M lithium iodide, and 0.5 M 4-tert-butylpyridine in 3-methoxyacetonitrile. Current-voltage (J-V) characteristics were measured under AM 1.5G condition (Keithley Model 2400 source measure unit, Keithley Instruments, Inc., Cleveland, OH, USA). Electrochemical impedance spectroscopy (EIS) behaviors were analyzed at open-circuit condition and in the frequency range of 100 kHz to 100 mHz. A 1,000-W xenon lamp (Oriel, 91193, Newport Co., Stratford, CT, USA) was used as a light source for one sun condition.

## Results and discussion

Figure 2 shows the FESEM images of the conical islands formed on the Ti surface by the RIE method. In our previous report, we fabricated arrays of protruding Ti cylinders [[30](#page-8-0)]. By increasing the etching time, the morphology of the Ti surface can be converted from protruding cylinders

 $\left( \mathbf{a}\right)$ 

to conical islands, as shown in Figure 2. In this process, negative photoresists were first spin-coated on the Ti metal surface. Usually, when photoresists are exposed to UV light, they are cross-linked and so block reactive etching ions during the RIE process. The cross-linked photoresist layer protects the Ti surface from physical and chemical etching.

However, photoresists have weaker cross-linking at an edge than at the center. At edges, cross-linked photoresists can be easily damaged during the RIE process if the etching time is long enough. In that case, the crosslinked photoresists at the edge areas can be crushed by the reactive ions after the formation of the Ti protruded cylinders. This allows the protruded cylinders to be converted into conical islands by increasing the dry etching time, as shown in Figure 2a.

The top and the bottom of the produced conical islands have diameters of approximately 1 μm and approximately 4 μm, respectively. The height of the conical islands is approximately 5 μm. Ti conical island arrays can be seen in Figure 2b. The distance between adjacent conical islands is approximately 8 μm and coincides with the photomask patterns.

Figure [3](#page-3-0) shows FESEM images of the conical islands composed of  $TiO<sub>2</sub>$  nanotubes. The  $TiO<sub>2</sub>$  nanotubes were grown by anodic oxidation of Ti metal in 0.5 wt%  $NH_4F$ - and 0.2 vol%  $H_2O$ -containing ethylene glycol solution. A constant voltage of 60 V with a ramping speed of 1 V/s was applied between the cathode and the anode during anodic oxidation. Fluoride ions in the ethylene glycol solution can anisotropically etch Ti and Ti oxide due to the voltage-biased direction. As a result, highly ordered  $TiO<sub>2</sub>$  nanotubes are vertically grown on the whole Ti surface, including the conical islands. The inclined planes of the conical islands also have vertically grown  $TiO<sub>2</sub>$  nanotubes on their surfaces. The underlying mechanism is discussed in a previous report [\[30](#page-8-0)]. The diameter and wall thickness of the fabricated  $TiO<sub>2</sub>$  nanotubes are approximately 80 nm and approximately 20 nm, respectively.



<span id="page-3-0"></span>

The length of the  $TiO<sub>2</sub>$  nanotube arrays was controlled by the anodizing time and was in the range of 1 to 2 μm.

Figure 4 shows the X-ray diffraction patterns of the as-anodized and annealed conical island structures composed of  $TiO<sub>2</sub>$  nanotube arrays on Ti substrates. The as-anodized conical island structure composed of

 $TiO<sub>2</sub>$  nanotube arrays, as shown in Figure 4a, has only Ti metal peaks, and there are no crystalline peaks of Ti oxides. The as-anodized conical island structure can be confirmed as amorphous phase. After annealing at 500°C for 1 h, the anatase peaks of  $TiO<sub>2</sub>$  can be found, as seen in Figure 4b. This reveals that the nanotubes on the conical



islands were converted to the anatase phase by the annealing process. The anatase phase has previously shown the best performance when applied in DSCs because the anatase (101) plane provides preferential sites for dye adsorption [\[54,55\]](#page-9-0).

Figure 5 shows the schematic diagrams and FESEM images of the DSC photoelectrodes prepared with a planar structure and with a conical island structure [\[30](#page-8-0)]. The photoelectrode with the planar  $TiO<sub>2</sub>$  nanotube array was prepared by the anodic oxidation of flat Ti foils. On the other hand, the photoelectrode with the conical island structure of  $TiO<sub>2</sub>$  nanotube arrays was fabricated by anodic oxidation of Ti foils which were patterned, etched, and finally shaped like conical islands.

For experimental purposes, both types of  $TiO<sub>2</sub>$  arrays were prepared with nanotube lengths of 1.0, 1.5, and 2.0 μm. Both types of photoelectrodes were fabricated with the identical active cell size of 5 mm  $\times$  5 mm. The photoelectrode with the planar  $TiO<sub>2</sub>$  nanotube arrays was later used as a reference sample to compare J-V characteristics and EIS behaviors with the conical island structure.

Figure [6](#page-5-0) and Table [1](#page-5-0) show the J-V characteristic performances of DSCs based on planar and conical island structures of  $TiO<sub>2</sub>$  nanotube arrays, with different lengths of  $TiO<sub>2</sub>$  nanotubes. The short-circuit current  $(J_{\rm sc})$  in the planar structure increased with the length of the  $TiO<sub>2</sub>$  nanotube arrays, and such data has been previ-ously reported [\[30](#page-8-0)]. Longer  $TiO<sub>2</sub>$  nanotubes provide a

For the same length of  $TiO<sub>2</sub>$  nanotubes, DSCs based on the conical island structure show higher  $J_{\rm sc}$  than DSCs based on the planar structure. The added  $TiO<sub>2</sub>$ nanotube arrays on the inclined surfaces of the conical island further increase the surface area available to adsorb the dyes and produce photoelectrons.

DSCs having  $2.0$ - $\mu$ m-long TiO<sub>2</sub> nanotube arrays showed the best performances for both planar and conical island structures. The short-circuit current and efficiency for the conical island structures and  $2.0$ - $\mu$ m-long TiO<sub>2</sub> nanotubes were  $5.877 \text{ mA/cm}^2$  and  $1.866\%$ , respectively. In all cases, the efficiencies of the conical island structure DSCs were higher than those based on the planar structure. The conical island structure was very effective at improving DSC power conversion efficiency. It is also anticipated that conical island structures with  $TiO<sub>2</sub>$  nanotube arrays longer than 2.0 μm will have still higher efficiency.

Figure [7](#page-6-0) shows EIS results of DSCs with planar and conical island structures. The Nyquist plots in Figure [7](#page-6-0)a show that DSCs based on conical island structures have a lower chemical resistance than DSCs based on planar structures. Lower chemical resistance is closely related to better efficiency in DSCs. These results exactly coincide with the J-V characteristics in Figure [6](#page-5-0) and Table [1](#page-5-0).

Chemical resistances in Figure [7](#page-6-0)a include the charge transfer resistance  $(R_{Pt/electrolyte})$  at the interface between the Pt counter electrode and electrolyte, charge transfer



<span id="page-5-0"></span>

resistance  $(R_{TiO2/electrolyte})$  at the interface between the TiO2 photoelectrode and electrolyte, and charge transport resistance  $(R_{\text{electrolyte}})$  in the electrolyte. DSCs based on the conical island structure have lower chemical resistance due to remarkably decreased  $R_{Pt/electrolyte}$ . The reason for this decrease in  $R_{Pt/electrolyte}$  is not clear. It seems that the conical island may be a better structure for transferring electrons to the counter electrode than the planar structure, since the  $J_{\rm sc}$  of the conical island structure is higher than that of the planar structure.

Figure [7](#page-6-0)b shows Bode plots of DSCs based on planar and conical island structures. Usually, Bode plots of DSCs have three main phase  $(\theta)$  peaks, according to the frequency range. The high-frequency (1 kHz to 100 kHz) region is attributed to charge transfer at the interface between the counter cathode and electrolyte. The mediumfrequency (1 Hz to 1 kHz) region is the response of charge transfer at the interface between the  $TiO<sub>2</sub>$  photoelectrode and electrolyte. The low-frequency (1 mHz to 1 Hz) region reflects charge transport associated with the Nernstian diffusion process in the electrolyte. If the frequency at the phase peaks can be determined, the electronic lifetimes at each interface and in the electrolyte can be calculated by just inversing the peak frequencies. Only two phase peaks, in the high- and medium-frequency regions, could be found in Figure [7](#page-6-0)b. The phase peak in the low-frequency region was not observed because the acquisition time was too short to obtain the diffusion peak.

In the high-frequency region, frequencies at the phase peaks are approximately 2.1 kHz and approximately 2.4 kHz for DSCs based on planar and conical island structures, respectively; therefore, the electronic lifetimes are 0.476 and 0.417 ms, respectively. On the other hand, in the medium-frequency region, the frequencies at phase peaks are 10 and 5.5 Hz for DSCs with planar and conical island structures; thus, the electronic lifetimes are 0.1 and 0.18 s, respectively. The difference of electronic lifetimes





<span id="page-6-0"></span>

in the high-frequency region is negligible; however, in the medium-frequency region, the electronic lifetime of DSCs based on the conical island structure is obviously longer than that of DSCs based on the planar structure. These results indicate that conical islands composed of  $TiO<sub>2</sub>$ nanotubes are more effective for charge transfer at the interface between the  $TiO<sub>2</sub>$  photoelectrode and electrolyte.

Figure [8](#page-7-0) shows the raw data and normalized values of open-circuit voltage  $(V_{oc})$  decay measurements for DSCs based on planar and conical island structures. To analyze DSC  $V_{\text{oc}}$  decay behaviors, light at AM 1.5G was supplied for approximately 20 s, and  $V_{\text{oc}}$  decays were measured by abruptly stopping the supply of illumination. When the illumination on DSCs is stopped, charge recombination at the interface of the  $TiO<sub>2</sub>$  photoelectrode or counter electrode gives rise to the decrease of  $V_{\text{oc}}$ . The slower the  $V_{\text{oc}}$ decay is, the higher the electronic lifetime is. This test revealed that the  $V_{oc}$  decay of DSCs based on the conical island structure was slower than that of DSCs based on the planar structure; therefore, the electronic lifetime of DSCs based on the conical island structure is higher

than that of DSCs based on the planar structure. This result is closely correlated to the result of Bode plots in Figure 7b.

## Conclusions

Ti conical islands were fabricated by coating photoresists on Ti foils, stacking dot-patterned photomasks on the photoresists, illuminating with UV light, etching the surface with the RIE method, and stripping the photoresist. When the produced Ti conical islands were anodized, conical islands composed of  $TiO<sub>2</sub>$  nanotubes were successfully fabricated. The conical islands composed of  $TiO<sub>2</sub>$ nanotubes were then employed as photoelectrodes of DSCs. The J-V characteristics of DSCs based on the conical island structures were compared with those of DSCs based on the planar structures covered with  $TiO<sub>2</sub>$  nanotubes. The  $J_{\rm sc}$  and the efficiency of DSCs based on the conical island structures were higher than those of DSCs based on the planar structures. The efficiency of the DSCs using conical island structures reached 1.87%. The charge transfer resistance at the counter cathode of DSCs based on the conical island structures was remarkably reduced

<span id="page-7-0"></span>

compared to that of DSCs based on the planar structures. The electron lifetime at the interface between the photoelectrode and electrolyte was longer for DSCs based on the conical island structure than that for DSCs based on the planar structure. The conical island structure effectively enhanced the performance of DSCs based on  $TiO<sub>2</sub>$ nanotube arrays. The relatively low efficiency of DSCs based on the conical island structure compared with that of typical DSCs can be overcome by controlling the microstructure of the conical island, dye, electrolyte, surface treatment of  $TiCl<sub>4</sub>$ , and so on.

#### Competing interests

The authors declare that they have no competing interests.

### Authors' contributions

WK and HP conceived of the study and drafted the manuscript. WK helped with the anodization of the Ti surface. HP helped with the J-V characterization and EIS analysis of the DSCs. WC supervised the whole work and revised the manuscript. All authors read and approved the final manuscript.

#### Acknowledgements

This research was financially supported by the Ministry of Education, Science and Technology (MEST) and by the National Research Foundation of Korea (NRF) through the Human Resource Training Project for Regional Innovation (Grant No. 2012H1B8A2026009).

#### <span id="page-8-0"></span>Author details

<sup>1</sup>Department of Metal and Materials Engineering, Gangneung-wonju National University, Gangneung 210-720, South Korea. <sup>2</sup>Korea Institute of Science and Technology Information, Seoul 130-741, South Korea. <sup>3</sup>Research Institute for Dental Engineering, Gangneung-Wonju National University, Gangneung 210-720, South Korea.

## Received: 25 November 2014 Accepted: 6 January 2015 Published online: 11 February 2015

## References

- 1. O'Regan B, Gratzel M. A low-cost, high-efficiency solar-cell based on dye-sensitized colloidal TiO<sub>2</sub> films. Nature. 1991;353(6346):737-40.
- 2. Burschka J, Pellet N, Moon S-J, Humphry-Baker R, Gao P, Nazeeruddin MK, et al. Sequential deposition as a route to high-performance perovskite-sensitized solar cells. Nature. 2013;499(7458):316–9.
- 3. Yu X, Ci Z, Liu T, Feng X, Wang C, Ma T, et al. Influence of different electron acceptors in organic sensitizers on the performance of dye-sensitized solar cells. Dyes Pigments. 2014;102:126–32.
- 4. Wang S, Wang H, Guo J, Tang H, Zhao J. Influence of the terminal electron donor in D–D–π–A phenothiazine dyes for dye-sensitized solar cells. Dyes Pigments. 2014;109:96–104.
- 5. Sharma GD, Patel KR, Roy MS, Misra R. Characterization of two new (A–π)2– D–A type dyes with different central D unit and their application for dye sensitized solar cells. Org Electron. 2014;15(8):1780–90.
- 6. Neuthe K, Bittner F, Stiemke F, Ziem B, Du J, Zellner M, et al. Phosphonic acid anchored ruthenium complexes for ZnO-based dye-sensitized solar cells. Dyes Pigments. 2014;104:24–33.
- 7. Liu L, Chen J, Ku Z, Li X, Han H. Unsymmetrical squaraine sensitizers containing auxiliary arylamine donor for NIR-harvesting on dye-sensitized solar cell. Dyes Pigments. 2014;106:128–35.
- 8. Liu D-S, Ding W-L, Zhu K-L, Geng Z-Y, Wang D-M, Zhao X-L. The master factors influencing the efficiency of D–A–π–A configurated organic sensitizers in dye-sensitized solar cell via theoretically characterization: design and verification. Dyes Pigments. 2014;105:192–201.
- 9. Lee W, Yuk SB, Choi J, Kim HJ, Kim HW, Kim SH, et al. The effects of the number of anchoring groups and N-substitution on the performance of phenoxazine dyes in dye-sensitized solar cells. Dyes Pigments. 2014;102:13–21.
- 10. Lee CH, Yun HJ, Jung MR, Lee JG, Kim SH, Kim JH. Preparation and characterization of squaraine dyes containing mono- and bis-anchoring groups as the light absorber in dye sensitized solar cells. Electrochim Acta. 2014;138:148–54.
- 11. He J, Hua J, Hu G, Yin XJ, Gong H, Li C. Organic dyes incorporating a thiophene or furan moiety for efficient dye-sensitized solar cells. Dyes Pigments. 2014;104:75–82.
- 12. Chen X, Jia C, Wan Z, Yao X. Organic dyes with imidazole derivatives as auxiliary donors for dye-sensitized solar cells: experimental and theoretical investigation. Dyes Pigments. 2014;104:48–56.
- 13. Li L-L, Diau EW-G. Porphyrin-sensitized solar cells. Chem Soc Rev. 2013;42(1):291–304.
- 14. Peng X, Feng Y, Meng S, Zhang B. Preparation of hierarchical  $TiO<sub>2</sub>$  films with uniformly or gradually changed pore size for use as photoelectrodes in dye-sensitized solar cells. Electrochim Acta. 2014;115:255–62.
- 15. Zhang J, Shen H, Guo W, Wang S, Zhu C, Xue F, et al. An upconversion NaYF<sub>4</sub>:Yb<sup>3+</sup>, Er<sup>3+</sup>/TiO<sub>2</sub> core–shell nanoparticle photoelectrode for improved efficiencies of dye-sensitized solar cells. J Power Sources. 2013;226:47–53.
- 16. Yong S-M, Nikolay T, Ahn BT, Kim DK. One-dimensional WO<sub>3</sub> nanorods as photoelectrodes for dye-sensitized solar cells. J Alloys Compd. 2013;547:113–7.
- 17. Xie Y, Huang N, Liu Y, Sun W, Mehnane HF, You S, et al. Photoelectrodes modification by N doping for dye-sensitized solar cells. Electrochim Acta. 2013;93:202–6.
- 18. Maldonado-Valdivia AI, Galindo EG, Ariza MJ, García-Salinas MJ. Surfactant influence in the performance of titanium dioxide photoelectrodes for dye-sensitized solar cells. Sol Energy. 2013;91:263–72.
- 19. Maçaira J, Andrade L, Mendes A. Review on nanostructured photoelectrodes for next generation dye-sensitized solar cells. Renew Sustain Energy Rev. 2013;27:334–49.
- 20. Mor GK, Shankar K, Paulose M, Varghese OK, Grimes CA. Use of highly-ordered TiO<sub>2</sub> nanotube arrays in dye-sensitized solar cells. Nano Lett. 2006;6(2):215–8.
- 21. Zhu K, Neale N, Miedaner A, Frank A. Enhanced charge-collection efficiencies and light scattering in dye-sensitized solar cells using oriented  $TiO<sub>2</sub>$  nanotubes arrays. Nano Lett. 2007;7(1):69-74.
- 22. Zhu K, Vinzant T, Neale N, Frank A. Removing structural disorder from oriented TiO<sub>2</sub> nanotube arrays: reducing the dimensionality of transport and recombination in dye-sensitized solar cells. Nano Lett. 2007;7(12):3739–46.
- 23. Yang DJ, Park H, Cho SJ, Kim HG, Choi WY. TiO<sub>2</sub>-nanotube-based dyesensitized solar cells fabricated by an efficient anodic oxidation for high surface area. J Phys Chem Solid. 2008;69(5–6):1272–5.
- 24. Park H, Yang DJ, Kim HG, Cho SJ, Yang SC, Lee H, et al. Fabrication of MgO-coated TiO<sub>2</sub> nanotubes and application to dye-sensitized solar cells. J Electroceramics. 2009;23(2–4):146–9.
- 25. Park H, Yang DJ, Yoo JS, Mun KS, Kim WR, Kim HG, et al. Surface passivation of highly ordered TiO<sub>2</sub> nanotube arrays and application to dye-sensitized solar cells using the concept of isoelectric point. J Ceram Soc Jpn. 2009;117(1365):596–9.
- 26. Park H, Kim W-R, Jeong H-T, Lee J-J, Kim H-G, Choi W-Y. Fabrication of dye-sensitized solar cells by transplanting highly ordered  $TiO<sub>2</sub>$  nanotube arrays. Sol Energy Mater Sol Cells. 2011;95(1):184–9.
- 27. Park H, Kim W-R, Yang C, Kim H-G, Choi W-Y. Effect of a fullerene derivative on the performance of TiO<sub>2</sub>-nanotube-based dye-sensitized solar cells. J Nanosci Nanotechnol. 2012;12(2):1535–8.
- 28. Park H, Yang C, Choi W-Y. Organic and inorganic surface passivations of TiO<sub>2</sub> nanotube arrays for dye-sensitized photoelectrodes. J Power Sources. 2012;216:36–41.
- 29. Kim W-R, Lee Y-J, Park H, Lee J-J, Choi W-Y. TiO<sub>2</sub>-nanotube-based dyesensitized solar cells containing fluorescent material. J Nanosci Nanotechnol. 2013;13(5):3487–90.
- 30. Kim W-R, Park H, Choi W-Y. TiO<sub>2</sub> micro-flowers composed of nanotubes and their application to dye-sensitized solar cells. Nanoscale Res Lett. 2014;9(1):1–10.
- 31. Law M, Greene LE, Johnson JC, Saykally R, Yang P. Nanowire dye-sensitized solar cells. Nat Mater. 2005;4(6):455–9.
- 32. Ko SH, Lee D, Kang HW, Nam KH, Yeo JY, Hong SJ, et al. Nanoforest of hydrothermally grown hierarchical ZnO nanowires for a high efficiency dye-sensitized solar cell. Nano Lett. 2011;11(2):666–71.
- 33. Liao J-Y, Lei B-X, Chen H-Y, Kuang D-B, Su C-Y. Oriented hierarchical single crystalline anatase  $TiO<sub>2</sub>$  nanowire arrays on Ti-foil substrate for efficient flexible dye-sensitized solar cells. Energy Environ Sci. 2012;5(2):5750–7.
- 34. Fan J, Hao Y, Cabot A, Johansson EM, Boschloo G, Hagfeldt A. Cobalt (II/III) redox electrolyte in ZnO nanowire-based dye-sensitized solar cells. ACS Appl Mater Interfaces. 2013;5(6):1902–6.
- 35. Yang SC, Yang DJ, Kim J, Hong JM, Kim HG, Kim ID, et al. Hollow TiO<sub>2</sub> hemispheres obtained by colloidal templating for application in dyesensitized solar cells. Adv Mater. 2008;20(5):1059–64.
- 36. Yang D-J, Yang S-C, Hong J-M, Lee H, Kim I-D. Size-dependent photovoltaic property in hollow hemisphere array based dye-sensitized solar cells. J Electroceramics. 2010;24(3):200–4.
- 37. Yip CT, Huang H, Zhou L, Xie K, Wang Y, Feng T, et al. Direct and seamless coupling of TiO<sub>2</sub> nanotube photonic crystal to dye-sensitized solar cell: a single‐step approach. Adv Mater. 2011;23(47):5624–8.
- 38. Lin J, Guo M, Yip CT, Lu W, Zhang G, Liu X, et al. High temperature crystallization of free-standing anatase  $TiO<sub>2</sub>$  nanotube membranes for high efficiency dye-sensitized solar cells. Adv Funct Mater. 2013;23(47):5952–60.
- 39. Zhou Y, Chen L, Dai H, Tao Y. LIU J-g, Hu Y. Porous, single crystalline titanium nitride nanoplates grown on carbon fibers: an excellent counter electrode for low-cost, high performance, fiber-shaped dye-sensitized solar cells. Chemical Communications. 2014;50:14321–4.
- 40. Yun D-J, Ra H, Rhee S-W. Concentration effect of multiwalled carbon nanotube and poly (3,4-ethylenedioxythiophene) polymerized with poly(4-styrenesulfonate) conjugated film on the catalytic activity for counter electrode in dye sensitized solar cells. Renew Energy. 2013;50:692–700.
- 41. Yi L, Liu Y, Yang N, Tang Z, Zhao H, Ma G, et al. One dimensional CulnS<sub>2</sub>-ZnS heterostructured nanomaterials as low-cost and high-performance counter electrodes of dye-sensitized solar cells. Energy Environ Sci. 2013;6(3):835–40.
- <span id="page-9-0"></span>42. Xu X, Huang D, Cao K, Wang M, Zakeeruddin SM, Grätzel M. Electrochemically reduced graphene oxide multilayer films as efficient counter electrode for dye-sensitized solar cells. Scientific Reports. 2013;3:1–7.
- 43. Gong F, Xu X, Li Z, Zhou G, Wang Z-S. NiSe<sub>2</sub> as an efficient electrocatalyst for a Pt-free counter electrode of dye-sensitized solar cells. Chem Commun. 2013;49(14):1437–9.
- 44. Ellis H, Vlachopoulos N, Häggman L, Perruchot C, Jouini M, Boschloo G, et al. PEDOT counter electrodes for dye-sensitized solar cells prepared by aqueous micellar electrodeposition. Electrochim Acta. 2013;107:45–51.
- 45. Chang L-H, Hsieh C-K, Hsiao M-C, Chiang J-C, Liu P-I, Ho K-K, et al. A graphene-multi-walled carbon nanotube hybrid supported on fluorinated tin oxide as a counter electrode of dye-sensitized solar cells. J Power Sources. 2013;222:518–25.
- 46. Bu C, Tai Q, Liu Y, Guo S, Zhao X. A transparent and stable polypyrrole counter electrode for dye-sensitized solar cell. J Power Sources. 2013;221:78–83.
- 47. Yella A, Lee H-W, Tsao HN, Yi C, Chandiran AK, Nazeeruddin MK, et al. Porphyrin-sensitized solar cells with cobalt (II/III)–based redox electrolyte exceed 12 percent efficiency. Science. 2011;334(6056):629–34.
- 48. Bai Y, Cao Y, Zhang J, Wang M, Li R, Wang P, et al. High-performance dye-sensitized solar cells based on solvent-free electrolytes produced from eutectic melts. Nat Mater. 2008;7(8):626–30.
- 49. Snaith HJ, Schmidt-Mende L. Advances in liquid electrolyte and solid-state dye-sensitized solar cells. Adv Mater. 2007;19(20):3187–200.
- 50. Wang P, Zakeeruddin SM, Moser JE, Nazeeruddin MK, Sekiguchi T, Grätzel M. A stable quasi-solid-state dye-sensitized solar cell with an amphiphilic ruthenium sensitizer and polymer gel electrolyte. Nat Mater. 2003;2(6):402–7.
- 51. Nogueira AF, Durrant JR, De Paoli MA. Dye-sensitized nanocrystalline solar cells employing a polymer electrolyte. Adv Mater. 2001;13(11):826–30.
- 52. Park H, Kim H-G, Choi W-Y. Characterizations of highly ordered TiO<sub>2</sub> nanotube arrays obtained by anodic oxidation. J Trans Electr Electron Mater. 2010;11:112–5.
- 53. Mor G, Varghese O, Paulose M, Shankar K, Grimes C. A review on highly ordered, vertically oriented TiO<sub>2</sub> nanotube arrays: fabrication, material properties, and solar energy applications. Sol Energy Mater Sol Cells. 2006;90(14):2011–75.
- 54. Adachi M, Murata Y, Takao J, Jiu J, Sakamoto M, Wang F. Highly efficient dye-sensitized solar cells with a titania thin-film electrode composed of a network structure of single-crystal-like  $TiO<sub>2</sub>$  nanowires made by the "oriented attachment" mechanism. J Am Chem Soc. 2004;126(45):14943–9.
- 55. Vittadini A, Selloni A, Rotzinger F, Grätzel M. Structure and energetics of water adsorbed at TiO<sub>2</sub> anatase (101) and (001) surfaces. Phys Rev Lett. 1998;81(14):2954.

# **Submit your manuscript to a journal and benefit from:**

- $\blacktriangleright$  Convenient online submission
- Rigorous peer review
- **Figure 1** Immediate publication on acceptance
- ▶ Open access: articles freely available online
- $\blacktriangleright$  High visibility within the field
- $\blacktriangleright$  Retaining the copyright to your article

**Submit your next manuscript at** 7 **springeropen.com**