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# Modelling non-symmetric collagen fibre dispersion in arterial walls

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New experimental results on collagen fibre dispersion in human arterial layers have shown that the dispersion in the tangential plane is more significant than that out of plane. A rotationally symmetric dispersion model is not able to capture this distinction. For this reason, we introduce a new nonsymmetric dispersion model, based on the bivariate von Mises distribution, which is used to construct a new structure tensor. The latter is incorporated in a strain-energy function that accommodates both the mechanical and structural features of the material, extending our rotationally symmetric dispersion model (Gasser et al. 2006 J. R. Soc. Interface 3, 15-35. (doi:10.1098/ rsif.2005.0073)). We provide specific ranges for the dispersion parameters and show how previous models can be deduced as special cases. We also provide explicit expressions for the stress and elasticity tensors in the Lagrangian description that are needed for a finite-element implementation. Material and structural parameters were obtained by fitting predictions of the model to experimental data obtained from human abdominal aortic adventitia. In a finite-element example, we analyse the influence of the fibre dispersion on the homogeneous biaxial mechanical response of aortic strips, and in a final example the non-homogeneous stress distribution is obtained for circumferential and axial strips under fixed extension. It has recently become apparent that this more general model is needed for describing the mechanical behaviour of a variety of fibrous tissues.

## 1. Introduction

Physiological and pathological changes in the cardiovascular system directly influence the mechanical behaviour of arterial walls [1]. It is, therefore, of crucial importance to improve understanding of the mechanical properties of the constituents of arterial walls, including the inherent features of anisotropy and nonlinearity. These properties, among others, pose formidable challenges in the constitutive modelling and numerical analysis of such tissues and can be clearly connected to the underlying structure of the tissues. The passive mechanical behaviour of an arterial wall is governed mainly by the matrix material (which consists of water, elastin, proteoglycans, etc.) and the collagen fibre reinforcement. The anisotropy is associated with the local mean alignment of the collagen fibres, which also stiffen their response when under tension, leading to their significant nonlinear characteristics. The fibres are not perfectly aligned but are dispersed around their mean direction, and the amount and character of the dispersion depends on the topography, the particular layer of the vessel considered and the respective (patho)physiological condition, inter alia. Fibre dispersion in arterial walls has been documented and analysed in, for example, [2–5]; for an overview of the structural quantification of collagen fibres in arterial walls, see [6]. In particular, the study in [5] identified the presence of two fibre families in the intima, media and adventitia of human aortas; they are helically and almost symmetrically arranged with respect to the cylinder axis. Often a third and sometimes a fourth family is present in the intima in the respective axial and circumferential directions. Recent work [7] has revealed that, while helical fibre structures are present in human elastic arteries, in more muscular arteries (as for the murine basilar artery) and veins (such as the porcine jugular vein) a transition from the helical arrangement to two nearly orthogonal fibre families aligned in the circumferential and axial directions can be observed, and it is suggested that this is to ensure optimal efficiency of the vasculature. Observations of dispersion for other tissues, including the myocardium, corneas and articular cartilage, can be found in [8,9], [10,11] and [12], respectively.

Several mechanical models accounting for the dispersion of collagen fibres have been proposed. Fibre dispersion can be represented either directly by incorporation in a strainenergy function via a probability density function (PDF) or by a generalized structure tensor, for example. Following [13], these two approaches are referred to as 'angular integration' (AI) and 'generalized structure tensor' (GST). Cortes *et al.* [13] compared the results of the two formulations on the basis of the energy function introduced in [14] and the GST in [15]. As was pointed out in, for example, [16] one approach is to consider the strain energy  $w(\lambda)$  of a single collagen fibre as a function of the fibre stretch  $\lambda$  and to integrate this over the unit sphere *S* to obtain the total free-energy function  $\Psi_{\rm f}$  of the fibres per unit reference volume, i.e.

$$\Psi_{\rm f} = n \int_{S} \rho(\mathbf{N}) w(\lambda) \mathrm{d}S, \qquad (1.1)$$

where *n* is the number of fibres per unit reference volume, **N** is a unit vector describing the orientation of an individual fibre and  $\rho$  is the relative angular density of fibres normalized according to

$$\frac{1}{4\pi} \int_{S} \rho(\mathbf{N}) \mathrm{d}S = 1. \tag{1.2}$$

The region of the unit sphere where fibres are in tension is defined by the set of N for which  $\lambda > 1$ , where  $\lambda = (C : N \otimes N)^{1/2}$  is the stretch in the direction N, C is the right Cauchy–Green tensor and a double contraction is defined by  $C : N \otimes N = (CN) \cdot N$ . This is the AI approach.

In the GST approach, the energy function is associated with a GST **H** and is given by

$$\Psi_{\rm f} = \Psi_{\rm f}({\rm C},\,{\rm H}),\tag{1.3}$$

where H is defined by

$$\mathbf{H} = \frac{1}{4\pi} \int_{S} \rho(\mathbf{N}) \mathbf{N} \otimes \mathbf{N} dS, \quad \text{with} \quad \text{tr} \mathbf{H} = 1, \qquad (1.4)$$

the latter following from (1.2).

By comparing these approaches, we see that (1.1) requires integration over the unit sphere at each point, while for the strain-energy function (1.3) no such integration is needed once **H** has been determined by  $(1.4)_1$  or prescribed. On the one hand, the AI approach allows identification and exclusion of individual fibres under compression, while, on the other hand, this is not quite so straightforward on the basis of  $(1.4)_1$ . For a detailed discussion of the latter issue, see [17,18].

In chronological order, we now provide a short overview of the main existing continuum mechanical models which take fibre dispersion into account. Probably the study in [19] was the first to consider fibre dispersion in the analysis of fibrous connective tissues, with the tissue structure (fibre orientation) being accounted for in the strain energy via an orientation density function. The approach [20] incorporates a two-dimensional AI distribution in a strain-energy function based on the Beta distribution, but neglects the out-of-plane dispersion of the fibres, while a planar fibre dispersion was used in [21] with a Gaussian PDF to study the biaxial behaviour of arterial walls and aortic valves. In [22], the Gaussian distribution was used to compute the so-called splay invariants to represent two- and three-dimensional fibre dispersions, which the authors applied to aortic valve tissues.

In [14], the use of an exponential strain-energy function was motivated following [23], and this model was extended to the case of fibre dispersion in [15] based on the structure tensor  $(1.4)_1$ . Therein, we used a rotationally symmetric distribution of collagen fibres. In addition, the model [15] was applied to several other tissues, including the cornea [24] and articular cartilage [25], while, in [26], the constitutive model of [14] was applied with the fibre orientations uniformly distributed over the azimuthal angle. A structure tensor based on a planar counterpart of that in [15] was defined in [27,28] and used in [29].

A PDF based on the von Mises distribution taking account of a non-rotationally symmetric fibre dispersion and based on the micro-sphere model was suggested in [30], and an ellipsoidal distribution with a power-law strain-energy function, based on the AI formulation, was employed in [31] and applied to cartilage. Based on the AI approach with the von Mises distribution Raghupathy & Barocas [32] derived a closedform solution for a simple exponential fibre stress-strain law and applied their model to planar biaxial extension of a bioartificial tissue. The study of Federico & Gasser [33] adopted the model of Gasser et al. [15] and included the limiting case of an in-plane arrangement of fibres following the AI approach, which was also used in [34] with a planar von Mises distribution to examine the in-plane dispersion of collagen fibres. In [35], the Bingham distribution was used and was claimed to be 'clearly superior to the  $\pi$ -periodic von Mises distribution in modelling the collagen organization in vascular tissue', which we have not found to be the case in this study. The model [15] was extended in [36] to incorporate a higher order statistical measure of dispersion in order to reduce differences between the GST and AI formulations. In a recent paper [17], we have introduced a fibre dispersion model with a weighted energy function that allows the exclusion of fibres which are under compression. The model, based on the AI approach, has been developed for plane strain and for three-dimensional deformations appropriate for finite-element implementation (see also the discussion in [18] on the exclusion of compressed fibres using the GST approach). A summary of the main models discussed above is listed in table 1.

The model in [15] has proved to be very successful, but recently it was shown in [5] that an axisymmetric model of collagen fibre dispersion is not appropriate, and a more general dispersion model is required to accommodate the new findings. Hence, based on the structure tensor approach initiated in [15], we introduce here a bivariate von Mises distribution that enables the dispersion data to be captured. In particular, for arteries, the out-of-plane dispersion is relatively narrow while the in-plane dispersion is more significant. This work provides a natural extension of the constitutive setting documented in [14,15] to a more general context.

This work is structured as follows. In §2, we introduce the mathematical framework for describing fibre dispersion illustrated by the use of the bivariate von Mises distribution. This is particularly appropriate for use with the new experimental data, which indicate that the current models are not

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**Table 1.** Overview of the main existing continuum mechanical models which take fibre dispersion into account, listed in chronological order of appearance. The column 'formulation' describes the approach originally presented in the cited paper (GST, generalized structure tensor; AI, angular integration); PDF, probability density function; 'parameters' refer to the parameters describing the fibre dispersion (not material or any other parameters). The abbreviations 'RS', 'NS' and 'IP' refer to rotationally symmetric, non-symmetric and in-plane, respectively. For the parameters in some of the references listed, we have used notation from this paper for consistency.

refs	formulation	PDF	parameters	2D/3D	dispersion
Sacks [20]	AI	Beta	γ, δ	2D	IP
Driessen <i>et al.</i> [21]	AI	Gaussian	σ	2D	IP
Freed & Doehring [37]	AI/GST	Gaussian	σ	2D/3D	RS
Gasser et al. [15]	GST	von Mises	к	3D	RS
Alastrué <i>et al</i> . [30]	AI	von Mises	various	3D	NS
Ateshian <i>et al.</i> [31]	AI	ellipsoidal	$\xi_{i},  \alpha_{i},  i = 1,  2,  3$	3D	NS
Raghupathy & Barocas [32]	AI	von Mises	Ь	2D	IP
Federico & Gasser [33]	AI	von Mises	Ь	3D	RS
Agianniotos <i>et al.</i> [34]	AI	von Mises	Ь	2D	IP
Gasser et al. [35]	AI	Bingham	κ <sub>1</sub> , κ <sub>2</sub>	3D	NS
Pandolfi & Vasta [36]	GST/AI	von Mises	κ, κ	3D	RS
Holzapfel & Ogden [17]	AI	von Mises	А, В, С, D, <i>к</i>	2D/3D	RS/NS
Melnik <i>et al.</i> [18]	GST	von Mises	к	3D	RS
this work	GST	bivariate von Mises	к <sub>ір</sub> , к <sub>ор</sub>	3D	RS/NS/IP

sufficiently general to capture the collagen fibre structure. Then we introduce a PDF which accounts for these experimental observations, and it is shown how the new model reproduces several special cases from the literature. The continuum mechanical framework associated with the dispersion structure introduced in §2 is presented in §3. In §4, the theory of §3 is applied in three representative numerical simulations with the aim of showing the efficacy and capability of the proposed structural model. Finally, §5 summarizes the proposed fibre dispersion model and discusses future developments of our work.

# 2. Mathematical representation of fibre dispersion

#### 2.1. Structure tensor for the fibre dispersion

Motivated by the experimental results documented in [38], we introduce the coordinate system shown in figure 1, where the unit vector **N** is an arbitrary fibre direction in the reference (undeformed) configuration, expressed in terms of the two angles  $\Phi$  and  $\Theta$  by

$$\mathbf{N}(\Phi, \Theta) = \cos \Theta \cos \Phi \mathbf{e}_1 + \cos \Theta \sin \Phi \mathbf{e}_2 + \sin \Theta \mathbf{e}_3, \quad (2.1)$$

where  $\Phi \in [0, 2\pi]$ ,  $\Theta \in [-\pi/2, \pi/2]$  and  $\mathbf{e}_1, \mathbf{e}_2, \mathbf{e}_3$  are unit rectangular Cartesian basis vectors. Locally,  $\mathbf{e}_1$  and  $\mathbf{e}_2$  define the tangential plane of a cylindrical coordinate system and  $\mathbf{e}_3$  the corresponding outward radial direction. For a circular cylinder,  $\mathbf{e}_1$  is taken to be the circumferential direction and  $\mathbf{e}_2$  the axial direction. Although the coordinate system is similar to that introduced in [15], there is a subtle but important difference: our approach does not involve symmetry about a preferred direction since recent experimental results [38] have suggested that the fibre dispersion is not rotationally symmetric.



**Figure 1.** A unit vector  $\mathbf{N}(\Phi, \Theta)$  representing a general fibre direction defined by the angles  $\Phi$  and  $\Theta$  with respect to rectangular Cartesian unit basis vectors  $\mathbf{e}_1$ ,  $\mathbf{e}_2$ ,  $\mathbf{e}_3$ .

We describe  $\rho(\mathbf{N})$ , which appears in (1.1), as the probability density of the fibre orientation  $\mathbf{N}$  in the reference configuration as a function of  $\Phi$  and  $\Theta$ . In the following, we write either  $\rho(\mathbf{N})$  or  $\rho(\Phi, \Theta)$ , and this is normalized according to (1.2), equivalently

$$\frac{1}{4\pi} \int_{S} \rho(\Phi, \Theta) \mathrm{d}S = 1, \qquad (2.2)$$

where  $dS = \cos\Theta d\Phi d\Theta$ . (Note that the usual spherical polar angles are  $\pi/2 - \Theta$  and  $\Phi$ .)

Another requirement is that the PDF has to be independent of the sense of **N** so that  $\rho(\mathbf{N}) \equiv \rho(-\mathbf{N})$ , which is equivalent to  $\rho(\Phi, \Theta) = \rho(\Phi + \pi, -\Theta)$ . Based on the experimental results presented in [38], we introduce two additional symmetries, namely the in-plane symmetry  $\rho(\Phi, \Theta) = \rho(\Phi + \pi, \Theta)$  and the out-of-plane symmetry  $\rho(\Phi, \Theta) = \rho(\Phi, -\Theta)$ .

It is assumed that the material behaviour does not depend on the sense of N, so that the strain energy depends on N only through the tensor product  $N \otimes N$  ([14]; see also [15,39]), via a symmetric second-order tensor, introduced in  $(1.4)_1$ , which we now write as

$$\mathbf{H} = \frac{1}{4\pi} \int_{\Phi=0}^{2\pi} \int_{\Theta=-\pi/2}^{\pi/2} \rho(\Phi, \Theta) \mathbf{N} \otimes \mathbf{N} \cos \Theta \mathrm{d}\Phi \mathrm{d}\Theta, \qquad (2.3)$$

the components of which involve only  $\rho(\mathbf{N})$  and the sines and cosines of the angles  $\Phi_{\ell} \Theta$ . The tensor **H** is a structure tensor involving the fibre dispersion via  $\rho(\mathbf{N})$ . This is associated with a dispersion of fibres about a single mean direction, the mean direction of N, which, according to the data in [38], has a very small component out of plane. We therefore assume here that the mean fibre direction, say M, lies in the tangential plane defined by  $\mathbf{e}_1$  and  $\mathbf{e}_2$ . An explicit expression for **M** will be written down below.

In terms of its components  $H_{ii}$ , we write  $\mathbf{H} = H_{ii} \mathbf{e}_i \otimes \mathbf{e}_i$  and note that, due to the symmetries of  $\rho(\mathbf{N})$ , the off-diagonal components  $H_{13}$  and  $H_{23}$  vanish, and the only non-zero components of **H** are therefore  $H_{12}$  and the diagonal components  $H_{11}$ ,  $H_{22}$ ,  $H_{33}$ . Thus, **H** has four non-zero components, and, because tr H = 1 by  $(1.4)_2$ , only three of them are independent:

$$H_{11}, H_{22}, H_{12}, \text{ with } H_{33} = 1 - H_{11} - H_{22}.$$
 (2.4)

In [38], it was also observed that the dispersions in the two planes are essentially independent, which means that the PDF can be decomposed in the form

$$\rho(\Phi, \Theta) = \rho_{\rm ip}(\Phi)\rho_{\rm op}(\Theta), \qquad (2.5)$$

where  $\rho_{\rm ip}(\Phi)$  and  $\rho_{\rm op}(\Theta)$  describe the in-plane and out-ofplane dispersions, respectively. Since the symmetry of the PDF has to be fulfilled, we require that

$$\rho_{\rm ip}(\Phi) = \rho_{\rm ip}(\Phi + \pi) \quad \text{and} \quad \rho_{\rm op}(\Theta) = \rho_{\rm op}(-\Theta)$$

Without taking explicit account of these symmetries, the normalization in equation (2.2) is now written as

$$\frac{1}{4\pi} \int_{0}^{2\pi} \rho_{\rm ip}(\Phi) \mathrm{d}\Phi \int_{-\pi/2}^{\pi/2} \rho_{\rm op}(\Theta) \cos\Theta \mathrm{d}\Theta = 1.$$
 (2.6)

Without loss of generality, we can choose the normalization of  $\rho_{ip}$  so that

$$\frac{1}{2\pi} \int_{0}^{2\pi} \rho_{\rm ip}(\Phi) \mathrm{d}\Phi = 1, \qquad (2.7)$$

and hence (2.6) reduces to

$$\int_{-\pi/2}^{\pi/2} \rho_{\rm op}(\Theta) \cos \Theta d\Theta = 2.$$
 (2.8)

Let us now define

$$\kappa_{\rm op} = \frac{1}{4} \int_{-\pi/2}^{\pi/2} \rho_{\rm op}(\Theta) \cos^3 \Theta d\Theta, \qquad (2.9)$$

which is a measure of the out-of-plane dispersion and is consistent with the definition, in slightly different notation, given in eqn (4.1) in [15] for the case in which the dispersion is rotationally symmetric. By definition,  $\kappa_{op}$  is non-negative and from (2.8) and (2.9) it must satisfy  $\kappa_{op} \leq 1/2$ . Thus,

$$0 \le \kappa_{\rm op} \le 1/2. \tag{2.10}$$

If  $\rho_{\rm op} \equiv 1$ , giving an isotropic distribution, then  $\kappa_{\rm op} = 1/3$ . Note that if  $\rho_{ip} = 1$  then (2.7) is automatically satisfied, but on the other hand equation (2.7) does not necessarily imply that  $\rho_{\rm ip} = 1$ . However, when  $\rho_{\rm ip} = 1$  we obtain  $H_{11} = H_{22} = \kappa_{\rm op}$ ,  $H_{33} = 1 - 2\kappa_{op}$  and  $H_{12} = 0$  and this corresponds to a



**Figure 2.** Unit vector **M** in the  $(\mathbf{e}_1, \mathbf{e}_2)$  plane,  $\alpha$  is the angle between **M** and  $\mathbf{e}_{1}$ , while  $\mathbf{M}_{n}=\mathbf{e}_{3}$  is the unit out-of-plane vector given by  $\mathbf{e}_{1}\times\mathbf{e}_{2}$ .

transversely isotropic distribution with mean fibre direction  $\mathbf{e}_3$ . The lower limit  $\kappa_{op} = 0$  corresponds to the case in which all the fibres are in the  $e_3$  direction (no dispersion), while the upper limit  $\kappa_{op} = 1/2$  corresponds to a planar distribution in the  $(\mathbf{e}_1, \mathbf{e}_2)$  plane. For isotropy, we have  $\rho_{ip} =$  $\rho_{\rm op}=1$  and  $\kappa_{\rm op}=1/3.$  In general,  $\rho_{\rm ip}$  and  $\rho_{\rm op}$  are separate measures of the fibre dispersions, in plane and out of plane, respectively.

We now define  $\bar{\kappa}_{11}$ ,  $\bar{\kappa}_{22}$  and  $\bar{\kappa}_{12}$  by

$$\bar{\kappa}_{11} = \frac{1}{2\pi} \int_{0}^{2\pi} \rho_{\rm ip}(\Phi) \cos^2 \Phi d\Phi,$$
  
$$\bar{\kappa}_{22} = \frac{1}{2\pi} \int_{0}^{2\pi} \rho_{\rm ip}(\Phi) \sin^2 \Phi d\Phi$$
 (2.11)

and

$$\bar{\kappa}_{12} = \frac{1}{2\pi} \int_0^{2\pi} \rho_{\rm ip}(\Phi) \sin \Phi \cos \Phi \mathrm{d}\Phi, \qquad (2.12)$$

and hence by (2.11)

$$\bar{\kappa}_{11} + \bar{\kappa}_{22} = 1.$$
 (2.13)

Together,  $\bar{\kappa}_{11}$ ,  $\bar{\kappa}_{22}$  and  $\bar{\kappa}_{12}$  characterize the in-plane dispersion. Note that  $\bar{\kappa}_{12} = 0$  when  $\rho_{ip} = 1$  and also when the mean in-plane direction is either  $\mathbf{e}_1$  or  $\mathbf{e}_2$ , as discussed below. In general, we now have

$$\begin{aligned} H_{ab} &= 2\kappa_{op}\bar{\kappa}_{ab}, \quad a, b \in \{1, 2\}, \quad H_{11} + H_{22} = 2\kappa_{op} \\ \text{and} \quad H_{33} &= 1 - 2\kappa_{op}. \end{aligned}$$
 (2.14)

At this point, we note that the structure tensor **H** has the form

$$\mathbf{H} = \sum_{j=1}^{3} H_{jj} \mathbf{e}_{j} \otimes \mathbf{e}_{j} + H_{12} (\mathbf{e}_{1} \otimes \mathbf{e}_{2} + \mathbf{e}_{2} \otimes \mathbf{e}_{1}), \qquad (2.15)$$

and we emphasize, as noted above, that it is associated with a dispersion that has a single mean direction. We shall relate this to the model of dispersion discussed in [15] when considering special cases in §2.2.

An alternative way of writing the above representation is the form

$$\mathbf{H} = A\mathbf{I} + B\mathbf{M} \otimes \mathbf{M} + (1 - 3A - B)\mathbf{M}_{n} \otimes \mathbf{M}_{n}, \qquad (2.16)$$

where the unit vector  $\mathbf{M} = \cos \alpha \mathbf{e}_1 + \sin \alpha \mathbf{e}_2$  is the in-plane mean fibre direction,  $\alpha$  being the angle between **M** and **e**<sub>1</sub>, and  $\mathbf{M}_n$  is a unit out-of-plane vector (see figure 2), while A and B are constants.

By comparing (2.15) and (2.16), we find that the components  $H_{ii}$  are related to A, B and the angle  $\alpha$  by

$$H_{11} = A + B\cos^2 \alpha, \quad H_{22} = A + B\sin^2 \alpha \quad \text{and} \\ H_{12} = B\sin\alpha\cos\alpha, \qquad (2.17)$$

from which we obtain

$$2A = 2\kappa_{\rm op} - B, \quad B^2 = (H_{11} - H_{22})^2 + 4H_{12}^2$$
 (2.18)

and

$$\tan 2\alpha = \frac{2\bar{\kappa}_{12}}{\bar{\kappa}_{11} - \bar{\kappa}_{22}}.$$
 (2.19)

By defining  $B = 2\kappa_{op}\bar{B}$  and  $A = \kappa_{op}\bar{A}$ , we obtain

$$\bar{A} = 1 - \bar{B}$$
 and  $\bar{B}^2 = (\bar{\kappa}_{11} - \bar{\kappa}_{22})^2 + 4\bar{\kappa}_{12}^2$ , (2.20)

where  $\overline{A}$  and  $\overline{B}$  depend only on the in-plane distribution. Note that  $\overline{B}$  is invariant with respect to the rotation of the in-plane axes.

For any given distribution  $\rho_{\rm ip}$ , we can calculate  $\bar{\kappa}_{11}$ ,  $\bar{\kappa}_{22}$ and  $\bar{\kappa}_{12}$  and hence the angle  $\alpha$  and the constants  $\bar{A}$  and  $\bar{B}$ , while  $\kappa_{\rm op}$  is obtained when  $\rho_{\rm op}$  is given. Note that **H** contains only two independent parameters. Note also that for  $\alpha = 0$  or  $\pi/2$  the dispersion parameter  $\bar{\kappa}_{12} = 0$ .

In §2.1.1, we consider a particular choice of the density functions  $\rho_{\rm ip}$  and  $\rho_{\rm op}$ .

# 2.1.1. Using the bivariate von Mises distribution as the dispersion distribution

The work of Schriefl *et al.* [38] documented angular datasets for the in-plane collagen dispersions of the intima, media and adventitia of human non-atherosclerotic thoracic and abdominal aortas and common iliac arteries. The out-ofplane angles were measured separately for each layer, showing that the out-of-plane dispersions are very similar at all anatomic locations and for each layer. Moreover, each mean fibre angle was found to be very close to tangential. Motivated by these results we now model, as illustrative examples, each of  $\rho_{ip}(\Phi)$  and  $\rho_{op}(\Theta)$  with  $\pi$ -periodic von Mises distributions [40], so the overall PDF is a bivariate von Mises distribution, featuring the symmetries that were discussed in §2.1.

For  $\rho_{\rm ip}(\Phi)$ , we consider the basic von Mises distribution

$$\rho_{\rm ip}(\Phi) = \frac{\exp\left(a\cos 2\Phi\right)}{I_0(a)},\tag{2.21}$$

where the so-called concentration parameter *a* is a constant and  $I_0(a)$  is the modified Bessel function of the first kind of order 0 defined by

$$I_0(x) = \frac{1}{\pi} \int_0^{\pi} \exp(x \cos \alpha) d\alpha, \qquad (2.22)$$

which provides a normalization factor leading to (2.11) independently of *a*. Equation (2.22) corresponds to a distribution that is symmetric about  $\Phi = 0$ . Plots of the circular distribution (2.25) for different concentration parameters *a* are shown in figure 3, and we note that, as  $a \to \infty$ ,  $\rho_{\rm ip}$  becomes a delta function. For this special case of symmetry, we use the notation  $\kappa_{11}$ ,  $\kappa_{22}$  and  $\kappa_{12}$ . It follows from (2.25) using the definitions (2.15) and (2.16) that  $\kappa_{12} = 0$  and

$$\kappa_{11} = \frac{1}{2} + \frac{I_1(a)}{2I_0(a)} \quad \text{and} \quad \kappa_{22} = \frac{1}{2} - \frac{I_1(a)}{2I_0(a)},$$
(2.23)

where

$$I_1(x) = \frac{1}{\pi} \int_0^{\pi} \exp(x \cos \alpha) \cos \alpha d\alpha \qquad (2.24)$$

is the modified Bessel function of the first kind of order 1.



**Figure 3.** The von Mises distribution (2.21) plotted against  $\Phi$  for three different values of the parameter *a*: (1, 3, 5), corresponding to the continuous, short dashed and long dashed curves, respectively. For a = 0 we have  $\rho_{\rm ip} \equiv 1$ , while, for  $a \to \infty$ ,  $\rho_{\rm ip}$  becomes a delta function.

If the distribution is symmetrical about  $\Phi = \alpha$  instead of  $\Phi = 0$ , then  $\Phi$  is replaced by  $\Phi - \alpha$  in (2.21) and the appropriate values of  $\bar{\kappa}_{11}$  and  $\bar{\kappa}_{12}$  are given by

$$\left. \begin{array}{l} \bar{\kappa}_{11} = \kappa_{11} \cos^2 \alpha + \kappa_{22} \sin^2 \alpha, \\ \bar{\kappa}_{12} = (\kappa_{11} - \kappa_{22}) \sin \alpha \cos \alpha \end{array} \right\}$$
(2.25)

and  $\bar{\kappa}_{22}$  is given by (2.13). From now on, whenever  $\kappa_{12} = 0$  we use the notation  $\kappa_{ip}$  instead of  $\kappa_{22}$ .

Equation (2.8) is satisfied by taking  $\rho_{\rm op}(\Theta)$  to be a von Mises distribution of the form

$$\rho_{\rm op}(\Theta) = 2\sqrt{\frac{2b}{\pi}} \frac{\exp\left[b(\cos 2\Theta - 1)\right]}{\operatorname{erf}(\sqrt{2b})},$$
(2.26)

where b is a constant concentration parameter and erf is the error function defined by

$$\operatorname{erf}(x) = \frac{2}{\sqrt{\pi}} \int_0^x \exp(-\xi^2) \mathrm{d}\xi.$$
 (2.27)

Note that (2.26) has a similar shape to (2.21) but is marginally different from the distribution used in [15], and gives a somewhat better fit to the narrow out-of-plane dispersion.

For the distribution (2.26), a closed-form expression for  $\kappa_{op}$  from (2.9) is obtained in the form

$$\kappa_{\rm op} = \frac{1}{2} - \frac{1}{8b} + \frac{1}{4} \sqrt{\frac{2}{\pi b}} \frac{\exp\left(-2b\right)}{\exp\left(\sqrt{2b}\right)}.$$
 (2.28)

Figure 4 shows a plot of  $\kappa_{op}$  as a function of the concentration parameter *b*. Note in particular that  $\kappa_{op} = 1/3$  when b = 0.

We use maximum-likelihood estimates to obtain the parameters *a* and *b* in the PDFs,  $\rho_{ip}(\Phi)$  and  $\rho_{op}(\Theta)$ , from the angular datasets of Schriefl *et al.* [38]. Although it is possible to determine the parameters of a PDF by minimizing the sum of squared errors, this method has several disadvantages, as pointed out in [5], and hence we prefer to identify the parameters by using a maximum-likelihood estimate. In figure 5, we show experimental data from the adventitia of a human non-atherosclerotic abdominal aorta obtained from picrosirius polarization, in combination with a universal stage [38]. To produce the fit, we used the maximum-likelihood estimate to obtain the concentration parameters *a* and *b*. In figure 5*a*, we show the in-plane bi-modal distribution of  $\rho_{ip}$  of  $\Phi$  with

6



**Figure 4.** Plot of  $\kappa_{op}$  versus the concentration parameter *b* according to equation (2.28). Note that  $0 \le \kappa_{op} \le 1/2$  and  $\kappa_{op} = 1/3$  when b = 0.

 $\alpha = \pm 47.99^{\circ}$  giving a = 2.54, which is obtained from (2.21) using  $\Phi + \alpha$  and  $\Phi - \alpha$  instead of  $\Phi$ . In figure 5*b*, we show the out-of-plane probability density  $\rho_{\rm op}$  of  $\Theta$  obtained from (2.26), where b = 19.44.

#### 2.2. Special cases of fibre dispersions

Our model includes several existing dispersion models as special cases of (2.16), which are discussed in the following section.

#### 2.2.1. Transversely isotropic dispersion

In [15], we considered a transversely isotropic dispersion for which **H** has the form

$$\mathbf{H} = \kappa \mathbf{I} + (1 - 3\kappa)\mathbf{M} \otimes \mathbf{M}, \qquad (2.29)$$

where  $\kappa$  is a single dispersion parameter associated with a transversely isotropic dispersion about **M**. Note that here we are taking **M** to lie in the ( $\mathbf{e}_1$ ,  $\mathbf{e}_2$ ) plane (figure 2). In [15],  $\kappa$  had the form

$$\kappa = \frac{1}{4} \int_0^{\pi} \rho(\Theta_{\kappa}) \sin^3 \Theta_{\kappa} \mathrm{d}\Theta_{\kappa},$$

where  $\Theta_{\kappa}$  is measured from **M** and is different from the  $\Theta$  used here. The connection between  $\Theta_{\kappa}$  and  $\Theta$  is obtained from  $\mathbf{M} \cdot \mathbf{N} = \cos \Theta_{\kappa}$ , with **N** given by (2.1) and  $\mathbf{M} = \cos \alpha \mathbf{e}_1 + \sin \alpha \mathbf{e}_2$ , in the form  $\cos \Theta_{\kappa} = \cos \Theta \cos(\Phi - \alpha)$ .

Equation (2.29) is recovered as a special case of (2.16) by taking  $\kappa = 1 - 2\kappa_{op}$ , which corresponds to  $A = \kappa$ ,  $B = 1 - 3\kappa$ .

#### 2.2.2. Perfect alignment

If both concentration parameters *a* and *b* become infinite, there is no dispersion in either plane and we obtain the model proposed in [14]. With  $a \to \infty$ ,  $\rho_{ip}$  becomes an inplane delta function, and with  $b \to \infty$  we have  $\kappa_{op} \to 1/2$ . The structure tensor is then  $\mathbf{H} = \mathbf{M} \otimes \mathbf{M}$ , and all fibres are oriented in the in-plane direction of  $\mathbf{M}$ . This corresponds to A = 0, B = 1 in (2.16).

#### 2.2.3. Isotropic dispersion

An isotropic fibre dispersion is represented by a uniform dispersion in each plane, meaning that  $\rho(\Phi, \Theta)$  is independent of  $\Phi$  and  $\Theta$ , with  $\rho_{\rm ip} = \rho_{\rm op} = 1$ , so that, for the von Mises distributions, a = b = 0, while  $\kappa_{\rm op} = 1/3$ , and the structure tensor is given by  $\mathbf{H} = (1/3)\mathbf{I}$ . Thus, there is no preferred direction ( $\kappa = 1/3$ ), with A = 1/3, B = 0 in (2.16).

#### 2.2.4. Planar dispersion

A dispersion with fibres oriented only in plane was presented in [27,39]. In this case, there is no out-of-plane contribution to  $\mathbf{H}$ , which can be written as

$$\mathbf{H} = \kappa \mathbf{1} + (1 - 2\kappa) \mathbf{M} \otimes \mathbf{M}, \qquad (2.30)$$

where **1** is the two-dimensional identity in the considered plane with the normalization and dispersion parameter given by

$$\frac{1}{\pi} \int_{-\pi/2}^{\pi/2} \rho(\Theta) d\Theta = 1 \quad \text{and} \quad \kappa = \frac{1}{\pi} \int_{-\pi/2}^{\pi/2} \rho(\Theta) \sin^2 \Theta d\Theta. \quad (2.31)$$

Note that the normalization  $(2.31)_1$  is equivalent to (2.7) bearing in mind the different ranges of angles used, and  $\kappa$  is equivalent to  $\bar{\kappa}_{22}$  as defined in  $(2.11)_2$ . The in-plane PDF  $\rho(\Theta)$  satisfies  $\rho(-\Theta) = \rho(\Theta)$ . For the von Mises distribution (2.26), this corresponds to  $b \to \infty$  and  $\kappa_{op} \to 1/2$ . Equation (2.30) is obtained from (2.16) by setting  $A = \kappa = \bar{\kappa}_{22}$ ,  $B = 1-2\kappa$ . Note that this is a two-dimensional distribution but its application is not restricted to use in two dimensions.

#### 2.2.5. Planar isotropic dispersion

If a dispersion features perfect out-of-plane alignment,  $b \rightarrow \infty$  and  $\kappa_{\rm op} \rightarrow 1/2$ , and is fully dispersed in plane so that  $a \rightarrow 0$ ,  $\rho_{\rm ip} = 1$ , then it is planar isotropic. For this case, the structure tensor is simply  $\mathbf{H} = (1/2)\mathbf{1}$ , corresponding to A = 1/2, B = 0 in (2.16).

#### 2.2.6. Summary of the special cases

The special cases discussed above are summarized in table 2. Figure 6 is a visualization of  $\rho(\mathbf{N})\mathbf{N}$  for (*a*) the general case for which **H** is given by equation (2.16), (*b*) the transversely isotropic dispersion given in §2.2.1 with **H** given by (2.29), (*c*) the case of perfect alignment according to §2.2.2, (*d*) the isotropic case according to §2.2.3 and (*e*) the case of planar dispersion given by equation (2.30).

### 3. Continuum mechanical framework

We consider a (stress-free) reference configuration, denoted  $\Omega_0$ , and a deformed (spatial) configuration, denoted  $\Omega$ . The deformation map  $\chi(X)$  transforms a material point  $X \in \Omega_0$  into a spatial point  $x \in \Omega$ . With this deformation map we define the deformation gradient  $F(X) = \partial \chi(x) / \partial X$  and its determinant  $J = \det F(X)$ , where J is the local volume ratio; we require J > 0.

Following [42,43], we apply the multiplicative decomposition of **F** into a spherical (dilatational) part  $J^{1/3}$ **I** and a unimodular (distortional) part  $\overline{\mathbf{F}} = J^{-1/3}\mathbf{F}$ , with det $\overline{\mathbf{F}} = 1$ . We define the right Cauchy–Green tensor and its modified counterpart as  $\mathbf{C} = \mathbf{F}^{\mathrm{T}}\mathbf{F}$  and  $\overline{\mathbf{C}} = \overline{\mathbf{F}}^{\mathrm{T}}\overline{\mathbf{F}}$ , respectively, with the related invariants  $I_1 = \text{tr}\mathbf{C}$  and  $\overline{I}_1 = \text{tr}\overline{\mathbf{C}}$ .



**Figure 5.** (*a*) Probability density  $\rho_{ip}(\Phi + \alpha)$ ,  $\rho_{ip}(\Phi - \alpha)$  plotted against  $\Phi$  with the concentration parameter a = 2.54 and with two mean fibre directions located at  $\alpha = \pm 47.99^{\circ}$  and (*b*) probability density  $\rho_{op}(\Theta)$  against  $\Theta$  with the parameter b = 19.44.



**Figure 6.** Visualization of the fibre dispersion defined by  $\rho(\mathbf{N})\mathbf{N}$ , with  $\rho = \rho_{ip}(\Psi)\rho_{op}(\Theta)$  according to (2.21) and (2.26), where the distance from the centre to the surface represents the probability of finding a fibre in the direction **N**. The plots have been scaled differently and represent (*a*) a non-rotationally symmetric dispersion, (*b*) a rotationally symmetric dispersion, (*c*) perfectly aligned fibres, (*d*) a three-dimensional isotropic fibre dispersion and (*e*) a planar fibre dispersion. The planar isotropic case corresponds to a circle in (*e*). (Online version in colour.)

 Table 2. Special cases of the present model based on the von Mises distributions (2.21) and (2.26). TI, transversely isotropic dispersion; ID, isotropic dispersion;

 PA, perfect alignment; PD, planar dispersion; PI, planar isotropic dispersion.

case	conc. parameter	dispersion parameter	structure tensor H	reference
TI	_	$\kappa = 1 - 2\kappa_{\rm op}$	$\kappa \mathbf{I} + (1 - 3\kappa) \mathbf{M} \otimes \mathbf{M}$	[15]
PA	$a ightarrow\infty$ , $b ightarrow\infty$	$\kappa_{ m op}  ightarrow$ 1/2, $\kappa = 0$	$\mathbf{M}\otimes\mathbf{M}$	[14]
ID	a = b = 0	$\kappa_{\rm op} = \kappa = 1/3$	(1/3) <b>I</b>	[15,41]
PD	$b  ightarrow \infty$	$\kappa_{ m op}  ightarrow$ 1/2, $\kappa = ar{\kappa}_{ m 22}$	$\kappa 1 + (1 - 2\kappa) \mathbf{M} \otimes \mathbf{M}$	[27,39]
PI	$a  ightarrow$ 0, $b  ightarrow \infty$	$\kappa_{ m op}  ightarrow 1/2$	(1/2) <b>1</b>	[35,39]

Since we treat an artery as an elastic material, we assume the existence of a strain-energy function  $\Psi(C, H_4, H_6)$  that depends on the macroscopic deformation through C and the underlying tissue structure through the structure tensors  $H_4$  and  $H_6$ , which describe the fibre alignment and dispersion for two fibre families. Based on (2.16), these are defined by

$$\mathbf{H}_i = A\mathbf{I} + B\mathbf{M}_i \otimes \mathbf{M}_i + (1 - 3A - B)\mathbf{M}_n \otimes \mathbf{M}_n, \quad i = 4, 6$$
(3.1)

and  $M_4$  and  $M_6$  lie in the  $(e_1, e_2)$  plane, and  $M_n$  normal to that plane.

For computational purposes, we assume that it is possible to split the strain-energy function into two parts as  $\Psi(\mathbf{C}, \mathbf{H}_4, \mathbf{H}_6) = \Psi_{vol}(J) + \Psi_{iso}(\bar{\mathbf{C}}, \mathbf{H}_4, \mathbf{H}_6)$ , as shown in [5,44]. The function  $\Psi_{vol}$  is a purely volumetric contribution while  $\Psi_{iso}$  represents the energy contribution of an isochoric (volume preserving) deformation through  $\bar{\mathbf{C}}$ . The second Piola–Kirchhoff stress tensor **S** is given by  $\mathbf{S} = 2\partial \Psi / \partial \mathbf{C}$ . Using the decoupled form of  $\Psi$ , we can identify two stress

parameter	interpretation	range	identification
c (kPa)	stiffness of (isotropic) ground matrix	(0, ∞)	
k <sub>1</sub> (kPa)	parameter with dimension of stress	[0, ∞)	mechanical tests
k <sub>2</sub> (–)	dimensionless parameter	[0, ∞)	
к <sub>ip</sub>	dispersion in plane	[0, 1]	)
к <sub>ор</sub>	dispersion out of plane	[0, 1/2]	hictology/imaging
<b>M</b> <sub>4</sub> , <b>M</b> <sub>6</sub>	mean fibre directions	$\alpha \in [0, \pi/2]$	
M <sub>n</sub>	normal to the mean fibre plane	—	J

contributions so that  $S = S_{\rm vol} + S_{\rm iso}$ . Using well-known results from tensor analysis (e.g. [44]), and the chain rule, we obtain

$$\mathbf{S}_{\text{vol}} = p J \mathbf{C}^{-1}, \quad \mathbf{S}_{\text{iso}} = J^{-2/3} \text{Dev} \bar{\mathbf{S}}, \quad (3.2)$$

where  $p = d\Psi_{vol}(J)/dJ$  is the constitutive equation for the hydrostatic pressure and

$$\bar{\mathbf{S}} = 2 \frac{\partial \Psi_{\text{iso}}}{\partial \bar{\mathbf{C}}}$$
(3.3)

is the so-called 'fictitious' isochoric second Piola–Kirchhoff stress tensor. The deviator in the Lagrangian configuration is defined by  $\text{Dev}(\bullet) = \mathbb{P} : (\bullet)$ , where  $\mathbb{P} = \mathbb{I} - (1/3)\mathbf{C}^{-1} \otimes \mathbf{C}$  is a projection tensor that furnishes the correct deviatoric operator in the Lagrangian setting, and  $(\mathbb{I})_{ABCD} = 1/2(\delta_{AC}\delta_{BD} + \delta_{AD}\delta_{BC})$  is a fourth-order identity tensor.

The related elasticity tensor  ${\mathbb C}$  in the Lagrangian description is written in the decoupled form

$$\mathbb{C} = 2 \frac{\partial \mathbf{S}(\mathbf{C})}{\partial \mathbf{C}} = \mathbb{C}_{vol} + \mathbb{C}_{iso}, \quad \mathbb{C}_{vol} = 2 \frac{\partial \mathbf{S}_{vol}}{\partial \mathbf{C}} \\ \text{and} \quad \mathbb{C}_{iso} = 2 \frac{\partial \mathbf{S}_{iso}}{\partial \mathbf{C}}. \tag{3.4}$$

According to [44], we have the specification

$$\mathbb{C}_{\text{vol}} = J\tilde{p}\mathbf{C}^{-1} \otimes \mathbf{C}^{-1} - 2Jp\mathbf{C}^{-1} \odot \mathbf{C}^{-1}$$
(3.5)

and

$$\mathbb{C}_{\text{iso}} = \mathbb{P} : \overline{\mathbb{C}} : \mathbb{P}^{\mathrm{T}} + \frac{2}{3} \text{Tr}(J^{-2/3}\overline{\mathbf{S}}) \widetilde{\mathbb{P}} - \frac{2}{3} (\mathbf{C}^{-1} \otimes \mathbf{S}_{\text{iso}} + \mathbf{S}_{\text{iso}} \otimes \mathbf{C}^{-1}).$$
(3.6)

In (3.5), we have used the notation  $\mathbf{C}^{-1} \odot \mathbf{C}^{-1} = -\partial \mathbf{C}^{-1} / \partial \mathbf{C}$ , where the symbol  $\odot$  denotes the tensor product according to the rule

$$(\mathbf{C}^{-1} \odot \mathbf{C}^{-1})_{ABCD} = \frac{1}{2} (C_{AC}^{-1} C_{BD}^{-1} + C_{AD}^{-1} C_{BC}^{-1}), \qquad (3.7)$$

and the scalar function  $\tilde{p}$  is defined by  $\tilde{p} = p + Jdp/dJ$  with the constitutive equation for p given in the line after equation (3.2). In (3.6), we have also used the definitions

$$\begin{split} \bar{\mathbb{C}} &= 4J^{-4/3} \frac{\partial^2 \Psi_{\rm iso}(\bar{\mathbf{C}})}{\partial \bar{\mathbf{C}} \partial \bar{\mathbf{C}}}, \quad \mathrm{Tr}(\bullet) = (\bullet):\mathbf{C}, \\ \tilde{\mathbb{P}} &= \mathbf{C}^{-1} \odot \mathbf{C}^{-1} - \frac{1}{3} \mathbf{C}^{-1} \otimes \mathbf{C}^{-1}, \end{split}$$

$$(3.8)$$

where  $\overline{\mathbb{C}}$  is the fourth-order fictitious elasticity tensor,  $\operatorname{Tr}(\bullet)$  is the trace and  $\widetilde{\mathbb{P}}$  is the modified projection tensor of fourth order. The related spatial stress and elasticity tensors may be derived by push-forward operations on (3.2) and (3.4)<sub>1</sub>, respectively [44]. It should be noted here that the compressible formulation is introduced for computational purposes, and the incompressibility condition has to be enforced by a numerical scheme, one example of which is the augmented Lagrangian method (e.g. [45]).

We emphasize that for a specific material we need to specify  $\Psi_{iso}$  and hence to calculate its derivatives with respect to  $\bar{C}$ , which affects the two expressions (3.3) and (3.8)<sub>1</sub>. In §3.1, such a specification is provided.

#### 3.1. Anisotropic strain-energy function

Each of the structure tensors  $H_4$  and  $H_6$  depends on two dispersion parameters. We assume that these are the same for each fibre family. Our approach follows the work of Holzapfel *et al.* [14] in which the contributions  $\Psi_g$  and  $\Psi_f$ of the ground matrix (non-collagenous material) and the fibres to the strain energy are added. The artery is treated as an incompressible, elastic and fibre-reinforced material with the fibre dispersion accounted for both in plane and out of plane. Hence, superposition of energies reads

$$\Psi_{\rm iso} = \Psi_{\rm g}(\bar{\mathbf{C}}) + \sum_{i=4,6} \Psi_{fi}(\bar{\mathbf{C}}, \mathbf{H}_i). \tag{3.9}$$

Following [14,46], we model the ground substance with a neo-Hookean material  $\Psi_g = c(\bar{I}_1 - 3)/2$ , where the parameter *c* is the shear modulus in the reference configuration. For the fibre contributions  $\Psi_{fi}$ , we adopt the exponential functions [15]

$$\Psi_{\rm fi}(\bar{\mathbf{C}},\mathbf{H}_i) = \frac{k_1}{2k_2} [\exp(k_2\bar{E}_i^2) - 1], \quad i = 4, 6, \tag{3.10}$$

where  $k_1$  is a parameter with the dimensions of stress and  $k_2$  a dimensionless parameter, while

$$\bar{E}_i = \mathbf{H}_i : (\mathbf{C} - \mathbf{I}) \tag{3.11}$$

is a Green–Lagrange strain-like quantity which can be interpreted as an averaged or weighted fibre strain, depending on the fibre dispersion through the structure tensor  $\mathbf{H}_i$  and the (isochoric) macroscopic deformation through  $\bar{\mathbf{C}}$ .

Since tr $\mathbf{H}_i = 1$ , we can write  $\bar{E}_i = \mathbf{H}_i : \bar{\mathbf{C}} - 1$ . Using the definitions (3.1) of the structure tensors  $\mathbf{H}_4$  and  $\mathbf{H}_6$ , we obtain

$$\bar{E}_i = A\bar{I}_1 + B\bar{I}_i + (1 - 3A - B)\bar{I}_n - 1, \qquad (3.12)$$

where

$$\bar{I}_i = \bar{\mathbf{C}} : \mathbf{M}_i \otimes \mathbf{M}_i, \quad i = 4, 6, \qquad \bar{I}_n = \bar{\mathbf{C}} : \mathbf{M}_n \otimes \mathbf{M}_n.$$
 (3.13)

The invariants  $I_i = J^{2/3} \bar{I}_i$  and  $I_n = J^{2/3} \bar{I}_n$  are the squares of the stretches in the directions  $\mathbf{M}_i$  and  $\mathbf{M}_{n\nu}$  respectively. A summary of the parameters used is provided in table 3.

The strain-energy function used in the proposed model reads

$$\Psi_{\rm iso} = \frac{c}{2}(\bar{I}_1 - 3) + \frac{k_1}{2k_2} \sum_{i=4,6} \left\{ \exp\left[k_2 \bar{E}_i^2\right] - 1 \right\}.$$
 (3.14)

Following [14], we make the common assumption that the fibres do not resist any compression and are only active in tension. The invariants  $I_i$  are used as switches between fibre compression and tension so that the anisotropic part  $\Psi_{fi}$  only contributes to the strain energy when  $I_4 > 1$  or  $I_6 > 1$ . If one or more of these conditions is not satisfied then the relevant part of the anisotropic function is omitted from (3.14). For example, if  $I_4$  and  $I_6$  are less than (or equal to) 1, then the tissue response is purely isotropic. For discussion of subtle points regarding the choice of switching criteria, see [17].

In the expressions for the stress and the elasticity tensor, we need to calculate  $\partial \Psi_{iso} / \partial \bar{C}$  and  $\partial^2 \Psi_{iso} / \partial \bar{C} \partial \bar{C}$ . By using the chain rule, these are obtained as

$$\frac{\partial \Psi_{iso}}{\partial \bar{\mathbf{C}}} = c\mathbf{I} + \psi'_{i}\mathbf{H}_{i}, \quad \frac{\partial^{2} \Psi_{iso}}{\partial \bar{\mathbf{C}}\partial \bar{\mathbf{C}}} = \psi''_{i}\mathbf{H}_{i} \otimes \mathbf{H}_{i}, \quad i = 4, 6, \quad (3.15)$$

where  $\mathbf{H}_i$  is given by (3.1), and we have introduced the notations

$$\psi_{i}^{\prime} = \frac{\partial \Psi_{iso}}{\partial \bar{E}_{i}} = k_{1} \bar{E}_{i} \exp\left(k_{2} \bar{E}_{i}^{2}\right) \text{ and}$$

$$\psi_{i}^{\prime\prime} = \frac{\partial^{2} \Psi_{iso}}{\partial \bar{E}_{i}^{2}} = k_{1} (1 + 2k_{2} \bar{E}_{i}^{2}) \exp\left(k_{2} \bar{E}_{i}^{2}\right).$$

$$\left. \right\}$$

$$(3.16)$$

#### 4. Representative examples

Having identified the *dispersion* parameters of the model in §3, we now identify the *mechanical* parameters by fitting the constitutive model to uniaxial data. The proposed constitutive model is then implemented in the finite-element analysis program FEAP, and the parameters obtained are used to obtain results for a simple biaxial extension of an eight-element cube and compared with the analytical (Matlab) result. Finally, numerical results are obtained for the stress distribution in the non-homogeneous extension of strips of an adventitial layer cut out along the axial and circumferential directions.

#### 4.1. Parameter fitting to experimental data

In this example, we consider the purely incompressible formulation where the strain-energy function is characterized by  $\Psi = \Psi(C, H_4, H_6)$ . The second Piola–Kirchhoff stress tensor **S** is then given by

$$\mathbf{S} = 2\frac{\partial \Psi}{\partial \mathbf{C}} - p\mathbf{C}^{-1}$$
  
=  $2\left(\frac{\partial \Psi}{\partial I_1}\mathbf{I} + \sum_{i=4,6}\frac{\partial \Psi}{\partial I_i}\mathbf{M}_i \otimes \mathbf{M}_i + \frac{\partial \Psi}{\partial I_n}\mathbf{M}_n \otimes \mathbf{M}_n\right) - p\mathbf{C}^{-1},$   
(4.1)

where *p* in this case denotes the Lagrange multiplier required to enforce incompressibility. From (4.1), the Cauchy stress tensor can be computed simply by  $\boldsymbol{\sigma} = \mathbf{FSF}^{\mathrm{T}}$ .

The strain-energy function in the present formulation reads

$$\Psi = \frac{c}{2}(I_1 - 3) + \frac{k_1}{2k_2} \sum_{i=4,6} \{ \exp[k_2 E_i^2] - 1 \}, \qquad (4.2)$$



**Figure 7.** A specimen with two symmetric fibre families with mean fibre directions  $\mathbf{M}_4$  and  $\mathbf{M}_6$  lying in the circumferential/axial plane, each making an angle  $\alpha$  with the circumferential direction. The normal direction to the plane is  $\mathbf{M}_n$ .

where

$$E_i = AI_1 + BI_i + (1 - 3A - B)I_n - 1, \quad i = 4, 6.$$
(4.3)

The anisotropic term in (4.2) only contributes when the fibres are extended, i.e. when  $I_4 > 1$  or  $I_6 > 1$ . For the case that one or more of these conditions is not satisfied then the relevant part is omitted from (4.2).

We now consider a tissue specimen with two fibre families in the reference configuration with mean directions illustrated in figure 7. The unit vectors  $\mathbf{M}_4$  and  $\mathbf{M}_6$ , which are symmetrically disposed in the circumferential/axial plane, each make an angle  $\alpha$  with the circumferential direction, so that

$$\begin{bmatrix} \mathbf{M}_4 \end{bmatrix} = \begin{bmatrix} \cos \alpha, & \sin \alpha, & 0 \end{bmatrix}^{\mathrm{T}} \text{ and} \\ \begin{bmatrix} \mathbf{M}_6 \end{bmatrix} = \begin{bmatrix} \cos \alpha, & -\sin \alpha, & 0 \end{bmatrix}^{\mathrm{T}}, \end{bmatrix}$$
(4.4)

while the normal direction to the plane is  $[\mathbf{M}_n] = [0, 0, 1]^T$ . The deformation gradient matrix and the right Cauchy–Green matrix are

$$[\mathbf{F}] = \operatorname{diag}[\lambda_1, \lambda_2, \lambda_3] \quad \text{and} \quad [\mathbf{C}] = \operatorname{diag}[\lambda_1^2, \lambda_2^2, \lambda_3^2], \quad (4.5)$$

where  $\lambda_1,\ \lambda_2,\ \lambda_3$  denote the principal stretches. Hence, the required invariants read

$$I_1 = \lambda_1^2 + \lambda_2^2 + \lambda_3^2,$$
 (4.6)

$$I_i = \mathbf{C} : (\mathbf{M}_i \otimes \mathbf{M}_i) = \lambda_1^2 \cos^2 \alpha + \lambda_2^2 \sin^2 \alpha, \quad i = 4, 6$$
 (4.7)

and 
$$I_n = \mathbf{C} : (\mathbf{M}_n \otimes \mathbf{M}_n) = \lambda_3^2.$$
 (4.8)

Since  $I_i$  is the same for i = 4 and 6, we obtain  $\psi'_4 = \psi'_6$ , where here, in analogy with  $(3.16)_1$ ,  $\psi'_i = k_1 E_i \exp(k_2 E_i^2)$ , i = 4, 6. By using  $E_i$  in the strain-energy function (4.2), we obtain the derivatives of the strain-energy function as

$$\frac{\partial \Psi}{\partial I_1} = \frac{c}{2} + 2A\psi'_4, \quad \frac{\partial \Psi}{\partial I_i} = B\psi'_i, \quad i = 4, 6, \\
\frac{\partial \Psi}{\partial I_n} = 2(1 - 3A - B)\psi'_4.$$
(4.9)

Hence, the non-zero components of the Cauchy stress are

$$\sigma_{11} = [c + 4(A + B\cos^2\alpha)\psi_4]\lambda_1^2 - p, \qquad (4.10)$$

$$\sigma_{22} = [c + 4(A + B\sin^2 \alpha)\psi_4]\lambda_2^2 - p \tag{4.11}$$

and 
$$\sigma_{33} = [c + 4(1 - 2A - B)\psi_4]\lambda_3^2 - p.$$
 (4.12)

The constants A and B can be deduced from (2.17) and (2.18) as

$$A = 2\kappa_{\rm op}\kappa_{\rm ip}$$
 and  $B = 2\kappa_{\rm op}(1 - 2\kappa_{\rm ip}).$  (4.13)

Together with the incompressibility condition ( $\lambda_1 \lambda_2 \lambda_3 = 1$ ), the implicit equations (4.10)–(4.12) with  $\sigma_{22} = \sigma_{33} = 0$  can then be used to obtain p,  $\lambda_2$  and  $\lambda_3$  in terms of  $\lambda_1$ , thus giving an expression for  $\sigma_{11}$  in terms of  $\lambda_1$ , the material parameters c,  $k_1$  and  $k_2$  and the structural parameters  $\kappa_{ip}$ ,  $\kappa_{op}$  and  $\alpha$ .

#### Table 4. Summary of material and structural parameters.

	material para	material parameters		structural parameters		
	c (kPa)	<i>k</i> 1 (kPa)	<b>k</b> <sub>2</sub> (-)	<b>к<sub>ip</sub> (</b> -)	<b>к</b> ор (-)	α (°)
value	10.07	5.89	21.62	0.116	0.493	$\pm$ 47.99 $^{\circ}$
R <sup>2</sup>	·	- 0.998		0.877	0.916	(-)

To determine the material parameters of our model, we use the structural data provided in figure 5 (mean fibre angle  $\alpha$  and concentration parameters *a*, *b*) and experimental data (from uniaxial tension tests) from the adventitia of a human nonatherosclerotic abdominal aorta. A least-squares objective function is chosen as the  $L^2$ -norm of the error between the model prediction of the Cauchy stress in the circumferential and the axial directions, and the corresponding experimental data. We use the function LSQNONLIN in Matlab (v. R2010b; The MathWorks Inc., Natick, MA, USA), which uses the Levenberg–Marquardt algorithm to find the minimum. The dispersion parameters  $\kappa_{ip}$  from (2.23)<sub>2</sub> with the value of *a*, and  $\kappa_{op}$ from (2.28) with the value of *b*, can be determined from the imaging data provided in figure 5. Imaging and material data were obtained from different but comparable tissues.

To quantify the goodness of the fit, we calculate the coefficient of determination  $R^2$ , which is given by

$$R^2 = 1 - \frac{S_{\rm err}}{S_{\rm tot}},\tag{4.14}$$

where  $S_{\text{err}}$  and  $S_{\text{tot}}$  are the sums of squares of the differences between model/experiment and the mean of experiment/ experiment, respectively [47]. We obtained a good fit with  $R^2 = 0.998$ . The parameters identified are listed in table 4. Figure 8 shows the result of fitting the proposed model to data from uniaxial tension tests.

# 4.2. Biaxial extension: comparison of Matlab and finite-element results

In this section, we demonstrate the efficacy of the proposed constitutive model by using eight hexahedral Q1/P0 finite elements for a cube  $(10 \times 10 \times 10)$ , which is reinforced by two symmetric fibre families, as depicted in figure 7. The cube is subjected to homogeneous biaxial extension in the 1,2-plane, which here we call 'in plane', as shown in figure 9; the 2,3-plane we call 'out of plane'. The mesh is unstructured according to figure 9. The material parameters used are those documented in table 4. We investigate four different cases of fibre dispersion: (i) structural parameters taken from table 4, which is the reference case; (ii) high alignment out of plane and isotropy in plane; (iv) isotropy in both out of plane and in plane. For a summary see table 5.

The maximum of each stretch,  $(\lambda_1, \lambda_2)$  in the 1 and 2 directions, respectively, is 1.15 and the corresponding Cauchy stresses ( $\sigma_{11}, \sigma_{22}$ ) are computed from equations (4.10)–(4.12) using Matlab. In addition, these relationships are compared with results obtained from a finite-element computation using FEAP [48]. The cube is stretched simultaneously in the 1 and 2 directions under displacement-driven conditions using the Newton–Raphson method. Figure 10*a*,*b* shows



**Figure 8.** Fitting of the proposed model to the results of a uniaxial tension test. The good quality of fit is reflected in the high coefficient of determination,  $R^2 = 0.998$ . The model (Cauchy) stresses at a stretch of 1.3 are 16.6 (axial) and 12.2 kPa (circumferential)—see also figure 11. (Online version in colour.)

plots of the Cauchy stresses  $\sigma_{11}$  and  $\sigma_{22}$  versus the stretches  $\lambda_1$  and  $\lambda_2$ , respectively. Note that, for case (iv), the stresses in the 1 and 2 directions are the same since the in-plane and out-of-plane dispersions are isotropic.

#### 4.3. Extension of adventitial strips

In this section, we illustrate the results of the finite-element implementation of the proposed constitutive model, simulating uniaxial extension tests related to experiments on strips performed in our laboratory in Graz, Austria. The strips were taken from an adventitial layer cut out along the axial and circumferential direction of a human non-atherosclerotic abdominal aorta. The strips, each with initial length, width and thickness of 10.0, 3.0 and 0.5 mm, respectively, were subjected to a stretch of 1.3. Both ends of each strip were constrained so as to model the mounting in the testing machine and were not allowed to deform. The resulting deformation of each strip was therefore non-homogeneous. We assume uniform material parameters over the adventitial strips with values provided in table 4. Two symmetric fibre families, as shown in figure 7, are assumed to make an angle  $\alpha$  of  $\pm 47.99^{\circ}$  with the circumferential direction and to show a distribution characterized by  $\kappa_{ip} = 0.116$  and  $\kappa_{op} = 0.493$ , as also provided in table 4. We use 3200 hexahedral elements, applying the mixed Q1/P0 element throughout the simulation. Figure 11 shows the element results of the circumferential and axial specimens, subjected to a stretch of 1.3. Both circumferential and axial specimens show that the Cauchy stresses in the adventitia can be modelled within the experimentally



Figure 9. Unstructured mesh of a cube discretized by eight hexahedral finite elements. The cube is reinforced by two symmetric fibre families and subjected to homogeneous equibiaxial extension in the 1,2-plane, illustrated by arrows.

Table 5. Four different cases of fibre dispersion for the homogeneous biaxial extension of a cube. The material parameters are taken from table 4.

case	description	dispersion parameters
i	structural parameters from table 4	$\kappa_{ m ip} = 0.116, \; \kappa_{ m op} = 0.493$
ii	high alignment out of plane and isotropy in plane	$\kappa_{\mathrm{ip}} = 0.5, \; \kappa_{\mathrm{op}} = 0.48$
iii	less alignment out of plane and isotropy in plane	$\kappa_{\mathrm{ip}}=$ 0.5, $\kappa_{\mathrm{op}}=$ 0.45
iv	isotropy in out of plane and in plane	$\kappa_{\mathrm{ip}}=$ 0.5, $\kappa_{\mathrm{op}}=$ 1/3

predicted range of stresses (compare with the experimental results in figure 8). It turns out that the (axial and circumferential) stresses in the middle of the specimens are slightly higher than the model stresses provided in figure 8, which is due to the boundary conditions used in the present example.

### 5. Discussion

As several previous approaches have shown (e.g. [15,25,37]) incorporation of fibre dispersion into a continuum mechanical framework for soft biological tissues is a challenging and important task. Indeed the mechanical response of such tissues depends significantly on the tissue structure, in particular the arrangement of the fibre dispersion.

One of the main goals of our previous papers, e.g. [14,15], has been to incorporate the structure of biological tissues into our models in order to capture the physiological and pathological mechanical mechanisms. Recent experimental data have shown the need for a more general model that takes account of the non-symmetric arrangement of collagen fibres. This motivates the introduction, in this paper, of the bivariate von Mises distribution for describing the collagen fibre dispersion. By fitting the bivariate PDF to angular distribution data gained from imaging/histological analysis, we can determine structural parameters that can be integrated within the framework of continuum mechanics, in particular hyperelasticity. This allows us to define a strain-energy function from which the stress and the elasticity tensors can be computed, thus facilitating an efficient implementation of the model into a finite-element code.

The proposed constitutive model introduces a new structure tensor which incorporates in-plane and out-of-plane fibre dispersions in a clear and simple way, leading to an invariant-based formulation of the strain-energy function, generalizing our previous models [14,15]. The strain-energy function can incorporate different structure tensors for different fibre families. In particular, the work of Schriefl *et al.* [38] showed that the number of fibre families depends on the location of the artery and the type of layer, although in most cases two collagen fibre families were reported. For the most general case considered here the structure tensor has three independent components, and these reduce to two when there is in-plane symmetry (with parameters  $\kappa_{ip}$  and  $\kappa_{op}$ ) or



**Figure 10.** Homogeneous biaxial extension of a cube (cf. figure 9): (*a*,*b*) plots of Cauchy stresses  $\sigma_{11}$ ,  $\sigma_{22}$  versus stretches  $\lambda_1$ ,  $\lambda_2$  in the 1 and 2 directions. The four different cases of fibre dispersion are listed in table 5, where (i) is the reference case with parameters given in table 4. The circles represent finite-element results, while the curves are obtained from a Matlab computation.



**Figure 11.** Finite-element results of circumferential and axial specimens from the adventitia, subjected to a stretch of 1.3. The Cauchy stress is plotted in the direction of the applied displacement. The undeformed (initial) configuration is indicated by solid lines.

one when there is rotational symmetry about a single preferred direction (with parameter  $\kappa$ ). Various special cases have been highlighted in §2.2, capturing the cases of rotational symmetry, in-plane dispersion, transverse isotropy and isotropy.

We have constructed a specific energy function that illustrates the efficacy of the model using three specific examples. In the first example, we have demonstrated a good fit of the model to experimental data using identified structural parameters as a fixed set. The model was then used to illustrate the effect of fibre dispersion on biaxial extension experiments, including a comparison of analytical and finite-element results. Finally, we have analysed the three-dimensional non-homogeneous stress response for a given overall extension of adventitial strips in the circumferential and axial directions obtained from a human abdominal aorta. The resulting stress responses deviate significantly from each other due to the different mechanical and structural properties in the two directions.

More data are required to determine the detailed dispersion of collagen fibres, not only for the arterial wall but also for other types of fibrous tissues. To inform the modelling process, second-harmonic generation, for example, in combination with optical clearing [49] is a powerful technique for obtaining collagen fibre dispersion data from various types of tissues.

In order to analyse the data within a continuum framework, two main approaches are used. First the AI approach, which is computationally expensive because it involves an integration over the unit sphere. It does, however, allow the exclusion of fibres undergoing compression. With the GST approach, it is possible to extract structural parameters and to include them in the continuum mechanical framework to account for the fibre dispersion. For an artery that exhibits a dispersion with a strong alignment in plane, the GST seems to be a feasible and very efficient method, although it does require a minor modification in order to exclude fibres which are under compression [18].

For further development, as mentioned above, many more data are needed to inform the modelling process. In particular, there is a pressing need to obtain *in vivo* data in order to construct more realistic models of tissue and organ mechanics. The model can also be extended to incorporate inelastic effects such as damage, viscoelasticity and muscle activation.

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