Multifunctional Polymer Nanofibers: UV Emission, Optical Gain, Anisotropic Wetting, and High Hydrophobicity for Next Flexible Excitation Sources

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Ⓢ *Supporting Information*

ABSTRACT: The use of UV light sources is highly relevant in many fields of science, being directly related to all those detection and diagnosis procedures that are based on fluorescence spectroscopy. Depending on the specific application, UV light-emitting materials are desired to feature a number of opto-mechanical properties, including brightness, optical gain for being used in laser devices, flexibility to conform with different lab-on-chip architectures, and tailorable wettability to control and minimize their interaction with ambient humidity and fluids. In this work, we introduce multifunctional, UV-emitting electrospun fibers with both optical gain and greatly enhanced anisotropic hydrophobicity compared to films. Fibers are described by the onset of a composite wetting state, and their arrangement in uniaxial arrays further favors liquid directional control. The low gain threshold, optical losses, plastic nature, flexibility, and stability of these UV-emitting fibers make them interesting for building light-emitting devices and microlasers. Furthermore, the anisotropic hydrophobicity found is strongly synergic with optical properties, reducing interfacial interactions with liquids and enabling smart functional surfaces for droplet microfluidic and wearable applications.

KEYWORDS: electrospinning, optical gain, UV light sources, plastic optical amplifier, dye-doped fibers, polymer nanofibers

INTRODUCTION

A lot of modern research on materials is in the broad field of human healthcare and safety, involving significant issues such as protection from chemical and biological hazards, the detection of toxic combustions, and the development of novel, low-cost, point-of-care diagnostics. In these areas, the use of UV light sources is highly relevant, being related to all of the detection and diagnosis methods that are based on fluorescence spectroscopy.[1,2] Depending on the specific application, UV light-emitting materials are desired to feature a number of opto-mechanical properties. These include bright emission, stimulated emission, and optical gain for use in laser devices, flexibility to conform with different lab-on-chip architectures, and tailorable wettability to control and minimize the interaction of emitting materials with ambient humidity and fluids.[3]

Inorganic materials and nanostructures have received a lot of attention for realizing UV light-emitting devices (e.g., in plasmonic nanolasers).[4–8] These materials can be processed by colloidal methods, controlled growth, or top-down lithographies. However, all of them suffer from some drawbacks, such as poor flexibility, relatively high production cost, and frequently harsh processing conditions. Organic

nanomaterials, conversely, are inherently soft and can be processed at low cost, which might greatly help in realizing flexible light sources. For instance, various UV light-emitting devices and lasers have been produced by spin-cast films and organic slabs,[9–12] whereas soft nanomaterials with emission and optical gain in the UV range are still largely unexplored.[13]

Nowadays, great progress in the structural flexibility and versatility of the surface properties of photonic materials is being offered by electrospun polymer nanowires.[14–20] These fibers show high optical anisotropy and internal molecular orientation,[21–24] enhanced quantum yield,[25] and eventually optical gain, which has led to demonstrated lasing in the visible[26,27] and near-infrared[28] spectral range by individual nanowires or mats[26,28] and to distributed feedback imprinted geometries.[27] However, electrospun nanofibers approaching or emitting in the UV spectral range have only been based on hybrid structures embedding ZnO or Ge nanocrystals whose fabrication involves additional steps of thermal annealing, laser ablation, or chemical reactions.[29,30] The development of purely

organic, flexible, UV-emitting fibers enabling multiple functions, hopefully including optical gain and tailored wetting, would open new perspectives for the integration of light-emitting materials in point-of-care diagnostics and sensors and for the implementation of novel device platforms, such as flexible microlasers and smart wearable fabrics. In this work, we introduce multifunctional, UV-emitting electrospun fibers with both optical gain and anisotropic hydrophobicity. Hydrophobic properties help in making optical properties stable and reducing interfacial interactions of active materials with water, which make these fibers good candidates for integration in lab-on-chip and in droplet microfluidic platforms for chromophore excitation in liquids. In particular, line narrowing is found for the first time in electrospun fibers emitting in the UV range with onset for excitation fluence of a few tens of $\mu J/cm^2$ and optical gain above 5 cm^{-1}. These features, together with the flexibility and conformability of the fiber mats, suggest applications in various light-emitting devices and microlaser architectures with controllable and complex shapes.

■ RESULTS AND DISCUSSION

Fibers embed the molecular dye 4,4‴-bis[[(2-butyloctyl)oxy]-1,1′:4′,1″:4″,1‴-quaterphenyl (BBQ, molecular structure in Figure 1a), and their bundles are highly effective excitation sources for other visible-emitting chromophores and labeled antibodies.

The absorption, photoluminescence, and amplified spontaneous emission (ASE) spectra of BBQ are displayed in Figure S1 in the Supporting Information. Electrospinning allows for obtaining defect-free, BBQ-doped fibers with uniform internal distribution of the light-emitting component (Figure 1b−d). Fibers can be easily deposited in compact, free-standing
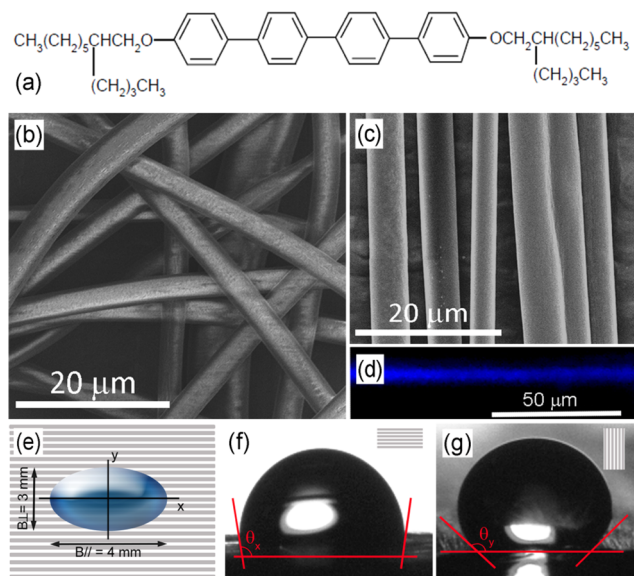

Figure 1. (a) BBQ molecular structure. (b) SEM micrograph of BBQ-doped electrospun fibers with random orientation. The fibers are uniform, defect-free, and highly flexible. (c) Unidirectional, aligned fibers. (d) Fluorescence micrograph of a single fiber, showing a uniform and bright emission, and highlighting the homogeneous presence of the embedded dye and the lack of appreciable clustering effects. (e−g) Schematics (e) and optical micrographs of water droplets and corresponding WCA values for the direction parallel (f, x-direction) and perpendicular (g, y-direction) to the fiber alignment axis.

bundles made by arrays of aligned wires (Figure 1c), thus favoring the assessment of optical gain properties, which have been found to be affected by the mutual alignment of electrospun fibers.[28] The topography of UV-emitting fiber surfaces also strongly affects their wettability. Indeed, although BBQ-doped poly(methylmethacrylate) (PMMA) films are hydrophilic, exhibiting a water contact angle (WCA) of $\theta = 61° \pm 2°$ (Figure S2a), randomly oriented fibers are remarkably hydrophobic with WCA $\theta_{F,Rand} = 132° \pm 2°$ (Figure S2b), which do not vary significantly upon changing the doping level in the range 0.1−2 wt/wt. Hence, the fiber topography leads to the onset of a composite wetting state described by the Cassie−Baxter law $\cos(\theta_{F,Rand}) = f_{CB}(\cos\theta + 1) - 1$, where $f_{CB} < 1$ is the fraction of the solid surface wet by the liquid.[31] This regime is related to the presence of air trapped within the surface of neighbor electrospun fibers. In addition, whereas on randomly oriented fibers wetting is isotropic and the deposited droplets take a spherical shape, aligning fibers in uniaxial arrays leads to an asymmetric surface topography and consequently to different WCA measured along a direction parallel (//) or perpendicular (⊥) to the fiber longitudinal axis, respectively. Pinning on the fibers tightly confines the fluid and induces an asymmetric shape of deposited water droplets (scheme in Figure 1e).[32−34] A stronger confinement is found perpendicular to the fiber length with WCA $\theta_{F,Align}^{//} = 99° \pm 2°$ (Figure 1f) and $\theta_{F,Align}^{\perp} = 140° \pm 2°$ (Figure 1g) approaching a super-hydrophobic regime.[35] This result is consistent with the Gibbs' criterion for liquid pinning, according to which the wetting contact line remains pinned on a microstructure when the advancing contact angle is comprised between θ and $\theta + \alpha$, where α (∼90°) is the maximum inclination of the surface of individual deposited fibers with respect to their top edge.[34] At equilibrium, the shape of the drops on the fiber surface can be predicted by minimizing the reduced free energy, $f = G/\gamma_{LV}$, where G is the overall free energy and γ_{LV} indicates the liquid−vapor surface tension.[36−38] f, in the direction parallel and perpendicular to the fiber longitudinal axis, can be estimated by[37] $f_{//} = [B_{//}\theta_{F,Align}^{//}/\sin(\theta_{F,Align}^{//})] - B_{//}\cos\theta$ and $f_{\perp} = [B_{\perp}\theta_{F,Align}^{\perp}/\sin(\theta_{F,Align}^{\perp})] - B_{\perp}\cos\theta$, where $B_{\perp}$ and $B_{//}$ are the lengths of the drop base in the two directions. For droplets of $B_{\perp} \cong 3$ mm and $B_{//} \cong 4$ mm (Figure 1e), we find a difference of ∼40% for the corresponding reduced free energies. Overall, electrospun fibers show greatly enhanced hydrophobicity compared to films, and their arrangement in uniaxial arrays further favors liquid directional control through pinning effects. Both of these effects might contribute to reduce interfacial interactions of the light-emitting material with water, thus improving both mechanical and optical stability. In fact, wetting properties are also related to the fiber surface chemistry and polarity, and variations of the surface polarity, like those associated with oxidation processes, can be unravelled by WCA changes.[39] The UV-emitting fibers, excited under conditions both below and above the ASE threshold (50−300 $\mu J/cm^2$, see below) do not show appreciable wettability variation, thus evidencing excellent surface chemical stability.

UV fiber bundles are then optically characterized with the aim of assessing ASE threshold, net gain, and optical losses. In Figure 2, we report the results of ASE measurements, showing emission spectra obtained at different excitation fluences and the resulting behavior of the spectral line width and emission intensity for fibers. At a low excitation level (<50 $\mu J/cm^2$), the signal stays low and the spectra are broad (see also Figure S3), mainly consisting of spontaneous emission.
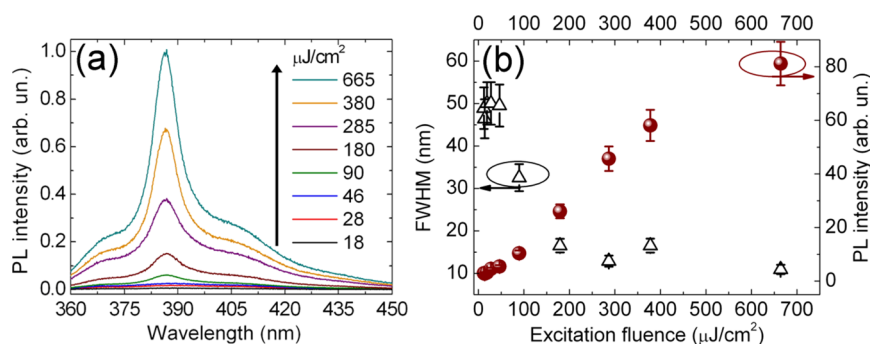
Figure 2. (a) ASE spectra from BBQ-doped fibers for various excitation fluences. (b) ASE intensity (full symbols) and fwhm (empty symbols) vs. excitation fluence.
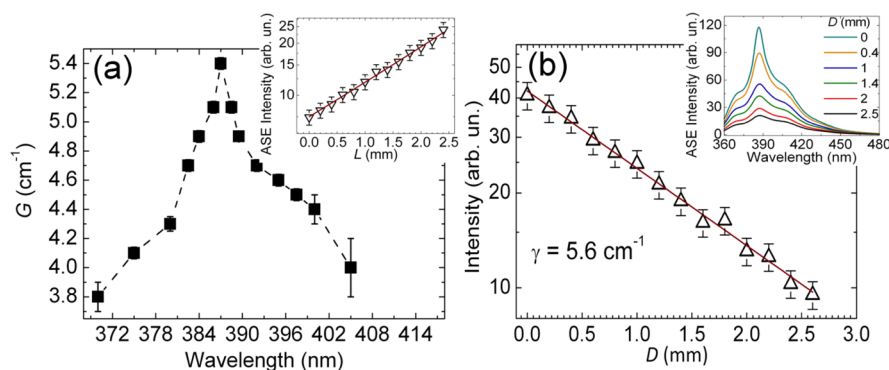
Figure 3. (a) Gain spectrum of BBQ-doped fibers. The plotted net gain values (symbols) are obtained by analyzing the ASE intensity as a function of the excitation stripe length at each wavelength by means of eq 1. The dashed line is a guide for the eyes. Inset: example of the dependence of the ASE intensity on the excitation stripe length (empty symbols) and corresponding best fit (continuous line). Data correspond to the maximum gain wavelength (387 nm). (b) Plot of the edge-emission intensity, recorded at different distances (D), of the excitation stripe from the fiber termination (empty symbols). The continuous line is the best fit by an exponential decay, providing the optical loss coefficient γ. Inset: spectra collected at different D values. Excitation fluence = 285 μJ/cm^2.

At ~50 μJ/cm^2, the spectra of fibers start to narrow, peaking around 387 nm. Spectral narrowing is an effect typically related to ASE, which is induced by photons traveling across the length of optical gain in an active medium.[40] In materials exhibiting stimulated emission under intense pump excitation, line-narrowing occurs in the spectral regions where the gain, which is related to the peak cross-section of the involved optical transition, is higher. The effect is frequently assisted by photon waveguiding along the gain material, which may constitute the core of an asymmetric waveguide in which the organics are sandwiched between two media with lower refractive indexes (the quartz substrate underneath and vacuum). The peak wavelength found here matches well with the spectral region of state-of-the-art inorganic microlasers, such as those based on ZnO nanowires.[6,7] Up to 300 μJ/cm^2, the ASE intensity grows in a superlinear way, whereas the full width at half-maximum (fwhm) undergoes an abrupt decrease down to 11 nm. Spectral narrowing mainly occurs around 100 μJ/cm^2. Overall, fibers show good ASE performances, which are particularly promising in view of building nanofiber-based, flexible UV microlasers. The net gain $G(\lambda)$ at 387 nm, shown in Figure 3a, is then obtained from the following expression

$$I_L = \frac{I_p A(\lambda)}{G(\lambda)} [e^{G(\lambda)L} - 1] \qquad (1)$$

where I_L is the emission output intensity, I_p represents the pump intensity, $A(\lambda)$ is related to the spontaneous emission cross-section, and L is the length of the excited region

(corresponding film data are shown in Figures S4 and S5). As expected, above the threshold, fibers show an exponential raise of the output intensity upon varying the stripe length (inset of Figure 3a), whereas the gain spectrum resembles the spectral profile of ASE. A maximum gain of 5.4 cm^{-1} is measured at 387 nm, which is comparable with values of bright, dye-doped polymer fibers emitting in the visible and near-infrared range.[28] The slightly higher value measured for films (Figure S5a) can be rationalized by considering that the estimated G is given by the difference of the optical gain coefficient in the active material and the optical loss coefficient γ. Optical losses are typically higher in electrospun fibers compared to film slabs due to more effective light scattering in the microstructures (Figure 3b and Figure S5b).

Another important feature of light-emitting nanofibers is the attenuation that occurs during the propagation of guided light. To assess optical losses, we measure the emission intensity as a function of the varying distance, D, of the excitation region from the sample emitting edge. The output intensity is then fitted by the expression, $I_{PL} = I_0 e^{-\gamma D}$, where I_0 is a constant and γ is the loss coefficient. Figure 3b shows the resulting output intensity from the fiber sample excited at 285 μJ/cm^2, providing $\gamma = 5.6$ cm^{-1}. This is a very favorable value compared to other light-emitting nanofiber systems[28] with the low optical losses here being related to the reduced contribution of self-absorption of emitted light because of the large Stokes shift of BBQ (766 meV, see Figure S1).
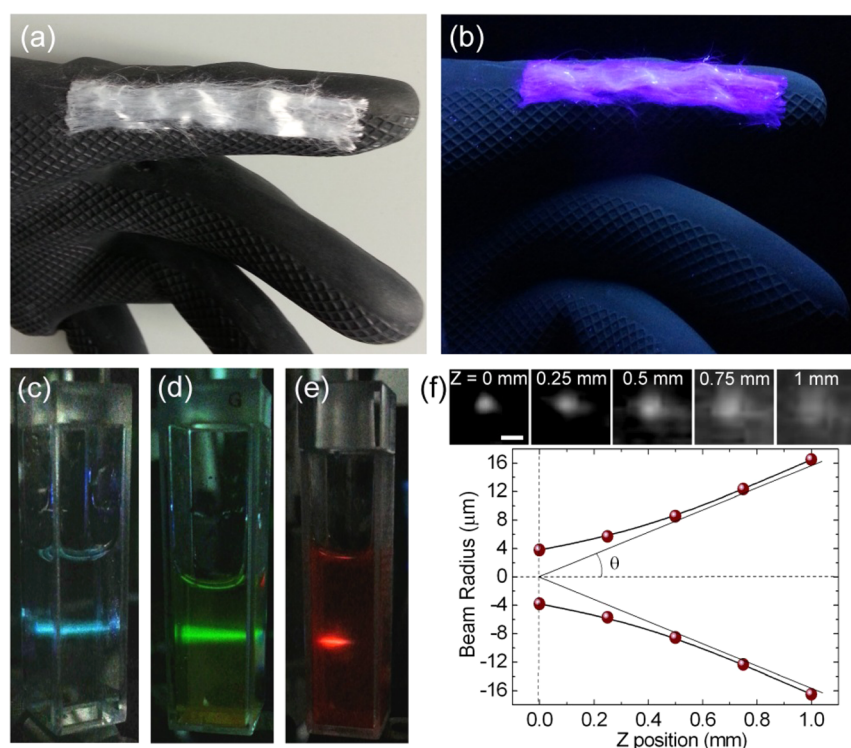
Figure 4. (a) Photograph of a bundle of aligned BBQ-doped fibers on curved surfaces. (b) The same bundle while emitting UV light. (c−e) Photoluminescence of 4′,6-diamidino-2-phenylindole (c, DAPI, blue-emitting), 2-(6-amino-3-imino-3H-xanthen-9-yl)benzoic acid methyl ester (d, Rhodamine 123, green-emitting), and [2-[2-[4-(dimethylamino)phenyl]ethenyl]-6-methyl-4H-pyran-4-ylidene]propanedinitrile (e, DCM, red-emitting) solutions, excited by the UV ASE from BBQ-doped fibers. Further experimental details for this configuration are shown in Figure S6. (f) ASE beam divergence measurement. Top: micrographs of the emission from a single fiber recorded at different positions from the emitting fiber tip image (Z position). Scale bar = 10 μm. Bottom: plot of the beam radius (dots) vs. Z. According to the definition of the divergence θ of a Gaussian beam, we consider the half angle corresponding to the asymptotic variation of the beam radius along the light propagation direction.

Furthermore, UV-emitting fiber bundles are mechanically robust, can be repeatedly bent, and reversibly conform to both planar and curved surfaces (Figure 4a,b). The reversible adhesion contact to other surfaces, mediated by van der Waals forces, allows for repeated use of the same bundles or mats with different substrates. Hence, these materials can be used as flexible excitation sources in a variety of spectroscopic measurements and lab-on-chip architectures, inducing fluorescence in chromophores and labeled antibodies with emission in the visible and near-infrared regions under either continuous or pulsed excitation conditions. For example, in Figure 4c−e, we show how the ASE signal from a bundle of aligned BBQ-based fibers excites three different dyes dissolved in solution. It is worth noting that the ASE beam profile is highly directional (Figure 4f) with an estimated divergence of 16.5 mrad, which is in line with the smaller divergence reported for polymers with optical gain.[41]

In particular, digital microfluidic applications can strongly benefit from the synergy of the emission and wetting properties of these fibers. For this aim, fibers aligned on a glass or quartz surface can be pumped by an evanescent field through total internal reflection (TIR)[42] and in turn be used to excite fluorescent droplets of solutions of chromophores or labeled antibodies (Figure 5). Droplets are placed and moved on fibers without touching the glass surface underneath, which is favored by the liquid confinement and hydrophobic character of the bundles, making this material ideal as an active substrate for fluorescence excitation in digital microfluidics.
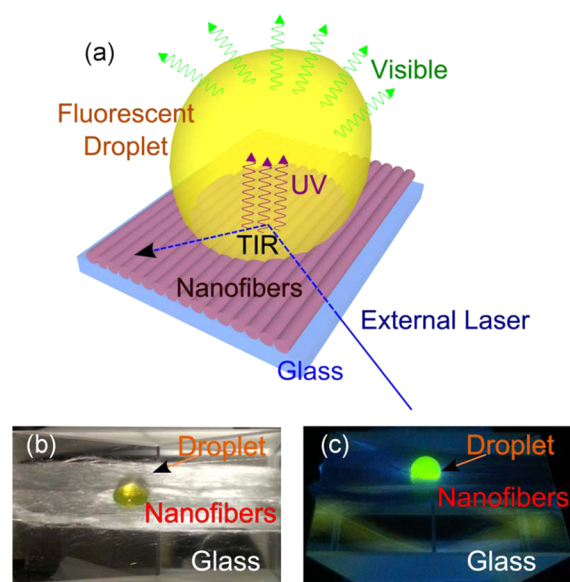
Figure 5. Aligned UV light-emitting nanofibers as a smart substrate for droplet microfluidics. (a) Schematics of a bundle of fibers with thickness of a few micrometers, deposited onto a prism, and evanescently excited by laser light (355 nm) in a TIR configuration. (b, c) Bright-field (b) and fluorescence (c) pictures of a droplet of Rhodamine 123 fluorescent solution in water. In (c), the fibers UV emission excites the droplet, and it is then removed by a long-pass filter.

CONCLUSIONS

In conclusion, we demonstrate multifunctional UV-emitting, BBQ-doped organic fibers featuring ASE, optical gain, as well as anisotropic wetting and hydrophobicity. The low ASE threshold and optical losses, the plastic nature, and the flexibility of these UV-emitting filaments make them interesting for building light-emitting devices and microlasers. Furthermore, the found anisotropic hydrophobicity is strongly synergic with optical properties, reducing interfacial interactions of the active material with liquids and enabling smart functional surfaces for microfluidic and wearable applications.

MATERIALS AND METHODS

Electrospinning. BBQ/PMMA solutions at 1% wt/wt are prepared by $CHCl_3$ (PMMA content = 3.75 wt % compared to solvent). The concentration is chosen in the range 0.1−2% following preliminary tests performed to optimize the emission. The solution is mixed by mechanical stirring for 12 h at room temperature prior to electrospinning, which is performed by a 1 mL syringe and a 21 gauge stainless needle. Randomly oriented fibers are spun by an applied voltage of 15 kV (EL60R0.6-22, Glassman High Voltage) and a needle-collector distance of 20 cm. The electrospinning injection flow rate is kept constant at 1 mL h^{-1} using a syringe pump (Harvard Apparatus). Aligned fibers are obtained using a collector having a diameter of 8 cm and rotating at 4000 rpm, placed 10 cm from the needle. Reference films are spin-cast from the same solution used for electrospinning. Sample inspection is performed by a Nova NanoSEM 450 system (FEI).

Contact Angle Measurements. The apparent WCA on the surface of films and electrospun fibers is measured by an optical video contact angle system (CAM-200, KSV Instruments), gently delivering a drop (∼20 μL) of ultrapure water from a capillary tip onto the fiber surface. The contact angles are determined by fitting the profile of at least three droplets. WCA measurements are performed both before and after UV pulsed light irradiation (355 nm, exposure time = 3 s).

Optical Characterization. Absorption measurements are carried out by a spectrophotometer. Spontaneous emission spectra are obtained by exciting samples below gain threshold. Fluorescence micrographs are collected by an inverted microscope (Nikon), exciting fibers with a mercury lamp through an objective lens (50×, NA = 0.75). The fiber emission is then collected by the same objective and measured by a charge-coupled device (CCD). Optical excitation of fluorescent dyes in water is performed by directing ASE from one edge of the fiber sample into filled cuvettes. Evanescent field excitation of solution droplets deposited onto fiber mats is obtained by using bundles of aligned fibers on top of a prism. The excitation laser (355 nm) is directed into the prism, being reflected at the internal surface at an angle larger than the critical one according to a TIR regime. The fiber emission is cut by a long-pass filter after exciting the fluorescent droplets and thus not being collected by the imaging system.

Optical Gain. Bundles of aligned fibers and spin-coated films are placed under vacuum at 10^{-4} mbar and excited by the third harmonic (355 nm) of a pulsed Nd:YAG laser (pulse duration = 10 ns, repetition rate = 10 Hz) focused on a stripe parallel to the fiber alignment direction. A system of lenses allows for collecting the emitted light from one edge of the fibers, coupling the photoluminescence signal into a spectrograph. The ASE threshold is then measured by systematically varying the excitation fluence, and the net gain is obtained by the variable stripe length technique. By keeping fixed the excitation fluence above threshold as well as the stripe position on the sample, we change the stripe length and measure the emitted intensity, finally analyzed by eq 1. In a different way, optical losses are measured through the emitted light intensity by varying the excitation stripe position (with fixed excitation fluence and length = 4 mm), namely, moving the stripe away from the sample emitting edge. For beam divergence measurements, a bundle of aligned fibers is excited above threshold, and ASE is coupled to a lens system (numerical aperture = 0.85, resolving power of ∼0.3 μm). A Si CCD is initially placed on the

focus plane at which the tip of the emitting fiber is imaged (Z = 0 mm in Figure 4f) and gradually moved away to record the divergent emission profile as a function of the distance from the focus position. The plot of the obtained spot diameter relative to the CCD position allows the beam divergence to be estimated, which is defined as the half angle corresponding to the asymptotic variation of the beam radius along the propagation direction.

ASSOCIATED CONTENT

Ⓢ Supporting Information

The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acsami.5b06483.

Further wettability and optical data along with experimental details (PDF)

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Notes

The authors declare no competing financial interest.

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