

A topologically driven glass in ring polymers

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The static and dynamic properties of ring polymers in concentrated solutions remains one of the last deep unsolved questions in polymer physics. At the same time, the nature of the glass transition in polymeric systems is also not well understood. In this work, we study a novel glass transition in systems made of circular polymers by exploiting the topological constraints that are conjectured to populate concentrated solutions of rings. We show that such rings strongly interpenetrate through one another, generating an extensive network of topological interactions that dramatically affects their dynamics. We show that a kinetically arrested state can be induced by randomly pinning a small fraction of the rings. This occurs well above the classical glass transition temperature at which microscopic mobility is lost. Our work both demonstrates the existence of long-lived inter-ring penetrations and realizes a novel, topologically induced, glass transition.

glass transition | ring polymers | topology | topological glass | molecular dynamics

The physics of ring polymers remains one of the last big mysteries in polymer physics (1). Concentrated systems of ring polymers have been observed, in both simulations and experiments, to display unique features that are not easily reconciled with the standard reptation theory of linear polymers (2–6). The main reason for this is that ring polymers do not possess free terminal segments, or ends, essential for end-directed curvilinear diffusion. In contrast, ring polymers possess a closed contour, which leads to markedly different relaxation and diffusion mechanisms. Recently, there has been much improvement in the production of purified systems of rings (6–8), with the consequent result that more and more experimental puzzling evidence requires a deeper understanding of their motion in concentrated solutions and melts from a theoretical point of view.

Recently, it has been conjectured that ring polymers assume crumpled, segregated conformations in concentrated solution or the melt (5). On the other hand, numerical and experimental findings (5, 6) suggest that rings exhibit strong intercoil correlations, which have proved difficult to address in simplified theoretical models (9–12). Because of this, there have been many recent attempts to rigorously characterize these interchains' interactions (13–16), although a precise definition and unambiguous identification of these "threadings" in concentrated solutions of rings remains elusive. The primary reason for this is that the rings are assumed to remain strictly topologically unlinked from one another throughout if synthesized in this state.

In the case of concentrated solutions of rings embedded in a gel, a method to identify these interpenetrating threadings has recently been proposed (13). Here it was shown that the number of threadings scales extensively in the polymer length (or mass) and can therefore be numerous for long rings, creating a hierarchical sequence of constraints that can span the entire system. It has also been conjectured that a kinetically frozen state, or a "topological glass" (17) can emerge, because such an extensive network of constraints can eventually suppress the translational degrees of freedom of the rings. However, the molten, or highly concentrated, state does differ from that of polymers embedded in a gel and so whether a similar jamming transition occurs for long enough polymers or even whether threadings are present in

the absence of a gel remain open problems and are the main questions addressed in this study. An example of interthreaded ring configuration is shown in Fig. 1. A spherical region (Fig. 1B) is carved from the configuration depicted in Fig. 1A, which represents a typical system studied in this work. The degree of interpenetration between the different coils is readily appreciable from the figure, and it can be boiled down even further into a network representation (13). The uncrossability constraint between chains transforms the threadings into topological hindrance in the motion of the coils, which we conjecture to form the basis for a dramatic slowing down in the dynamics of long enough coils.

Glass-forming systems exhibit degrees of freedom that become constrained as the temperature, or the density, of the system approaches the glass transition temperature T_g or the critical jamming density ρ_c , thereby (super)exponentially increasing the viscosity of the system (18). Understanding the origin of these constraints, being kinetic or thermodynamic in nature, is still an open topic that animates intense research (19, 20).

Recently, a novel and promising theoretical approach to study the glass transition in glass-forming liquids has been advanced: It involves perturbing a system by randomly pinning some fraction of the constituents and by observing the behavior of the unfrozen fraction. This method introduces a field of "quenched" disorder by freezing in space and time a subset of the system (21–27).

Inspired by this approach, we focus our attention on a concentrated solution of rings and apply a similar protocol: We freeze in space and time a fraction of polymers in the system and observe the response of the unfrozen constituents. We find that, although linear polymers are substantially insensitive to this perturbation, ring polymers become irreversibly trapped in a network of intercoil constraints (threadings), which, in the limit of long rings, allows us to drive a kinetically arrested state with only a small fraction of permanently frozen chains. We conjecture that a spontaneous glassy state might therefore emerge in

Significance

The glass transition is commonly associated with a reduction in the temperature of liquids or by an increase in density of granular materials. In this work, we propose a radically different approach to study dynamical arrest that relies on the topology of the components. We find that a concentrated solution of ring polymers can be driven to a kinetically arrested state by randomly pinning a small fraction of rings, a transition not observed in linear polymers. We attribute this jamming to topological interactions, called "threadings," that populate solutions of rings. Our work provides the first evidence for these threadings and suggests that very long rings may be expected to be kinetically arrested even as the fraction of pinned rings approaches zero.

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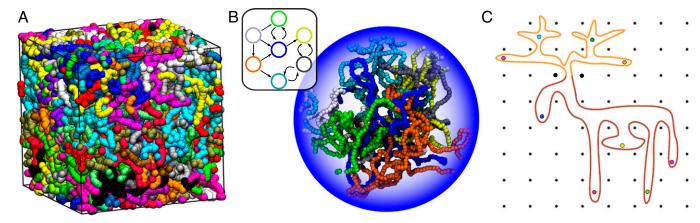


Fig. 1. Networks of threadings. (A) Snapshot from a simulation representing a system with N = 50 chains M = 512 beads long. (B) Snapshot of a spherical region and some of the rings from A. One can easily appreciate the interpenetration between the rings in the system (for instance, cyan through dark gray or orange through blue). The sketched network in the top corner is obtained by visual inspection and depicts the penetrations of the rings. Two circles are connected by a directed arrow when the former is threading through the latter (13) (see also Movie S1). (C) Lattice animal representation of a single ring polymer surrounded by nonthreading rings (black dots) and penetrated by threading rings (colored dots). Adapted from ref. 6 with permission from Macmillan Publishers Ltd: Nature Materials, copyright 2008.

the long chain limit, even as the fraction of explicitly pinned chains goes to zero. Because these constraints are topological in nature, originating from noncrossability of the chains, this glassy state has the potential to be produced at arbitrary temperature or monomer density, provided only that these intercoil topological interactions remain abundant in the system, i.e., the rings are sufficiently long and not too dilute. This system is therefore a candidate for a novel kind of glass transition in systems made of polymers or other elements with nontrivial topology.

Rings in Solution Assume Crumpled but Largely Overlapping **Conformations**

The static properties of rings in solution have been studied for some decades, and it is nowadays thought that rings assume crumpled conformations characterized by a scaling law for their gyration radius $R_g \approx M^{\nu}$ with $\nu \simeq 1/3$ in the limit of rings with large polymerization M (11, 28–31). On the other hand, they also have been found to maintain numerous contacts with their neighbors. This is often quantified by measuring the "contact surface" of the polymers, which is defined as the number of segments m_{surf} in any one chain that are in contact with any segment belonging to any other chain. This has been found to scale nearly extensively with the size of rings (SI Appendix, Figs. S1 and S2) $m_{\rm surf} \approx M^{\beta}$ with $\beta \simeq 0.98$. This has been interpreted as a clear signature of abundant interactions and coil overlap. This picture is also supported by the fact that a coil's pair correlation function g(r) is peaked at $r_c < 2r_{\text{max}} \simeq 2.6R_g$ (see SI Appendix, Fig. S3), in light of which the coils can be viewed as interpenetrating ultrasoft colloids of radius $r_{\text{max}} \simeq 1.3 R_g$ (32–34).

The counterintuitive fact that rings have numerous intercoil interactions while assuming a scaling exponent $\nu \simeq 1/3$ can be understood within the fractal globule conjecture (11, 35) in which the rings assume a fractal, hierarchical and virtually entanglement-free, conformation that can accommodate nearextensive interactions with other chains with a contact exponent γ near unity (29) (SI Appendix, Fig. S3).

These interchain interactions may include threadings that have been conjectured to be intimately related to the slow overall diffusion of the rings' center of mass (5, 8, 13, 16) (see also SI Appendix, Fig. S4), an observation that is in apparent contrast with the very fast stress relaxation (6, 7), characterized by a power-law decay of the relaxation modulus G(t) and by a remarkable absence of the entanglement plateau that characterizes concentrated solutions of linear polymers. In light of these findings, we believe that further investigation of the role of these inter-ring interactions is crucial.

Contiguity Is Persistent for Longer Chains

To probe intercoil interactions, we first adopt a definition of surface monomers where the *i*th monomer of chain *I* is a surface monomer of that chain if its distance from a monomer j belonging to a different chain *J* is $d_{ij} < \rho^{-1/3}$ where $\rho = 0.1\sigma^{-3}$ is the monomer concentration. We then define as "contiguous" two coils that share surface monomers. From this definition, we propose a method to track the exchange dynamics of contiguous chains, I and J, by computing a dynamic $N \times N$ matrix P(t) whose elements are defined as

$$P_{IJ}(t) = \begin{cases} 0 & \text{if } d_{ij} \ge \rho^{-1/3} \forall i, j \\ 1 & \text{otherwise} \end{cases}.$$
 [1]

From this, it is straightforward to obtain the correlation function

$$\varphi_{nc}(t) = \left\langle \frac{1}{N} \sum_{J=1}^{N} P_{IJ}(t) P_{IJ}(t - \Delta t) \dots P_{IJ}(0) \right\rangle$$
 [2]

where $\langle ... \rangle$ indicates the ensemble average over rings I and initial times. This function quantifies the exchange dynamics of contiguous chains and tracks the time that the chains first become noncontiguous, because it involves the product of P_{IJ} over all of the intermediate time steps up to time t. The behavior of $\varphi_{\rm nc}(t)$ is reported in Fig. 2. Eq. 2 gives a more strict measurement of the interchain cooperativity than would be obtained from a standard contiguity correlation function and, unlike the latter, should decay to zero over time scales that are comparable with the time taken for the chains' centers of mass to diffuse away from one another. Although, for short chains, this is well described by a simple exponential, we observe that its behavior for longer chains can be fitted for about two decades by stretched exponentials

$$\varphi_{\rm nc} = \exp\left[-\left(\frac{t}{\tau_{\rm nc}}\right)^{\beta_{\rm nc}}\right]$$
 [3]

with an exponent $\beta_{\rm nc}$ that varies from $\beta_{\rm nc}=1$ for M=256 to $\beta_{\rm nc}\simeq 1/2$ for the longest chains in the system (see *SI Appendix*,

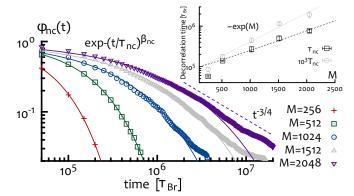


Fig. 2. The rate at which chains become noncontiguous slows with their length. Correlation function $\varphi_{\rm nc}(t)$ as a function of time (in simulation time units τ_{Br} ; see Materials and Methods) for different chain lengths. Solid lines represent stretched exponentials, and the dashed line on the right represents a power-law decay with exponent of -3/4 suggesting an even slower inter-ring decorrelation dynamics. (Inset) The exponential increase of the typical exchange time $\tau_{\rm nc}$ and $T_{\rm nc}$ for the systems displaying a stretched exponential relaxation (SI Appendix, Fig. S5).

Fig. S5). Even more striking is the exponential increase of the typical time to become noncontiguous $\tau_{\rm nc}$, indicating a very slow decorrelation between chains (or very long "exchange time"), often interpreted as the onset of glassy dynamics (34). As $\varphi_{\rm nc}$ shows fat tails at long times, we also compute $T_{\rm nc}$ as the (numerical) integral of the correlation function. We find that the functional behavior of $T_{\rm nc}$ is in agreement with $\tau_{\rm nc}$ and shows an even steeper increase (see Fig. 2, *Inset*). This is most likely due to the fact that only the long-time tails show deviations from the stretched exponential behavior, and these act to further increase the exchange time of the coils.

These findings are a strong signature that rings display long-lasting intercoil correlations that are present even after they have traveled beyond their own size, in agreement with previous numerical and experimental findings (5, 8). In addition, this is clear evidence that the exchange dynamics of the rings becomes slower, more glass-like (16, 36), as the polymerization M increases. The increasingly stretched decay of $\varphi_{\rm nc}$ also implies that the relaxation dynamics of the chains becomes more heterogeneous, i.e., some parts of the chains are much slower to separate from one another than other parts. One might conjecture that the increasingly slow and heterogeneous exchange dynamics will eventually appear in the long-time dynamics of the ring displacement because contiguity will ultimately constrain motion.

It is natural to now ask whether we can understand the nature of these correlations, and the heterogenity in the relaxation dynamics observed in Fig. 2, as being directly related to the threadings recently proved to exist in concentrated solution of rings embedded in gels (13, 14). In the case of pure solutions of rings, such threadings have never been rigorously identified.

Our approach consists in artificially immobilizing ("freezing"), in space and time, a fraction c of randomly selected rings from equilibrated configurations and then tracking the dynamics of the "unfrozen" fraction. This protocol is inspired by recent theoretical and experimental studies on the nature of glass transitions and idealizes the case in which a fraction c of polymers in the system might be arbitrarily frozen, perhaps by mixing two polymeric species with different T_g or by using optical tweezers (24, 25), although the primary interest in this work is in its role as a conceptual tool.

If the rings are mutually threaded, perhaps in a way that resembles the threaded lattice animal in Fig. 1C, then one would expect those unfrozen rings that are threaded by frozen ones to have their mobility substantially limited. They would appear to

be immobilized within effective "cages," being the region of space that they can explore limited by the threadings that they experience. In the alternative picture where rings remain unthreaded, as is often envisaged in simplified models, the mobility would be substantially unaffected by the presence of frozen chains.

A primary result of the present work, discussed in *Randomly Pinning Rings Induces a Kinetically Arrested State*, is that we do indeed observe immobilization of the unfrozen fraction. We believe that this represents excellent evidence for the existence of threadings in concentrated solutions of rings, something that has not previously been demonstrated.

Randomly Pinning Rings Induces a Kinetically Arrested State

Starting from an equilibrated configuration, we perturb the system by randomly freezing, in space and time, a fraction c of coils. As a comparison, we first consider a system of linear polymers with one unfrozen linear "probe" chain diffusing through cN = N - 1artificially immobilized (frozen) linear polymers. The same perturbation is then applied to a dense solution of ring polymers, and the two cases are compared in Fig. 3. This figure shows that the longtime dynamics of the unfrozen linear chain (green circles) is substantially insensitive to the presence of frozen neighbors. This is because the linear polymer can undergo reptation and simply snake through the frozen surroundings. [There is a weak correlation effect due to the lack of mass relaxation in the frozen chains, leaving a "hole" (and corresponding "bump") in the density as the mobile chain moves. We find that this only weakly affects the reptative dynamics.] On the other hand, when we repeated this procedure on a corresponding system of rings, we observed the probe ring's diffusion to be arrested, with it becoming irreversibly trapped within a region of space of size somewhat smaller than its gyration radius R_g (see red squares in Fig. 3).

Because nothing other than the topology of the polymers was changed, this dramatically different dynamical response should be attributed to the presence of topological interactions between ring polymers, which we identify as the threadings. This immediately implies that the equilibrated state of the rings in our systems is one in which threadings constrain the free diffusion of the rings (see sketch in Fig. 1C), therefore limiting their motion and, in the extreme case, leading to caged diffusion when neighboring rings that thread them are permanently frozen. The

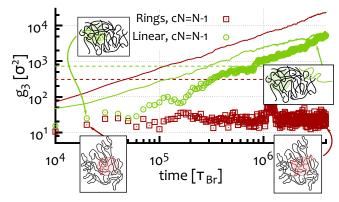


Fig. 3. A single unfrozen ring in a system of frozen chains becomes permanently caged whereas a linear polymer is substantially unaffected. Mean square displacement of the center of mass $g_3(t)$ of a probe ring (red squares) and of a probe linear polymer (green circles) diffusing through, N-1 frozen ring and linear polymers, respectively. The curves are averaged over 10 different probes (samples). The two systems have the same monomer density $\rho=0.1\sigma^{-3}$ and comprise N=50 polymers each containing M=256 beads that differ only in their topology. Solid lines represent the behavior of the free solution, i.e., the c=0 case and dashed lines represent the diameter (squared) for rings and linear polymers (see Movie S2).

constraints provided by threading between rings in the unperturbed (c=0) solution will be transient to a greater or lesser extent, but they must nonetheless exist. Our results therefore represent, to our knowledge, the first unambiguous evidence for inter-ring threading in dense solutions of rings.

It is also worth stressing that the measurement that we perform should be interpreted as carrying a statistical meaning: With enough attempts, one will always be able to find a ring that is not caged, however unlikely that might be. When we interrogate the rings regarding their state, caged or free, we are implicitly drawing from a binomial distribution, and, therefore, we can calculate the probability p of observing a noncaged ring in any one test for a given fraction of frozen rings. Having performed 10 tries freezing N-1 rings and having observed 10 caged rings, we can set a 95% confidence bound on the fact that $p \simeq 0.26$, although the precise values are not particularly important and any statistical confidence criterion will give qualitatively similar results in what follows.

Given that, for all systems studied here, we observe a regime in which all of the unfrozen rings are caged in regions smaller than their sizes $(2R_{\sigma})$ for the whole simulation run time, it is natural to ask how the behavior crosses over from the unperturbed system (c=0), in which all of the rings are free to diffuse and none are explicitly frozen, to the case in which enough rings are explicitly frozen to implicitly pin, or cage, the others (at some level of statistical confidence).

We study this transition by tracking the behavior of each chain's center of mass diffusion $g_3(t)$, averaged over the unfrozen rings. The observed behavior of $g_3(t)$ is reported in Fig. 4: For all systems, there exists a critical frozen fraction (c^{\dagger}) for which every single unfrozen polymer is permanently trapped by the network of threadings. In other words, at $c = c^{\dagger}$, the systems exhibit a transition from (at least partially) liquid, or diffusive, behavior to a glassy state in which the unfrozen chains, although free to rearrange their conformations to some extent, are all irreversibly caged.

More practically, one can define c^{\dagger} by introducing the effective diffusion coefficient (37)

$$D_{\text{eff}}(c) \equiv \lim_{t \to \infty} \frac{\langle g_3(t) \rangle}{6t}$$
 [4]

whose average is taken over the unfrozen rings and vanishes when all of the unfrozen rings are "caged" by the topological interactions. It is therefore natural to identify c^{\dagger} as the value of c at which $D_{\text{eff}} = 0$. In practice, we run the simulations for a time much longer than the relaxation time of the rings measured at c = 0 (see also *SI Appendix*, Fig. S4) and define as caged all of the systems that display a $D_{\rm eff}$ 50 times smaller than the diffusion coefficient of the unperturbed (c=0) case, D_0 (see Fig. 4F). From this figure, one can notice that the decay of the $D_{
m eff}$ becomes increasingly steeper as the rings become longer, suggesting that the systems with long chains are more susceptible to small contaminations of frozen chains, whereas, from Fig. 4F, Inset, one can also appreciate the nearly exponential decay of $D_{\rm eff}$ with c (27).

Although the diffusion coefficient of the center of mass of the rings informs us about the overall diffusion of the chains, it is also interesting to study the relaxation of the chains at different length scales in response to this external perturbation. This can be done by computing the dynamic scattering function

$$S_c(q,t) = \frac{1}{S_c(q)} \left\langle \frac{1}{f N} \sum_{I} \left\langle \frac{1}{M} \sum_{ij \in I} e^{iq \left[\mathbf{r}_i(t) - \mathbf{r}_j(0) \right]} \right\rangle \right\rangle, \quad [5]$$

where \sum' stands for the summation over the (1-c)N = fN nonfrozen chains and the average is performed over time and orientations of q. In Fig. 5 (and SI Appendix, Figs. S6-S8), we report the behavior of this quantity computed for two choices of wave vector q probing length scales $(l = 2\pi/q)$ comparable to the rings' diameter $2R_g$. We observe that $S_c(2/R_g,t)$ decays much more slowly than the scattering function measured at $q = 4/R_e$. To compare their behavior, we choose an arbitrarily long time $(\bar{t} = 10^7 \tau_{Br})$ at which we evaluate the scattering function and report their difference ΔS_c for a range of values of c and rings' length (see Fig. 5, and see SI Appendix for different choices of \overline{t}).

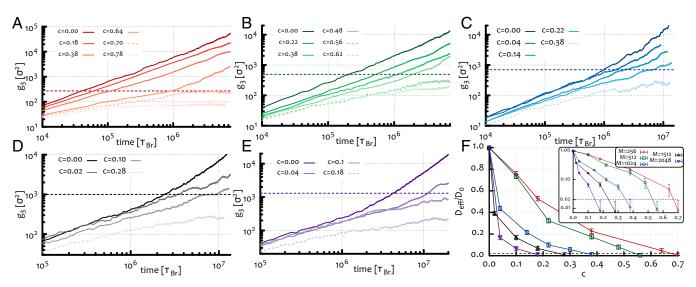


Fig. 4. The critical fraction of frozen chains c^{\dagger} at which all of the nonexplicitly frozen chains are usually caged by threadings decreases with ring length. The mean squared displacement of the center of mass $[g_3(t)]$ averaged over only the nonfrozen fraction (1-c) of rings plotted against time for different values of the frozen fraction c ranging from c = 0 (free solution) to $c \ge c^{\dagger}$ at which all rings in the unfrozen fraction have become permanently caged. (A) M = 256, (B) M=512, (C) M=1,024, (D) M=1,512, and (E) M=2,048. The thick horizontal dashed lines indicate the diameter of the chains. (F) The value of Deff (Eq. 4) normalized by $D_0 \equiv D_{\text{eff}}(c=0)$ is shown in linear-linear and (Inset) linear-log scale to appreciate the nearly exponential dependence. The critical value c^{\dagger} is defined as the one for which $D_{\rm eff}(c^{\dagger})/D_0$ is below a critical threshold that we here set to 0.02 (dashed line).

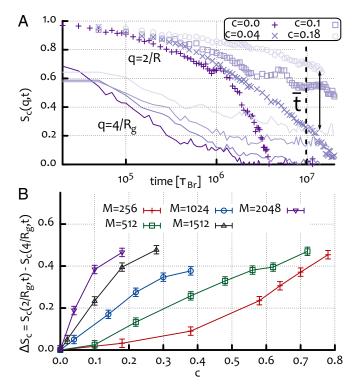


Fig. 5. The freezing procedure affects long length scales more strongly than short ones. (A) Behavior of the dynamic scattering function $S_c(q,t)$ for the system with M=2,048 and for $q=2/R_g$ ($I=\pi R_g$) (symbols) and $q=4/R_g$ ($I=\pi R_g/2$) (solid lines) (more examples can be found in SI Appendix, Figs. S7 and S8). The difference $\Delta S_c=S_c(2/R_g,\bar{t})-S_c(4/R_g,\bar{t})$ is computed at an arbitrary long time $\bar{t}=10^7$ τ_{Br} and reported in B for the different cases. From this plot, it is clear that the relaxation of length scales longer than $2R_g$ is slowed down more severely than the relaxation of shorter ones by the pinning procedure. In turn, this suggests that threadings act mainly by constraining low wavenumber modes.

The increasing trend of ΔS_c suggests that length scales longer than the diameter of the rings are more susceptible to the freezing procedure. This is consistent with the fact that threadings between ring polymers constrain their translational degrees of freedom on length scales comparable to the size of the rings whereas there can exist internal modes that are left unhindered and free to relax. In addition, we observe that the systems with longer chains require a smaller value of c to significantly slow down the relaxation of the scattering function with respect to systems with shorter chains. This observation is, in turn, in agreement with the fact that threadings are more numerous for systems with longer chains (13), and therefore a smaller fraction of frozen rings is sufficient to achieve a similar slowing down.

Fig. 5 also suggests that although coils are strongly constrained by the presence of frozen rings, the configurations can still partially relax by internal rearrangements, and this allows rings to relax their internal stress. The picture that emerges is rather different from that for linear polymers, where the center of mass diffusion is intimately related to its ability to undergo any and all conformational rearrangements. In the latter case, motion is only arrested by the onset of a microscopically glassy state for $T < T_g$. In the case of ring polymers, given their closed topology, threadings instead decouple the dynamics involved in displacing the center of mass and those involved in the internal rearrangement. Any glassy jamming that emerges in this work is unusual in that it is not associated with arrest of microscopic degrees of freedom but rather with the topological arrest of low wavenumber modes, including that associated with the motion of

the center of mass. This is the reason why an entanglement plateau is absent. It also means that the slowing down of the dynamics in systems of rings might not be clearly captured by the stress relaxation function G(t) frequently studied in the literature (6–8, 30), which is mostly dominated by the unconstrained internal modes, but rather by the mean square displacement of the unfrozen rings (as in Fig. 4) or the dynamic scattering function $S_c(q,t)$.

Phase Diagram of the System

The behavior of both the effective diffusion coefficient $D_{\rm eff}$ and ΔS_c show an increasingly steeper dependence on the freezing parameter c as the length of the rings increases (see Figs. 4F and 5B). This implies that systems made of long chains become extremely sensitive to very small perturbations. Indeed, one can think of these observables as quantifying a form of susceptibility that captures how the dynamic mobility of the system responds to the freezing of (very few) threading constraints by chain immobilization.

Finally, in Fig. 6, we show the transition line $c = c^{\dagger}$ in the space of parameters (1/M,c). The colored data points represent the position of the simulated systems in the phase space and whether their behavior was liquid-like (finite D_{eff} -red) or solid-like (vanishing D_{eff} -blue). The diagram in Fig. 6 is reminiscent of that observed in more traditional glass-forming systems subject to random pinning fields (22) where the temperature T is here replaced by the inverse length 1/M; a substitution not unfamiliar to field-theoretic treatments of polymer systems (2). Along the transition line, we also report the value of the probability p of finding an uncaged ring in any one sample as obtained from the binomial distribution at 95% confidence interval $(1-p)^{(1-c^{\dagger})N} = 0.05$. It is also worth noticing that the curve is well fitted by an exponential function of the form $M^{\dagger}(c)=M_ge^{-3.3c}$ with $M_g\simeq 3{,}500$. Computational limitations forbid a thorough exploration of the small c region. Nonetheless, and somewhat remarkably, our results show that the number of caged chains per explicitly frozen ring is exponential in the ring length (see SI Appendix, Fig. S9B). This would seem to mean that an arbitrarily large fraction of caged chains can be achieved from an arbitrarily small fraction of frozen rings provided that the rings are long enough. This raises the possibility that a glassy state could emerge spontaneously in the critical regime near c = 0 in the (universal) limit of large M.

From these results, it is clear that concentrated solutions of long rings are extremely sensitive to a small external pinning field

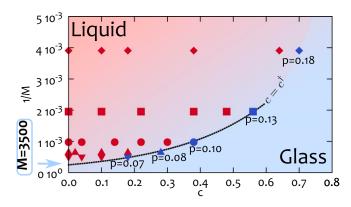


Fig. 6. The phase diagram of the system suggests a spontaneous vitrification at large M. The figure shows the phase space (1/M,c) for systems of rings with length M and in which a fraction c of rings are permanently frozen in space and time. The transition line $(1/M^{\dagger},c^{\dagger})$ is shown together with an exponential fit (dotted line). The colored data points in the diagram indicate whether the system displays a finite diffusion coefficient at large times (red) or whether it is irreversibly caged with vanishing $D_{\rm eff}$ (blue). Along the transition line, we also report the value of the probability p of finding an uncaged ring in any one test performed at fixed c (see *Phase Diagram of the System* for details).

that can drive the center of mass motion of the rings to become glassy (jammed). We have here traced this phenomenon to the unusual, topological constraints provided by the threadings between ring polymers. Although branched polymers also display exponentially long relaxation times, the origin of this is not cooperative in nature and would therefore be rather insensitive to the freezing of some components. On the other hand, cooperativity, such as the inter-ring threadings we study, is thought to be an essential ingredient of a genuine glass transition (18).

Our work therefore elucidates the conformation of rings in concentrated solutions and unambiguously characterizes, for the first time to our knowledge, the presence of threadings between rings. Furthermore, we show the dynamics of ring polymers to be sensitive to these interactions, and, finally, we provide strong evidence for the emergence of a kinetically arrested state solely driven by topological constraints. This system is therefore a novel instance of a glassy state induced through the topology of the constituents.

- 1. McLeish T (2008) Polymer dynamics: Floored by the rings. Nat Mater 7(12):933-935.
- 2. de Gennes PG (1979) Scaling Concepts in Polymer Physics (Cornell Univ Press, Ithaca, NY).
- 3. Doi M, Edwards S (1988) The Theory of Polymer Dynamics (Oxford Univ Press, Oxford).
- Halverson JD, Lee WB, Grest GS, Grosberg AY, Kremer K (2011) Molecular dynamics simulation study of nonconcatenated ring polymers in a melt. I. Statics. J Chem Phys 134(20):204904.
- 5. Halverson JD, Lee WB, Grest GS, Grosberg AY, Kremer K (2011) Molecular dynamics simulation study of nonconcatenated ring polymers in a melt. II. Dynamics. J Chem Phys 134(20):204905.
- 6. Kapnistos M. et al. (2008) Unexpected power-law stress relaxation of entangled ring polymers. Nat Mater 7(12):997-1002.
- 7. Pasquino R, et al. (2013) Viscosity of ring polymer melts. ACS Macro Lett 2(10):
- 8. Doi Y, et al. (2015) Melt rheology of ring polystyrenes with ultrahigh purity. Macromolecules 48(9):3140-3147. 9. Cates ME, Deutsch J (1986) Conjectures on the statistics of ring polymers. J Phys 47(12):
- 2121-2128.
- 10. Rubinstein M (1986) Dynamics of ring polymers in the presence of fixed obstacles. Phys Rev Lett 57(24):3023-3026.
- 11. Grosberg AY (2014) Annealed lattice animal model and Flory theory for the melt of non-concatenated rings: Towards the physics of crumpling. Soft Matter 10(4): 560-565.
- 12. Smrek J, Grosberg AY (2015) Understanding the dynamics of rings in the melt in terms of the annealed tree model. J Phys Condens Matter 27(6):064117.
- 13. Michieletto D, Marenduzzo D, Orlandini E, Alexander GP, Turner MS (2014) Threading dynamics of ring polymers in a gel. ACS Macro Lett 3(3):255-259.
- 14. Michieletto D, Marenduzzo D, Orlandini E, Alexander GP, Turner MS (2014) Dynamics of self-threading ring polymers in a gel. Soft Matter 10(32):5936-5944.
- Tsalikis D, Mavrantzas V (2014) Threading of ring poly (ethylene oxide) Molecules by linear chains in the melt. ACS Macro Lett 3(8):763-766.
- 16. Lee E, Kim S, Jung Y (2015) Slowing down of ring polymer diffusion caused by interring threading. Macromol Rapid Commun 36(11):1115-1121.
- 17. Lo W-C, Turner MS (2013) The topological glass in ring polymers. EPL 102(5):58005.
- 18. Gibbs JH. DiMarzio EA (1958) Nature of the glass transition and the glassy state. J Chem Phys 28(3):373.
- 19. Berthier L, Biroli G (2011) Theoretical perspective on the glass transition and amorphous materials. Rev Mod Phys 83(2):587-645.
- 20. Ritort F, Sollich P (2003) Glassy dynamics of kinetically constrained models. Adv Phys 52(4):219-342.

Materials and Methods

We model the system by enclosing N semiflexible polymers modeled using the standard Kremer-Grest model (38) and formed by M beads of nominal size σ in a box with periodic boundary conditions of linear size $\it L$. The monomer fraction is fixed at $\rho = NM/L^3 = 0.1\sigma^{-3}$, and we vary the length of the polymers M as our main control parameter. The simulation time step is denoted with τ_{Br} and corresponds to the time taken by a bead of size σ experiencing a friction ξ to diffuse its size. The simulation is carried out in NVT ensemble with the large-scale atomic/molecular massively parallel simulator (LAMMPS) engine. Further computational details are described in SI Appendix.

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- 21. Biroli G. Bouchaud J-P. Cavagna A. Grigera TS. Verrocchio P (2008) Thermodynamic signature of growing amorphous order in glass-forming liquids. Nat Phys 4(10): 771-775.
- 22. Cammarota C, Biroli G (2012) Ideal glass transitions by random pinning. Proc Natl Acad Sci USA 109(23):8850-8855.
- 23. Karmakar S, Parisi G (2013) Random pinning glass model. Proc Natl Acad Sci USA 110(8):2752-2757.
- 24. Gokhale S, Hima Nagamanasa K, Ganapathy R, Sood AK (2014) Growing dynamical facilitation on approaching the random pinning colloidal glass transition. Nat Commun 5:4685.
- 25. Nagamanasa KH, Gokhale S, Sood K, Ganapathy R (2015) Direct measurements of growing amorphous order and non-monotonic dynamic correlations in a colloidal glass-former. Nat Phys 11(5):403-408.
- 26. Ozawa M, Kob W, Ikeda A, Miyazaki K (2015) Equilibrium phase diagram of a randomly pinned glass-former. Proc Natl Acad Sci USA 112(22):6914-6919.
- 27. Kob W, Coslovich D (2014) Nonlinear dynamic response of glass-forming liquids to random pinning. Phys Rev E Stat Nonlin Soft Matter Phys 90(5-1):052305
- 28. Grosberg AY, Rabin Y, Haylin S, Neer A (1993) Crumpled globule model of the threedimensional structure of DNA. Europhys Lett 23(5):373-378.
- 29. Halverson JD, Smrek J, Kremer K, Grosberg AY (2014) From a melt of rings to chromosome territories: The role of topological constraints in genome folding. Rep Prog Phys 77(2):022601.
- 30. Brás AR, et al. (2014) Compact structure and non-Gaussian dynamics of ring polymer melts. Soft Matter 10(20):3649-3655.
- 31. Rosa A, Everaers R (2014) Ring polymers in the melt state: The physics of crumpling. Phys Rev Lett 112(11):118302.
- 32. Bernabei M, Bacova P, Moreno AJ, Narros A, Likos CN (2013) Fluids of semiflexible ring polymers: Effective potentials and clustering. Soft Matter 9(4):1287.
- 33. Likos CN, Narros A, Moreno A, Capone B (2014) Multi-blob coarse graining for ring polymer solutions. Soft Matter 10(48):9601-9614.
- 34. Mattsson J, et al. (2009) Soft colloids make strong glasses. Nature 462(7269):83-86.
- 35. Mirny LA (2011) The fractal globule as a model of chromatin architecture in the cell. Chromosome Res 19(1):37-51.
- 36. Kang H, Yoon YG, Thirumalai D, Hyeon C (2015) Confinement-induced glassy dynamics in a model for chromosome organization. Phys Rev Lett 115(19):198102.
- 37. Bi D, Yang X, Marchetti MC, Manning ML (2015) Motility-driven glass and jamming transitions in biological tissues, arXiv:150906578.
- 38. Kremer K, Grest GS (1990) Dynamics of entangled linear polymer melts: A moleculardynamics simulation. J Chem Phys 92(8):5057.