

# Broadband high-efficiency dielectric metasurfaces for the visible spectrum

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**Metasurfaces are planar optical elements that hold promise for overcoming the limitations of refractive and conventional diffractive optics. Original dielectric metasurfaces are limited to transparency windows at infrared wavelengths because of significant optical absorption and loss at visible wavelengths. Thus, it is critical that new materials and nanofabrication techniques be developed to extend dielectric metasurfaces across the visible spectrum and to enable applications such as high numerical aperture lenses, color holograms, and wearable optics. Here, we demonstrate high performance dielectric metasurfaces in the form of holograms for red, green, and blue wavelengths with record absolute efficiency (>78%). We use atomic layer deposition of amorphous titanium dioxide with surface roughness less than 1 nm and negligible optical loss. We use a process for fabricating dielectric metasurfaces that allows us to produce anisotropic, subwavelength-spaced dielectric nanostructures with shape birefringence. This process is capable of realizing any high-efficiency metasurface optical element, e.g., metalenses and axicons.**

metasurface | hologram | nanophotonics

**T**ransmissive dielectric metasurfaces (1–6) (DM)—optical devices composed of subwavelength-spaced units and near-flat profiles compared with refractive optics (7–9)—have allowed unprecedented control over optical wavefronts (5) while circumventing Ohmic losses associated with plasmonic metasurfaces. Due to the scientific and technological importance of visible wavelengths, there has been increasing effort to realize DMs at these wavelengths. For example, silicon nitride DMs have been realized at a red wavelength ( $\lambda = 633$  nm) (10), scattering properties of coupled TiO<sub>2</sub> resonators (11) have been examined, and TiO<sub>2</sub> binary diffraction gratings have been fabricated via top-down etching (12). However, no current DM implementation has been capable of providing arbitrary phase control of an optical wavefront while maintaining high efficiency across the entire visible spectrum (especially at blue and green wavelengths). Furthermore, the typical top-down techniques used to implement these metasurfaces can introduce significant surface roughness and make it difficult to create subwavelength sampling of a desired optical phase profile.

In this paper, we demonstrate amorphous TiO<sub>2</sub> metasurfaces that maintain high efficiency across the entire visible spectrum. Our approach to creating DMs uses a bottom-up nanofabrication via atomic layer deposition providing high-aspect ratio, anisotropic dielectric nanostructures with minimal surface roughness. As proof of the concept that we can provide control of the phase of a wavefront from 0 to  $2\pi$ , a requirement for many optical components, we produced metasurface holograms based on geometric phase. Efficient metasurfaces with metallic components operating in reflection have been demonstrated at red and near-infrared wavelengths (13, 14) but have efficiencies of <1% and <10% at blue and green wavelengths, respectively (15, 16). Thus, the TiO<sub>2</sub> metasurfaces we demonstrate here provide substantial improvement with efficiencies from 78 to 82%. Additionally, the use of a dielectric material allows our devices to function in transmission with high efficiency, which is necessary for many optical components.

The ability to realize high-efficiency dielectric metasurfaces is critically dependent on the optical properties of the constituent material, as characterized by the complex refractive index,  $\tilde{n} = n + ik$ . The material should have negligible absorption loss ( $k \approx 0$ ) with relatively high refractive index ( $n > 2$ ) at visible wavelengths. Although a negligible absorption is necessary for high transmission efficiency, a high refractive index ensures strong confinement of the light, which ultimately allows full control over the phase of the exiting wavefront (phase changes from 0 to  $2\pi$ ). The material should be optically smooth, having a surface roughness that is much smaller than the wavelength of light. High surface roughness introduces extraneous light scattering, which is a significant source of loss. To minimize roughness, the material must be amorphous or single-crystalline, as the grain boundaries in polycrystalline materials increase roughness and act as light-scattering centers.

We chose amorphous TiO<sub>2</sub> as our base material because it has a transparency window ( $k \approx 0$ ) for wavelengths longer than  $\sim 360$  nm and its interband transition lies just outside of the visible spectrum, resulting in a sufficiently high index of refraction for strong light–matter interactions. We deposit TiO<sub>2</sub> by atomic layer deposition (ALD) at 90 °C with tetrakis(dimethylamido)titanium (TDMAT) as the precursor due to its high deposition rate and absence of defect-driven absorption that is typical of TiCl<sub>4</sub>-based precursors (17). Additionally, use of an ALD process offers several advantages. ALD is a self-limiting process providing conformal coverage and monolayer control of the film thickness (18). The conformal nature of the technique is essential to produce high

## Significance

**Metasurfaces are optical elements that can mimic and expand on the functionality of refractive optics with a comparatively thin and planar profile. Transmissive metasurfaces suffer from high optical loss at visible wavelengths because of currently used materials and fabrication techniques. This work introduces metasurfaces that operate across the visible spectrum with high efficiencies. As the basis of our metasurfaces, we used a common material, titanium dioxide, and a fabrication method based on atomic layer deposition that creates highly anisotropic nanostructures. The results presented here provide an important advance for realizing optical components at visible wavelengths—e.g., lenses, holograms, and phase shifters—with orders of magnitude reduction in thickness compared with traditional refractive optics.**

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result of interband absorption. From the ellipsometry data and corresponding TL model, we determine the bandgap,  $E_g$ , to be 3.456 eV [the full set of fitting parameters (Table S1) and raw ellipsometry data (Fig. S1) are included in Supporting Information]. This value of the bandgap is in good agreement with previously reported values for amorphous  $\text{TiO}_2$  (22).

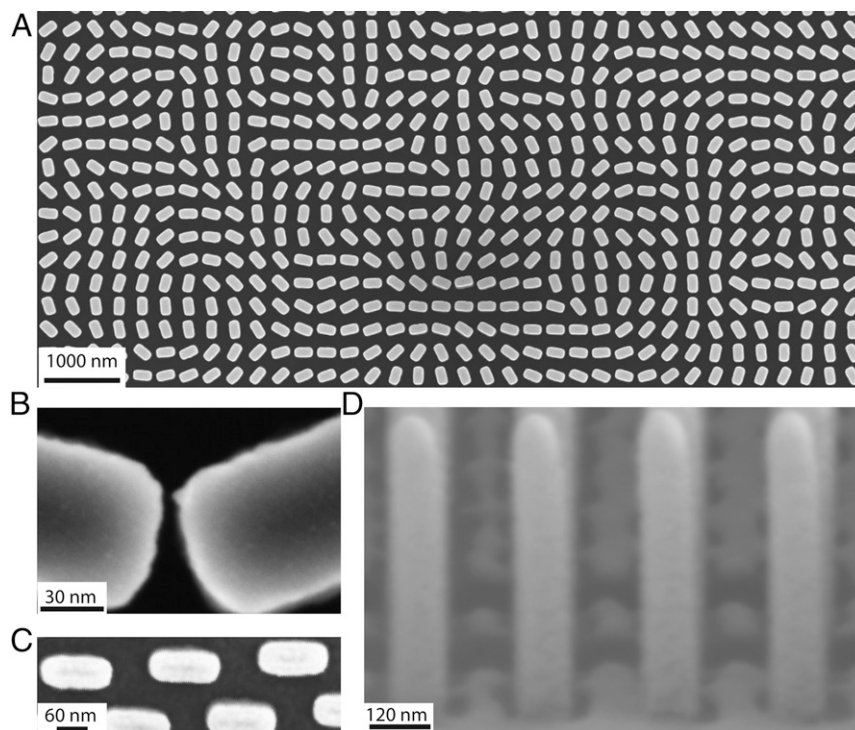
Our  $\text{TiO}_2$  films also exhibit surface roughness that is much smaller than the incident wavelength, as characterized by atomic force microscopy (AFM). Fig. 1B shows an AFM scan of a typical  $\text{TiO}_2$  film deposited on a fused silica substrate. The scan has a root mean square (RMS) roughness of 0.738 nm, which is on the order of the surface roughness of the underlying substrate (Fig. S2). Due to the absence of grain boundaries verified by AFM, coupled with the X-ray diffraction data (Fig. S3), we conclude that the material is amorphous. The measured roughness of the amorphous  $\text{TiO}_2$  is an order of magnitude less than other phases such as polycrystalline rutile or anatase (20). The latter two phases generally have grain boundaries and RMS roughness as high as 5–10 nm, both of which contribute to light-scattering loss (23).

To achieve highly efficient metasurface devices while preserving the optical properties of ALD-prepared  $\text{TiO}_2$ , we used the fabrication process shown in Fig. 2. We spun electron beam resist (EBR) onto a fused silica substrate to produce a layer with thickness,  $t_{\text{resist}}$  (Fig. 2A). Control of  $t_{\text{resist}}$  is important because it sets the height of our final nanostructures. We patterned the resist using electron beam lithography and subsequent development in solution to remove the exposed EBR. This pattern is the inverse of our final metasurface (Fig. 2B). We transferred the exposed sample to an ALD chamber set to 90 °C. The purpose of this temperature is twofold: It produces the desired amorphous phase and keeps the EBR below its glass transition temperature (i.e., prevents deterioration of the nanoscale patterns). During the deposition, the gaseous  $\text{TiO}_2$  precursor (TDMAT) coated all

exposed surfaces, producing a conformal film on the top and side of the EBR as well as on the exposed fused silica substrate (Fig. 2C). We allowed the ALD process to reach a specific thickness such that all features had been completely filled with  $\text{TiO}_2$ . Because the conformal ALD process fills exposed features from both sides, the total ALD film thickness required is  $t_{\text{film}} \geq w/2$ , where  $w$  is the maximum width of all gaps (Fig. 2D). In practice, we allowed the deposition to proceed well beyond the minimum requirement of half the feature width to ensure that  $\text{TiO}_2$  had sufficiently diffused into all pores and that there are no voids in the final nanostructures. We removed the residual  $\text{TiO}_2$  film that coats the top surface of the resist by reactive-ion-etching the sample in a mixture of  $\text{BCl}_3$  and  $\text{Cl}_2$  gas (8:2), similar to a planarization technique. The etch depth was equal to  $t_{\text{film}}$  so that the etching process exposes the underlying resist and the top of the nanostructures (Fig. 2E). We removed the remaining resist and left only the nanostructures that make up our metasurface (Fig. 2F). In this way, we obtained structures of heights  $t_{\text{resist}}$  while only needing to deposit a film of thickness  $t_{\text{film}} \approx w/2$ , which is timesaving and efficient.

We note that this approach is different from standard liftoff techniques, as one cannot use conventional liftoff process due to the conformal coating of the ALD films. In addition, creating high-aspect ratio nanostructures via dry etching of  $\text{TiO}_2$ , similar to other dielectric materials, is difficult and can lead to increased sidewall roughness. This ALD process is different from previous techniques (24, 25) where the patterns created are generally fixed by a set template (e.g., anodic alumina or inverse opal). Here we are free to define more complex nanostructures because we use ALD directly on exposed EBR.

Investigation of the structures, fabricated using the process above, shows that we produce nanostructures (nanofins) with the desired anisotropy and subwavelength dimensions. Fig. 3 shows



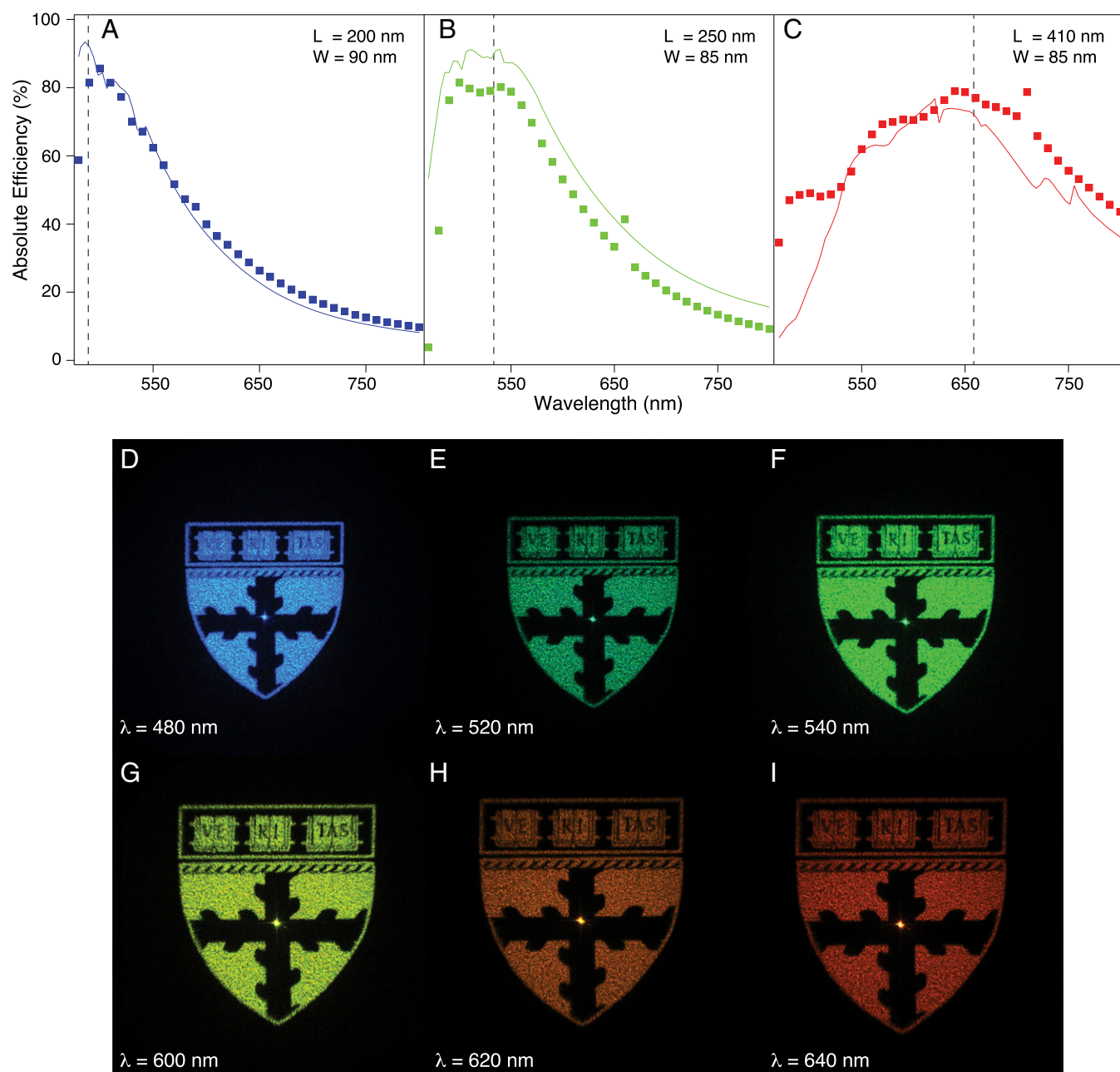
**Fig. 3.** Scanning electron microscope images of fabricated structures. (A) Large-scale view of a fabricated metasurface hologram composed of  $\text{TiO}_2$  nanofins. (B) Zoomed top view of the metasurface showing individual nanofins. The fabricated nanofins are free from residual resist and have dimensions  $\pm 10$  nm of the designed dimensions of 250 nm  $\times$  85 nm. It can also be seen that, with this fabrication technique, we can achieve gaps between structures as small as 6 nm. (C) Top view of structures with lateral dimensions of  $\sim 40$  nm. (D) Cross-section (side view) of nanofins exhibiting vertical sidewalls with a height of  $\sim 600$  nm. The oscillations in contrast between nanofins result from shadowing effects during deposition of a metal film used to prevent charging while we image the samples.



SEM images of a fabricated metasurface hologram. As seen in Fig. 3A, we can densely pattern large areas with subwavelength-spaced  $\text{TiO}_2$  nanofins. This ultimately ensures that the majority of the incident light is imprinted with a desired phase, higher diffraction orders are suppressed, and we can produce, in our case, holographic images with high efficiency. Fig. 3B shows an SEM of the metasurface at high magnification, allowing resolution of individual nanofins. With this process, we can also obtain extremely small spacing between individual nanofins, as shown by the 6-nm gap in Fig. 3B. The structures in this case have dimensions of

$250 \text{ nm} \times 90 \text{ nm}$ , but we can produce structures with dimensions as small as  $40 \text{ nm}$  (Fig. 3C).

Structures that deviate from  $90^\circ$  sidewalls, taking on a more triangular cross-section, can introduce phase errors in the metasurface. Similarly, structures with large voids or defects will reduce the effective refractive index of the nanofins. Fig. 3D shows a cross-section SEM of the nanofins (vertical direction in the image is the direction of light propagation). We observe that the angle of the structures in the vertical direction is  $89^\circ$ ; that is, the nanostructures are highly anisotropic. This anisotropy is despite the fact that the nanostructures shown here are relatively tall compared with their



**Fig. 4.** Measured absolute efficiencies and holographic images. (A–C) Measured (square markers) and simulated (solid line) hologram efficiencies. We define the absolute efficiency as the ratio of the total optical power of the hologram to the total optical power transmitted through an aperture of the same size as the hologram ( $300 \times 300 \mu\text{m}^2$ ). The vertical dashed line marks the design wavelengths of each device, and device dimensions are (A)  $200 \text{ nm} \times 90 \text{ nm}$ , (B)  $250 \text{ nm} \times 85 \text{ nm}$ , and (C)  $410 \text{ nm} \times 85 \text{ nm}$ . (D–I) Holographic images covering the visible spectrum. The input wavelength is (D)  $480 \text{ nm}$ , (E)  $520 \text{ nm}$ , (F)  $540 \text{ nm}$ , (G)  $600 \text{ nm}$ , (H)  $620 \text{ nm}$ , and (I)  $640 \text{ nm}$ . All images were obtained from the device designed for  $480 \text{ nm}$  and show the broadband behavior of a single device. The bright spot in the center of the image is due to the propagation of zero-order light.

other dimensions, with heights of 600 nm. Similarly, we do not see any void formation in the center of the nanofins. From the SEMs, it can also be seen that the final nanostructures have minimal surface and sidewall roughness.

To demonstrate the efficiency and functionality of our TiO<sub>2</sub> metasurfaces, we designed three metaholograms (metaholograms) to have peak efficiencies at wavelengths of 480, 532, and 660 nm. We impart the required phase via rotation of the birefringent TiO<sub>2</sub> nanofins shown in Fig. 3—known as geometric or Pancharatnam–Berry (PB) phase (26–28). (For more detail on this technique, see refs. 1 and 27, *Materials and Methods*, and *Supporting Information*). An advantage of the PB phase is that the imparted phase is wavelength-independent (only the efficiency changes with wavelength), thus providing an ideal platform to test the performance of our metasurface over the entire visible range. We compute the phase map of a holographic image, the binary Harvard logo, by means of the Gerchberg–Saxton phase-retrieval algorithm (29) (Fig. S4).

Fig. 4 A–C shows the measured and simulated efficiencies as a function of wavelength from 480 nm to 800 nm for the metaholograms designed at  $\lambda = 480, 532,$  and  $660$  nm, respectively. (Figs. S5 and S6 show simulated field profiles and transmission for the nanopillars.) We define absolute efficiency as the total optical power of the reconstructed Harvard logo divided by the optical power measured through a  $300 \times 300 \mu\text{m}^2$  square aperture, which is the same size as the hologram (see *Materials and Methods* and Fig. S7 for measurement details). The experimental results generally follow the simulation data and reach maxima of 82, 81, and 78% near the design wavelengths of 480, 532, and 660 nm, respectively. These values are, to our knowledge, the highest reported to date, even compared with reflective metasurfaces (16, 30, 31).

We note that, in certain cases, there are discrepancies between the simulated and measured trend. Differences between designed and fabricated nanofin dimensions and the possibility of weak coupling between fins likely cause these discrepancies. Additionally, the total efficiency of a geometric phase device depends on the transmission for an electric field polarized along the long or short axis of the individual elements (32). We see high transmission through the elements (Fig. S6), especially near the design wavelengths, but these transmission values could be further increased by exploring antireflective coatings or by impedance matching techniques.

Because the geometric phase is a wavelength-independent effect, we can generate holographic images across the visible spectrum from a single design. Fig. 4 D–I shows the holographic images across the visible spectrum for the hologram with a design wavelength of 480 nm. The subwavelength spacing and oversampling of the phase map can be seen in the images because there is sharp resolution of fine features such as the word “VERITAS” at the top of the Harvard crest. The bright spot near the center of the cross of the Harvard logo is from the zero order. However, at the design wavelength, the ratio of the intensity in the zero order to the total intensity contained in the hologram image is  $\sim 1\%$ .

Although we have chosen to demonstrate our process using PB-phase metaholograms, the TiO<sub>2</sub> properties and fabrication process are not limited to this specific type of metasurface. For example, simulations using the measured optical constants of our TiO<sub>2</sub> and structural dimensions achievable with our fabrication process show that we can vary pillar dimensions to provide full  $2\pi$ -phase coverage rather than using the geometric phase (Fig. S8). Thus, one can also use the results we presented here to produce DMs that use linear birefringent resonators to encode phase information (5). Moreover, the demonstration of full 0- to  $2\pi$ -phase coverage enables the implementation of any DM components with high efficiency at visible wavelengths such as axicons, lenses, gratings, and polarization beam splitters.

We have detailed the experimental realization of highly efficient DMs that span the visible spectrum. We used ALD to produce smooth amorphous TiO<sub>2</sub> films that are transparent for wavelengths longer than 360 nm and have an index of refraction that is sufficiently high to provide complete phase control over an optical wavefront. The fabrication technique for these metasurfaces, requiring only single-step lithography, provides a simple method to produce the highly anisotropic nanostructures that are necessary for DMs. Although we used TiO<sub>2</sub>, our process is applicable to any material that can be deposited via ALD. The fabricated metasurface holograms have the highest recorded efficiencies to date (82, 81, and 78%) at their respective design wavelengths. The technique presented here is general and can be applied to any type of metasurface. Careful consideration of the optical properties of our base material and the precision of the fabrication technique allowed us to expand the utility of DMs to visible wavelengths. This work can enable the creation of new compact optical systems with thicknesses that are orders of magnitude less than traditional optical systems.

## Materials and Methods

**Device Fabrication.** All devices used above were fabricated on an f-silica substrate. Substrates were first coated with hexamethyldisilazane to promote resist adhesion followed by spin coating of undiluted positive-tone EBR (ZEP-520A; Zeon Chemicals). For the devices shown, we spun the resist at 1,750 rpm to achieve the desired resist thickness of 600 nm. The resist was then baked at 180 °C for 5 min. Subsequently, samples were coated with 10 nm of chromium via electron beam evaporation to avoid charging effects during the writing process. The patterns were exposed using an accelerating voltage of 125 kV (ELS-F125; Elionix Inc.) and developed in *o*-xylene for 60 s under gentle agitation. For the ALD (Savannah; Cambridge Nanotech) of TiO<sub>2</sub>, a standard two-pulse system of water and the TDMAT precursor was used with a 0.2-s water pulse followed by a 7-s delay and a 0.4-s TDMAT pulse followed by a 10-s delay. The system was left under continuous 20 cm<sup>3</sup>/min flow of N<sub>2</sub> carrier gas and was maintained at 90 °C throughout the process. This led to an overall deposition rate of  $\sim 0.7$  nm per cycle. Reactive ion etching was carried out on Unaxis inductively coupled plasma (ICP) reactive ion etching with a mixture of Cl<sub>2</sub> and BCl<sub>3</sub> gas (3 and 8 cm<sup>3</sup>/min, respectively) at a pressure of 4 mTorr, substrate bias of 150 V, and ICP power of 400 W. Etch rates were typically between 1.3 nm/s and 1.6 nm/s. After processing was complete, the samples were exposed to UV irradiation and ozone, followed by soaking in Remover PG (MicroChem Corporation) for 24 h.

**Device Design and Simulation.** We translate the computed phase map,  $\varphi(x, y)$ , into a spatial distribution of angles,  $\theta(x, y) = \varphi(x, y)/2$ , that sets the rotation angle of a given nanofin at position  $(x, y)$ . The size of each metahologram is  $300 \times 300 \mu\text{m}^2$ , consisting of 600-nm-tall TiO<sub>2</sub> nanofins. Simulations were performed using 3D finite-difference time-domain method (FDTD; Lumerical Inc.). The nanofin’s height, width, and length were optimized to provide a  $\pi$ -phase shift between its major and minor axis, which is required to achieve maximum conversion efficiency based on the geometrical phase (see Figs. S5 and S6 and surrounding discussion).

**Device Measurement.** Devices were characterized using the setup shown in Fig. S7 consisting of a fiber-coupled laser source (supercontinuum laser; NKT Photonics), linear polarizers, quarter waveplates, a lens, and a camera/detector. All measurements were carried out using circularly polarized light as required for PB phase. A linear polarizer and quarter waveplate at the input generated the circularly polarized light, and the set of quarter waveplate and linear polarizer at the output filtered out unconverted light—light whose helicity is the same as the input light.

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