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Microribbon-based hydrogels accelerate stem cell-based bone regeneration in a mouse critical-size cranial defect model

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Abstract

Stem cell-based therapies hold great promise for enhancing tissue regeneration. However, the majority of cells die shortly after transplantation, which greatly diminishes the efficacy of stem cell-based therapies. Poor cell engraftment and survival remain a major bottleneck to fully exploiting the power of stem cells for regenerative medicine. Biomaterials such as hydrogels can serve as artificial matrices to protect cells during delivery and guide desirable cell fates. However, conventional hydrogels often lack macroporosity, which restricts cell proliferation and delays matrix deposition. Here we report the use of injectable, macroporous microribbon (µRB) hydrogels as stem cell carriers for bone repair, which supports direct cell encapsulation into a macroporous scaffold with rapid spreading. When transplanted in a criticalsized, mouse cranial defect model, µRB-based hydrogels significantly enhanced the survival of transplanted adiposederived stromal cells (ADSCs) (81%) and enabled up to three-fold cell proliferation after 7 days. In contrast, conventional hydrogels only led to 27% cell survival, which continued to decrease over time. MicroCT imaging showed µRBs enhanced and accelerated mineralized bone repair compared to hydrogels (61% vs. 34% by week 6), and stem cells were required for bone repair to occur. These results suggest that paracrine signaling of transplanted stem cells are responsible for the observed bone repair, and enhancing cell survival and proliferation using µRBs further

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promoted the paracrine-signaling effects of ADSCs for stimulating endogenous bone repair. We envision μ RB-based scaffolds can be broadly useful as a novel scaffold for enhancing stem cell survival and regeneration of other tissue types.

Keywords

macroporous; injectable; hydrogels; stem cells; bone

INTRODUCTION

Large size bone defects affect millions of people and can be caused by trauma, disease, congenital malfunction, or tumor resection.¹ Bone autografts and allografts are common treatments that account for 1.3 million operations per year in the United States.² Despite their well-established efficacy and immune compatibility,¹ autografts are often challenged by limited autologous bone supply, donor-site morbidity, and chronic pain. Allografts are more abundantly available, but place the patient at higher risk of immune rejection and disease transmission. Stem cell-based therapy for bone repair has emerged in the last decade to help overcome these challenges.^{3–9} Adult stem cells with osteogenic potential, such as bone marrow-derived mesenchymal stem cells (MSCs) or adipose-derived stromal cells (ADSCs), can be harvested from the patients with minimal invasiveness and expanded in vitro for repairing bony defects.^{3,4} Pluripotent stem cells, including embryonic stem cells (ESCs) and induced pluripotent stem cells (iPSCs) have also been explored for bone repair.^{5,6} Stem cells may contribute to bone regeneration either directly via osteogenic differentiation⁷ or indirectly by paracrine signaling to stimulate endogenous bone healing.⁸ However, the efficacy of stem cells alone for repairing bony defects is often limited due to poor cell survival and engraftment,⁹ lack of structural support,⁹ and inefficient nutrient supply.¹⁰

To enhance the efficacy of stem cell-based therapy for bone repair, extensive attempts have been made to develop tissue-engineering scaffolds as the carriers for transplanting stem cells.^{11–13} Hydrogels are a class of scaffolds that have been widely used to aid tissue regeneration due to their injectability, tunable biochemical compositions, and ease of direct cell encapsulation for achieving uniform cell distribution.^{11,12} However, most hydrogels lack macropores larger than the size of encapsulated cells, which is critical for bone-healing bioactivities including cell spreading, vascularization, and new tissue deposition.¹³ Most hydrogels also lack the mechanical strength for engineering load-bearing tissues such as cartilage and bones.¹⁴ Prefabricated macroporous scaffolds¹⁵⁻²⁰ such as silk-based scaffolds,¹⁵ poly (lactic-*co*-glycolic acid) scaffolds,¹⁶ collagen sponges,¹⁷ and hydroxyapatite-based scaffolds¹⁸ provide macroporosity and stronger mechanical strength and have been used for bone-tissue engineering. However, current fabrication of macroporous scaffolds often require nonbiocompatible conditions, such as chemical erosion¹⁹ and phase separation.²⁰ As such, cells can only be seeded onto the prefabricated porous scaffolds, often leading to poor cell-seeding efficiency and non-uniform cell distributions.²¹ Furthermore, unlike hydrogels, prefabricated macroporous scaffolds are not

moldable or injectable to fit into a complex bone defect,²² which may otherwise promote scaffold-bone contact and subsequent integration.

To overcome the above limitations, we recently developed microribbon (μ RB)-like, crosslinkable elastomers as scaffold building blocks which can encapsulate cells in three dimensions (3D) while simultaneously forming a highly macroporous scaffold.²³ Our previous studies showed that μ RBs facilitate uniform cell encapsulation, allow formation of 3D scaffolds with tunable macroporosity, and support rapid cell spreading and proliferation *in vitro*. Flatness of μ RBs facilitated force absorption and allowed the scaffolds to sustain large cyclic strain without losing integrity, which made them particularly attractive for engineering load bearing tissues.²³ The goal of the current study was to evaluate the potential of μ RB-based scaffolds for *in vivo* bone repair using a critical-sized, mouse cranial defect model. We hypothesized that μ RB-based scaffolds would promote stem cell engraftment and survival after transplantation, and that scaffold macroporosity would enhance host tissue ingrowth and promote bone regeneration.

MATERIALS AND METHODS

Materials

Type A gelatin, methacrylic anhydride, L-lysine hydrochloride, glutaraldehyde, 2,4,6trimethylbenzoyl chloride, dimethylphenylphosphonite were purchased from Sigma–Aldrich (St. Louis, MO). All materials were used as received.

Synthesizing gelatin microribbons (µRBs)

Type A gelatin (GelA) was stirred in dimethyl sulfoxide (15 wt %) at 60 rpm and 50°C for 12 h to form a viscous solution, transferred into a 20-mL syringe pump, and ejected at 5 mL per h at room temperature into a tank of ethanol (3.5 L), which was located 1.8 m under the syringe; the tank was stirred at 1100 rpm. In ethanol, the stream of GelA was partially dried and turned into microfibers, which were further dried with acetone for 3 h to form μ RBs. Asformed μ RBs were chopped into short segments (<3 mm) in ethanol using a homogenizer. To enable photocrosslinking, μ RBs were stirred at 25°C for 3 h in methacrylic anhydride (15 wt % in 100 mL methanol). The methacrylated μ RBs were pre-fixed with glutaraldehyde (0.1% in 200 mL methanol) under vigorous stirring at 25°C for 3 h, washed three times with deionized water, and neutralized for 12 h in L-lysine hydrochloride (1% in 200 mL phosphate-buffered saline (PBS)). These μ RBs were washed eight times with deionized water, freeze-dried, and stored at -20°C before use.

Needle injection of µRBs

 μ RBs were rehydrated in PBS by 7.5 wt % density, incubated at 37°C for 1 h, and transferred into a 1 mL syringe by the plunger side. The μ RBs were injected through a 16-gauge needle into scaffold molds, under a roughly 3 mL min⁻¹ ejection rate. The ejection was recorded using a digital camera. A video showing the injection is uploaded to the journal website.

Synthesizing hydrogel precursor: Methacrylated gelatin

To introduce methacrylate groups, GelA (10 g) was dissolved in 100 mL PBS under 50°C, and methacrylic anhydride (20 mL) was added by drops under constant stirring at 1000 rpm. The reaction continued for 2 h at 50°C. Crude product of methacrylated gelatin (GelMA) was extracted by dripping the solution into acetone (3 L), which precipitated GelMA and removed excessive methacrylic anhydride and by products. GelMA was purified by dialysis in deionized water, freeze-dried, and stored at -20° C before use.

Synthesizing photoinitiator lithium phenyl-2,4,6-trimethylbenzoylphosphinate

At room temperature and under argon, 2,4,6-trimethylbenzoyl chloride (3.2 g) was added dropwise to dimethylphenylphosphonite (3.0 g) under vigorous agitation.²⁴ The mixture was stirred at room temperature for 18 h, then heated to 50°C and mixed with excess lithium bromide (6.1 g) in 2-butanone (100 mL). After 10 min, the product was cooled to room temperature, allowed to rest for 4 h, and washed three times with 2-butanone to remove unreacted lithium bromide. Excess solvent was removed by vacuum.

Isolation and culture of adipose-derived stromal cells (ADSCs)

Mouse ADSCs were collected by harvesting inguinal fat pads isolated from 5- to 6-weekold, luciferase-positive/GFP-positive transgenic mice (supplied by the laboratory of Dr. Joseph Wu, Stanford University). All procedures involving animals were approved by the Stanford Institutional Review Board and NIH guidelines for the care and use of laboratory animals (NIH Publication #85–23 Rev. 1985) have been observed. Fat tissues were washed with PBS and digested at 37°C for 30 min with Blendzyme 3 (0.5 U mL⁻¹; Roche Diagnostics, Indianapolis, IN). Enzyme activity was neutralized with Dulbecco's Modified Eagle Medium containing 10% fetal bovine serum (Invitrogen, Carlsbad, CA) and 1% penicillin/streptomycin solution. The cells were then filtered through a 70- μ m cell strainer to remove cellular debris, counted by using a hemocytometer, and seeded into tissue-culture flasks. Following an initial 48 h of incubation at 37°C and 5% CO₂, cells were washed with PBS and expanded in Dulbecco's Modified Eagle Medium with 10% fetal bovine serum, 1% penicillin/ streptomycin, and 10 ng mL⁻¹ basic fibroblast growth factor (PeproTech, Rocky Hill, NJ). Cells were passaged upon 85–90% confluence and second-passage cells were used for all experiments.

Formation of macroporous scaffolds from µRBs

Freeze-dried μ RBs were rehydrated in PBS containing 0.05% photoinitiator to reach a μ RB density of 150 mg mL⁻¹. Following 1 h of incubation at 37°C, the precursor was sandwiched between two glass slides (with a gap of 0.25 mm) and exposed to ultraviolet light (365 nm, 2 mW cm⁻²) for 4 min. This exposure photocrosslinked the μ RBs to produce a macroporous scaffold. For cell encapsulation, trypsinized ADSCs were gently mixed with the μ RB precursor to reach a cell density of 10 million cells/mL prior to photocrosslinking. After 24 h of incubation (37°C, 5% CO₂), cell-laden scaffolds were cut into circular samples (4 mm in diameter) and transferred to 24-well plates before further use.

Formation of hydrogel scaffolds

GelMA were rehydrated in PBS containing 0.05% photoinitiator to reach a GelMA density of 150 mg mL⁻¹. Trypsinized ADSCs were mixed with the hydrogel precursor to reach a cell density of 10 million cells/mL prior to photocrosslinking. The precursor with ADSCs was sandwiched between two glass slides (with a gap of 0.25 mm) and exposed to ultraviolet light (365 nm, 2 mW cm⁻²) for 4 min, forming hydrogel scaffolds. After 24 h of incubation (37°C, 5% CO₂), cell-laden scaffolds were cut into circular samples (4 mm in diameter) and transferred to 24-well plates before further use.

To prepare cryo-sectioning samples, μ RB-based or hydrogel-based scaffolds were rinsed twice in 37°C PBS, fixed in 4% paraformaldehyde for 2 h, permeabilized by 0.5% Triton-X 100 (Sigma–Aldrich, St. Louis, MO) in PBS for 30 min, soaked in 60 wt % sucrose in PBS for 24 h, soaked in optimal cutting temperature (OCT) compound solution (VWR International, Radnor, PA) for 1 h, and frozen by liquid nitrogen vapor in imbedding molds. 8 μ m-thick slides were prepared by cryo-sectioning the OCT-imbedded samples, and were stored at 5°C before use. To visualize cell morphology, the slides were stained for actin cytoskeletons using fluorescently labeled phalloidin (Sigma–Aldrich, St. Louis, MO) at 1:100 dilution, counter-stained for cell nuclei using Hoechst 33342 stain (Thermo Scientific, Waltham, MA) at 1:500 dilution and imaged using confocal microscopy (SP5, Leica, Buffalo Grove, IL) with oil-immersion lenses, 20× magnification, and 0.7 μ m scanning resolution.

Animal surgery

Animal surgeries were performed in accordance with Stanford University Animal Care and Use Committee guidelines. NIH guidlines for the care and use of laboratory animals (NIH Publication #85–23 Rev. 1985) have been observed. CD-1 Foxn1 nude mice (Charles River Laboratories, Hollister, CA) were used for cranial defect procedures. Each treatment group contained at least four animals. The mice were deeply anesthetized by using an isoflurane vaporizer with 2.5% anesthetic dose. After removing the overlying pericranium, 4-mm cranial defects were created with a trephine drill on the contralateral left parietal bone with meticulous care to avoid damaging the underlying dura mater. The bone defects were treated with a 4-mm diameter µRB scaffold with ADSCs, hydrogel scaffolds with ADSCs, or either type of scaffold without cells. Mice with empty defects were prepared as a negative control.

Bioluminescence imaging

From day 0 to week 6 after defect generation and scaffold implantation, ADSC viability and implant location were monitored weekly via bioluminescence imaging. Mice were anesthetized with 2.5% isoflurane and injected subcutaneously under the scruff with 200 μ L reporter probe D-luciferin (30 mg mL⁻¹). An IVIS Spectrum system (Caliper Life Sciences, Hopkinton, MA) was used to image the mice under 1.5% isoflurane anesthesia, at 60-s exposure time and 5 min acquisition period. Each mouse was imaged for photon radiance from the defect site, and was scanned until the peak signal was reached. Radiance was quantified in photons per second per centimeter squared per steradian.

X-ray microtomography

At weeks 0, 2, 4, and 6, the mineralization level of the implants was monitored via X-ray microtomography using a large-field Inveon PET-CT (Siemens, Washington, D.C.), with settings of 80 kVp X-ray voltage, 500 μ A anode current, 80 μ m voxel resolution, and 500 ms time for each 360° rotational step. The two-dimensional projection images were reconstructed into three-dimensional models with ImageJ (National Institutes of Health, Bethesda, MD). Voxels at the supraoccipital part of the occipital bone were used as the threshold to identify mineralized bone formation. Percentage healing in each mouse was determined by calculating the amount of reduction in the defect area.

Histology and immunostaining

Mice were harvested at week 6 for histology. Tissue samples were fixed for 2 h in 4% paraformaldehyde, demineralized for 2 weeks in 16% ethylenediaminetetraacetic acid, and embedded in optimal cutting temperature compound before cryo-sectioning. Tissue morphology was examined with Masson trichrome staining (Thermo Scientific, Waltham, MA). For immunostaining, sectioned slices were treated with blocking buffer consisting of 2% goat serum and 3% bovine serum albumin in 1X PBS and incubated overnight at 4°C in rabbit polyclonal antibody against CD31 (1:100 dilution), osteocalcin (1:100 dilution), or green fluorescent protein (GFP) (1:100 dilution) (all from Abcam, Cambridge, MA). For secondary staining, sectioned slices were incubated for 1 h at room temperature with secondary antibodies: goat anti-rabbit horseradish peroxidase/3, 3′-diaminobenzidine for CD31/GFP (Abcam), and goat anti-rabbit Alexa Fluor 488 for osteocalcin (1:200 dilution) (Invitrogen, Carlsbad, CA). Nuclei were counterstained with Hoechst 33342 stain (Thermo Scientific, Waltham, MA) and images were taken with a Zeiss fluorescence microscope. Sections were stained with all reagents without primary antibody for negative controls.

Stiffness measurements via atomic force microscopy

The stiffness of μ RB and hydrogel scaffolds was measured with a silicon nitride cantilever with 5.4- μ m colloidal probe (AppNano), 0.03–0.08 N m⁻¹ spring constant, and 4–6 kHz fundamental resonance frequency. All measurements were performed in warm PBS at 37°C using a MFP-3D system (Asylum Research). Stiffness maps of μ RB/hydrogel (10 × 10 grid across the width of the ribbon) were measured in force-volume mode on three samples; each map yielded 100 force-distance curves recorded at a probing rate of 0.5 Hz. The stiffness value was calculated by fitting the loading part of each force-distance curve to the Hertz model, assuming a Poisson's ratio of 0.4 for the samples.

Statistical analysis

All data were expressed as mean \pm standard error. Statistical significance was determined via analysis of variance using Student's *t* test with equal variance. *p* values (two-tailed) <0.05 were considered statistically significant, and *p* values <0.005 were considered statistically highly significant.

RESULTS

Gelatin µRBs form macroporous scaffolds via photocrosslinking

Photocrosslinkable µRBs were synthesized by wet-spinning as previously reported.²³ Gelatin dissolved in dimethyl sulfoxide was ejected from a syringe into stirred ethanol and precipitated to form microfibers [Fig. 1(A)]. The fibers were turned into 40–70 µm wide ribbons via acetone drying [Fig. 1(B)], dissociated, methacrylated, glutaraldehyde fixed, and freeze-dried for storage [Fig. 1(C,D)]. Prior to use, µRBs were rehydrated in phosphatebuffered saline and mixed with trypsinized mouse ADSCs [Fig. 1(G,H)]. The paste-like physical property of the rehydrated µRBs allows their injection into scaffold molding through needles of 16 G or larger size [Fig. 1(I) and Supporting Information Video]. To fabricate macroporous scaffolds, molded µRBs were photocrosslinked (365 nm, 4 min, 2.5 mW cm⁻²) in the presence of a photoinitiator [Fig. 1(E–J)]. The resulting scaffolds were characterized by high mechanical flexibility [Fig. 2(A-C)] and highly interconnected macropores that provided abundant internal space and surface area [Fig. 1(K)]. All these features have been reported to promote osteogenic differentiation of stem cells and facilitate bone-matrix production.^{13,25} To provide stiffness conditions reported to promote bone differentiation,^{12,26} we modulated our aldehyde-fixation procedures to obtain µRBs with stiffness of 75.5 ± 22.7 kPa [300 point measurements across 3 microribbons; Fig. 2(D,E)]. Stiffness was measured with atomic force microscopy.²⁷

To compare the efficacy of μ RB-based scaffolds for bone repair with that of existing hydrogel platforms, we also fabricated hydrogels of methacrylated gelatin as control groups.²⁸ These hydrogels provided collagen-derived ligands that are similar to those displayed by the μ RBs. To provide the similar matrix stiffness, the hydrogel scaffolds were fabricated with a stiffness of 65.1 ± 2.1 kPa using 15% (w/v) methacrylated gelatin as precursor.

µRB-based scaffolds promote cell spreading in 3D

To examine the effects of macroporosity on cell morphology in 3D, μ RB or hydrogel-based precursors (both at 15% (w/v)) were mixed with ADSCs and photocrosslinked into 250- μ m thick, sheet-form scaffolds. Cell-laden scaffolds were incubated in ADSC growth medium for 1 day before implantation or analyzed for cell morphology. Our μ RB-based scaffolds enabled direct cell encapsulation, while forming macropores ranging from 150 to 300 μ m [Fig. 2(F)], a range which was previously reported to facilitate bone-tissue formation *in vivo*. In contrast, conventional gelatin hydrogels exhibited substantially smaller pore sizes (~20 μ m) [Fig. 2(G)]. Confocal imaging 24 h after encapsulation revealed extensive spreading of ADSCs throughout the μ RB-based scaffolds [Fig. 2(H) and Supporting Information Video], while ADSCs in the hydrogel-based scaffolds maintained a round morphology with minimal cell spreading [Fig. 2(I) and Supporting Information Video]. Such dramatic difference in cell morphology indicates the advantage of μ RB-based scaffolds in supporting cell survival and proliferation, which are known to prefer extensive cell spreading.

µRB-based scaffolds promote cell viability in vivo

Critical-sized cranial defects 4 mm in diameter (which do not heal endogenously in the absence of treatment) were created in athymic mice on the right parietal bones as described previously³⁰; these defects were implanted with μ RB- or hydrogel-based scaffolds harvested from the scaffold sheets [Fig. 3(A)]. Four implanted groups were prepared in total: mice with μ RB or hydrogel scaffolds plus ADSCs (μ RB + ADSC and hydrogel + ADSC groups), mice with either type of scaffold without cells (μ RB-only and hydrogel-only groups), and mice with defects that were not implanted with anything (negative control).

Over 6-weeks post-surgery, ADSCs in µRB scaffolds showed significantly higher cell numbers (in percentages relative to day 0) than ADSCs in hydrogel scaffolds. The observed cell numbers were influenced by both cell survival and cell proliferation. On day 1, bioluminescence imaging (BLI) revealed $81.1\% \pm 21.3\%$ ADSC survival in the μ RB + ADSC group, but significantly decreased cell survival ($26.6\% \pm 10.1\%$) in the hydrogel + ADSC group (p < 0.002) [Fig. 3(B–F)]. At week 1, the μ RB + ADSC group displayed substantial ADSC proliferation (289.2% 698.2% vs. day 0, p = 0.002 vs. day 1), while the hydrogel + ADSC group presented roughly unchanged cell numbers ($23.3\% \pm 5.9\%$ vs. day (0, p > 0.5 vs. day 1) [Fig. 3(G)]. From weeks 2 to 6, both groups experienced gradually decreased cell viability, while the ADSCs in µRB scaffolds maintained twice as much viability as those in hydrogel scaffolds (p < 0.05) [Fig. 3(G)]. Because ADSCs in μ RB and hydrogel scaffolds were exposed to similar biochemical ligands and matrix stiffness (75.5 \pm 22.7 kPa vs. 65.1 \pm 2.1 kPa), these BLI results indicated that the dramatically enhanced survival of ADSCs in μ RB scaffolds stemmed from the high macroporosity in the μ RBbased scaffolds, which was the main difference between the $\mu RB + ADSC$ group and the hydrogel + ADSC group.

µRB-based scaffolds with ADSCs promote bone healing, bone-scaffold integration, and vascularization

X-ray microtomography (micro-CT) and histology were used to investigate how scaffold types and ADSC transplantation influence bone healing. The effects of scaffold type were studied with μ RB-based and hydrogel-based implants, and the efficacy of ADSC incorporation was elucidated with cell-laden and acellular implants. Results from empty defects provided a point of reference. To exclude other variables that could affect bone repair, the ADSCs were cultured in growth medium without osteogenic growth factors, and the scaffolds were fabricated without osteoconductive cues such as hydroxyapatite coating³¹ or osteogenic growth factors such as bone morphogenetic protein.³²

Both the μ RBs and HGs seeded with ADSCs led to noticeable mineralized bone formation as early as week 2 (32% ± 14% in μ RB vs. 21% ± 23% in HG), with higher variability across samples in the HG group (Fig. 4). Compared to HG with ADSCs, μ RB with ADSCs led to 52% more mineralized bone in week 2, and this trend continued to increase over time (Fig. 4). By 6 weeks, μ RB with ADSCs led to 79% more mineralized bone repair than HG with ADSCs (Fig. 4). While the difference is not statistically significant due to the high variability in the HG + ADSC group, there is a clear trend that cell seeded μ RBs led to faster and greater mineralized bone formation than cell seeded HG. For both μ RBs and HGs, new

mineralized bone formation was the greatest within the first 4 weeks, and slowed down afterwards (Fig. 4). In contrast to the cell-laden scaffolds, negligible bone filling (<5%) was observed in defects with scaffolds alone without cells and in defects without implants (Figs. 4 and 5). Because μ RB and HG scaffolds provided similar stiffness and biochemical cues, the micro-CT results indicate that the observed faster bone filling stemmed from the macroporosity of the μ RB-based scaffolds. Furthermore, the dramatic difference between the cell-laden groups and the acellular groups demonstrates that transplanted ADSCs play an essential role in bone regeneration.

Trichrome staining showed extensive collagen-rich tissue ingrowth in defects implanted with μ RB scaffolds harboring ADSCs, both at the center and at the edges of the defect [Fig. 5(F,K,P)]. In defects implanted with μ RB scaffolds without ADSCs, tissue ingrowth was observed, but with lower collagen content [Fig. 5(HMR)]. Collagen-rich tissues appeared in defects implanted with hydrogel scaffolds with ADSCs [Fig. 5(G,L,Q)]. In defects implanted with hydrogel scaffolds with ADSCs, tissue ingrowth was absent; instead, fibrous capsules formed around the hydrogel scaffolds [Fig. 5(I,N,S)]. In terms of implant integrity, μ RB scaffolds with or without ADSCs exhibited well-preserved macroporous structures, whereas hydrogel scaffolds with or without ADSCs were mostly degraded and disappeared after 42 days post-implantation.

Vascularization is essential for bone healing.³³ Here, immunostaining against cluster of differentiation 31 (CD31), an endothelial cell marker, revealed signs of angiogenesis throughout defects implanted with μ RB scaffolds with or without ADSCs; the macropores resembled interconnected "tunnels" for vascular-tissue expansion [Fig. 6(A,C)]. Vascularization was much more limited in defects implanted with hydrogel scaffolds with ADSCs [Fig. 6(B)]. Signs of angiogenesis were absent in defects without scaffolds and in defects implanted with hydrogel scaffolds without ADSCs [Fig. 6(D,E)]. CD31 staining showed that both μ RB macroporosity and transplanted ADSCs promote implant vascularization. The enhanced vascularization of the μ RB + ADSC group may also support the viability of ADSCs via increased nutrient supply.

ADSCs stimulate endogenous bone regeneration via paracrine signaling

To elucidate how ADSCs participate in bone regeneration (via osteogenic differentiation or via paracrine signaling), we immunostained defects for osteocalcin, a pro-osteoblastic molecule. This staining revealed extensive osteogenesis throughout defects implanted with μ RB scaffolds with ADSCs [Fig. 6(F)]. Osteogenesis was detected in defects with hydrogel scaffolds with ADSCs, but was limited due to hydrogel collapse [Fig. 6(G)]. Osteogenesis was negligible in empty defects [Fig. 6(E,J)] and in defects implanted with either scaffold without ADSCs [Fig. 6(H,I)]. These observations are consistent with our micro-CT results [Fig. 5(A–E)], which showed that bone healing occurred most rapidly in defects implanted with hydrogel scaffolds with ADSCs, occurred more slowly in defects implanted with hydrogel scaffolds with ADSCs, and was absent from the defects lacking scaffolds or ADSCs.

Because our ADSCs were harvested from mice expressing green fluorescent protein (GFP), we used GFP expression to track ADSC fate in the cranial defects. A few transplanted ADSCs were detected at the center and the periphery of defect filled with μ RB scaffolds

[Fig. 6(F)], but not near the hydrogel scaffolds [Fig. 6(G)]. In both groups, new bone formation was mostly contributed by endogenous cells, which did not express GFP. Thus, the observed bone healing [Fig. 5(A,B)] was mainly due to bone formation by host cells that were stimulated by transplanted ADSCs via paracrine signaling.

DISCUSSION

Healing of large bony defects represents a significant challenge clinically, as traditional autografts or allografts are subject to drawbacks that include donor-site complications, insufficient donor tissue supply, and the risk of immune rejection and disease transmission. Stem cell-based therapies augment native bone healing with osteogenic bioactivities, such as osteogenic differentiation and paracrine signaling stimulation.^{7,8} These bioactivities are most effective when the transplanted stem cells are supported physically and biochemically to facilitate cell engraftment, cell proliferation, nutrient supply, and bone-tissue deposition.^{10,34}

Here, we approached such goals by developing microporous scaffolds using photocrosslinkable µRBs as building blocks. *In vivo* transplantation of ADSCs in µRB-based scaffolds dramatically enhanced ADSC engraftment, survival, and proliferation [Fig. 2(B–G)]. The macroporosity formed among µRBs promoted endogenous tissue integration [Fig. 5(K,P)], vascularization [Fig. 6(A)], and mineralized bone formation [Fig. 5(A)]. In comparison with conventional grafting materials such as hydrogels and prefabricated macroporous scaffolds, this novel biomaterial possesses several important advantages, which are described below.

Promotion of cell spreading, survival, and proliferation

Low cell viability is a common challenge for stem cell-based therapies, and is often due to a lack of cell adhesion and spreading after transplantation.^{10,34} In contrast, integrin-mediated cell attachment and the resultant signaling cascades are known to rescue cells from apoptosis,²⁹ stimulated DNA synthesis, and promoted cell survival and growth.³⁵ Nutrient supply also plays an important role in sustaining stem-cell survival.¹⁰ In our μ RB-based scaffolds, cell adhesion is supported by the gelatin-based ligands of the μ RBs, while nutrient diffusion and cell spreading are promoted by the macropores [Fig. 2(H) and Supporting Information Video]. Macropore-supported vascularization [Fig. 6(A,C)] may also promote long-term cell survival. In addition to the beneficial effects on bone repair, such enhancement of cell viability is promising for a broad variety of cell-based therapies, such as the stem cell regeneration of fat,³⁶ cartilage,³⁷ or blood vessels.³⁸

Tunability of macropore formation

Unlike μ RB-based scaffolds, most conventional hydrogels lack macropores that are bigger than cells [Fig. 2(G)], therefore restricting cell proliferation, migration, and new tissue deposition. Here, transplanting ADSCs to cranial defects with methacrylated gelatin hydrogels instead of μ RBs caused significantly lower cell viability [Fig. 3(F,G)], limited vascularization [Fig. 6(A,C)], and slower bone formation [Fig. 5(A,B)]. Efforts have been made to overcome such limitations. Improvements in hydrogel porosity are often achieved

by lowering the hydrogel's polymer concentration.³⁹ Enzyme-cleavable hydrogels allow cell-mediated matrix degradation,⁴⁰ while biocompatible porogens have been embedded and dissolved in hydrogels to produce macropores.⁴¹ Nonetheless, these methods often simultaneously degrade scaffold integrity and the hydrogel's biochemical composition, complicating the control of scaffold properties. In contrast, our macroporous scaffolds are generated in one step by crosslinking the μ RBs as building blocks [Fig. 1(G– K)]. As a result, scaffold macroporosity is determined only by μ RB packing density [Fig. 2(F)], while matrix stiffness [Fig. 2(D,E)] and the biochemical properties that cells sense from the scaffold are determined separately by the properties of individual μ RBs [Fig. 1(D)]. This building block-based approach will simplify the future design of macroporous scaffolds for bone repair and for other types of tissue regeneration.

Direct cell encapsulation and injectability

Conventional macroporous scaffolds are mostly prefabricated, and incorporating cells into them requires external cell seeding, which is prone to inefficient cell distribution²¹ and may become problematic for scaffolds with clinically relevant dimensions. In contrast, in the current strategy, μ RBs are paste-like in phosphate-buffered saline and can be thoroughly mixed with cells before photocrosslinking [Fig. 1(H–K)]. The resulting cell distribution is highly uniform regardless of the level of porosity or scaffold dimensions [Fig. 2(H)]. Such direct cell encapsulation simplifies the control of cell density, which determines the efficiency of cell-to-cell communication and is known to directly influence bone-tissue regeneration.⁴² Moreover, μ RBs can be injected via syringe to fill irregular defects [Fig. 1(I) and Supporting Information Video), which is promising for minimally invasive surgeries²² that aim for less surgical trauma, lower infection risk, and shorter patient recovery time.

Support for ADSC paracrine signaling for endogenous bone healing

Adipose tissues represent an abundant and readily accessible source of stem cells in our body.³ ADSCs are known to stimulate wound healing by delivering paracrine-signaling factors that promote cell survival,⁴³ vascularization,⁴⁴ endogenous stem-cell recruitment, and the differentiation of stem cells along desirable lineages.^{8,43,44} Results from our cranial defect model suggest that such paracrine signaling was primarily responsible for the observed bone healing [Fig. 5(A–E) versus Fig. 6(F–J)]. The relation between higher cell proliferation [Fig. 3(B,C)] and faster defect filling [Fig. 5(A,B)] in the μ RB + ADSC group suggests that μ RBs may further such paracrine-signaling effects by enhancing the proliferation and survival of ADSCs.

Our results also highlight the therapeutic potential of transplanting primary ADSCs with the μ RB-based scaffolds. ADSCs without *ex vivo* manipulations, such as predifferentiation, may be incorporated through a point-of-care process,⁴⁵ in which ADSCs are extracted as primary cells and transplanted with μ RBs immediately upon isolation. This process aims to save time and resources by eliminating *ex vivo* cell manipulation, which may help increase the efficiency and cost-effectiveness of stem cell-based bone therapy.

Potential incorporation of osteoconductive reagents

The efficacy of μ RBs for bone repair was evidenced by progressive bone formation in the cranial defects of mice that received μ RBs and ADSCs [Fig. 5(A,F,K,P)]. However, it is worth noting that our μ RB synthesis remains to be optimized. To evaluate the potential of using μ RBs for stem cell-based bone repair, here we excluded the osteoconductive reagents and osteogenic growth factors that are commonly incorporated into bone-grafting biomaterials.^{31,32} In future studies, our μ RBs may be incorporated with these components to further accelerate bone regeneration.

Versatility

In addition to bone repair, our μ RBs are highly versatile and can be recustomized for various types of tissue regeneration. For example, glycosaminoglycans may be introduced into the μ RBs to promote the formation of cartilage tissue.¹⁴ To enable crosslinking in dark environments, such as inside the body, the methacrylate groups may be replaced with molecules that can be crosslinked through other mechanisms, such as the thiol and maleimide-based ligands that crosslink each other via thiolene coupling.¹¹ μ RB stiffness and packing density may be optimized for other wound-healing processes that are sensitive to matrix stiffness and porosity. Finally, μ RBs may be embedded with drug-releasing vehicles, such as poly (lactic-*co*-glycolic acid) particles,³² to deliver biomolecules for tissue repair, such as bone morphogenetic protein for osteogenesis,³² TGF- β for chondrogenesis,³⁷ and vascular endothelial growth factor for vascularization.³⁸

In summary, the challenges of stem cell-based therapies for large bone defects necessitate new strategies for cell transplantation. Here, we approach this goal by using photocrosslinkable µRBs as carriers for stem cell transplantation, which dramatically enhanced ADSC survival, accelerated defect vascularization and host tissue integration, and promoted the paracrine-signaling effects for endogenous bone healing. This µRB-based platform is highly versatile and can be customized for engineering a broad variety of tissues.

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FIGURE 1.

Fabrication scheme for a cell-laden, μRB-based scaffold. (A) Wet spinning. DMSO, dimethyl sulfoxide. (B) Acetone drying. (C, D) Dissociation, methacrylation, and aldehyde fixation. (E) Cell mixing. (F) Photocrosslinking with photoinitiator (PI) and ultraviolet light (UV). (G) Rehydrating μRBs. (H) Cell mixing. (I) Injection of cells and μRBs via syringe. (J) As-fabricated scaffold. (K) Scaffold macroporosity under scanning electron microscopy.



FIGURE 2.

Physical properties of μ RB-based scaffold. (A–B) μ RB-based scaffold under 90% cyclic compression. (C) Stress–strain loops of μ RB scaffolds show scaffolds maintained mechanical integrity upon cyclic compression, as indicated by overlapping loops. (D) Stiffness of individual μ RB measured by atomic force microscopy (AFM). (E) Stiffness quantified by AFM measurement (n = 300). The stiffness of microribbon (75.5 ± 22.7 kPa, mean ± standard deviation) is within the range that reportedly promotes bone differentiation. (F, G) Scaffold porosity under variable-pressure SEM showed large macropores in the μ RB

scaffolds, and much smaller micropores in the conventional gelatin hydrogels. (H, I) Two days post-encapsulation, adipose-derived stem cells exhibited extensive spreading in μ RB-based scaffolds (H), while remained mostly round in conventional gelatin hydrogels (I) (green: actin; blue: cell nuclei). Scale bars indicate 50 μ m in (D) and (F–I).



FIGURE 3.

Bioluminescence imaging (BLI) showed μ RB-based scaffolds enhanced stem cell survival and proliferation *in vivo* compared to conventional hydrogels after being transplanted in a mouse critical size cranial defect model. (A) Transplanted scaffolds in the cranial defect. (B– E) BLI on days 1 and 7. The red end of the color scale indicates higher signal (in radiance unit, p/sec/cm³/sr). (F) BLI imaging showed μ RB scaffolds led to significantly higher cell survival at day 1 than in conventional hydrogels (p = 0.002). μ RBs enabled cell proliferation up to three-fold after 7 days, whereas the cell number in HG controls remained low; (G) μ RB scaffolds led to consistently higher number of transplanted stem cells than conventional hydrogels throughout 6 weeks (p < 0.05), with the greatest difference observed within the first 2 weeks. n = 4 per group, error-bars: standard deviations.



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FIGURE 4.

Analyzed microCT imaging data showed cell-laden μ RBs led to faster and greater mineralized bone repair than conventional hydrogels in a critical sized mouse cranial defect over 6 weeks. Percent of bone healing was normalized to the defect size at day 1. n = 4 per group, error-bars: standard deviations.

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FIGURE 5.

Representative micro-CT images and histological cross sections of mouse cranial defects at 6 weeks after treatment with μRB-based scaffolds or conventional hydrogels, with or without adipose-derived stem cells. Untreated cranial defects were included as controls. (A–E) Representative micro-CT images indicate only stem cell-seeded scaffolds led to mineralized bone formation in the critical size cranial defects, with ADSC + μRB group showing the highest degree of mineralized bone deposition. (F–T) Masson's Trichrome stain of collagen at cranial defect sites at week 6 after transplantation. The images display a cross-section of the skull, and connective tissue is stained blue. Representative images with higher resolutions of locations of images in F–J are shown in I–T. (F–J) Low magnification views of entire cranial defect region. Arrows: edges of cranial defects. (K–O) High magnification views of central region within the cranial defects, as indicated by yellow box in F. (P–T) High magnification views of interface between implants and the original cranial defects, as indicated by red box in F. Yellow line: edge of original cranial defects. Scale bars: 500 μm in (F–J); 100 μm in (K–T).



FIGURE 6.

Representative staining of CD-31 (for endothelial cells) and osteocalcin (for bone) in cranial defects at 6 weeks *in vivo*. (A–E) CD31-positive cells (brown) counter-stained with Harris's hematoxylin (purple). (F–J) Immunofluorescence staining of osteocalcin (red) counter-stained by transplanted ADSCs (green) and cell nuclei (blue). At 6-week time point, transplanted ADSCs were only detectable in groups treated with μ RB + ADSCs. Most regenerated bone tissues were contributed by endogenous bone regeneration. Scale bars, 100 μ m.