

# Architecture and coevolution of allosteric materials

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**We introduce a numerical scheme to evolve functional elastic materials that can accomplish a specified mechanical task. In this scheme, the number of solutions, their spatial architectures, and the correlations among them can be computed. As an example, we consider an “allosteric” task, which requires the material to respond specifically to a stimulus at a distant active site. We find that functioning materials evolve a less-constrained trumpet-shaped region connecting the stimulus and active sites, and that the amplitude of the elastic response varies nonmonotonically along the trumpet. As previously shown for some proteins, we find that correlations appearing during evolution alone are sufficient to identify key aspects of this design. Finally, we show that the success of this architecture stems from the emergence of soft edge modes recently found to appear near the surface of marginally connected materials. Overall, our *in silico* evolution experiment offers a window to study the relationship between structure, function, and correlations emerging during evolution.**

disordered materials | proteins | evolution

Proteins are long polymers that can fold in a reproducible way and achieve a specific function. Often, the activity of the main functional site depends on the binding of an effector on a distant site (1, 2). Such an allosteric behavior can occur over large distances, such as 20 residues or more (3), and often involves only a sparse subset of residues in the protein (3, 4). Allosteric regulation offers an appealing target for drug design (5), and there is considerable interest in predicting allosteric pathways (6, 7). One central difficulty is that the physical mechanisms allowing such an “action at a distance” remain elusive. In some cases, allostery can be understood as the modulation of a hinge connecting two extended rigid parts of the protein (8, 9), but, often, the displacement field induced by the binding of the effector cannot be described in these terms (4, 10, 11). Another route, statistical coupling analysis (12), considers correlations within sequences of proteins of the same family to infer allosteric pathways (4, 7). The generality of this elegant approach is, however, debated (13).

From a physical viewpoint, specific response at a distance is surprising. The structure of proteins is similar to randomly packed spheres (14). Generically, the response of such systems is nonspecific and decays rapidly in space (in a manner similar to a continuum medium) at distances larger than the particle size; this is true, except close to a critical point where the number of constraints coming from strongly interacting particles is just sufficient to match the number of degrees of freedom of the particles (15). There, the elastic response becomes heterogeneous on all scales (16, 17). This point is illustrated in Fig. 1*A*, showing the rapidly decaying response of a random spring network to a stimulus. However, as shown in Fig. 1*B* (and independently found in ref. 18 using a different algorithm), springs can be moved so that the response extends farther and specifically matches a target response on the other side of the system. This observation raises various questions, including the following: (i) Which network architectures allow for such allosteric response? (ii) Why are these architectures functional? (iii) What is the number of solutions? (iv) As we shall see, a network can be represented by a sequence; knowing only a family of sequences of solutions, can

one predict which sites are important for function from their correlations alone?

In this work, we answer these questions by introducing a model of elastic networks that can evolve according to some fitness function  $F$ , which depends on the response of the material to a well-defined stimulus. Our approach allows for considerable freedom in the choice of the fitness function. As an illustration, we impose here that a displacement of four nodes on one side of the material (the “stimulus”) elicits a given displacement of identical amplitude but different direction on four target nodes on the other side of the system. A key advantage of our scheme is that our algorithm uniformly samples the fitness landscape (we use a Monte Carlo algorithm that turns out to equilibrate rapidly), which allows us to count the number of solutions and compute the entropy  $S(F)$ , as well as to guarantee that the solutions generated are the typical (most numerous) ones. The quality of the solutions can be monitored by an “evolution temperature”  $T_e$  that controls the fitness of the solutions probed. Our central findings are as follows. (i) There exists a transition temperature below which high-quality solutions appear and above which solutions are poor. (ii) High-quality solutions share a specific design. They present a trumpet-shaped region where the material is less constrained, which ends by a marginally connected region in the vicinity of the target. (iii) The response amplitude varies nonmonotonically between the stimulus and active sites. (iv) We rationalize this design based on a recent theory of edge modes in marginally connected disordered media (19). (v) We show that coevolution—the correlations in the structures of the family of solutions—alone is sufficient to identify the trumpet structure. Finally, this detailed characterization of the solutions also points to some of their limitations in using them in thermal environments. We discuss how the fitness function can be changed to alleviate such problems.

## Description of the Evolution Model

Scalar models, where the response of a node is described by a scalar instead of a vector, have been introduced to study

### Significance

**In allosteric proteins, binding a ligand affects activity at a distant site. The physical principles allowing for such an action at a distance are not well understood. Here we introduce a numerical scheme to evolve allosteric materials in which the number of solutions, their spatial architectures, and the correlations among them can be computed. We show that allostery in these materials uses recently discovered elastic edge modes near the active site to transmit information, and that correlations generated during evolution alone can reveal key aspects of this architecture.**

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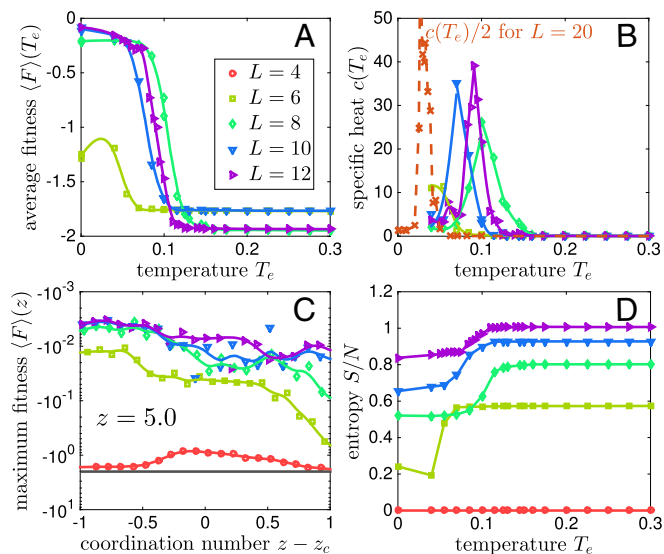
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**Fig. 2.** (A) Average fitness  $\langle F \rangle$  versus evolution temperature  $T_e$  for various system sizes where  $z = 5.0$ . A steep change in fitness is seen near  $T_c \approx 0.09$  for relatively large systems. (B) Specific heat  $c$  versus temperature  $T_e$ . The maximal specific heat increases with the system size  $L$ , suggesting the existence of a transition at some  $T_c$ . (C) Fitness averaged over local maxima  $\langle F \rangle_{T_e=0}$  versus coordination number  $z$  in log-linear scale. The black line shows the fitness if no mechanical response is present at the active sites. (D) Entropy density  $S/N$  versus temperature  $T_e$ . The entropy jump near  $T_c$  indicates the number of degrees of freedom that must be tuned to achieve the desired response.

average of local maxima in the fitness landscape, as reported in Fig. 2C. These structures result from a pure gradient ascent in the fitness landscape. We find that, in the range of coordination we probed, the cost decreases by at least 200-fold with respect to random networks, i.e., the response converges very precisely toward the desired one. Thus, the system does not get stuck in local maxima of poor quality in the fitness landscape.

Finally, we can quantify the number of allosteric networks; it follows  $e^{S(T_e)}$ , where  $S(T_e)$  is the entropy. It satisfies  $dS = c(T_e)dT_e/T_e$ , and is shown in Fig. 2D. For example, at  $T_e = 0.05$ , where networks perform very well, we find that their number is very large, exponential to the system size,  $e^{S(T_e)} \approx 10^{53}$  for  $L = 12$ , but the probability  $p_A$  to obtain such a network by chance is also exponentially low,  $p_A = e^{S(T_e) - S(\infty)} \approx 10^{-10}$  for  $L = 12$ .

### Architecture of Allosteric Networks

Hypostatic networks with  $\delta z < 0$  are extremely floppy. It may be an interesting case to study intrinsically disordered proteins (29), but, for folded proteins, considering  $\delta z > 0$  is more realistic. Henceforth we focus on that case and choose  $z = 5$ . In *SI Appendix*, our results are presented for the floppy case  $\delta z < 0$ .

Which architecture allows for such a long-distance, specific response? A systematic design is revealed by averaging the occupancy of various solutions that our algorithm generates, as shown in Fig. 3A for  $L = 12$  (see *SI Appendix* for larger systems). At high temperature, the structures are essentially random and not functioning. At low temperature, a trumpet-shaped region appears that connects the allosteric and active sites. Specific features are that (i) inside the trumpet, the mean occupancy is lower than the mean, but there are no floppy modes (i.e., modes that do not deform the springs); (ii) the mean occupancy or coordination decreases monotonically from the system center to the active site; (iii) the mouthpiece of the trumpet is surrounded by two more rigid regions, which appear in dark in Fig. 3A; and (iv)

the coordination number is close to its critical isostatic value in the vicinity of the active site (see *Physical Processes Underlying Allostery*).

The trumpet-like architecture is robust: It remains qualitatively unchanged as the mean coordination number is varied, as long as  $\delta z > 0$ . For  $\delta z < 0$ , however, a trumpet still exists (see *SI Appendix*), but it is inverted: It is more coordinated than the rest of the system.

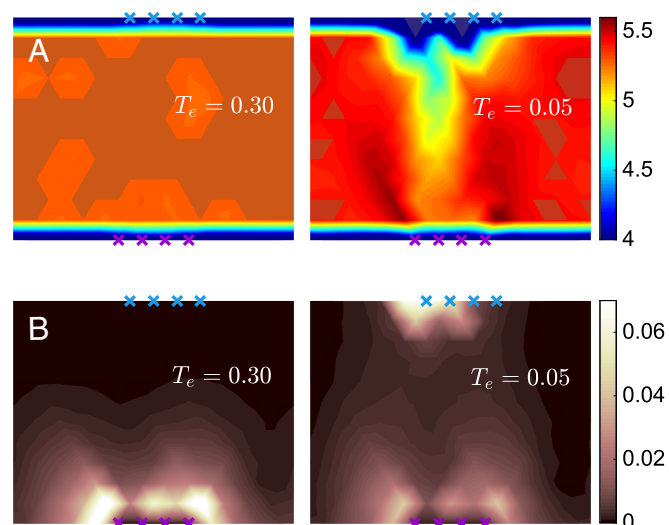
Next we study how such trumpets shape the response to a binding event, by considering the mean-squared magnitude of the normalized response at different nodes  $i$ ,  $\langle |\delta R_i^r|^2 / \sum_i |\delta R_i^r|^2 \rangle$ , as shown in Fig. 3. For random networks, unsurprisingly, the response is large only close to the stimulus site. However, the response of fit networks displays a striking feature: It varies nonmonotonically between the allosteric and the active site. It almost vanishes in the bulk of the material, but reappears near the active site, where it is the strongest.

### Physical Processes Underlying Allostery

The observation that fit networks develop a less-constrained region connecting the stimulus to the active site is not very surprising, because the elastic point response can remain heterogeneous on longer length scales in that case (17). This argument does not explain, however, the strong asymmetry of the trumpet, more coordinated near the stimulus and nearly marginally connected near the active site. We now argue that this design is selected for because it prevents the decay in the amplitude of the signal one expects in normal elastic materials.

Recent works have shown that marginally connected crystals can display edge modes, leading to exponentially growing response when displacements are imposed at the boundary of the system (30–32). It was very recently shown that such “explosive” modes must be present in disordered marginally connected materials as well (19). Such systems, if sufficiently constrained at some of their boundaries, can act as a lever that can amplify complex motions exponentially toward free boundaries (19).

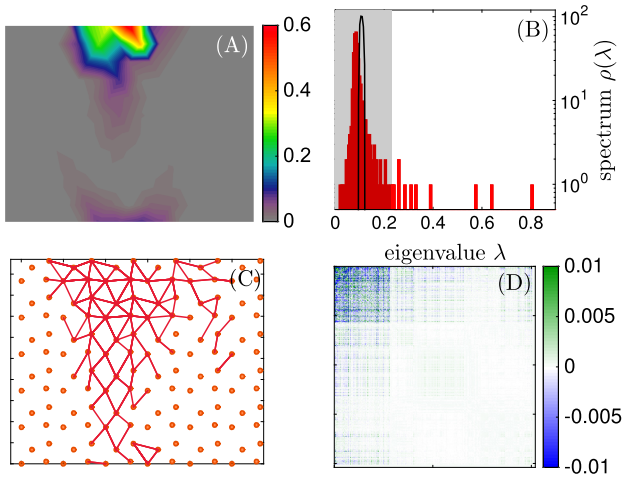
We argue that our allosteric networks are built along this principle. As sketched in Fig. 4A, their structure is approximately that of a well-connected elastic material surrounding a marginally connected network near the active site. If a stimulus is imposed



**Fig. 3.** (A) Map of the mean coordination number and (B) spatial distribution of the average response magnitude for configurations equilibrated at  $T_e = 0.30$  (Left) and  $T_e = 0.05$  (Right). In the functioning networks (Right), a trumpet connecting the allosteric and active sites appears in A, and the response to stimulus varies nonmonotonically inside the trumpet in B.







**Fig. 5.** (A) Spatial distribution of conservation, as defined in Eq. 2, for  $T_e = 0.05$  and  $z = 5.0$ . (B) Spectrum of eigenvalues  $\rho(\lambda)$  of  $C$  for the high-temperature case ( $T_e = 0.30$ ) in black and the low-temperature case ( $T_e = 0.05$ ) in red. The white region indicates which eigenvalues are used to identify the springs shown in C. (C) Springs selected using the procedure explained in *Conservation and Coevolution*. (D)  $\tilde{C}$  is built using the same parameters as in C.  $\tilde{C}$  presents a clear separation in a region where the correlations are stronger, which corresponds to the trumpet shown in C. All these images are made using  $L = 12$  and  $z = 5$ .

For completeness, we define a correlation matrix reconstructed from its 10 top eigenvectors,

$$\tilde{C}_{\alpha\beta} = \sum_{\gamma=1}^{N_T} \lambda_{\gamma} |\psi^{\gamma}\rangle \langle \psi^{\gamma}|. \quad [4]$$

$\tilde{C}$  is shown in Fig. 5D after reordering links in terms of the strength of their components in the top 10 modes, clearly showing a sector of links where correlations are strong in amplitude (but vary in sign).

### Conclusion

We have introduced a scheme to discover materials that accomplish a specified task. The scheme allows us to characterize the architecture of the solutions, their entropy, and how correlated they are. We illustrated this approach using a specific allosteric task, where a strain imposed on an allosteric site must lead to a given strain on a distant active site. The architectures we obtain are highly anisotropic. Our analysis revealed that the physical mechanisms that enable allostery include the recently discovered presence of soft edge modes in marginally connected elastic materials (19). It would be very interesting to test if some proteins have evolved to exploit such effects by (i) measuring the displacement field induced by binding the ligand, and by checking if its magnitude varies nonmonotonically inside the protein and is amplified near the active site, as reported in Fig. 3B; and (ii) testing if the region around the active site is marginally connected, using algorithms developed in proteins to quantify connectivity and flexibility (33).

The detailed study of the architectures we found also reveals some of their limitations. Real proteins have additional constraints other than those we have considered: Among others, they are made of a chain that folds and remains relatively stable despite thermal noise. Our asymmetric structures are quite soft near the active site: As documented in *SI Appendix*, the thermally induced motion would be about 4 times larger there than in the other nodes also located at the system surface, which may not be desirable. Note, however, that such features will improve if alternative fitness functions are considered, which our approach

allows for. This procedure could be implemented by explicitly penalizing thermal motion at the active site. An intriguing extension of our work is to reason in terms of energy, as is done, in particular, in the Monod–Wyman–Changeux model (1, 2), instead of displacement. A natural quantity to maximize is the cooperativity between two distant sites. Denoting by  $E_1$  and  $E_2$  the mechanical energies associated with a binding event in some site 1 and site 2, respectively, and denoting  $E_{12}$  the energy of binding both, we can consider  $F = E_1 + E_2 - E_{12}$ . Fitness can be large only if the two sites are strongly coupled together elastically, which, from the symmetry of the fitness, presumably corresponds to more symmetric architectures than those discovered here.

In addition, our scheme can be used to benchmark (and seek novel) methods aiming at inferring function and structure from protein sequence data alone. As an illustration, we performed a statistical coupling analysis and identified “protein sectors” from top eigenvectors of the correlation matrix. Our results support that such a method (4, 7) can indeed reveal compact regions central for function. It would be very interesting to extend this analysis to multiple and possibly time-varying tasks, and to consider how many sectors appear under these conditions. Another approach, direct coevolutionary couplings (35), identifies evolutionary interactions between amino acids. Our framework can be used to measure these interactions, and to study their relationship with elasticity, structure, and function.

### Methods

**Computing the Linear Response to an Imposed Displacement.** The linear response to an external force field  $|\mathbf{F}\rangle$  reads

$$|\mathbf{F}\rangle = \mathcal{M}|\delta\mathbf{R}\rangle, \quad [5]$$

where the stiffness matrix  $\mathcal{M}$  depends only on connection  $|\sigma\rangle$  and the link directions. Here  $|\delta\mathbf{R}\rangle$  is the entire displacement field, of dimension  $2N$ . Its components are noted  $\delta R_i$ . We use a basis for which  $i = 1 \dots 4$  corresponds to the stimulus nodes, whereas  $i = 5 \dots N$  labels the other nodes.

To impose the stimulus at the allosteric nodes  $|\delta\mathbf{R}^{\mathcal{E}}\rangle$ , forces must be applied on these nodes. All other nodes adapt to a new mechanical equilibrium with no net forces on them, and they follow a displacement  $|\delta\mathbf{R}(\sigma)^{\mathcal{F}}\rangle$ . Thus, Eq. 5 becomes, for this choice of basis,

$$\begin{pmatrix} |\mathbf{F}^{\mathcal{E}}\rangle \\ |\mathbf{0}\rangle \end{pmatrix} = \mathcal{M} \begin{pmatrix} |\delta\mathbf{R}^{\mathcal{E}}\rangle \\ |\delta\mathbf{R}(\sigma)^{\mathcal{F}}\rangle \end{pmatrix}, \quad [6]$$

which leads to

$$\begin{pmatrix} |\mathbf{F}^{\mathcal{E}}\rangle \\ |\delta\mathbf{R}(\sigma)^{\mathcal{F}}\rangle \end{pmatrix} = \mathcal{Q}^{-1} \mathcal{M} \begin{pmatrix} |\delta\mathbf{R}^{\mathcal{E}}\rangle \\ |\mathbf{0}\rangle \end{pmatrix}, \quad [7]$$

with

$$\mathcal{Q}_{ij} = \begin{cases} \delta_{ij} & \text{if } j \in \mathcal{E} \\ -\mathcal{M}_{ij} & \text{if } j \notin \mathcal{E} \end{cases}. \quad [8]$$

When there are floppy modes in the network,  $\mathcal{Q}$  is not necessarily invertible. Then, in the above formula,  $\mathcal{Q}^{-1}$  should be understood as the pseudoinverse. Another possibility is to regularize the problem, for example, by imposing that each node also interacts with all its next-nearest neighbors via weak springs of stiffness  $k_w \ll 1$ . Both methods lead to qualitatively identical results. Our results were computed using the second approach with the very small value  $k_w = 10^{-4}$ .

**Computing the Fitness.** Minimizing Eq. 1 with respect to the global translation and rotation leads to

$$\sum_{i \in \mathcal{T}} (\delta\mathbf{R}_i^{\mathcal{F}} - \delta\mathbf{R}_i^{\mathcal{T}})^2 - \sum_{i \in \mathcal{T}} (\delta\mathbf{R}_i^{\mathcal{F}} - \delta\mathbf{R}_i^{\mathcal{T}})(\delta\mathbf{R}_{i+1}^{\mathcal{F}} - \delta\mathbf{R}_{i+1}^{\mathcal{T}}), \quad [9]$$

where  $i + 1 = \min(\mathcal{T})$  if  $i = \max(\mathcal{T})$ .

**Metropolis Algorithm.** Starting from a configuration  $|\sigma\rangle$ , we consider the move toward a new configuration  $|\sigma'\rangle$  that differs only by the motion of a spring. The move is accepted with the probability

$$P(|\sigma\rangle \rightarrow |\sigma'\rangle) = \min \left[ 1, \exp \left( \frac{F(|\sigma'\rangle) - F(|\sigma\rangle)}{T_e} \right) \right], \quad [10]$$

which satisfies detailed balance. Based on whether the move is accepted or not, the algorithm samples the new configuration  $|\sigma'\rangle$  or the original one  $|\sigma\rangle$ . The next step starts from the configuration just sampled.

For each coordination number  $z$  and each evolution temperature  $T_e$ , we sample 20 Monte Carlo sampling series with  $10^5$  Monte Carlo steps in each, and do not consider the first half of these time series (which is sufficient to eliminate transient effects). Our results are thus averaged over  $10^6$  configurations.

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