

# Cell contraction induces long-ranged stress stiffening in the extracellular matrix

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Animal cells in tissues are supported by biopolymer matrices, which typically exhibit highly nonlinear mechanical properties. While the linear elasticity of the matrix can significantly impact cell mechanics and functionality, it remains largely unknown how cells, in turn, affect the nonlinear mechanics of their surrounding matrix. Here, we show that living contractile cells are able to generate a massive stiffness gradient in three distinct 3D extracellular matrix model systems: collagen, fibrin, and Matrigel. We decipher this remarkable behavior by introducing nonlinear stress inference microscopy (NSIM), a technique to infer stress fields in a 3D matrix from nonlinear microrheology measurements with optical tweezers. Using NSIM and simulations, we reveal large long-ranged cell-generated stresses capable of buckling filaments in the matrix. These stresses give rise to the large spatial extent of the observed cell-induced matrix stiffness gradient, which can provide a mechanism for mechanical communication between cells.

biopolymer networks | microrheology | nonlinear elasticity | cell-matrix interactions | cell mechanics

Living cells interact mechanically with their 3D microenvironment. Many basic cell functions, including migration, proliferation, gene expression, and differentiation, depend on how these forces deform and shape the surrounding soft extracellular matrix (ECM) (1–4). In addition, externally imposed forces on the matrix can impact cell behavior, for instance in beating cardiac cells on a 2D substrate (5–7). Such external forces may be generated by other cells and act as mechanical signals (8–10) leading to emergent collective cell dynamics (11, 12). Nevertheless, it remains unclear how cell-generated forces propagate through the ECM and impact the mechanics of their 3D extracellular environment.

The ECM is composed of several types of biopolymers (13), such as collagen or fibrin, which are largely responsible for its mechanical properties. Experiment and theory have shown that biopolymer networks exhibit a highly nonlinear mechanical response (14), involving the entropic elasticity of individual filaments, geometric effects due to fiber bending and buckling, and even collective network effects governed by critical phenomena (15–21). Recent works have indicated that this nonlinear response is highly relevant to cell–ECM interactions (22–25). Although these nonlinear mechanical properties of biopolymer gels have been studied extensively with bulk rheology, the direct characterization of microscale mechanics inside a 3D matrix in the vicinity of a cell is still lacking. Consequently, the role of elastic nonlinearities in mechanical cell–ECM interactions has remained elusive.

Ideally, cell–ECM interactions would be analyzed by determining the stress field generated by the cell. Unfortunately, standard microscopy techniques do not reveal this information in a straightforward and unambiguous way. Some information about internal network forces can be accessed by adding deformable particles (26) or by creating an interface, for example by laser ablation, and observing the resulting deformation of the system (27, 28). However, obtaining internal stresses with such invasive and destructive approaches requires additional assumptions about the network's local mechanical properties. The same is true of approaches that infer stresses from a combination of microscopy imaging and finite element modeling (23, 29, 30). The intrinsic heterogeneity (31–33) and a highly nonlinear mechanical response (14, 34) of extracellular networks pose a daunting challenge to these techniques (35).

To investigate how living cells mechanically modify their microenvironment, we use microrheology with optical tweezers to directly measure the local nonlinear elastic properties in a 3D ECM network. We observe that remarkably far-reaching stiffening gradients are generated toward the cell in a variety of biopolymer matrices. To investigate this, we introduce a model-independent measurement technique termed nonlinear stress inference microscopy (NSIM), enabling us to determine the stress in a region around the cell and study stress propagation inside a 3D ECM. We use a combination of theory and simulations to demonstrate the ability of NSIM to accurately measure 3D local stress with high spatial resolution. Using NSIM, we show that the observed extended stiffness gradient around cells results from remarkably large stresses, which are capable of exciting the matrix's nonlinear response over distances exceeding the size of the cell. Our results demonstrate that contractile cells strongly modify the mechanics of the surrounding ECM, which could be crucial in shaping matrix-mediated interactions between cells.

# Cells Strongly Stiffen Their Surrounding ECM by Actively Contracting

To study the mechanical interactions between cells and their surrounding ECM, we culture MDA-MB-231 cells in a 1.5 mg/mL reconstituted 3D collagen network. The network is infused with

## Significance

The behavior of cells is strongly affected by the mechanics of their surroundings. In tissues, cells interact with the extracellular matrix, a 3D network of biopolymers with a highly nonlinear elastic response. We introduce a method exploiting this matrix nonlinearity to infer mechanical stresses in 3D. Using this method, we demonstrate that cell contractility induces large stresses, which generate a massive stiffness gradient over an extended region in 3D matrices of collagen, fibrin, and Matrigel. Our work highlights the importance of nonlinear matrix mechanics at the microscopic scale and suggests a concrete mechanism through which cells can control their microenvironment and mechanically communicate with each other.

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4.5-µm-diameter latex beads, large enough to prevent slippage through the mesh. The cells spread and start contracting the surrounding network within 4 h (Fig. 1*A*). We probe the local micromechanics of the matrix using optical tweezers to pull these beads away from the cell at a constant speed of 1 µm/s (Fig. 1 *B–D* and *SI Appendix*, Fig. S1). This low speed ensures that the viscous drag on the bead from the background fluid is negligible compared with the network's restoring force, and at this speed the mechanical response of the matrix is rate independent, fully reversible, and therefore predominately elastic (*SI Appendix*, Fig. S2). Thus, this protocol enables us to obtain the local force–displacement relationship F(x) that characterizes the micromechanics of the matrix.

By probing a bead located far from the cell (>200  $\mu$ m), we determine the intrinsic response of the collagen matrix. The resulting force–displacement relationship is shown by the black line in Fig. 1*E*. The nonlinear differential stiffness  $k_{nl}(F) = dF/dx$ , defined as the slope of this force–displacement curve, increases with applied force *F*, revealing a strong force-stiffening behavior. This is reminiscent of the well-characterized stress-stiffening behavior measured at large scales using macrorheology on collagen gels (14, 34).

Interestingly, the matrix becomes substantially stiffer closer to the MDA-MB-231 cell (Fig. 1*E*). Indeed, the local linear stiffness  $k_{lin}$  of the matrix, defined as the small force limit of  $k_{nl}(F)$ , is two orders of magnitude larger near the cell than at a remote location (Fig. 1*F*, red squares). This direct observation of cellinduced matrix stiffening in the bulk of the network is consistent with prior 2D experiments showing cell-induced stiffening of the surface of a collagen matrix with a cell migrating on top (22), as well as with simulations (23).

This dramatic stiffness gradient in the vicinity of the cell originates from the active forces it exerts together with the nonlinear elasticity of the matrix. To demonstrate this, we first note that we can rule out the effect of the passive rigidity of the cell on the matrix stiffness, which is proven theoretically to be very shortranged in 3D (33). Next, we measure the stiffness gradient around MCF-10A cells, a normal human mammary epithelial cell type with weak contractility. In this case, we observe negligible stiffening of the surrounding matrix (Fig. 1*F*, light blue polygons), in stark contrast to their highly contractile counterpart. Furthermore, inhibiting cell contractility of MDA-MB-231 cells using 2 µM cytochalasin D results in a strong attenuation of the cell-induced stiffening (Fig. 1F, blue circles). The weak residual stiffness gradient we observe with weakly contractile cells is well explained by increased ECM density near the cell, under the assumption that the matrix rigidity scales as the square of the collagen concentration c (36); by estimating c using confocal reflection microscopy, we determine that the enhanced matrix density near the cell can account for a stiffening of up to a factor of  $\sim 3$  (Fig. 1F, gray diamonds). However, the enhanced matrix concentration near the cell clearly cannot account for the much larger stiffness gradient generated by contractile MDA-MB-231 in collagen (Fig. 1F). Macrorheology experiments show that the stiffness of collagen gels increases not only with collagen concentration but also with stress (34). To test whether this nonlinear matrix response can account for the large stiffness gradient induced by the cell, we measure the local matrix stiffness in the vicinity of a MDA-MB-231 cell in a linear elastic matrix (37) with a similar macrorheological linear modulus to that of collagen (RGD-alginate, 5 mg/mL). In this linear elastic matrix, we observe no local stiffening effect around contracting cells (SI Appendix, Fig. S5). Taken together, our results demonstrate that active forces exerted by the cell result in an extended stiffened region in the 3D collagen matrix, reflecting the presence of a stress field decaying away from the cell with stress values sufficiently large to excite the nonlinear response of the collagen network, as illustrated in Fig. 1D.

#### **Nonlinear Stress Inference Microscopy**

To study the cell-induced stress fields, we use the network's nonlinear microrheological response to our advantage and infer local stress values from our stiffness measurements. The nonlinear stiffening evidenced in Fig. 1*E* originates from two contributions: the force *F* exerted by the optical tweezers acting on the bead, and the local stress  $\sigma_{loc}$  induced by the cell. This similar influence of force and stress suggests that we may be able to extract  $\sigma_{loc}$  at a specific distance from the cell by comparing the corresponding force–displacement relationship to the remote measurement at which  $\sigma_{loc}$  is negligible. This comparison is confounded, however, because of force and stress being fundamentally different quantities:



Fig. 1. Far-reaching stiffness gradient of ECM caused by a single contracting cell in a 3D collagen network. (A) Image of a MDA-MB-231 cell (blue) in a 3D collagen network (green). (Scale bar, 10 µm.) (B-D) Schematics illustrating the force-displacement measurement with laser tweezers and the relation between matrix stiffening (blue potential wells) and the cell-generated stress field in the cell contraction direction. (E) Local force-displacement curves, showing the local nonlinear stiffening response in the collagen network. Different colors represent measurements at various distances from the cell along the contraction direction. (F) Quantification of the linear stiffness  $k_{lin}$  of the local 3D matrix as a function of distance to the cell r. Red squares and yellow triangles represent measurements along and perpendicular to the main contraction direction of MDA-MB-231 cells, respectively. Blue circles are measured along the contraction direction of MDA-MB-231 cells but with cell contraction inhibited by cytochalasin D treatment. Gray diamonds indicate the stiffness expected solely from the increased collagen concentration c. Light blue polygons represent measurements in the contraction direction of MCF-10A cells. Here, "remote" stands for the locations that are far away from the cell (>200 μm), where the matrix's response is not affected by cell contraction. Error bars represent SD (n = 15).

beyond having different dimensions, force transforms as an axial vector under spatial symmetry operations, while stress is a rank 2 tensor. This has an essential implication for the difference in the nonlinear response due to a force as opposed to a stress: The local stiffness should be invariant under reversal of the force vector, F to -F, while reversal of the stress tensor  $\sigma$  to  $-\sigma$  exchanges compression and tension, which can have a qualitatively different effect on the nonlinear mechanical response.

Despite these differences, here we show that a correspondence between force and stress controlled stiffening can be established in the strongly nonlinear regime. First, consider a simple 1D system of nonlinear springs representing the network surrounding a bead in a geometry with fixed network stress  $\sigma$  (Fig. 24) and one with fixed tweezer force F (Fig. 2B). For nonlinear springs that stiffen under tension and soften under compression-a generic characteristic of biopolymers (14, 18, 38)-we find that the functional form of the stiffness curves actually becomes similar at large  $\sigma_{loc}$ and F, despite being qualitatively different in the weakly nonlinear regime. Indeed, the tensed spring in Fig. 2B dominates the differential stiffness experienced by the bead in the strongly nonlinear regime, rendering this case similar to the stress-controlled geometry, where the mechanical response is equally shared by two similarly tensed bonds (Fig. 2A). This quantitative similarity be-tween the  $k_{\text{lin}}$  vs.  $\sigma_{\text{loc}}$  and  $k_{\text{nl}}$  vs. F curves in the strongly nonlinear regime enables us to use the latter, which we measure by nonlinear microrheology, as a "dictionary" to infer local stresses.

This intuitive correspondence between the force- and stresscontrolled geometries in the nonlinear regime becomes mathematically exact when the springs' differential stiffness has a power-law dependence on tension, as widely observed for biopolymer networks (34, 39) (*SI Appendix*, section 4). Specifically, from a measurement of  $k_{\text{lin}}$  in a network with an unknown local stress  $\sigma_{\text{loc}}$ , we can obtain an effective force  $F_{\text{eff}}$  defined such that  $k_{\text{nl}}(F_{\text{eff}}) = k_{\text{lin}}(\sigma_{\text{loc}})$ , and this effective force is directly proportional to the local stress:

$$\sigma_{\rm loc} \approx a \ F_{\rm eff},\tag{1}$$

provided large local stresses, such that  $k_{\text{lin}} \gg k_0$ , where  $k_0$  is the linear stiffness of the unstressed network. We determine the proportionality factor *a* by assuming that nonlinearity sets in at a similar stress  $\sigma^*$  at a macro and microscopic level. In practice, we adjust *a* to match the low- and high-stress asymptotes, in a log-log plot, of the macroscopic differential shear modulus *K* ( $\sigma_{\text{macro}}$ ) to those of the microrheology curve  $k_{\text{nl}}(F)$  (Fig. 2 *C*-*E* and *SI Appendix*, Figs. S7 and S13). Together with Eq. 1, this provides a procedure to infer stresses from nonlinear microrheology, which we term nonlinear stress inference microscopy (NSIM).

To demonstrate the validity and accuracy of NSIM, we perform simulations of the experimental scenario presented in Fig. 1. We embed a contractile cell in a disordered 3D network of fibers with power-law stiffening (SI Appendix, section 2). We model the cell as a rigid ellipsoidal body that contracts along its long axis, inducing strong stiffening in an extended conic region as depicted in Fig. 3.4. We then simulate a microrheology experiment by applying a force on a selection of network nodes to obtain a local forcedisplacement curves at various distances r along the contraction direction of the cell (Fig. 3B). From this, we determine the linear stiffness  $k_{lin}$  of the network as a function of r (Fig. 3C), which exhibits a dependence similar to the experimental measurements shown in Fig. 1F. We further confirm that this dependence vanishes in the direction perpendicular to contraction and in the absence of an active contractile force, as in experiments (Fig. 1F). We infer the local stress field from these linear stiffnesses using NSIM, as shown in Fig. 3D. We find excellent agreement with the "true" local stress in the strong stiffening regime even when mechanical disorder gives rise to fluctuations in the stress field (Fig. 3E and SI Appendix, section 3), thereby validating NSIM as a quantitative method to capture the spatial stress distribution around a contractile cell in a disordered 3D fiber matrix (Fig. 3 D-F).



**Fig. 2.** Nonlinear elastic responses can be used to infer cell-induced local stresses. (*A*) One-dimensional system of nonlinear springs in a stress-controlled geometry with local stress  $\sigma_{loc}$ . (*B*) Force-controlled geometry with force *F* applied to the central bead, together with an expansion of stiffness dictated by symmetry properties of the two scenarios and a schematic of the nonlinear response. The linear stiffness,  $k_{linv}$  of the system in *A* can be measured by applying a small perturbation to the central bead, while the nonlinear stiffness,  $k_{nik}$  is defined as the derivative of the force–displacement curve of the central bead in *B*. The springs represent the surrounding network. (*C*) Schematics of linear micro-rheological stiffness as a function of the local stress in the stress-controlled geometry on a logarithmic scale. (*D*) Nonlinear microrheological stiffness for the force-controlled geometry. (*E*) Differential shear modulus, *K*, as a function of applied shear stress  $\sigma_{macro}$  as in a macroscopic rheology experiment. Our inference technique exploits a correspondence between the stress-controlled and force-controlled geometries in the strongly nonlinear regime.

#### Tensile Stress Propagation Leads to Extended Stiffness Gradients Around Cells

To unravel the mechanical origins of the far-reaching cellinduced stiffness gradient in collagen (Figs. 1F and 4A), we use NSIM to experimentally infer the local stresses  $\sigma_{loc}(r)$  around a cell inside the matrix. The inferred stress decays with distance rfrom the cell consistent with a power law  $\sigma_{loc} \sim r^{-2}$  (Fig. 4B), in contrast with the power law  $\sigma_{loc} \sim r^{-3}$  expected in the far field for a linear material (40). Our model-independent measurement of slow stress decay is consistent with previous theoretical predictions for models of fiber networks with various nonlinear force-extension relationships (41-44), as well as with the observed deviations from linear elasticity in experimental deformation fields (23, 29). Importantly, however, in this specific geometry where an elongated cell exerts opposite forces at two distant points (5, 23, 29), near-field linear elasticity also predicts an inverse quadratic stress decay at distances smaller than the cell diameter, rendering it difficult to distinguish linear and nonlinear force transmission. Nevertheless, our simulations predict that this long-ranged decay extends further than the cell size, showing a clear deviation from the linear elastic prediction (Fig. 3F).

Conceptually, this increased range of stresses in fibrous materials found in simulation results from their asymmetric response to tension and compression: Fibers stiffen under tension and soften due to buckling under compression (18, 45). Simply speaking, the matrix around a strong contractile cell effectively behaves as a network of ropes, where only tensile forces are transmitted,



**Fig. 3.** Three-dimensional simulations of cell-generated stress fields inducing nonlinear network response and validations of NSIM. (*A*) Simulated rigid ellipsoidal cell contracting inside a 3D nonlinear fiber network (in green). The linear stiffness  $k_{lin}$  is depicted by the spheres in a green–white logarithmic color gradient. (*B*) Local force–displacement curves, showing the local nonlinear stiffning response in the simulated network. Different colors represent measurements at various distances from the cell along the contraction direction. (*C*) Local linear stiffness  $k_{lin}$  of the 3D matrix as a function of distance to the cell *r* from simulations. Red and yellow symbols represent data parallel and perpendicular to the main contraction direction, respectively. Blue symbols correspond to a noncontracting rigid cell. (*D*) The inferred stress depicted by spheres in a red–white logarithmic color gradient in the same simulation as in *A*. Absent points along the direction perpendicular to the cell's contraction axis correspond to soft compressed regions where the local stiffness is smaller than  $k_0$ , precluding the use of NSIM. (*E*) Inferred stresses from simulated data in *B* using NSIM vs. direct numerically determined stress, demonstrating that NSIM allows to correctly infer stresses within a factor of order 1 in the nonlinear regime. (*F*) The local stress along the cell axis decays as  $r^{-2}$ , faster than the linear elastic prediction in this geometry (dotted line).

unimpeded by orthoradial compressive counterforces. Hence the total contractile force exerted by the cell is conserved with distance, and the decay of radial stress simply reflects this force spreading over an increasing surface area (41). This buckling-based mechanism for long-range stress transmission is supported by observations with confocal reflection microscopy of a larger amount of highly curved collagen filaments in the vicinity of a contractile cell, compared with the case where contraction is inhibited with cytochalasin D (Fig. 4 E and F).

To explore the generality of our observations in collagen, we perform the same nonlinear microrheological experiments with MDA-MB-231 cells in a 2.5 mg/mL Matrigel (Fig. 4A, green circles), a blend of biopolymers more complex than pure collagen (46), and for human umbilical vein endothelial cells (HUVECs) in a 3.0 mg/mL fibrin gel (light blue triangles in Fig. 4A). In both cases, we find that cells are capable of generating large extended stiffness gradients along the cell's contraction direction (Fig. 4A). Using NSIM, we find that the slow stress decay consistent with rope-like force transmission ( $\sigma_{loc} \sim r^{-2}$ ) is observed in all three cases, despite significant variability in the absolute magnitude of the stresses (solid lines in Fig. 4B). The stresses induced by the cell enhance the linear stiffness  $k_{lin}$  over an extended region of the ECM (Figs. 1F and 4A), thus exciting the nonlinear elastic response of the matrix over a distance exceeding the cell size. These results also highlight the wide applicability of NSIM.

Cells actively modify not only the linear stiffness of their 3D matrix environment (Fig. 4A) but also the nonlinear mechanical response. To reveal how cell stress and probe forces combine to stiffen the surrounding network, we measure the full nonlinear microrheological response of the network both in the vicinity of the cell and at a remote location in all three types of ECM model systems, as shown in Fig. 4C. The nonlinear  $k_{nl}$  vs. F curves measured at different distances from the cell are clearly separated from the remote measurement. This observation cannot be accounted for by network heterogeneities (*SI Appendix*, Fig. S3) or the increase of network concentration near the cell (*SI Appendix*, Fig. S4), indicating a significant contribution of cell-generated stress on the nonlinear mechanical response. This contribution could be through nonlinear network elastic stiffening or network plastic deformation (47). We note that our

stress inference is largely independent of the specifics of the ECM's nonlinear response but does assume a predominantly elastic response to the forces generated by the cell. Indeed, significant plastic deformations could imply that the ECM's nonlinear response is systematically modified as a function of the distance from the cell. In the absence of plastic deformations, we expect that further stiffening a prestressed matrix by a large tweezer force would result in a nonlinear response that is functionally similar for all levels of cell stress. To test this, we plot all nonlinear stiffening curves as a function of the combined force  $F + F_{\rm eff}$ , where the effective force  $F_{\rm eff} \propto \sigma_{\rm loc}$  is determined as in Fig. 2A and B (SI Appendix, Table S1). Remarkably, we find that the data taken at different distances to the cell collapse in any network composition onto a smooth master curve (Fig. 4D). The large cell-generated stress thus locally drives the ECM into an elastic nonlinear regime, which can be further extended by the probe force we apply with optical tweezers.

Several studies have reported experimentally measured displacement fields induced by a contracting cell in 3D contexts (23, 29, 30, 35, 48, 49). By using these displacement fields together with a continuum elasticity model, it was suggested that the matrix may stiffen near the cell. However, to our knowledge, no direct measurements of the local stiffness have been done in 3D contexts, at the scale of a cell level. Furthermore, we infer the stresses responsible for this stiffening using NSIM, a conceptually unique inference technique that does not require knowledge of the materials' constitutive stress-strain relationship nor of a reference undeformed state. Due to its simplicity and insensitivity to the detailed material's properties, NSIM could be used in various conditions, including embryo or tumor development. The stresses inferred using this technique far from the cell are consistent with prior measurements (29). Close to the cell, strong stiffening renders the technique most accurate and corresponds to stresses of the order of 200 Pa, larger than previously reported (29). These cell-induced stresses decay more slowly than in a linear continuum material, which can be accounted for by buckling of fibers in the network, impeding the transmission of compressive stresses. This slow stress decay has also been inferred in previous studies by using a finite-element model in conjunction with imaged deformation fields (23, 29). Here, we provide direct evidence for



**Fig. 4.** Nonlinear matrix stiffening and cell-generated stress propagation in various 3D biopolymer networks. (*A*) Local linear stiffness  $k_{lin}$  is plotted against the distance to the cell *r* along its principal contraction direction in collagen (red square), fibrin (blue triangle), and Matrigel (green circle). All three different ECM model systems exhibit a strong cell-induced stiffening gradient. (*B*) The stress field  $\sigma$  generated by the cell determined using NSIM is shown as a function of distance to the cell *r*, and the dashed line indicates a slope of -2. (*C*) Local nonlinear differential stiffness  $k_{nl}$  is plotted against the applied probe force *F* for all three ECM model systems. (*D*) Collapse of the data from *C* onto a master curve in each respective matrix obtained by plotting  $k_{nl}$  as a function of combined local force  $F + F_{eff}$ , where the  $F_{eff}$  is determined using NSIM. (*E*) Time-lapse imaging shows the buckling process of a single fiber around a contracting cell. The fiber curvature distributions (*Bottom*) and the cumulative probability (*Top*) near the cell, within a 60-µm distance along the principal cell contraction direction, before and after cytochalasin D treatment. Error bars in *A* and *B* represent SD (n = 15).

long-range stress transmission by using a model-independent measurement of local stresses and their decay around a cell. These cell-induced stresses result in far-reaching stiffness gradients as high as 50 Pa/µm over a cell diameter. Other cells in the surrounding matrix could sense and respond to such large gradients, suggesting that cell-induced ECM stiffening could mediate intercell mechanical communication and collective durotaxis. These observations highlight the critical role of nonlinear matrix mechanics not only in shaping cell–ECM interactions (8, 50) but also for matrix-mediated interactions between cells.

### Methods

**Cell Culture and Matrix Preparation.** Cells are maintained under 37 °C, 5% CO<sub>2</sub> and 95% humidity. MDA-MB-231 cells were cultured in DMEM with 10% FBS, 1% penicillin and streptomycin. HUVECs (Lonza) were cultured on collagen l-coated flasks in EGM-2 growth medium (Lonza) and used between passages 6 and 8. To prepare the collagen gel, 800  $\mu$ L of type I bovine collagen solution (3.0 mg/mL; PureCol; Advanced BioMatrix) was mixed with 100  $\mu$ L of PBS (10×). We adjusted the solution to pH 7.2 with ~70  $\mu$ L of 0.1 M NaOH. The collagen solution is then mixed with PBS (1×) to reach a final collagen concentration of 1.5 mg/mL and polymerized in the cell culture incubator for 30 min. To prepare the fibrin gel, fibrinogen from bovine plasma (F8630; Sigma) was dissolved in PBS at 6 mg/mL. Thrombin (T4648; Sigma) was dissolved at 2 U/mL in PBS (for experiments without cells) or in EGM-2 (for experiments with cells). Then we mixed thrombin and fibrinogen at 1:1 volume ratio and polymerized it in the cell culture incubator for 15 min. For Matrigel preparation, the basement

membrane matrix (10 mg/mL; Corning) was diluted to 2.5 mg/mL with DMEM and polymerized in the cell culture incubator for 30 min. For all cell-loaded gels, cell and bead suspensions were added to the gel solution before polymerization, with a cell density around  $10^4$ /mL, and all measurements were conducted 12 h after polymerization. To inhibit contractility of MDA-MB-231 cells, we disrupted filamentous actin structures using 2  $\mu$ M cytochalasin D (PHZ1063; Invitrogen) for 30 min.

**Optical Tweezer Measurements.** We used a Thorlabs optical tweezers system to perform all measurements. Briefly, to optically trap a bead (4.5- $\mu$ m carbox-ylate microspheres; Polyscience) that is embedded and confined in a 3D biopolymer network, the laser beam (5 W, 1,064 nm) is tightly focused through a series of Keplerian beam expanders and a high-N.A. objective (100 × 1.4; oil; Leica). A high-resolution quadrant detector was used for position detection. The linear region of the detector and the trap stiffness (0.04 pN/nm) were calibrated with the same bead in pure cell culture media by using an active power-spectrum method and equipartition theorem (51). To manipulate the trapped bead, a high-resolution piezo stage (P-545; PI nano) was moved at a constant velocity of 1  $\mu$ m/s, and the relative distance between laser and bead was recorded, from which local force-displacement curves inside the matrix were determined (52) (see *SI Appendix* for details).

**Bulk Rheology.** We performed bulk rheology measurements on a DHR-3 rheometer (TA Instruments) using a plate–plate geometry, with a 40-mm glass disk as the top plate and a 60-mm Petri dish as the bottom plate with a gap of 500  $\mu$ m. All gels were formed in the gap at 37 °C and were sealed by mineral oil to avoid evaporation. The polymerization process was monitored by strain oscillations with a strain amplitude of 0.005 at a frequency of 1 rad/s. After polymerization, a strain ramp was applied to the gel at a rate of 0.01/s, and the resulting stresses were measured.

**Theoretical Modeling and Simulations.** Numerical simulations presented in Fig. 3 are performed using a model of nonlinear springs [force–extension relation  $f(x) = \exp(\mu x) - 1$ ; see *SI Appendix*, Figs. S10 and S11 for other types of nonlinearities], with regular removal of springs to introduce disorder in the network, while ensuring a fiber length  $L_f = 10$ , in a spherical system of radius R = 50.5. The contractile cell is a rigid ellipsoidal body of size  $14.2 \times 2.8 \times 2.8$ , with force and torque balance, contracted by 50% along its long axis. The surrounding network is flexibly clamped at the surface of the cell and at the boundary of the system. Mechanical equilibrium is attained by minimization of the energy using the BFGS algorithm. Further details are provided in *SI Appendix*, sections 2 and 3.

**Imaging of Collagen Networks and Image Analysis.** The 3D collagen networks near a contracting cell were imaged with confocal reflection microscopy using a 63×, 1.2 N.A. water objective (Leica SP8). To determine the boundary of the cell, the cytoplasm was stained with CellTracker Green (C7025; Thermo Fisher) and imaged at the same time under confocal microscope. To capture the fiber buckling process, we imaged the cell and its surrounding 3D fiber

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networks at a 5-min interval for 4 h at 37 °C and with 5% CO<sub>2</sub>. To analyze the curvature of single collagen fiber, we manually selected 20 points on each individual collagen fiber; the fiber outline was determined by cubic spline interpolation, from which the average curvature of the fiber was calculated.

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