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# Temperature-Dependent HfO<sub>2</sub>/Si Interface Structural Evolution and its Mechanism



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#### **Abstract**

In this work, hafnium oxide ( $HfO_2$ ) thin films are deposited on p-type Si substrates by remote plasma atomic layer deposition on p-type Si at 250 °C, followed by a rapid thermal annealing in nitrogen. Effect of post-annealing temperature on the crystallization of  $HfO_2$  films and  $HfO_2$ /Si interfaces is investigated. The crystallization of the  $HfO_2$  films and  $HfO_2$ /Si interface is studied by field emission transmission electron microscopy, X-ray photoelectron spectroscopy, X-ray diffraction, and atomic force microscopy. The experimental results show that during annealing, the oxygen diffuse from  $HfO_2$  to Si interface. For annealing temperature below 400 °C, the  $HfO_2$  film and interfacial layer are amorphous, and the latter consists of  $HfO_2$  and silicon dioxide ( $SiO_2$ ). At annealing temperature of 450-550 °C, the  $HfO_2$  film become multiphase polycrystalline, and a crystalline  $SiO_2$  is found at the interface. Finally, at annealing temperature beyond 550 °C, the  $HfO_2$  film is dominated by single-phase polycrystalline, and the interfacial layer is completely transformed to crystalline  $SiO_2$ .

Keywords: Hafnium oxide, Atomic layer deposition, Interface, Annealing, Crystallization

#### Introduction

Hafnium oxide (HfO<sub>2</sub>) thin film is an interesting material for a variety of applications. It can be used in multilayer optical coating [1], protective coating [2], gate dielectric [3], passivating layer [4–6], and so on due to its excellent properties, such as high density, high refractive index, wide band gap, and relatively high thermal stability. Many methods have been used to prepare HfO<sub>2</sub> thin film, such as electron beam evaporation [7], chemical solution deposition [8], reactive sputtering [9], metal organic chemical vapor deposition [10], molecular beam epitaxy [11], and atomic layer deposition (ALD). ALD is a promising method for obtaining thin films with both high-precision thickness control and high accuracy uniformity. Post-annealing is found to have significant influences on ALD HfO<sub>2</sub> films [12–15]. According to the research, HfO<sub>2</sub> thin films can crystalize for an annealing temperature higher than 500 °C [16-18]. The crystalline structure of HfO2 strongly affects optical and electrical

In this work, the HfO<sub>2</sub> thin films were fabricated by a remote plasma atomic layer deposition (RP-ALD) on p-type silicon substrates. Post-annealing was performed by a rapid thermal annealing (RTA) system at different temperatures. The structural changes and crystallization properties of HfO<sub>2</sub> thin films by RTA were characterized by atomic force microscopy (AFM), grazing incident X-ray diffraction (GIXRD), X-ray photoelectron spectroscopy

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properties. For example, the structural change of HfO<sub>2</sub> from amorphous to monoclinic crystalline phase could lead to changes of refractive index from 1.7 to 2.09, optical gap from 5.75 to 6.13 eV, and dielectric constant from 24.5 to 14.49 [19, 20]. For ALD HfO<sub>2</sub> deposited on silicon substrates, an oxide layer is usually observed at HfO2/Si interface [21, 22]. The presence of this interfacial layer is reported to decrease the dielectric constant [22]. In addition, Kopani et al. [23] presented the structural properties of 5-nm HfO2 films after nitric acid oxidation of n-doped Si substrates. They found that high annealing temperature increases the growth rate of crystalline nuclei. However, their crystallization properties particularly HfO<sub>2</sub>/ substrate interface have scantly been studied. Therefore, the annealing temperature affecting the crystallization properties of HfO2 thin films prepared by ALD was worth for further investigation.

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(XPS), and high-resolution transmission electron microscopy (HR-TEM). The temperature-dependent  $HfO_2/Si$  interface structural evolution and its mechanism are also investigated.

#### Method

Doubled-sided polished (100) oriented p-type 2-inch 250-µm Czochralski Si wafers with a resistivity of 30  $\Omega$  cm were used. Prior to the deposition, Si wafers were cleaned by a standard Radio Corporation of America method followed by dipping in diluted hydrofluoric acid solution (5%) for 2 min to remove possible stray oxides without final water rinse. After cleaning, all of the wafers were dried with pure nitrogen (N2) gas and mounted onto the substrate holder. Approximately 15 nm HfO<sub>2</sub> (168 ALD cycles) thin films were deposited on Si wafers by RP-ALD (Picosun R-200, Finland) using tetrakis (ethylmethylamino) hafnium (TEMAH) and oxygen (O<sub>2</sub>) in alternating pulse with N<sub>2</sub> purge of the reaction chamber between pulses. The TEMAH and O<sub>2</sub> plasma were pulsed into the reactor in the following sequence: TEMAH pulse 1.6 s; N2 purge 10 s; O<sub>2</sub> plasma pulse 10 s, and N<sub>2</sub> purge 12 s. After depositing the HfO2 thin films, the rapid thermal annealing was performed in N<sub>2</sub> ambient for 10 min. The annealing temperatures were varied from 400 to 600 °C to investigate the effect on crystallization of the HfO2 thin films and HfO<sub>2</sub>/Si interface. Table 1 lists the typical conditions of RPALD and post-annealing.

AFM measurements were performed in tapping mode for investigating the surface morphology of the  $HfO_2$  thin films. The AFM images shown in this work are  $2 \, \mu m \times 2 \, \mu m$  scans with a resolution of 256 points  $\times$  256 lines. The structure of  $HfO_2$  films were characterized by grazing incident X-ray diffraction (GIXRD, Rigaku TTRAXIII, Japan) measurements with a Cu long-fine-focus X-ray tube. X-rays with a wavelength of 0.154 nm were produced at an operating voltage of 50 kV and a current of 300 mA. An incident angle of 0.5° was selected to obtain diffraction

Table 1 RPALD HfO<sub>2</sub> deposition parameters

RPALD- HfO <sub>2</sub> thin film	
Parameter	Value
Substrate temperature (°C)	250
TEMAH pulse time (s)	1.6
O <sub>2</sub> plasma pulse time (s)	10
O <sub>2</sub> plasma power (W)	2500
Thickness (nm)	15
RTA-post annealing process	
Parameter	Value
Temperature (°C)	400-600
Time (min)	20
Ambient	N <sub>2</sub>

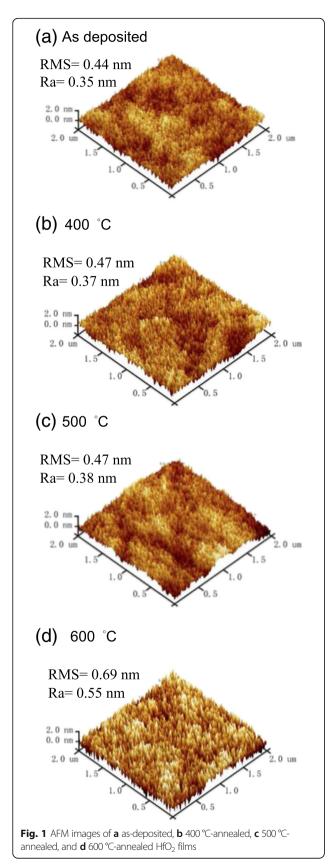
patterns over a  $2\theta$  range of 20– $60^\circ$ . X-ray photoelectron spectroscopy (XPS, Thermo Fisher K-alpha) was also performed using monochromatic Al K $\alpha$  X-ray radiation (hv = 1486.6 eV). For the XPS analysis, a 100- $\mu$ m diameter spot was used, and photoelectrons were collected at a take-off angle of 45°. The cross sections of the HfO<sub>2</sub> thin films were prepared by a focused ion beam lift-out technique in a Hitachi NX2OOO system. The cross-sectional images of the HfO<sub>2</sub> thin films were examined by a field emission high-resolution transmission electron microscopy (HR-TEM, JEM-2100F, USA).

#### **Results and Discussion**

Figure 1 shows the AFM images for the  $HfO_2$  films annealed at different temperatures. The root-mean-square (RMS) and average surface roughness (Ra) values are shown for indicating the surface roughness. The RMS value is 0.44 nm for the as-deposited film. It slightly increases to 0.47 nm when the annealing temperature rises to 500 °C. Further increasing the annealing temperature to 600 °C leads to a significant enhancement in surface roughness with a RMS increasing to 0.69 nm. Same tendency is observed in Ra values. The increase in surface roughness for the annealed films might infer a structural change.

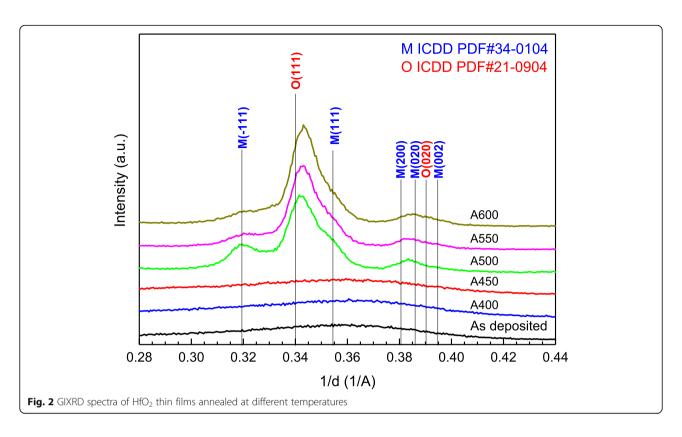
Figure 2 shows the temperature-dependent GIXRD spectra of various HfO<sub>2</sub> thin films. The as-deposited HfO<sub>2</sub> films is amorphous and remains amorphous after annealing at 400 and 450 °C. At an annealing temperature higher than 500 °C, diffraction peaks appear, indicating the formation of crystalline  $HfO_2$ . The peaks at 1/d = 0.319 and  $0.354 \,\mathrm{\AA^{-1}}$  correspond to the -111 and 111 planes to the monoclinic phase (ICDD PDF#34-0104, space group P21/ c), respectively. The peak at  $1/d = 0.340 \text{ Å}^{-1}$  corresponds to the (111) plane of the orthorhombic phase (ICDD PDF#21-0904, space group Pbcm). Other peaks near 1/d  $= 0.380 \sim 0.395$  are the 200, 020, and 002 planes of the monoclinic and the 020 plane of the orthorhombic phases. The results also reveal that the monoclinic phase decrease and the orthorhombic phases increase with the annealing temperature. The orthorhombic HfO<sub>2</sub> dominates the crystalline structure at higher annealing temperatures. However, the diffraction peaks of orthorhombic HfO2 were observed at a lower 1/d (a smaller d-spacing) as compared to that in the ICDD PDF#21-0904. In addition, the shift of  $1/d = 0.340 \,\text{Å}^{-1}$  towards a higher value indicates that the d-spacing decreases with the annealing temperature.

The concentrations of Hf and O within the  $HfO_2$  films were measured using depth profiled XPS. Figure 3 shows the O/Hf composition ratio of the as-deposited and post-annealed  $HfO_2$  films. The O/Hf ratio decreases from 1.60 to 1.29 with the annealing temperature. Due to the use of  $N_2$  during the annealing, the  $HfO_2$  becomes oxygen deficient with the temperature. The oxygen



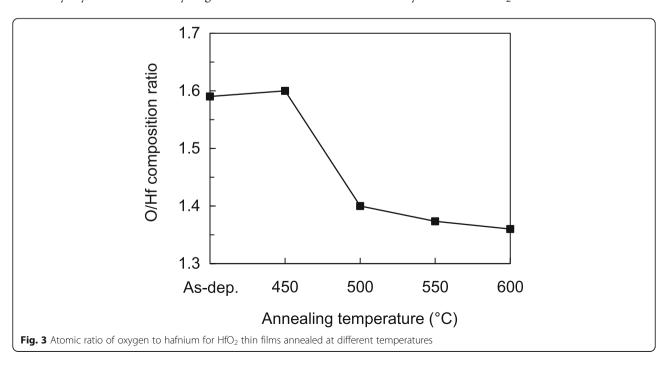
deficient HfO<sub>2</sub> film also results a smaller d-spacing as mentioned previously.

Figure 4a, b, c, d, e, and f show the high-resolution cross-sectional HR-TEM images of as-deposited 400 °C-, 450 °C-, 500 °C-, 550 °C-, and 600 °C-annealed HfO<sub>2</sub> thin films on Si substrates, respectively. It can be seen that the HfO<sub>2</sub> layer and Si substrate are clearly exhibited in these images. Additionally, a thin layer with the thickness of 1-2 nm between HfO2 and Si substrate could be the SiO<sub>2</sub> film. As shown in Fig. 4a, there is no obvious lattice arrangement in the as-deposited HfO2 film, indicating that this film is amorphous. After annealing at 400 °C, although most regions of HfO<sub>2</sub> film are still amorphous, we can observe that a fraction of lattice arrangements with the d-spacing values of 2.82 and 3.12 Å are formed in this film. These two d-spacing values are indexed to monoclinic HfO<sub>2</sub> (111) and monoclinic HfO<sub>2</sub> (-111) planes, respectively, and the 400 °C-annealed film shows the nanocrystalline structure. With increasing the annealing temperature from 400 to 600 °C, the crystal quality of HfO2 film is gradually enhanced. When the HfO<sub>2</sub> film is annealed at 500–550 °C, the main lattice arrangements consisting of monoclinic HfO<sub>2</sub> (-111), monoclinic HfO<sub>2</sub> (200), and orthorhombic HfO<sub>2</sub> (111) can be identified. However, further increasing the annealing temperature to 600 °C, the lattice structure of orthorhombic HfO<sub>2</sub> (111) still exists in the film, and the other two lattice arrangements gradually disappear. On the other hand, the d-spacing values of orthorhombic  $HfO_2$  (111) planes for the 500 °C-, 550 °C- and 600 ° C-annealed HfO2 films are determined to be 2.93, 2.90, and 2.88 Å, respectively. This agrees well with the XRD result that the orthorhombic HfO<sub>2</sub> (111) diffraction peak shifts towards to the high angle direction with increasing the annealing temperature from 500 to 600 °C. The result reveals that the oxygen content of HfO2 film reduces gradually as the annealing temperature is increased. The other interesting phenomenon can be found in the changes of crystal structure and thickness of the SiO<sub>2</sub> layer. At the as-deposited state, the SiO<sub>2</sub> layer is amorphous. Even if the sample is annealed at 400 °C, the thermal energy is not high enough to transform the structure of SiO2 layer from amorphous to crystalline. Nevertheless, by increasing the annealing temperature from 450 to 600 °C, the crystalline SiO<sub>2</sub> layer (with the cubic SiO<sub>2</sub> (220) structure) is formed and its thickness increases from 1.0 to 1.6 nm. It can be observed that the amorphous SiO<sub>2</sub> layer completely transforms to cubic SiO<sub>2</sub> structure after annealing the sample at 600 °C. With an increment of annealing temperature from 550 to 600 °C, the d-spacing value of cubic SiO<sub>2</sub> (220) increases from 2.48 to 2.56 Å. This means that the oxygen content of SiO<sub>2</sub> layer increases by increasing the annealing temperature. It can be reasonably speculated

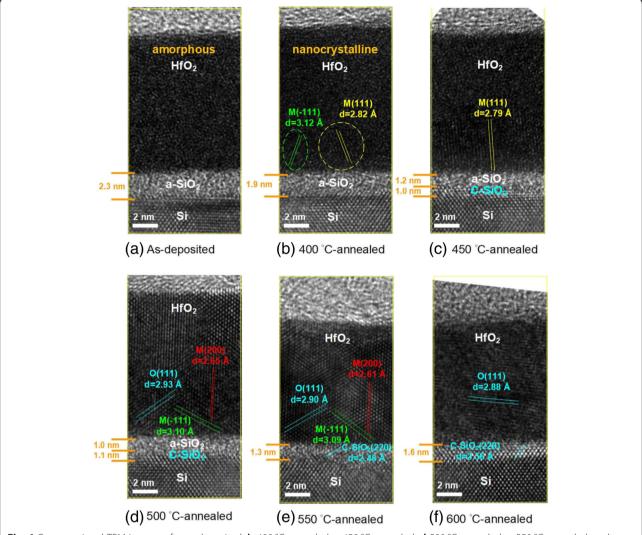


that the addition of oxygen content in the  $\mathrm{SiO}_2$  layer is attributed to the diffusion of oxygen atoms sourced from the  $\mathrm{HfO}_2$  film. Moreover, the overall thickness decreases for the annealing temperature of 550 and 600 °C and might be related to the increase of the film density caused by crystallization and hydrogen removal.

Based on the above results, Fig. 5 illustrates the mechanisms of the  $HfO_2$  films with different annealing temperatures. Considering the annealing temperature is smaller than 400 °C (Fig. 5a), the film is amorphous where Hf and O atoms are randomly arranged. The interfacial layer between  $HfO_2$  and c-Si wafer is a mixed

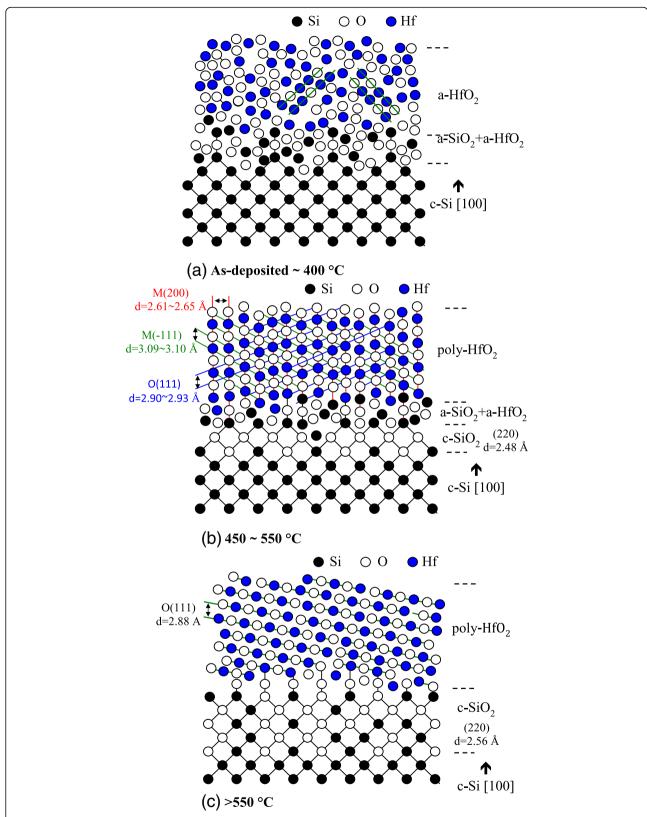


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**Fig. 4** Cross-sectional TEM images of **a** as-deposited, **b** 400 °C-annealed, **c** 450 °C-annealed, **d** 500 °C-annealed, **e** 550 °C-annealed, and **f** 600 °C-annealed HfO₂/Si

oxide consisting of a-SiO<sub>2</sub> and a-HfO<sub>2</sub>. At an annealing temperature of 450-550 °C (Fig. 5b), the HfO<sub>2</sub> film receives thermal energy leading to a structural change from amorphous to polycrystalline with monoclinic and orthorhombic phases. The crystalline orientation and d-spacing are indicated according to the HR-TEM and GIXRD results. A crystalline SiO<sub>2</sub> layer is formed. Several works reported an ordered silicon oxide layer at the interface of a-SiO<sub>2</sub> and (100) c-Si, but the mechanism and atomic-scale structure have remained controversial. Silicon thermal oxidation could be regarded as sequential inserting operations of oxygen atoms into Si-Si bonds, and this induces a large accumulation of compressive strains in the oxidized regions and might possibly cause a structural transformation into ordered oxide at the  $SiO_2/c$ -Si interface [24]. It has also been reported that crystalline oxygen-containing phase could be formed under conditions of high oxygen oversaturation of Si [25] or low interface defect density [26]. From the XPS and TEM images in this work, the HfO2 layer is oxygen deficient. The significant amounts of oxygen diffuse from HfO2 towards silicon substrate, and this might lead to oversaturation of oxygen at the c-Si interface and formation of crystalline SiO<sub>2</sub>. In this temperature range, the crystalline SiO<sub>2</sub> layer thickness would increase but the a-HfO<sub>2</sub> + a-SiO<sub>2</sub> mixed layer thickness decreases with increasing annealing temperature. At an annealing temperature higher than 550 °C (Fig. 5c), the HfO<sub>2</sub> structure is dominated by polycrystalline orthorhombic (111) single phase. The interfacial layer is entirely governed by crystalline SiO<sub>2</sub>. The d-spacing decreases for orthorhombic HfO2 layer and increases for c-SiO2. Although annealing of HfO2 is necessary for achieving high Si wafer passivation and dielectric constant, at high temperatures,



**Fig. 5** Diagrams of mechanism of crystallization of  $HfO_2$  films and interfacial layer in the temperature ranges **a** as-deposited to 400 °C, **b** 450 to 550 °C, and **c** beyond 550 °C. The d-spacing value and crystalline orientation are also indicated

the resultant crystallization of the  $HfO_2$  and the interfacial  $SiO_2$  may reduce the film properties. The annealing temperature of  $500\,^{\circ}\mathrm{C}$  is found to obtain the best dielectric constant of 17.2. Further increasing the annealing temperature leads to a reduction in dielectric constant, possibly due to the change in the crystalline phase. Tomida et al. reported that the dielectric constant of  $HfO_2$  decreases when the structure transformed from polycrystalline to monoclinic single phase [27]. The best passivation of  $HfO_2/Si$  can also be obtained at the annealing temperature of  $500\,^{\circ}\mathrm{C}$ , as higher temperatures might lead to a complete  $c\textsc{-}SiO_2$  interfacial layer and dehydrogenation at the interface.

#### **Conclusion**

HfO2 films are prepared using RP-ALD, and effect of annealing temperature on crystalline structure of the HfO<sub>2</sub> has been investigated. For as-deposited HfO2 and that annealed below 400 °C, the HfO2 and the interfacial layer are amorphous. With increasing annealing temperature, the d-spacing of orthorhombic reduces while that of the c-SiO<sub>2</sub> interfacial layer increases, indicating the oxygen diffusion from HfO2 to Si interface. Annealing temperature higher than 550 °C shows a HfO2 layer with polycrystalline orthorhombic single-phase, and the interfacial layer completely transforms to c-SiO<sub>2</sub>. Although annealing is required for HfO2 in many applications such as achieving high passivation of Si wafers and high dielectric constant, the crystallization could be harmful to the film properties. The annealing temperature of 500 °C can have the best Si wafer passivation quality and dielectric constant.

#### Abbreviations

AFM: Atomic force microscopy; a-HfO $_2$ : Amorphous hafnium oxide; ALD: Atomic layer deposition; a-SiO $_2$ : Amorphous silicon dioxide; c-SiO $_2$ : Crystalline silicon dioxide; GIXRD: Grazing incident X-ray diffraction; HfO $_2$ : Hafnium oxide; HR-TEM: High-resolution transmission electron microscopy; N $_2$ : Nitrogen; O $_2$ : Oxygen; RMS: Root-mean-square; RP-ALD: Remote plasma atomic layer deposition; RTA: Rapid thermal annealing; TEMAH: Tetrakis (ethylmethylamino) hafnium; XPS: X-ray photoelectron spectroscopy

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#### Availability of Data and Materials

All data supporting the conclusions of this article are included within the article.

#### Authors' contributions

XYZ carried out the characterization of the  $HfO_2$  thin films and drafted the manuscript. CHH, WYW, SLO, and SYL led the experimental and analytical effort. SYC, WH, WZZ, FBX, and SZ contributed to the valuable discussion on

experimental and theoretical results. All authors read and approved the final manuscript.

#### Competing interests

The authors declare that they have no competing interests.

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