



# Solar Paint from  $TiO<sub>2</sub>$  Particles Supported Quantum Dots for Photoanodes in Quantum Dot−Sensitized Solar Cells

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**S** [Supporting Information](#page-5-0)

ABSTRACT: The preparation of quantum dot (QD)−sensitized photoanodes, especially the deposition of QDs on  $TiO<sub>2</sub>$  matrix, is usually a time-extensive and performance-determinant step in the construction of QD-sensitized solar cells (QDSCs). Herein, a transformative approach for immobilizing QD on the  $TiO<sub>2</sub>$ matrix was developed by simply mixing the as-prepared oil-soluble QDs with  $TiO<sub>2</sub>$ P25 particles suspension for a period as short as half a minute. The solar paint was prepared by adding the  $TiO<sub>2</sub>/QD$  composite in a binder solution under ultrasonication. The QD-sensitized photoanodes were then obtained by simply brushing the solar paint on a fluorine-doped tin oxide substrate followed by a lowtemperature annealing at ambient atmosphere. Sandwich-structured complete QDSCs were assembled with the use of  $Cu<sub>2</sub>S$ /brass as counter electrode andpolysulfide redox couple as an electrolyte. The photovoltaic performance of the resulting Zn−Cu−In−Se (ZCISe) QDSCs was evaluated after primary optimization of the  $QD/TiO<sub>2</sub>$  ratio as well as the thicknesses of photoanode



films. In this proof of concept with a simple solar paint approach for photoanode films, an average power conversion efficiency of 4.13%  $(J_{sc} = 11.11 \text{ mA/cm}^2$ ,  $V_{oc} = 0.590 \text{ V}$ , fill factor = 0.631) was obtained under standard irradiation condition. This facile solar paint approach offers a simple and convenient approach for QD-sensitized photoanodes in the construction of QDSCs.

# ■ INTRODUCTION

The development of high-efficiency and low-cost photovoltaic cells is an effective way to solve the increasing concerns on global warming and the exhaustion of fossil fuels.<sup>[1,2](#page-6-0)</sup> Quantum dot−sensitized solar cells (QDSCs) are considered as one of the promising third-generation solar cells due to the excellent optoelectronic properties of QD light absorbers, such as high absorption coefficient, tunable light-harvesting range, high stability, and low-cost availability.<sup>[2](#page-6-0)−[6](#page-6-0)</sup> Furthermore, due to the multiexciton generation possibility, the theoretical maximum power conversion efficiency (PCE) of QDSCs can go beyond the Shockley−Queisser limit and reach 44%.[7,8](#page-6-0) The structure of QDSCs is similar to that of dye-sensitized solar cells with a light-harvesting material-sensitized photoanode, electrolyte, and a counter electrode. In the photoanode, light-harvesting material QDs are immobilized on nanocrystallite metal oxide (mainly  $TiO<sub>2</sub>$ ) mesoporous films. The mesoporous film electrode plays a crucial role in determining the performance of a cell device, such as providing enormous surface areas for supporting enough amounts of QD light absorbers, collecting and transmitting photogenerated electrons, and allowing redox couple electrolyte penetrating into the interior of the film to regenerate QD sensitizers.<sup>[9](#page-6-0)</sup> Usually, a series of time-consuming procedures are required to get access to a well-performing photoanode. First, crystallite  $TiO<sub>2</sub>$  mesoporous film electrodes are prepared by coating  $TiO<sub>2</sub>$  paste on an optically transparent

electrode (herein, fluorine-doped tin oxide (FTO) conducting glass) substrate and following a high-temperature sintering process.<sup>[10,11](#page-6-0)</sup> Then, QD sensitizers are immobilized on the film electrode. This QD deposition step is considered as a crucial and challenging step in determining the photovoltaic perform-ance of the resultant cell device.<sup>[12](#page-6-0)−[14](#page-6-0)</sup>

Currently, two major routes, i.e., in situ growth, and postsynthesis assembly with use of preprepared QDs, have been developed for the immobilization of QDs on a mesoporous film electrode.<sup>[13](#page-6-0)−[27](#page-6-0)</sup> For the first route, QDs are formed in situ on a mesoporous  $TiO<sub>2</sub>$  film electrode starting from a solution containing the corresponding cationic and anionic precursors for  $QDs$ .<sup>[15](#page-6-0)−[20](#page-6-0)</sup> For the second one, the preprepared QDs are tethered on the film electrode via direct adsorption, molecular linker-, or electrophoresis-assisted assembly process.<sup>[21](#page-6-0)−[27](#page-6-0)</sup> Benefiting from the feasible penetration of ionic precursors into the nanoscaled channels of the mesoporous film electrode, in situ growth route can result in an effective and high coverage of QD on the electrode surface.<sup>[28](#page-6-0)</sup> However, the drawback of this route is the high density of trap state defects and heterogeneous size distribution of the formed QDs. These disadvantages limit the photovoltaic

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performance of the resulting cell devices. In contrast, the postsynthesis assembly route is favored by the high quality of the preprepared QDs, but is shadowed by the low uploading amount of QDs due to the relatively large size of QD particles.<sup>[29](#page-6-0)−[32](#page-6-0)</sup> The deposition of QD sensitizers on the mesoporous film electrodes has thus being a bottleneck in the development of high-efficiency QDSCs.

To construct versatile and low-cost QDSCs, it is necessary to simplify the currently delicate and time-intensive preparation procedure for photoanodes, especially the QD sensitization procedure. Unambiguously, the invention that QD-sensitized photoanodes can be attained by simply brushing a solar paint on a conductive substrate is a transformative step in manufacturing QDSCs. This pioneering concept was tried by Hodes and co-workers.<sup>[33](#page-7-0)</sup> They developed a solar paint using CdSe, CdTe powder with a fluxing material such as  $CdCl<sub>2</sub>$ , but the corresponding photovoltaic performance was not obtained. Recently, Kamat and co-workers reported solar paints for photoanode by simply mixing the commercially available  $TiO<sub>2</sub>$ P25 powder with bulk CdS powder in water and t-butanol mixture solvent or depositing CdS (or CdSe) nanoparticles on the suspended P25 particles by adsorption of  $Cd^{2+}$  and  $S^{2-}$  (or Se<sup>2−</sup>) ions alternatively. The obtained highest PCE was 1.08% in their work. $34$  During the preparation of this article, Bang and co-workers increased the highest PCE to 1.35% through this solar paint route with use of PbS QDs as the light-harvesting material.<sup>[35](#page-7-0)</sup> However, all of these reported solar paint approaches to fabricate the photoanode applied the successive ion layer absorption and reaction method to deposit QDs on the  $TiO<sub>2</sub>$  substrate. As discussed above, this in situ growth method usually resulted in the high density of trap state defects and heterogeneous size distribution of the formed QDs, seriously deteriorating photovoltaic performance of the cell devices. Thus, preparing solar paint with use of presynthesized high-quality QDs owns a great potential to obtain highefficiency QDSC based on solar paint.

Herein, we target at simplifying the preparation procedure of QD-sensitized photoanode by simply brushing a solar paint on the FTO substrate, followed by a low temperature annealing treatment at ambient atmosphere. The solar paint was made from  $TiO<sub>2</sub>/QD$  composites, which were obtained by direct adsorption of presynthesized oil-soluble QDs on the  $TiO<sub>2</sub>$  P25 suspension in a period as short as half a minute. Less-toxic and environmental-benign Zn−Cu−In−Se (ZCISe) QDs were selected as the model QD light absorber. An average PCE of 4.13% was achieved in the resulting QDSCs with use of  $Cu<sub>2</sub>S$ / brass counter electrode.

## ■ RESULTS AND DISCUSSION

Solar Paint and Photoanode Preparation. In QDSCs, the QD-sensitized photoanodes are normally prepared by tethering QD on TiO<sub>2</sub> nanocrystallite mesoporous films.<sup>[15](#page-6-0)−[27](#page-6-0)</sup> On one hand, the adopted mesoporous structured film can enormously enhance the surface face of the matrix for supporting much more QDs, and therefore improving the light-harvesting capability; on the other hand, the nanoscale channel throughout the film renders the QD deposition process difficult and time-consuming. To overcome the disadvantage of the conventional route in the preparation of photoanodes, herein QD sensitizers are immobilized on the suspending  $TiO<sub>2</sub>$ P25 particles to get the  $TiO<sub>2</sub>$  particles supported QD composite powder (noted as  $TiO_2/QD$  hereafter). Then, the obtained  $TiO<sub>2</sub>/QD$  powder is mixed with a binder solution to

get the so-called solar paint. From this solar paint, QDsensitized photoanodes could be feasibly obtained via a screen printing process. Scheme 1 illustrates the preparation of QDsensitized photoanodes with use of solar paint starting from the TiO<sub>2</sub>/QD composite powder.

Scheme 1. Schematic Procedure for the Preparation of QD-Sensitized Photoanodes



In the experiment, oleylamine (OAm) capped oil-soluble ZCISe QD light-harvesting materials with particle size of 4.1 nm and absorption onset of 980 nm are first synthesized according to literature method. $32,38$  $32,38$  $32,38$  The corresponding optical spectra are available in [Figure S1.](http://pubs.acs.org/doi/suppl/10.1021/acsomega.7b01761/suppl_file/ao7b01761_si_001.pdf) Then, a fixed amount of P25  $(0.5 \text{ g})$  suspension in CH<sub>2</sub>Cl<sub>2</sub> is mixed with ZCISe QD dispersion in  $CH_2Cl_2$  to get the TiO<sub>2</sub>/QD composite. To get an optimal loading amount, the amount of QD used is varied. Without the limitation of nanoscaled narrow channels, as encountered in previously commonly used mesoporous film,<sup>15−[27](#page-6-0)</sup> in our case, both QD and P25 particles are suspended in the solution; therefore, QDs could get access and tether on P25 particles from any direction. As a result, the deposition of QD on P25 could be completed in a period as less as 30 s. Meanwhile, a high loading amount of QD on the P25 particles can also be obtained as discussed below. It is highlighted that in this work the initial OAm-capped oil-soluble QDs are used directly in this sensitization process. Due to the chemical inertness of the long hydrocarbon chain in OAm ligand around QDs, the deposition of oil-soluble QD on the conventional  $TiO<sub>2</sub>$  film electrode is usually an extensively time-consuming process with the deposition time up to several even to tens of hours, as well as a low uploading amount.<sup>[39](#page-7-0)–[44](#page-7-0)</sup>

To avoid the degradation of ZCISe QDs at the following high-temperature annealing process, a recipe for binder solution borrowed from the paste suitable for conducting polymer substrate is adopted in the preparation of solar paint in our case.[36](#page-7-0) In the experiment, solar paint is obtained by adding  $TiO<sub>2</sub>/QD$  powder to a poly(vinylidene difluoride) (PVDF) binder solution under ultrasonication. The resulting thick, brown solar paint could then be screen printed directly on FTO to obtain QD-sensitized photoanodes after a low-temperature (120 °C) annealing process. TiO<sub>2</sub>/QD powder is obtained by direct adsorption of the initial oil-soluble QD on the P25 particles suspension. The solar paint is then prepared by mixing the obtained  $TiO<sub>2</sub>/QD$  powder with a PVDF binder solution. Finally, the photoanodes are obtained by screen printing the solar paint on the FTO substrate followed with a 120 °C annealing procedure at ambient atmosphere for 1 h. It is noted that this low annealing temperature cannot eliminate the PVDF completely, rendering the PVDF to remain in the photoanode film. The insulating feature of the PVDF in the intervals among neighboring TiO<sub>2</sub> particles hinders the photogenerated transportation among the  $TiO<sub>2</sub>$  particles and decrease the photovoltaic performance accordingly. Thus, the amount of PVDF in the solar paint is optimized, and it is found that the addition of 20 mg PVDF gives the best photovoltaic performance. The corresponding photovoltaic parameters of QDSCs prepared with different amounts of PVDF in the solar paint are shown in [Table S3](http://pubs.acs.org/doi/suppl/10.1021/acsomega.7b01761/suppl_file/ao7b01761_si_001.pdf). On the basis of these obtained photoanodes, the sandwich-structured complete cells are assembled with the use of  $Cu<sub>2</sub>S$ /brass counter electrode, andthe corresponding photovoltaic performances are evaluated.

Figure 1 shows the absorption spectra of ZCISe QDsensitized TiO<sub>2</sub> film electrodes derived from different amounts



Figure 1. UV-vis absorption spectra of TiO<sub>2</sub> film corresponding to different contents of ZCISe QDs in the  $TiO<sub>2</sub>$  suspension. Inset: photographs of corresponding film electrodes.

of ZCISe QDs used in the preparation of  $TiO<sub>2</sub>/QD$  composite. It is found that with the increase in the QD amount in the  $TiO<sub>2</sub>$ suspension, the absorbance in the resulting film electrode with identical thickness is improved accordingly to reach a saturation value at the QD amount of 75.0 mg. This indicates that the increase in QD loading on the  $TiO<sub>2</sub>$  particles is translated into the enhancement of loading amount on the final film electrode. For convenience, the used QD amount in the  $TiO<sub>2</sub>$  suspension is used to distinguish the loading amount in film electrodes hereafter. The increase in absorbance in the photoanode film indicates the enhancement of QD loading amount on the  $TiO<sub>2</sub>$ particles, and also the improvement of the light-harvesting capacity of the film electrode. The enhancement of QD loading amount of P25 can also be reflected by the deepening of the photograph color of the film electrode, as shown in the inset of

Figure 1. Similar to the case of the absorption spectra from the QD dispersion in  $CH_2Cl_2$  ([Figure S1](http://pubs.acs.org/doi/suppl/10.1021/acsomega.7b01761/suppl_file/ao7b01761_si_001.pdf)), the light absorption range for the resulting photoanodes covers the whole visible spectrum and expands to the near-infrared window with the longest wavelength extending to near 1000 nm. This excellent light-harvesting capacity paves the way for achieving an outstanding photovoltaic performance of the resulting cell devices as discussed below.

Figure 2a shows the transition electron microscopy (TEM) images of ZCISe QDs-sensitized  $TiO<sub>2</sub>$  film electrodes. From the TEM images, it can be seen that the surface of P25 particles (about 20−50 nm) is covered densely by smaller spherical particles (ZCISe QD, about 4 nm). This gives a visual evidence for the uniform distribution of QDs onto the P25 particles. Figure 2b shows the cross-sectional field emission scanning electron microscopy (SEM) images of ZCISe QDs-sensitized  $TiO<sub>2</sub>$  film electrode with the film thickness of 12  $\mu$ m. From the high-magnification SEM images in Figure 2c, it can be seen that the  $TiO<sub>2</sub>$  film electrodes are composed of numerous mesoporous, which allows the fluent diffusion of electrolyte solution.<sup>[45](#page-7-0)</sup>

Photovoltaic Performance. To suppress the charge recombination loss at the photoanode/electrolyte interfaces in assembled cell devices, a thin passivation layer of ZnS is overcoated around the surface of photoanodes, as done in a standard QDSC.<sup>[44,46](#page-7-0)</sup> The treated photoanodes are assembled into a sandwich-type QDSC device with the use of  $Cu<sub>2</sub>S/brass$ counter electrode and polysulfide redox electrolyte. To minimize the measurement deviation, at least five devices are prepared and tested in parallel. To obtain higher efficiency of QDSCs, we investigate systematically the influence of a series of experimental variables, including the amount of QDs used in  $TiO<sub>2</sub>$  suspension for the preparation of  $TiO<sub>2</sub>/QD$  composite, and the film thickness.

[Figure 3a](#page-3-0) shows the photocurrent−voltage (J−V) curves of cells under the irradiation of 1 full sun intensity based on different amounts of QDs in the preparation of  $TiO<sub>2</sub>/QD$ composite powder. The measured short-circuit current density  $J_{\rm sc}$  values (9.19–11.11 mA/cm<sup>2</sup>) are consistent with the integrated current density (8.95−10.84 mA/cm<sup>2</sup> ) based on incident photon-to-current efficiency (IPCE) spectra in [Figure](#page-3-0) [3](#page-3-0)b. Photovoltaic parameters  $(V_{\alpha\sigma} J_{\rm sc}$  fill factor (FF), and PCE) that are obtained from the J−V curves are listed in [Table 1.](#page-3-0) The individual photovoltaic parameters of each tested cell based on the solar paint containing different amounts of QDs are listed in [Table S1](http://pubs.acs.org/doi/suppl/10.1021/acsomega.7b01761/suppl_file/ao7b01761_si_001.pdf). When the amount of QDs in the  $TiO<sub>2</sub>$  suspension is increased from 45.0 to 75.0 mg, the  $V_{\text{od}}$   $J_{\text{sol}}$  and PCE values are increased systematically, whereas the FF values are kept nearly constant. With further increase in the QD amount, the



Figure 2. (a) TEM images of TiO<sub>2</sub> nanoparticles supported QDs. (b) Low-magnification and (c) high-magnification SEM images of the cross section of QD-sensitized  $TiO<sub>2</sub>$  film.

<span id="page-3-0"></span>

Figure 3. (a) J−V and (b) IPCE curves of QDSCs corresponding to different contents of ZCISe QDs in TiO2 suspension.

Table 1. Average Photovoltaic Performance and Standard Deviation Extracted from J−V Measurement Curves of QDSCs Corresponding to Different Contents of ZCISe QDs in the  $TiO<sub>2</sub>$  Suspension<sup>a</sup>

content $(mg)$	$V_{\rm oc}$ (V)	$J_{sc}$ (mA/cm <sup>2</sup> )	FF	PCE $(\% )$
45	0.576(0.578)	9.19(9.47)	0.632(0.625)	$3.34 \pm 0.09$ (3.42)
60	0.584(0.586)	10.56(10.88)	0.620(0.610)	$3.82 \pm 0.04$ (3.89)
75	0.590(0.599)	11.11(11.08)	0.631(0.634)	$4.13 \pm 0.07$ (4.21)
90	0.589(0.590)	11.06(11.58)	0.635(0.626)	$4.12 \pm 0.11$ (4.25)

<sup>a</sup>The numbers in parentheses represent the values obtained from the champion cells.



Figure 4. EIS characterization of QDSCs corresponding to different contents of ZCISe QDs in the TiO<sub>2</sub> suspension: (a) chemical capacitance C<sub>u</sub>; (b) recombination resistance R<sub>rec</sub>; (c) dark current on corrected voltage V<sub>app</sub>; (d) Nyquist plots at the forward bias of −0.6 V.

photovoltaic performance is kept at a plateau value. It is noted that the unabsorbed free QDs could be reused, and this accords with the concept of "green" chemistry. What is more, this method is demonstrated to be beneficial for the fabrication of larger-sized device. We have constructed the cells with a larger active area of  $1.0 \text{ cm}^2$  and an average PCE of  $3.83\%$  was obtained, which is comparable with that of the small-sized cells. The detailed photovoltaic parameters are shown in [Table S4.](http://pubs.acs.org/doi/suppl/10.1021/acsomega.7b01761/suppl_file/ao7b01761_si_001.pdf)

The relatively low efficiency obtained in this work can be attributed to the low  $J_{\rm sc}$  value. This work highlights a facile approach to prepare the QD-sensitized photoanode in QDSCs

using a solar paint and without a need of high-temperature treatment. Nevertheless, the thickness of the photoanode film is limited in this method, which will be discussed in the following sections. What is more, the residual PVDF in the film arising from the low annealing temperature for the preparation of film electrodes also leads to the relatively low  $J_{\rm sc}$  values.

Electrochemical Impedance Spectroscopy. To unveil the inherent mechanism for the dependence of photovoltaic performance on the amounts of QDs used in the preparation of  $TiO<sub>2</sub>/QD$  composite, electrochemical impedance spectroscopy (EIS) is employed. All of the cell samples are measured at

<span id="page-4-0"></span>

Figure 5. (a) OCVD curves of QDSCs corresponding to different contents of ZCISe QDs in the TiO<sub>2</sub> suspension. (b) Electron lifetime derived from OCVD measurements.



Figure 6. (a) J−V and (b) IPCE curves of QDSCs based on different thicknesses of photoanode film.

different forward biases (from −0.2 to −0.6 V) under darkness. The corresponding parameters of EIS are obtained relying on the standard equivalent circuit model.[47](#page-7-0),[48](#page-7-0) [Figure 4](#page-3-0)a−c shows the applied forward bias voltage  $(V_{\rm app})$  dependent chemical capacitance  $(C_u)$ , recombination resistance  $(R_{\text{rec}})$ , and dark current. The similar  $C_{\mu}$  values [\(Figure 4](#page-3-0)a) of all of the cells illustrate that different QD loading amounts did not change the state of the conduction band or the distribution of traps states of the  $TiO<sub>2</sub>$  substrate. However, with the increase in the loading amount, the  $R_{\text{rec}}$  values [\(Figure 4](#page-3-0)b) are improved gradually. It is well known that the charge recombination rate occurring at the photoanode/electrolyte interfaces is inversely proportional to the  $R_{\text{rec}}$  value.<sup>[49,50](#page-7-0)</sup> With the increase in the QD loading amount, a higher surface coverage for the  $TiO<sub>2</sub>$  matrix is achieved, and the surface areas exposed directly to the electrolyte solution is decreased.<sup>[9](#page-6-0)</sup> It is noted that the uncovered  $TiO<sub>2</sub>$  surface could act as a recombination center for the capture of photogenerated electron by the oxidized species (i.e.,  $S_n^2$ ) in the electrolyte[.51](#page-7-0),[52](#page-7-0) Therefore, the improvement in surface coverage favored the suppression of charge recombination process occurring at photoanode/electrolyte interfaces and brought forward the enhancement of photovoltaic performance of the resultant cell devices. Furthermore, the dependence of dark current on  $V_{app}$  as shown in [Figure 4c](#page-3-0) gives further support for the suppressed charge recombination rate with the increase in the QD loading amount. To distinguish the recombination difference of cells based on different QD loading amounts, [Figure 4d](#page-3-0) shows the Nyquist plots of the corresponding cells under the forward bias near the  $V_{\text{oc}}$  value. The observed greater diameter for the semicircles corresponding to larger QD loading amount indicates a greater charge recombination resistance. Overall, the EIS results indicate that the charge recombination occurring at  $TiO<sub>2</sub>/QDs/electrolyte interfaces is$  inhibited with an increase in the loading amount of QD on the  $TiO<sub>2</sub>$  particles. The EIS results are in well-accordance with the results from the J−V measurements as shown in [Table 1](#page-3-0) and [Figure 3.](#page-3-0)

Open-Circuit Voltage Decay. To illustrate further the performance difference of the cells based on different amounts of QDs in film electrodes, the open-circuit voltage decay (OCVD) measurement is used to analyze the electron recombination processes. A white light-emitting diode of 100  $mW/cm<sup>2</sup>$  continuously illuminates on the studied cell to get to a stable voltage, and then the open-circuit voltage decay information is recorded after switching off the light.[53,54](#page-7-0) As shown in Figure 5a, it can be seen that the rate of voltage decay becomes slower by increasing the amount of QDs. The electron lifetime  $(\tau_n)$  can be visually reflected by the open-circuit voltage decay curves. Figure 5b shows the electron lifetime, which can be calculated using the equation  $\tau_n = -(k_B T/e)(dV_{oc}/dt)^{-1}$ , , where  $k_B$ , T, and e are the Boltzmann constant, the absolute temperature  $(298 \text{ K})$ , and the electronic charge, respectively.<sup>[45](#page-7-0)</sup> From Figure 5, it is obvious that the solar cells based on a higher QD loading amount correspond to a longer electron lifetime related to those with a lesser QD loading amount at the identical potential, indicating that the charge recombination process is retarded by the increasing QD loading amount. This observation is consistent with that observed in the EIS measurement, and therefore gives further evidence for the obtained better photovoltaic performance as observed from the J−V measurements.

Effect of Film Thickness on Photovoltaic Performance. In addition to the different amounts of QDs in the solar paint, the mesoporous film thickness of a photoanode is another important parameter in determining the performance of the resulting cell devices. In a certain range, a thicker film can

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thickness $(\mu m)$	$V_{\alpha}$ (V)	$J_{sc}$ (mA/cm <sup>2</sup> )	FF	PCE $(\% )$
	0.598(0.601)	9.99(10.22)	0.615(0.614)	$3.67 \pm 0.08$ (3.77)
12	0.590(0.599)	11.11(11.08)	0.631(0.634)	$4.13 \pm 0.07$ (4.21)
15	0.585(0.589)	10.01(10.13)	0.639(0.640)	$3.74 \pm 0.06$ (3.82)
		<sup>a</sup> The numbers in parentheses represent the values obtained from the champion cells.		

<span id="page-5-0"></span>Table 2. Average Photovoltaic Parameters of QDSCs Based on Different Thicknesses of Photoanode Film<sup>a</sup>

enhance the light absorption capacity and thus improve the photocurrent, whereas it will increase the possibility of a charge recombination simultaneously.[55](#page-7-0) In the experiment for optimizing film thickness, the amounts of QDs and PVDF in the solar paint is kept at an optimum value. The J−V and IPCE curves for champion cells under variable film thickness are shown in [Figure 6.](#page-4-0) As shown in Table 2, the average photovoltaic performances ( $V_{\text{occ}}$  J<sub>sc</sub>, FF, and PCE) are improved with the increase in the film thickness from 9 to 12  $\mu$ m. The individual photovoltaic parameters of each tested cell with different film thickness are displayed in [Table S2](http://pubs.acs.org/doi/suppl/10.1021/acsomega.7b01761/suppl_file/ao7b01761_si_001.pdf). We can see that the improved PCE is mainly derived from the increased  $J_{\rm sc}$ . When the film thickness is increased furthermore from 12 to 15  $\mu$ m,  $V_{\rm oo}$  J<sub>sc</sub>, and PCE are decreased; however, the FF is increased. This demonstrates that the photovoltaic performance of the resulting cell device could not be improved by just monotonously increasing the thickness of the photoanode film. Furthermore, our experiment results indicate that the strength of the photoanode film is sacrificed with an increase in the thickness of the film, and the resulting thick film is easier to fall off from the FTO glass substrate. Our experimental results indicate that the thickness of 12  $\mu$ m is the optimized value in consideration of both photovoltaic performance and device stability.

## ■ **CONCLUSIONS**

In summary, a transformative approach for the preparation of QD-sensitized photoanode in QDSCs was developed by simply brushing a solar paint on the FTO conductive glass. The solar paint was derived from the  $TiO<sub>2</sub>/QD$  composite, which can be obtained in a time period as short as half minute by mixing the initial oil-soluble QDs and  $TiO<sub>2</sub>$  P25 suspension. Obviously, this approach can simplify the preparation process of QDsensitized photoanodes, especially the QD sensitization step, and, therefore, speed up the industrialization process of QDSCs. An average PCE of 4.13% ( $V_{\text{oc}} = 0.590$  V,  $J_{\text{sc}} =$ 11.11 mA/cm<sup>2</sup>,  $FF = 0.631$ ) was achieved on ZCISe QDSCs prepared through this solar paint approach. Further optimization of the solar paint composition and other components of the device are necessary to boost further the performance of the resultant cell devices.

## **EXPERIMENTAL SECTION**

Chemicals. Selenium powder (200 mesh, 99.99%), oleylamine (OAm, 95%), 1-octadecene (90%), and poly(vinylidene difluoride) (PVDF) were purchased from Aldrich. Diphenylphosphine (98%) and zinc acetate  $(Zn(OAc))$ , 99.99%) were obtained from J&K. Copper iodide (CuI, 99.998%) and indium acetate  $(In(OAc)<sub>3</sub>)$  were received from Alfa Aesar. P25 powder (20% rutile and 80% anatase) was obtained from Degussa. All of the chemicals were used as received without further treatment.

Preparation of Solar Paint and Fabrication of Solar Cells. The procedure and all of the chemicals used for the preparation of oil-soluble ZCISe QDs were identical to

previous report.<sup>[32](#page-6-0)</sup> The TiO<sub>2</sub>/QD composites were obtained by simply mixing the as-prepared oil-soluble QDs with  $TiO<sub>2</sub>$ P25 nanoparticles suspension in the dichloromethane medium. Typically, 10.0 mL of P25 suspension in dichloromethane (containing 0.5 g of P25) was mixed with an equal volume of as-prepared ZCISe QD dispersion in dichloromethane (containing 75.0 mg of ZCISe QDs), and the resulting mixture was subjected to sonication for 30 s to complete the process of adsorption of QDs on the surface of TiO<sub>2</sub> particles. The TiO<sub>2</sub>/ QDs powder was collected via a centrifugation/precipitation process for twice to get rid of the unattached free-dispersion QDs in solution. The powder product was then dried under hot air with typical yields of 0.571 g.

Binder solution for the following solar paint was prepared according to standard literature method.<sup>[36](#page-7-0)</sup> In the procedure for solar paint preparation, the obtained  $TiO<sub>2</sub>/QD$  powder (0.571) g), 1.0 mL as-prepared binder solution were loaded to a mixture containing 6.0 mL of anhydrous ethanol and 100  $\mu$ L of  $TiO<sub>2</sub>$ -sol in a 50 mL round-bottom flask and then sonicated for 10 min. The solar paint was obtained by the removal of redundant ethanol from this mixture through a rotary evaporation process.

The photoanode film were fabricated by screen-printing the obtained solar paint on a FTO glass  $(8 \Omega/sq)$  and then sintered at 120 °C for 60 min in a muffle-type furnace at ambient atmosphere. The sandwich-type cells were fabricated by assembling the  $Cu<sub>2</sub>S/brass$  counter electrode and the photoanode film using a scotch spacer (50  $\mu$ m). Modified polysulfide aqueous solution containing 2.0 M of S, 2.0 M Na<sub>2</sub>S, and 20 wt % poly(vinyl pyrrolidone) was used as redox electrolyte. $3<sup>3</sup>$ 

Characterization. The morphology of the photoanode film was studied by a Hitachi S4800 SEM. The TEM images were obtained on a JEOL JEM-2100 instrument with an operation voltage of 200 kV. The absorption spectra of the photoanode film were measured on a Shimadzu UV-2600 spectrometer. The Keithley 2000 multimeter with illumination from a 300 W tungsten lamp with a Spectral Product DK240 monochromator was used to measure IPCE curves. The J−V characteristics of the cell devices were recorded on a Keithley 2400 source meter under illumination of simulated AM 1.5G solar light (Oriel, model no. 94022A). A NREL standard silicon solar cell was used to calibrate the light intensity before each measurement. The photoactive area was determined by a black metal mask of  $0.237$  cm<sup>2</sup>. OCVD and EIS were obtained using an electrochemical workstation (Zahner, Zennium).

## ■ ASSOCIATED CONTENT

#### **S** Supporting Information

The Supporting Information is available free of charge on the [ACS Publications website](http://pubs.acs.org) at DOI: [10.1021/acsomega.7b01761.](http://pubs.acs.org/doi/abs/10.1021/acsomega.7b01761)

> Absorption spectra of oil-soluble ZCISe QD dispersions in dichloromethane; the individual photovoltaic parameters extracted from J−V measurement curves of QDSCs corresponding to different contents of ZCISe QDs in the

<span id="page-6-0"></span> $TiO<sub>2</sub>$  suspension; the individual photovoltaic parameters based on different thicknesses of photoanode film ([PDF\)](http://pubs.acs.org/doi/suppl/10.1021/acsomega.7b01761/suppl_file/ao7b01761_si_001.pdf)

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# Notes

The authors declare no competing financial interest.

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