

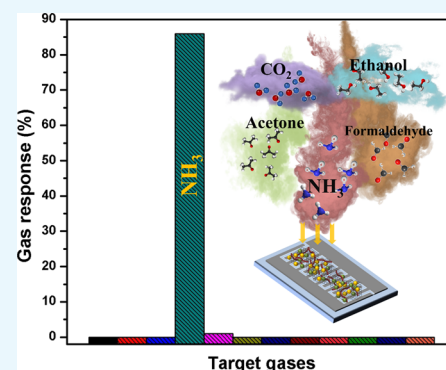
Ultrahigh Selective Room-Temperature Ammonia Gas Sensor Based on Tin–Titanium Dioxide/reduced Graphene/Carbon Nanotube Nanocomposites by the Solvothermal Method

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S Supporting Information

ABSTRACT: Resistive-based gas sensors have been considered as the most favorable gas sensors for detection of toxic gases and volatile organic compounds (VOCs) because of their simple structure, low cost, high sensitivity, ease of use, and high stability. Unfortunately, wide application of resistive-based gas sensors is limited by their low selectivity. In this article, we present the fabrication of ultrahigh selective NH₃ gas sensor based on tin–titanium dioxide/reduced graphene/carbon nanotube (Sn–TiO₂@rGO/CNT) nanocomposites. The Sn–TiO₂@rGO/CNT nanocomposites with different molar ratios of Sn/Ti (1:10, 3:10, and 5:10) were synthesized via the solvothermal method. Characterizations by scanning electron microscopy, transmission electron microscopy, and X-ray photoelectron spectroscopy confirmed the decoration of Sn–TiO₂ nanoparticles on rGO/CNT nanocomposite surfaces. The Sn–TiO₂@rGO/CNT nanocomposite gas sensor exhibited high response and ultrahigh selectivity to NH₃ against toluene, dimethylformamide, acetone, ethanol, methanol, isopropanol, formaldehyde, hydrogen, carbon dioxide, acetylene, and VOCs in paint thinners at room temperature. The Sn–TiO₂@rGO/CNT nanocomposite gas sensor with molar ratio of Sn/Ti = 1:10 showed the highest response to NH₃ over other molar ratios of Sn/Ti as well as pure rGO/CNT and Sn–TiO₂ gas sensors. The ammonia-sensing mechanisms of the Sn–TiO₂@rGO/CNT gas sensor were proposed based on the formation of p–n heterojunctions of p-type rGO/CNT and n-type Sn–TiO₂ nanoparticles via a low-temperature oxidizing reaction process.



1. INTRODUCTION

Ammonia (NH₃) is one of the toxic gases that can cause illness or death when it is inhaled or absorbed by eyes, nose, skin, and the respiratory tract at high concentration.¹ NH₃ has been widely used in several commercial products and industrial applications such as ice factory, adhesives, rubber cements, automotive fuels, and laboratory solvents.^{2–4} Moreover, NH₃ is also found in the exhaled breath of humans, in which it plays an important biomarker for diagnosis of kidney disorders or ulcers caused by the *Helicobacter pylori* bacterial stomach infection.^{5,6} The exhaled breath of patients with kidney disorders and peptic ulcer releases NH₃ in the concentration range of 0.82–14.7 ppm.⁷ Furthermore, NH₃ is produced from farming areas or animal agriculture that can affect health of humans, animals, and environment.⁸ Therefore, detection of NH₃ is necessary for life saving, environment protection, and medical applications. Nowadays, sensing nanomaterials and fabrication techniques for NH₃ gas sensors have been extensively studied in term of high sensitivity, high selectivity, high stability, fast response, low cost, and ease of use. Metal oxides are one of the most popular NH₃-sensing nanomaterials. However, most NH₃ gas sensors based on metal oxides require high working temperatures of 150–300 °C and exhibit cross-sensitivity.^{9,10} The development of room-temperature NH₃ gas sensors based on nanocomposites^{11–14} is a way to overcome

these crucial problems but until now no nanocomposites exhibited perfect selectivity to NH₃, leading to impractical use in real-world applications based on single gas detection with quantitative purpose.

A nanocomposite is defined as a multicomponent material in which at least one of components are of nanoscale size. It generally comprises multiphase inorganic/organic materials such as metal–metal oxides (Ni₆₀Fe₃₀Mn₁₀, Sn/Cu/ZnO NPs, MoS₂–Au, MoS₂/ZnO, and Cu–BTC/BNNT),^{15–19} metal oxide–carbon nanomaterials (TiO₂–RGO, CoFe₂O₄/graphene, and rGO/MoS₂),^{20–22} metal oxide–polymer (Pd–TiO₂@PPy, PANI/SnO₂, and PPy/Fe₂O₃),^{23,24} and metal oxide–carbon nanomaterials–polymer (Pd–PANI–rGO, TiO₂@PPy–GN, graphene–PEDOT:PSS, and Pt/PAN–MWCNTs/WGE).^{25–28} Among them, the hybrid carbon-based nanostructures have become one of the most attractive materials that can provide multidisciplinary applications, including flexible battery, biosensor, solar cell, super-capacitors, chemical sensor, and gas sensing,^{29–31} due to their unique properties such as low-cost production, strong covalent-bond structures, large specific surface area, high electrical con-

Received: July 15, 2019

Accepted: September 23, 2019

Published: October 3, 2019

ductivity, low redox potential, and high mechanical strength.^{32–34} In this work, we focus on new metal oxide–carbon nanocomposites consisting of tin–titanium dioxide/reduced graphene/carbon nanotube (Sn–TiO₂@rGO/CNT) for NH₃-sensing application at room temperature. The Sn–TiO₂@rGO/CNT nanocomposites were synthesized by the solvothermal process and characterized via scanning electron microscopy (SEM), transmission electron microscopy (TEM), and X-ray photoelectron spectroscopy (XPS). The sensing performances of Sn–TiO₂@rGO/CNT nanocomposites were systematically investigated. The NH₃-sensing mechanism was proposed based on the formation of p–n heterojunctions.

2. RESULTS AND DISCUSSION

2.1. Characterization of Sensing Films. The SEM image of Sn–TiO₂@rGO/CNT nanocomposite is shown in Figure 1a. It illustrates agglomeration of rGO and CNT leading to

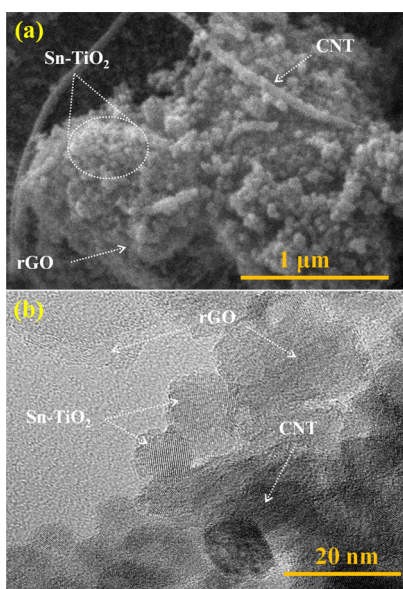


Figure 1. (a) SEM and (b) TEM images of Sn–TiO₂@rGO/CNT nanocomposite.

self-assembled 3D hierarchical interconnected morphology, whereas the Sn–TiO₂ nanoparticles are randomly embedded on the surface of the rGO/CNT nanocomposite. The corresponding TEM image confirms the coexistence of Sn–TiO₂ nanoparticles, graphene, and CNTs as demonstrated in Figure 1b. It can be seen that the Sn–TiO₂ nanoparticles with diameters of <15 nm are widely coated on graphene and CNT surfaces.

XPS survey scan measurements were performed to investigate the chemical composition of the Sn–TiO₂@rGO/CNT nanocomposite as illustrated in Figure 2. The survey scan XPS spectra of Sn–TiO₂@rGO/CNT with different molar ratios of Sn/Ti confirm the presence of C, Ti, Sn, and O in agreement with the expected chemical compositions of the Sn–TiO₂@rGO/CNT nanocomposite. The Sn atomic contents of Sn–TiO₂@rGO/CNT nanocomposites were calculated from the XPS data to be around 2.9, 3.8, and 5.4 at. % for nanocomposite I, nanocomposite II, and nanocomposite III, respectively. It can be seen that the Sn adding amounts increase approximately linearly with increasing Sn in TiO₂@rGO/CNT based on solvothermal method. In addition, all

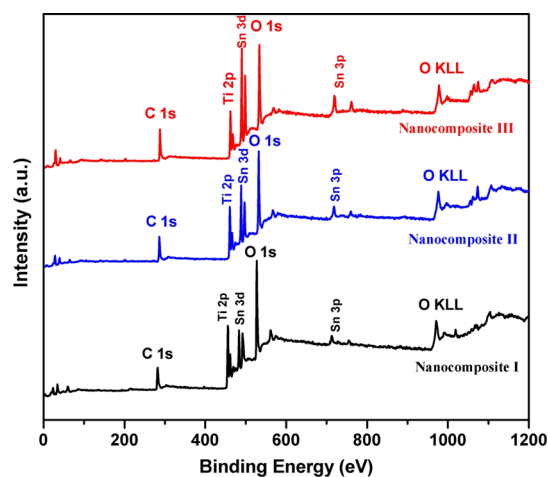


Figure 2. XPS survey scan spectra of Sn–TiO₂@rGO/CNT nanocomposite.

element atomic contents of nanocomposite I were found to be around C 1s (40.7 at. %), Ti (12.6 at. %), Sn (2.9 at. %), and O 1s (43.8 at. %).

The high-resolution C 1s peak (Figure 3a) can be deconvoluted into five peaks located at 283.3, 284.8, 286.1, 286.9 and 288.6 eV which correspond to C=C, C–C, C–OH, C–O, and COOH groups in rGO/CNT, respectively.^{35–40} The C=C and C–C components refer to the backbone of the rGO/CNT structure, whereas the C–OH, C–O, and COOH bonds indicate the oxygen-containing functional groups. It can be seen that the C=C and C–C bonds have higher intensity than the oxygen-containing functional groups of C–OH, C–O, and COOH bonds. XPS spectra of Sn 3d and Ti 2p in nanocomposite I are presented in Figure 3b,c, respectively. Two peaks of Sn 3d spectrum at 486.6 and 495.0 eV are attributable to Sn 3d_{5/2} and Sn 3d_{3/2} spin orbit peaks of the Sn⁴⁺ state in SnO₂, confirming the formation of SnO₂ nanoparticles in the Sn–TiO₂@rGO/CNT nanocomposite.^{41,42} For Ti 2p spectrum, two spin–orbit peaks of Ti 2p_{3/2} and Ti 2p_{1/2} are located at 458.8 and 464.5 eV, respectively. The two prominent peaks can be assigned to the Ti⁴⁺ state of TiO₂.^{41,43} The high-resolution O 1s XPS spectrum of the nanocomposite I is shown in Figure 3d, the O 1s peak can be decomposed into four contributions centered at 529.9, 531.1, 531.9 and 532.9 eV, which can be attributed to O–metal (Sn, Ti) bonding,^{44–46} Sn–O–C or Ti–O–C,^{47,48} Sn–OH or Ti–OH,⁴⁹ and C–OH/C–O–C,⁵⁰ respectively.

2.2. Gas Sensing Properties. Figure 4a shows the change in resistance of the nanocomposite I gas sensor upon exposure to NH₃ vapor with different concentrations ranging from 25 to 250 ppm at room temperature. It is seen that the Sn–TiO₂@rGO/CNT gas sensor exhibits a rapid decrease of resistance upon exposure to NH₃ vapor before recovering to their baseline values in dry air. The resistance changing behaviors may be attributed to the adsorption and desorption of NH₃ molecules of the sensing films. The details of sensing mechanism for the Sn–TiO₂@rGO/CNT gas sensor will be discussed in the next section. The nanocomposite I gas sensor has been further tested to evaluate reproducibility under exposure to 250 ppm NH₃ vapor at room temperature as demonstrated in Figure 4b. It demonstrates that the Sn–TiO₂@rGO/CNT gas sensor can fully recover to their initial baselines for several repeated cycles. Consequently, the Sn–

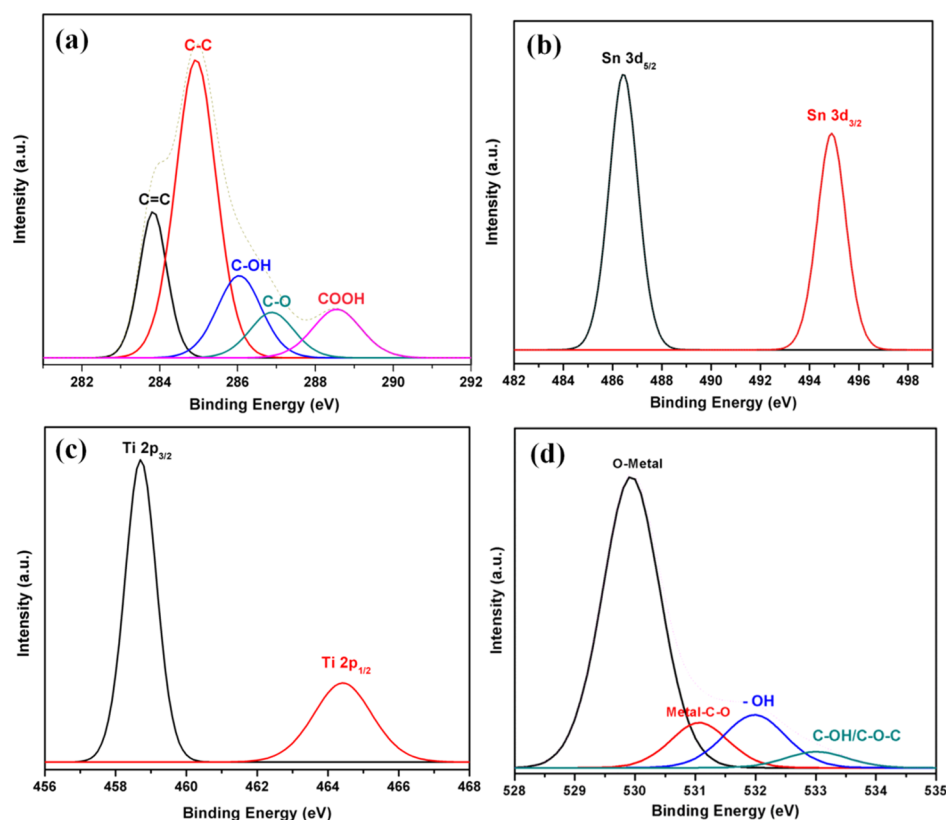


Figure 3. The high-resolution XPS spectra of the nanocomposite I for (a) C 1s, (b) Sn 3d, (c) Ti 2p, and (d) O 1s.

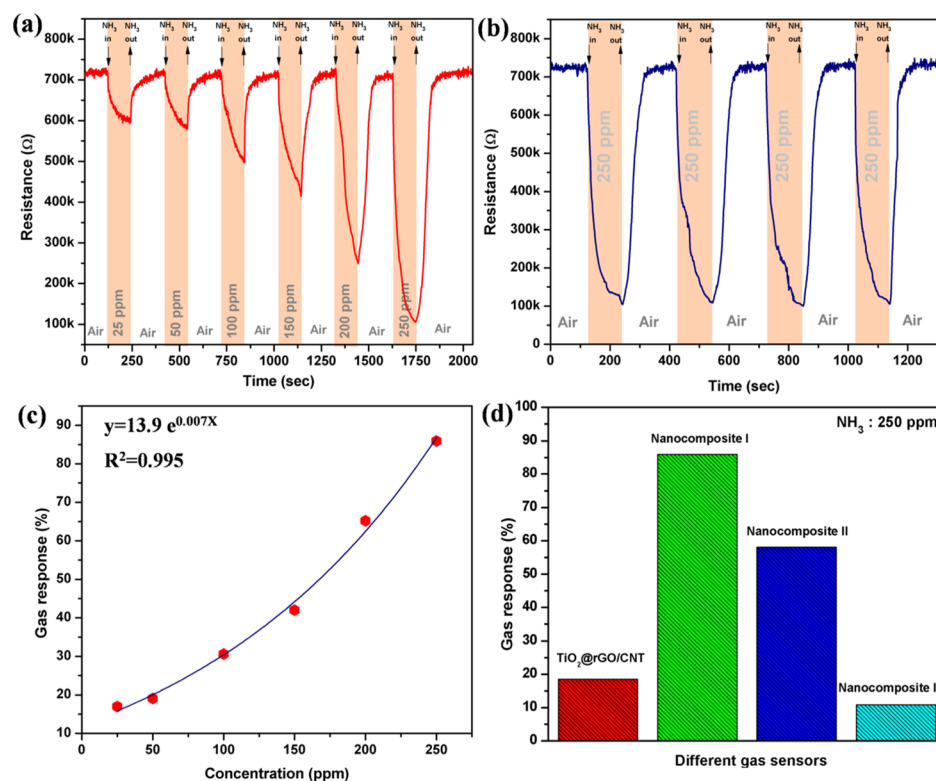


Figure 4. Changes in resistance of (a) nanocomposite I gas sensors to ammonia vapor with various concentrations and (b) four repeated pulses of 250 ppm ammonia at room temperature. (c) Gas responses of nanocomposite I gas sensor to ammonia vapor with various concentrations. (d) Gas responses of nanocomposite I to 250 ppm ammonia vapor at room temperature compared with nanocomposite II, nanocomposite III, and $\text{TiO}_2@r\text{GO}/\text{CNT}$ gas sensors.

TiO₂@rGO/CNT gas sensor indicates its good reproducibility for NH₃ detection at room temperature. The stability of the Sn–TiO₂@rGO/CNT nanocomposite gas sensor has been investigated with exposure to 250 ppm ammonia for 30 days. Based on storage of gas sensor at room temperature with uncontrolled relative humidity (RH), a slight gas response drop (only about 3%) could be observed after 30 days (see Figure S1 in the Supporting Information). It indicates that the Sn–TiO₂@rGO/CNT nanocomposite gas sensor exhibits good stability over 30 days.

The nanocomposite I gas sensor as a function of NH₃ concentration at room temperature is shown in Figure 4c. It can be seen that the gas response increases exponentially over the NH₃ concentration range of 25–250 ppm. The gas responses of the nanocomposite I gas sensor at 25, 50, 100, 150, 200, and 250 ppm NH₃ are 17.0, 19.1, 30.6, 42.0, 65.2 and 85.9%, respectively. The gas response as functions of NH₃ concentration (C_{NH_3}) can be fitted by a power law function relation according to the equation: $S = 13.9 e^{0.007C_{\text{NH}_3}}$ with a good correlation coefficient (0.995). The calculated response and recovery times of the nanocomposite I gas sensor for detection of 250 ppm NH₃ are 99 and 66 s, respectively, which are relatively short for a room-temperature sensor.

The effect of different molar ratios of Sn/Ti on the gas response toward 250 ppm NH₃ vapor at room temperature was investigated as shown in Figure 4d. It can be observed that the gas response increases from 18.5 to 85.9% after adding Sn/SnO₂ to TiO₂@rGO/CNT nanocomposites. However, more Sn/SnO₂-loading contents into the TiO₂@rGO/CNT nanocomposites (nanocomposite II and nanocomposite III) exhibit a decrease in response. Thus, nanocomposite I is the optimal condition of the Sn–TiO₂@rGO/CNT gas sensor that yields the highest NH₃ response at room temperature. The observed results may be explained based on the formation of p–n heterojunctions. The gas response of the nanocomposite sensor has been improved after adding Sn/SnO₂ due to the increasing number of Sn/SnO₂ nanoparticles corresponding to the number of p–n heterojunctions on the surfaces. The increasing number of Sn/SnO₂ nanoparticles on TiO₂@rGO/CNT surfaces may enlarge the specific surface area of the sensing film to absorb more oxygen molecules on the sensing film surface and increase the interface of p–n heterojunction. When the Sn/SnO₂ loading amount becomes too high, the Sn/SnO₂ molecules may aggregate into larger nanoparticles or cover the active centers of the nanocomposite, leading to a reduced number of p–n heterojunctions, causing a lower response. To distinguish the performance contribution of each ingredient within the nanocomposites, rGO/CNT and Sn–TiO₂ gas sensors were fabricated and tested toward 250 ppm NH₃ at room temperature (see Figure S2 in the Supporting Information). One can observe that the initial resistances of the rGO/CNT and Sn–TiO₂ gas sensors are too low and too high, respectively, whereas the resistance of the Sn–TiO₂@rGO/CNT sensor is between the resistance of rGO/CNT and Sn–TiO₂ sensors. The resistance of the rGO/CNT sensor increases and Sn–TiO₂ sensor resistance decreases in the present of NH₃ according to p-type and n-type characteristics, respectively. The gas responses of the rGO/CNT sensor and Sn–TiO₂ are calculated to be 30.0 and 28.8%, respectively. These results confirm the benefit of p–n heterojunction formation that gives better gas responses than pure p-type or n-

type materials, that is over 55% for sensing NH₃ at room temperature.

The selectivity of the optimal Sn–TiO₂@rGO/CNT gas sensor has been characterized toward various gases including toluene (C₇H₈), dimethylformamide (DMF), acetone (C₃H₆O), ammonia (NH₃), ethanol (C₂H₆O), methanol (CH₃OH), isopropanol (C₃H₈O), formaldehyde (CH₂O), thinner, hydrogen (H₂), carbon dioxide (CO₂), and acetylene (C₂H₂) at concentrations as shown in Figure 5. It can be seen

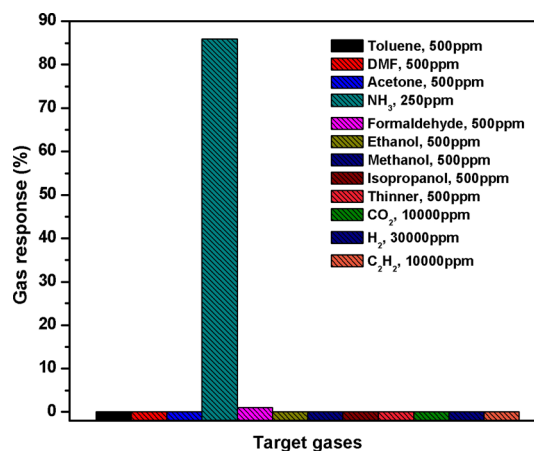


Figure 5. Selectivity histogram of the nanocomposite I gas sensor to various gases/VOCs at room temperature.

that the optimal Sn–TiO₂@rGO/CNT gas sensor exhibits ultrahigh selective NH₃ over several tested gases at room temperature. The high selectivity toward NH₃ may be related to the interaction between polarity of NH₃ molecules and hydroxyl groups on the Sn–TiO₂ surface leading to strong chemisorptions, whereas it is too difficult for other tested gases to form hydrogen bonds with these hydroxyl groups on the Sn–TiO₂ surface at room temperature.⁵¹ Moreover, NH₃ molecules were favorable to promoting molecular adsorption on the surface acidity of Sn–TiO₂ existing Lewis acid sites (Sn⁴⁺ and Ti⁴⁺) and Brønsted acid sites (Sn–OH and Ti–OH).⁵² To investigate the humidity effect, the resistance of the Sn–TiO₂@rGO/CNT sensor was measured in real time at various humidity levels from 30 to 90% RH (see Figure S3 in the Supporting Information). It is seen that the sensor resistance does not change significantly in the range of 30–70% RH at room temperature. At high humidity level (RH > 70%), the sensor resistance decreases with increasing RH. However, when humidity response of the Sn–TiO₂@rGO/CNT sensor (defined as $[R_{30\%RH} - R_x] \times 100 / R_{30\%RH}$, where $R_{30\%RH}$ and R_x are the sensor resistance values at 30% RH and the target value at another RH, respectively) is calculated, the sensor exhibits only 13.8% at 90% RH that is lower than the NH₃ gas response for all tested NH₃ concentrations (25–250 ppm). Thus, it can confirm the humidity-independent sensor for normal living condition. Moreover, the NH₃-sensing performance of the optimal Sn–TiO₂@rGO/CNT gas sensor in this study is superior to other recent reports of room-temperature NH₃ as listed in Table 1.

2.3. Sensing Mechanism of Sn–TiO₂@rGO/CNT Nanocomposite Gas Sensor. From the results, the resistance changing behaviors may be attributed to the adsorption and desorption of NH₃ molecules on the Sn–TiO₂@rGO/CNT nanocomposite sensing film. The NH₃-sensing mechanisms of

Table 1. Comparison between the Present Study and Other Reported Ammonia Gas Sensors Operating at Room Temperature

materials	response (%)	ammonia (ppm)	refs
CuSbS ₂ QDs/rGO composites	42 ^a	250	53
PANI/Fe ₃ O ₄	47 ^a	300	54
OP-G/G channel	29.3 ^a	1000	55
CCO-Co(10)	7.9 ^a	400	56
O ₂ functional groups on rGO	3.1 ^a	300	57
0.1-Pt QDs/WS ₂ NSs	14.1 ^b	500	58
TiO ₂ QDs/WS ₂ nano hybrids	43.75 ^b	250	59
graphite oxide	30 ^a	500	60
SnO ₂ /rGO hybrids	4.73 ^a	300	61
Sn-TiO ₂ @rGO/CNT	85.9 ^a	250	this work

^a $\Delta R/R_a \times 100$. ^b $\Delta I/I_a \times 100$.

the Sn-TiO₂@rGO/CNT nanocomposite gas sensor at room temperature may be explained based on the formation of p-n heterojunctions between p-type rGO/CNT and n-type metal-oxide-semiconductor (MOS) of Sn-TiO₂.⁶²⁻⁶⁴ In air, oxygen species (O₂⁻) chemisorbed on the surface of Sn-TiO₂@rGO/CNT nanocomposite film will capture free electrons from the conduction band of Sn-TiO₂ and the Fermi level of rGO/CNT according to the surface reaction: (1) O₂ (ads) + e⁻ → O₂⁻ (ads), leading to the formation of surface depletion layers on the interface of the Sn-TiO₂@rGO/CNT sensing film and cause the high resistance of the sensor in air as illustrated in Figure 6a.⁶⁵⁻⁶⁷ Upon exposure to NH₃ vapor, NH₃ molecules as a reducing gas will react with oxygen species according to the reaction: (2) 4NH₃ (gas) + 5O₂⁻ (ads) → 4NO (gas) + 6H₂O (gas) + 5e⁻, leading to discharge of electrons into the conduction band of Sn-TiO₂ on the sensor surface (as seen in Figure 6a).^{27,68,69} This process results in a decrease of the

thickness of the electron depletion layer leading to decrease in the resistance of the sensor. Moreover, the NH₃-sensing process of the Sn-TiO₂@rGO/CNT nanocomposite generates a charge between Sn-TiO₂ MOS and NH₃ molecules to electron transport through the rGO/CNT nanostructure. The rGO/CNT behaves as the electrical conduction pathway between Sn-TiO₂ grain boundaries due to the high carrier mobility of rGO/CNT. The Sn-TiO₂ nanoparticles are randomly interwoven with the rGO/CNT nanostructure to form a nanocomposite network that can enhance the surface area of the sensing film to increase the ability for adsorption and desorption of gas molecules.

Furthermore, Sn-TiO₂ and rGO/CNT are typical n-type MOS and p-type semiconductors, respectively. When the p-n heterojunctions are formed, the electrons in Sn-TiO₂ will transfer to rGO/CNT, whereas the holes will transfer from rGO/CNT to Sn-TiO₂ because the n-type Sn-TiO₂ has a higher Fermi level than the p-type rGO/CNT, as shown in Figure 6b.⁷⁰⁻⁷² Once the Sn-TiO₂@rGO/CNT sensor is exposed to NH₃ vapor, the reaction between NH₃ molecules and oxygen species will occur on the n-type Sn-TiO₂ surface, so that the electrons are released from the ionized oxygen species back to the conduction band of n-type Sn-TiO₂ and combine with holes of p-type rGO/CNT, leading to reduction of the p-n heterojunction electron depletion layer width and decrease the potential barrier height ($\Delta\Phi$). The resistance of the Sn-TiO₂@rGO/CNT sensor thus decreases upon exposure to NH₃ vapor at room temperature. In addition, after the sensor is exposed to NH₃ molecules, the resistance will decrease due to the decrease of the potential barrier height at the p-n heterojunction according to the relationship: $R = B \exp(\Delta\Phi q/kT)$, where R is the sensor resistance, B is a constant, q is the electron charge, $\Delta\Phi$ is an effective potential barrier height at the p-n heterojunction, k is Boltzmann's constant, and T is the temperature of the sensing layer.^{39,73}

3. CONCLUSIONS

In conclusion, Sn-TiO₂@rGO/CNT nanocomposites were successfully synthesized by the solvothermal method and systematically characterized for NH₃ sensing at room temperature. SEM, TEM, and XPS characterizations confirmed the presence of Sn-TiO₂ nanoparticles on the surface of rGO/CNT nanocomposite. As seen from the gas-sensing results, the Sn-TiO₂@rGO/CNT gas sensor demonstrated a rapid decrease of resistance upon exposure to NH₃ and fully recovered to its baseline values in air with good repeatability. The nanocomposite I gas sensor Sn-TiO₂@rGO/CNT with molar ratio of Sn/Ti (1:10) showed the highest response to NH₃ at room temperature due to balance of p-n heterojunctions and high active sensing area. Moreover, the Sn-TiO₂@rGO/CNT gas sensor exhibited ultrahigh selectivity toward NH₃ against various volatile organic compounds (VOCs) and environmental gases at room temperature with no effect of humidity in the range of 30–70% RH. The NH₃-sensing mechanism of the Sn-TiO₂@rGO/CNT nanocomposite gas sensor has been proposed based on the formation of p-n heterojunctions of p-type rGO/CNT and n-type Sn-TiO₂ via the low-temperature oxidizing reaction process. The observed ultrahigh NH₃ selectivity, high response, relatively short response, and recovery times proved that the Sn-TiO₂@rGO/CNT nanocomposite can be considered as a

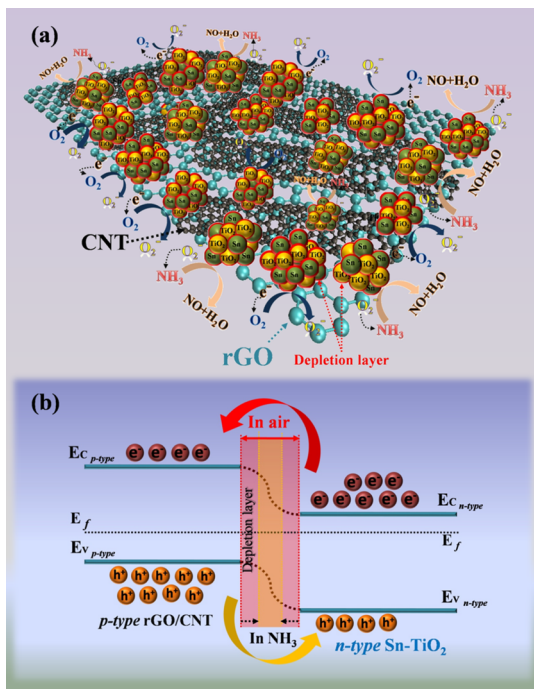


Figure 6. (a) Schematic of the proposed NH₃-sensing mechanism and (b) p-n heterojunction of the Sn-TiO₂@rGO/CNT nanocomposite gas sensor.

promising material for practical application of NH_3 detection at room temperature.

4. EXPERIMENTAL DETAILS

4.1. Synthesis of $\text{Sn-TiO}_2@\text{rGO}/\text{CNT}$ Nanocomposites. The synthesis process of the $\text{Sn-TiO}_2@\text{rGO}/\text{CNT}$ nanocomposite is demonstrated in Figure 7. Graphene oxide

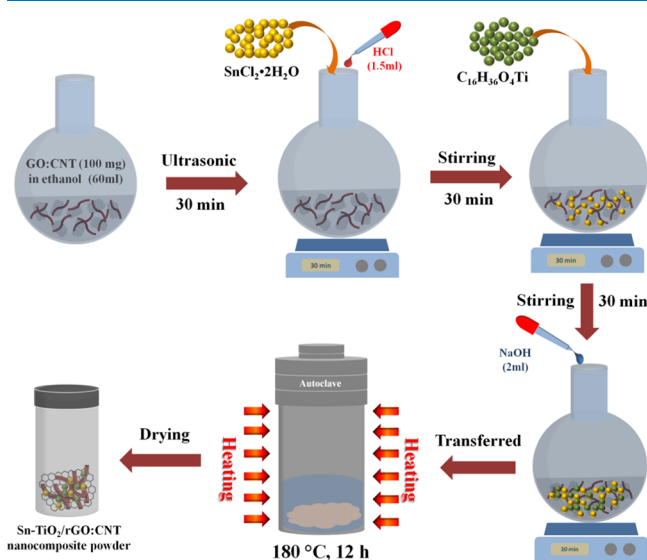


Figure 7. Schematic diagram of $\text{Sn-TiO}_2@\text{rGO}/\text{CNT}$ nanocomposite synthesis process.

(GO) was purchased from ACS Material, LLC. CNT (with diameter 20–40 nm, length $\approx 30 \mu\text{m}$) were purchased from Timesnano Co., Ltd. (Chengdu, China). $\text{Sn-TiO}_2@\text{rGO}/\text{CNT}$ composites were synthesized by the solvothermal method. The amount of 100 mg of GO and CNT was dispersed in 60 mL of ethanol and treated in an ultrasonic processor for 30 min. Then, stannous chloride ($\text{SnCl}_2 \cdot 2\text{H}_2\text{O}$) and 1.5 mL of HCl (1 M) were added into the rGO/CNT solution under mechanical stirring for 30 min. Tetrabutyl titanate ($\text{C}_{16}\text{H}_{36}\text{O}_4\text{Ti}$) was mixed into the above suspension and stirred till molar ratios of Sn/Ti is 1:10. This was followed by the addition of 2 mL NaOH (1 M) dropwise and mechanically stirred for 30 min. The solution was then transferred to Teflon-lined stainless-steel autoclave and heated oven at $180 \text{ }^\circ\text{C}$ for 12 h. At the end of reaction, the precipitates were collected, washed with deionized (DI) water and ethanol 3 times, respectively, and then dried at $80 \text{ }^\circ\text{C}$ overnight to obtain the as-prepared $\text{Sn-TiO}_2@\text{rGO}/\text{CNT}$ nanocomposites. Finally, the well crystalline solid powders were sintered at $500 \text{ }^\circ\text{C}$ for 3 h. In addition, the $\text{Sn-TiO}_2@\text{rGO}/\text{CNT}$ nanocomposite with different molar ratios of Sn/Ti (3:10 and 5:10) were also produced with the same process in order to investigate the effects of Sn content on sensing properties.

4.2. Fabrication of $\text{Sn-TiO}_2@\text{rGO}/\text{CNT}$ Nanocomposite Gas Sensors. The fabrication process of the $\text{Sn-TiO}_2@\text{rGO}/\text{CNT}$ nanocomposite gas sensor is displayed in Figure 8. $\text{Sn-TiO}_2@\text{rGO}/\text{CNT}$ nanocomposite powder (100 mg) was dispersed in 1 mL of DI water under ultrasonication for 5 min. The aluminum-interdigitated electrodes with the size of $4 \text{ mm} \times 4 \text{ mm}$ and spacing $100 \mu\text{m}$ were prefabricated on an alumina substrate by conventional photolithography, radio-frequency sputtering, and lift-off processes. Before drop coating of the

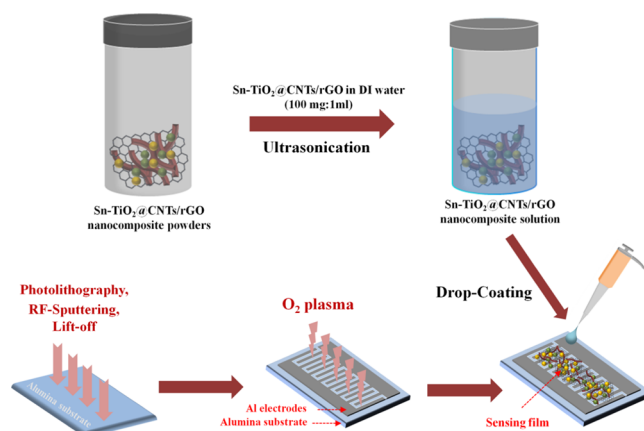


Figure 8. Schematic diagram of $\text{Sn-TiO}_2@\text{rGO}/\text{CNT}$ nanocomposite gas sensor fabrication process.

sensing film, the aluminum interdigitated electrodes on the alumina substrate were treated by O_2 plasma in order to improve the adhesion of the sensing film on the substrates and render a hydrophilic surface. Subsequently, the $\text{Sn-TiO}_2@\text{rGO}/\text{CNT}$ aqueous dispersion was drop-coated on the aluminum-interdigitated electrodes and dried at $80 \text{ }^\circ\text{C}$. The fabricated $\text{Sn-TiO}_2@\text{rGO}/\text{CNT}$ gas sensors with different molar ratios of Sn/Ti (1:10, 3:10, and 5:10) were defined as nanocomposite I, nanocomposite II, and nanocomposite III gas sensors, respectively.

4.3. Gas-Sensing Measurement. The gas-sensing properties of $\text{Sn-TiO}_2@\text{rGO}/\text{CNT}$ nanocomposite gas sensors were tested in a Teflon chamber with the dynamic flow measurement. Various gases/VOCs including toluene, DMF, acetone, ammonia, formaldehyde, ethanol, methanol, isopropanol, thinner, carbon dioxide, hydrogen, and acetylene were used to evaluate the response and selectivity of fabricated gas sensors at the room temperature ($26 \pm 2 \text{ }^\circ\text{C}$) with RH ($56 \pm 2\%$). The concentrations of test gases were varied using mass flow controllers with a flux of synthetic air. The total gas rate was fixed at 1000 sccm. The baseline of sensors was obtained by clean air for 3 min, and then the tested gas at a particular concentration was introduced into the sensor chamber for 2 min. A simple voltage divider circuit at a fixed voltage of 5 V was employed to measure the sensor resistances. The resistances of gas sensors were recorded every 1 s via a USB NI-DAQ 6008 under our developed LabVIEW software. The performances of the gas sensors were determined by means of gas response and selectivity. The gas response is defined as $S (\%) = [(R_{\text{air}} - R_{\text{gas}})/R_{\text{air}}] \times 100$, where R_{air} and R_{gas} are the resistance of the fabricated gas sensor in clean air and the test gas, respectively. Selectivity is the ability of a sensor to identify a target gas that can be evaluated from relative response between different gases. The response time is defined as the time required for the sensor resistance to reach 90% of the final equilibrium signal upon exposure to the target gas, while the recovery time is the time needed to recover 90% of the initial baseline.

■ ASSOCIATED CONTENT

📄 Supporting Information

The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acsomega.9b02185.

Stability of gas sensor and changes in resistance of Sn–TiO₂ and rGO/CNT gas sensors to 250 ppm ammonia as well as the effect of humidity on sensor resistance (PDF)

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Notes

The authors declare no competing financial interest.

ACKNOWLEDGMENTS

This work was financially supported by a grant (TRF-KURDI Research Career Development Grant) from Thailand Research Fund and Kasetsart University Research and Development Institute (RSA6180062).

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