

# Characterization, Selection, and Microassembly of Nanowire Laser Systems

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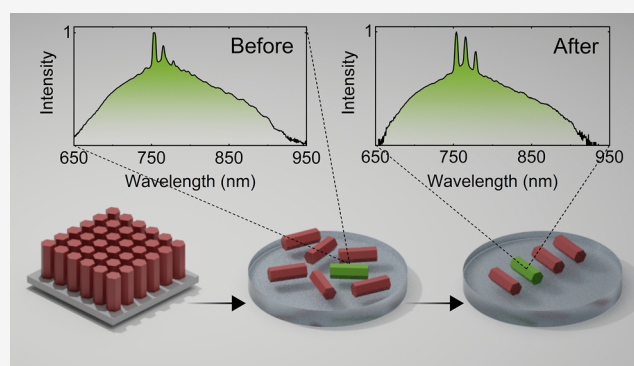
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**ABSTRACT:** Semiconductor nanowire (NW) lasers are a promising technology for the realization of coherent optical sources with ultrasmall footprint. To fully realize their potential in on-chip photonic systems, scalable methods are required for dealing with large populations of inhomogeneous devices that are typically randomly distributed on host substrates. In this work two complementary, high-throughput techniques are combined: the characterization of nanowire laser populations using automated optical microscopy, and a high-accuracy transfer-printing process with automatic device spatial registration and transfer. Here, a population of NW lasers is characterized, binned by threshold energy density, and subsequently printed in arrays onto a secondary substrate. Statistical analysis of the transferred and control devices shows that the transfer process does not incur measurable laser damage, and the threshold binning can be maintained. Analysis on the threshold and mode spectra of the device populations proves the potential for using NW lasers for integrated systems fabrication.

**KEYWORDS:** III–V Nanowire Lasers, Transfer-Printing, Photoluminescence, Nanophotonics



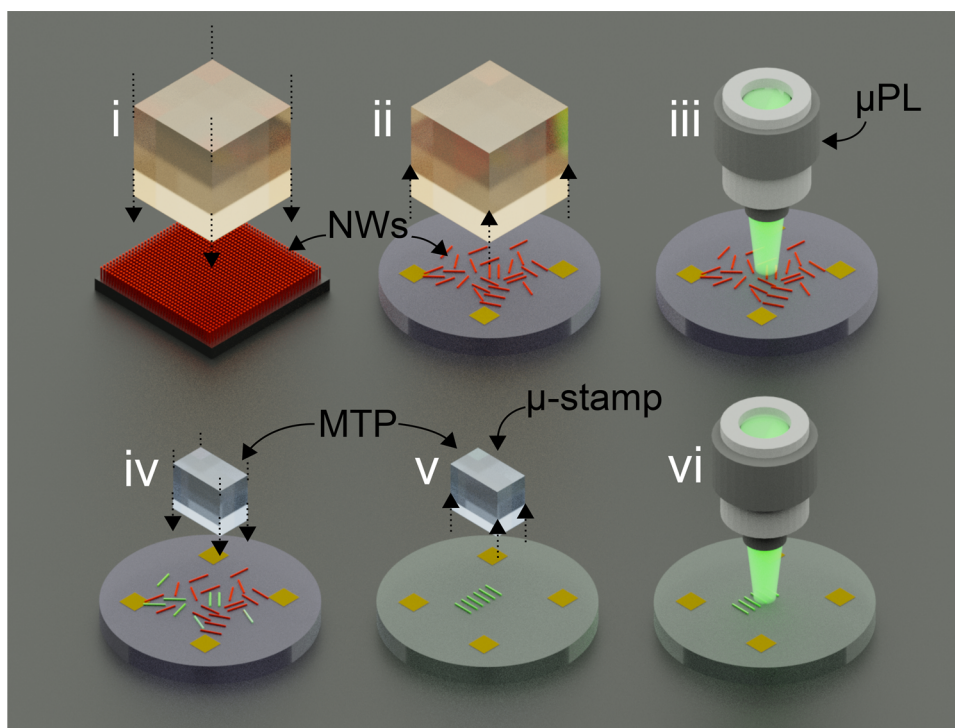
Nanowire (NW) lasers are ultracompact, energy efficient sources of coherent light<sup>1</sup> with potential applications ranging from distributed on-chip sensing<sup>2,3</sup> to optical signal processing.<sup>4</sup> A significant amount of effort has been made to optimize the growth processes and physical structure of NW devices to improve brightness and control emission wavelength and modal structure.<sup>5,6</sup> Furthermore, a range of schemes have been developed to integrate these lasers with the necessary on-chip optical components to provide a toolbox to produce future systems. Vertically structured NWs have been directly grown onto silicon waveguide platforms,<sup>7</sup> and discrete NW devices have been transferred to host substrates, postgrowth, for integration with waveguides,<sup>8</sup> plasmonics,<sup>9,10</sup> and complementary NW structures.<sup>11</sup> Typically, however, such systems have been proof-of-principle demonstrations of limited numbers of devices, requiring preselection of suitable devices and skilled manually controlled microassembly techniques. Nevertheless, pick-and-place assembly is a promising route toward automated fabrication of future systems based on NWs. Future scalable integration of NW devices will have two major requirements: (1) that as-grown NWs can be removed from their growth structure with high yield and (2) that large sets of these devices can be easily characterized before integration to ensure performance matching the application. The transfer of NWs from their growth substrate to a host wafer usually includes a fracturing stage where devices are physically

“snapped” at some point along their length, producing one of the two reflective facets required for lasing.<sup>12</sup> This mechanical process produces populations of devices across a sample with variations in length and facet quality that in turn affect lasing threshold and modal spectrum. Therefore, in order to progress toward systems incorporating NW laser sources, a scalable approach is required to map the individual device performance and allow subsequent integration of particular devices with high-accuracy spatial positioning. Previous work has shown that it is possible to characterize large populations of NWs using automated microscopy-based systems.<sup>13</sup> In this work we present the combination of this population measurement method with a spatially resolved pick-and-place technique<sup>8,14,15</sup> to allow device binning and transfer. A randomly distributed population of NW lasers are characterized and binned by their lasing threshold. The subsets of devices are then transferred to a second substrate using computer controlled spatial registration to allow rapid location and submicrometer accurate alignment of NWs. Finally, automated characterization of the NW devices is repeated on

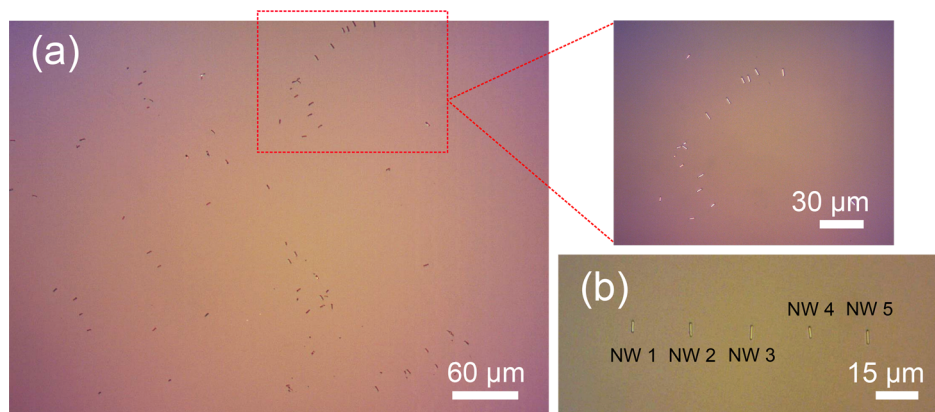
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**Figure 1.** Schematic of the process: (i) NW lasers on their growth substrate are released using a large PDMS block, (ii) transferred onto a target quartz disk, and randomly distributed on the surface. (iii) Device populations are characterized using an automated scanning microscopy system and binned by threshold. (iv) Selected NW devices are individually picked using a PDMS  $\mu$ -stamp and (v) printed onto a second substrate in regular arrays. (vi) Target devices are recharacterized with the system.



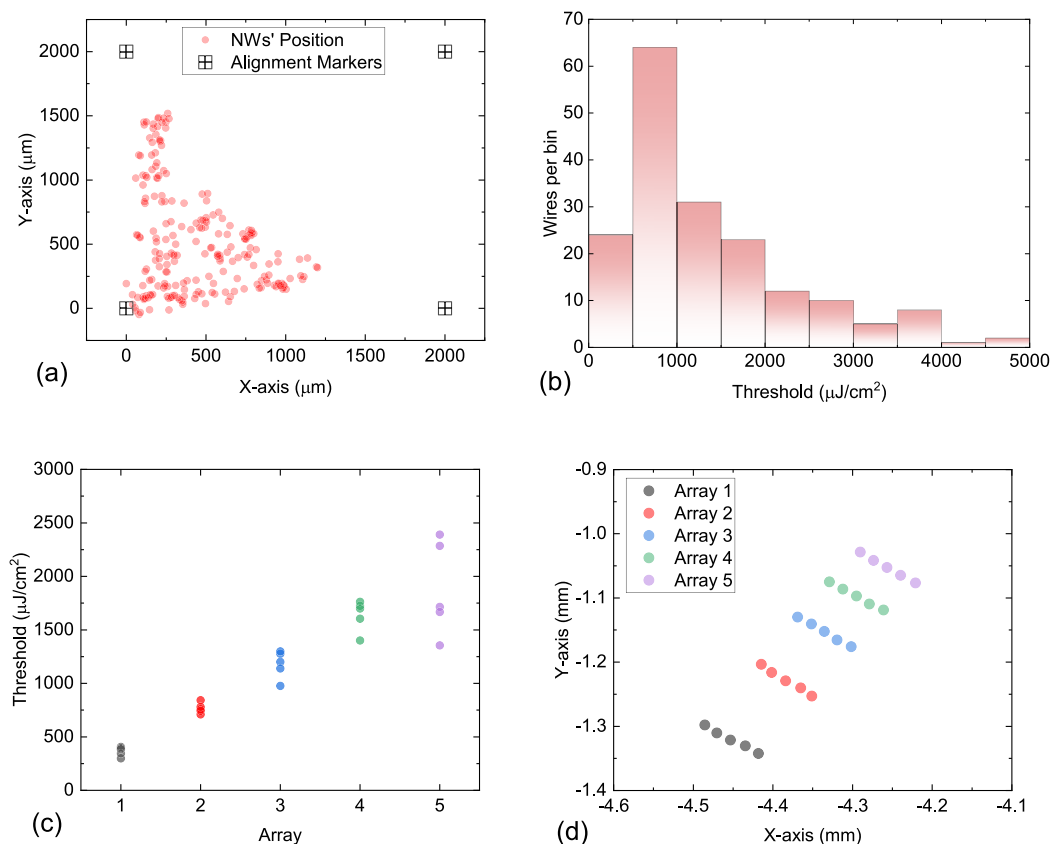
**Figure 2.** Quartz substrates with printed NW laser devices. (a) Randomly distributed first transfer and (b) 5 device array of binned and deterministically placed devices.

both the original sample and the sample with printed laser arrays to assess the effects of the printing process on device performance.

The NW lasers are composed of GaAs-AlGaAs core-shell wires grown on a GaAs substrate using a bottom-up approach, as reported in ref 16. The devices support multiple transverse mode solutions and have a diameter of  $\sim 450$  nm and a length of  $\sim 4$   $\mu$ m. Figure 1 shows a schematic flow diagram of the main stages of NW laser processing, measurement, and transfer from the growth substrate onto the host samples. Full process details can be found in the Supporting Information. In summary: (i, ii) Vertically grown NWs are released and coarsely distributed on a quartz substrate with random orientation. (iii) These devices are then optically measured and spatially mapped to allow device binning by a specific

parameter, such as threshold or lasing wavelength. (iv, v) Devices from the population are selected for transfer to a second substrate. An automatically spatially aligned transfer-printing process is used to transfer these devices into regular arrays on a second receiver substrate. (vi) Finally, the devices are remeasured using the same setup as in iii.

To achieve a controlled pick-and-place of the NW devices, the microtransfer-printing technique (MTP), described in ref 14, was used. The technique exploits a tacky nature of a polymeric microstamp to capture/release target NW lasers.<sup>14</sup> Two stamp designs were used: (1) to transfer large populations of NW from their growth substrate, and (2) to pick-and-place individually selected devices without affecting other NWs on the substrate. In the initial step, NW lasers were captured from their growth substrate using a  $\sim 2$  mm<sup>2</sup> flat polydimethylsilox-



**Figure 3.** (a) Spatial map of 221 NWs (red circles) on quartz substrate identified by the characterization rig. Dark grids indicate alignment markers on the substrate, with a marker at (0, 0) being the reference point. (b) Histogram of measured NW thresholds before the selection and transfer processes. (c) Binning of selected NW lasers by lasing threshold energy. (d) Absolute coordinates of the transfer-printed arrays on the second quartz disk.

ane (PDMS) stamp. The large stamp area was brought into direct contact with the vertical array of NWs and translated in one direction in order to induce shearing near the base of the NWs. The released NWs were bonded to the PDMS surface and can be subsequently transferred to a host substrate. In this work a *z*-cut quartz disk, coated with a  $\sim 4 \mu\text{m}$  thick photopolymer<sup>17</sup> layer, was used as the receiver. The photopolymer provided an adhesive surface to improve capture of the NWs, and the quartz and photopolymer are transparent in the visible spectral range, allowing for measurement of the devices in a transmission microscopy arrangement. The NWs were transferred from the large-area PDMS stamp to the quartz sample producing a randomly distributed population of devices on the sample. The spatial density of these devices was coarsely controlled through the initial release and subsequent printing process parameters. In this work, a relatively sparse population was targeted to ensure the automatic characterization method addressed single, easily distinguishable devices. Figure 2a shows an optical microscope image of a random distribution of NWs on the quartz substrate with low spatial density. Furthermore, both the initial and final quartz receiver samples were patterned with lithographic markers to allow spatial mapping of the NWs to a predefined grid. Previous work on transfer-printing of thin membrane devices has shown positional accuracy of the process in the 100s of nanometers range.<sup>18</sup>

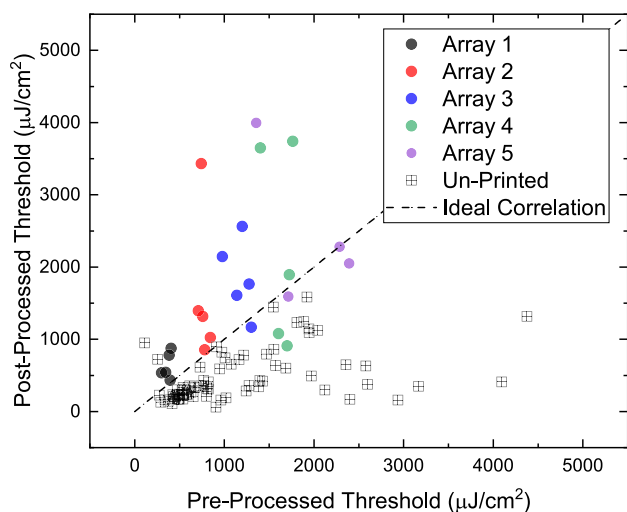
The randomly deposited NW lasers were measured using the automated optical characterization setup at room temperature as previously described in ref 13. Measurement details are

presented in full in the Supporting Information. The automated scanning measurement technique allows the direct mapping of NW spatial position to be associated with the optical measurements. The measurements taken include below and above threshold emission spectra, pump energy vs emission power, and both dark- and bright-field micrographs of the device obtained with a CCD camera.<sup>13</sup> In this work we selected the NW laser threshold energy as the binning parameter, as this will be a key metric in the fabrication of future integrated NW systems, designed to operate with similar performance across a chip or to yield lasing emission with comparable pump density across the array. The initial transfer step created a population of 221 NW devices, out of which 180 of them showed lasing emission, on the first quartz substrate. Figure 3a shows a spatial mapping scatter plot of NW lasers across the disk substrate: red dots are coordinates of the NW devices, and dark grids depict the alignment markers on the substrate. Figure 3b shows a histogram of the initially measured threshold energy densities of the NW laser population. From this population, five bins of lasing threshold energy were selected as shown in Figure 3c, varying between devices with thresholds below  $500 \mu\text{J}/\text{cm}^2$  and an upper band at around  $2500 \mu\text{J}/\text{cm}^2$ , with mean values separated by  $\sim 500 \mu\text{J}/\text{cm}^2$ , to produce distinguishable sets. Five devices were selected per bin and subsequently printed onto the second quartz sample. To transfer individual devices, a custom PDMS  $\mu$ -stamp was used as detailed in ref 14. The selected devices were distributed throughout the population with randomly distributed orientation of their long axes. The printing process

was semiautomated, taking as inputs the position of the NW relative to the registration marker at (0, 0) and the orientation taken from the bright-field microscopy image of the NW. The NW target position was defined as a set of coordinates on the target sample, relative to a registration marker on that substrate, and all devices were rotated to have their long axes colinear. The translation and rotation of the NWs during the printing process were computer controlled based on the measured registration marker and NW device positions. Figure 3d shows the measured coordinates of the printed arrays as measured by the characterization system. Printed devices were separated by  $\sim 20 \mu\text{m}$  to allow individual optical injection and measurement.

The postprocessing round of measurements was performed on both printed and unprinted samples after a period of six months, during which time both sets of samples were stored together in a desiccator cabinet in a temperature controlled lab. By using the unprinted sample as a reference, we were able to control for variation in experimental conditions (and device aging) between these two measurements. In particular, our comparison allows compensation for changes made to the laser characterization tool during this period, that may introduce some absolute variation between first and second measurement rounds, but should not affect relative changes between printed and unprinted device sets.

Figure 4 shows the extracted threshold energy densities of the printed (in color) and unprinted (black grids) devices



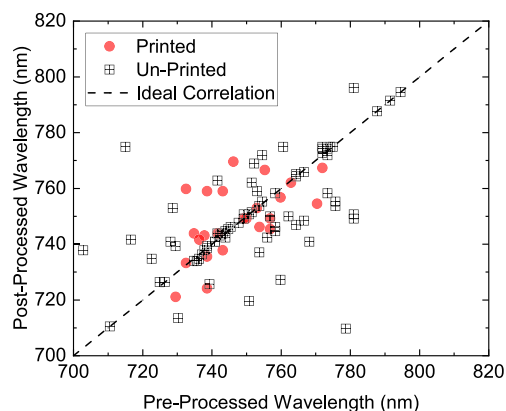
**Figure 4.** Correlation of laser threshold values between two measurements for printed and unprinted NW lasers. Dashed line represents ideal correlation.

before and after the transfer process, a dashed line is included as a visual aid corresponding to ideal correlation. In the printed device group, 24 out of 25 NW devices retained their lasing emission properties at room temperature, one device did not exhibit lasing. In the case of the unprinted devices, 74 out of the remaining 155 devices exhibited lasing during the second round of measurements. The reason for the significant decrease in lasing population in the unprinted devices is not clear at this stage, but may be due to thermal effects on the polymer layer at the laser surface during repeated measurements.

Individually, both populations of devices show correlation of their lasing threshold energy density from the first to second

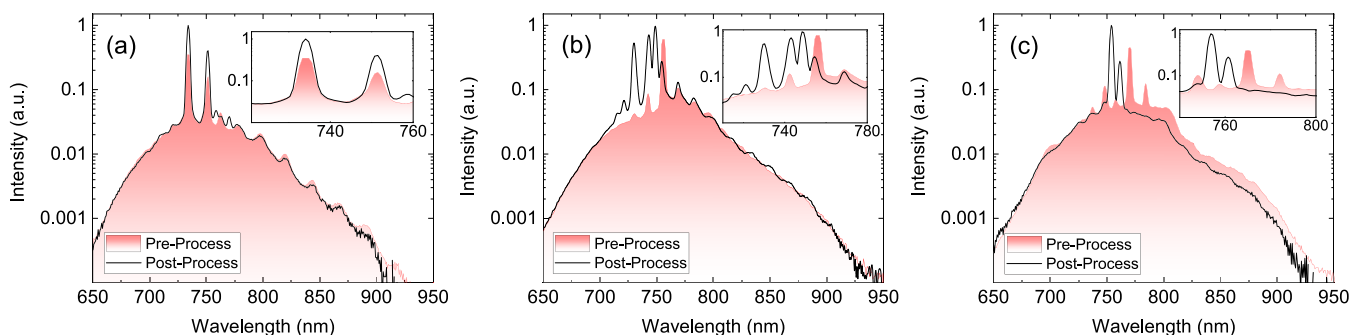
measurements, though there are clear differences. The linear correlation coefficient of threshold for the printed devices is moderate and significant, at 0.462 with  $p$ -value of 0.023, showing that the binned devices retain the global trend of threshold ranges, although a clear increase in absolute threshold levels is observed. Similarly, the unprinted devices exhibit a moderate and significant correlation coefficient of 0.435. In this case the second measurement round exhibits a decrease in absolute threshold levels. The threshold energy densities for both populations show systematic variation in absolute values, but retain their global relative trends. Therefore, the low threshold binned devices can be selected for, even though absolute values are not stable for either printed or unprinted populations, due to changes in the laser excitation system described above.

As noted earlier, the NW lasers in this work support multiple modes, and therefore any changes in the lasing threshold may be due to a change in dominant lasing mode between first and second measurement rounds. Before considering the lasing spectra, the subthreshold photoluminescence of the NWs was measured. As previously shown in ref 13, the photoluminescence spectral form of these devices consists of the two distinct peaks, at  $\sim 810$  and  $870 \text{ nm}$ , which correspond to the emission from the core and MQW regions of the devices. The shape of emission is unique from wire-to-wire due to the growth inhomogeneity. To verify that the microtransfer-printing techniques do not affect the printed devices, both pre- and postprocessing spectra were compared (see the Supporting Information). As no significant variations in the PL spectra were apparent, the peak lasing wavelength for both populations of devices was measured, as shown in Figure 5, where red circles represent printed devices, and black grids represent unprinted devices.

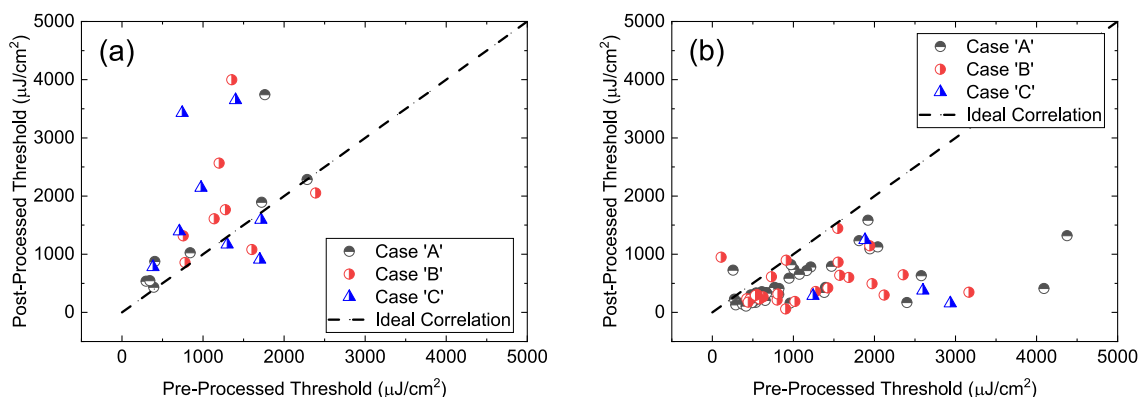


**Figure 5.** Correlation of peak lasing wavelength between two measurements for printed (red circles) and unprinted (black grids) devices. Dashed line represents ideal correlation.

Both populations of devices exhibit similar scatter around the ideal correlation curve, suggesting that the printing process does not dominate variations in peak lasing wavelength. The measured variation in peak lasing wavelength is significant for these multimode devices, extending beyond 10 nm in some cases. The peak wavelength measurement does not give a full account of the laser modal structure, and so a further qualitative measure of the spectra can be defined that may help to ascertain if the NW has been physically altered between measurements. This modal structure of the device is directly



**Figure 6.** Three representative cases of the printed and unprinted devices which depict state of the spectrum pre- and postprocessing: (a) reconstructed, (b) partially reconstructed, (c) non-reconstructed spectrum.



**Figure 7.** Scatter plot showing modal cases relatively to the threshold change in the NW devices: (a) printed and (b) unprinted devices.

related to its physical geometry and so provides a useful probe of any damage induced in the NWs. Three possible cases, taken from measured spectra, are presented in Figure 6 that show how the modal structure of the laser may change between measurements. For consistency, the pre- and postprocessed spectra of the selected device measurements were obtained significantly above their threshold level.

In the first case “A”, shown in Figure 6a, the Fabry–Pérot (FP) modes of the device in both measurements fully overlap; i.e., the modes overlap in wavelength and exhibit a similar distribution of optical power. For the particular device presented, the lasing threshold changed from 734 to 356  $\mu\text{J}/\text{cm}^2$ , but the modal structure shows very little variation. This suggests that the device losses, gain, or absorption properties may be affected, but its geometry and hence FP mode structure are consistent. The second common case “B”, Figure 6b, exhibits a spectrum where the FP modes of the device overlap between measurements, but the distribution of optical power has changed significantly. Again, in this case it would suggest that the geometry of the device is unaffected but the internal laser properties, or pumping conditions, have changed enough to favor lasing of an alternative mode. Finally, in case “C”, in Figure 6c the two modal spectra do not appear to overlap well, suggesting that the device has been physically affected, inducing a change in the FP mode structure. Figure 7 shows the correlation between measured laser threshold values between first and second measurement rounds. The form of each data point marker (black, red, and blue) indicates whether the lasing spectral overlap between measurement “A” and “B” resembles case “A”, “B”, or “C” most closely.

Table 1 presents the correlation coefficients and  $p$ -values for each of three spectral overlap cases and the full printed and

**Table 1. Threshold Fluence Correlations for Devices Grouped by Modal Spectrum Correlations**

mode spectrum case [unprinted printed]	unprinted		printed	
	$\rho$ -value	$p$ -value	$\rho$ -value	$p$ -value
all devices [74 24]	0.435	<.001	0.462	0.023
case A [45 8]	0.584	<.001	0.841	0.009
case B [25 8]	0.265	0.200	0.239	0.568
case C [4 8]	-0.294	0.706	0.006	0.989

unprinted device populations. In both the printed and unprinted devices, case “A” (full mode overlap) shows good correlation between first and second measurements. Cases “B” and “C” exhibit no significant correlation values for both populations. This suggests that where the device lasing modes are consistent between first and second measurements, the printing process does not introduce significant detrimental effects to the devices, and lasing thresholds are consistent. Nevertheless, the multimodal nature of the NW lasers studied here allows for variation in lasing mode between measurements and therefore a shift in lasing threshold. In order to maintain tight control over device characteristics single transverse-mode NW lasers could be used.

In conclusion, we have demonstrated that large populations of NW lasers with random spatial distribution can be rapidly characterized and sorted using their measured lasing parameters. These populations can then be transferred to separate substrates using a semiautomated pick and place technique that uses the same spatial registration markers as the characterization measurements. Devices can be binned by lasing threshold and spatially arranged into regular arrays on a

secondary substrate. Repeated measurements on both unprinted and printed device populations show that the multimode nature of the devices allows for variations in measured laser characteristics in both cases. Nevertheless, in cases where the laser modal spectrum is similar in both measurements, the lasing threshold energy densities are strongly correlated, showing that effective binning and device assembly can be achieved. These results suggest that the printing process does not induce any particular detrimental effects in the transferred devices and could be a route to scalable systems assembly from large inhomogeneous distributions of NW devices. Furthermore, by targeting single-mode devices, some of this variability may be mitigated.

**Methods. NW Laser Characterization.** The as-transferred nanowires were located, imaged, and characterized using an automated microscopy platform as previously described.<sup>13,19</sup> In brief, the sample was moved beneath a 60 $\times$  objective lens and imaged using a CCD camera. Nanowires were identified on the substrate using machine vision, and each wire was sequentially imaged. A HeNe laser (632.8 nm) at low power was used to excite each wire, and the low-power photoluminescence was collected using a fiber-coupled spectrometer. For the lasing study, an ultrafast laser pulse (250 kHz,  $\sim$ 150 fs, 620 nm) was produced using an optical parametric amplifier, with the pulse energy controlled using a computer controlled neutral density filter. The pulse was defocused to around 800  $\mu\text{m}^2$  (a circle of 16  $\mu\text{m}^2$  radius) to uniformly excite the wires. Optical excitation light, in the both configurations, was circularly polarized to ensure that excitation polarization effects in the NW devices are minimized. The laser power was continually monitored using a calibrated silicon photodiode. The threshold for each wire was measured by increasing the excitation fluence until a narrow lasing line emerged which was accompanied by an increase in gradient in the integrated emission as a function of excitation energy as described previously.<sup>19</sup> The position (relative to the markers), the lasing threshold, the microscopy, the low-power photoluminescence, and the lasing spectrum were recorded and stored for every nanowire. The process took approximately 30–60 s per nanowire, including threshold measurement. After transfer, the same process was performed. The unprinted wires were relocated and matched to their initial data using the positional relocation approach outlined in ref 20.

**Microtransfer-Printing.** Two main forms of transfer-printing were used in this work. In the first case a large PDMS block was used to release NWs from their growth substrate using a shearing motion when the surface of the block was placed in contact with the NW substrate. The released NWs adhered to the surface of the PDMS block which was subsequently brought into contact with a photopolymer coated quartz disk. The block was sheared against the quartz substrate, thereby releasing the NWs. By controlling the speed of the PDMS translation relative to the quartz disk and applied contact force, the density of released NWs could be roughly controlled. In the second printing process devices selected during the laser measurement and binning stages were identified with positions defined relative to the on-chip registration marks. The orientation of the NWs was measured using a bright-field microscopy image from the characterization measurements. The NWs were then defined relative to the markers in the  $xy$ -plane and with a relative rotation of their major axis. A second set of markers on the receiver substrate allowed positioning of the NWs in a predefined grid and with

their major axes aligned. The printing process was carried out using a PDMS microblock with a surface area of  $10 \times 30 \mu\text{m}$ . This block was aligned with the NW of interest and brought into contact. The NW was then released from the substrate using a rapid vertical translation of the PDMS block. The NW was then aligned to the receiver substrate and brought into contact. A slow release then allowed transfer of the NW to the receiver substrate.

## ■ ASSOCIATED CONTENT

### Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acs.nanolett.9b05078>.

Additional experimental details and figures including a schematics, optical microscopy image, spatial map, bright-field micrograph, dark-field micrograph, threshold plot, and PL spectra (PDF)

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### Author Contributions

D.J., J.M., B.G., J.A.A., and P.P. carried out the experiments. D.J. and M.J.S. wrote the manuscript with support from all

authors. M.J.S., A.H., B.G., and M.D.D. supervised the project. H.H.T. and C.J. fabricated the GaAs-AlGaAs nanowire lasers. A.H., P.P., and M.J.S. conceived the original idea.

### Notes

The authors declare no competing financial interest.

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### REFERENCES

- (1) Huang, M. H. Room-Temperature Ultraviolet Nanowire Nanolasers. *Science* **2001**, *292*, 1897–1899.
- (2) Yang, Z.; et al. Single-nanowire spectrometers. *Science* **2019**, *365*, 1017–1020.
- (3) Cui, Y. Nanowire Nanosensors for Highly Sensitive and Selective Detection of Biological and Chemical Species. *Science* **2001**, *293*, 1289–1292.
- (4) Takiguchi, M.; Yokoo, A.; Nozaki, K.; Birowosuto, M. D.; Tateno, K.; Zhang, G.; Kuramochi, E.; Shinya, A.; Notomi, M. Continuous-wave operation and 10-Gb/s direct modulation of InAsP/InP sub-wavelength nanowire laser on silicon photonic crystal. *APL Photonics* **2017**, *2*, 046106.
- (5) Quan, L. N.; Kang, J.; Ning, C.-Z.; Yang, P. Nanowires for Photonics. *Chem. Rev.* **2019**, *119*, 9153–9169.
- (6) Güniat, L.; Caroff, P.; i Morral, A. F. Vapor Phase Growth of Semiconductor Nanowires: Key Developments and Open Questions. *Chem. Rev.* **2019**, *119*, 8958–8971.
- (7) Kim, H.; Lee, W.-J.; Farrell, A. C.; Morales, J. S. D.; Senanayake, P.; Prikhodko, S. V.; Ochalski, T. J.; Huffaker, D. L. Monolithic InGaAs Nanowire Array Lasers on Silicon-on-Insulator Operating at Room Temperature. *Nano Lett.* **2017**, *17*, 3465–3470.
- (8) Jevtics, D.; Hurtado, A.; Guilhabert, B.; McPhillimy, J.; Cantarella, G.; Gao, Q.; Tan, H. H.; Jagadish, C.; Strain, M. J.; Dawson, M. D. Integration of Semiconductor Nanowire Lasers with Polymeric Waveguide Devices on a Mechanically Flexible Substrate. *Nano Lett.* **2017**, *17*, 5990–5994.
- (9) Sidiropoulos, T. P. H.; Röder, R.; Geburt, S.; Hess, O.; Maier, S. A.; Ronning, C.; Oulton, R. F. Ultrafast plasmonic nanowire lasers near the surface plasmon frequency. *Nat. Phys.* **2014**, *10*, 870–876.
- (10) Bermúdez-Ureña, E.; Tutuncuoglu, G.; Cuerda, J.; Smith, C. L. C.; Bravo-Abad, J.; Bozhevolnyi, S. I.; i Morral, A. F.; García-Vidal, F. J.; Quidant, R. Plasmonic Waveguide-Integrated Nanowire Laser. *Nano Lett.* **2017**, *17*, 747–754.
- (11) Zhang, C.; Zou, C.-L.; Dong, H.; Yan, Y.; Yao, J.; Zhao, Y. S. Dual-color single-mode lasing in axially coupled organic nanowire resonators. *Science Advances* **2017**, *3*, No. e1700225.
- (12) Alanis, J. A.; Chen, Q.; Lysevych, M.; Burgess, T.; Li, L.; Liu, Z.; Tan, H. H.; Jagadish, C.; Parkinson, P. Threshold reduction and yield improvement of semiconductor nanowire lasers via processing-related end-facet optimization. *Nanoscale Advances* **2019**, *1*, 4393.
- (13) Alanis, J. A.; Saxena, D.; Mokkaapati, S.; Jiang, N.; Peng, K.; Tang, X.; Fu, L.; Tan, H. H.; Jagadish, C.; Parkinson, P. Large-Scale Statistics for Threshold Optimization of Optically Pumped Nanowire Lasers. *Nano Lett.* **2017**, *17*, 4860–4865.
- (14) Guilhabert, B.; Hurtado, A.; Jevtics, D.; Gao, Q.; Tan, H. H.; Jagadish, C.; Dawson, M. D. Transfer Printing of Semiconductor Nanowires with Lasing Emission for Controllable Nanophotonic Device Fabrication. *ACS Nano* **2016**, *10*, 3951–3958.
- (15) Xu, W.-Z.; Ren, F.-F.; Jevtics, D.; Hurtado, A.; Li, L.; Gao, Q.; Ye, J.; Wang, F.; Guilhabert, B.; Fu, L.; Lu, H.; Zhang, R.; Tan, H. H.; Dawson, M. D.; Jagadish, C. Vertically Emitting Indium Phosphide Nanowire Lasers. *Nano Lett.* **2018**, *18*, 3414–3420.
- (16) Saxena, D.; Jiang, N.; Yuan, X.; Mokkaapati, S.; Guo, Y.; Tan, H. H.; Jagadish, C. Design and Room-Temperature Operation of GaAs/

AlGaAs Multiple Quantum Well Nanowire Lasers. *Nano Lett.* **2016**, *16*, 5080–5086.

(17) Norland Optical Adhesive 65. <https://www.norlandprod.com/literature/65Stds.pdf>.

(18) McPhillimy, J.; Guilhabert, B.; Klitis, C.; Dawson, M. D.; Sorel, M.; Strain, M. J. High accuracy transfer printing of single-mode membrane silicon photonic devices. *Opt. Express* **2018**, *26*, 16679.

(19) Alanis, J. A.; Lysevych, M.; Burgess, T.; Saxena, D.; Mokkaapati, S.; Skalsky, S.; Tang, X.; Mitchell, P.; Walton, A. S.; Tan, H. H.; Jagadish, C.; Parkinson, P. Optical Study of p-Doping in GaAs Nanowires for Low-Threshold and High-Yield Lasing. *Nano Lett.* **2019**, *19*, 362–368.

(20) Parkinson, P.; Alanis, J. A.; Peng, K.; Saxena, D.; Mokkaapati, S.; Jiang, N.; Fu, L.; Tan, H. H.; Jagadish, C. Modal refractive index measurement in nanowire lasers—a correlative approach. *Nano Futures* **2018**, *2*, 035004.