

MATERIALS SCIENCE

Nitrogen in black phosphorus structure

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Group V elements in crystal structure isostructural to black phosphorus with unique puckered two-dimensional layers exhibit exciting physical and chemical phenomena. However, as the first element of group V, nitrogen has never been found in the black phosphorus structure. Here, we report the synthesis of the black phosphorus-structured nitrogen at 146 GPa and 2200 K. Metastable black phosphorus-structured nitrogen was retained after quenching it to room temperature under compression and characterized in situ during decompression to 48 GPa, using synchrotron x-ray diffraction and Raman spectroscopy. We show that the original molecular nitrogen is transformed into extended single-bonded structure through gauche and trans conformations. Raman spectroscopy shows that black phosphorus-structured nitrogen is strongly anisotropic and exhibits high Raman intensities in two A_g normal modes. Synthesis of black phosphorus-structured nitrogen provides a firm base for exploring new type of high-energy-density nitrogen and a new direction of two-dimensional nitrogen.

INTRODUCTION

The black phosphorus (BP) structure is widely adopted by group V elements, with nitrogen being the apparent exception. Unlike other group V elements, with atoms connected by single bonds that can then more readily form extended BP-structured solids, nitrogen forms diatomic N_2 molecules held together by a very strong triple bond that is the strongest bond among all known molecules. To transform N_2 molecules into BP-structured nitrogen, enormous energy would be required to convert the $N\equiv N$ triple bond into $N-N$ single bond. This requires extreme conditions to be applied and is very difficult to achieve. However, once BP nitrogen is formed, it would release a huge amount of energy when transforming back to N_2 , making it a powerful high-energy-density material (1–3). In addition, the emergence of the group V element family with two-dimensional (2D) BP crystalline structure (4–7) is of particular interest during the past 5 years. The existence of BP structure in phosphorus, arsenic, antimony, and bismuth allows the synthesis of the BP-structured 2D materials phosphorene (5), arsenene, antimonene, and bismuthine (8), which have highly anisotropic layers (Fig. 1, C to E) being “puckered” in one direction and smooth in the other direction. Obtaining BP structure in nitrogen would complete the set of BP-structured materials in group V and open up the possibility for developing nitrogen-based 2D materials.

The experimental search of BP-structured nitrogen under extreme conditions, which revealed a couple of polymeric nitrogen in other crystal structures, remained unsuccessful. A series of theoretical models (including BP structure) have been proposed for polymeric nitrogen, among which “cubic gauche (CG)”-structured single-bonded

nitrogen was calculated to be the thermodynamic ground state of nitrogen at high pressure (9) and was synthesized experimentally at 110 GPa and 2000 K (10). An additional single-bonded nitrogen phase in the layered polymeric (LP) structure was found to coexist with the CG-nitrogen over a similar pressure-temperature (P - T) range (11). LP-N has a crystal structure close to a previously predicted *Pba2* structure (12) and exhibits interesting properties such as large Raman cross section and high stiffness. Very recently, a hexagonal layered polymeric nitrogen (HLP-N) was identified above 180 GPa (13). Despite these discoveries of single-bonded nitrogen, BP-structured nitrogen has remained elusive. Analogous to the black, white, and red phosphorus polymorphism, single-bonded nitrogen at high pressure may also have multiple polymorphs that are very close in energy, and the BP-structured nitrogen may coexist with other polymorphs and require very specific P - T conditions to synthesize and to distinguish experimentally.

In the present study using laser-heated diamond anvil cell (DAC) techniques, we conducted an in-depth search for BP-structured nitrogen in a P - T range, where other single-bonded nitrogen polymorphs have been reported. Through diagnostic probes of single-crystal x-ray diffraction (SXRD) and Raman spectroscopy, BP-structured nitrogen is observed unambiguously to coexist with CG-nitrogen at 146 GPa and >2200 K. BP-structured nitrogen and CG-nitrogen are crystallized in micrometer-sized crystals, and BP-structured, nitrogen-rich regions can be identified by XRD (Fig. 1 and Table 1) and Raman spectroscopic microprobes (Fig. 2). In addition, the newly developed high-pressure multigrain SXRD method (14) has been applied to the spotty pattern of BP-structured nitrogen (table S1) to unequivocally separate BP-structured nitrogen from CG-nitrogen. Unlike CG-nitrogen, BP-structured nitrogen has only two-thirds of the single bonds in gauche conformation with the others in trans. Notably, BP-structured nitrogen exhibits high Raman intensity as anticipated from the anisotropic nature of the layered structure.

RESULTS AND DISCUSSION

Sample 1 was laser heated at 146 GPa. At this pressure, the sample becomes dark (fig. S1B) and can absorb the 1064-nm laser radiation directly. The temperature was increased to 2200 K at several sample positions. Once the phase transformation was complete, the heated

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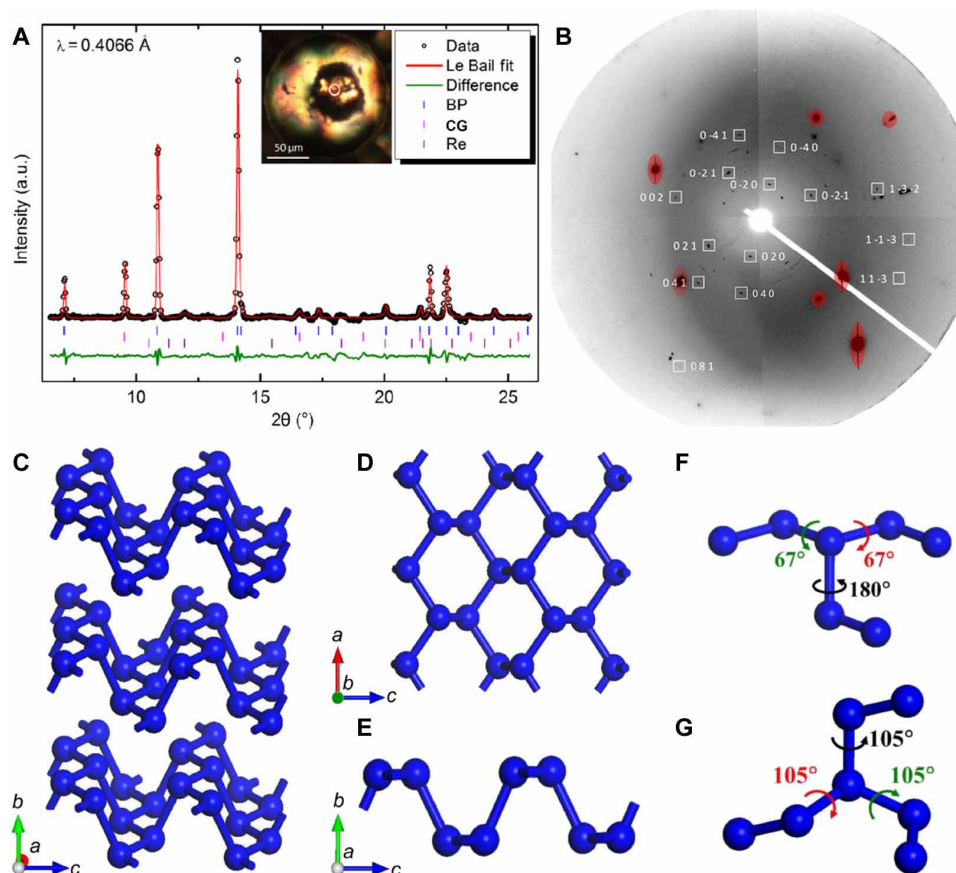


Fig. 1. XRD characterization and illustration of the crystal structure of BP-structured nitrogen. (A) Le Bail fit of XRD data at 132 GPa. Black circles, red line, and green line represent experimental data, Le Bail fitting, and residual, respectively. Blue, pink, and purple ticks mark reflections for BP-structured nitrogen, CG-nitrogen, and Re, respectively. Inset shows the microscopic image of the sample under illumination with both reflected and transmitted light, with the white circle marking the sample position being measured. BP and CG stand for BP-structured nitrogen and CG-nitrogen, respectively. (B) 2D x-ray diffraction (XRD) image from grain #2. White boxes mark the diffraction spots of BP-structured nitrogen. Numbers indicate corresponding Miller indices. Image is obtained by merging selected step scan patterns that contain BP-structured nitrogen reflections for demonstration. (C) Crystal structure of BP-structured nitrogen. (D and E) Single nitrogen layer in BP-structured nitrogen. (F and G) Dihedral angles in BP-structured nitrogen and CG-nitrogen calculated at 132 GPa, respectively. a.u., arbitrary units.

spot immediately became transparent and ceased to absorb the laser. Both the Raman spectrum and XRD (by oscillating the DAC from -21° to 21° about the Ω axis, which is the vertical goniostat axis) measurements after laser heating at a series of sample positions indicate that the transformed regions contain a mixture of CG-nitrogen and a new phase (fig. S2). The spatial distribution of the two phases was resolved by 2D XRD contrast imaging of the entire sample using a submicrometer-focused x-ray probe, as shown in fig. S1 (C and D).

The unit cell parameters and symmetry of the BP-structured nitrogen phase were determined using SXRD with the high-pressure multigrain method, which can uniquely select the diffraction spots of a single crystal among multiple crystals and determine its corresponding orientation (14). One sample region with quite spotty XRD pattern was selected for the SXRD data collection (inset of Fig. 1A). The data were collected using 36-keV x-ray beam while rotating the DAC from -21° to 21° by a step of 0.5° . A total of 23 independent reflections covering a broad spectrum of d -spacing, from 3.28 to 0.65 Å (Table 1), were observed and indexed by the BP structure (orthorhombic in the $Cmce$ space group), confirming unequivocally the successful synthesis of BP-structured nitrogen. The polycrystalline XRD pattern, which integrates all XRD spots and lines, collect-

ed at 132 GPa does not show any unassigned reflections other than these belonging to BP-structured nitrogen, CG-nitrogen, and Re (Fig. 1A). Individual diffraction spots were further indexed on the basis of the angles between each pair of diffraction spots in reciprocal space. Because of the weakness in intensity and scarcity of the diffraction spots (41 identified in total), a manual search and match process had to be performed. In this search, we were able to index two different grains with 19 spots (grain #1) and 14 spots (grain #2), respectively. Figure 1B shows the diffraction spots of grain #2 with corresponding indices on the raw image (see fig. S3 for grain #1 and table S1 for more details regarding the indexing). Thus, BP-structured nitrogen structure explains the XRD data both in extinction rule and angular relationships between diffraction spots in reciprocal space. The obtained unit cell parameters of BP-structured nitrogen at 140 GPa are $a = 2.1438(6)$ Å, $b = 6.5554(8)$ Å, and $c = 2.8534(3)$ Å, which agree well with the calculated values using density functional theory (DFT), $a = 2.1399$ Å, $b = 6.5577$ Å, and $c = 2.8255$ Å. This excellent agreement supports the identified BP crystal structure in which N atoms in the puckered 2D layers are arranged in zigzag (a axis) and armchair (b axis) forms and the layers are stacked along the normal (c axis) direction (Fig. 1, C to E).

Table 1. List of 23 reflections indexed with the BP-structured nitrogen structure at 140 GPa. d_{obs} and d_{cal} represent the d -spacing of observed and calculated reflections, respectively. d_{cal} was obtained using unit cell parameters $a = 2.1438 \text{ \AA}$, $b = 6.5554 \text{ \AA}$, and $c = 2.8534 \text{ \AA}$.

#	hkl	d_{obs} (Å)	d_{cal} (Å)	$d_{\text{obs}} - d_{\text{cal}}$ (Å)
1	0 2 0	3.2779	3.2777	0.0002
2	0 2 1	2.1531	2.1521	0.0010
3	1 1 1	1.6606	1.6582	0.0024
4	0 4 0	1.6383	1.6389	-0.0006
5	0 0 2	1.4269	1.4267	0.0002
6	0 4 1	1.4212	1.4211	0.0001
7	1 3 1	1.3475	1.3486	-0.0011
8	1 1 2	1.1678	1.1687	-0.0009
9	2 0 0	1.0733	1.0719	0.0014
10	1 5 1	1.0421	1.0414	0.0007
11	2 2 0	1.0197	1.0188	0.0009
12	2 2 1	0.9597	0.9595	0.0002
13	0 2 3	0.9146	0.9135	0.0011
14	1 1 3	0.8616	0.8619	-0.0003
15	2 4 1	0.8547	0.8558	-0.0011
16	0 4 3	0.8230	0.8226	0.0004
17	0 8 1	0.7874	0.7876	-0.0002
18	2 6 0	0.7646	0.7652	-0.0006
19	1 7 2	0.7347	0.7354	-0.0007
20	1 5 3	0.7245	0.7246	-0.0001
21	0 6 3	0.7178	0.7174	0.0004
22	2 2 3	0.6944	0.6953	-0.0009
23	2 8 0	0.6517	0.6510	0.0007

BP-structured nitrogen was also identified from the measured Raman spectrum. We synthesized a pure CG sample (sample 2) at 120 GPa as a reference (see the Supplementary Materials for details of the Raman spectrum of sample 2) and used its Raman spectrum as the reference to identify CG peaks. We are able to identify four Raman peaks uniquely belonging to BP-structured nitrogen in sample 1 at 138 GPa (Fig. 2A). Group theory analysis indicates that the BP-structured nitrogen has six Raman active modes, $2A_g$ (named A_g and A_g') + B_{1g} + $2B_{2g}$ (named B_{2g} and B_{2g}') + B_{3g} . The four observed peaks at 796, 830, 993, and 1296 cm^{-1} can be assigned sequentially to B_{1g} , B_{2g} , A_g , and A_g' modes, respectively. The calculated frequencies (799, 818, 977, and 1291 cm^{-1}) of these modes agree very well with the measurements (Fig. 2A). B_{3g} and B_{2g}' with calculated frequencies of 1431 and 1053 cm^{-1} are not observed. At this pressure, the B_{3g} peak is buried by the Raman signal of the stressed diamond anvil, which moves out the diamond Raman region and becomes visible at lower pressures (fig. S4). The B_{2g}' has negligible theoretical intensities, i.e., <1% of the intensity of the A_g mode, and therefore not observable. A weak peak is observed at 1132 cm^{-1} , which is not able to be assigned with either CG-nitrogen or BP-structured nitrogen and could be induced by impurities, defects, or other phases of nitrogen. The concurrent appearance of multiple orientation-dependent

Raman peaks indicates that the measured sample is likely polycrystalline, given the laser spot size being $\sim 5 \mu\text{m}$.

The pressure dependences of XRD and Raman shift of BP-structured nitrogen were measured during decompression from 138 to 48 GPa at 300 K. The lattice parameters and unit cell volumes obtained from the XRD show an excellent agreement with the calculations (Fig. 3, A and B). A unique feature of the BP family of 2D materials is the in-plane anisotropy (15), and BP-structured nitrogen exhibits very strong in-plane anisotropic compressibility. In this range of decompression, the in-plane zigzag direction (a axis) expands 2.0% and the armchair direction (c axis) expands 7.3%, which is almost as large as the b axis, with 7.4% expansion in the layer stacking normal direction. The bulk modulus (B_0) and unit cell volume (V_0) at ambient pressure are obtained from fitting the second-order Birch-Murnaghan equation of state, which are $B_0 = 183.0 \text{ GPa}$ and $V_0 = 55.66 \text{ \AA}^3$ for BP-structured nitrogen, and $B_0 = 254.3 \text{ GPa}$ and $V_0 = 52.87 \text{ \AA}^3$ for CG-nitrogen. BP-structured nitrogen has a similar density to CG-nitrogen but becomes 1.5% denser than CG-nitrogen at the high pressure of 128 GPa, indicating pressure stabilization of BP-structured nitrogen relative to CG-nitrogen.

The measured Raman shifts of BP-structured nitrogen are compared with the calculated and previously reported experimental values (Fig. 3C). All the observed Raman peaks of BP-structured nitrogen soften under decompression with the largest softening in the A_g mode. The calculated frequencies of the B_{1g} , B_{2g} , and A_g' modes are in good agreement with the measurements, while that of the A_g mode shows a notable deviation below 100 GPa. In BP-structured nitrogen, all Raman modes, except A_g , have the eigenvectors parallel to the basal plane. A_g is the only mode that vibrates perpendicular to the basal plane. The exaggerated softening in A_g indicates, possibly, an insufficient treatment of van der Waals interactions between the layers in the calculation. Raman signals from both BP-structured nitrogen and CG-nitrogen were observed down to 48 GPa, below which sample escaped due to failure of a diamond anvil. Our calculation shows that BP-structured nitrogen is not dynamically stable at ambient pressure. Auxiliary confinements, such as nanotube (16) in the case of CG-nitrogen, or doping is anticipated to preserve BP-structured nitrogen under ambient conditions.

We calculated the Raman spectra for BP-structured nitrogen and CG-nitrogen using the Placzek approximation (see the Supplementary Materials for details), which essentially reproduces the experimental spectrum of sample 1 (Fig. 2A). Phonon analysis shows that the A_g/A_g' mode has a puckering/stretching motion of the basal plane with the eigenvector along the normal/armchair direction (Fig. 2, B and C). Both modes have a diagonal Raman tensor, i.e., $(R_{xx}, 0, 0)$, $(0, R_{yy}, 0)$, and $(0, 0, R_{zz})$, which is effective when the polarizations of incident and scattered radiations are collinear. The Raman tensors for both the A_g and A_g' modes are strongly anisotropic, with a very large R_{zz} (armchair) compared with R_{xx} (zigzag) and R_{yy} (normal). The $R_{xx}:R_{yy}:R_{zz}$ ratios are 1:1.3:8.3 for A_g and 1:0.5:13.5 for A_g' . The large intensities of the two A_g peaks therefore arise from excessively large R_{zz} components. As a comparison, the A mode in CG-nitrogen has an isotropic Raman tensor ($R_{xx} = R_{yy} = R_{zz}$), the magnitude of which is only 3.3% of the R_{zz} in A_g and 4.4% of the R_{zz} in A_g' , which explains the low Raman signal of CG-nitrogen in the calculated spectrum, as illustrated in Fig. 2A (the Supplementary Materials include more details regarding the calculated Raman tensors). This finding confirms that the polarizability in BP-structured nitrogen changes most rapidly along the A_g coordinate and is highly active to the incident

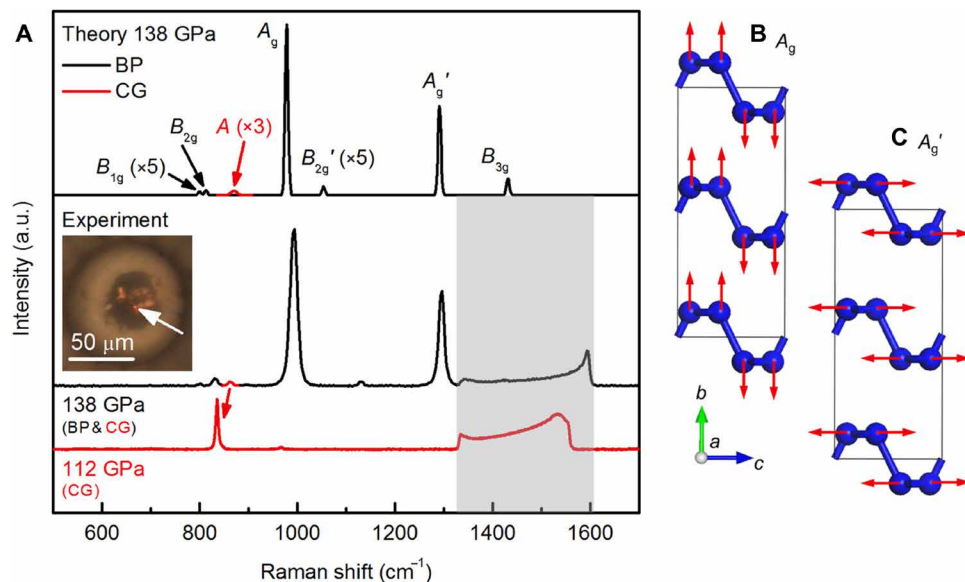


Fig. 2. Identification of the BP structure by Raman spectrum. (A) Raman spectra from sample 1 measured at 138 GPa and sample 2 measured at 112 GPa compared with the superposition of two calculated spectra for BP-structured nitrogen (black) and CG-nitrogen (red) at 138 GPa. Gray shade marks the region of phonon from stressed diamond anvil, which hides B_{3g} of BP-structured nitrogen. Inset shows photomicrograph of the sample, with the white arrow marking the sample position being measured. A mode of CG-nitrogen at 112 GPa shifts to a lower frequency compared with the spectrum at 138 GPa. Red arrow between the experiment spectra correlates the two A modes of CG-nitrogen at different pressures. (B and C) Schematic of atomic displacements of the A_g and A_g' modes.

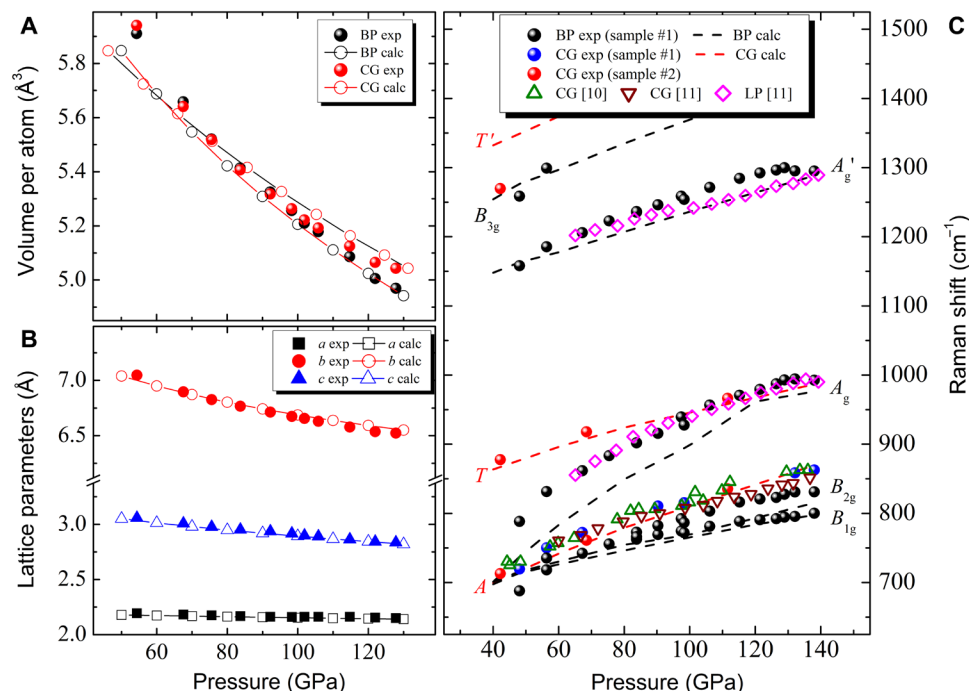


Fig. 3. Measured and calculated pressure-dependent evolution of unit cell volume, lattice parameters, and Raman phonons of BP-structured nitrogen. (A) Unit cell volumes of BP-structured nitrogen and CG-nitrogen. (B) Lattice parameters a , b , and c of BP-structured nitrogen. (C) Measured Raman shift compared with calculations and previously reported values (10, 11). All measurements were performed during decompression. In (C), LP stands for LP-N (11).

radiation with the polarization along the armchair direction. Intuitively, this scenario makes sense since the puckering motion of the layers would be very effective in disturbing the electrons in lone pairs above and below the layers. The calculated relative dielectric con-

stants (ϵ_r) are 6.6, 4.7, and 11.3 along the zigzag, normal, and armchair directions, respectively, showing strong anisotropy. The unique optical response is therefore expected from the anisotropy of the BP structure. However, we like to note that since the sample is mostly

polycrystalline, the experimentally measured intensities of Raman peaks are influenced by the crystal orientation and, therefore, are not directly comparable to the calculated intensities.

We discuss here why the synthesis of BP-structured nitrogen is more difficult from the perspective of conformational isomerism. The buckled layer structure of BP-structured nitrogen (Fig. 1C) is formed by interconnecting the buckled honeycombs where each N atom is bonded to three neighbors. Two distinct directions define the basal plane: the armchair (c axis) and zigzag (a axis) directions (Fig. 1, D and E). The layers are stacked along the normal direction (b axis) through van der Waals interactions. The local structure of BP-structured nitrogen features sp^3 hybridized N atoms, with three σ bonds and one lone pair on each atom, which is the common motif in all single-bonded N phases. Calculated band structure shows that this structure has a semiconducting ground state with a direct band gap of 1.75 eV (150 GPa). The long-range order of BP-structured nitrogen is determined by the dihedral angle within each N_2N - NN_2 conformer. This angle is defined as the angle by which the N_2N group must be rotated about the N-N axis to be located precisely above the NN_2 group. Quantum chemical calculations on isomers of X_2N-NX_2 ($X = H, CH_3$, etc.) shows that the repulsion between lone-pair electrons on adjacent N atoms has a substantial impact on the preferred dihedral angle. To a first approximation, the dihedral angle can be viewed as a property of the N-N bond, which determines the crystal structures of single-bonded nitrogen allotropes (17, 18). BP-structured nitrogen has two-thirds of its bonds in global minimum gauche (DFT values at 132 GPa, $\pm 67^\circ$) and one-third in local minimum trans conformation (180°) (Fig. 1F). The CG-nitrogen have all gauche helicity (DFT value at 132 GPa, 105°) at the same pressure (Fig. 1G). The N-N bond length in trans is 1.45 Å, notably longer than that in gauche (1.33 Å). The integrated crystal-orbital Hamilton population (ICOHP) analysis shows that the gauche bond has higher strength than the trans bond, consistent with the “shorter bond = stronger bond” paradigm. The presence of trans conformers in BP-structured nitrogen, thus, yields a higher internal energy (E) than the all-gauche CG-nitrogen. Following the same logic, the A7 structure, as another structural candidate for high-pressure nitrogen and possessed by group V element As, is formed by an all-trans helicity and therefore thermodynamically more unfavorable. A7-N has never been found in experiments and remains hypothetical.

Our success in obtaining BP-structured nitrogen demonstrates the importance of high pressure and high temperature in synthesizing novel metastable nitrogen allotropes that are energetically less favorable. The calculated enthalpies ($H = E + pV$) of BP-structured nitrogen and other relevant types of nitrogen phases (12, 19, 20) at the ground state show that BP-structured nitrogen never has a region of thermodynamic stability (Fig. 4A). Molecular solid (N_2) and CG-nitrogen are found to be the ground-state phases below and above 56 GPa, respectively, which agrees well with previous calculations (21). However, BP-structured nitrogen is more compressible than CG-nitrogen (Fig. 3A), which means a smaller pV work term to compensate for internal energy at high pressures and a lower vibrational free energy to compensate for the enthalpy at high temperatures. To account for the temperature effects, the vibrational free energy [$F_{\text{vib}} = -\ln(Z)/\beta$] was estimated for BP-structured nitrogen and CG-nitrogen using the harmonic approximation. The calculated finite-temperature energy ($H + F_{\text{vib}}$) shows that BP-structured nitrogen is increasingly more stable at higher temperatures at pressures above 80 GPa (Fig. 4B). The small energy deficiency in BP-structured ni-

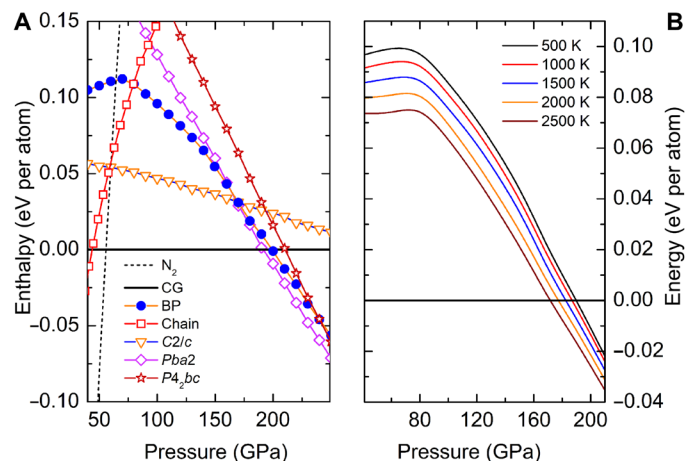


Fig. 4. Calculated enthalpies of BP-structured nitrogen under high pressure at the ground state and at high temperature. (A) Calculated enthalpies as functions of pressure for various structures of solid nitrogen with respect to CG-nitrogen. *Pba2*, *C2/c*, *P4bc*, and chain structures are reported in (12), (20), (13), and (19), respectively. “ N_2 ” denotes the enthalpy of the molecular solid of N_2 , calculated using the lowest-enthalpy structures (21) in respective pressure ranges. (B) Finite-temperature energy ($H + F_{\text{vib}}$) of BP-structured nitrogen relative to CG-nitrogen.

trogen can be overcome with sufficient compression and laser heating, as demonstrated in our experiment. Another predicted metastable (*C2/c*) structure (20) in this pressure region is not observed in our experiment, which probably requires different reaction paths/conditions.

In summary, we synthesized the long-sought-after BP allotrope of nitrogen at high pressure and high temperature. This new phase of nitrogen exhibits large Raman intensities anticipated for the anisotropic nature of the single-bonded layered structure. This is the first confirmation of the existence of the BP structure in nitrogen, the first element in group V, which has not been found to share any crystal structure possessed by other group V elements previously. The discovery of BP-structured nitrogen opens up the possibility of searching for new high energetic materials with clean decomposition products and layer-structured 2D nitrogen.

MATERIALS AND METHODS

Symmetric DAC with beveled diamond anvils (100 μm beveled to 300 μm) were used for pressure generation. Rhenium gaskets, pre-indented to 20- μm thickness approximately, were used to confine the pressurized samples. Sample chambers were fabricated, with initial diameter of approximately 70 μm . A thin flake of gold was loaded near the edge of a chamber for pressure measurement (22). The diamond Raman edge (23) was also measured to check pressures. Ultra-pure nitrogen gas (99.999%) commercially obtained (Airgas Inc.) was loaded into DACs at approximately 0.15 GPa using gas loaders at HPSynC and GeoSoilEnviroCARS (GSECARS), Advanced Photon Source (APS), Argonne National Laboratory (ANL). Laser heating was performed on two samples at 146 GPa (sample 1) and 120 GPa (sample 2) using ytterbium fiber lasers (1064 nm) at HPSynC and 13 IDD, respectively. Sample 1 was heated without a laser coupler at a pressure where nitrogen directly absorbs laser irradiation. Sample 2 was heated at 120 GPa, where nitrogen does not show substantial absorption of the infrared laser. Thus, a thin flake of compressed

graphite powder was used as the laser coupler. SXRD measurements of sample 1 were carried out at 16-IDB of High-Pressure Collaborative Access Team (HPCAT), APS using 3 μm by 6 μm focused monochromatic beam (36 keV) and a Mar165 charge-coupled device (CCD) area detector. 2D XRD contrast imaging was performed at 34-IDE, APS using 500 nm by 500 nm focused monochromatic beam (24 keV) and a Mar165 CCD area detector. The XRD data during decompression were collected at 16-IDB using 3 μm by 6 μm focused monochromatic beam (30.5 keV) and Pilatus 1M area detector. Raman spectra during decompression were measured using a microconfocal Raman system in backscattering geometry at GSECARS, APS, with 660-nm excitation laser, 1200-cm⁻¹ grating, and a CCD camera (24). General Structure Analysis System (GSAS) (25) was used for performing the Le Bail fit. X-ray Diffractive Imaging (XDI) was used to perform the 2D XRD contrast imaging data analysis (26). Dioptas (27) was used for XRD data reduction. Fityk (28) was used to perform peak fitting to subtract individual peak positions from the reduced XRD data and Raman spectra. EoSFit7-GUI (29) was used for fitting the EOS.

Enthalpy and phonon calculations were performed using the Vienna Ab Initio Simulation (VASP) program (30). A five-electron projector-augmented plane wave potential (31) with the Perdew-Burke-Ernzerhof functional (32) was used with a 900-eV kinetic energy cutoff. A k -spacing of $2\pi \times 0.02 \text{ \AA}^{-1}$ was used for Brillouin zone sampling. Raman spectra were calculated using the Quantum ESPRESSO package (33), with norm-conserving pseudopotential and an energy cutoff of 80 rydbergs, along with a $12 \times 12 \times 16$ k -point mesh. The crystal-orbital Hamilton population (COHP) and ICOHP analyses were performed using the LOBSTER program (34). Band structure was calculated using the Heyd-Scuseria-Ernzerhof exchange correlation function (35).

SUPPLEMENTARY MATERIALS

Supplementary material for this article is available at <http://advances.sciencemag.org/cgi/content/full/6/23/eaba9206/DC1>

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