

Disentangling Nano- and Macroscopic Viscosities of Aqueous Polymer Solutions Using a Fluorescent Molecular Rotor

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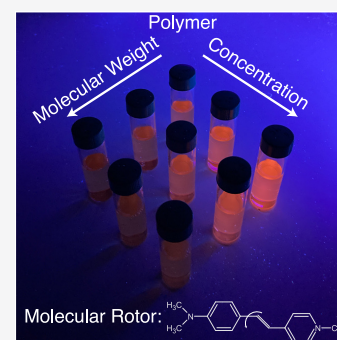
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ABSTRACT: The macroscopic viscosity of polymer solutions in general differs strongly from the viscosity at the nanometer scale, and the relation between the two can be complicated. To investigate this relation, we use a fluorescent molecular rotor that probes the local viscosity of its molecular environment. For a range of chain lengths and concentrations, the dependence of the fluorescence on the macroscopic viscosity is well described by the classical Förster–Hoffmann (FH) equation, but the value of the FH exponent depends on the polymer chain length. We show that all data can be collapsed onto a master curve by plotting the fluorescence versus polymer concentration, which we explain in terms of the characteristic mesh size of the polymer solution. Using known scaling laws for polymers then allows us to quantitatively explain the relation between the FH exponent and the polymer chain length, allowing us to link the nano- to the macroviscosity.



Polymer melts and solutions are omnipresent in everyday life; for instance, in food and cosmetic products polymers are often added as thickening agents since small amounts allow researchers to significantly increase the macroscopic viscosity. However, at the scale of the polymer molecules themselves, the viscosity that molecules or probe particles experience can be very different from the macroscopic one. Brochard-Wyart and de Gennes already asked the question of how to measure and describe the viscosity at small scales in such systems, and they asserted that flows around small colloidal particles or in thin films, capillaries, etc., cannot be understood in terms of the macroscopic polymer viscosity.¹ This takes all of its importance when one realizes that our cells contain biopolymers, and many different types of molecules have to be transported in this crowded environment. Microrheology is a much-used technique for the study of the viscoelastic properties of cells and complex fluids and relies precisely on the measurement of the Brownian motion of small colloidal probes, generally assuming that what they probe are the macroscopic mechanical properties. There is consequently a strong need to understand the viscosity of simple model systems at the scale of individual molecules. We pursue this here with molecular rotors, whose fluorescence depends on the viscosity of the suspending medium. Such molecular rotors are widely used probes for local viscosities in condensed matter.^{2–10} Generically, intramolecular twisting upon photoexcitation enables rapid non-emissive relaxation,^{11–14} but when the twisting is hindered by a high degree of molecular crowding, the energy dissipation mechanism is disabled; this then leads to fluorescence emission from the excited state. The central question we ask here is what happens to the fluorescence of molecular rotors in systems, in which the local nanoscopic viscosity is different from the

macroscopic viscosity. Arguably the best known and characterized complex fluids are aqueous polymer solutions for which it is known that the viscosity reported by conventional macroscopic rheology can be completely different from the nanoscopic viscosity sensed by molecular probes.^{15–27}

We study the fluorescence of the water-soluble hemicyclic molecular rotor *trans*-4-[4-(dimethylamino)styryl]-1-methylpyridinium iodide (4-DASPI)^{28–32} (Figure 1) in aqueous

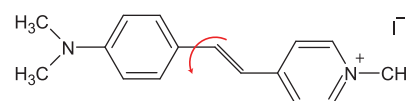


Figure 1. Chemical structure of the molecular rotor 4-DASPI. The red arrow indicates intramolecular rotation.²⁸

poly(ethylene oxide) (PEO) solutions, covering a broad range of concentrations and four orders of magnitude in molecular weight. Our systematic study shows that the local viscosity reported by the molecular rotor is determined by the polymer concentration rather than by the macroscopic viscosity of the solutions. Polymer scaling laws prescribe the relation between the total number of monomers and the polymer correlation length, allowing for a nanoscale interpretation of the

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fluorescence of the molecular rotor in semidilute polymer solutions. The usual Förster–Hoffmann equation relates the fluorescence to the macroscopic viscosity, and we show here a correlation between the fluorescence and the polymer chain length. This then enables us to use the environment-sensitive molecular rotor to not only measure correlation lengths but also to predict the molecular weight of the polymers from the fluorescence data.

Stock solutions of PEO (Sigma-Aldrich) with average molecular weights M (documented as M_n for $M < 10^5$ and M_w for $M > 10^5$ g mol⁻¹) in the range of $[4 \times 10^2$ to $2 \times 10^6]$ g mol⁻¹ in ultrapure water were prepared by gently stirring the mixtures over a week to guarantee complete dissolution. The solutions were then further diluted to cover a wide range of polymer concentrations c (in percentage by weight) and viscosities η . We focus on the semidilute regime as the viscosities are high enough to be distinguishable by molecular rotors and scaling laws are well-established³³ and verified in this regime.^{34,35} The semidilute regime of polymer solutions starts above the overlap concentration, which we estimate as $c^* = M\rho^{-1}/\left(\frac{4}{3}\pi R_g^3 N_A\right)$,¹⁹ where ρ is the solution density, $R_g = 0.02M^{0.58}$ is the radius of gyration, and N_A is Avogadro's number.³⁶ Table 1 lists the calculated c^* values for our

Table 1. Characteristics of the Poly(ethylene oxide) Solutions Studied

M [g mol ⁻¹]	c^* [wt %]	c [wt %]
4×10^2	36.7	[40–100]
6×10^3	7.9	[8–50]
2×10^4	3.3	[4–30]
2×10^5	0.6	[0.8–4]
9×10^5	0.2	[0.2–1]
2×10^6	0.1	[0.1–1]

solutions. 4-DASPI (Sigma-Aldrich, 98% purity, Figure 1) in ultrapure water was added to the polymer solutions to produce

molar concentrations of ≈ 10 μ M. These concentrations guarantee sufficient fluorescence intensity while excluding possible inner filter effects. Macroscopic viscosities were obtained from rheological measurements performed on an Anton Paar MCR 302 rheometer using a cone–plate geometry (diameter 50 mm, angle 1°) at ambient temperature and included preshear and rest periods to establish reproducible and controllable initial states of all samples. To account for non-Newtonian behavior observed in the high molecular weight samples (9×10^5 and 2×10^6 g mol⁻¹),³⁴ we extracted viscosities at low-shear rates in the Newtonian plateau of these samples (see Supporting Information, Figure S1). Prior to each fluorescence measurement, UV–vis absorption spectra were recorded using an HP/Agilent 8453 spectrometer. Fluorescence emission spectra were measured using a SPEX Fluorolog 3-22 fluorimeter, equipped with double grating monochromators. To avoid spectral contamination by incident light, the fluorescence emission spectra were recorded by a photomultiplier at a viewing angle of 90°. All polymer solutions were excited at 488 nm, and the emitted light, corrected for both spectral sensitivity of the detection system and intensity fluctuations of the lamp, was collected between 510 and 700 nm. Absorption spectra were background-corrected and fluorescence intensity compensated for light attenuation using the relation $I = I_{\text{raw}}/(1 - 10^{-A})$ with A the absorbance at the excitation wavelength and I_{raw} the uncorrected fluorescence intensity at its peak wavelength close to 600 nm. All spectra shown in this work were smoothed to increase readability.

Absorption and fluorescence emission spectra of 4-DASPI in aqueous PEO solutions are presented in Figure S2 and Figure 2, respectively. The hemicyanine dye is known to exhibit an absorption maximum at 452 nm in water,²⁸ which we also find for solutions of low polymer concentrations. A moderate redshift at higher concentrations is due to solvatochromism. We found the influence of polarity effects on the fluorescence to be negligible in these systems (see Supporting Information,

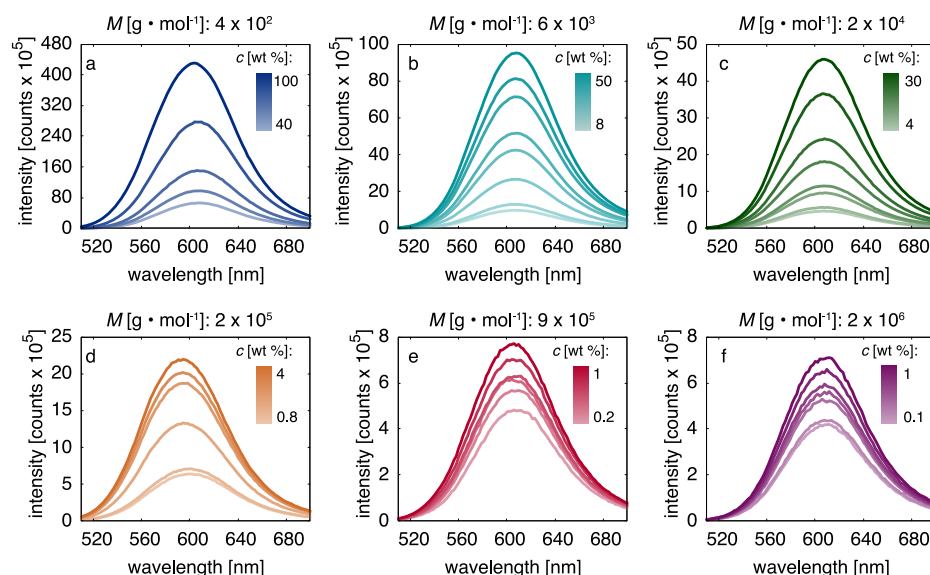


Figure 2. (a–f) Fluorescence emission spectra of 4-DASPI in aqueous PEO solutions of different molecular weights M and concentrations c . At constant dye concentration all samples show a gradual increase in fluorescence intensity with polymer concentration and, thus, solution viscosity. The effect is more pronounced for the lower molecular weight samples (a–c) than for the higher ones (d–f). The corresponding UV–vis absorption spectra can be found in Figure S2.

Figure S3). As shown in Figure 2, the fluorescence intensity increases with polymer concentration for all molecular weights, in a way that is more pronounced for lower molecular weights. The increase in fluorescence intensity is primarily due to an increase in solvent viscosity, preventing the formation of 4-DASPI's twisted intramolecular charge-transfer state.^{11–14} Plotting the absorption-corrected fluorescence intensity at its peak versus the macroscopic viscosity η_{macro} (Figure 3) reveals

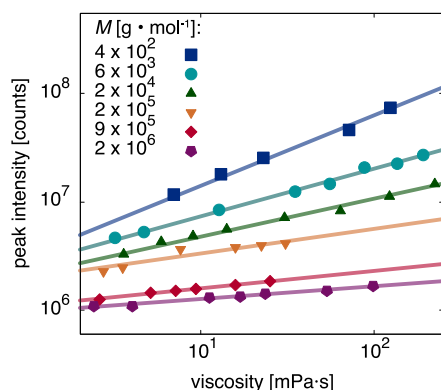


Figure 3. Fluorescence peak intensity versus viscosity of 4-DASPI in aqueous PEO solutions of different molecular weights M and concentrations c . Viscosities were obtained from conventional macroscopic rheological measurements. Lines are obtained by fitting eq 1 to the experimental data.

a different power law behavior for all molecular weights. The conclusion is that the fluorescence is not only sensitive to the macroviscosity (that we fit with the Förster–Hoffmann equation relation^{11,13,14}) but also to the polymer molecular weight. The Förster–Hoffmann fits

$$I = k\eta_{\text{macro}}^x \quad (1)$$

have k and x which are solvent- and dye-dependent parameters,³⁷ and the slope x varies with the molecular weight.³⁸

We find that, as the molecular weight increases from 4×10^2 to 2×10^6 g mol⁻¹, x decreases from 0.65 ± 0.05 to 0.12 ± 0.01 (Figure 3). Our upper value agrees well with the theoretical value¹¹ of $2/3$, which has also been reported for 4-DASPI in water/glycerol mixtures.³⁰ This means that for the smallest molecular weight, the solution behaves like a simple homogeneous fluid, but for polymers that are much larger than the probe size, the probe “feels” an effective viscosity that is between that of the solvent and the macroscopic viscosity.

By plotting the fluorescence intensity as a function of polymer concentration c (Figure 4), we find all data points to collapse onto a single master curve. This indicates that the local viscosity reported by 4-DASPI scales with the total number of monomers rather than with the macroscopic viscosity of the solutions. Our measurements are in the semidilute regime, which de Gennes described as a network of polymer chains with a characteristic mesh size ξ , corresponding to the screening distance of excluded volume interactions.³³ The correlation length or “blob” size is independent of the molecular weight and thus varies only with the polymer concentration, $\xi \propto c^\gamma$, where γ is a solvent-dependent parameter.³³ It is therefore tempting to see whether the rotors probe the fluid properties at this nanoscopic length scale. De Gennes indeed postulated that the nanoscopic viscosity η_{nano}

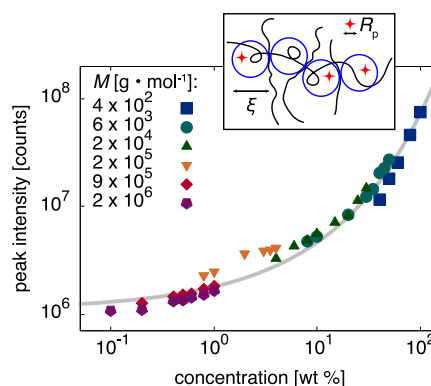


Figure 4. Fluorescence peak intensity versus polymer concentration of 4-DASPI in aqueous PEO solutions of different molecular weights M and concentrations c . The data fall onto a master curve, which is fitted by eq 2 (gray line). The inset shows an illustration of the partition of 4-DASPI probe molecules of size R_p (red stars) within PEO “blobs” of size ξ (blue circles).

experienced by a nanoprobe of radius R_p is a function depending only on R_p/ξ ,³³ which was subsequently verified experimentally.¹⁶ This is also in line with our measurements; Figure 4 shows that the data can be fitted using the stretched exponential suggested by polymer theory^{1,20–22,39} and experiments:^{15,23,40}

$$I = I_0 \exp(bc^a) \quad (2)$$

where $I_0 = 1.10 \times 10^6$ counts is the measured fluorescence intensity of 4-DASPI in pure water and b and a are fitting parameters ($b = 0.49 \pm 0.03$ and $a = 0.46 \pm 0.01$ for the fit of Figure 4). This nanoscopic interpretation has some interesting implications. Given the proportionality $c \propto \xi^{1/\gamma}$ and the independence of ξ on the molecular weight, molecular rotors not only probe local viscosities within polymer solutions but also yield direct information on the characteristic length scales of the semidilute polymer solutions, ξ (Figure 4, inset). From a molecular viewpoint, we conclude that an increase in the total number of PEO monomers results in a decrease of the correlation length or “blob” size and thus exposes the molecular rotor to a higher degree of confinement on a nanoscopic scale. This confinement then restricts the formation of 4-DASPI's twisted intramolecular charge-transfer state and increases the fluorescence.

The remaining question is then how to connect the Förster–Hoffmann eq 1 to our molecular picture and relate the fluorescence intensity to both the macro and the nanoscopic viscosity. To do so, we first express c in the stretched exponential in terms of the macroscopic viscosity using the scaling law $\eta_{\text{macro}} \propto M^\alpha c^\beta$.³⁵ Using $\alpha = 1$ and $\beta = 2$ in the range of our semidilute PEO solutions^{34,35} (see Supporting Information, Figure S4), we find

$$I = I_0 \exp(c_1 M^{-a/2}) \eta_{\text{macro}}^{(c_2 M^{-a/2})} \quad (3)$$

where c_1 and c_2 are constants (see Supporting Information). Equation 3 indeed has the shape of Förster–Hoffmann's eq 1 and indicates that the slope x is related to the molecular weight by a power law, $x \propto M^{-a/2}$. Compared with our measurements (Figure 5), we find a good agreement with this theoretical prediction, providing strong evidence for our molecular picture.

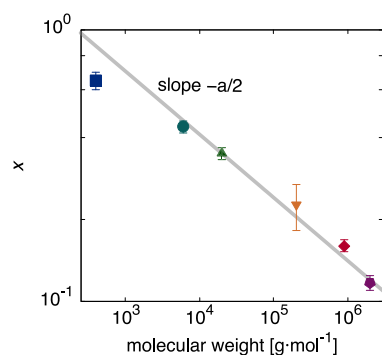


Figure 5. Förster–Hoffmann parameter x as a function of the molecular weight M . The gray line is the predicted slope from eq 3.

In conclusion, considering a molecular interpretation of the molecular rotors residing in “blobs” of polymers in the semidilute regime, we disentangled the effect of polymer molecular weight and monomer concentration on the fluorescence. We also derived a quantitative relation between the exponent of the empirical Förster–Hoffmann power law and well-established scaling laws for the viscosity in polymer solutions. Our observations contribute to the understanding of molecular rotors in systems exhibiting large deviations between macroscopic and nanoscopic viscosities, and they allow us to better understand the small-scale viscosity experienced by molecules in crowded systems such as polymer solutions.

■ ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acs.jpcllett.1c00512>.

Rheological measurements, UV–vis spectroscopy, effect of polarity, scaling laws for the viscosity of PEO solutions, derivation of Förster–Hoffmann’s relation from the stretched exponential (PDF)

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Notes

The authors declare no competing financial interest.

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