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Highly efficient Ag doped δ -MnO₂ decorated graphene: Comparison and application in electrochemical detection of H₂O₂



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ABSTRACT

Keywords: Nonenzymatic sensor δ-MnO₂ Ag NPs H₂O₂ Ternary nanocomposite Cytotoxic H₂O₂ is an inevitable part of our life, even during this contemporary pandemic COVID-19. Personal protective equipment of the front line fighter against coronavirus could be sterilized easily by H₂O₂ for reuse. In this study, Ag doped δ -MnO₂ nanorods supported graphene nanocomposite (denoted as Ag@ δ -MnO₂/G) was synthesized as a nonenzymatic electrochemical sensor for the sensitive detection of H₂O₂. The ternary nanocomposite has overcome the poor electrical conductivity of δ -MnO₂ and also the severe aggregation of Ag NPs. Furthermore, δ -MnO₂/G provided a rougher surface and large numbers of functional groups for doping more numbers of Ag atoms, which effectively modulate the electronic properties of the nanocomposite. As a result, electroactive surface area and electrical conductivity of Ag@ δ -MnO₂/G increased remarkably as well as excellent catalytic activity observed towards H₂O₂ reduction. The modified glassy carbon electrode exhibited fast amperometric response time (<2 s) in H₂O₂ determination. The limit of detection was calculated as 68 nM in the broad linear range (0.005–90.64 mM) with high sensitivity of 104.43 μ A mM⁻¹ cm⁻². No significant interference, long-term stability, excellent reproducibility, satisfactory repeatability, practical applicability towards food samples and wastewater proved the efficiency of the proposed sensor.

1. Introduction

Hydrogen peroxide (H₂O₂) is one of the most significant and versatile analytes available in our daily life [1,2]. It is widely used as a disinfecting agent in food safety and medical care [1,3], as a bleaching agent in the textile, wood, and paper industry [4], and as an oxidizing agent in environmental systems [5]. Recently, the Food and Drug Administration of the United States suggested sterilizing N95 respirator masks and personal protective equipment aerosolized by 35% concentrated vapor phase of H₂O₂ for the safety of doctors and their assistants during this contemporary pandemic COVID-19 [6]. On the other hand, H₂O₂ is also well known as a cytotoxic agent due to its reactive oxygen species (ROS) producing capability [7]. Excess secretions of ROS through abnormal production of H₂O₂ in the human body could damage cell components [1] and play a vital role in diseases mechanism [8]. So, H_2O_2 is highly responsible for some serious diseases such as cancer [9], inflammation [10], neurodegeneration [11], DNA damage [12], heart attack [1], Alzheimer's [13], Parkinson's [14], cardiovascular diseases [15], etc. Furthermore, H₂O₂ could easily contaminate the environment, it has adverse effects on the ecosystem also [16]. Though H_2O_2 is harmful to our health and environment, but we could not imagine a single day without it now a day. Because of significant applications of H_2O_2 , an accurate, reliable, sensitive, rapid, and low-cost detection method developing is of great consideration [17]. Among several analytical detection methods of H_2O_2 , electrochemical sensing is considered as the best technique because of its low cost, fast response, wide linear range, high sensitivity, better selectivity, low detection capability, and good reproducibility [18-20]. Owing to sluggish reaction kinetics, bare glassy carbon electrodes (GCE) provide a negligible current response towards H_2O_2 reduction [21]. Modification of GCE is necessary to enhance the large surface area to volume ratio, high conductivity, and fast response towards H_2O_2 [22]. Recently, non-enzymatic metal and metal oxide nanomaterials H_2O_2 sensors attracted much attention due to their low limit of detection, wide linear range, and long-term stability [3,23].

Metal oxides are favorable for the synthesis of electrochemical sensors due to their chemical stability, good catalytic activity, outstanding ion exchange capability, and environmental compatibility [24]. Among various metal oxides, manganese oxide (MnO₂) has been specifically chosen due to its non-toxicity, low cost, natural abundance and its better

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catalytic activity towards H₂O₂ [9,25]. The different crystallographic polymorphs of MnO₂ (α , β , γ , δ , λ , and ε) depend on the presence of fundamental [MnO₆] octahedral units in its face, edge, and corner [24,26]. Among them, δ -MnO₂ demonstrates a two-dimensional (2D) layer-type structure with [MnO₆] octahedral unit at the edge and provides the highest electrochemical performances due to its higher interlayer distance (\sim 7.0 Å) and largest tunnel size [26,27]. The significant physicochemical characterizations of δ -MnO₂ offer cation vacancies [27]. On the other hand, the poor electrical conductivity $(10^{-5}-10^{-6} \text{ S})$ $\rm cm^{-1})$ of $\rm MnO_2$ is the main obstacle to its broad applications in electrochemical sensors [28]. This problem could be solved by loading MnO₂ on highly conductive material such as graphene (G); which is one of the most remarkable substances in electrochemical sensors due to better electrical conductivity (550 S cm⁻¹), excellent thermal conductivity (up to 5300 W m⁻¹ K⁻¹), large surface area (2630 m² g⁻¹), high charge carrier mobility (200,000 $\text{cm}^2 \text{ V}^{-1}$ s), and a large number of electrochemically favorable carbon edges [3,29].

The combination of δ -MnO₂ and G provides a rougher surface and large numbers of functional groups for doping of heteroatoms, which could play an important role to facilitate electron transportation, increasing electron density, creating p- or n-junctions and tuning band gap [30,31]. Doping of pure metal as a heteroatom on the defect surface of δ-MnO₂/G can effectively modulate their physical and electronic properties and demonstrate the synergistic effect on the electrochemical performances [30]. A short number of metal doping reports are available for electrocatalysis due to its complex creation and tough acceptance [31]. Among different metal nanoparticles (NPs), silver NPs (Ag NPs) have been significantly reported as an effective H₂O₂ sensor due to large surface area, high conductivity, good biocompatibility, and low toxicity [1,5,18,32,33]. But Ag NPs suffer from severe aggregations because of the strong van der Waals force and high surface energy of nano-sized particles [5,31]. The δ -MnO₂/G surface could play a synergistic effect for the uniform distribution and effective size and shape of Ag NPs [31]. According to quantum chemical calculation, Mn doped can weaken the O-O bond and improve Ag catalysis performance, and Ag doped can enhance the adsorption of O₂ by MnO₂ [34].

Based on the above discussion, the limitations of δ -MnO₂ could be exceeded by the combination of G. The prepared substrate (δ -MnO₂/G) would be suitable for decreasing the aggregation of Ag NPs. And doping of Ag-atom on the defect surface of δ -MnO₂/G will enhance their physical and electronic properties.

2. Experimental

2.1. Synthesis of $Ag@\delta-MnO_2/G$

In this study, we synthesized Ag doped δ -MnO₂ nanorods (NRs) supported graphene nanocomposite (denoted as Ag@ô-MnO2/G) in a two-step facile route. In the first step, δ -MnO₂ nanorods was synthesized in a simple hydrothermal process, and a perfectly oriented structure of δ -MnO₂ (Scheme 1b) was obtained. GO was obtained by modified Hummers' method from graphite. In the second step, 30 mL of 5 mM AgNO₃ aqueous solution was added into the as-prepared homogeneous solution of GO (30 mg) and δ -MnO₂ (30 mg) slowly (1 mg mL⁻¹), followed by the dropwise addition of 1% NaBH₄ (450 µL). Then the mixture was refluxed at 110 °C for 6 h, after that kept under magnetic stirring for further 18 h at room temperature. The final product Ag@δ-MnO₂/G was obtained after centrifugation several times with distilled water and ethanol, and drying at 60 °C for 72 h in a vacuum oven. In the second step, δ -MnO₂ offer cation vacancies (Scheme 1c), and δ -MnO₂/GO provided a rougher surface area and a large number of functional groups for absorption of heteroatom (i.e. Ag). When silver precursor was added, Ag ions were absorbed by electrostatic interaction into the δ -MnO₂/GO and NaBH₄ converted it to metallic Ag, which also convert GO to RGO. Subsequently, Ag atoms occupied the created vacancies of δ -MnO₂, and Ag@ô-MnO₂/G achieved higher structural benefits (Scheme 1d). Reagents, instrumental characterization and fabrication of working electrodes are discussed into the supporting information.

3. Results and discussions

3.1. Morphology analysis

Transmission electron microscopy (TEM) images demonstrated the surface morphology and defects of the as-prepared samples for analysis in Fig. 1. Fig. S1a displayed nanorods (NRs) like morphology of δ -MnO₂ on the graphene surface in δ -MnO₂/G sample. High-resolution TEM (HRTEM) of δ -MnO₂/G revealed that the interlayer spacing of 0.69 nm, corresponding to the (001) plane of δ -MnO₂ (Fig. S1b) [24,35]. Fig. 1a illustrated homogeneous distribution of spherical Ag NPs on the graphene surface with average particle size of 11.18 ± 0.1 nm. HRTEM of Ag/G provided interlayer spacing of 0.24 nm, corresponding to the (111) lattice plane of pure Ag (Fig. S1c) [36]. Fig. 1b shows the TEM



Scheme 1. (a) basic structure of $[MnO_6]$ octahedral unit; schematic illustration of (b) perfectly oriented δ -MnO₂ NRs, (c) δ -MnO₂ NRs with Mn vacancies, and (d) δ -MnO₂ NRs with Ag-doping.



Fig. 1. TEM images of (a) pure Ag/G, and (b) Ag@ δ -MnO₂/G; (c) HRTEM image of δ -MnO₂ NRs and Ag NPs in Ag@ δ -MnO₂/G, corresponding magnified HRTEM images of (d) pure Ag/G, and (e and f) Ag@ δ -MnO₂/G in different regions.

image of Ag@ô-MnO2/G, in which numerous number of Ag NPs are distributed on different lengths and diameters of 8-MnO2 decorated graphene surface. The average particle size of Ag NPs in Ag@δ-MnO₂/G was calculated as 7.71 \pm 0.1 nm. HRTEM of Ag@ $\delta\text{-MnO}_2/\text{G}$ exhibited the lattice d-spacing of 0.24 nm, and 0.69 nm corresponding to (111) plane of Ag, and (001) plane of δ -MnO₂, respectively (Fig. 1c) [37]. It is notable that aggregation of Ag NPs was decreased in Ag@ô-MnO2/G than Ag/G. Furthermore, particle size of Ag NPs was smaller in Ag@δ- MnO_2/G than Ag/G, which suggests the doping of Ag on δ -MnO₂/G surface [34]. For the confirmation of doping of Ag on δ -MnO₂/G surface, magnified HRTEM images were analyzed also. The magnified HRTEM image of Ag/G revealed uniformly distribute same size of dark spots without local lattice distortions indicates no impurity (Fig. 1d). Fig. 1e and 1f demonstrated the magnified HRTEM image of Ag@ô-MnO2/G in different regions. Fig. 1e exhibited clean areas due to the regular octahedral structure of δ -MnO₂ NRs, also demonstrated dark spotted areas due to lattice disorder and dislocations of δ -MnO₂ NRs [38,39]. Fig. 1f exhibited thick and ink-like surface areas compared to Fig. 1e, which may be due to the doping of Ag atoms in cation vacancies of δ-MnO₂ NRs [39,40]. EDX spectrum revealed the successful synthesis of Ag@δ-MnO₂/G (Fig. S1d). Furthermore, elemental mapping images of Ag@δ-MnO₂/G portraying the uniform distribution of C, O, Mn, and Ag (Fig. S1 (e-h)).

3.2. Structural analysis

Fig. 2a shows the X-ray diffraction (XRD) patterns of RGO, δ -MnO₂/G and Ag@ δ -MnO₂/G in 2 θ = 10-90°. RGO contains an intense broad diffraction peak (002) and a small diffraction peak (100) of carbon at 2 θ = 24.82° and 43.07°, respectively [41]. δ -MnO₂/G demonstrated four diffraction peaks at 2 θ = 12.31, 25.31, 36.75 and 65.61° corresponding to the (001), (002), (111) and (020), respectively, reflections of δ -MnO₂ (JCDPS card No. 80–1098) [42], and a carbon peak (002) at 2 θ = 24.71° (inset of Fig. 2a), reflection of graphene like structure. After deposition of Ag precursor, the resulted Ag@ δ -MnO₂/G illustrated five

diffraction peaks at $2\theta = 38.13$, 44.39, 65.54, 77.44 and 81.71° corresponding to (111), (200), (220), (311) and (222), reflections of facecentered cubic (fcc) phase of metallic Ag (JCDPS card No. 04–0783) [43]. The slightly positive shift of 2 θ value and higher intensities of diffraction peaks of Ag@ δ -MnO₂/G compared to Ag/G (Fig. S2a), indicating the successful attachment of δ -MnO₂ [43,44]. There was no characteristic diffraction peak for RGO in Ag@ δ -MnO₂/G, due to quite uniform dispersion and more disorder stacking of RGO nanosheet [45].

Fig. 2b exhibited the Raman spectrum of RGO, δ -MnO₂/G and Ag@ δ -MnO₂/G. All samples demonstrated two characteristic D and G bands at \sim 1349 and \sim 1586 cm⁻¹, respectively. The I_D/I_G ratios were obtained as 1.02, 1.06 and 1.08 for RGO, δ -MnO₂/G and Ag@ δ -MnO₂/G, respectively. Representative peaks of δ -MnO₂ (Fig. S2b) at 643 cm⁻¹ and 504 cm⁻¹ corresponding to symmetric stretching vibration of Mn-O of [MnO₆], 565 cm⁻¹ corresponding to symmetric stretching vibration of Mn-O of Mn-O in the basal plane of [MnO₆], and 360 cm⁻¹ corresponding to stretching vibrations of Mn-O-Mn in MnO₂ [42,46]. δ -MnO₂/G contain a peak at 645 cm⁻¹, which shifted to 613 cm⁻¹ in Ag@ δ -MnO₂/G, confirm the presence of MnO₂ in both samples. Ag@ δ -MnO₂/G displayed additional peak at 466, 843 and 1111 cm⁻¹, due to presence of Ag in the nanocomposite [47].

Fig. 2c shows Fourier-transform infrared spectroscopy (FTIR) of the GO, RGO, δ -MnO₂/G, and Ag@ δ -MnO₂/G. FTIR spectrum of GO represents the characteristics peaks related to —OH (3000–3500 cm⁻¹), C=O (1733 cm⁻¹), C=C (1626 cm⁻¹), C—OH (1405 cm⁻¹), C—O-C (1227 cm⁻¹), and C—O (1043 cm⁻¹) [41,48]. The intensities of all functional groups decreased dramatically in RGO, assured the successful reduction of GO. FTIR spectrum of δ -MnO₂/G and Ag@ δ -MnO₂/G contains two intense peaks at 489 and 604 cm⁻¹ due to stretching vibrations of Mn-O, and one sharp peak at 1565 cm⁻¹ due to bending vibration of Mn-OH [49,50]. The –OH peak of δ -MnO₂/G and Ag@ δ -MnO₂/G shifted to 3137 and 3029 cm⁻¹, which confirm the presence of MnO₂ surface [49]. Furthermore, the movement of –OH peak to lower wavelength and decrease of intensities in Mn-O of Ag@ δ -MnO₂/G, compared to δ -MnO₂/G, indicates the successful formation of Ag@



Fig. 2. (a) XRD patterns of RGO, δ -MnO₂/G, and Ag@ δ -MnO₂/G, (b) Raman spectra of RGO, δ -MnO₂/G, and Ag@ δ -MnO₂/G, (c) FTIR spectra of RGO, δ -MnO₂/G, and Ag@ δ -MnO₂/G, with GO as reference, and (d) UV–Vis spectra of RGO, δ -MnO₂/G, Ag/G, and Ag@ δ -MnO₂/G; inset: the magnified XRD region of C(002) in δ -MnO₂/G.

δ-MnO₂/G.

Fig. 2d shows the UV–Vis spectra of RGO, δ -MnO₂/G, Ag/G, and Ag@ δ -MnO₂/G. RGO consists of a sharp π - π^* absorption peak at 260 nm due to aromatic C-C bonds, and a shoulder n- π^* absorption peak at 370 nm due to C = O bonds [51,52]. δ -MnO₂/G demonstrated a broad absorption peak at 350 nm due to δ -MnO₂, and the characteristic absorption peak of RGO shifted slightly to 255 nm and absorption decreased dramatically [52,53]. Ag/G displayed two well define absorption peaks at 260 nm and 400 nm due to RGO and Ag nanoparticles, respectively [54]. UV–Vis spectrum of Ag@ δ -MnO₂/G illustrated all absorption peaks of its synthesized materials and confirmed the successful formation. The absorption peaks at 275, 375 and 400 nm due to RGO, δ -MnO₂ and Ag NPs, respectively.

Fig. 3a shows the X-ray photoelectron spectroscopy (XPS) survey spectrum of RGO, δ -MnO₂/G, and Ag@ δ -MnO₂/G. RGO consists of two prominent signal peaks at 285.08 and 532.08 eV, due to the presence of carbon and oxygen, respectively, which confirmed the presence of graphene in all prepared samples [55]. δ -MnO₂/G and Ag@ δ -MnO₂/G exhibited conjugated peaks in the range of 642.08–654.08 eV, which indicated the existence of Mn2p [56]. Furthermore, Ag@ δ -MnO₂/G displayed conjugated peaks in the range of 368.08–374.08 eV due to Ag3d, and additional two peaks at 573.08 eV and 604.08 eV due to Ag3p [43]. The C/O ratio was calculated as 2.50, 1.25, and 1.52 RGO, δ -MnO₂/G, and Ag@ δ -MnO₂/G, respectively. Following order of C/O ratio: RGO > Ag@ δ -MnO₂/G > δ -MnO₂/G proved that Mn is in oxidized form in δ -MnO₂/G and Ag@ δ -MnO₂/G [44], and degree of Mn oxidation is higher in Ag@ δ -MnO₂/G than that of δ -MnO₂/G [43]. Fig. 3b shows

the high-resolution C1s spectra of RGO, δ -MnO₂/G, and Ag@ δ -MnO₂/G. C1s spectra of RGO was deconvoluted by four identical peaks at 284.58, 286.48, 288.18 and 290.48 eV corresponding to the C-C/C=C, C-O-C, C=O and O-C=O, respectively. The slightly shift of peak positions in binding energy of δ -MnO₂/G and Ag@ δ -MnO₂/G compared with RGO due to the strong electronic effect of carbon and metals [57]. The higher intensity of δ -MnO₂/G at 286.58 eV compared to RGO, due to the abundance of C—O—C in δ -MnO₂. Fig. S3 shows the high-resolution O1s spectra of RGO, δ -MnO₂/G, and Ag@ δ -MnO₂/G. O1s spectra of RGO was deconvoluted by three identical peaks at 532.68, 533.68 and 534.48 eV corresponding to the C=O, C-O, and O-C=O, respectively. The slightly negative shift of peak position in binding energy of $\delta\text{-MnO}_2/$ G and Ag@\delta-MnO₂/G than that of RGO due to the anchoring of positively charged Mn and negatively charged O, and an additional peak at 530.08 eV corresponding to metallic oxide [24]. Fig. 3c illustrated the Mn2p spectra of δ -MnO₂/G and Ag@ δ -MnO₂/G. Mn2p spectra of δ -MnO₂/G exhibited two peaks at 642.38 and 654.08 eV, corresponding to the Mn2p3/2 and Mn2p1/2 spins, respectively, with a typical spin energy separation of 11.7 eV. Whereas the spin energy separation between Mn2p_{3/2} and Mn2p_{1/2} was 11.5 eV in Ag@ δ -MnO₂/G, and this ~ 0.2 eV lower indicates bimetallic alloy zone formation between δ -MnO₂ and Ag in the Ag@ô-MnO2/G [58]. Fig. 3d shows the core-level Ag3d spectrum of Ag@ δ -MnO₂/G with two major peaks located at 368.28 and 374.28 eV for the Ag3d_{5/2} and Ag3d_{3/2} spins, respectively. The spin energy separation of 6.0 eV and no peaks for AgO and Ag₂O, confirms the only presence of metallic Ag (Ag⁰) without any other silver oxide species [43].



Fig. 3. (a) XPS survey spectra, and (b) core level spectra of C1s of RGO, δ -MnO₂/G, and Ag@ δ -MnO₂/G, (c) Mn2p survey spectra of δ -MnO₂/G, and Ag@ δ -MnO₂/G, and (d) Ag3d spectrum of Ag@ δ -MnO₂/G.

3.3. Electrochemical performance of modified electrodes

The electroactive surface area (EASA) and electrical conductivity of the as-prepared samples were investigated by using 5.0 mM [Fe(CN)₆]^{4-/} ³⁻ in 0.05 M KCl. Fig. 4a shows the cyclic voltammetry (CV) curves of bare glassy carbon electrode (GCE), δ -MnO₂/G/GCE, Ag/G/GCE and Ag@ δ -MnO₂/G/GCE at a scan rate of 50 mV s⁻¹ in Ar-saturated environment. All of the modified electrodes provided symmetric redox

peaks, which indicated their charge transfer capability due to redox reactions. Among them Ag@ δ -MnO₂/G demonstrated highest redox peak currents and bare GCE demonstrated lowest redox peak currents. Ag/G/GCE showed higher redox peak currents than that of δ -MnO₂/G/ GCE, revealed that electron transfer ability of δ -MnO₂/G/GCE is lower than Ag/G/GCE because of the presence of MnO₂. Above statement implies that Ag NPs played an important role to enhance the facile electron transfer of Ag@ δ -MnO₂/G/GCE. The EASA of different



Fig. 4. (a) CV curves at a scan rate of 50 mV s⁻¹ and (b) Nyquist plots at a frequency range of 0.1 Hz to 100 KHz of bare GCE, δ -MnO₂/G, Ag/G, and Ag@ δ -MnO₂/G in 0.05 M KCl with 5 mM [Fe(CN)₆]^{4/3-}.

modified electrodes were calculated from the slope of the peak current (I_p) versus the square root of the scan rate ($\nu^{1/2}$) of different scan rates (40–400 mV s⁻¹) (Fig. S4) by using the Randles–Sevcik equation [59]:

$$I_n = 2.69 \times 10^5 A D^{1/2} n^{3/2} \nu^{1/2} C \tag{1}$$

where the number of transferred electrons (n = 1), and diffusion coefficient $(D = 7.60 \times 10^{-6} \text{ cm}^2 \text{ s}^{-1})$ for $[\text{Fe}(\text{CN})_6]^{4-/3-}$, and concentration, C are constant parameters; so, the surface area (A) of the modified electrode is directly proportional to the peak current of the plot. The EASA was obtained as 0.1503 cm², 0.1381 cm², 0.0987 cm², and 0.0707 cm², for Ag@ô-MnO2/G/GCE, Ag/G/GCE, ô-MnO2/G/GCE, and bare GCE, respectively. About 8.14% and 34.33% higher EASA of Ag@ô-MnO2/G/ GCE than that of Ag/G/GCE and δ -MnO₂/G/GCE, respectively, proved that Ag NPs modified the defective surface area of δ -MnO₂ due to better absorption on δ -MnO₂/G surface. The conductivity of bare GCE, δ-MnO₂/G/GCE, Ag/G/GCE and Ag@δ-MnO₂/G/GCE was monitored by electrochemical impedance spectroscopy (EIS). Fig. 4b displays the Nyquist plots of the modified electrodes. Bare GCE showed the bigger semicircle diameter than that of δ -MnO₂/G/GCE, and Ag@ δ -MnO₂/G/ GCE. Ag/G/GCE exhibited almost straight line like plot with smallest semicircle shape. The semicircle diameter was obtained as 1930.17 Ω . 1448.30 Ω , 691.97 Ω , and 317.39 Ω for bare GCE, δ -MnO₂/G/GCE, Ag@δ-MnO₂/G/GCE, Ag/G/GCE, respectively. Usually, conductivity decrease with the increase of semicircle diameter of modified electrodes. Here, Ag@ô-MnO₂/G/GCE provided lower conductivity than that of Ag/ G/GCE, which revealed that semiconductive MnO2 decreased the

conductivity and Ag NPs played an important role to enhance the conductivity of the prepared samples. On the other hand, the conductivity of Ag@ δ -MnO₂/G/GCE was significantly greater compared with δ -MnO₂/G/GCE, which revealed that Ag NPs enhanced the conductivity of the proposed sample [60].

3.4. Electrochemical behaviors of H_2O_2 on fabricated electrodes

Fig. 5 shows the CV curves of different modified electrodes towards 2.0 mM H_2O_2 at a scan rate of 50 mV s⁻¹ in 0.2 M PBS (pH 7.0). Fig. 5a demonstrated the several cycles of bare GCE sweeping from -2.0 to 2.0 V. First cycle of the figure displayed a reduction peak at -0.546 V and no oxidation peak, which indicated a chemically irreversible reduction process. When H_2O_2 introduced in the PBS solution, the reduction mechanism followed the following reactions [61]:

$$H_2O_2 + e^- \leftrightarrow OH_{(ads)} + OH^-$$
 (2)

$$OH_{(ads)} + e^- \leftrightarrow OH^-$$
 (3)

$$OH^- + 2H^+ \leftrightarrow 2H_2O \tag{4}$$

With the increment of cycles, the reduction peak shifted to lower potential and peak current increased. On the other hand, after third cycle, there was a tiny oxidation peak due to the saturation of H_2O_2 [62].

Fig. 5b illustrated the electrochemical performance of Ag@ δ -MnO₂/G/GCE in absence and presence of 2.0 mM H₂O₂. In which there was no electrochemical response in absence of H₂O₂, and provided a



Fig. 5. CV curves of (a) bare GCE in 2.0 mM H_2O_2 (4 cycles), (b) Ag@ δ -MnO $_2$ /G in absence and presence of 2.0 mM H_2O_2 , (c) comparison of bare GCE, δ -MnO $_2$ /G, Ag/G, and Ag@ δ -MnO $_2$ /G in presence of 2.0 mM H_2O_2 , and (d) different concentrations of Ag@ δ -MnO $_2$ /G (0 to 12 mM) in 0.2 M PBS (pH 7.0) at a scan rate of 50 mV s⁻¹; insets: (a) magnified reduction and oxidation peaks, and (d) plot of current vs. [H_2O_2]. Error bars set for 5% of standard error.

remarkable reduction peak at -0.528 V after addition of 2.0 mM H_2O_2 following the irreversible reactions:

$$H_2O_2 \rightarrow \frac{1}{2}O_2 + H_2O$$
 (5)

$$O_2 + 2e^- + 2H^+ \rightarrow H_2O_2 \tag{6}$$

Similarly, bare GCE, 8-MnO2/G/GCE, and Ag/G/GCE showed reduction peak towards H₂O₂, whereas there was no peak in absence of H₂O₂. (Fig. S5). Fig. 5c exhibited CV comparison of bare GCE, δ -MnO₂/ G/GCE, Ag/G/GCE, and Ag@ô-MnO2/G/GCE. The catalytic current density of Ag@ δ -MnO₂/G/GCE was about 77% greater than δ -MnO₂/G/ GCE, which indicated the synergistic effect of Ag NPs on the defective surface of δ -MnO₂/G. Furthermore, about 32% greater current density and 60 mV positive shift of reduction peak potential of Ag@ô-MnO2/G/ GCE than Ag/G/GCE, proved that the δ -MnO₂ enhanced the electrochemical performances of the modified electrodes towards H₂O₂ [60]. Fig. 5d demonstrated the effect of concentration on Ag@δ-MnO₂/G/GCE from 0 to $12 \text{ mM H}_2\text{O}_2$ at a scan rate of 50 mV s⁻¹ in 0.2 M PBS (pH 7.0). There was no characteristic peak in absence of H₂O₂. When H₂O₂ was introduced in PBS solution, due to catalytic current response (I_{pc}) a peak was appeared. The reduction peak increased gradually and peak potential shifted negatively with the increment of $[H_2O_2]$, which indicated the effective electrocatalytic activity of Ag@ δ -MnO₂/G/GCE towards H_2O_2 . The plot of I_{pc} versus $[H_2O_2]$ from 1 to 10 mM provided a linear regression equation: I_{pc} (µA) = 20.81[H_2O_2] (mM) + 19.748, R² = 0.9902 (Fig. 5d inset); which confirmed the direct electro-reduction of H_2O_2 on the surface of Ag@ δ -MnO $_2/G$ nanocomposite. where R² is the correlation coefficient. Similarly, in the differential pulse voltammetry (DPV) curves, the reduction peaks current increased and the peak potential shifted negatively with the increment of $[H_2O_2]$ from 0 to 10 mM (Fig. S6) according to the following linear fit: I_{pc} (μ A) = 6.1679[H_2O_2] $(mM) + 1.0251, R^2 = 0.9945 [100 \ \mu M \text{ to } 10 \ mM] \text{ (inset)}.$

3.5. Kinetic behaviors of $Ag@\delta-MnO_2/G$ electrode

Fig. 6a showed the influence of different pH (5.0 to 9.0) of Ag@ δ -MnO₂/G/GCE towards 2.0 mM H₂O₂ at a scan rate of 50 mV s⁻¹ in 0.2 M PBS. Fig. 6b illustrated the plot of catalytic current (I_{pc}) versus *pH*, and the plot of reduction potential (E_{pc}) versus *pH*. The plot of E_{pc} versus *pH* showed that peak potential decreased with the increased of pH, which indicated the active participation of protons in the electrochemical reaction [63]. The slope of E_{pc} versus *pH* was obtained as -61.4 mV/pH, which matched with the equation (6) for two-electron and two proton process [64]. From the plot of I_{pc} versus *pH* the highest peak current was observed at pH 7.0, therefore pH 7.0 was selected for the further experiment of this study.

Fig. 7a illustrated the effect of different scan rates (20 to 500 mV s⁻¹) of Ag@ δ -MnO₂/G/GCE towards 2.0 mM H₂O₂ in 0.2 M PBS. The catalytic current increased with the increase of scan rate. Fig. 7b demonstrated that the plot of I_{pc} versus square root of scan rate ($\nu^{1/2}$) provided a linear regression equation: I_{pc} (μ A) = 5.6521 $\nu^{1/2}$ + 7.9547, R² = 0.9956; which indicated that diffusion controlled reduction reaction between Ag@ δ -MnO₂/G/GCE and H₂O₂ [39]. Furthermore, E_{pc} versus ln ν (Fig. 7c) also provided a linear regression equation: E_{pc} (V) = -0.0449 ln ν – 0.2957, R² = 0.9956; which confirmed that the electron transfers kinetics of H₂O₂ remained unchanged in wide range of scan rates at the surface of Ag@ δ -MnO₂/G [65].

3.6. Quantitative determination of H_2O_2 at $\delta\text{-Mn}O_2/G$ and Ag@ $\delta\text{-Mn}O_2/G$

Fig. 8a demonstrated the choronoamperometric (CA) current-time curve of Ag@δ-MnO₂/G/GCE and δ-MnO₂/G/GCE at -0.5 V for successive addition of H₂O₂ in every 50 s into stirred 0.2 M PBS (pH 7.0). When there was no H₂O₂ in the system, none of modified electrodes showed any current response. At 50 s, when trace quantity of H₂O₂ was added, both of them showed almost equal current response. But after 150 s, Ag@ô-MnO2/G/GCE showed much higher catalytic current response than δ-MnO₂/G/GCE for each addition of H₂O₂. Fig. S6 implied that Ag@ô-MnO₂/G/GCE reached to a steady-state current within 2 s, whereas δ -MnO₂/G/GCE took around 5 s (Fig. S7). The faster and remarkable current response of Ag@ô-MnO2/G/GCE compared with δ-MnO₂/G/GCE, ascribed that Ag NPs significantly improved the EASA of the defective surface of δ -MnO₂/G as well as surface/interface properties. The sharp electrochemical response of Ag@ô-MnO2/G/GCE was observed due to fast adsorption, diffusion and electron transfer capacity of the modified electrode towards H₂O₂. Fig. 8b showed that the plot of Ipc versus [H2O2] of δ-MnO2/G/GCE provided two linear regression equations as below:

 I_{pc} (µA) = 2.4601 [H_2O_2] (mM) + 2.2791, R² = 0.9932; [0.05 to 4.0 mM].

and.

 I_{pc} (µA) = 0.08408 [H_2O_2] (mM) + 8.0802, R² = 0.9924; [4.0 to 16.0 mM].

The limit of detection (LOD) and sensitivity of δ -MnO₂/G was obtained as 0.756 µM and 34.79 µA mM⁻¹ cm⁻² (0.05 to 4.0 mM), and 2.46 µM and 11.89 µA mM⁻¹ cm⁻² (4.0 to 16.0 mM), respectively (where, S/N = 3). Fig. 8c showed that the plot of I_{pc} versus [H_2O_2] of Ag@ δ -MnO₂/G/GCE provided a linear regression equation as below:

 I_{pc} (µA) = 7.3829 [H_2O_2] (mM) + 5.3533, R² = 0.9987; [0.005 to 90.64 mM].

The LOD and sensitivity of Ag@ δ -MnO₂/G was obtained as 0.068 μ M



Fig. 6. (a) CV curves of $Ag@\delta-MnO_2/G$ at different pH (5.0 to 9.0) in the presence of 2.0 mM H_2O_2 in 0.2 M PBS at a scan rate of 50 mV/s. (b) Peak potential and peak current as the function of pH. Error bars set for 5% of standard error.



Fig. 7. (a) CV curves of Ag@ δ -MnO₂/G at different scan rates (20 to 500 mV/s) in presence of 2.0 mM H₂O₂ in 0.2 M PBS (pH 7.0). (b) Linear dependence of the peak current with the square root of the scan rate ($\nu^{1/2}$), and (c) linear relation between the peak potential and ln ν . Error bars set for 5% of standard error.

and 104.43 μ A mM⁻¹ cm⁻², respectively (where, S/N = 3). Table 1. and Table 2. represent the comparison in electrochemical parameters of Ag@ δ -MnO₂/G with other previously reported non-Ag and Ag modified sensors for H₂O₂ determination, which implied the proposed modified electrode as one of the best sensors.

3.7. Interference study

Fig. 9a exhibited the interference effect of Ag@ δ -MnO₂/G/GCE towards 750 μ M H₂O₂ in presence of 750 μ M coexist biomolecules such as ascorbic acid (AA), glucose (Glu), acetaminophen (AP), uric acid (UA), dopamine (DA), 10-fold excess of ethanol (EtOH), and 25-fold excess of common ions such as Na⁺, Mn²⁺, Γ , and CO₃²⁻ in Ar-saturated 0.2 M PBS (pH 7.0) at -0.5 V. None of them showed significant current response, whereas H₂O₂ brought out well defined current response for every successive addition. Each of the successive addition of 750 μ M H₂O₂ provided almost equal current response, interestingly third addition was slight higher than that of second one (Fig. 9a, inset). Fig. 9b showed the CA difference between the Ar and O₂-saturated 0.2 M PBS (pH 7.0) at -0.5 V for successive addition of 750 μ M H₂O₂ at every 100 s. Ar-saturated CA provided smooth and stable current response along with good signal-to-noise ratio, whereas background noise in O₂-saturation

increased with time due to electro-reduction of oxygen.

3.8. Long-term stability, reproducibility and repeatability

The stability, reproducibility and repeatability of $Ag@\delta-MnO_2/G/GCE$ towards 2.0 mM H_2O_2 were investigated by CV at a scan rate of 50 mV/s in 0.2 M PBS (pH 7.0). The stability of the modified electrode was checked every three days during a period of 30 days while the modified electrode was at room temperature under a plastic cover. After one month 87.54% relative current of initial current was obtained (Fig. S8a), which indicated the long-term stability of the modified electrode. Electrode-to-electrode reproducibility was evaluated by six different electrodes (Fig. S8b) and the relative standard deviation (RSD) was calculated as 1.19%. Repeatability was examined by ten consecutive measurements of same electrode and RSD was calculated as 4.34%.

3.9. Real samples analysis

The practical application of Ag@ δ -MnO₂/G/GCE towards different concentration of H₂O₂ was investigated by DPV (Fig. S9) in food samples (honey, milk and tomato sauce (TS)) and industrial sample (tab water (TW)) in 0.2 M PBS (pH 7.0). Honey, liquid milk and TS were collected



Fig. 8. (a) CA current response of Ag@ δ -MnO₂/G and δ -MnO₂/G at an applied potential of -0.5 V with successive additions of H₂O₂ in 0.2 M PBS (pH 7.0), relationship between current vs. [H₂O₂] for (b) δ -MnO₂/G (50 μ M to 16 mM), and (c) Ag@ δ -MnO₂/G (5 μ M to 90.64 mM); inset: magnified CA curves of Ag@ δ -MnO₂/G at lower concentration. Error bars set for 5% of standard error.

Table 1	
Comparison of the detection parameters of $Ag@\delta-MnO_2/G$ with the previously reported non-Ag catalysts for determination of H_2O_2 .	

Sensors	Response time (s)	Linear range (mM)	Sensitivity ($\mu A \ mM^{-1} \ cm^{-2}$)	Detection limit (µM)	Refs ^[published year]
Ag@δ-MnO ₂ /G/GCE	<2	0.005-90.64	104.43	0.068	This work
TS-HTC/GCE	3.61	0.05–0.5 0.5–2.7	122.02 80.79	0.015 0.05	[7][2021]
NCNT MOF CoCu	<3	0.0–3.5	639.5	0.206	[23][2021]
MnFe ₂ O ₄ /rGO/SPCE	~5	0.1-4.0	6.337*	0.528 nM	[25] ^[2021] [01 ^[2020]
GN@FeOOH	~3	0.0003-1.2	 265.7	0.08	$[12]^{[2020]}$
CuCo ₂ O ₄	<3	0.01-8.90	94.1	3.0	[66] [2020]
CC/Co@C-CNTs MnO ₂ (δ)/MWCNT	<4 <2	0.0004-7.2 0.100-20.5	388 13.9	0.27 6.97	[67][2020] [68][2019]
Mn(II)-PLH-CMWCNT/GCE	<3	0.002–1.0	464.18	0.5	[69] [2019]
δ-MnO ₂ /CNTs/GCE	<3	0.05–22.0	243.9	1	[24] ^[2016]

 * $\mu A m M^{-1}$.

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Table 2

Comparison of the detection parameters of Ag@o-MnO ₂ /G with the previously reported Ag catalysts for G
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Sensors	Response time (s)	Linear range (mM)	Sensitivity (μ A mM ⁻¹ cm ⁻²)	Detection limit (µM)	Refs ^[published year]
Sensors Ag@&-MnO ₂ /G/GCE LIG@Ag HPCS-Ag C _{rod} @Ag-Ps Ag-Pt core-shell NWs Ag-Pt core-shell NWs Ag-Pt core-shell NWs Ag-Pt core-shell NWs Ag-Ws0.5/GCE Ag/H-ZIF-67/GCE PDDA-AuPtAg/RGO	Response time (s) <2	Linear range (mM) 0.005–90.64 0.01–2.61 0.004–0.24 0.5–5.0 0.017–0.99 0.3–3.3 0.001–1.075 0.005–67 0.00005–5.5	Sensitivity (µA mM ⁺ cm ⁻) 104.43 28.6 637 128 - 271 - 421.4 2838.95	0.068 2.8 0.188 67.0 10.95 0.2 0.05 1.1 0.0012	Reis This work [1] [2021] [5] [2021] [18] [2021] [20] [2020] [32] [2020] [33] [2020] [70] [2019]
Nf/Pd@Ag/rGO-NH ₂ /GCE Ag/MnO ₂ /GO/GCE Ag-MnO ₂ -MWCNTs/GCE	<2 <5 <2	0.002–19.5 0.003–7 0.005–10.4	1307.46 105.40 82.5	0.7 0.7 1.7	[71] ^[2018] [31] ^[2015] [59] ^[2013]



Fig. 9. CA current response of the $Ag@\delta-MnO_2/G$ towards 750 μ M H_2O_2 (a) in presence of 750 μ M AA, Glu, AP, UA, DA, 10-fold excess of EtOH, and 25-fold excess of Na⁺, CO₃²⁻, Mn²⁺, and Γ in Ar-saturated 0.2 M PBS (pH 7.0), and (b) comparison between O₂-saturated and Ar-saturated system at -0.5 V.

from local supermarket of Gwangju, South Korea. All of the food samples centrifuged and diluted in distilled water for homogeneous solution, and reserved in refrigerator before used. TW was taken from our lab basin and used without any purification. 50 μ L of each sample was added in 3950 μ L 0.2 M PBS (pH 7.0) individually for the preparation of real samples. Then 0.10, 0.25, 0.5, and 1.0 mM H₂O₂ was added subsequently in those samples, and obtained current was measured. The founded concentration was calculated from the obtained current for each real samples (Fig. S9 insets). The result of practicability was summarized in Table 3. The recovery ranges and relative standard deviation (RSD) of all the real samples indicated that the sensor was effective for the practical applications.

4. Conclusion

The Ag@ δ -MnO₂/G ternary nanocomposite was synthesized to increase the electrical conductivity of δ -MnO₂ and reduce the aggregation of Ag NPs. Furthermore, we have planned to occupy the cation vacancies of δ -MnO₂ by the Ag-atom for achieve a better electronic structure and surface/interface properties. TEM images revealed that Ag atom was successfully doped on MnO₂/G surface. The defective bimetallic zone created by Ag atom in the prepared sample provided higher electrochemical performances. As a result, Ag@ δ -MnO₂/G/GCE exhibited a faster and remarkable current response towards H₂O₂ compared with δ -MnO₂/G/GCE in CA. The LOD, sensitivity, linear range, and response time of Ag@ δ -MnO₂/G/GCE were better than δ -MnO₂/G/GCE. The proposed sensor also showed negligible interference, long-term stability, repeatability, reproducibility, and excellent practicability in food and industrial samples.

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Determination	result of H ₂ O	in food and	l wastewater sam	nles at Aσ@δ-MnO	$^{/G}$

Samples	Added H ₂ O ₂ (mM)	Founded H ₂ O ₂ (mM)	Recovery (%)	RSD (%)
Honey	1.0	1.027	102.7	1.69
	0.50	0.506	101.2	
	0.25	0.252	100.8	
	0.10	0.098	98.0	
Milk	1.0	1.031	103.1	1.12
	0.50	0.509	101.8	
	0.25	0.244	97.6	
	0.10	0.099	99.0	
Tomato sauce	1.0	1.009	100.9	1.63
(TS)	0.50	0.492	98.4	
	0.25	0.251	100.4	
	0.10	0.103	103.0	
Tab water (TW)	1.0	0.993	99.3	1.99
	0.50	0.513	102.6	
	0.25	0.249	99.6	
	0.10	0.097	97.0	

CRediT authorship contribution statement

Abdul Kader Mohiuddin: Conceptualization, Methodology, Data curation, Visualization, Investigation, Writing – original draft. **Seung-won Jeon:** Supervision, Writing – review & editing.

Declaration of Competing Interest

The authors declare that they have no known competing financial

interests or personal relationships that could have appeared to influence the work reported in this paper.

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Appendix A. Supplementary material

Supplementary data to this article can be found online at https://doi.org/10.1016/j.apsusc.2022.153162.

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