



Simple and Inexpensive Fabrication of Zero Mode Waveguides for High Concentration Single Molecule Microscopy

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Abstract

In single molecule fluorescence enzymology, background fluorescence from labeled substrates in solution often limits fluorophore concentration to pico- to nanomolar ranges, several orders of magnitude less than many physiological ligand concentrations. Optical nanostructures called zero mode waveguides (ZMWs), which are 100–200 nm in diameter apertures fabricated in a thin conducting metal such as aluminum or gold, allow imaging of individual molecules at micromolar concentrations of fluorophores by confining visible light excitation to zeptoliter effective volumes. However, the need for expensive and specialized nanofabrication equipment has precluded the widespread use of ZMWs. Typically, nanostructures such as ZMWs are obtained by direct writing using electron beam lithography, which is sequential and slow. Here, we use colloidal, or nanosphere, lithography as an alternative strategy to create nanometer-scale masks for waveguide fabrication. This report describes the approach in detail, with practical considerations for each phase. The method allows thousands of aluminum or gold ZMWs to be made in parallel, with waveguide diameters and depths of 100–200 nm. Only common lab equipment and a thermal evaporator for metal deposition are required. By making ZMWs more accessible to the biochemical community, this method can facilitate the study of molecular processes at cellular concentrations and rates.

SUMMARY:

We describe a nanosphere lithography method for parallel fabrication of zero mode waveguides, which are arrays of nanoapertures in a metal-clad glass microscopy coverslip for single molecule imaging at nano- to micromolar concentrations of fluorophores. The method takes advantage of colloidal crystal self-assembly to create a waveguide template.

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The authors have nothing to disclose.

Keywords

Zero Mode Waveguides; Nano-aperture; Single Molecule Fluorescence; Nanosphere Lithography; Colloidal Crystal; Self-assembly

INTRODUCTION:

Single-molecule techniques such as single molecule fluorescence resonance energy transfer (smFRET) or single molecule fluorescence correlation spectroscopy (FCS) are powerful tools for molecular biophysics, allowing the study of dynamic movements, conformations, and interactions of individual biomolecules in processes such as transcription^{1–3}, translation^{4–6}, and many others⁷. For smFRET, total internal reflection fluorescence (TIRF) microscopy is a common method because many tethered molecules may be followed over time, and the evanescent wave generated by TIR is limited to a 100–200 nm region adjacent to the coverslip⁸. However, even with this restriction on excitation volume, fluorophores of interest still need to be diluted to pM or nM ranges in order to detect single molecule signals above background fluorescence⁹. Since the Michaelis-Menten constants of cellular enzymes are typically in the μM to mM range¹⁰, biochemical reactions in single molecule studies are usually much slower than those in the cell. For example, protein synthesis occurs at 15–20 amino acids per second in *E. coli*^{11,12}, while most prokaryotic ribosomes in smFRET experiments translate at 0.1–1 amino acid per second¹³. In protein synthesis, crystal structures and smFRET on stalled ribosomes showed that transfer RNAs (tRNAs) fluctuate between ‘hybrid’ and ‘classical’ states before the tRNA-mRNA translocation step^{14,15}. However, when physiological concentrations of the translocation GTPase factor, EF-G, was present, a different conformation, intermediate between the hybrid and classical states, was observed in smFRET⁶. Studying dynamic molecular processes at rates and concentrations similar to those in the cell is important, but remains a technical challenge.

A strategy to increase the fluorescent substrate concentration is the use of metal-based, sub-visible wavelength apertures, called zero mode waveguides (ZMWs), to generate confined excitation fields that selectively excite biomolecules localized within the apertures¹⁶ (Figure 1). The apertures are typically 100–200 nm in diameter and 100–150 nm in depth¹⁷. Above a cutoff wavelength related to the size and shape of the wells ($\lambda_c \approx 2.3$ times the diameter for circular waveguides with water as the dielectric medium¹⁸), no propagating modes are allowed in the waveguide, hence the term zero mode waveguides. However, an oscillating electromagnetic field, termed an evanescent wave, exponentially decaying in intensity still tunnels a short distance into the waveguide^{18,19}. Although similar to TIR evanescent waves, ZMW evanescent waves have a shorter decay constant, resulting in 10–30 nm effective excitation region within the waveguide. At micromolar concentrations of fluorescently labeled ligands, only one or a few molecules are simultaneously present within the excitation region. This restriction of the excitation volume and consequent reduction of background fluorescence enables fluorescence imaging of single molecules at biologically relevant concentrations. This has been applied to many systems²⁰, including FCS measurements of single protein diffusion²¹, single molecule FRET measurements of low-affinity ligand-

protein²² and protein-protein interactions²³, and spectro-electrochemical measurements of single molecular turnover events²⁴.

ZMWs have been produced by directly patterning a metal layer using ion beam milling^{25,26} or electron beam lithography (EBL) followed by plasma-etching^{16,27}. These mask-less lithography methods create waveguides in series and typically require access to specialized nanofabrication facilities, preventing widespread adoption of ZMW technology. Another method, ultraviolet nanoimprint lithography lift-off²⁸, uses a quartz slide mold to press an inverse ZMW template onto a resist film like a stamp. While this method is more streamlined, it still requires EBL for fabrication of the quartz mold. This paper presents the protocol for a simple and inexpensive templated fabrication method that does not require EBL or ion-beam milling and is based on close-packing of nanospheres to form a lithographic mask.

Nanosphere or “natural” lithography, which was first proposed in 1982 by Deckman and Dunsmuir^{29,30}, uses the self-assembly of monodisperse colloidal particles, ranging from tens of nanometers to tens of microns³¹, to create templates for surface patterning via etching and/or deposition of materials. The 2D or 3D extended periodic arrays of colloidal particles, referred to as colloidal crystals, are characterized by a bright iridescence from scattering and diffraction³². Although less widely used than electron-beam or photolithography, this masking methodology is simple, low cost, and easily scaled down to create feature sizes below 100 nm.

Directing the self-assembly of colloidal particles determines the success of using colloidal crystals as masks for surface patterning. Provided the particles’ size and shape are homogeneous, colloidal particles can be readily self-assembled with hexagonal packing, driven by entropic depletion³³. Water evaporation after drop-coating is an effective route to sediment the colloidal particles, although other methods include dip-coating³⁴, spin coating³⁵, electrophoretic deposition³⁶, and consolidation at an air-water interface³⁷. The protocol presented below is based on the evaporation sedimentation method, which was the simplest to implement. The triangular interstices between close-packed polystyrene beads form openings in which to plate a sacrificial metal, forming posts (Figures 2 and S1). Brief annealing of the beads before this step adjusts the shape and diameter of these posts. The beads are removed, a final metal layer is deposited around the posts, and then the posts are removed. After the two metal deposition steps onto the colloidal nanomask, removal of the intermediate posts, and surface chemistry modification for passivation and tethering, ZMW arrays are ready to use for single molecule imaging. More extensive characterization of the ZMW optical properties after fabrication can be found in an accompanying article³⁸. Besides a thermal evaporator for vapor deposition of the metals, no specialized tools are required.

PROTOCOL:

Note: All steps can be completed in general lab space.

1. Glass Coverslip Cleaning

1.1. To provide a clean surface for evaporative deposition of colloidal particles, place 24×30 mm optical borosilicate glass coverslips (0.16–0.19 mm thickness) within the grooved inserts of a coplin glass staining jar for cleaning.

Note: Make sure the coverslips stand upright and are well-separated so that all surfaces are clearly exposed during the cleaning process.

1.2. Pour enough acetone in the staining jar to cover the coverslips, place the cover on, and sonicate for 10 minutes at 40 °C.

1.3. Pour out the acetone and rinse the coverslips by filling the staining jar with distilled H₂O and pouring out the water. Repeat 2 more times.

1.4. Repeat the acetone sonication once more (steps 1.2–1.3).

1.5. Pour enough 200 mM KOH in the jar to cover the coverslips and sonicate, covered, for 20 min at 40 °C.

Note: The KOH slightly etches the glass.

1.6. Rinse the coverslips with distilled H₂O 6 times.

1.7. Add ethanol to cover the coverslips, add the lid, and sonicate for 10 min at 40 °C.

1.8. Rinse the coverslips with distilled H₂O 3 times.

1.9. Pick up each coverslip at the edge using gentle forceps and dry the coverslips with N₂ gas. Touch only the edges of the coverslip. Place each of the dried, cleaned coverslips in an individual clean petri dish.

2. Evaporative Deposition of Polystyrene Beads

2.1. To create the colloidal crystal mask for the ZMW array, centrifuge 50 µL of 1 µm diameter, non-functionalized polystyrene beads (2.5% weight/volume in water) at 15,000 g, 25 °C for 5 min.

Note: Before pipetting the beads, the stock solution should be briefly vortexed in case the beads have settled to the bottom of the bottle.

2.2. Discard the supernatant, leaving as little water remaining as possible.

Note: Residual water can change the evaporation properties of the ethanol resuspension³⁹, so removal of a small amount of beads in order to remove all of the water is acceptable.

2.3. Resuspend the beads from step 2.2 in 50 µL of 1:400 TritonX-100:EtOH solvent. Pipette up and down several times to thoroughly mix the beads with the solvent.

Note: TritonX-100:EtOH solvent should be sealed with parafilm after use and prepared fresh once a month.

Note: The beads tend to adhere to the sides of a plastic vessel, such as an Eppendorf tube, so pipette along the sides to ensure that all beads are resuspended.

2.4. To set up a humidity chamber for deposition, place 6 petri dishes, each with one coverslip, on a bench in a line with lids left slightly ajar. In each dish, move the coverslip to the open region so that the coverslips are exposed to the environment when the humidity is increased in the next step.

2.5. Place a hygrometer and a small electric fan centered behind the petri dishes.

2.6. Record the starting relative humidity (RH) in the lab. Fill a 200 mL beaker with 150–200 mL of ~75 °C water and place it behind the fan.

2.7. Turn on the fan and cover the petri dishes, fan, beaker, and hygrometer with an overturned, transparent plastic storage container (66 cm x 46 cm x 38 cm).

2.8. Let the RH in the chamber rise to 70–75%, which typically takes 5–10 min.

Note: If the ambient lab RH is low (below ~50%), let the chamber reach a higher RH, but no higher than 80%, to compensate for loss of humidity during deposition (see below).

2.9. When the RH reaches 70–75%, record the RH and lift up the plastic storage container slightly to quickly place covers on the petri dishes, which prevents over-wetting of the coverslips.

Note: The temperature in the chamber will be slightly warmer than room temperature, typically 25–26 °C, as a result of humidification. If moisture is visible on the coverslips, then the glass surfaces are too wet. A commercial glove box might simplify this part of the protocol.

2.10. Let the RH in the chamber continue to rise to 85%. At that point, record the RH in the humidity chamber and pipette 5 μ L of the bead suspension onto the center of each coverslip.

2.11. Close the chamber and petri dishes after each deposition to minimize loss of humidity. Aim to finish all 6 depositions within 2 minutes.

2.12. Record the RH in the chamber after the deposition.

Note: The RH after deposition will help gauge how fast humidity was lost during deposition, which depends on ambient lab conditions. For a typical successful run, the chamber will start at 85% RH prior to deposition and end at 70–75% RH after the deposition.

2.13. Let the bead droplets spread and dry for 5 minutes.

Note: If the colloidal crystals have many holes or multilayered regions, then the chamber was likely too humid or dry, respectively. Adjust the relative humidity at which to close the petri dishes and begin the depositions (see Results section for further discussion of optimization).

3. Bead Annealing for Reducing Pore Size in the Colloidal Crystal Template

3.1. To provide a uniform temperature surface for annealing of the polystyrene beads, which narrows the inter-bead interstices and rounds the interstices' corners, place a flat, milled aluminum plate on top of a standard ceramic hot plate.

3.2. Set the temperature of the hot plate to 107 °C, the glass transition temperature of polystyrene⁴⁰.

Note: To obtain stable and accurate temperature, a thermocouple probe was held in a 2–3 mm wide and 4–5 mm deep hole in the aluminum plate.

3.3. Place a coverslip containing the bead template on the hot aluminum plate and anneal for 20 s (see Discussion section for explanation of melting time).

3.4. After heating, remove the coverslip from the aluminum plate and promptly place it on another room temperature aluminum surface to cool it.

Note: It is helpful to either have the coverslips hang slightly over the edge of the plate or mill shallow channels (see video) into the plate to facilitate pickup of the coverslips.

Note: The protocol can be interrupted here, but bead templates should be stored in covered, clean petri dishes in a cool, dry environment to preserve the polystyrene templates.

4. Nanofabrication of Zero Mode Waveguides Using the Colloidal Crystal Template

Note: The protocol to fabricate aluminum ZMWs (Figure 2) is provided in this section. The accompanying video shows fabrication of aluminum ZMWs.

4a.1. Using thermal or electron-beam evaporative deposition, deposit 300 nm of copper at 2 Å per sec over the colloidal crystal template to generate posts in the interstices between the beads.

4a.2. Remove excess metal on top of the beads by gently pressing the surface with Scotch Magic tape. Slowly peel the tape to pull off the metal.

Note: Some small patches of reflective excess metal may remain after the tape pull, and these can often be removed by a stream of N₂ gas. If substantial patches of reflective excess metal remain after the tape pull, try soaking the templates in toluene for 2 hours to partially dissolve the polystyrene beads. Wash the coverslips with distilled water, dry with N₂, and repeat the tape pull. The additional soak should not completely dissolve the beads, as the beads help protect the posts from damage during the tape pull.

4a.3. To dissolve the polystyrene beads, place the bead templates in toluene and soak overnight.

Caution: Toluene fumes may be toxic. Work with toluene under a well-ventilated hood and wear personal protective equipment, including gloves, safety glasses, and a lab coat. Toluene should be stored in ventilated cabinets designated for flammable liquids.

4a.4. After the toluene incubation, rinse the templates once with chloroform and twice with ethanol. Handle coverslips carefully at this point because the delicate 200 – 300 nm tall metal posts are now exposed. Dry the templates with N₂ and remove residual polymer and contaminants in an oxygen plasma cleaner for 30 minutes.

Caution: Chloroform fumes may be toxic. Work with chloroform under a well-ventilated hood and wear personal protective equipment, including gloves, safety glasses, and a lab coat. Chloroform should be stored in ventilated cabinets away from other flammable solvents.

4a.5. Using thermal or electron-beam evaporative deposition, deposit 3 nm of a titanium adhesion layer at 1 Å per sec followed by 100–150 nm of aluminum at 4 Å per sec around and on top of the copper posts.

Note: One can use thicker cladding to obtain deeper guides and better attenuation of background fluorescence, but this also decreases the yield after exposing and dissolving the posts in the next step (see Discussion section).

4a.6. To dissolve the metal posts, soak the coverslips in copper etchant (Transene copper etchant 49–1, citric acid-based) for 2 hours.

Caution: Metal etchant can cause skin burns. Work with etchants under a well-ventilated hood and wear protective equipment. Wash hands thoroughly after handling. Metal etchant should be stored in ventilated cabinets designated for corrosive liquids.

4a.7. Rinse the coverslips with distilled water, dry with N₂, and gently buff the surface of the metal cladding with lens paper to expose any posts that are still covered in cladding. Place the coverslips back in copper etchant for another 2 hours, then rinse again with distilled water and dry with N₂.

Note: ZMW slides should be stored in covered, clean petri dishes to keep them free of contaminants. Note: The protocol to fabricate gold ZMWs (Figure S1), which mirrors the protocol to fabricate aluminum ZMWs, is provided in this section.

4b.1. Using thermal or electron beam evaporative deposition, deposit 3 nm of a titanium adhesion layer at 1 Å per sec followed by 300 nm of aluminum at 4 Å per sec.

4b.2. Remove excess metal on top of the beads by gently pressing the surface with Scotch Magic tape. Slowly peel the tape to pull off the metal.

4b.3. To dissolve the polystyrene beads, place the bead templates in toluene and soak overnight.

4b.4. After the toluene incubation, rinse the templates once with chloroform and twice with ethanol. Dry the templates with N₂ and remove residual polymer contaminants in an oxygen plasma cleaner for 30 minutes.

4b.5. Using thermal or electron beam evaporative deposition, deposit 100–150 nm of gold at 5 Å per sec around and on top of the aluminum posts.

4b.6. To dissolve the metal posts, soak the coverslips in aluminum etchant (Transene aluminum etchant type A, phosphoric acid-based) for 1 hour.

4b.7. Rinse the coverslips with distilled water, dry with N₂, and gently buff the surface of the metal cladding with lens paper to expose any posts that are still covered in cladding. Place the coverslips back in aluminum etchant for 1 hour, then rinse again with distilled water and dry with N₂.

Note: ZMW slides should be stored in covered, clean petri dishes.

REPRESENTATIVE RESULTS:

The self-assembly of the polystyrene colloidal particles via evaporative sedimentation (steps 2.1 – 2.13) can produce a range of results since it requires control of the solvent evaporation rate. However, because the depositions are fast (10–15 min per round), the procedure can be quickly optimized for different ambient lab conditions. Figure 3a shows a well-formed colloidal template after deposition and evaporation. Macroscopically, the region of beads is circular, with borders defined by an opaque, multilayered ring of beads. The translucent, but not white regions in the image are the desired monolayer areas. Figure 3b shows a colloidal template that was packed in an overly humid environment (80% RH when petri dishes were closed). These templates tend not to have a clean circular boundary and have multilayered tendrils extending outward. The deposition is acceptable and can be used in subsequent phases, but the holes in the lattice reduce the number of useable ZMW array areas for single molecule imaging. Figure 3c shows a colloidal template that was packed in an overly dry environment (65% RH when petri dishes were closed). These templates are usually smaller in diameter compared to the ideal, well spread templates. The deposition can be used, but the multilayered, white regions, which streak inward, reduce the area usable for imaging. Thus, we do not recommend performing more than 6 depositions at a time since depositions toward the end of the process will occur at a lower humidity as the chamber is opened and closed. Figure 3d shows the rainbow pattern produced by diffraction of reflected light from the polystyrene crystal. This pattern can be used to confirm success and quality of crystal packing by eye. Figures 3e and 3f show atomic force microscopy (AFM) images of well-packed colloidal templates. The defects between grains arise from jamming during the evaporative sedimentation⁴¹, and distinct grains can be seen with a 10x objective. Thus, examining the colloid depositions with a low-power light microscope can also be used to assess packing.

After deposition of copper onto the annealed colloidal templates (step 4a.1), the rainbow diffraction pattern should still be visible and enhanced by the reflective metal coating the tops of the beads (Figure 4a–b). The templates lose the reflective rainbow diffraction pattern

after the Scotch tape pull (step 4a.2) that removes excess copper (Figure 4c). Figures 4d–e show AFM images of a typical field of copper posts after metal deposition. The defects in between the colloidal crystal grains in Figure 3e are visible in the copper post images as larger regions of copper. Analysis of AFM images shows that, for a copper deposition thickness of 300 nm, the eventual copper posts average 255 nm (Figure 4f) in height and 121 nm in diameter (Figure 4g).

Deposition of the aluminum cladding (step 4a.5) where the beads were and on top of the copper posts, and subsequent dissolution of the posts (steps 4a.6–4a.7) results in the aluminum ZMWs shown in Figure 5a–c. Defects between the colloidal crystal grains are visible as larger openings (Figure 5b). The average distance between the ZMW centers in Figure 5c is 559 nm, consistent with the spacing set by the hexagonal close packing geometry of the 1 μm beads ($\frac{2}{\sqrt{3}}*500 \text{ nm} = 557 \text{ nm}$). Using polystyrene templates that were annealed for 20 s results in waveguides that are on average 118 nm in diameter (Figure 5d–e), consistent with the post diameters and sufficiently small to cut off propagation of visible light. A height profile of a waveguide from Figure 5d also shows that it is ~ 120 nm deep.

Single molecule FRET was performed in the ZMWs to test for functionality (Figure 6a). A typical field of ZMWs for imaging is shown in Figure 6b, which contains >3000 waveguides in a $40 \times 80 \mu\text{m}$ field of view. We first passivated the ZMWs using protocols described previously^{42,43}. Briefly, the aluminum ZMWs were passivated with poly(vinylphosphonic acid) to coat the aluminum cladding followed by methoxy-terminated polyethylene glycol doped with biotin-terminated PEG to coat the glass bottoms of the ZMWs. Gold ZMWs can be passivated with thiol-derivatized PEG to coat the gold cladding followed by a similar PEG treatment for the glass bottoms. Flow chambers, $\sim 20 \mu\text{L}$ volume, were then constructed for single molecule imaging⁴⁴. Single molecule FRET imaging of the DNA duplexes was performed as described previously³⁸. Briefly, 100 pM – 1 nM of Cyanine-3/Cyanine-5 (Cy3/Cy5), biotinylated DNA duplexes (33 base-pair length) were incubated for 10 min in flow channels functionalized with streptavidin (5 min incubation, 0.5 mg/mL solution). The concentration of labeled macromolecules can be titrated to achieve $\sim 20\%$ loading of the waveguides with 1 molecule, leading to $<5\%$ waveguides loaded with more than 1 molecule, based on Poisson distributed loading (most waveguides, $\sim 75\%$, will have no molecules)⁴⁵. Unbound DNA was washed away with nuclease-free duplex buffer followed by illumination buffer (0.3% (w/v) glucose, 300 $\mu\text{g}/\text{ml}$ glucose oxidase, 120 $\mu\text{g}/\text{ml}$ catalase, and 1.5 mM TROLOX (6-hydroxy-2,5,7,8-tetramethyl-chromane-2-carboxylic acid)). Non-biotinylated Cy5-labeled DNA duplexes (33 base pair length) were present in the illumination buffer at 0, 50, 100, and 500 nM as background fluorophores in solution. Single molecule FRET traces from the immobilized Cy3/Cy5 duplex DNA molecules were recorded with a custom-built TIRF microscope adjusted to epi-fluorescence conditions. Movies were recorded with a 1.48 NA 100x oil immersion objective with alternating 532 nm and 640 nm excitation (100 ms exposure) and a Dual View spectral splitter (Photometrics, Inc.) to record Cy3 and Cy5 emission simultaneously on an EMCCD camera. Single molecule FRET traces with single step bleaches in the Cy5 channel were detectable at all concentrations of ambient Cy5

tested (Figure 6c–f). In comparison, single molecules would only be detectable in TIRF illumination with pM to low nM solution fluorophore concentrations⁴⁶.

DISCUSSION:

For the colloidal self-assembly (steps 2.1 – 2.13), the use of ethanol rather than water as the suspension solvent speeds the evaporation process so that templates are ready in 2–3 minutes after deposition rather than 1–2 hours as in previous methods^{48,49}. The evaporative sedimentation protocol presented here is also simpler than previous sedimentation protocols that require controlling surface tilt, temperature, and air volume above the suspension^{49–51}. The particle volume fraction used in this protocol is 2.4%, higher than the 0.2–0.5% used in previous sedimentation methods⁴⁸, which resuspended colloids in water-glycerol mixtures for much longer settling time-scales. However, the quality of the depositions is robust to changes in the particle volume fraction, with past studies finding that it can be varied between 2–10%^{49–51}. The grain sizes of the colloidal crystals obtained in a successful deposition from this protocol are 20–30 μm across, larger than grains from previous sedimentation methods (typically several hundred nanometers across)^{48,49}. Macroscopically, the roughly 2 cm in diameter areas of colloidal monolayer are also comparable to the 1 cm areas produced by previous methods⁴⁹. The large size of the colloidal crystal templates produced in this method also allows 3–5 separated flow chambers⁴⁴, each about 3–4 mm wide, to be made on each ZMW slide. Thus, multiple independent single molecule experiments can be performed on each slide.

Annealing the colloidal bead templates (steps 3.1–3.4) after self-assembly is a simple, but crucial step for sufficiently reducing background fluorescence with the ZMWs. As Figure S2 shows, the effective cutoff wavelength for a waveguide with the triangular cross-section from an un-annealed template is 894 nm. In comparison, the effective cutoff wavelength for a 130 nm diameter circular waveguide from an annealed template is 221 nm, as determined both analytically (1.7 times the diameter of the guide¹⁸) and numerically. Using smaller beads for deposition could also reduce the size of the template pores, but the waveguides would then be spaced closer than 200 nm, which is around the diffraction limit of visible light. Furthermore, the waveguides would remain triangular in cross-section, which leads to nonsymmetric power propagation through the waveguide (Figure S2a–d). One drawback of the annealing step is that variability in melting time can introduce fluctuations in waveguide diameter, so accurate timing helps to minimize variation between batches. The interstices start to close at annealing times longer than 25 sec, and post diameters do not decrease very much between 20 and 25 sec (Figure S3b–d). A quick test for interstice closure is to check whether annealed templates still produce a rainbow diffraction pattern when illuminated with light and viewed at an angle. If not, the majority of interstices likely have closed. The relationship between annealing time and typical pore diameters has been presented earlier³⁸.

After achieving the desired pore size during the annealing step, copper is deposited (step 4a.1) onto the templates to create a shadow of the mask. It is important to use line of sight deposition, with the metal approaching the template as perpendicularly as possible. Thus, increasing the distance between the sample and metal source as well as ensuring the plate holding the templates is not spinning, as is automatically done in some vapor deposition

machines, will aid in minimizing lateral deposition of metal onto the polystyrene substrate. However, some lateral deposition is inevitable, which reduces the size of the interstitial hole and thus the post cross-section as more metal is deposited⁵². This results in pyramidal metal posts rather than prism-like structures⁵².

Because the copper posts are likely pyramidal rather than prism shaped, aluminum deposition (step 4a.5) on top of the posts also covers some of the sloped sides, blocking accessibility of the copper etchant for some of the posts. Thus, the lens paper buffing step (step 4a.7 or 4b.7) was added after the first soak in etchant to mechanically disrupt any copper posts still covered in aluminum. Depositing more copper to create taller posts also makes the posts more susceptible to mechanical disruption during the lens paper buffing. However, more than 500 nm of copper should not be deposited since the goal of deposition is to project the interstitial hole at the 500 nm midline of the 1 μm beads.

Another potential difficulty is the unintended removal of aluminum cladding during the lens paper buffing (step 4a.7 or 4b.7). It was found that loss of aluminum cladding during buffing became more frequent after the annealing step was added, likely due to increased polystyrene residue, which can interfere with aluminum adherence to glass (steps 3.1–3.4). However, the overnight toluene soak (step 4a.3/4b.3) after the tape pull resolved this issue. In the AFM image in Figure 4e, some residual rings of polystyrene can be seen between the posts, but the aluminum cladding still resisted multiple buffs in step 4a.7. If loss of the aluminum cladding remains an issue after the overnight toluene soak, an RCA-1 (standard clean-1) wash, piranha wash, or further oxygen plasma cleaning can be added to step 4a.4/4b.4.

The performance of the ZMWs in single molecule FRET experiments was similar to that of ZMWs fabricated with EBL. In a previous study⁵³ performing single molecule FRET on Cy3 labeled single stranded DNA with Cy5 labeled DNA helicase loader protein in solution (the same donor-acceptor arrangement as that in Fig. 6a), FRET events were clearly discernible at 100 nM Cy5 background, were less clear (lower acceptor trace signal to noise) at 1 μM , and not discernible at 10 μM . We note that a previous study with commercial ZMWs reported single molecule FRET acceptor signals at background concentrations as high as 1 mM⁵⁴, higher than we and other previous studies^{42,53} with in-house fabricated ZMWs have achieved. Further discussion of the signal-to-background performance among ZMWs is given in reference 38. Nonspecific interaction of fluorescent sample with the ZMW surfaces⁵³ is a common challenge limiting access to higher concentrations, especially if the diffusing fluorescent species in solution is a large macromolecule. Studies with ZMWs on complex biochemical systems such as translation have typically limited free fluorescent substrate concentrations to 100–250 nM^{55–58}. Regardless of the intended application of the ZMWs, optimization of passivation methods for different systems will likely be necessary to maintain acceptable signal to noise at high concentrations.

Overall, the method presented here requires no specialized skills or equipment, allows parallel fabrication of many templates at once, and can be adapted to fabricate ZMWs in different metals. In this work, copper and aluminum were substituted with aluminum and gold, respectively, to fabricate gold ZMWs (Figure S3). This is advantageous for labs that

use passivation methods of gold rather than aluminum. In addition, gold ZMWs have been shown to enhance emission for fluorophores that absorb in the red region of the visible spectrum, while aluminum ZMWs enhance emission for fluorophores that absorb in the green region⁵⁹. In the future, fluorescent signal intensity from ZMWs fabricated with this method might be enhanced by etching into the glass below the ZMW metal cladding using Piranha solution or HF^{16,26,60}. This brings the immobilized biomolecules farther away from the metal walls, which can quench fluorophores⁶¹ and has the added benefit of allowing an extra aluminum compatible cleaning step before passivation. Furthermore, there is a maximum in excitation illumination intensity below the entrance of the aperture, and this has been exploited previously to enhance single molecule emission^{26,60}.

Supplementary Material

Refer to Web version on PubMed Central for supplementary material.

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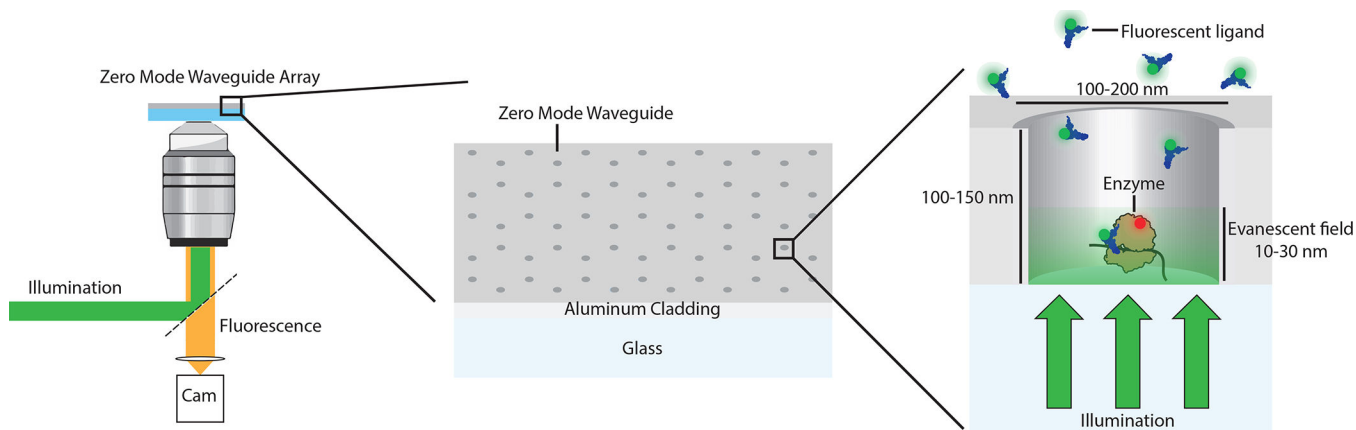


Figure 1. Schematic of zero mode waveguides.

Diagram of ZMW array with expanded cross-sectional diagram of a single ZMW on the right. Single fluorescently tagged enzymes of interest (brown ribosome with red circle to represent fluorescent dye) chemically immobilized (via the mRNA in this example) to the glass bottom of the ZMWs (typically functionalized with biotinylated-PEG) can be imaged with a typical laser-based epifluorescence microscopy setup. The 532 nm excitation light (green arrows) is reflected at the glass-metal boundary due to the small size of the aperture (100–200 nm diameter), but a non-propagating evanescent wave that decays exponentially in intensity is present within the ZMW. This results in a 10–30 nm effective illumination depth (green shading in aperture). Individual fluorescent ligands (blue tRNAs with green circles as fluorescent tags) at nM to μ M concentrations are added. An individual ligand that diffuses into the aperture and interacts with the enzyme is imaged without prohibitive background fluorescence.

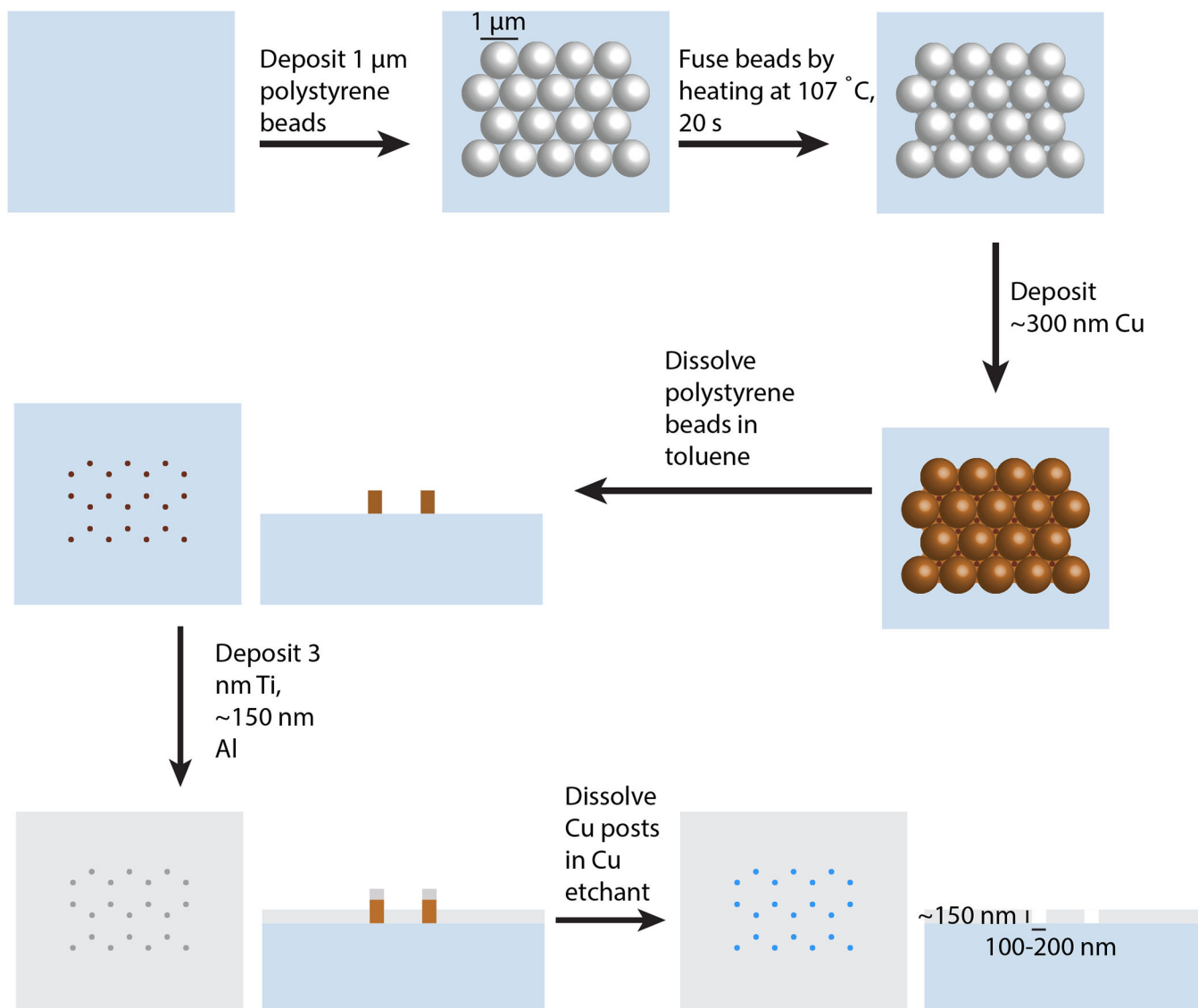


Figure 2. Schematic of colloidal templating method developed to fabricate aluminum ZMW arrays.

Polystyrene beads, 1 μm in diameter, are deposited and self-assembled on a cleaned glass coverslip, as described in section 2 of Protocol. Beads are then annealed to reduce pore sizes (section 3 of Protocol), followed by copper deposition and bead dissolution in toluene. Aluminum is deposited around and on top of the copper posts, which are then selectively etched away to leave behind a hexagonal array of nanoapertures (section 4 of Protocol). For the last three steps, cross-sectional views are provided to the right of the plan views to show the widths and heights of the copper posts and aluminum ZMWs.

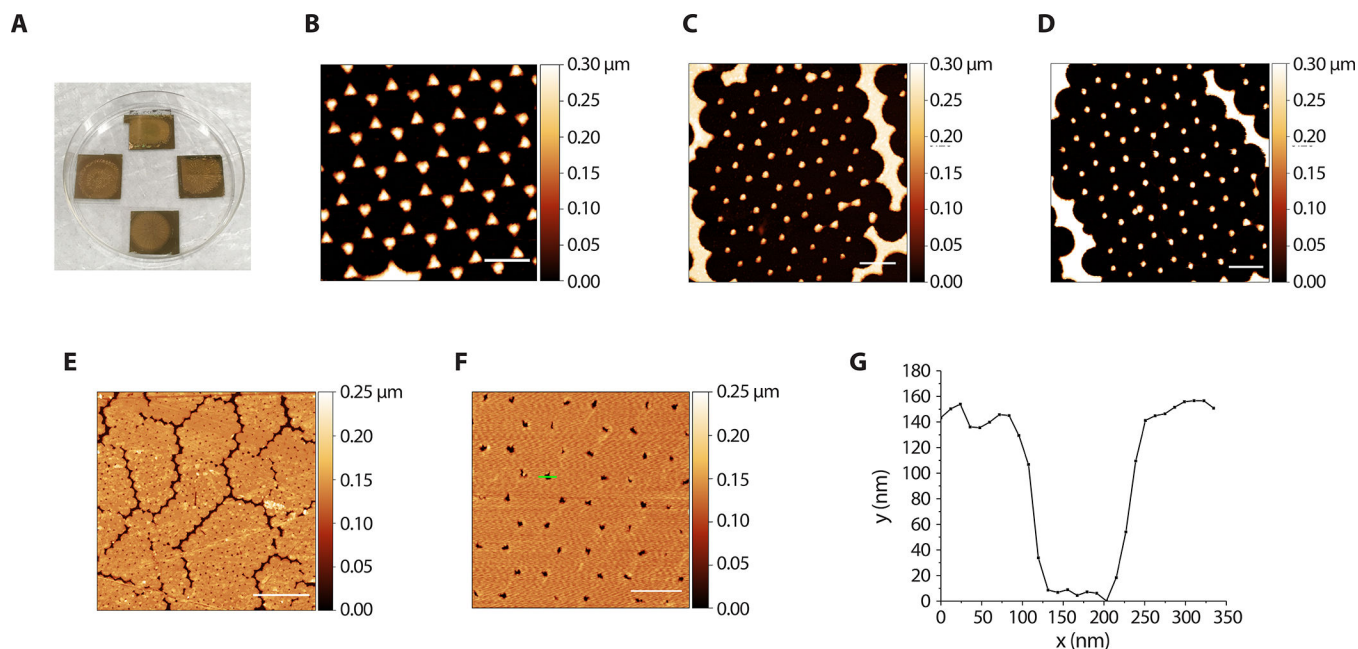


Figure 3. Representative results from evaporative deposition of colloids.

A. Example of optimal colloid deposition. **B.** Example of an acceptable colloid deposition in which conditions were more humid (80% RH) than ideal. Holes in the crystal monolayer are apparent. **C.** Example of an acceptable colloid deposition in which conditions were more dry (65% RH) than optimal. The monolayer regions are slightly translucent while multilayered areas are white and opaque (perimeter and streaks inward). **D.** A colloidal crystal illuminated with white light to highlight the rainbow diffraction from the crystals. **E.** Atomic force microscopy (AFM) image (tapping probe AFM in air) of a monolayer of hexagonally packed polystyrene beads from a successful colloid deposition (scale bar 10 μm). **F.** Expanded AFM image of packed beads (scale bar 2 μm).

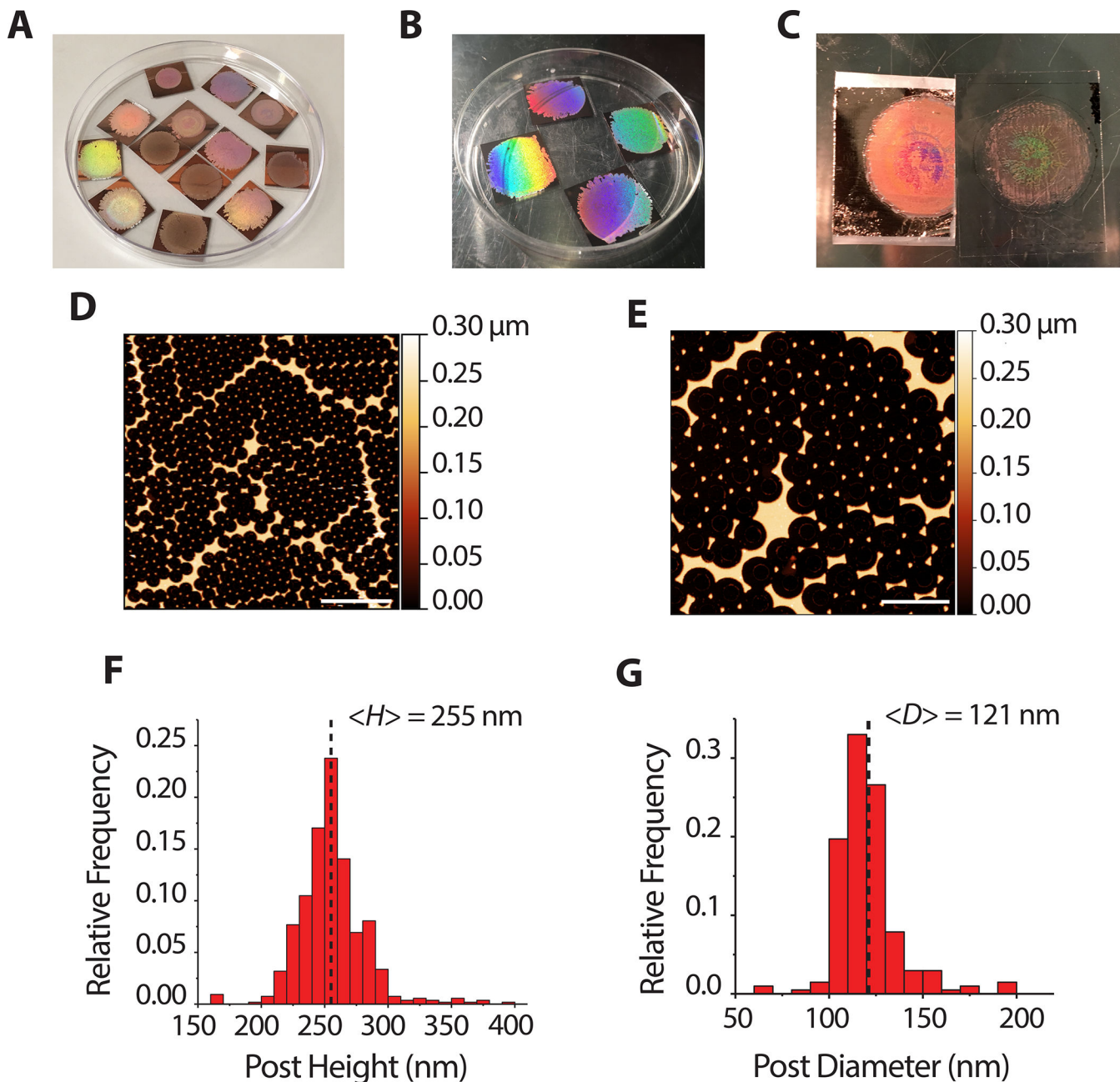


Figure 4. Macroscopic and microscopic images of ZMW templates after copper deposition.

A. Picture of slides after physical evaporation of copper on top of the bead templates. **B.** Rainbow diffraction pattern from bead templates after copper deposition. **C.** Picture of a template (right) after a tape pull to remove excess copper and the tape (left). **D.** AFM image of copper posts after tape pull and complete dissolution of the polystyrene beads (scale bar 5 μm). **E.** Higher magnification AFM image of D (scale bar 2 μm). **F.** Histogram of the copper post heights (defined as maximum height measurement within each post), $n = 534$. **G.** Histogram of the copper post Feret diameters, $n = 201$. Feret diameter is the maximum distance between two parallel lines tangent to the post boundary (quantified in ImageJ⁴⁷). To

identify particles for analysis, a threshold halfway between the top of the cladding and the bottom glass surface was used.

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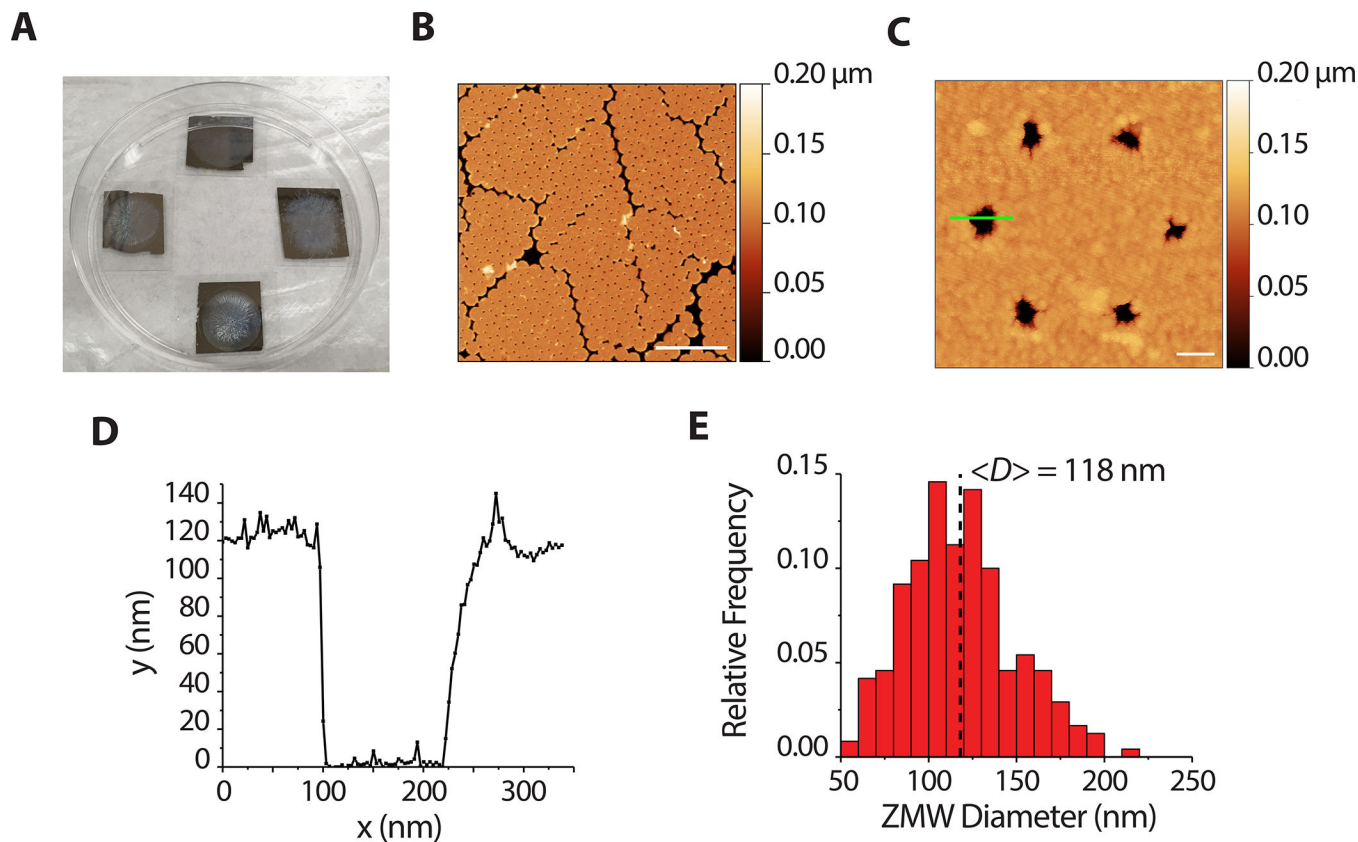


Figure 5. Macroscopic and microscopic images of aluminum ZMWs.

A. Picture of slides after physical evaporative deposition of 150 nm of aluminum around and on top of the copper posts. **B.** AFM image of aluminum ZMWs after post-dissolution (scale bar 5 μm). **C.** Higher magnification image of B (scale bar 0.2 μm). **D.** Typical depth profile of an individual ZMW from C. Profile taken from the green line drawn in C. **E.** Histogram of ZMW Feret diameters, $n = 240$. Feret diameters were measured as in Figure 4.

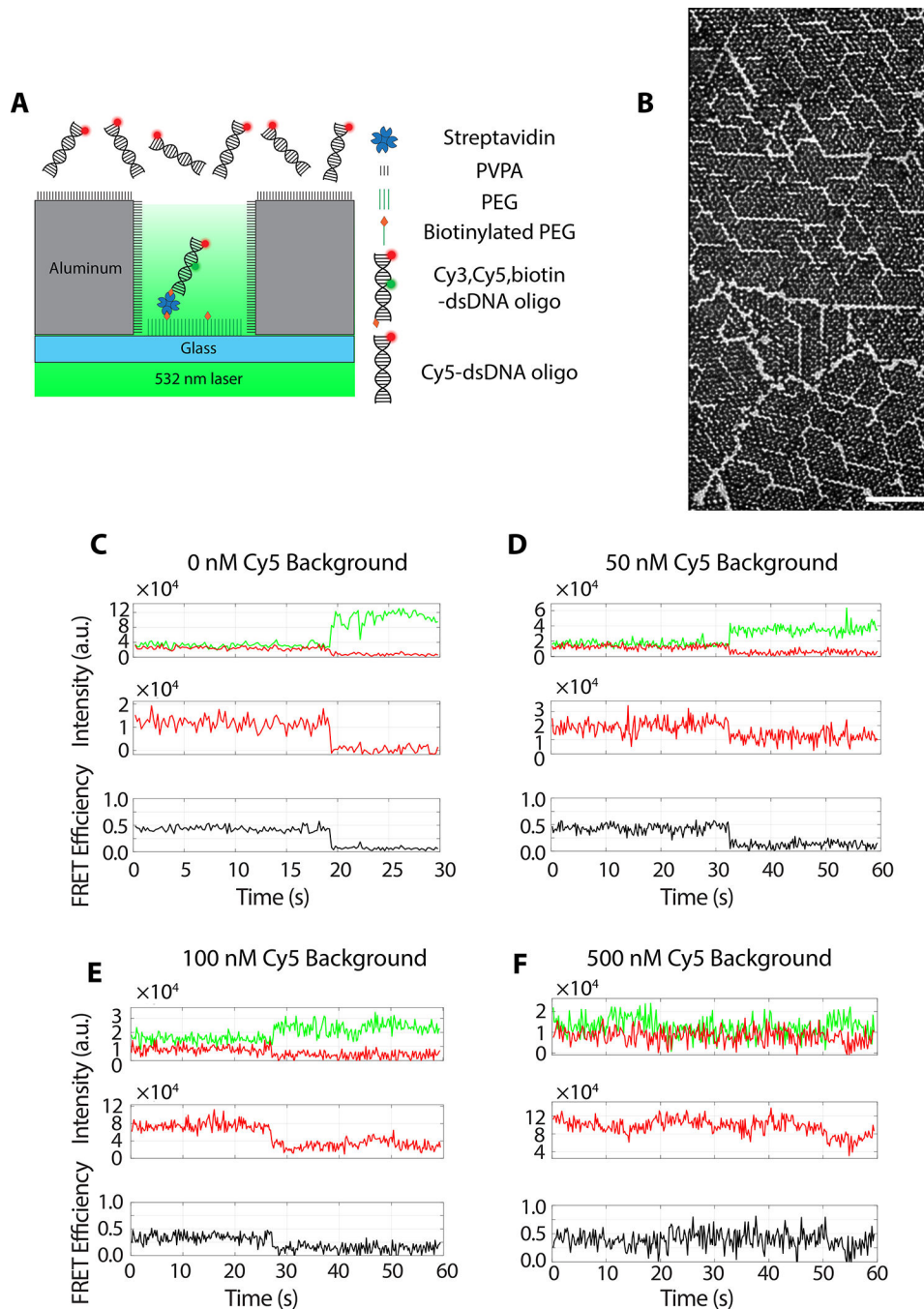


Figure 6. Single molecule FRET imaging in ZMWs.

A. Schematic (not to scale) of single molecule FRET imaging of Cy3, Cy5-labeled DNA duplexes in ZMWs with Cy5 labeled duplexes in the background. **B.** Example field of ZMWs under white light illumination (scale bar 10 μm). **C-F.** Single molecule FRET recordings of DNA duplexes immobilized in the ZMWs in the presence of 0 (**C**), 50 (**D**), 100 (**E**), and 500 nM (**F**) Cy5-labeled duplexes in solution. For each concentration, the top panel shows the Cy3 (green) and Cy5 (red) fluorescence intensity under 532 nm laser illumination (FRET imaging), middle panel shows the Cy5 fluorescence intensity under 640

nm laser illumination (direct acceptor excitation), and the lowest panel shows the FRET efficiency ($E_{apparent} = \frac{I_A}{I_A + I_D}$) calculated from the raw Cy3 (I_D) and Cy5 (I_A) fluorescence intensities. During imaging, the excitation wavelength alternated between 532 and 640 nm every 100 ms.