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Original Research

\cos e₂ nanobelt coupled with \cos OMoO₄ nanosheet as efficient electrocatalysts for hydrogen and oxygen evolution reaction

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1. Introduction

With the rising concern in energy crisis and environmental pollution, there has been an urgent need to exploit renewable and clean energy resources to reduce our reliance on fossil fuel. Hydrogen $(H₂)$ has been considered as one of the most promising energy carriers to satisfy future energy systems, due to its high energy density, renewability and pollution-free characteristics [\[1\]](#page-6-0). For example, H_2 has been used to produce electricity for vehicular and stationary applications to diminish the consumption of fossil fuel and $CO₂$ emissions [\[2,](#page-6-1)[3](#page-6-2)]. Until now, tremendous efforts have been devoted to synthesize H_2 , among which electrocatalytic water splitting for the generation of H_2 is one of the most environmentally friendly and economical way for the future H_2 economy [[4](#page-6-3),[5](#page-6-4)]. The electrochemical water splitting involves two half reactions, including hydrogen evolution reaction (HER) and oxygen evolution reaction (OER). Comparing with the HER, OER is concerned with various energy systems, such as metal-air batteries and regenerative fuel cells. Moreover, since the process usually involves four proton-coupled electrons transfer, O–H bond breaking, as well as attendant O–O bond formation, OER has been considered as a major bottleneck for realize large-scale water splitting [\[6\]](#page-6-5). Up to now, Pt, Ru and their alloys, are the most active electrocatalysts for the HER and OER, while their widespread application is serious restricted by their high cost and low earth-abundance [\[7,](#page-6-6)[8](#page-6-7)]. Hence, designing and exploiting active and cost-effective catalysts with earth abundant elements for replacing precious metals is still a major challenge [[9](#page-6-8)].

OER with a small Tafel slope of 43 mV dec⁻¹, as well as excellent stability in 1.0 M KOH. Therefore, this noblemetal-free and highly efficient catalyst enables prospective applications for electrochemical applications.

> According to the Volmer-Heyrovsky-Tafel models, the reaction rate for the HER is relies on the adsorption and desorption of H atoms on the catalyst surface in acidic conditions [\[10](#page-6-9)]. The Sabatier principle shows that an efficient catalyst should have a free energy of hydrogen adsorption (ΔG_H) that close to zero, in which the overall reaction rate is maximum [\[11\]](#page-6-10). Currently, numerous studies have focused on the development of non-noble-metal sulfides [[12,](#page-6-11)[13\]](#page-6-12), phosphides [\[14](#page-6-13)[,15](#page-6-14)], nitrides [\[16](#page-6-15)[,17](#page-6-16)], selenides [[18,](#page-6-17)[19\]](#page-6-18), and carbides [\[20](#page-6-19)[,21](#page-6-20)]. Due to the unique electronic configuration and the intrinsic metallic nature, cobalt chalcogenides, including CoS_2 and $CoSe_2$, have been prepared and offered a unique advantage as electrocatalysts for the HER, oxygen reduction reaction (ORR) [[22\]](#page-6-21), OER [\[23](#page-6-22)] and sodium-ion battery [\[24](#page-6-23)]. Though CoSe_2 shows extraordinary achievement towards the HER and OER, the low electrochemical performance of pure \cos ₂ are still restricted in terms of activity and stability. Therefore, it is highly desirable to exploit high active and stable catalyst with remarkable activity. Recently, the composites of $Cose_2$ with CeO_2 [[25\]](#page-6-24), WS_2 [\[26](#page-6-25)], $MoSe_2$ [[27\]](#page-6-26), and CoS_2^{28} were designed to enhance electrocatalytic activity. For

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example, Kang et al. $[29]$ $[29]$ prepared CoSe_2/CNT composite microspheres by spray pyrolysis and selenization, and the synthesized catalyst showed the superior electrocatalytic activity. Hou et al. [\[30](#page-6-28)] reported the fabrication of tubular-structured orthorhombic $CoSe₂$ (o-CoSe₂) and cubic CoSe₂ (c-CoSe₂) from Co₃Se₄ nanotubes and found they could be used as bifunctional electrocatalysts with high catalytic activity in alkaline medium for both dye-sensitized solar cells (DSSCs) and HER.

The strategy for the preparation of high activity electrocatalysts can be generally divided into two categories. The first one is dedicated to the construction and synthesis of different nanostructure catalysts, such as nanowires [\[31](#page-6-29)], nanoparticles [\[32](#page-6-30)], nanotubes [\[33](#page-6-31)], nanosheets [\[34](#page-6-32)], and nanobelts [\[35\]](#page-6-33), because they provide large surface area, enhance diffusion of active sites, and expose the reactive lattice planes by releasing the gas bubbles adsorbed onto the catalyst surface [\[36](#page-6-34)]. The other method involves the optimization of the electronic structure of catalysts via doping [[37,](#page-6-35)[38\]](#page-6-36), tuning the metal valence states and producing complex compounds [[39,](#page-6-37)[40\]](#page-6-38). In particular, chemical doping has been proposed as an attractive method to improve the catalytic activity of catalyst by means of altering the surface properties of catalyst, mainly including electronic band structure, synergistic coupling effect, and wettability properties [[41\]](#page-6-39). Additionally, the introduction of heteroatom into nanostructure could facilitate the hydrophily of the solid-state surface towards water, and then increase the contact surface between electrolyte and electrode, leading to a better electron transfer and electrocatalytic activity [\[42\]](#page-6-40). Further studies found that the introduction of S with unsaturated edge sites could efficiently promote proton discharge in the HER system [43]. Furthermore, the previous report suggested that the oxygen vacancy play a key role in improving the catalytic activity for the the HER system [\[43](#page-6-41)]. Furthermore, the previous report suggested that the oxygen vacancy play a key role in improving the catalytic activity for $CoSe₂/CoMoO₄$ at air $[44]$ $[44]$.

Hence, we put forward a facile route to prepare a novel hybrid nonnoble electrocatalyst via hydrothermal sulfuration or oxidation using the hybrid of $CoSe_2/CoMoO_4$ as precursor. The morphology, surface properties, structural composition and electrocatalytic activity of the synthesized hybrid catalysts were systematically investigated. The results show that the strongly coupled nature of CoMoO₄ and CoSe₂/DETA, the assistance of S heteroatom and unique nanosheet structure, promoted the electrocatalyst exhibits a highly electrocatalytic activity for the HER. Moreover, the introduction of oxygen vacancy improved the OER perassistance of S heteroatom and unique nanosheet structure, promoted the
electrocatalyst exhibits a highly electrocatalytic activity for the HER.
Moreover, the introduction of oxygen vacancy improved the OER per-
formance o significant approach to investigate inexpensive non-noble metal hybrid materials for application in water splitting.

2. Materials and methods

In this work, $Na₂Si$, $Na₂SeO₃$, diethylenetriamine (DETA) and CoCl2⋅6H2O are purchased from Tianjin Damao Chemical Reagent Factory. Co(CH3COO)2⋅4H2O Tianjin Fuchen Chemical Reagent Factory. All of reagents were used without further purification.

2.1. Materials preparation

The fabrication of $\cos \epsilon_2$ was according to the previous report [\[45](#page-6-43)]. In briefly, 1 mmol $Co(CH_3COO)_2 \cdot 4H_2O$ and 1 mmol Na₂SeO₃ were added into 40 mL solution that composed of diethylenetriamine (DETA) and distilled water (V:V $=$ 2:1), and the solution was stirred uniformly by using a magnetic stirrer for 0.5 h. The obtained mixture was then transferred to a 50 mL autoclave for reaction at 180 $^{\circ}$ C for 16 h. The precipitate was filtered, washed, and dried at 60 °C overnight.

2.2. Synthesis of CoSe₂/DETA

In a typical synthesis, 1 mmol CoCl2⋅6H2O, 10 mmol urea and 0.1766 g of (NH4)6Mo7O24⋅4H2O were dissolved in 40 mL of distilled water to form a clear solution. Next, 30 mg of CoSe₂/DETA was added and mixed in an ultrasonication bath for 15 min. After that, the mixed solution was transferred into a 50 mL autoclave and maintained at 180 $^{\circ}$ C for 12 h, followed by filtering and drying. The obtained precursor was denoted as CoMoO₄/CoSe₂.

2.3. Synthesis of $S-CoSe_2/CoMoO_4$

For preparing S –CoSe₂/CoMoO₄, 1 g of Na₂S was dissolved in the 40 mL DI water. Subsequently, 50 mg of CoSe₂/CoMoO₄ was added and the mixture was sonicated for 15 min to form a homogeneous dispersion, which was then transferred into a 50 mL autoclave and heated at 200 $^{\circ} \mathrm{C}$ for 6 h. The precipitate was filtered, washed, and dried at 60 °C overnight.

2.4. Synthesis of O-CoSe₂/CoMoO₄

For preparing O–CoSe₂/CoMoO₄, CoSe₂/CoMoO₄ was carbonized for 2 h at different target temperatures with a heating rate of 5 $^{\circ}$ C min⁻¹. The For preparing O–CoSe₂/CoMoO₄, CoSe₂
2 h at different target temperatures with a hefter was denoted as O–CoSe₂/CoMoO₄.

2.5. Physical measurements

Scanning electron microscopy (SEM) images were carried out on a Merlin scanning electron microscope at an accelerating voltage of 5 kV (Zeiss, Germany) to examine the morphologies of the samples. For analyzing the species of the catalysts, Energy-dispersive X-ray (EDX) elemental mapping mages were conducted on an X-MaxN20 dual detector system (Oxford, UK). X-ray photoelectron spectroscopy (XPS) measurements were determined on an ESCALAB 250Xi scanning X-ray microprobe (Thermo, USA). X-ray diffraction (XRD) patterns were performed on a Bruker diffractometer using Cu K α radiation ($\lambda = 1.5406$ nm) (Germany). Transmission electron microscopy (TEM) images were performed by using JEM-2100F microscope (Japan).

2.6. Electrochemical measurements

HER performances were conducted using CHI 760E electrochemical workstation (Shanghai Chenhua Co., China) in a conventional threeelectrode cell configuration in 0.5 M H₂SO₄. A saturated calomel electrode (SCE) and graphite rod were used as the reference and counter electrodes, respectively. A glassy carbon electrode (3 mm in diameter) was selected as the working electrode. For preparing catalyst ink, 2.5 mg of catalyst was dispersed in 500 μL of the mixture solution composed of 240 μL of ethanol, 240 μL of water and 20 μL of 5 wt% nafion 117 solution and sonicated for 30 min. Then, 5 μL of catalyst suspension was dropped onto the glassy carbon electrode surface and dried at room temperature. Linear sweep voltammetry was performed at a scan rate of 5 mV s^{-1} . EIS measurements were evaluated with frequency from 0.01 to $10⁵$ Hz at a potential of 0.15 V vs RHE. The electrochemical double-layer capacitances (C_d) was test within a non-Faradaic region from -0.1 to -0.3 V vs SCE. The EDLC was calculated according to the equation of C_{dl} $=(j_{anodic}j_{cathodic})/2R_{scan}.j_{anodic}$ and $j_{cathodic}$ are current densities measured at -0.2 V vs SCE. All the potentials reported in our work were converted to a reversible hydrogen electrode (RHE) and without iR-corrected.

The electrochemical OER test were carried out in a three-electrode configuration, with an Ag/AgCl and platinum foil as the reference and counter electrode, respectively. The working electrode was prepared by dropping 10 μL of the inks onto a polish and cleaned glassy carbon rotating disc electrode (RDE) with a diameter of 5 mm. The mass loading is about 0.225 mg cm⁻². The electrolyte (1.0 M KOH) was bubbled with O_2 for 30 min to ensure O_2 -saturation. The polarization curves were obtained at a scan rate of 5 mV s⁻¹ and rotating speed of 1600 rpm. The st $O₂$ for 30 min to ensure $O₂$ -saturation. The polarization curves were obtained at a scan rate of 5 mV s^{-1} and rotating speed of 1600 rpm. The for 1000 cycles by sweeping the potential from 0.8 to 0.2 V vs Ag/AgCl. stability test of O–CoSe₂/CoMoO₄ was performed by cyclic voltammetry for 1000 cycles by sweeping the potential from 0.8 to 0.2 V vs Ag/AgCl. EDLC was obtained in the non-Faradaic region (0.4–0.2 V vs. Ag/AgCl).

Fig. 1. Top, schematic illustration of the formation of S -CoSe2/CoMoO₄ and O -CoSe₂/CoMoO₄ catalysts. Bottom, SEM images of CoSe₂/DETA (a, b), CoSe₂/CoMoO₄ (c, d) and S -CoSe₂/CoMoO₄ (e, f) .

of S–CoSe₂/CoMoO₄. (e) XRD patterns of CoSe₂/DETA, CoSe₂/CoMoO₄, and S–CoSe₂/CoMoO₄. (f) Elemental mapping images of S–CoSe₂/CoMoO₄.

3. Results and discussion

The synthesis procedure of S –CoSe₂/CoMoO₄ is schematically illus-trated in [Fig. 1](#page-2-0). Firstly, CoSe₂/DETA nanobelts were prepared through used DETA as the reaction medium and $Na₂SeO₃$ as selenide source. Subsequently, metal ions and urea were simultaneously added into the solution containing CoSe₂/DETA nanobelts. This step involves the forused DETA as the reaction medium and Na₂SeO₃ as selenide source.
Subsequently, metal ions and urea were simultaneously added into the
solution containing CoSe₂/DETA nanobelts. This step involves the for-
mation of Co of nanobelts through the coordination effect of CoSe₂/DETA and metal solution containing CoSe₂/DETA nanobelts. This step involves the formation of Co–Mo precursors, which preferred nucleation on the surface of nanobelts through the coordination effect of CoSe₂/DETA and metal ions during was acquired via a facile thermal sulfuration process. In this step, the active sulfur anions that release from sodium sulfide could replace the OH⁻ or CO²⁻ anions of the precursors through the anion exchange reaction, leading to the formation of S-doped $\text{CoSe}_2/\text{CoMoO}_4$ hybrid nanosheet [[47\]](#page-6-45).

The morphology of CoSe₂/DETA and other hybrid materials are shown in [Fig. 1.](#page-2-0) CoSe₂/DETA with a belt morphology is clearly observed. Moreover, some particles on the surrounding of belt could be found, which may be attributed to incomplete reaction [\(Fig. 1](#page-2-0)a and b). As shown in [Fig. 1](#page-2-0)c and d, the $\text{CoSe}_2/\text{DETA}$ nanobelt is covered completely by the CoMoO₄ nanosheets with smooth surface, implying the excellent couple of nanobelt and nanosheet. As can be seen from Fig. 1e and f, the resultin CoMoO4 nanosheets with smooth surface, implying the excellent couple of nanobelt and nanosheet. As can be seen from [Fig. 1e](#page-2-0) and f, the nanoparticles are deposited onto the surface, which facilitated the adsorption of H atoms. The thickness of the nanosheets after sulfuration is increased compared with the initial precursor, suggesting the formation of S-doped CoSe₂/CoMoO₄ hybrid nanosheet.

XPS measurements were carried out to analyze the electronic state and elemental compositions of catalyst. As shown in [Fig. 2](#page-2-1), the XPS spectrum of CoSe₂/CoMoO₄ and S-CoSe₂/CoMoO₄ revealed the obvious signals of the C, Mo, Se, and O, as well as Co and S, indicating S was successfully introduced into $\text{CoSe}_2/\text{CoMoO}_4$ as expected ([Fig. 2a](#page-2-1)). As depicted in Co 2p spectrum ([Fig. 2b](#page-2-1)), the Co^{2+} oxidation state of CoMoO₄ is located at 780.5 eV and the binding energy at 782.5 eV corresponded to Co $2p_{3/2}$ [\[48](#page-6-46)]. The signal appeared at 798.4 eV is originated from the $Co²⁺$ cations of CoSe₂ [\[30](#page-6-28)[,49](#page-6-47)], while the satellite peaks at 786.7 and 803.3 eV corresponded to the shakeup-type peaks of Co at the high binding energy side of the edge [[50](#page-6-48)]. The peak fitting of high-resolution Mo 3d spectra suggested that there were two oxidation states for Mo on the surface of CoSe₂/CoMoO₄ [\(Fig. 2](#page-2-1)c). Corresponding features in the Mo 3d core level spectra at 232.3 and 235.4 eV are the characteristic of Mo

 $3d_{5/2}$ and Mo $3d_{3/2}$ of Mo⁶⁺, implying the presence of the Mo⁶⁺ oxidacan rates (e).
 $3d_{5/2}$ and Mo $3d_{3/2}$ of Mo⁶⁺, implying the presence of the Mo⁶⁺ oxidation state [[51\]](#page-6-49). For S–CoSe₂/CoMoO₄, the Co 2p region also exhibits several similar peaks at 781.3 (Co 2 $p_{3/2}$), 797.2 (oxidation state, Co²⁺), 786.5 and 803.1 eV (satellite peaks) [[52\]](#page-6-50). The Se 3d spectrum of tion state [51]. For S–CoSe₂/CoMoO₄, the Co 2p region also exhibits several similar peaks at 781.3 (Co 2p_{3/2}), 797.2 (oxidation state, Co²⁺), 786.5 and 803.1 eV (satellite peaks) [52]. The Se 3d spectrum of S–CoSe to CoSe2/CoMoO4. The peak at 54.7 eV can be attributed to the metal-selenium bond, whereas the peak at 59.8 eV is most likely related to SeO_x (Fig. S1) [[23,](#page-6-22)[53\]](#page-6-51). The S 2p singlet at 164.9 eV can be attributed to the sulfur-metal bonds [\(Fig. 2d](#page-2-1)). The high energy of 170.6 eV is in good agreement with the S^{4+} species at the surface or edges of to SeO_x (Fig. S1) [23,53]. The S 2p singlet at 164.9 eV can be attributed to the sulfur-metal bonds (Fig. 2d). The high energy of 170.6 eV is in good agreement with the S^{4+} species at the surface or edges of S–CoSe the sulfur-metal bonds (Fig. 2d). The high energy of 170.6 eV is in good agreement with the S^{4+} species at the surface or edges of S–CoSe₂/CoMoO₄ [54]. Thus, the XPS analysis confirmed that the composition of S–Co

into the structure of CoSe₂/CoMoO₄ through forming sulfur-metal bonds.
 XRD measurements were performed to investigate the structural information of samples. The XRD patterns of CoSe₂/DETA, CoSe₂/CoMoO₄, and S–C XRD measurements were performed to investigate the structural information of samples. The XRD patterns of $\text{CoSe}_2/\text{DETA}$, $\text{CoSe}_2/\text{CoMoO}_4$, and S– $\text{CoSe}_2/\text{CoMoO}_4$ are illustrated in Fig. 2e. The typical diffraction peaks of $CoSe_2/DETA$ at 33.5°, 45.1°, 50.8° and 62.4° can be indexed to the (210), (221), (311) and (400) planes, which was consistent with that of a pure cubic phase of the CoSe₂ standard pattern (JCPDS#09-234) [[55\]](#page-6-53). The peaks of $CoSe_2/CoMoO_4$ hybrid at 24.9° , 27.8° , 32.6° , 36.3° , 38.4°, 42.8°, 46.5° and 53.8° can be ascribed to $CoMoO₄$ % of a pure cubic phase of the CoSe₂ standard pattern (JCPDS#09–234) [55]. The peaks of CoSe₂/CoMoO₄ hybrid at 24.9°, 27.8°, 32.6°, 36.3°, 38.4°, 42.8°, 46.5° and 53.8° can be ascribed to CoMoO₄ (PDF#21–0868), and little shift, which further confirmed the $CoSe_2/DETA$ and $CoMoO₄$ is (PDF#21–0868), and the diffraction peaks of $\text{CoSe}_2/\text{DETA}$ presented a little shift, which further confirmed the $\text{CoSe}_2/\text{DETA}$ and CoMoO_4 is coupled. The XRD pattern of S–CoSe $_2/\text{CoMoO}_4$ exhibited obvious peaks at 23.7°, 26.2°, 33.7°, 37.3°, 43.2°, 46.0°, 47.8°, 53.8°, and 56.7°, corresponding to the (021), (003), (220), (312), (015), (042), (125), and coupled. The XRD pattern of S-CoSe₂/CoMoO₄ exhibited obvious peaks at 23.7°, 26.2°, 33.7°, 37.3°, 43.2°, 46.0°, 47.8°, 53.8°, and 56.7°, corresponding to the (021), (003), (220), (312), (015), (042), (125), and (051) at 23.7°, 26.2°, 33.7°, 37.3°, 43.2°, 46.0°, 47.8°, 53.8°, and 56.7°, corresponding to the (021), (003), (220), (312), (015), (042), (125), and (051) planes (PDF#30–0450). The TEM images and elemental mapping images of S– existed in the surface of sheet, which may be related to the incorporation of sulfur element. The elemental mapping further demonstrated the uniform distribution of Co, S, and Se. All these results revealed that existed in the surface of sheet, which may be related to the incorporation
of sulfur element. The elemental mapping further demonstrated the
uniform distribution of Co, S, and Se. All these results revealed that
S–CoSe₂/ accordance with the results of SEM and XPS. $S-CoSe₂/CoMoO₄$ was successfully prepared via sulfuration, which is in accordance with the results of SEM and XPS.
The HER catalytic activity of samples prepared with different molar ratios between Co and Mo were

The HER catalytic activity of samples prepared with different molar produces a current density (j) of 10 mA cm⁻² at an overpotential (η) of The HER catalytic activity of samples prepared with different molar
ratios between Co and Mo were shown in Fig. S2. S–CoSe₂/CoMoO₄ (1:1)
produces a current density (*j*) of 10 mA cm⁻² at an overpotential (*η*) of
17 the overpotential of 182 and 180 mV, respectively. The Tafel slope is another important criterion to evaluate the performance of catalyst, which refers to the additional voltage required to increase the catalytic current density [\[56](#page-6-54)]. A smaller Tafel slope means a faster increase in the

Fig. 4. SEM images of O-
prepared at 500 °C (g–j).

HER rate. The Tafel slope could be obtained by means of fitting the linear regions of Tafel plots according to the Tafel equation $\eta = a+b \log j$ (η is the overpotential, a is the Tafel constant, b is the Tafel slope, and regions of Tafel plots according to the Tafel equation $\eta = a + b \log j$ (η is the overpotential, a is the Tafel constant, b is the Tafel slope, and j is the shows a Tafel slope of 54 mV dec $^{-1}$, which is slightly higher than that of the overpotential, a is the Tafel constant, b is the Tafel slope, and *j* is the current density) [57].As depicted in Fig. 3b, S–CoSe₂/CoMoO₄ (1:1) shows a Tafel slope of 54 mV dec⁻¹, which is slightly higher than t $\rm dec^{-1}$). Three principal steps, including Volmer, Heyrovsky, and Tafel reactions with the corresponding Tafel slope of 120, 40, and 30 mV dec $^{-1}$, have been proposed for the conversion of \rm{H}^{+} to \rm{H}_{2} in an acidic medium [[58\]](#page-6-56). The Tafel slop of 42–54 mV dec⁻¹ reveals that electrochemical desorption is the rate-limiting step and the HER process follows the Volmer-Heyrovsky mechanism. This process contains two steps, a discharge step that converted protons into hydrogen atoms adsorbed on the surface of catalyst, and the second step involved the formation of surface-bound hydrogen molecules [\[59](#page-6-57)[,60](#page-6-58)]. Additionally, exchange current densities (j_0) is another inherent measure of activity for HER for the catalyst, which can be acquired by applying extrapolation method to the Tafel plots. The exchange current density was calculated to be 6.27 \times current densities (j_0) is another inherent measure of activity for HER for
the catalyst, which can be acquired by applying extrapolation method to
the Tafel plots. The exchange current density was calculated to be 6.27 \times 10⁻³ mA cm⁻²) and S-CoSe₂/CoMoO₄ (3:1, 8.30 \times 10⁻⁴ mA cm⁻²), indicating the best catalytic efficiency of S-CoSe₂/CoMoO₄ (1:1). % and be acquired by applying extrapolation method to e exchange current density was calculated to be 6.27 \times aich is larger than those of S–CoSe₂/CoMoO₄ (1:3, 1.96) and S–CoSe₂/CoMoO₄ (3:1, 8.30 \times 10⁻⁴ mA the Tafel plots. The exchange current density was calculated to be 10^{-3} mA cm⁻², which is larger than those of S–CoSe₂/CoMoO₄ (1:1).
 $\times 10^{-3}$ mA cm⁻²) and S–CoSe₂/CoMoO₄ (3:1, 8.30 $\times 10^{-4}$ mA indicatin ³ mA cm⁻², which is larger than those of S-CoSe₂/CoMoO₄ (1:3, 1.96
0⁻³ mA cm⁻²) and S-CoSe₂/CoMoO₄ (3:1, 8.30 × 10⁻⁴ mA cm⁻²),
icating the best catalytic efficiency of S-CoSe₂/CoMoO₄ (1:1).
As disp

41 mVdec $^{-1}$, while it exhibits the higher overpotential of 199 mV and the lower exchange current density (1.68×10^{-4} mA cm $^{-2}$) compared to that As displayed in Fig. S3. S–Mo/CoSe₂ shows the smallest Tafel slope of 41 mVdec⁻¹, while it exhibits the higher overpotential of 199 mV and the lower exchange current density $(1.68 \times 10^{-4} \text{ mA cm}^{-2})$ compared to that o activity with the overpotential of 235 mV and the Tafel slope of 60 mV ${\rm dec}^{-1}.$ The results demonstrated that there is a synergetic effect between of S–CoSe₂/CoMoO₄. In addition, S–Co/CoSe₂ presented the worst HER activity with the overpotential of 235 mV and the Tafel slope of 60 mV dec⁻¹. The results demonstrated that there is a synergetic effect between C activity with the overpotential of 235 mV and the Tafel s
dec⁻¹. The results demonstrated that there is a synergetic
Co and Mo in the S–CoSe₂/CoMoO₄ catalyst, resulting in th
of S–CoSe₂/CoMoO₄ superior to those ⁻¹. The results demonstrated that there is a synergetic effect between
and Mo in the S–CoSe₂/CoMoO₄ catalyst, resulting in the HER activity
i–CoSe₂/CoMoO₄ superior to those of each individual.
Electrochemical pe Co and Mo in the S–CoSe₂/CoMoO₄ catalyst, resulting in the HER activity of S–CoSe₂/CoMoO₄ superior to those of each individual.
Electrochemical performance for S–CoSe₂ and S–CoMoO₄ catalysts are depicted in Fi

inferior HER electrocatalytic activity with a lower overpotential of 203

mV at a current density of 10 mA $\rm cm^{-2}$ and a Tafel slope of 42 mV dec $^{-1}$. mV at a current density of 10 mA cm⁻² and a Tafel slope of 42 mV dec⁻¹.
HER performance of CoSe₂/DETA and S–CoSe₂/CoMoO₄ can be observed mV at a current density of 10 mA cm^{-2} and a Tafel slope of 42 mV dec⁻¹.
HER performance of $\text{CoSe}_2/\text{DETA}$ and S-CoSe₂/CoMoO₄ can be observed
in [Fig. 3.](#page-3-0) S-CoSe₂/CoMoO₄ shows higher electrocatalytic activity th CoSe₂/DETA. Such results suggested the excellent catalytic activity of HER performance of $\text{CoSe}_2/\text{DETA}$ and S–CoSe₂/CoMoO₄ can be observed
in Fig. 3. S–CoSe₂/CoMoO₄ shows higher electrocatalytic activity than
CoSe₂/DETA. Such results suggested the excellent catalytic activity of
 and CoSe₂/DETA, which is similar to the reported catalytic system of CoS₂/CoSe₂ [[28](#page-6-59)]. Electrochemical impedance spectroscopy (EIS) measurements were conducted to investigate the charge-transfer mechanism of HER involved. The low charge-transfer resistance (R_{ct}) that relates to a fast charge transfer at the interface between electrocatalyst and electro-lyte, implies excellent electrocatalysis for the HER [\[61](#page-6-60)]. As presented in Fig. 3c, the solution resistance calculated is about 11.58 Ω and R_{ct} decreases with the increase of overpotential, which is in agreement with [Fig. 3c](#page-3-0), the solution resistance calculated is about 11.58 Ω and R_{ct} decreases with the increase of overpotential, which is in agreement with the parameter to evaluate the catalytic response of catalyst. The current density-time (i-t) test was performed at overpotential of 178 mV vs RHE for 10000 s [\(Fig. 3d](#page-3-0)), while the current density was found slightly decline, which may be due to the H_2 bubbles gather on the electrode surface and result in a larger interface resistance [[62\]](#page-6-61). The electrochemical double-layer capacitance (C_{dl}) between the catalysts and electrolyte was obtained by using the cyclic voltammetry (CV) method. CV curves was acquired in the potential range without a redox process from 0.173 to - 0.027 V vs RHE at a scan rates from 20 to 200 mV s^{-1} at 0.5 M H2SO4 (Fig. S5- Fig. S6). Based on the CV curves and the plots of the curves was acquired in the potential range without a redox process from 0.173 to - 0.027 V vs RHE at a scan rates from 20 to 200 mV s^{-1} at 0.5 M H₂SO₄ (Fig. S5- Fig. S6). Based on the CV curves and the plots of the presents the highest C_{dl}, implying its high effective surface area and abundant active sites for the HER. Therefore, the comparison of electrocatalytic properties of different catalysts suggested that S doping and the interaction of CoMoO₄ and CoSe₂/DETA are in favor of the improvement of catalytic activity of catalysts.
SEM images and EDX mapping of O–C the interaction of CoMoO₄ and CoSe₂/DETA are in favor of the improvement of catalytic activity of catalysts.

different oxidization temperatures are displayed in [Fig. 4.](#page-4-0) The obtained

1000 th cycles (c) and Tafel plots (d) of O–CoSe₂/CoMoO₄ prepared at 500 °C. The chronoamperometric curve (e). Capacitive current at 0.3 V (vs Ag/AgCl) as a function of scan rate for O–CoSe₂/CoMoO₄.
O–CoSe₂/CoMo Fig. 5. LSV curves (a) and the correspondin
1000 th cycles (c) and Tafel plots (d) of O-
function of scan rate for O–CoSe₂/CoMoO₄.

aggregated particles [\(Fig. 4a](#page-4-0)–f). The void that existed in the sheet can efficiently boost the electron transport and provides enough void volume
for releasing gas, and then enhances the catalytic activity of catalyst.
Compared with the initial morphology of $\text{CoSe}_2/\text{CoMO}_4$, the thickness
of for releasing gas, and then enhances the catalytic activity of catalyst. Compared with the initial morphology of CoSe₂/CoMoO₄, the thickness dization, which may be due to the low thermal stability of CoSe₂/CoMoO₄. Furthermore, with the increase of oxidization temperature, most of sheet structures are transferred into the granular, leading to the decrease of CoMoO₄. Furthermore, with the increase of oxidization temperature, most of sheet structures are transferred into the granular, leading to the decrease of available active sites. EDX mapping of O-CoSe₂/CoMoO₄ was con most of sheet structures are transferred into the granular, leading to the most of sheet structures are transferred into the granular, leading to the decrease of available active sites. EDX mapping of O–CoSe₂/CoMoO₄ was conducted to confirm the presence of CoSe₂/DETA (Fig. 4g–j), the elemen decrease of available active sites. EDX mapping of O–CoSe₂/CoMoO₄ was
conducted to confirm the presence of CoSe₂/DETA (Fig. 4g–j), the
elemental mappings of O–CoSe₂/CoMoO₄ demonstrate Co, Mo, Se and O
are distrib different oxidization temperatures are shown in Fig. S7. XRD pattern of elemental mappings of O–CoSe₂/CoMoO₄ demonstrate Co, Mo, Se and O
are distributed uniformly. XRD patterns of O–CoSe₂/CoMoO₄ prepared at
different oxidization temperatures are shown in Fig. S7. XRD pattern of
the hy are distributed uniformly. XRD patterns of O–CoSe₂/CoMoO₄ prepared at different oxidization temperatures are shown in Fig. S7. XRD pattern of the hybrid that contains the diffraction peak of CoMoO₄ (PDF#21–0868) and higher temperature leads to the enhancement of crystalline forms of $Co₃O₄$ and a decrease of the diffraction peak of $CoMoO₄$, which may be related to the transformation occurred from CoMoO₄ to metal oxides.

The electrocatalytic OER performance was first evaluated by polarization curves using RDE in O_2 -saturated 1.0 M KOH electrolyte. The influence of oxidization temperature on electrochemical activity of catalyst is shown in [Fig. 5.](#page-5-0) $RuO₂$ exhibits superior performance compared to zation curves using RDE in O₂-saturated 1.0 M KOH electrolyte. The in-
fluence of oxidization temperature on electrochemical activity of catalyst
is shown in Fig. 5. RuO₂ exhibits superior performance compared to
O–Co density of 10 mA cm⁻² with a Tafel slope of 53 mV dec⁻¹. O–CoSe₂/CoMoO₄ (500 °C) (η = 471 mV) presents the optimal OER catalytic activity, which is better than that of O–CoSe₂/CoMoO₄ (600 °C) (η = 485 mV % y of catalyst
ompared to
the current
. O–CoSe₂/ CoMoO₄ (500 °C) (η = 471 mV) presents the optimal OER catalytic activity, which is better than that of O-CoSe₂/CoMoO₄ (600 °C) (η = 485 O–CoSe₂/CoMoO₄ with a low overpotential (η) of 379 mV at the current density of 10 mA cm⁻² with a Tafel slope of 53 mV dec⁻¹. O–CoSe₂/CoMoO₄ (500 °C) (η = 471 mV) presents the optimal OER catalytic activ CoMoO₄ (500 °C) (η = 471 mV) presents the optimal OER catalytic activity, which is better than that of O–CoSe₂/CoMoO₄ (600 °C) (η = 485 mV) and O–CoSe₂/CoMoO₄ (700 °C) (η = 497 mV) at j = 10 mA cm⁻² tivity, which is better than that of O–CoSe₂/CoMoO₄ (600 °C) ($\eta = 485$ mV) and O–CoSe₂/CoMoO₄ (700 °C) ($\eta = 497$ mV) at $j = 10$ mA cm⁻² (Fig. 5a). For O–CoSe₂/CoMoO₄ (500 °C), O–CoSe₂/CoMoO₄ (600 °C) a dec $^{-1}$, which are significantly lower than commercial RuO $_2$, suggesting (Fig. 5a). For O–CoSe₂/CoMoO₄ (500 °C), O–CoSe₂/CoMoO₄ (600 °C)
and O–CoSe₂/CoMoO₄ (700 °C), the Tafel slopes are 43, 44 and 49 mV
dec⁻¹, which are significantly lower than commercial RuO₂, suggesting
the decreases with the increase of oxidization temperature, due to the disappearance of sheet-like structure and hole, leading to the decline of exposed active sites. Furthermore, according to the results of XRD, several phases existed in hybrid may restrain the utilization of efficient active sites.

Besides such extraordinary catalytic activity, O-CoSe₂/CoMoO₄ also

displays excellent stability toward the OER. As illustrated in [Fig. 5](#page-5-0)c and d, the polarization curves of initial and 1000 th cycles are almost coincident. There is merely 4 mV positive shift at $j = 10$ mA cm⁻² and negli-
gible change of Tafel slopes after 1000 th cycles. High stability is great
importance for energy conversion systems, the chronoamperometric
method gible change of Tafel slopes after 1000 th cycles. High stability is great importance for energy conversion systems, the chronoamperometric $CoMoO₄$ at an overpotential of 0.64 V (vs Ag/AgCl). A high current retention of 96.71% can be observed in [Fig. 5e](#page-5-0), implying its outstanding durability in continuous oxygen evolution process. The high catalytic activity could be elucidated by ECSA of the materials, which can be calculated by measuring the CV curves at different scan rates (Figs. S8 \sim durability in continuous oxygen evolution process. The high catalytic activity could be elucidated by ECSA of the materials, which can be calculated by measuring the CV curves at different scan rates (Figs. S8 \sim S10). cm^{-2}) is much higher than those of O–CoSe₂/CoMoO₄ (600 °C, 15.04 mF ity could be elucidated by ECSA of the materials, which can be lated by measuring the CV curves at different scan rates (Figs. S8 \sim As shown in Fig. 5f, the C_{dl} of O–CoSe₂/CoMoO₄ (500 °C, 21.16 mF) is much higher calculated by measuring the CV curves at different scan rates (Figs. S8 ~ S10). As shown in Fig. 5f, the C_{dl} of O–CoSe₂/CoMoO₄ (500 °C, 21.16 mF cm⁻²) is much higher than those of O–CoSe₂/CoMoO₄ (600 °C, 15.04 attributed to enhanced dispersion of nanosheet with fully exposed and easily accessible active sites. XPS characterization was further applied to CoMoO4. As can be seen from Fig. S11, Co, Mo, Se and O are homogeneously distributed among the whole hybrid samples. The deconvolureveal the existence of oxygen vacancies on the surface of O–CoSe₂/CoMoO₄. As can be seen from Fig. S11, Co, Mo, Se and O are homogeneously distributed among the whole hybrid samples. The deconvolutions of the O 1s sp assumption of two species. The peak centered at 530.5 eV is associated with oxygen atoms bound to metals, and the peak at 531.7 eV is related to abundant defect sites with a low oxygen coordination [\[63](#page-6-62)]. It is believed that a number of oxygen vacancies in ultrathin nanosheets of with oxygen atoms bound to metals, and the peak at 531.7 eV is related to abundant defect sites with a low oxygen coordination [63]. It is believed that a number of oxygen vacancies in ultrathin nanosheets of $O-CoSe_2/CoMo_4$

4. Conclusion

We have successfully fabricated the S –CoSe₂/CoMoO₄ and O–CoSe₂/ CoMoO4 hybrid composite as earth-abundant and high-performance catalysts for the HER and OER. The as-prepared $S-CoSe_2/CoMoO_4$ hybrid exhibits superior HER catalytic activity with a small of overpotential (177 mV) at the current density of 10 mA cm^{-2} , and a small catalysts for the HER and OER. The as-prepared S–CoSe₂/CoMoO₄ hybrid exhibits superior HER catalytic activity with a small of over-potential (177 mV) at the current density of 10 mA cm⁻², and a small Tafel slope of also displays excellent electrocatalytic activity toward OER, affording a current density of at an overpotential of 471 mV and a low Tafel slope of 43 mV dec⁻¹, as well as negligible catalytic deactivation after 8 h operation. The excellent properties of the two hybrids may be due to four reasons: (1) $\cos\theta$ /DETA with partial activity could provide a substrate to support the generation of $CoMoO₄$ nanosheet. (2) The excellent coupling between the CoSe₂/DETA nanobelts and CoMoO₄ nanosheet offers abundant active sites for the HER and OER. (3) S doping changes the electronic energy structures and enhances adsorption of H atoms on the catalysts offers abundant active sites for the HER and OER. (3) S doping changes the electronic energy structures and enhances adsorption of H atoms on (4) The formation of oxygen vacancies in the ultrathin nanosheets can significantly increase the number of active sites and facilitate adsorption of H2O, resulting in an increase in the OER catalytic activity.

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Appendix A. Supplementary data

Supplementary data to this article can be found online at [https://doi.](https://doi.org/10.1016/j.ese.2019.100004) [org/10.1016/j.ese.2019.100004](https://doi.org/10.1016/j.ese.2019.100004).

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