

Article **Stretchable Surface Electrode Arrays Using an Alginate/PEDOT:PSS-Based Conductive Hydrogel for Conformal Brain Interfacing**

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Abstract: An electrocorticogram (ECoG) is the electrical activity obtainable from the cerebral cortex and an informative source with considerable potential for future advanced applications in various brain-interfacing technologies. Considerable effort has been devoted to developing biocompatible, conformal, soft, and conductive interfacial materials for bridging devices and brain tissue; however, the implementation of brain-adaptive materials with optimized electrical and mechanical characteristics remains challenging. Herein, we present surface electrode arrays using the soft tough ionic conductive hydrogel (STICH). The newly proposed STICH features brain-adaptive softness with Young's modulus of ~9.46 kPa, which is sufficient to form a conformal interface with the cortex. Additionally, the STICH has high toughness of ~36.85 kJ/mm³, highlighting its robustness for maintaining the solid structure during interfacing with wet brain tissue. The stretchable metal electrodes with a wavy pattern printed on the elastomer were coated with the STICH as an interfacial layer, resulting in an improvement of the impedance from 60 k Ω to 10 k Ω at 1 kHz after coating. Acute in vivo experiments for ECoG monitoring were performed in anesthetized rodents, thereby successfully realizing conformal interfacing to the animal's cortex and the sensitive recording of electrical activity using the STICH-coated electrodes, which exhibited a higher visual-evoked potential (VEP) amplitude than that of the control device.

Keywords: stretchable electronics; soft electronics; implantable electronics; brain interface; electrocorticogram; conductive hydrogel

1. Introduction

Brain interface technology is a bio-electronic bridging platform for monitoring brain activity $[1-5]$ $[1-5]$ or modulating brain function $[6-17]$ $[6-17]$ by connecting electronic devices to the neurological system. Owing to its technological-convergence-related benefits [\[18–](#page-11-4)[25\]](#page-12-0), brain interface platforms could be applied to various advanced biomechatronic-associated areas that include: (i) biomedical applications, such as the diagnosis and therapy of neuropathy, daily biomonitoring, healthcare, recovery of brain function from trauma or injury, rehabilitation, and prosthetics for quadriplegia and motor or sensory dysfunction;

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(ii) human–machine connection for the remote control of objects and avatar robotics; (iii) human enhancement technology, such as memory reinforcement and cognitive function extension; and (iv) tangible user-experience media, such as immersive virtual or augmented reality, metaverse contents, and realistic interactive gaming.

Recently, the expected ripple effect and utilization of brain interface technology have become significant in the biomedical engineering and neuroscience domains. In particular, the acquisition of the electrical activity generated from cranial neurons using large-area, high-density integrated multielectrode arrays permits the digitization of diagnostic information, such as the location, range, and degree of brain damage, qualitatively and quantitatively, by capturing the pathological neurosignals caused by brain disorders [\[26](#page-12-1)[–37\]](#page-12-2). Furthermore, the real-time recording of neural activity progression enables detailed observations of the developmental pattern of pathological symptoms and the therapeutic effects of neurostimulation [\[8](#page-11-5)[,38\]](#page-12-3). Ultimately, the multidimensional spatiotemporal mapping of the genesis, propagation, and distribution of electrical activity from the brain with high resolution and fidelity facilitates advanced neuroscience research by identifying the brainwave patterns that involve and control physical activities, such as thinking, association, sensation, perception, cognition, behavior, motion, exercise, learning, memory, and noegenesis [\[27](#page-12-4)[–31](#page-12-5)[,39–](#page-12-6)[56\]](#page-13-0).

To pursue the aforementioned objectives, sensor systems for acquiring brain-activityrelated electrophysiological information were developed and actively researched. These brain-interfacing sensor systems acquire the electrical brain activity propagated from cranial nerves through the following approaches based on the access to brain nerves: (i) electroencephalography (EEG), (ii) electrocorticography (ECoG), (iii) local field potential (LFP) analysis, and (iv) single-neuron spiking [\[1\]](#page-11-0). In particular, the ECoG monitoring platform has various advantages compared to the others. Because electrical signals are recorded from intracranial-implanted electrodes in this approach, an intimate bio-electronic interface with excellent impedance properties could be readily formed via close contact with neurons, leading to the acquisition of brain activity signals with a higher signal-to-noise ratio (SNR) than that of EEG platforms that are often disturbed by hair [\[57\]](#page-13-1). In terms of diagnosing brain disease in clinical settings, ECoG directly provides more diagnostically significant information than that derived from EEG methods such as high-frequency oscillation (HFO) as a presymptom of epileptic seizure frequently not transmitted across the skull and cannot be detected from the scalp [\[27](#page-12-4)[–29](#page-12-7)[,31](#page-12-5)[,33\]](#page-12-8). Compared to the probe-type electrodes penetrating brain tissue with high modulus and stiffness for recording LFPs or spikes [\[4,](#page-11-6)[5,](#page-11-1)[58–](#page-13-2)[60\]](#page-13-3), ECoG devices are safe and biocompatible tools that could be used for prolonged durations owing to their non-invasiveness into the brain tissue, thereby advantageous for large-area brain monitoring with multielectrode arrays in a matrix formation. Implementing the spatiotemporal topography of ECoG activity with high resolution, fidelity, and an SNR with large-area coverage is critical for the detailed documentation and imaging of brain activity development, tracing segmentalized functional brain maps for use in various brainassociated applications, and in-depth neuroscience studies [\[27](#page-12-4)[–31\]](#page-12-5). Advanced surface electrode devices have been developed for ECoG technology using inorganic materials, such as metals and ceramics, to conveniently achieve high scalability for ultrafine patterning and high integration density, large-area uniformity, high yield, high throughput, mass productivity, guaranteed electrical performance, and long-term durability [\[27–](#page-12-4)[31](#page-12-5)[,61\]](#page-13-4). Ultrathin-film devices that are sub-tens of micrometers in thickness exhibit mechanical deformation characteristics, such as flexibility, bendability, and foldability, by drastically lowering the stiffness and bending the radius of the system, and their mesh-like electronic pattern improves adaptability by lowering the effective surface moduli [\[62\]](#page-13-5). The wavy interconnect pattern of electronic devices imparts mechanical stretchability to the systems, and stretchable thin-film devices printed on the soft stretchable substrates could be integrated with biological tissue as artificial electronic integuments.

However, these rigid inorganic materials and polymers with high Young's moduli $(10^2 \text{ MPa} \sim 10^2 \text{ GPa})$ have intrinsic limitations in achieving adaptive conformal interfacing with the cortex, which is the softest tissue (almost 1 kPa Young's modulus) in the human body [\[63\]](#page-13-6). In addition, connecting a biometric system operated with ion transportation with electrical sensing devices without ionic conductivity also has fundamental limitations to implementing an adaptive bio-electronics interface with low electrochemical impedance $[63,64]$ $[63,64]$. There have been massive efforts to introduce soft conductive interfacial organic materials to be used in the form of a coating layer on the electrode surface to bridge electronic devices and biological tissues while matching surface moduli for adaptive conformal bio-electronic interfaces [\[65](#page-13-8)[–80\]](#page-14-0). However, the implementation of functional brain-interfacing materials capable of both high toughness and low Young's moduli, which desirably feature high compatibility to soft brain tissue and high durability while being coated on an elastic electrode surface, is still challenging.

Herein, we developed stretchable surface electrode arrays using the newly proposed soft tough ionic conductive hydrogel (STICH) composites with a triple-network structure consisting of poly (3,4-ethylenedioxythiophene):polystyrene sulfonate (PEDOT:PSS), alginate (Alg), polyacrylamide (PAAm), and carboxymethyl cellulose (CMC) for adaptive conformal brain interfacing (Figure [1\)](#page-2-0). Alg, CMC, and PAAm are most widely used for preparing hydrogel-based applications which feature low moduli, high biocompatibility,
(PECs) good processability, and good ionic conductivity via polyelectrolyte complexes (PECs) between polyanions and polycations [\[81,](#page-14-1)[82\]](#page-14-2). In addition, compared to other hydrogels, alg−die to heat, so when used as when used a Alg-based hydrogels have less dynamic change properties due to heat, so when used as implant electrodes later, significant changes in the adjusted mechanical properties are not expected [\[83](#page-14-3)[,84\]](#page-14-4). We fabricated and optimized the STICH for brain interfacing using these expected [83-84]. We fabricated and optimized the STICH for brain interfacing using these biomaterials and employed the new composite material for ECoG monitoring. The in vivo ECoG monitoring capability of our stretchable electrodes with the STICH was illustrated ECoG monitoring capability of our stretchable electrodes with the STICH was illustrated The thermo-curable STICH comprises a densely networked structure featuring (Figure [1a](#page-2-0)). The thermo-curable STICH comprises a densely networked structure featuring ing interactions and entanglements (Figure [1b](#page-2-0)). biomaterials and employed the new composite material for ECoG monitoring. The in vivo

Figure 1. Illustrations of the stretchable surface electrode arrays using the soft tough ionic conductive tive hydrogel (STICH) for adaptive conformal brain interfacing. (**a**) In vivo demonstration of ECoG hydrogel (STICH) for adaptive conformal brain interfacing. (**a**) In vivo demonstration of ECoG monitoring using an animal model, in which the electrode device is mounted on the cerebral cortex monitoring using an animal model, in which the electrode device is mounted on the cerebral cortex of a rat (left). The STICH layer coated on the metal electrode channel provides adaptive brainelectronics interfacing owing to its soft modulus and conformal contact with the cortex (right). (**b**) Three types of gel networks: intertwined, joined via covalent bonding (black ovals), and showing μ and μ and μ and μ and μ and the aminometric groups of polyacrylamide chains and the carboxylamide chains and the carboxylamide chains and the carboxylamide chains and the carboxylamide chains and μ entanglements (yellow ovals) between the amine groups on polyacrylamide chains and the carboxyl

entanglements (yellow ovals) between the amine groups on polyacrylamide chains and the carboxyl groups on the Alg and CMC chains. (acrylamide, AAm; alginate, Alg; carboxymethyl cellulose, CMC; polyacrylamide, PAAm; poly(3,4-ethylenedioxythiophene):poly(styrenesulfonate), PEDOT:PSS).

2. Materials and Methods

2.1. Preparation of the Alg, CMC, and PAAm-Based Conductive Hydrogels

The hydrogel materials were prepared using our previously reported one-pot synthesis procedure for Alg/CMC/PAAm-based conductive hydrogels [\[83\]](#page-14-3). Initially, polyanion precursors in various proportions (100% Alg (sodium alginate, Sigma-Aldrich, Inc., St. Louis, MO, USA), 100% carboxymethylcellulose sodium salt (CMC, low viscosity, Sigma-Aldrich, Inc., St. Louis, MO, USA), and a 1:1 Alg:CMC mixture were mixed with deionized

water at 2 wt.%. Subsequently, acrylamide (AAm, suitable for electrophoresis, ≥99%, Sigma-Aldrich, Inc., St. Louis, MO, USA) (16 wt.% of the polyanion precursor) was dissolved in the different polyanion solutions, which were then vigorously stirred for 30 min until homogeneous solutions were obtained. N,N'-methylenebisacrylamide (MBAA, powder, for molecular biology, suitable for electrophoresis, ≥99.5%, Sigma-Aldrich, Inc., St. Louis, MO, USA) (10⁻² wt.% of the total deionized water content), Clevios[™] PH 1000 (aqueous PEDOT/PSS dispersion, blue liquid, Heraeus, Ohio, Dayton, USA) (20 wt.% of the total deionized water content), and ammonium persulfate (APS, ACS reagent, ≥98.0%, Sigma-Aldrich, Inc., St. Louis, MO, USA) (6 wt.% of the total AAm content) were then added to the mixed solutions, followed by *N*,*N*,*N'*,*N'*-Tetramethylethylenediamine (TEMED, BioReagent, suitable for electrophoresis, ~99%, Sigma-Aldrich, Inc., St. Louis, MO, USA) (0.1 wt.% of the deionized water). The resulting solutions were transferred to manually manufactured molds and maintained at 60 \degree C for 30 min. The compositions of the prepared STICH and control materials are listed in Table [1.](#page-3-0)

Table 1. The precursor compositions of different Alg, CMC, and PAAm-based conductive hydrogels.

Alg = sodium alginate; AAm = acrylamide; CMC = carboxymethyl cellulose; Clevios™ PH 1000 = commercial PEDOT/PSS solution product; $MBAA = N N'$ methylenebisacrylamide.

2.2. Physicochemical Characterization of the Alg, CMC, and PAAm-Based Conductive Hydrogels

The morphology of the STICH and control materials were analyzed with scanning electron microscopy (SEM; JSM-7600F Schottky-field-emission scanning electron microscope; JEOL USA, Inc., Peabody, MA, USA). To investigate the chemical compositions of the Alg, CMC, and PAAm in the STICH as well as their interactions, the STICH and the controls were analyzed with attenuated-total-reflectance Fourier transform infrared (ATR-FTIR) spectroscopy (Bruker IFS-66/S, TENSOR27, Bruker, Republic of Korea).

2.3. Mechanical and Rheological Characterization of the Alg, CMC, and PAAm-Based Conductive Hydrogels

To measure the toughness and Young's moduli of the STICH and controls, cylindrical samples were fabricated using a premade cylindrical mold (2.5 \times 2.5 \times 30 mm³). The measurements were performed using a universal testing machine (UTM; 34SC-1 Single Column Table Model, INSTRON, Norwood, MA, USA) in two modes: continuous strain at a rate of 20 mm/min and cyclic stretching (approximately 10 times) from 0% to 100% strain at a rate of 10 mm/min. The Young's moduli (*E,* kPa) and toughness (*U_T,* kJ/m³) were calculated using the following equations:

$$
E = \sigma/\varepsilon \tag{1}
$$

$$
U_T = \int_0^{\varepsilon} \sigma d\varepsilon \tag{2}
$$

where σ is the uniaxial stress (N/m^2) and ε is the uniaxial strain.

2.4. Mechanical and Rheological Characterization of the Alg, CMC, and PAAm-Based Conductive Hydrogels

The rheological behavior of the STICH and the controls were investigated using a Discovery Hybrid Rheometer 2 (TA Instruments, New Castle, DE, USA) in the oscillationfrequency-sweep mode. The samples used in these experiments were prepared using manually fabricated cylindrical molds (12.5 \times 12.5 \times 1 mm³). All rheological measurements were performed using parallel-plate geometry (diameter: 20 mm). The oscillationfrequency-sweep measurements were used to evaluate the variation in storage (G') and loss moduli (*G*") with the frequency from 0.01 Hz to 10 Hz. The geometric gap and axial force were fixed at 1 mm and 0.1 N, respectively, during the rheological measurements. The storage modulus (*G*'), loss modulus (*G*"), energy dissipation value, tan delta (*tan* δ), and frequency-dependent complex modulus (*G**) were calculated using the following equations [\[84\]](#page-14-4):

$$
\varepsilon = \varepsilon_0 \sin(\omega t), \sigma = \sigma_0 \sin(\omega t + \delta)
$$
 (3)

$$
G' = \sigma_0 / \varepsilon_0 \left(\cos \delta \right) \tag{4}
$$

$$
G'' = \sigma_0 / \varepsilon_0 \left(\sin \delta \right) \tag{5}
$$

$$
\text{Tan } \delta = G''/G' \tag{6}
$$

$$
Dynamic modulus (G^*) = (G'^2 + G''^2)^{1/2}
$$
\n(7)

where σ is the stress (Pa), ε is the strain (mm/mm), ω is the frequency of strain oscillation (rad[\degree]/s), *t* is the time (s), and δ is the phase lag between stress and strain (rad[\degree]/s)

2.5. Electrical Characterization under Strain of the Stretchable Surface Electrode Arrays

Stretching tests were performed to characterize the electrical performance of the stretchable surface electrodes under strain using the following protocol: A piece of a polyethylene terephthalate film was attached to the auto-stretching stage (Motorizer Xtranslation Stages, Jaeil Optical System) to hold both edges of the stretchable device patch. The sample was loaded onto the stretching stage and robustly fixed using tape (3M Co., Ltd.) and silicon epoxy (Sil-poxy, Smooth-On Co., Ltd.). A droplet of liquid metal (gallium indium eutectic, Sigma Aldrich) was applied to the electrode channel and contact pad, followed by wiring cables up to a measurement instrument. A source meter (Keithley 2450 SourceMeter, TEKTRONIX, Inc.) was used to measure the electrical performance of the stretchable electrodes. The resistance of the stretchable electrode was measured according to the stretching range using LabVIEW customized software (National Instrument Corp.).

2.6. Electrochemical Impedance of the Surface Electrode Arrays Using the STICH

Electrochemical impedance spectroscopy (EIS) was performed using a potentiostat (SP1, ZIVE Lab Co., Ltd.) to characterize the electrochemical impedance properties of the stretchable surface electrodes. A three-electrode configuration was used to measure the device impedance. A saturated calomel electrode (BASiAg/AgCl/3 M NaCl) and Pt plate (RDE0021, AT Frontier Co., Ltd.) were used as the reference and counter electrodes, respectively. Using a root-mean-square voltage of 10 mV as the input source of the potentiostatic EIS measurements, impedance plots of the electrodes were acquired over a frequency range of 100 kHz–10 Hz using phosphate-buffered saline (PBS; Samshun Pure Chemical Co., Ltd.). To evaluate the possible improvement in electrochemical impedance induced by the STICH, impedance plots were acquired for the same electrode before and after the application of the STICH coating.

2.7. Acute in Vivo ECoG Recording and Visual Evoked Potential (VEP) Activation Tests

Animal experiments for in vivo ECoG monitoring and VEP activation were conducted on anesthetized rodents (Sprague–Dawley rats, 8 weeks old). For the acute tests, the animals were anesthetized via the intramuscular injection of a ketamine/xylazine mixture, and the head of the animal was fixed on a stereotaxic instrument (Model 900, David Kopf Instrument, Inc.). The scalp of the anesthetized animal was incised, and the skull was removed using an electric drill (Strong 207s, Saeshin, Inc.) to secure the cerebral cortex in the left hemisphere. The stretchable surface electrode arrays were connected to a neural recording instrument (Digital Lynx SX, Neuralynx, Inc.) using an assembled, customized adaptor. The device was mounted on the cerebral cortex, and baseline ECoG signal recordings were performed for 5 min. After the baseline signal recordings, the anesthetized animals were subjected to dark adaptation for approximately 5 min, followed by a VEP activation test for 5 min. To achieve VEP activation in the left hemisphere, light stimulation was conducted at a frequency of 0.2 Hz with respect to the right eye of the animal using a closely attached green LED while its left eye was covered. To evaluate the contribution of the conductive hydrogel to the signal acquisition performance, the baseline signal recording and VEP activation tests were sequentially performed on the same electrode device before and after the hydrogel coating. The documented ECoG data were processed and profiled using MATLAB.

3. Results and Discussion

3.1. Physicochemical Properties of the STICH

Cross-sectional SEM images and photographs (Figure [2a](#page-5-0), top, and bottom, respectively) confirmed the morphological stability of the STICH and other control materials of the Alg, CMC, and PAAm-based conductive hydrogels fabricated according to the conditions in Table [1.](#page-3-0) Overall, the composite hydrogels showed a nonporous structure (except PAAm) owing to the intense interactions between the backbones (particularly between PEDOT:PSS owing to the intense interactions between the backbones (particularly between PEDOT:PSS
and the polyelectrolytes). Additionally, as the Alg content (wt.%) increased, the strong ionic interactions between the PEDOT:PSS and carboxylate increased with the transparency of the STICH; however, a large amount of PEDOT:PSS precipitates was present in the gels (Figure 2a, third box). Nevertheless, the pristine CMC and the [Al](#page-5-0)g/CMC mixture exhibited reduced aggregation, suggesting that the CMC presumably interrupted the ionic interactions between Alg and PAAm.

Figure 2. Physicochemical characterization of the Alg, CMC, and PAAm−based conductive hydro-**Figure 2.** Physicochemical characterization of the Alg, CMC, and PAAm-based conductive hydrogels. gels. (**a**) Cross−sectional SEM images (top, scale bar = 20 µm) and photographs (bottom, scale bar = (**a**) Cross-sectional SEM images (top, scale bar = 20 µm) and photographs (bottom, scale bar = 1 cm) of 1 cm) of each Alg, CMC, and PAAm−based conductive hydrogel. (**b**) FT−IR spectra of the Alg, CMC, each Alg, CMC, and PAAm-based conductive hydrogel. (**b**) FT-IR spectra of the Alg, CMC, and PAAmbased conductive hydrogels. (**c**) Mechanism of the ionic interactions between the PEDOT:PSS/PAAm, D_{DMA} $\left(\frac{1}{2} \right)$ $\left(\frac{1}{2} \right)$ PAAm/polyanions (Alg or CMC), and PEDOT:PSS/Alg. ("PAAm", black; "Alg/PAAm", green; "CMC/PAAm", red; and "Alg/CMC/PAAm", the STICH, blue).

The FT-IR peaks provided clear evidence for the aforementioned hypothesis (Figure [2b](#page-5-0)). The broad peaks from 3000 to 3600 cm⁻¹ correspond to the stretching vibrations of the N–H, O–H, ether, and carboxylic functional groups [85,86]. The three peaks at approximately 1419, 1352, and 1049 cm⁻¹ were assigned to the stretching vibration of the S=O bond in the PSS chain [\[87,](#page-14-7)88]. The other three peaks at 1201, 1141, and 1083 cm⁻¹ were assigned to the stretching vibrations of C–O. The peaks at approximately 1750, 1640, and 1292 cm⁻¹ were assigned to the "stretching vibration of C=O" and the "bending vibrations of N–H, (C=O)NH₂

and C–N". Finally, the three small peaks at 2962, 2924, and 2850 cm−¹ were assigned to the stretching vibrations of $C-H_2$, $C-H_3$, and $C=O$, respectively. Compared to the FT-IR spectra of the "Alg/PAAm" and "CMC/PAAm", that of the "Alg/CMC/PAAm/PEDOT:PSS" (STICH) showed three peaks (C–H₂, C–H₃, and C=O), which could be attributed to the disruption of the ionic interactions between the amide and carboxylate. In summary, the STICH specimen that had a compact entangled network (e.g., the complex between Alg, CMC, and PSS) and various ionic interactions between polyanions (Alg, CMC, PSS) and polycations (PEDOT, PAAm) induced both ionic interactions and dimensional interruptions in the system, which ruptured the binding and increased the durability (Figure [2c](#page-5-0)).

3.2. Mechanical Properties of the STICH

The results of the mechanical property characterization also supported our hypothesis based on the aforementioned physicochemical characterization (Figure [3a](#page-6-0)–d). The UTM measurements of the Alg, CMC, and PAAm-based conductive hydrogels were performed using our previously reported protocol to evaluate the maximum tensile strain per stress of each Alg, CMC, and PAAm-based conductive hydrogel with varying toughness values and Young's moduli [\[89\]](#page-14-9). Notably, the single-network system of conductive hydrogels (such as "PAAm") could not endure a strain of more than 100%. However, the additional polyanion backbones strengthened the conductive hydrogel system, with the triple-network (such as "STICH") enduring a strain of more than 250% (Figure [3a](#page-6-0)) with high toughness (36.85 kJ/m³; Figure [3b](#page-6-0)) but a low Young's modulus (9.46 kPa; Figure [3c](#page-6-0)). On the other hand, in contrast to the single-network system of a conductive hydrogel, the elongation ratio in the triple-network system containing different kinds of polyanion volumes was significantly increased by three-fold while Young's modulus was decreased because of the CMC-originated energy relaxation (based on the presumable interruption of ionic interactions between Alg and PAAm). The cyclic stretching tests indicated that the STICH exhibited a lower Young's modulus without residual strain than the dual-network gel system (such as "CMC/PAAm") (Figure [3d](#page-6-0)). Normally, brain-targeted bioelectronics requires a low Young's modulus (~100 kPa) for modulus matching with brain tissue $(\sim)1$ kPa) but high toughness to support the upper elastic substrate of the bioelectronic (such as PDMS). Therefore, the STICH is a particularly appropriate hydrogel platform for bioelectronics compared to the other investigated hydrogel systems (Figure [3e](#page-6-0)).

Figure 3. Tensile strength measurements of the Alg, CMC, and PAAm−based conductive hydrogels. **Figure 3.** Tensile strength measurements of the Alg, CMC, and PAAm-based conductive hydrogels. (**a**) The graph shows the tensile stress−strain curves (inset image scale bar = 2 cm) of the Alg, CMC, (**a**) The graph shows the tensile stress-strain curves (inset image scale bar = 2 cm) of the Alg, CMC, and PAAm−based conductive hydrogels (*n* = 3). (**b**) Toughness of the Alg, CMC, and PAAm−based and PAAm-based conductive hydrogels (*n* = 3). (**b**) Toughness of the Alg, CMC, and PAAm-based conductive hydrogels (**c**) Young's moduli of the Alg, CMC, and PAAm-based conductive hydrogels. (d) Cyclic stretching data (10 times) of "CMC/PAAm" and STICH. (e) Young's moduli on the log scale of the Alg, CMC, and PAAm-based conductive hydrogels, brain, and PDMS. ("PAAm", black; *3.3. Rheological Properties of the STICH* "Alg/PAAm", green; "CMC/PAAm", red; and "Alg/CMC/PAAm", the STICH, blue).

3.3. Rheological Properties of the STICH

Normally, rheological behaviors that could be determined by analyzing the oscillatory behaviors of viscoelastic materials in parallel-plate geometry were typically evaluated by varying the frequency and monitoring through the absolute values of the storage modulus (*G'*), loss modulus (*G*"), and absolute rate distinctions between *G'* and *G*" (nominated as *tan* δ). The dynamic mechanical properties of the Alg, CMC, and PAAm-based conductive hydrogels were easily calculated using Equations (3)–(6). The schematic in Figure [4a](#page-7-0) illustrates the principle of the rheological experiments; the *G'* and *G*" data are shown in Figure [4b](#page-7-0) and 4c, respectively. The oscillation frequency sweep results of *G'*, *G*", *tan* δ, and *G** ultimately supported our hypothesis based on the aforementioned Alg, CMC, and PAAm-based conductive hydrogel characterization (Figure [4a](#page-7-0)–f). The tendencies of the oscillation results were mostly the same as the mechanical test results, which exhibited higher moduli than those of the single (such as "PAAm"), dual, (such as "CMC/PAAm" and "Alg/PAAm"), and triple-networks (STICH) (Figure [4b](#page-7-0),c). Especially, the frequency dependence result of *G** showed that the storage modulus dominates all of the viscoelastic responses from the whole frequency range (Figure [4d](#page-7-0)). Furthermore, the "PAAm" (singlenetwork), "CMC/PAAm" and "Alg/PAAm" (dual-networks), and STICH (triple-network) comparison graph of the initial dynamic moduli clearly demonstrated the correlation between the frequency dependence and tensile deformation rate, and the complex network system increased the whole elasticity criterion. In contrast, the *tan* δ results showed that STICH exhibited similar results to the single-network hydrogel, evidently dissipating the dynamic stress effectively (Figure [4f](#page-7-0)). In summary, our STICH is the most optimum Alg, CMC, and PAAm-based conductive hydrogel to facilitate soft bioelectronic applications, the softest hydrogel platform compared with single and dual-networks.

Figure 4. Rheological characterization of the Alg, CMC, and PAAm−based conductive hydrogels. **Figure 4.** Rheological characterization of the Alg, CMC, and PAAm-based conductive hydrogels. (**a**) The experimental setting illustration of the rheological property measurements of the Alg, CMC, (**a**) The experimental setting illustration of the rheological property measurements of the Alg, CMC, and PAAm-based conductive hydrogels ($n = 3$). (b) Storage modulus (G') of each Alg, CMC, and PAAm−based conductive hydrogel. (**c**) Loss modulus (*G*") of each Alg, CMC, and PAAm−based PAAm-based conductive hydrogel. (**c**) Loss modulus (*G*") of each Alg, CMC, and PAAm-based conductive hydrogel. (**d**) Dynamic modulus (*G**) of each Alg, CMC, and PAAm−based conductive conductive hydrogel. (**d**) Dynamic modulus (*G**) of each Alg, CMC, and PAAm-based conductive hydrogel. (**e**) Initial dynamic modulus on a log scale for "PAAm" (single−network), "CMC/PAAm" hydrogel. (**e**) Initial dynamic modulus on a log scale for "PAAm" (single-network), "CMC/PAAm" and "Alg/PAAm" (dual−network), and STICH (triple−network). (**f**) *Tan* δ of each Alg, CMC, and and "Alg/PAAm" (dual-network), and STICH (triple-network). (**f**) *Tan* δ of each Alg, CMC, and PAAm−based conductive hydrogel. ("PAAm", black; "Alg/PAAm", green; "CMC/PAAm", red; and PAAm-based conductive hydrogel. ("PAAm", black; "Alg/PAAm", green; "CMC/PAAm", red; and "Alg/CMC/PAAm", the STICH, blue). "Alg/CMC/PAAm", the STICH, blue).

3.4. Electrical Performance of the Stretchable Electrode and the Electrochemical Impedance of the Device Using the STICH

The multi-channel (12-channel for sensing, a reference, and a ground channel) surface electrode array devices were finely manufactured via a microfabrication process (Figure [5a](#page-8-0)). The resistance of the stretchable Au electrode with a 200 μ m diameter was approximately 80 $Ω$. The wavy, serpentine pattern of the lateral interconnects provided stretchability for the thin-film devices without any degradation of electrical characteristics up to 50% maximum strain, which was sufficient to achieve adaptive integration with the curved cerebral cortex (Figure [5b](#page-8-0)). This structural design exhibited mechanical durability in terms of maintaining a constant electrical performance during cyclic stretching at 30% strain (Figure [5c](#page-8-0)). The mechanical deformability and durability of our electrode design are enough to be used in brain interfacing devices. The electrochemical impedance of an Au electrode with a 200 µm diameter) was marked at approximately 60 kΩ at 1 kHz. The impedance plot of the Au electrode increased linearly up to approximately 5 M Ω at 10 Hz (Figure [5d](#page-8-0), red line). In comparison, the impedance corresponding to the frequency band in the same electrode channel with the STICH coating considerably decreased, particularly in the low-frequency range (Figure [5d](#page-8-0), blue line). The impedance of the STICH-coated electrode at 1 kHz was reduced to approximately 10 k Ω , and the maximum value at 10 Hz was also decreased to 860 kΩ. This reduction in impedance across the frequency band means that the electrophysiology recording capabilities of the electrode devices were further improved, so it could be expected that a higher sensitive recording of brain activity would be achieved during ECoG usage. These results demonstrated the STICHinduced improvements in the electrochemical characteristics, particularly with respect to electrophysiological performance.

Figure 5. Electrical characteristics of the stretchable surface electrode arrays and the in vitro EIS **Figure 5.** Electrical characteristics of the stretchable surface electrode arrays and the in vitro EIS analysis of the STICH-coated devices. (a) Exploded-view schematic of the stretchable analysis of the STICH-coated devices. (**a**) Exploded-view schematic of the stretchable surface electrode

arrays. (**b**) Electrical resistance of the stretchable surface electrode while stretching up to 50% strain. (**c**) Normalized electrical resistance during 100 stretching cycles up to 30% strain (inset: a representative plot of 10 cycles). (**d**) EIS plots of the Au electrode (red line) and the STICH-coated electrode (blue) with respect to frequency (from 100 kHz to 10 Hz).

3.5. In Vivo Experiment Results of Acute ECoG Monitoring

To demonstrate the feasibility of the STICH for brain-interfacing applications, in vivo experiments for ECoG monitoring were conducted using the stretchable electrode arrays coated with the STICH brain-interfacing layer. In the acute experiments, an electrode array device with the STICH layer was attached to the left hemisphere of an anesthetized rat, and the baseline ECoG signals and event-related potentials (ERPs) from the visual cortex (VEP) activated by light stimulation of the right eye were recorded (Figure [6a](#page-9-0)). The direct contact between the STICH layers with the cortex permitted the construction of adaptive brain interfacing. The brain-like softness of the STICH provided excellent conformality to electrodes, and the sufficient toughness allowed the STICH layer to maintain its solid form between the device and the tissue. The raw signal plot of the baseline ECoG activity from the surface electrodes mounted on the animal's cortex (Figure [6b](#page-9-0)) and activated VEP emphasized the selectivity of the 12-channel STICH-based electrode arrays (Figure [6c](#page-9-0)). Moreover, the STICH-coated electrodes featured the capability of clearly recording visual cortex-originated ERPs (Figure [6d](#page-9-0)). The contribution of the STICH was intuitively demon-*Polymers* **2023**, *15*, x FOR PEER REVIEW 11 of 15 strated by comparing the average amplitude of multiple activated VEPs recorded from the electrode using the STICH with that from the control device without the STICH coating (Figure [6e](#page-9-0)). The average amplitude of multiple VEPs (a total of 140 accumulated trials) from the STICH electrodes was significantly larger (\sim 133 μ V) than that from the control device $({\sim}109~\mu$ V). As expected, the improvement of electrochemical impedance performance by the use of the STICH coating resulted in a higher sensitive recording of ECoG activity in vivo. These results demonstrate the STICH-induced improvements in the neural recording performance of the electrodes in terms of both electrical and ionic conductivities. (rigule ve). The average amplitude of mumple vells (a total of pedance by the use of the STICH coating resulted in a higher sensitive icconductivity per

Figure 6. Experimental results of the agute in vivo ECoC monitoring a Figure 6. Experimental results of the acute in vivo ECoG monitoring and VEP activation tests conducted using an anesthetized rodent model. (**a**) Schematic of ECoG monitoring during VEP activation

on the left hemisphere of a rat by stimulating its right eye using a green LED (inset: a photograph of the STICH-coated stretchable surface electrode arrays mounted on the left hemispheric cortex, the scale bar indicates 3 mm). (**b**) Two-dimensional deployment map of the multi-channel ECoG electrodes mounted on the rodent's cortex. (**c**) Representative raw plot of the baseline ECoG and VEP signals recorded using the multi-channel surface electrode arrays. (**d**) An activated VEP signal recorded from a representative channel. (**e**) Average VEP plot derived from 140 trials of light stimulation tests. The difference in amplitude of the average VEPs before (left) and after (right) the STICH coating demonstrates the significant contribution of the STICH in improving the neural recording performance of the electrodes.

4. Conclusions

In this study, a new functional composite hydrogel, the STICH, was developed to construct an adaptive conformal brain interface. The triple-network-based STICH exhibited both high toughness and a low Young's modulus, which was ideal in terms of applicability to soft brain tissue. This mechanical improvement originated from the dissipated hydrogen bonding and entanglement between Alg, CMC, and PAAm. Furthermore, the STICH, with both electrical and ionic conductivities, bestowed improved electrochemical impedance properties to metal electrodes upon coating. Consequently, adaptive conformal brain interfacing and capable ECoG monitoring were successfully achieved in a rodent model using the stretchable surface electrode arrays with the STICH. Based on the improvement in electrochemical impedance, the electrode device with the STICH layer exhibited a more sensitive electrophysiology recording capability with a higher amplitude of visual ERPs than that of the control device. The implementation of these functionalities, which could be realized for translational applications, could permit the development of advanced bioelectronic materials and devices that are more practical and applicable in clinical settings.

Author Contributions: S.L., M.S. and D.S. conceptualized this study. K.P. characterized and optimized the hydrogel materials. S.L. designed and fabricated the electrode device. S.L. performed the electrical characterization and in vitro electrochemical impedance analysis of the devices. S.L., K.P. and J.K. performed the quantitative and qualitative experiments. S.L. and K.P. performed the validation and formal analysis. S.L. and S.A. drew the illustration. M.S. and D.S. supervised the work. S.L. and K.P. wrote the manuscript. The authors read and approved the final manuscript. S.A. conducted acute animal surgery. S.L., J.K. and S.A. demonstrated the application. S.L. and J.K. conducted the neural signal analysis and visualization. K.J.Y. and H.K. provided technical advice for animal experiments and data analysis. M.S. and D.S. were responsible for supervision. All authors have read and agreed to the published version of the manuscript.

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