Non-collinear magnetism in the post-perovskite thiocyanate frameworks CsM(NCS)₃

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Contents

Li[st of Figures](#page-3-0)

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Li[st of Tables](#page-10-1)

1 Experimental

1.1 Synthesis of CsM(NCS)³

Aqueous solutions of MnSO₄ · H₂O (254 mg, 1.5 mmol, 3 mL) and Ba(SCN)₂ · 3H₂O (461 mg, 1.5 mmol, 15 mL) were combined and stirred for 24 h. The opaque white solution was filtered, and the white residue of BaSO $_4$ removed to leave a filtrate of aqueous Mn(NCS) $_2$. Cs $_2$ SO $_4$ (272 mg, 0.75 mmol) and Ba ${\rm (SCN)}_2 \cdot$ 3 H $_2$ O (231 mg, 0.75 mmol) were dissolved in distilled water (10 mL) and added to the Mn(NCS) $_{\rm 2}$ solution. The mixture was stirred for 24 h, and filtered, removing a white residue of BaSO $_4$. The colourless filtrate solution of CsMn(NCS) $_3$ was concentrated under reduced pressure and left to stand for 48 h yielding pale green single crystals of $\mathsf{CSMn}(\mathsf{NCS})_3.$

 CsCo(NCS)_3 and CsNi(NCS)_3 were synthesised in an analogous method from $\text{CoSO}_4 \cdot 7\,\text{H}_2\text{O}$ (421 mg, 1.5 mmol) and NiSO₄ \cdot 6 H₂O (394 mg, 1.5 mmol) respectively. This yielded purple, CsCo(NCS)₃, and green, $\mathsf{CsNi(NCS)}_3$, microcrystalline powders. $\,$ Single crystals of these compounds were obtained by slow evaporation recrystallisations from concentrated aqueous solutions. The synthesis of CsNi(NCS) $_3$ has been previously reported through the reaction of NiCl $_2$ \cdot 6 H $_2$ O, Cs $_2$ CO $_3$ and $NH₄$ SCN.³

1.2 Single Crystal X-ray Diffraction

Single cr[ys](#page-14-0)tals were selected and mounted using Fomblin® (YR-1800 perfluoropolyether oil) on a polymer-tipped MiTeGen MicroMountTM and cooled rapidly to 120 K in a stream of cold N_2 using an Oxford Cryosystems open flow cryostat.⁴ Single crystal X-ray diffraction data were collected on an Oxford Diffraction GV1000 (AtlasS2 CCD area detector, mirror-monochromated Cu-K*α* radiation source; $\lambda = 1.541$ Å; ω scans). Cell parameters were refined from the observed positions of all strong reflections and absorption correction[s](#page-14-1) were applied using a Gaussian numerical method with beam profile correction (CrysAlisPro).⁵ The structure was solved within Olex2⁶ by dual space iterative methods (SHELXT),⁷ least squares refinement of the structural model was carried using (SHELXL). 8 Structures were checked with checkCIF (https://checkcif.iucr.org).

1.3 Single Cryst[al](#page-14-2) Neutron Diffraction

Monochromatic single crystal neutron diffraction data for CsMn(NCS)₃ and CsNi(NCS)₃ were collected on the four-circle D19 diffractometer at the Institut Laue Langevin (ILL) Grenoble, France, using neutrons with a wavelength of 1.455 Å (CsMn(NCS) $_3)$ and 1.457 Å (CsNi(NCS) $_3)$ provided by a flat Cu monochromator using the 220 reflection at $2\theta_M = 69.91^\circ$ take-off angle. The samples were each placed in a closed-circuit displex cooling device, which was operated following a ramp of 2 K min*−*¹ .

NOMAD software from the ILL was used for data collection. Unit cell determinations were performed using PFIND and DIRAX programs, and processing of the raw data was applied using RETREAT, <code>RAFD19</code> and Int3D programs. $^{\rm 9-12}$ For CsNi(NCS) $_3$, the data were corrected for the absorption of the low-temperature device using the D19ABS program¹³ and for the size and composition of the crystal. For CsMn(NCS) $_3$ the data were corrected for the absorption of the low-temperature device using the D19ABS program. Structural [m](#page-14-3)[od](#page-14-4)els were solved using the *SUPERFLIP* program and refined using Jana2006.14,15

1.4 Powder Neutron Diffraction

Constant [wa](#page-14-5)[ve](#page-14-6)length powder neutron diffraction data for $\text{CSMn}(NCS)_{3}$ and $\text{CSNi}(NCS)_{3}$ were collected on the high-intensity medium resolution D1b diffractometer¹⁶ at ILL, France. The incident wavelength was *λ* = 2*.*52 Å and the scattering was measured over an angular range of 2 *<* 2*θ <* 128°. Thermal diffractograms for CsMn(NCS)₃ were collected between 1.5 K and 20 K heated with a programmed ramp of 0.06 K min^{−1}. Long acquisition measurements [wer](#page-14-7)e collected at 1.5 K and 20 K.

Thermal diffractograms for CsNi(NCS)₃ were collected between 1.5 K and 10 K heated with a programmed ramp of 0.025 K min*−*¹ . Long acquisition measurements were collected at 1.5 K and 10 K. NOMAD software from the ILL was used for data collection. Refinements of the magnetic model were completed using the FullProf program.¹⁷

1.5 Magnetic Measurements

Measurements of the magnetic susceptibilit[y w](#page-14-8)ere carried out on samples of $\mathsf{CsNi(NCS)}_3$ (34.5 mg) and CsCo(NCS)₃ (14.3 mg) using a Quantum Design Magnetic Property Measurements System (MPMS) 3 Superconducting Quantum Interference Device (SQUID) magnetometer, and for CsMn(NCS) $_3$ (9.6 mg) using an MPMS XL. The zero-field-cooled (ZFC) and field-cooled (FC) susceptibility was measured in an applied field of 0.01 T over the temperature range 2–300 K. As *M*(H) is linear in this field, the small-field approximation for the susceptibility, $\chi(\mathcal{T}) \simeq \frac{M}{H}$ $\frac{M}{H}$, where \bm{M} is the magnetisation and *H* is the magnetic field intensity, was taken to be valid.

Isothermal magnetisation measurements were carried out at 2 K over the field range *−*7 to +7 T for CsMn(NCS)₃ and CsCo(NCS)₃, and a range of -6 to $+6$ T for CsNi(NCS)₃. Data were corrected for diamagnetism of the sample using Pascal's constants.¹⁸

1.6 Density Functional Theory Calculations

We have performed density functional theory (DFT) [calc](#page-14-9)ulations to probe the structures and energetics of spin order of the compounds considered in this study. The spin-polarised DFT+*U* method (with Grimme's D3 van der Waals correction)¹⁹ was employed in structural optimisations and energy calculations, using the Vienna Ab initio Simulation Package (VASP 5.4.4).²⁰ In our DFT+U calculations, we used U values of 3.6, 5.0 and 5.1 eV for d-electrons of Mn^{2+} , Co^{2+} and Ni^{2+} cations, $21,22$ respectively, and a range of ferromagnetic a[nd](#page-14-10) anti-ferromagnetic spin solutions were considered for divalent magnetic cations (Table 1). We used a plane-wave basis set with [a](#page-14-11) kinetic energy cutoff of 520 eV to expand the [w](#page-14-12)ave functions. The Perdew-Burke-Ernzerhof functional²³ in combination w[ith](#page-14-13) the projector augmented wave method $24,25$ were used to solve the Kohn-Sham equations. An energy convergence threshold of 10*−*⁴ eV was used for all total energy calculations, and the structural optimisations, including cell parameter[s a](#page-3-2)nd atomic positions, were considered conv[erg](#page-14-14)ed if all interatomic forces fall below 0.01 eV Å*−*¹ . All DFT [ca](#page-14-15)[lcu](#page-14-16)lations have been performed in the 2 *×* 1 *×* 1 supercell (8 formula units per cell) using a 6 *×* 5 *×* 5 k-grid (which corresponds to a k-points spacing of around 0.1 Å^{−1}).

Table 1: The configurations used to calculate the ground state interactions and the relative energy of each configuration for $\mathsf{CSM}(NCS)_3$, M = Ni, Mn. Each interaction is given as a relative interaction where + is ferromagnetic, *−* is antiferromagnetic, and 0 where the interactions are compensated.

Figure 1: Variable temperature magnetic susceptibility product measured over the range 2–300 K (zero-field cooled). The high temperature spin only Curie value, *C*, is indicated with a dashed line for a) CsNi(NCS) $_3$ ($C=1.0$ emu K mol^{−1}); b) CsMn(NCS) $_3$ ($C=4.375$ emu K mol^{−1}); c) CsCo(NCS)³ (*C* = 1*.*875 emu K mol*−*¹).

Figure 2: Inverse of magnetic susceptibility (zero-field cooled) between 1.5 and 300 K. The high temperature data was modelled using the Curie-Weiss law (180 *< T <* 300 K for Ni; $100 < T < 300$ K for Co and Mn), black line.

Figure 3: Inverse of magnetic susceptibility for CsNi(NCS) $_3$. The high temperature data was modelled using the Curie-Weiss law for a) $100 < T < 300$ K, $\theta_{\text{CW}} = +1.5(4)$ and b) $200 < T < 300$ K, $\theta_{\text{CW}} = -12.7(8)$ K, black line.

Figure 4: An expansion of the isothermal magnetisation data around the hysteresis for a) CsNi(NCS)₃ and b) $\mathsf{CsCo}(\mathsf{NCS})_3.$

Figure 5: Rietveld fits (black line) for CsNi $(NCS)_3$ data (green circles) collected with the D1b diffractometer at a) 10 K and b) 1.5 K. The black markers indicate the position of nuclear Bragg reflections and green markers for magnetic Bragg reflections. Data at 0.85 Å have been omitted due to the presence of an impurity.

Figure 6: CsNi(NCS) $_3$ lattice parameters between 1.5 and 10 K, obtained from Rietveld refinements of neutron diffraction data measured using the D1b diffractometer (ILL). The verical dashed black line indicates the magnetic ordering temperature ($T_{\text{C}} = 8.5$ K). A linear fit (dark green line) was used to calculated the coefficient of thermal expansion. As the change in sin(*β*) is at least one of magnitude smaller than the change in the *a*, *b* and *c* axes, the thermal expansion tensor will not significantly rotate over this temperature range. a) a axis, $\alpha_a = 19(3)$ MK^{−1}. b) *b* axis, $\alpha_b = -13(3) \text{ MK}^{-1}$. c) *c* axis, $\alpha_c = 5(2) \text{ MK}^{-1}$. d) sin(β) , $\alpha_{sin(\beta)} = 0.3(1) \text{ MK}^{-1}$. e) Volume, $\alpha_V=11(5)$ MK⁻¹.

Figure 7: Other AM(NCS) $_3$ structures with perovskite-analogue structure types: a) CsCd(NCS) $_3;^1$ b) [NH₄]₂NiCd(NCS)₆.² Caesium = pink, cadmium = purple, nickel = grey, nitrogen = blue, $carbon = black$, sulfur = yellow, hydrogen = pale pink.

Figure 8: Orientation of the ordered magnetic moments for a) CsNi(NCS) $_3$ viewed down the magnetic b axis, b) CsNi(NCS) $_3$ viewed down the magnetic a axis and c) CsMn(NCS) $_3$ viewed down the magnetic *a* axis.

Figure 9: Angles between the *ac* plane (black line) and M*−*N and M*−*S bond angles in the direction of a) corner-sharing octahedra and b) edge-sharing octahedra in ${\tt CsM(NCS)}_3.$ These angels are the same (to the nearest integer) for both M1 and M2. Metal = pink, nitrogen = blue, carbon = black, sulfur = yellow.

Figure 10: Single crystals used in the neutron diffraction experiments for a) $CsNi(NCS)₃$ and b) $\mathsf{CsMn}(NCS)_{3}.$

Table 2: Single crystal experimental details for CsCo(NCS) $_3$ (X-ray data) and CsNi(NCS) $_3$ (neutron data)

* Absorption correction method: gaussian, *CrysAlis PRO* 1.171.40.53 (Rigaku Oxford Diffraction, 2019). Numerical absorption correction based on gaussian integration over a multifaceted crystal model Empirical absorption correction using spherical harmonics, implemented in SCALE3 ABSPACK scaling algorithm.

** Absorption correction was applied using D19ABS program for the low-temperature device¹³ and the size and composition of the crystal.

*Absorption correction was applied using D19ABS program for the low-temperature device. $^{\rm 13}$

Table 4: Single crystal experimental details for CsMn(NCS)₃, 50 $-$ 300 K (50 and 300 K neutron data, 120 K X-ray data)

*Absorption correction method: gaussian, *CrysAlis PRO* 1.171.40.53 (Rigaku Oxford Diffraction, 2019). Numerical absorption correction based on gaussian integration over a multifaceted crystal model Empirical absorption correction using spherical harmonics, implemented in SCALE3 ABSPACK scaling algorithm.

**Absorption correction was applied using D19ABS program for the low-temperature device.¹³

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