Supplementary Information

Mesoporous multimetallic nanospheres with exposed highly entropic alloy

sites

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CONTENTS

Supplementary Figures

Supplementary Figure 1. The photographs of micellization process through the Tyndall effect and a chemical reduction process of PtPdRhRuCu MMNs.

Supplementary Figure 2. FTIR spectra of bare PEO₁₀₅₀₀-*b*-PMMA₁₈₀₀₀, PtPdRhRuCu MMNs before and after solvent extraction.

Supplementary Figure 3. SEM images and corresponding particle/pore size distribution of PtPdRhRuCu MMNs.

Supplementary Figure 4. The SAXS pattern of PtPdRhRuCu MMNs.

Supplementary Figure 5. HAADF-STEM image of PtPdRhRuCu MMNs showing possible atomic steps/kinks.

Supplementary Figure 6. SEM image and corresponding SEM-EDS maps of PtPdRhRuCu MMNs.

Supplementary Figure 7. Low-magnification HAADF-STEM image and corresponding map of PtPdRhRuCu MMNs.

Supplementary Figure 8. XPS spectra of PtPdRhRuCu MMNs.

Supplementary Figure 9. The time-depending SEM images for preparing PtPdRhRuCu MMNs.

Supplementary Figure 10. The optimized chemical reduction conditions.

Supplementary Figure 11. SEM images and corresponding SEM-EDS results of PtPdRhRuCu MMNs prepared at different temperature.

Supplementary Figure 12. SEM images of PtPdRhRuCu MMNs prepared using different amount of HCl.

Supplementary Figure 13. SEM images and XRD patterns of monometallic sample.

Supplementary Figure 14. SEM images and XRD patterns of mesoporous bimetallic alloy.

Supplementary Figure 15. SEM images and XRD patterns of mesoporous trimetallic alloy.

Supplementary Figure 16. SEM images and XRD patterns of mesoporous tetrametallic alloy.

Supplementary Figure 17. SEM images, N₂ adsorption–desorption isotherms, and EDS atomic maps of PtPdRhRuCu-1, PtPdRhRuCu, and PtPdRhRuCu-2 MMNs.

Supplementary Figure 18. Structural characterization of multimetallic PtPdCuNiCo alloy.

2

Supplementary Figure 19. Structural characterization of multimetallic PtPdCuNiCoMo alloy.

Supplementary Figure 20. Comparing HER performance of PtPdRhRuCu MMNs with other Ptbased mesoporous alloys.

Supplementary Figure 21. Powder XRD pattern of Pt MNs and PtPdRhRuCu MMNs.

Supplementary Figure 22. Electrochemical performance of PtPdRhRuCu MMNs and nonporous PtPdRhRuCu nanospheres.

Supplementary Figure 23. Electrochemical performance of PtPdRhRuCu MMNs and Pt MNs.

Supplementary Figure 24. DFT results toward HER reaction on PtPdRhRuCu MMNs.

Supplementary Figure 25. The valence electron densities of surface metal in pure metal and HEASs.

Supplementary Figure 26. The effect of pore size effect on HER performance.

Supplementary Figure 27. Structural characterization of PtPdRhRuCu-1, PtPdRhRuCu, and PtPdRhRuCu-2 samples.

Supplementary Figure 28. The characterizations of PtPdRhRuCu MMNs after HER stability test.

Supplementary Figure 29. HAADF-STEM image and corresponding map of PtPdRhRuCu MMNs after the durability HER test.

Supplementary Figure 30. Electrochemical performance in 0.5 M H₂SO₄ and 1.0 M PBS electrolytes. **Supplementary Figure 31**. The experimentally measured and theoretically calculated amounts of H₂ on PtPdRhRuCu MMNs in different electrolytes.

Supplementary Figure 32. Calibration curves in different electrolytes.

Supplementary Tables

Supplementary Table 1. The ICP–OES analysis of as-prepared and after alkaline HER test of PtPdRhRuCu MMNs.

Supplementary Table 2. The standard reduction potential of different half reactions for various metal precursors.

Supplementary Table 3. The feeding amounts of various metal precursors and the final compositions of Pt-based MNs.

Supplementary Table 4. The synthesis conditions for noble-mental-based MNs.

Supplementary Table 5: The calculation of Löwdin charge analysis for each metal of HEAS.Supplementary Table 6. The HER performance of recent reported MMA electrocatalysis.Supplementary Table 7: Summary of recently reported Pt-based HER electrocatalysts.

Supplementary References

Supplementary Figures



Supplementary Figure 1. (a-b) The photographs of PEO₁₀₅₀₀-*b*-PMMA₁₈₀₀₀ dissolved in DMF and following micellization process driven by adding aqueous solution, as seen through the Tyndall effect. (Note that the pure water was used instead of metal precursors because that the reaction mixture appears dark brown, leading to the optical path cannot be observed). (c) The TEM image of the spherical PEO₁₀₅₀₀-*b*-PMMA₁₈₀₀₀ micelles composed of PMMA cores surrounded by PEO shells, obtained by using a negative staining technique with phosphotungstic acid. (d) Photograph of a chemical reduction process for synthesis of PtPdRhRuCu MMNs.



Supplementary Figure 2. Fourier transform infrared spectroscopy spectra of bare PEO₁₀₅₀₀-*b*-PMMA₁₈₀₀₀, PtPdRhRuCu MMNs before and after solvent extraction, indicating that most of the micelles were successfully removed by acetone/ethanol washing. Source data are provided as a Source Data file.



Supplementary Figure 3. (a, b) SEM images of PtPdRhRuCu MMNs. The corresponding (c) particle size and (d) pore size distribution. Source data are provided as a Source Data file.



Supplementary Figure 4. The SAXS pattern of PtPdRhRuCu MMNs. (The position of each pixel on the SAXS image was converted into the scattering angle 2θ or the scattering vector q (its modulus was $q = 4\pi \sin(\theta)/\lambda$, where λ is the wavelength of the X-ray). And q was more common used because the value of θ varies in a small range.) Source data are provided as a Source Data file.



Supplementary Figure 5. (a) HAADF-STEM image of PtPdRhRuCu MMNs. (b) Schematic illustration for the atomic steps and kinks for enhancing the electrocatalytic properties.



Supplementary Figure 6. SEM image and corresponding SEM-EDS maps of PtPdRhRuCu MMNs.



Supplementary Figure 7. Low-magnification HAADF-STEM image and corresponding map of PtPdRhRuCu MMNs. Scale bars, 200 nm.



Supplementary Figure 8. (a) Pt 4*f* XPS spectra of PtPdRhRuCu and Pt MNs. (b) Pd 3*d*, Rh 3*d*, Ru 3*p* and Cu 2*p* XPS spectra of PtPdRhRuCu MMNs. Source data are provided as a Source Data file.



Supplementary Figure 9. The time-depending SEM images for preparing PtPdRhRuCu MMNs: (a) 9, (b) 11, (c) 20, (d) 30, (e) 70, (f) 120, and (g) 240 min. Scale bars: 100 nm (h) The changes of the composition in different deposition time, obtained by SEM-EDS. Source data are provided as a Source Data file.



Supplementary Figure 10. The optimized chemical reduction conditions. SEM images of PtPdRhRuCu prepared (a) without using surfactant, (b) with F127, and (c) with polyvinylpyrrolidone (PVP, MW=40,000 g mol⁻¹) instead of PEO₁₀₅₀₀-*b*-PMMA₁₈₀₀₀. SEM images of PtPdRhRuCu-based samples prepared using different reducing agents: (d) dimethylamine borane and (e) formic acid. SEM images of PtPdRhRuCu samples prepared using (f) THF (3 mL) and (g) H₂SO₄ (3 M, 0.4 mL) instead of DMF (3 mL) and HCl (6 M, 0.4 mL), respectively.



Supplementary Figure 11. (a–d) SEM images (scale bars, 200 nm.) and (e) corresponding SEM-EDS results of PtPdRhRuCu MMNs prepared at different temperature: (a) 65, (b) 75, (c) 80, and (d) 85 °C. Source data are provided as a Source Data file.



Supplementary Figure 12. SEM images of PtPdRhRuCu MMNs prepared using different amount of 6.0 M HCl: (a) 0.2, (b) 0.3, (c) 0.4, and (d) 0.5 mL. Scale bars: 500 nm.



Supplementary Figure 13. SEM images and XRD patterns of monometallic sample: (a, f) Pt MNs, (b, g) Pd MNs, (c, h) Rh MNs, (d, i) Ru NPs, and (e, j) Cu NPs. Scale bars in (a–e): 200 nm. To date, mesoporous nanospheres morphology of metallic Ru and Cu were still limited to their alloys with other precious metals. The monometallic Ru and Cu with mesoporous structure were not yet successfully prepared. Therefore, Ru and Cu NPs were used instead of mesoporous structures. Source data are provided as a Source Data file.



Supplementary Figure 14. SEM images of mesoporous bimetallic alloy: (a) PtPd, (b) PtRh, (c) PtCu, and (d) PtRu MNs. Scale bars: 200 nm. (e) XRD patterns of different samples. Compared to Pt (PDF#04-0802), the XRD peaks of PtPd do not significantly shift probably due to a low lattice mismatch of 0.77%.¹ However, it is interesting to note that we did not find a shift in the XRD peaks of PtRu and PtRh except for PtCu, which might be explained by the fact that Pt dominates in the alloys and the atomic radii of Ru and Rh are only slightly smaller than Pt (atomic radii order: Pt (130 pm) > Pd (128 pm) > Rh/Ru (125 pm) > Cu (117 pm).² Although the XRD peaks of PtRh and PtRu are not shifted, their full width at half maximum (FWHM) become larger, compared to PtPd alloy. According to Sherrer's formula, a larger FWHM represents a decrease in crystallite size³. The possible reason is the increase of defects caused in the introduction of Ru and Rh. Source data are provided as a Source Data file.



Supplementary Figure 15. (a–c) SEM images and (d) XRD patterns of mesoporous trimetallic alloy: (a) PtRuCu, (b) PtRuPd, and (c) PtRuRh MNs. Scale bars: 200 nm. Source data are provided as a Source Data file.



Supplementary Figure 16. (a, b) SEM images and (d) XRD patterns of mesoporous tetrametallic alloy: (a) PtRhRuCu, (b) PtPdRhRu MNs. Scale bars: 200 nm. Source data are provided as a Source Data file.



Supplementary Figure 17. SEM images and N₂ adsorption–desorption isotherms of (a, d) PtPdRhRuCu-1, (b, e) PtPdRhRuCu, and (c, f) PtPdRhRuCu-2 MMNs, which was prepared using block polymer PEO10000-*b*-PMMA5500, PEO10500-*b*-PMMA18000, and (c) PEO10500-*b*-PMMA22000, respectively. EDS atomic maps of (g) PtPdRhRuCu-1 and (h) PtPdRhRuCu-2 MMNs. Scale bars: 100 nm. Insets of (d-f) show the corresponding pore size distributions. Source data are provided as a Source Data file.



Supplementary Figure 18. (a) SEM image (Scale bar, 500 nm), (b) HAADF-STEM image (Scale bar, 100 nm), (c) EDS maps (Scale bar, 50 nm), (d) SAXS pattern, and (e) XRD pattern of multimetallic PtPdCuNiCo alloy. The atomic ratios in PtPdCuNiCo MNNs were determined to be Pt:Pd:Cu:Ni:Co = 25:28:33:7:7, based on EDS profiles. The lower molar ratio of Ni and Co probably due to their lower standard reduction potentials, which indicated that they were difficult to be reduced by *L*-AA at given experimental conditions. Source data are provided as a Source Data file.



Supplementary Figure 19. (a) SEM image (Scale bar, 500 nm), (b) HAADF-STEM image (Scale bar, 100 nm), (c) EDS maps (Scale bar, 50 nm), (d) SAXS pattern, and (e) XRD pattern of multimetallic PtPdCuNiCoMo alloy. The atomic ratios in PtPdCuNiCoMo MNNs were determined to be Pt:Pd:Cu:Ni:Co:Mo = 23:24:28:10:9:6, based on EDS profiles. The lower molar ratio of Ni, Co, and Mo probably due to their lower standard reduction potentials, which indicated that they were difficult to be reduced by *L*-AA at given experimental conditions. Source data are provided as a Source Data file.



Supplementary Figure 20. HER polarization curves of (a) monometallic MNs/NPs, (b) bimetallic, (c) trimetallic, (d) tetrametallic, and (e) comparing PtPdRhRuCu MMNs with other Pt-based mesoporous alloys at scan rate of 5 mV s⁻¹ without *iR* correction in 1.0 M KOH electrolyte. (e) The decline ratio between C_{d1} _CV and C_{d1} _EIS at scan rates of 25, 50, 100, and 200 mV s⁻¹ of different samples. Source data are provided as a Source Data file.



Supplementary Figure 21. Powder XRD pattern of Pt MNs and PtPdRhRuCu MMNs. Source data are provided as a Source Data file.



Supplementary Figure 22. (a) Polarization curves of PtPdRhRuCu MMNs and nonporous PtPdRhRuCu nanospheres. Cu-UPD measurement in 0.5 M H₂SO₄ with (I, II) and without (III) of 5 mM CuSO₄ on (b) PtPdRhRuCu MMNs and (c) nonporous PtPdRhRuCu nanospheres at a scan rate of 10 mV/s. The electrodes for II and III were fixed at certain deposition potential for 100 s to form UPD layer and followed by a CV scan. (f) Histograms showing the ECSA values. Source data are provided as a Source Data file.



Supplementary Figure 23. (a) Tafel plots, (b) Nyquist plots (operated at the overpotential of 20 mV) of PtPdRhRuCu MMNs and Pt MNs. (c) N₂ adsorption–desorption isotherms of Pt MNs (Inset shows the corresponding pore size distribution). (d) BET-normalized LSV curves of PtPdRhRuCu MMNs and Pt MNs. Source data are provided as a Source Data file.



Supplementary Figure 24. (a) The atomic configurations at the four stages toward HER reaction on PtPdRhRuCu MMNs. The calculated free energy (ΔG) for (b) water dissociation and (c) hydrogen adsorption at various catalytic sites of PtPdRhRuCu. DFT calculations were performed to understand the superior HER performance of PtPdRhRuCu MMNs in alkaline media. Based on the experimentally measured Tafel slope value (87 mV dec⁻¹), the HER reaction mechanism of PtPdRhRuCu MMNs in alkaline solution might follow the Volmer-Heyrovsky mechanism (*eqs* (i-ii)).

Volmer step:	$\mathrm{H_2O} + \mathrm{e^-} + * \longrightarrow \mathrm{H^*} + \mathrm{OH^-}$	(i)
Heyrovsky step:	$\mathrm{H}^{*} + \mathrm{H}_{2}\mathrm{O} + \mathrm{e}^{-} \rightarrow \mathrm{H}_{2} + \mathrm{O}\mathrm{H}^{-}$	(ii)

where * represented the catalytic sites on the surface; H* was the adsorbed hydrogen intermediate.

Unlike HER in acidic media, water dissociation (Volmer step) was a key step to provide the free proton for the HER reaction in alkaline solution. Pt presented a superior HER performance in acidic environment but had a much lower reaction rate in alkaline solution.⁴ An ideal alkaline HER electrocatalyst should combine an optimal free energy of hydrogen adsorption (ΔG_{H^*}) and a low water dissociation barrier. **Supplementary Figure 24a** displays the atomic configurations of the four elemental stages during alkaline HER at the catalytic sites of Pt₂₃Pd₂₂Rh₂₀Ru₁₃Cu₂₂ (The atomic ratios of each metal was based on ICP–OES, which was close to the HEASs in PtPdRhRuCu MMNs (**Figure 2**)): (Stage 1) the adsorption of H₂O* molecule; (Stage 2) the destabilization of adsorbed H₂O* molecule to form H–OH bond; (Stage 3) the dissociated H–OH to generate H* and OH* intermediates; (Stage 4) the adsorbed H* will react with proton to produce H₂. As shown in the energy profile of **Supplementary Figure 24b**, in the energy barrier to water dissociation on diverse active sites of Pt₂₃Pd₂₂Rh₂₀Ru₁₃Cu₂₂ Rh site has the lowest barrier of 0.29 eV to break the H–OH bond (*i.e.*, stage 1 \rightarrow stage 2). As a key step of HER reaction in basic environment, the energy cost of 0.29 eV for water dissociation was much lower than the 0.89 eV occurred on the pure Pt (111) surface,⁵ indicating a facilitated water dissociation on the synthesized MMA of Pt₂₃Pd₂₂Rh₂₀Ru₁₃Cu₂₂.

The evolution of hydrogen intermediate on the catalyst for HER was usually descripted by hydrogen adsorption free energies ΔG_{H^*} , and the optimal value of ΔG_{H^*} should be close to zero (*i.e.*, $|\Delta G_{H^*}| = 0$).⁶ In HER, Pt (111) was reported as an efficient HER catalyst due to its near-zero value of $|\Delta G_{H^*}| = 0.09 \text{ eV.}^5$ From the calculated ΔG_{H^*} on different sites of *fcc* Pt₂₃Pd₂₂Rh₂₀Ru₁₃Cu₂₂ (**Supplementary Figure 24c**), Cu site has a much lower $|\Delta G_{H^*}|$ value of 0.05 eV than the reported result on Pt (111). The synergistic effect between the active metallic site of MMA, which has altered the surface reactivity to hydrogen adsorption, was common in metal alloy or hybrid catalysts.^{7, 8, 9, 10} For example, as reported by Li and coworkers,⁸ the nearby Ni, Co sites in Pt₁₈Ni₂₆Fe₁₅Co₁₄Cu₂₇ catalyst were considered as the preferred sites for H* adsorption to stabilize H in the alkaline HER. The localized effect of each metal coupled to its neighboring metals in MMA catalysts lead to a more chemical complexity on the catalytic surface of MMA.^{11, 12} Therefore, the catalytic performance of PtPdRhRuCu MMNs with multi-metal involved is not simply dependent on the each separate metal, but on a cooperative effect from the active metal site and its coordinated metals, which is beneficial to a pH-universal HER catalytic activity and stability. Source data are provided as a Source Data file.



Supplementary Figure 25. The valence electron densities of surface metal in pure metal and HEASs of PtPdRhRuCu MMNs respectively from Löwdin charge analysis, which was study the charge transfer effects. The change of electronic structure might change Fermi level and the *d*-band center position of metals, thus optimized the adsorption for the reaction intermediates to influence the electrocatalytic activity.¹³ Source data are provided as a Source Data file.



Supplementary Figure 26. (a) Polarization curves of HER without *iR* correction of PtPdRhRuCu-1, PtPdRhRuCu, and PtPdRhRuCu-2 MMNs. (b) The ECSA/EWSA ratio (dynamic (CV)/static (EIS) capacitance and (c) the ratio between $C_{dl_{c}CV}$ and $C_{dl_{EIS}}$ at scan rates of 25, 50, 100, and 200 mV s⁻¹ of different samples (The numbers at the top of the histogram in (c) indicate decrease rate (%) of different scan rate). To investigate the effect of the role of each kind of mesopore on performance, we conducted in-depth electrochemical analysis with EIS. The electric double layer capacitance (C_{dl}) was calculated from EIS and cyclic voltammetry (CV) to obtain $C_{dl_{eIS}}$ and $C_{dl_{cV}}$, which were proportional to EWSA and ECSA, respectively. The high $C_{dl_{cV}}/C_{dl_{eIS}}$ values and low decrease rate at increased scan rates indicate the efficient ion migration. Source data are provided as a Source Data file.



Supplementary Figure 27. (a, d, g) HAADF-STEM images, (b, e, h) enlarged image of one pore area, and (c, f, i) the corresponding ΔS_{mix} value at the selected area (i.e., 1, 2, 3, and 4) of (a–c) PtPdRhRuCu-1, (d–f) PtPdRhRuCu, and (g–i) PtPdRhRuCu-2 samples. Corresponding cartoon structure drawings are shown inset of (a, d, g). Error bars in (c, f, i) based on the measurements at three points within the selected region. Scale bars: (a, d, g) 50 nm; (b, e, h) 10 nm. Source data are provided as a Source Data file.



Supplementary Figure 28. The characterizations of (a) HAADF-STEM (scale bar, 200 nm), (b) HRTEM (scale bar, 2 nm), (c) XRD, (d) SAXS pattern, and (e–i) XPS spectra of PtPdRhRuCu MMNs after the durability HER test in 1.0 M KOH. Inset of (b) shows the HAADF-STEM image and as marked possible atomic steps/kinks (scale bar, 1 nm). The mesoporous morphology of PtPdRhRuCu MMNs were maintained after HER test. The atomic steps and kinks were still observed in **Supplementary Figure 28b**, which might be a contributing factor for the stability. However, considering that local surface changes on the atomic scale on nanomaterials probably occurred during the electrochemical testing, rearrangement of surfaces of the electrocatalysts was not uncommon. The stable bulk morphology, composition, and porous structure of the nanosphere might be the main factor for maintaining long-term HER measurement. Source data are provided as a Source Data file.



Supplementary Figure 29. HAADF-STEM image and corresponding map of PtPdRhRuCu MMNs mixed with carbon after the durability HER test in 1.0 M KOH. Scale bars, 200 nm.



Supplementary Figure 30. Polarization curves of PtPdRhRuCu MMNs and Pt MNs in (a) 0.5 M H₂SO₄ and (b) 1.0 M PBS solutions. (c) Comparison of overpotentials required to achieve a current density of 10 mA cm⁻². Error bars obtained from three independent experiments. Long-term chronopotentiometry tests for PtPdRhRuCu MMNs in (d) 0.5 M H₂SO₄ and (e) 1.0 M PBS electrolytes at different current densities. Source data are provided as a Source Data file.



Supplementary Figure 31. The experimentally measured and theoretically calculated amounts of H₂ on PtPdRhRuCu MMNs in different electrolytes. The commercial RuO₂ was used as oxygen evolution reaction catalyst. The applied potential of water splitting for 1.0 M KOH, 0.5 M H₂SO₄, and 1.0 M PBS solution is -2.0, -2.0, and -3.0 V, respectively. Source data are provided as a Source Data file.



Supplementary Figure 32. Calibration curves of (a) Hg/HgO, (b) SCE, and (c) Ag/AgCl reference in 1.0 M KOH, 0.5 M H₂SO₄, and 1.0 M PBS electrolytes, respectively. The reference electrodes were calibrated with respect to the reversible hydrogen electrode three-electrode system using Pt wires as both the working and counter electrodes.¹⁴ The electrolytes, including 1.0 M KOH, 0.5 M H₂SO₄, and 1.0 M PBS was saturated with high-purity hydrogen for at least 30 min at room temperature and followed CV measurement performed at a scan rate of 1 mV s⁻¹. The potentials at which the currents crossed zero were taken as the thermodynamic potentials. Source data are provided as a Source Data file.

Supplementary Tables

Supplementary Table 1. The ICP–OES analysis of as-prepared and after alkaline HER test of PtPdRhRuCu MMNs.

Samula		Atomic co	ntent of the elem	ents (at%)	
Sample	Pt	Pd	Rh	Ru	Cu
As-prepared	23	22	20	13	22
After HER	24	24	19	12	21

Supplementary Table 2. The standard reduction potential of different half reactions for various metal precursors.^{15, 16}

Half reaction	E^0 (V vs. SHE)
$[PtCl_4]^{2-} + 2e^- \rightarrow Pt + 4Cl^-$	+0.76
$[PdCl_4]^{2-}+2e^- \rightarrow Pd + 4Cl^-$	+0.59
$[RhCl_6]^{3-}+ 3e^- \rightarrow Rh + 6Cl^-$	+0.43
$Ru^{3+} + 3e^- \rightarrow Ru$	+0.39
$Cu^{2+} + 2e^- \rightarrow Cu$	+0.34

Sample	Feeding amounts (mL) (40 mM)			Final composition (at%) ^a						
	[PtCl4] ²⁻	[PdCl4] ²⁻	[RhCl6] ³⁻	Ru^{3+}	Cu ²⁺	Pt	Pd	Rh	Ru	Cu
Pt	2.0	-	-	-	-	100	-	-	-	-
PtPd	1.6	0.4	-	-	-	78	22	-	-	-
PtRh	1.6	-	0.4	-	-	83	-	17	-	-
PtRu	1.6	-	-	0.6	-	90	-	-	10	-
PtCu	1.6	-	-	-	0.4	81	-	-	-	19
PtPdRu	1.2	0.4	-	0.6	-	68	21	-	11	-
PtRhRu	1.2	-	0.4	0.6	-	73	-	18	9	-
PtRuCu	-	-	-	0.6	0.4	83	-	-	9	18
PtPdRhRu	0.8	0.4	0.4	0.6	-	46	22	20	12	-
PtRhRuCu	0.8	-	0.4	0.6	0.4	51	-	19	10	20
PtPdRhRuCu	0.4	0.4	0.4	0.6	0.4	23	22	20	11	24

Supplementary Table 3. The feeding amounts of various metal precursors and the final compositions of Pt-based MNs.

^aThe final compositions of different Pt-based samples were got from SEM-EDS.

	Preparation parameters							
Sample		Organic	Deionized		T (0C)	T ' (1)		
	Block copolymers	solvent (mL)	water (mL)	Acid (M, mL)	Temp. (°C)	Time (n)		
Pt	PEO10500-b-PMMA18000	DMF (0.4)	0.5	HCl (6, 0.1)	55	4		
Pd ^a	PS5000- <i>b</i> -PEO2200	THF (0.2)	0.7	HCl (2, 0.1)	40	4		
Rh	PEO10500-b-PMMA18000	DMF (0.6)	0.4	0	60	12		
PtPd	PEO10500-b-PMMA18000	DMF (0.8)	0.2	HCl (6, 0.4)	60	4		
PtM (M = Rh/Ru)	PEO10500-b-PMMA18000	DMF (0.4)	0	HClO ₄ (2, 0.7)	50	6		
PtRuCu	PEO10500-b-PMMA18000	DMF (0.4)	0	HClO ₄ (2, 0.7)	60	6		
PtPdRu	PEO10500-b-PMMA18000	DMF (0.8)	0.2	HCl (6, 0.4)	70	6		
PtPdRhRu	PEO10500-b-PMMA18000	DMF (0.8)	0.2	HCl (6, 0.4)	70	6		
PtRhRuCu	PEO10500-b-PMMA18000	DMF (0.8)	0.2	HCl (6, 0.4)	80	6		
PtPdRhRuCu	PEO10500-b-PMMA18000	DMF (0.8)	0.2	HCl (6, 0.4)	80	6		
PtPdRhRuCu-1	PEO10000-b-PMMA5500	DMF (0.8)	0.2	HCl (6, 0.4)	80	6		
PtPdRhRuCu-2	PEO10500-b-PMMA22000	DMF (0.8)	0.2	HCl (6, 0.4)	80	6		
PtPdCuNiCo	PEO10500-b-PMMA18000	DMF (0.4)	0	HClO ₄ (2, 0.8)	110	8		
PtPdCuNiCoMo	PEO10500- <i>b</i> -PMMA18000	DMF (0.4)	0	HClO ₄ (2, 0.8)	110	8		

Supplementary Table 4. The synthesis conditions for noble-mental-based MNs.

^a The aqueous H₂PdCl₄ was used for synthesis of Pd MNs while the aqueous Na₂PdCl₄ was used for other PtPd-based MNs.

Atom	Löwdin charge	Initial valence electrons	Partial charge transfer (e)
Pt	10.37	10.0	+0.37
Pd	8.51	9.00	-0.49
Rh	11.19	11.0	+0.19
Ru	9.51	10.0	-0.49
Cu	8.34	8.00	+0.34

Supplementary Table 5: The calculation of Löwdin charge^a analysis for each metal of HEAS, in comparison with the initial valence electron.

^aLöwdin charge was study the charge transfer effects. It depicted the total number of valence electrons for each element. By subtract it to the initial valence electrons of the atom would produce the partial charge of each atom.

Supplementary Table 6. The HER performance of recent reported MMA electrocatalysis at different pH.

Materials	Shape	Preparation method	Electrolyte	Overpotential (mV) @10 mA cm ⁻²	Reference	
			1.0 M KOH	10		
PtPdRhRuCu	mesoporous	soft-templating chemical	1.0 M PBS	28	This work	
	nanospheres	reduction (80 °C)	0.5 M H ₂ SO ₄	13		
NiCoFePtRh	nanoparticles	chemical reduction by SBH and H ₂ treatment	0.5 M H ₂ SO ₄	27	13	
N'EM C.C.	1 11	1. 1200 %	1.0 M KOH	172	17	
N120Fe20M010C035Cr15	bulk	arc-melting 1200 °C	0.5 M H ₂ SO ₄	107	17	
	1 11	melting, fast cooling, and	0.1.0 M KOH	-	18	
AINiCuPtPdAu	bulk	dealloying	$0.5 \mathrm{~M~H_2SO_4}$	-	10	
AlNiCoIrMo	bulk	melting pure melts	0.5 M H ₂ SO ₄	18.5	19	
PtAuPdRhRu	nanoparticles	chemical reduction by ethylene glycol	1.0 M KOH	-	20	
FeCoNiAlTi intermetallic	L1 ₂	arc-melting and dealloying	1.0 M KOH	88.2	21	
		chemical reduction by	1.0 M KOH	17		
IrPdPtRhRu	nanoparticles	triethylene glycol	0.05 M H2SO4	33	12	
FeCoPdIrPt@GO	nanoparticles	fast moving bed pyrolysis	1.0 M KOH	42	22	
CoCrFeMnNiP	bulk	heating with ethylene glycol- 400 °C	1.0 M KOH	136	23	
Pt ₁₈ Ni ₂₆ Fe ₁₅ Co ₁₄ Cu ₂₇	nanoparticles	one-pot oil phase synthesis (220 °C)	1.0 M KOH	11	8	
	111-		1.0 M PBS	24	24	
CUAINIMOFE	bulk	arc-melting and dealloying	1.0 M KOH	9.7	24	
Al ₈₉ Ag ₁ Au ₁ Co ₁ Cu ₁ Fe ₁ Ir ₁ Ni ₁ Pd ₁ Pt ₁ Rh ₁ Ru ₁	bulk	arc-melting and dealloying	0.5 M H ₂ SO ₄	32	25	
PdFeCoNiCu	nanoparticles	one-pot oil phase synthesis (220 °C)	1.0 M KOH	18	26	
DIDAC N'D			1.0 M KOH	32	27	
PdPtCuNiP	metallic glass	large-scalable melt-spinning	$0.5 \mathrm{~M~H_2SO_4}$	62	27	
RuRhPdAgOsIrPtAu	nanoparticles	Heating with triethylene glycol (230 °C)	0.15 M H ₂ SO ₄	-	11	
		arc melting and single-roller				
Ni14Co14Fe14Mo6Mn52	bulk	melt spinning, followed	1.0 M KOH	14	28	
		dealloying				
PtRuRhCoNi	nanowire	olevlamine-200 °C	0.5 M H ₂ SO ₄	13	29	
		oleylanine-200 C	1.0 M KOH	15		
CuCoNiFeMn	bulk	high-energy hall milling	1.0 M KOH	152 @ 70	30	
	oun	men energy our mining	$0.5 \mathrm{~M~H_2SO_4}$	214 @ 20		

			1.0 M PBS	320 @ 50	
		traditional solid-state			
Co _{0.6} (VMnNiZn) _{0.4} PS ₃	bulk	reaction with their nominal	1.0 M KOH	65.9	31
		composition powders			
FaCaNiCu	hierarchical	physical metallurgy and	1.0 M VOH	42.2	32
reconicu	porous	dealloying strategies	1.0 M KOH	42.2	
FaCaNiCuDd	bull	pulsed DC reactive	1.0 M VOH	20	33
reconicura	UUIK	magnetron sputtering	1.0 M KOH	29	
CoZnCdCuMnS	nanoarrays	hydrothermal	1.0 M KOH	173	34
		electrospinning technology			
FeCoNiMnRu	nanoparticles	and graphitization	1.0 M KOH	5	35
		process.			
	nononarticles	ethylene glycol and thermal	1.0 M KOH	158	36
ContentingZil/graphene	nanoparticles	treat at 900 °C	$0.5 \ M \ H_2 SO_4$	131	
IrPdPtRhRu	nanoparticles	continuous-flow reactor	1.0 M HClO ₄	6	37
	nonowiro/non	hydrothermal reaction			
FeCoNiCuMnN/CC-400	oshaat	calcination at 400 °C under	1.0 M KOH	184	38
	osheet	Ar/NH ₃			
		mechanical alloying and			
CoCrFeNiAl	bulk	spark plasma sintering	$0.5 \ M \ H_2 SO_4$	73	39
		consolidation			
Al82Ni6Co3Mn3Y3Au3	bulk	arc-melting-dealloying	$0.5 \ M \ H_2 SO_4$	24	40

Supplementary Table 7: Summary of recently reported Pt-based HER electrocatalysts in different media.

	C1		Overpotential (mV)	D (
Materials	Snape	Electrolyte	$@10 \text{ mA cm}^{-2}$	Reference
		1.0 M KOH	10	
PtPdRhRuCu	mesoporous nanospheres	1.0 M PBS	28	This work
		0.5 M H ₂ SO ₄	13	
PtNi5-0.3	nanoparticles	1.0 M KOH	26.8	41
hcp-Pt-Ni	nano-multipods	1.0 M KOH	65	42
PtSn ₄	bulk single crystal	1.0 M KOH	37	43
Pt ₃ Ni ₃ -NWs	nanowires	1.0 M KOH	40	44
Pt-SAs/MoSe ₂	single atom	1.0 M KOH	29	45
PtRu NCs/BP	nanoclusters	1.0 M KOH	22	46
PtNi-O/C	nanoparticles	1.0 M KOH	40	47
c-PtPd@a-NiB CSHs	hollow nanopolyhedra	1.0 M KOH	31	48
Ni-N-C-250/Pt	nanoparticles	1.0 M KOH	83.5	49
PtTe ₂ -600 NSs	nanosheets	1.0 M KOH	22	50
Pt/np-Co _{0.85} Se	single-atom	1.0 M PBS	55	51
Pt@CoOx	nanoparticles	1.0 M PBS	82	52
BC ₃ N@Pt	nanoparticles	1.0 M PBS	38.5	53
Pt@Cu-0.3	blackberry-shaped nanocrystals	1.0 M PBS	35	54
Pt/PtTe2/NiCoTe2	hollow microspheres	1.0 M PBS	36	55
GaPt ₃	nanoparticles	1.0 M PBS	103	56
PtAC-MnO ₂	cluster	1.0 M PBS	41	57
Pt/Ni ₃ S ₂	nanoparticles	1.0 M PBS	109	58
Pt-Co(OH) ₂ /CC	nanoparticles	1.0 M PBS	84	59
Pd@Pt	nanocube	0.5 M H ₂ SO ₄	15	60
Pt3.21Ni@Ti3C2	ultrathin Nanowires	0.5 M H ₂ SO ₄	18.5	61
$Pt_1Cu_{1.03}$	branched nanodendrites	$0.5 \mathrm{~M~H_2SO_4}$	20	62
PtRh	dendritic nanoassemblies	0.5 M H ₂ SO ₄	27	63
Pt-RuO ₂ HNSs	hollow nanospheres	0.5 M H ₂ SO ₄	26	64
Pt ₁ /OLC	atomically dispersed	0.5 M H ₂ SO ₄	38	65
Pt ₃ Co NPs	nanoparticles	0.5 M H ₂ SO ₄	42	66
$La_2Sr_2PtO_{7+\delta}$	bulk	0.5 M H ₂ SO ₄	27	9
Pt-WO ₃	clusters	0.5 M H ₂ SO ₄	42	67
Pt-ACs/CoNC	atomic clusters	0.5 M H ₂ SO ₄	24	68
Pt-NPs/CoNC	nanoparticles	0.5 M H ₂ SO ₄	42	68

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