Nanoscale reshaping of resonant dielectric microstructures by light-driven explosions. Supplementary Information

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Supplementary Figure 1. PIC results for a scaled-down microresonator and pump wavelength. The scaling factor is 0.5, where $\lambda_L = \lambda_R = 1.95 \,\mu\text{m}$, and all the dimensions of the simulation are halved. The minimum projected feature from the final ion temperature distribution is around 40 nm.

Supplementary Note 1. The Role of Native Oxide

Since the experiments were conducted at ambient conditions, the existence of the native oxide layer on silicon was presumed. However, we render the role of the native oxide (thickness ~2 nm) negligible compared to the bulk of the microresonators with their characteristic dimensions. Since the wavelength of the resonance is, as an approximation, proportional to the $n_{eff} - 1$, where n_{eff} is the effective refractive index of the resonator. The overall change of the resonator mode's central wavelength due to the native oxide can be roughly estimated as $\Delta\lambda/\lambda \approx \Delta n_{eff}/(n_{eff} - 1)$, where $\Delta n_{eff} \approx n_{Si} - n_{Si+SiO_2}$, $n_{Si} = 3.4$, and n_{Si+SiO_2} is the adjusted effective refractive index due to the addition of the native oxide, estimated at $n_{Si+SiO_2} = (V_{Si}n_{Si} + V_{SiO_2}n_{SiO_2})/(V_{Si} + V_{SiO_2})$, where *V* is the respective volume of the material. At 2 nm thickness of SiO₂ and approximating the microresonator as a cuboid with 0.6 µm × 1 µm × 1 µm dimensions, the ratio of the volumes $\frac{V_{SiO_2}}{V_{Si}} \approx 0.015$, and $n_{SiO_2} = 1.5$, the overall n_{eff} correction due to the oxide is $\Delta n_{eff} \approx 0.03$, making the central wavelength correction $\Delta\lambda \approx 50$ nm, which is well within the bandwidths of the pulse and the resonance.

Supplementary Note 2. The Role of Resonator Heating

In our particle-in-cell calculations, we have estimated the amount of heat that is generated through the relevant processes of free carrier generation and thermalization with the lattice. Fig. 5 of the main text shows the overall lattice temperature elevation and its distribution throughout the volume of the material, peaking at around 6,000-7,000 K in the hot spot. In this dynamic picture, the elevated temperature profile will remove the material from the V groove in the microresonator. However, when averaged over the material's volume, the resonator's overall temperature rise is mild. With the hot spot volume being roughy 0.1 μ m × 0.1 μ m × 0.6 μ m = 0.006 μ m³, and the resonator volume estimated at 1 μ m × 1 μ m × 0.6 μ m = 0.6 μ m³, the averaged overall temperature rise of the resonator is estimated around 50 - 60 K which is far below the melting point of silicon (1,683 K), without taking into account the heat dissipation in the substrate and the removal of heat by the ejected particulates. Therefore, our finetuning of the laser beam intensity allowed us highly localized material modification and removal without any significant modification to the remainder of the material. The studies of the microscopic changes, such as localized amorphization of Si, represent an exciting avenue for research that will be considered for our future studies.