

Supporting Information

High-Performance Semiconducting Carbon Nanotube Transistors Using Naphthalene Diimide-based Polymers with Biaxially Extended Conjugated Side Chains

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Table S1. Optical, electrochemical, and thermal properties of the polymers.

CPs	λ_{\max} (nm) ^a	HOMO (eV) ^b	LUMO (eV) ^b	E_g^{CV} (eV) ^c	E_g^{UV} (eV) ^d	T_d (°C) ^e
P1	389, 705	-5.96	-4.09	1.87	1.76	436
P2	369, 637	-5.91	-3.77	2.14	1.95	312
P3	375, 635	-5.89	-3.73	2.16	1.95	358

^a UV-Vis absorption maximum position of the CPs dissolved in toluene. ^b Derived from the CV oxidative onset potential determined using Fc/Fc⁺ as an internal potential reference. ^c HOMO and LUMO gap determined from the difference between the oxidative and reductive onsets in CV profiles. ^d Energy bandgap determined from the onset wavelength in UV-Vis spectrum. ^e Determined from TGA at 5% weight loss in a nitrogen atmosphere.

Table S2. FET dielectric layer parameters.

Layer	Thickness (nm)	Dielectric constant (ϵ_r)
SiO₂	300	3.9
SBS	30	2.4

Table S3. Summary of the sorting parameters of CP/*sc*-SWNT solutions and the crystallographic parameters of the pristine CP films.

CPs	$C_{sc\text{-SWNTs}}$ (g L ⁻¹) ^a	ϕ ^a	Purity (%)	Yield (%)	d_{100} (Å) ^b	q_{100} (Å ⁻¹) ^b
P1	0.044	0.325	~99	34.0	25.1	0.25
P2	0.003	0.429	>99	2.3	31.9	0.20
P3	0.032	0.586	>99	23.3	34.3	0.18

^a *sc*-SWNT concentration of the as-sorted solutions and ϕ value determined from the UV-Vis absorption spectra in **Figure S19**. ^b d -spacing and q values are extracted from the 1D GIXD profiles.

Table S4. Crystallographic parameters of the CP/*sc*-SWNT films, including the *d*-spacing, full-width at half-maximum (FWHM), and paracrystalline disorder of the IP(010) diffraction peaks.

CPs/ <i>sc</i> -SWNTs	d_{010} (Å)	q_{010} (Å ⁻¹)	FWHM	g_{010} (%)
P1	4.65	1.352	0.245	16.9
P2	4.61	1.362	0.262	17.5
P3	4.59	1.369	0.267	17.6

Table S5. Device performances of FETs comprising the CP/*sc*-SWNTs.

CPs/ <i>sc</i> -SWNTs	μ_h (cm ² V ⁻¹ s ⁻¹) ^a	V_{th} (V) ^b	I_{on}/I_{off} ^c	S_s (V/dec) ^d	N_{tr} ($\times 10^{12}$ cm ⁻¹ eV ⁻¹) ^e
P1 ($V_d = -1$ V)	0.058	3.8	10 ⁵	5.8	5.27
P3 ($V_d = -1$ V)	0.079	11.1	10 ⁵	4.6	4.18
P1 ($V_d = -10$ V)	0.48	4.4	10 ⁵	6.3	5.73
P3 ($V_d = -10$ V)	0.99	11.6	10 ⁶	5.6	5.09
P1 ($V_d = -100$ V)	2.21	13.7	10 ⁵	15.2	13.8
P3 ($V_d = -100$ V)	4.72	24.5	10 ³	12.6	11.5

^a Average hole mobility of the P-type FET device measured at $V_d = -10$ or -100 V. ^b Threshold voltage.

^c On-off current ratio. ^d Subthreshold swing. ^e Maximum interfacial trap density calculated from the subthreshold slope.

Table S6. Device performances of FETs comprising the CP thin films as the channel.

CPs	μ_e (cm ² V ⁻¹ s ⁻¹) ^a	V_{th} (V) ^b	I_{on}/I_{off} ^c
P1	3.5×10^{-2}	37.0	10 ³
P2	1.8×10^{-3}	14.4	10 ⁵
P3	3.4×10^{-5}	27.7	10 ³

^a Average electron mobility of the N-type FET device measured at $V_d = 100$ V. ^b Threshold voltage.

^c On-off current ratio.

Materials Characterizations.

^1H and ^{13}C nuclear magnetic resonance (NMR) spectra were recorded on a Bruker AVNEO500 with working frequencies of 500 MHz for ^1H and 125 MHz for ^{13}C , respectively, and with *d*-chloroform (CDCl_3) and dimethyl sulfoxide-*d*₆ ($\text{DMSO-}d_6$) as the *d*-solvent. Number-average molecular weights (M_n), weight-average molecular weights (M_w), and PDI of polymers were measured at 40 °C and flowing rate of 0.3 mL min⁻¹ in tetrahydrofuran (THF) by gel permeation chromatography (GPC) on a LC-20AT (Shodex GPC LF-604). Thermogravimetry analysis (TGA) measurement was performed from 100 to 750 °C with a heating rate of 10 °C min⁻¹ on a Perkin Elmer TGA4000. Differential scanning calorimetry (DSC) was measured under N₂ protection from 30 to 300 °C with a heating rate of 10 °C min⁻¹ on a Perkin Elmer DSC6000. The elemental analysis of reported polymers was performed on a UNICUBE (Elementar, Germany). Cyclic voltammetry (CV) analysis was conducted by a CHI 6273E electrochemical analyzer (CH Instrument Inc.), and ITO glass, Pt wire, and Ag/AgNO₃ (acetonitrile (*sat.*)) were served in a three-electrode cell system as the working, auxiliary, and reference electrodes, respectively.

Physical properties of the synthesized CPs:

P1: ^1H NMR (500 MHz, CDCl_3 , δ ppm, 25 °C, **Figure S10**): 8.67 (d, $J = 155$ Hz, 2H), 7.48–7.07 (br, 4H), 4.11 (s, 4H), 1.98 (s, 2H) 1.51–0.78 (br, 76H). M_n (GPC) = 66000, PDI = 1.69. Yield = 78% (550 mg).

P2: ^1H NMR (500 MHz, CDCl_3 , δ ppm, 25 °C, **Figure S12**): 8.89 (s, 2H), 8.22 (d, $J = 5$ Hz, 4H), 7.41 (d, $J = 6$ Hz, 4H), 7.32 (s, 2H), 7.22 (s, 2H) 4.20 (d, $J = 5.5$ Hz, 4H), 1.58 (s, 2H), 1.44–0.79 (br, 76H). M_n (GPC) = 5,200, PDI = 1.23 (bimodal molecular weight distribution due to the polymer aggregates). Anal. Calcd for $\text{C}_{76}\text{H}_{98}\text{N}_2\text{O}_8\text{S}_2$ (%): C, 74.1; H, 8.9; N, 2.3; S, 5.2. Found (%): C, 67.2; H, 7.4; N, 2.14; S, 3.6. Yield = 55% (100 mg).

P3: ^1H NMR (500 MHz, CDCl_3 , δ ppm, 25 °C, **Figure S14**): 8.91 (s, 2H), 8.76 (s, 2H), 7.18 (s, 4H), 7.38–7.29 (br, 2H), 7.24–7.16 (br, 2H) 4.26 (s, 4H), 1.57 (s, 2H), 1.42–0.75 (br, 154H). M_n (GPC) = 5,950, PDI = 1.4 (bimodal molecular weight distribution due to the polymer aggregates). Anal. Calcd for $\text{C}_{118}\text{H}_{178}\text{N}_2\text{O}_{12}\text{S}_2$ (%): C, 75.4; H, 9.5; N, 1.5; S, 3.4. Found (%): C, 80.0; H, 11.1; N, 0.7; S, 1.4. Yield = 67% (130 mg).

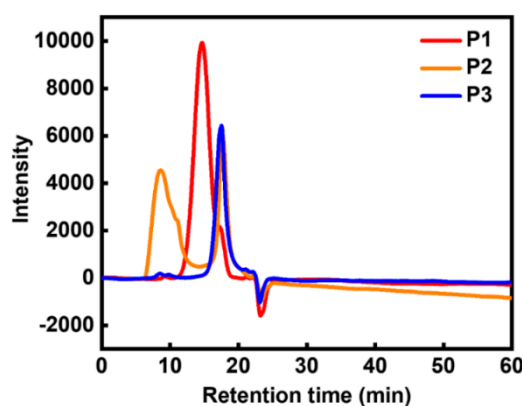


Figure S1. GPC profiles of the CPs in THF at 40 °C and 0.3 mL min^{-1} . Note that the molecular weight used poly(methyl methacrylate) (PMMA) as the molecular weight standard.

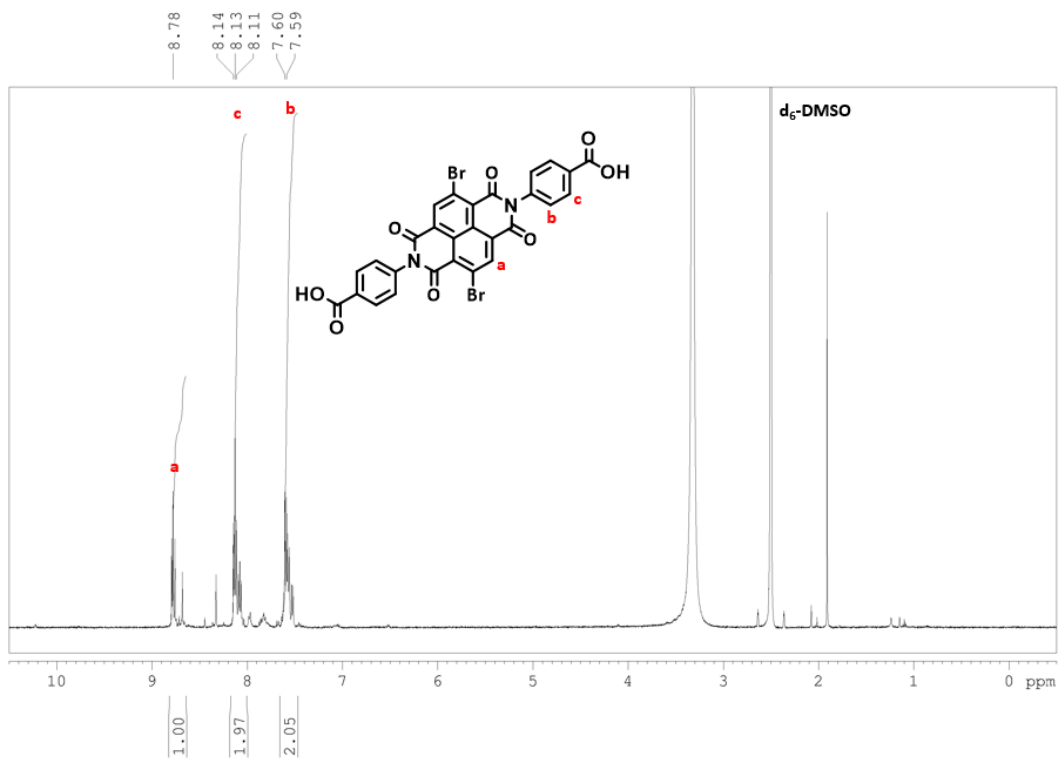


Figure S2. ^1H NMR spectrum of M1 in $\text{DMSO-}d_6$.

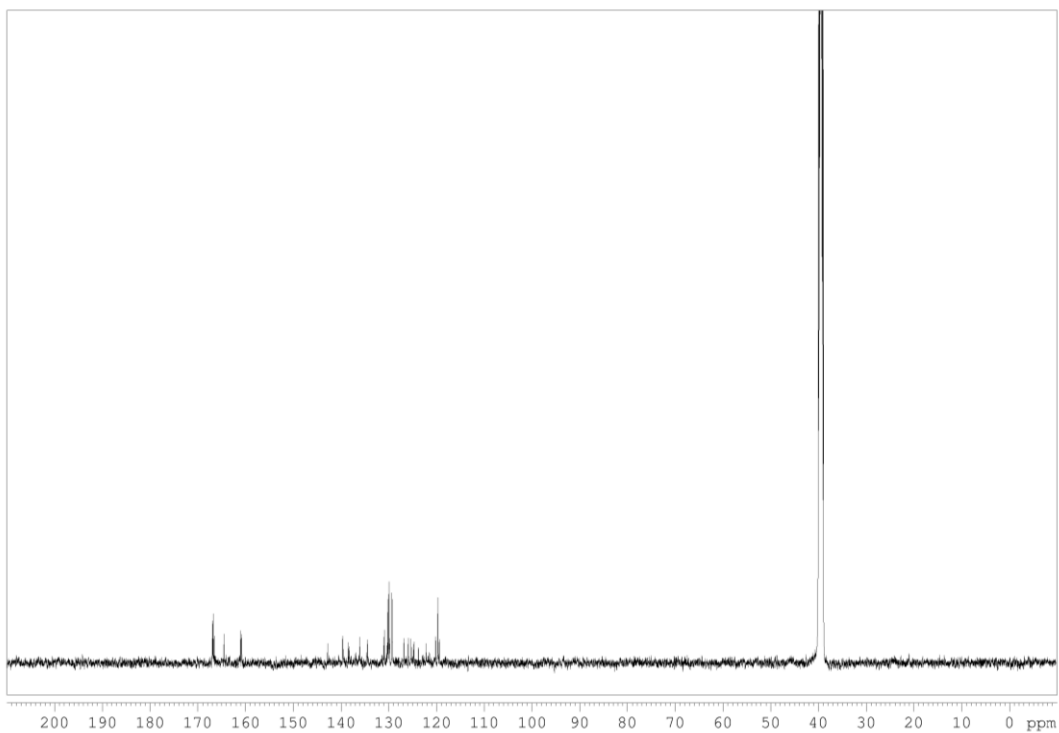


Figure S3. ^{13}C NMR spectrum of M1 in $\text{DMSO-}d_6$.

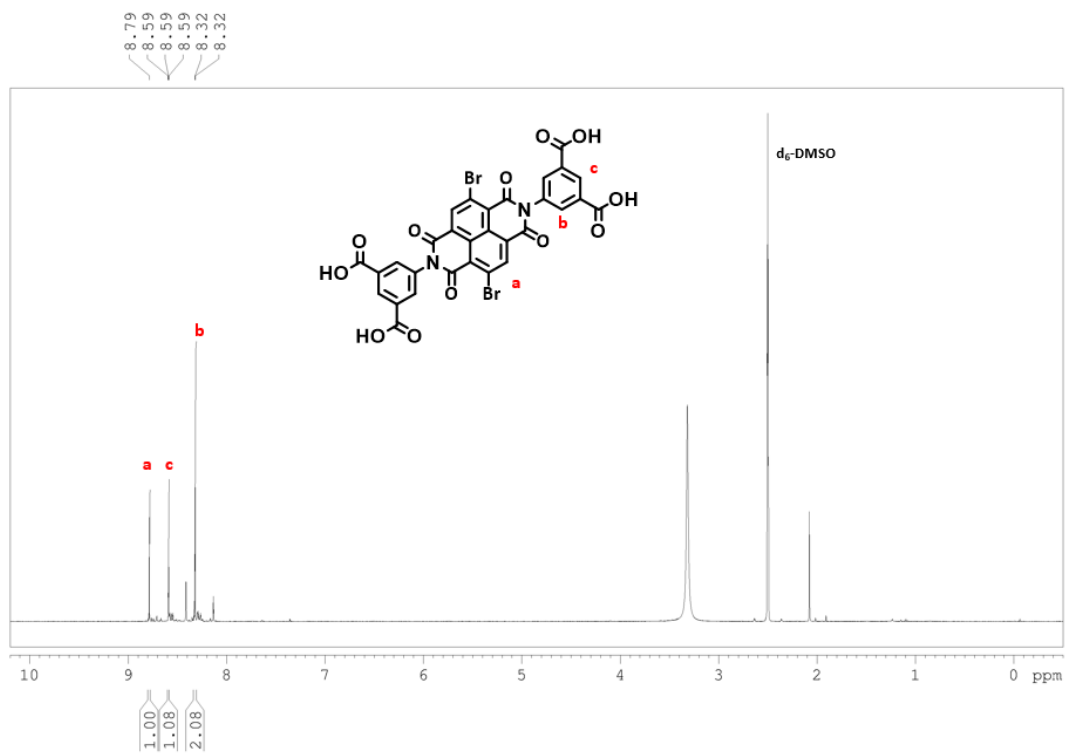


Figure S4. ^1H NMR spectrum of M2 in $\text{DMSO-}d_6$.

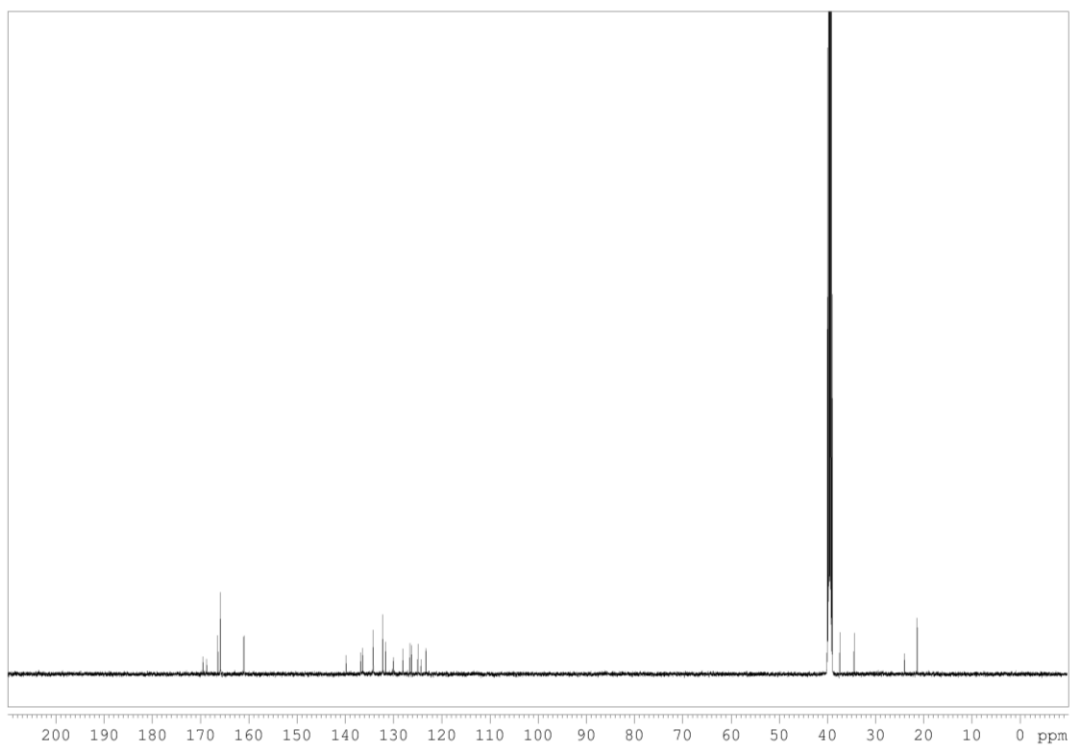


Figure S5. ^{13}C NMR spectrum of M2 in $\text{DMSO-}d_6$.

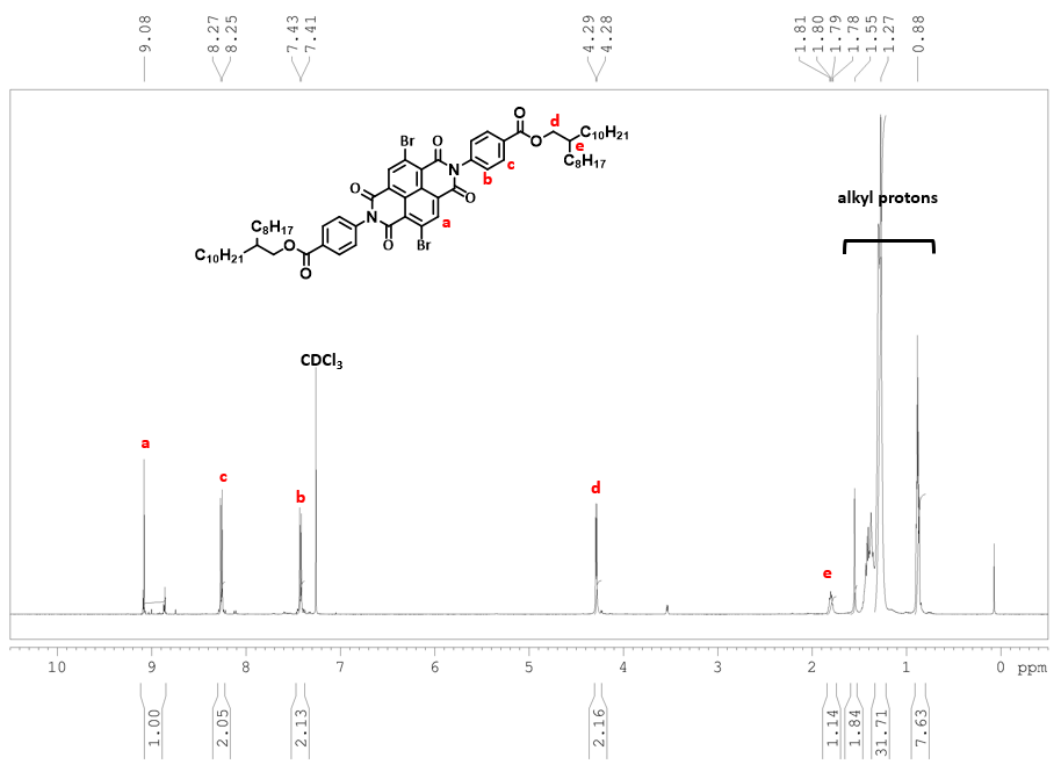


Figure S6. ^1H NMR spectrum of **M3** in CDCl_3 .

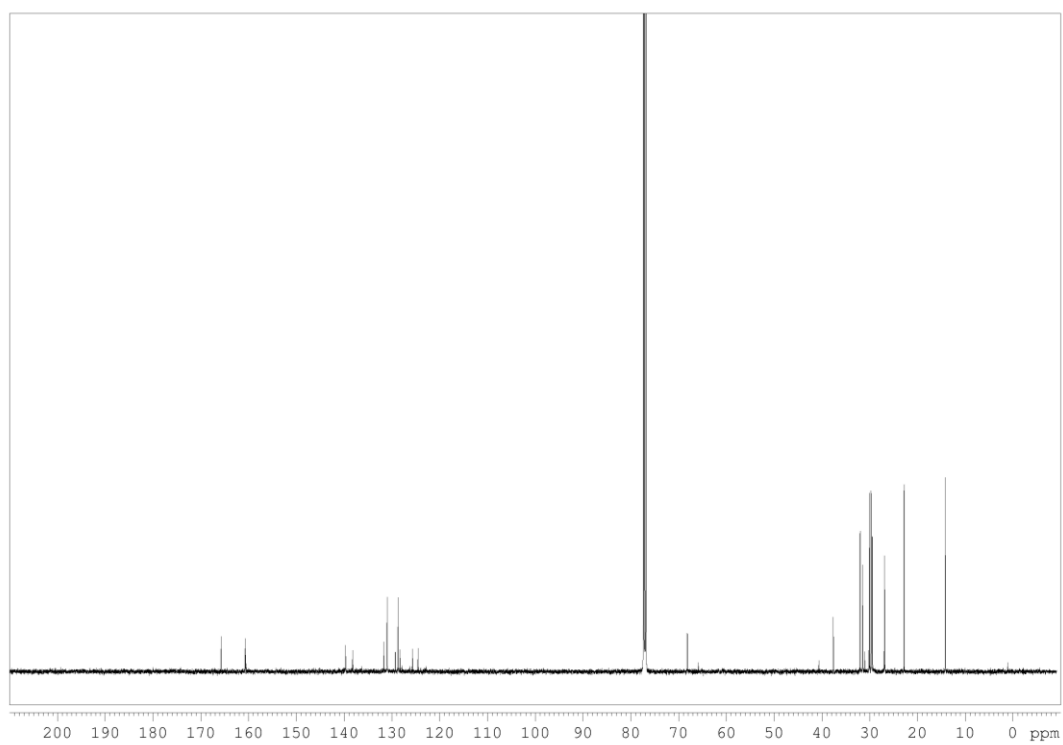


Figure S7. ^{13}C NMR spectrum of **M3** in CDCl_3 .

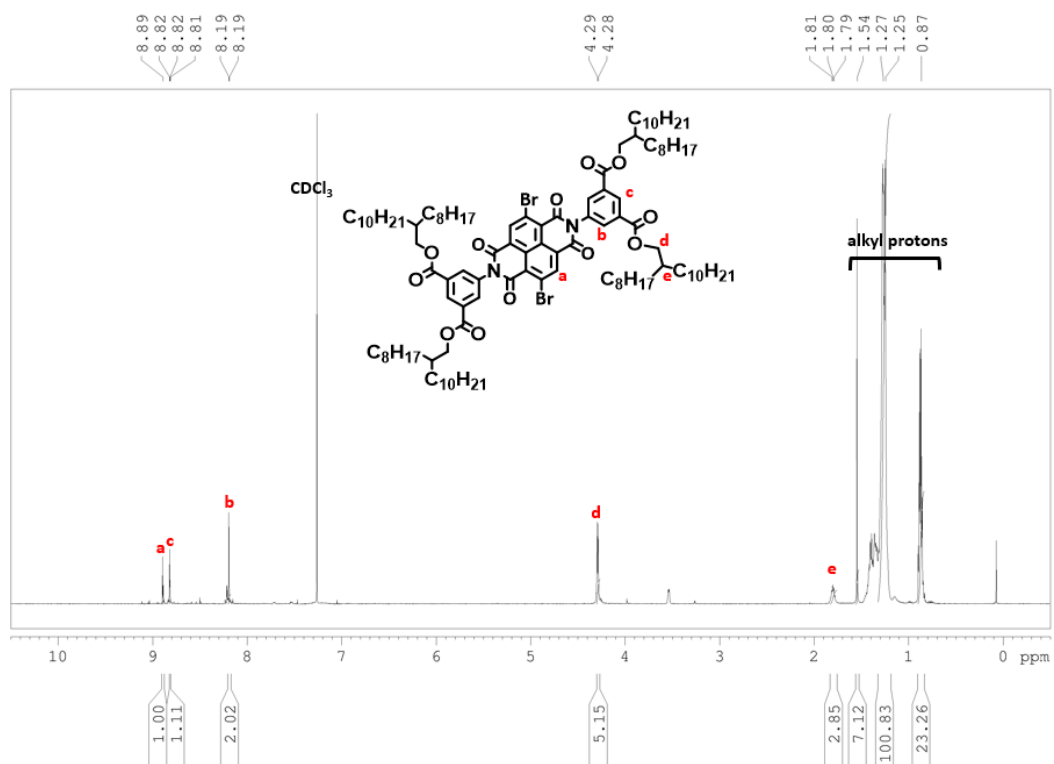


Figure S8. ^1H NMR spectrum of M4 in CDCl_3 .

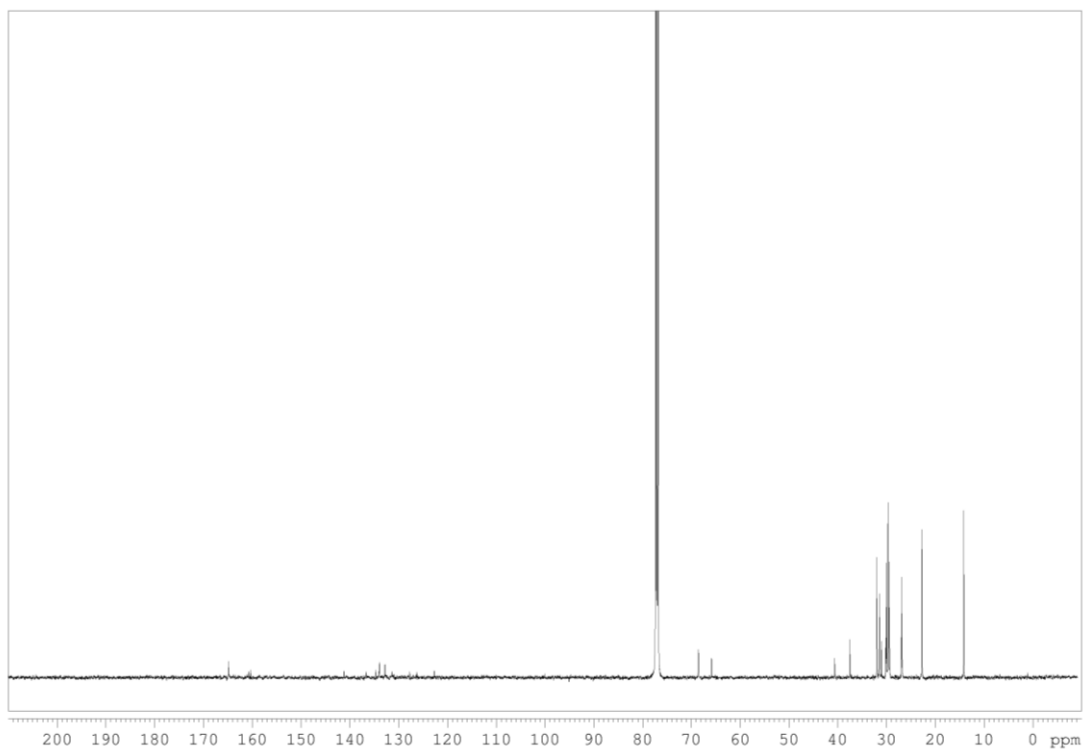


Figure S9. ^{13}C NMR spectrum of M4 in CDCl_3 .

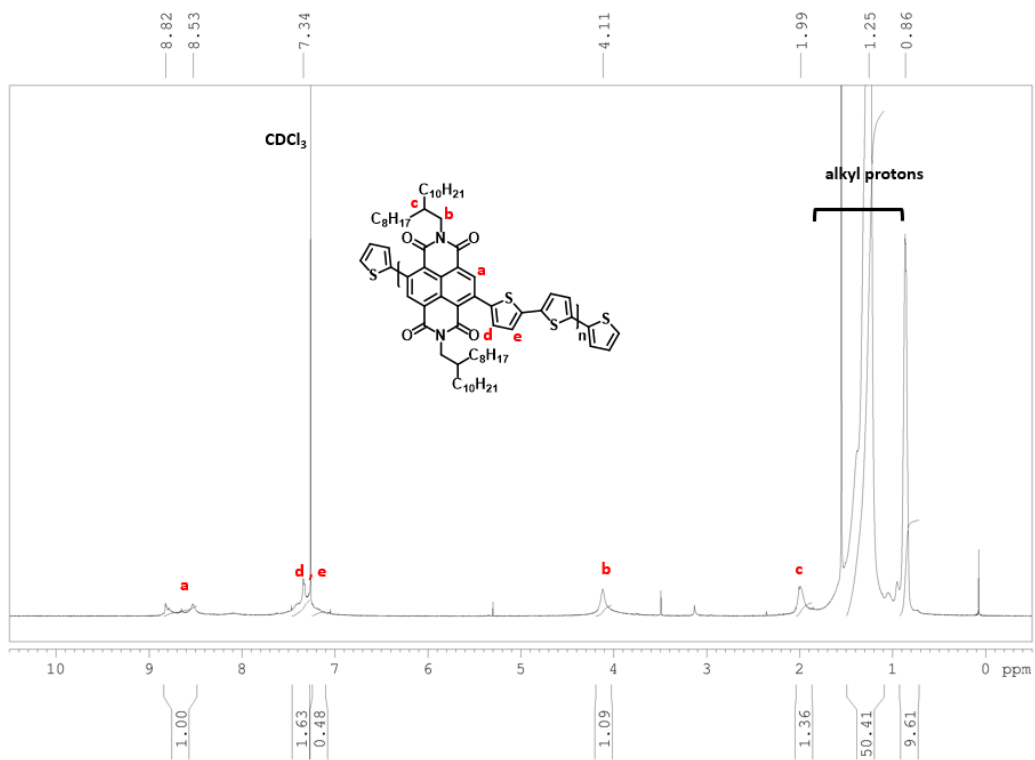


Figure S10. ¹H NMR spectrum of P1 in CDCl₃.

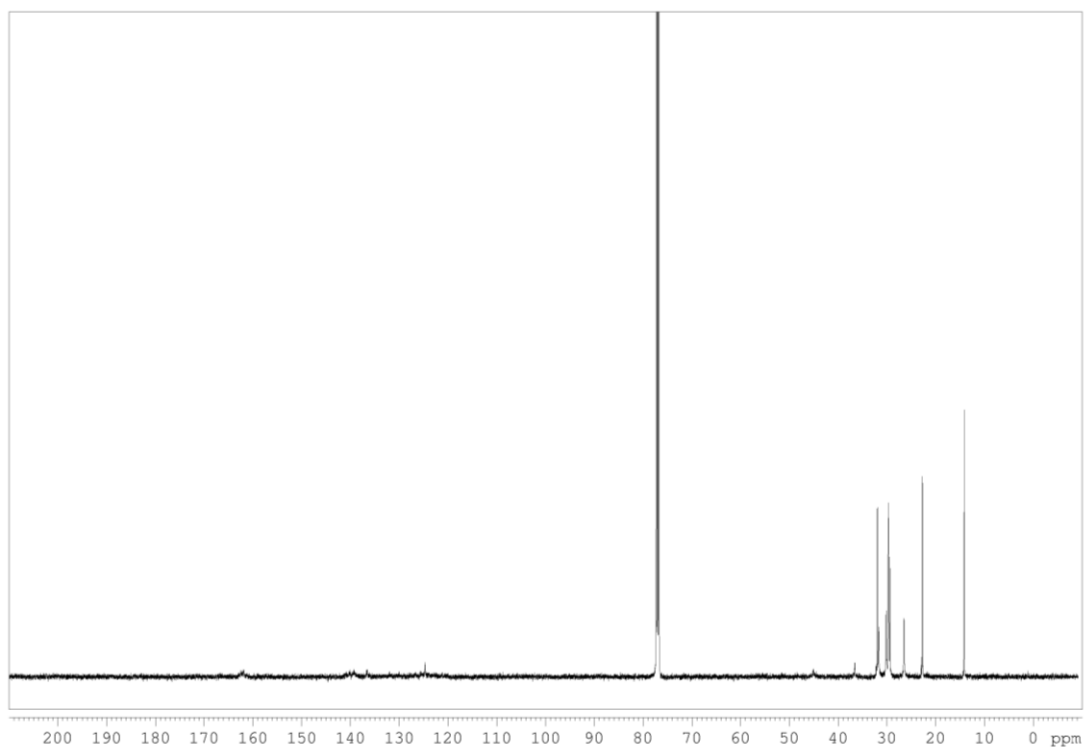


Figure S11. ¹³C NMR spectrum of P1 in CDCl₃.

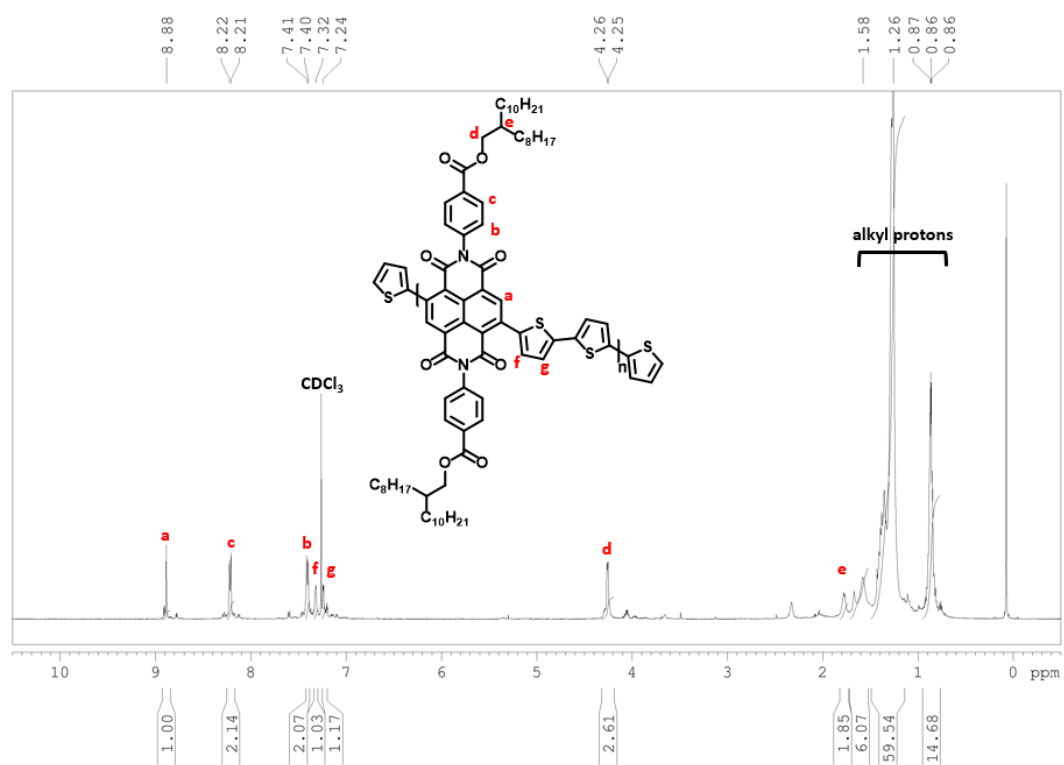


Figure S12. ^1H NMR spectrum of **P2** in CDCl_3 .

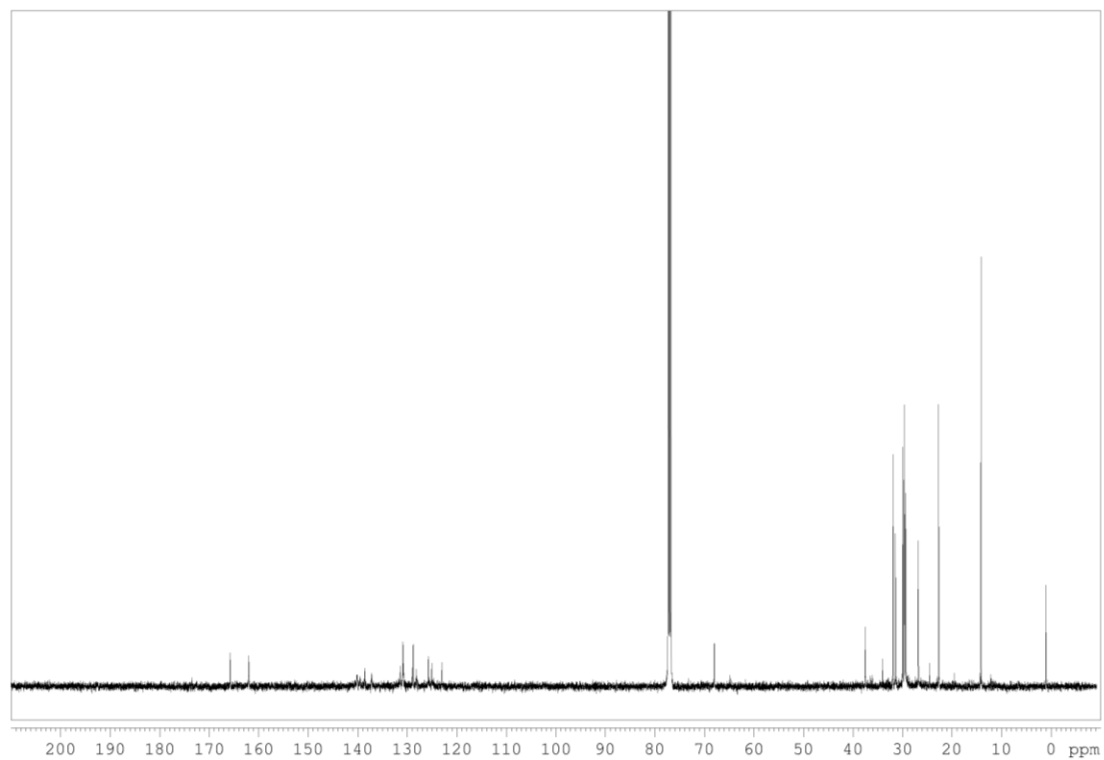


Figure S13. ^{13}C NMR spectrum of **P2** in CDCl_3 .

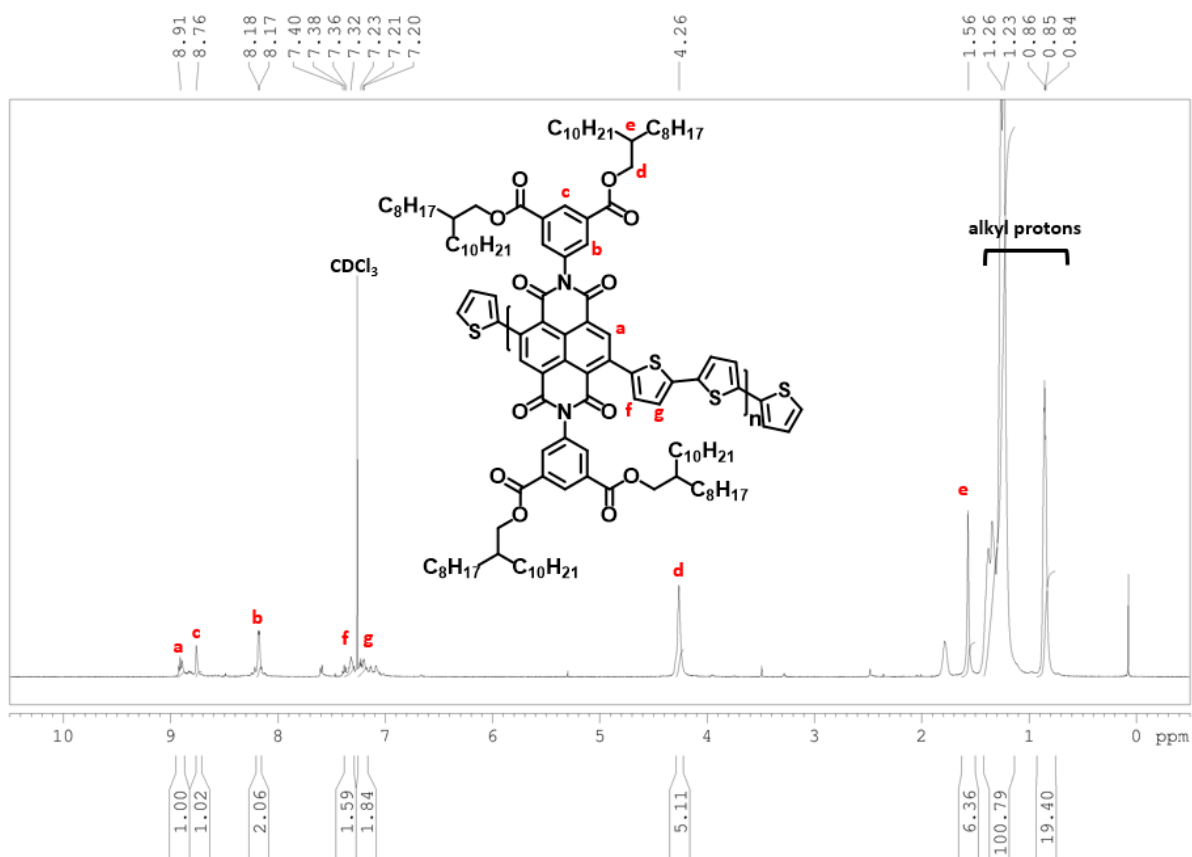


Figure S14. ^1H NMR spectrum of **P3** in CDCl_3 .

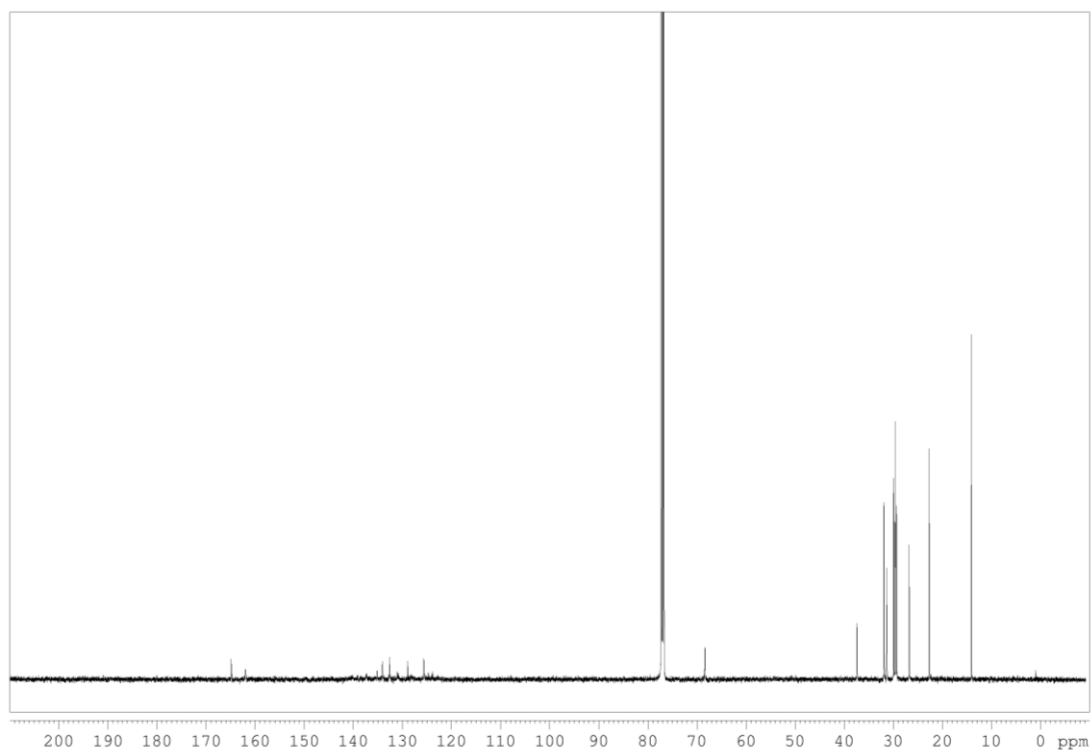


Figure S15. ^{13}C NMR spectrum of **P3** in CDCl_3 .

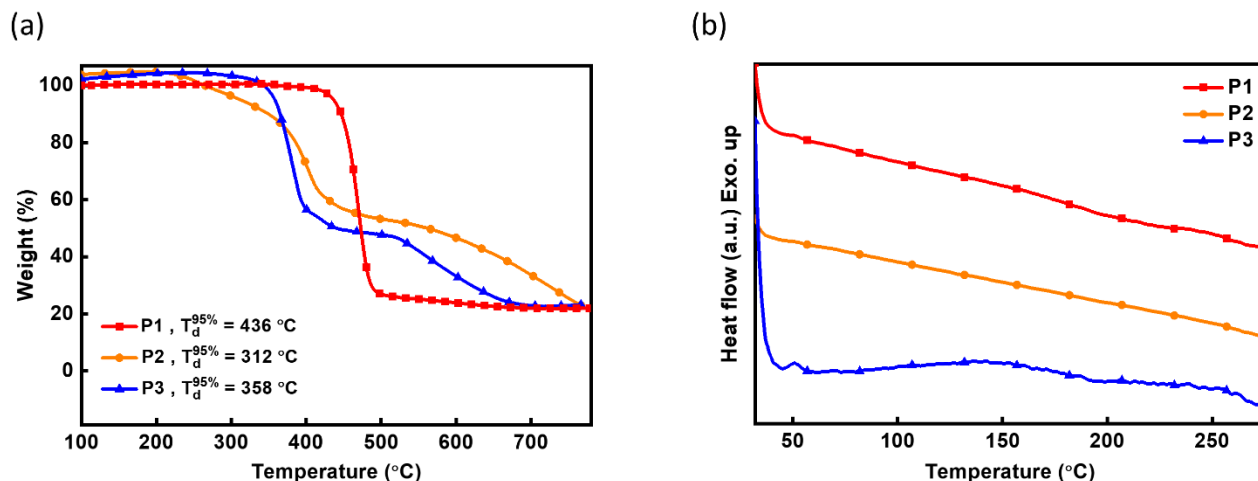


Figure S16. Thermal properties of the polymers studied: (a) TGA and (b) DSC profiles at a ramping rate of $10\text{ }^{\circ}\text{C min}^{-1}$.

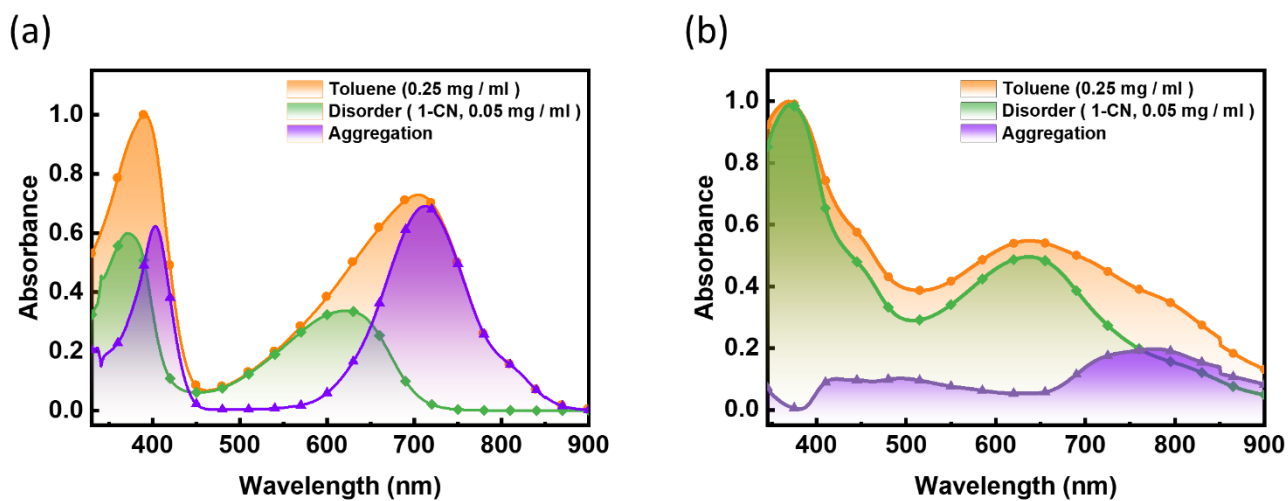


Figure S17. Aggregation and disorder fractions in the UV–Vis absorption spectra of polymers (a) **P1** and (b) **P2**. Note that the polymer solutions in toluene were prepared at a concentration of 0.25 mg mL^{-1} , and the disordered polymer solutions were prepared in 1-chloronaphthalene (1-CN) at a concentration of 0.05 mg mL^{-1} .

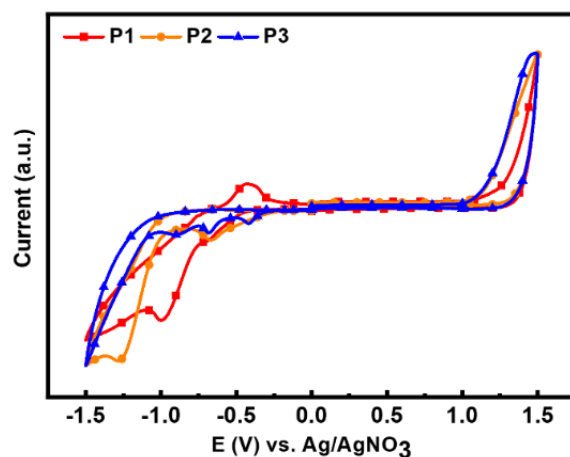


Figure S18. CV profiles of the polymer films coated on an ITO glass. The measurement was conducted at a scanning rate of 0.1 V s^{-1} .

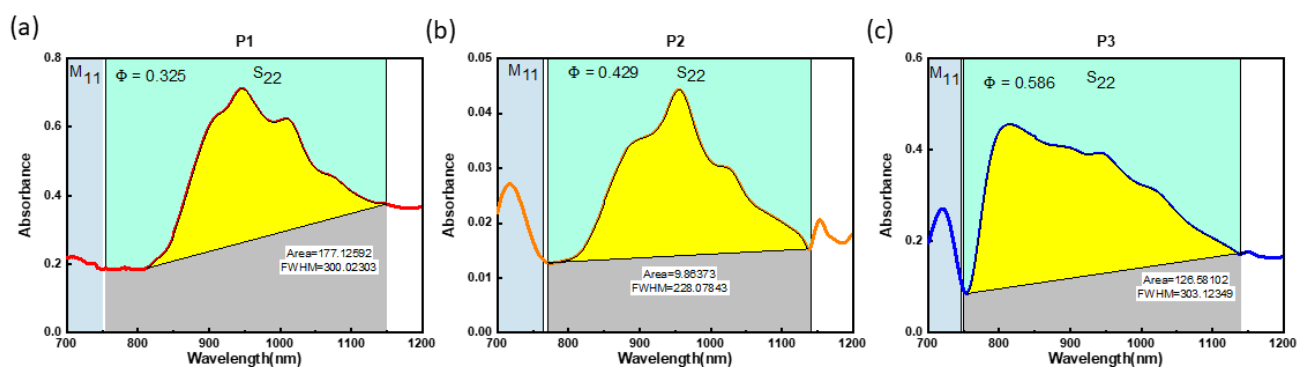


Figure S19. UV-Vis absorption spectra of the *sc*-SWNTs sorting solutions for calculating its purity: (a) **P1**/*sc*-SWNT, (b) **P2**/*sc*-SWNTs, and (c) **P3**/*sc*-SWNTs. The absorbances of NDI-based CPs were deconvoluted and subtracted from the absorption spectra of CP/*sc*-SWNT solutions.

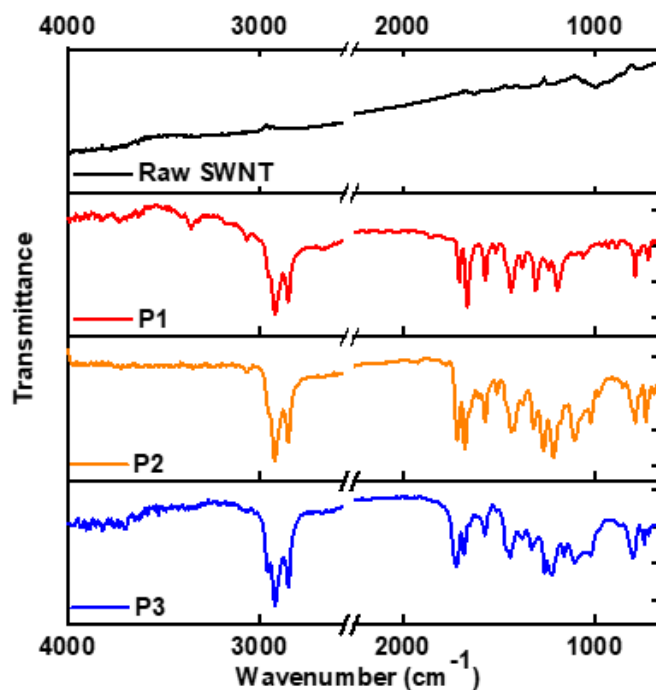


Figure S20. FT-IR spectra of the raw SWNT, and pure polymer powder of **P1**, **P2**, and **P3**.

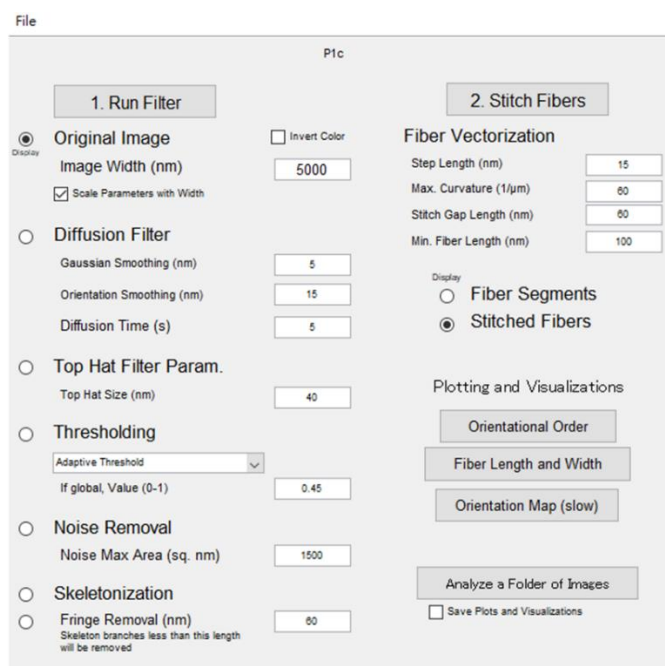


Figure S21. SWNT morphology fitting parameters in GTFiber software for extracting the lengths of *sc*-SWNTs in AFM topography. The program was developed by Persson *et al.* and reported in *Chem. Mater.* **2017**, *29*, 3-14.

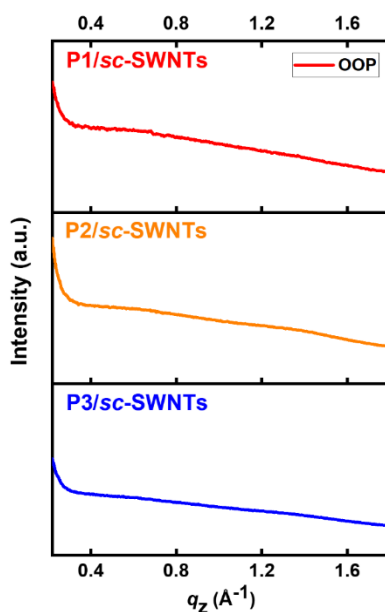


Figure S22. 1D GIXD profiles of CP/*sc*-SWNT films extracted along the OOP direction: the lamellar stacking is inhibited with the existence of *sc*-SWNTs.

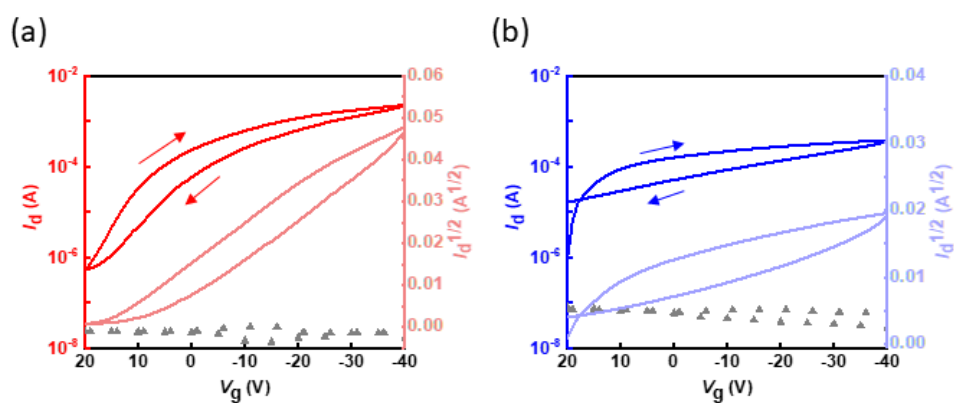


Figure S23. Transfer characteristics curves with $V_d = -100$ V for CP/*sc*-SWNT devices of (a) **P1**/*sc*-SWNTs and (b) **P3**/*sc*-SWNTs. Note that gray scattered dots represent the gate current measured, and the curves were swept from 20 to -40 V for *p*-type operation.

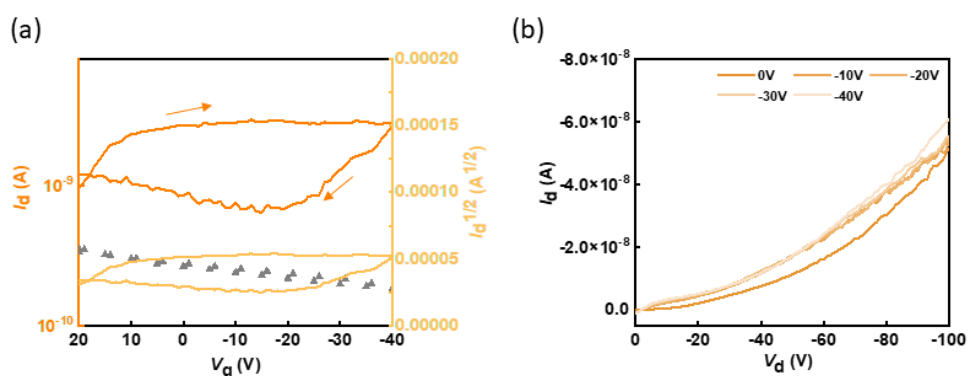


Figure S24. (a) Transfer curve with $V_d = -10$ V and (b) output curve of **P2**/*sc*-SWNTs device. Note that the device's drain current is low due to the low yield of *sc*-SWNTs sorted by **P2**.

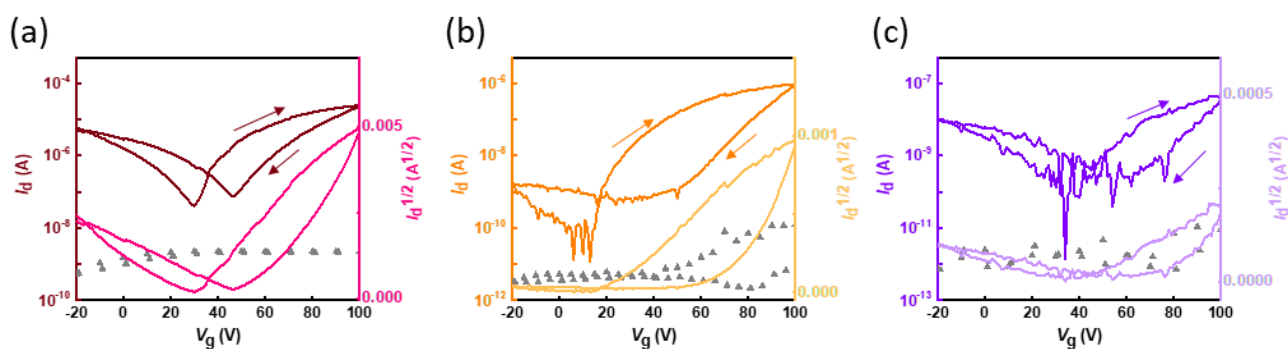


Figure S25. FET transfer characteristics of the *n*-type CP films of (a) **P1**, (b) **P2**, and (c) **P3**. Note that the gray scattered dots represent the gate currents measured, and the curves were swept from -20 to 100 V for *n*-type operation.

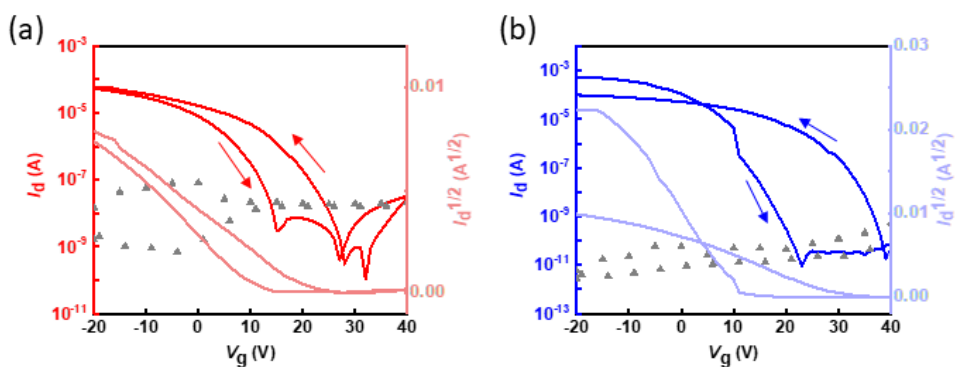


Figure S26. *N*-type transfer characteristic curves of the CP/*sc*-SWNTs device of (a) **P1**/*sc*-SWNTs and (b) **P3**/*sc*-SWNTs. Note that the curves were swept from -20 to 40 V at $V_d = 10$ V.

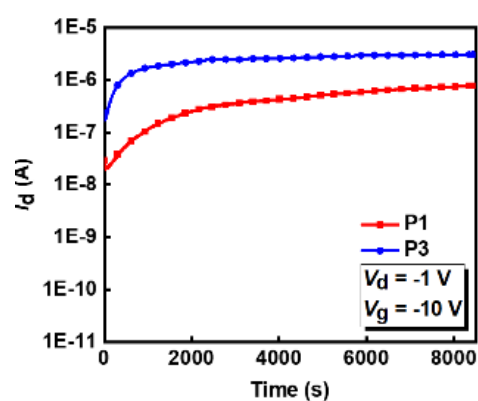


Figure S27. Bias stability test of CP/*sc*-SWNT based devices by applying $V_g = -10$ V at $V_d = -1$ V.

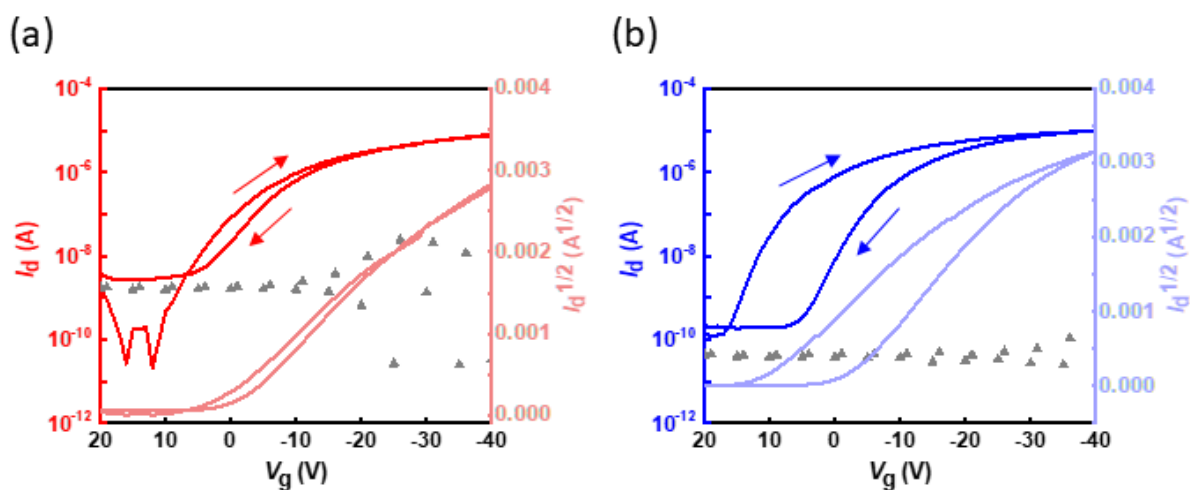


Figure S28. Transfer characteristics curves with $V_d = -1$ V for the CP/*sc*-SWNT devices of (a) **P1**/*sc*-SWNTs and (b) **P3**/*sc*-SWNTs. Note that gray scattered dots represent the gate current measured, and the curves were swept from 20 to -40 V for *p*-type operation.