

Supplementary information

**Formation of Nanofoam carbon and re-emergence of
Superconductivity in compressed CaC₆**

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This file includes Three Tables and nine Figures.

Supplementary Table S1. Structures of stable phases of CaC₆ and carbon foam.

Only the fractional coordinates of symmetry inequivalent atoms are given. For C222 phase, under pressure, it is easy to transform into *Cmmm* symmetry, which is verified by total enthalpy calculation.

Pressure (GPa)	Space group (No.)	Lattice parameters			Atomic fractional coordinates					
		<i>a</i>	<i>b</i>	<i>c</i>	α	β	γ			
0	<i>R-3m</i>	5.1683	5.1683	5.1683	Ca 4 <i>i</i>	0.0000	0.0000	0.0000		
	(164)	49.7	49.7	49.7	C1 6 <i>g</i>	0.1667	0.8333	0.5000		
19	<i>C222</i>	9.1472	3.6947	3.5964	Ca 2 <i>a</i>	0.0000	0.0000	0.0000		
	(29)	90	90	90	C1 4 <i>f</i>	0.4213	0.0000	0.5000		
					C2 8 <i>l</i>	0.1751	0.8082	0.4999		
39	<i>Pmmn</i>	5.8904	2.4893	6.4903	Ca 2 <i>b</i>	0.0000	0.5000	0.7425		
	(59)	90	90	90	C1 2 <i>b</i>	0.0000	0.5000	0.3159		
					C2 2 <i>a</i>	0.0000	0.0000	0.1766		
					C3 4 <i>f</i>	0.1994	0.0000	0.0286		
					C4 4 <i>f</i>	0.2791	0.0000	0.5537		
126	<i>Cmcm</i>	2.3828	16.1113	4.0533	Ca 4 <i>c</i>	0.0000	0.4874	0.2500		
	(63)	90	90	90	C1 8 <i>f</i>	0.0000	0.1044	0.4303		
					C2 8 <i>f</i>	0.0000	0.2705	0.5647		
					C3 8 <i>f</i>	0.0000	0.3551	0.4371		
0	<i>C₆:Cmcm</i>	5.9354	6.3121	2.4675	C1 4 <i>c</i>	0.0000	0.5716	0.2500		
	(63)	90	90	90	C2 8 <i>g</i>	0.2949	0.2163	0.2500		

Supplementary Table S2. The bond lengths and bond angle of carbon nanofoam (C_6 , space group $Cmcm$), $Pm\bar{m}n$, $Cmcm$ phases of compressed CaC_6 compared to experimental data of graphite and diamond.

Structure	Density(g/cm ³)	Bond length (Å)		Bond angle (°)
Graphite	2.27	sp^2-sp^2	1.421	$sp^2-sp^2-sp^2$ 120.00
diamond	3.54	sp^3-sp^3	1.545	$sp^3-sp^3-sp^3$ 109.47
0 GPa				
nanofoam	2.589	sp^2-sp^2	1.410	$sp^2-sp^3-sp^2$ 106.24
C_6 ($Cmcm$)		sp^2-sp^3	1.522	$sp^3-sp^3-sp^3$ 107.55
		sp^3-sp^3	1.529	$sp^3-sp^3-sp^2$ 110.77
				$sp^3-sp^2-sp^2$ 118.93
				$sp^2-sp^2-sp^2$ 120.11
$Pm\bar{m}n$	3.499	sp^2-sp^2	1.470,1.513	$sp^2-sp^3-sp^2$ 101.70,112.70
		sp^2-sp^3	1.582,1.620	$sp^3-sp^3-sp^3$ 107.43
		sp^3-sp^3	1.596	$sp^3-sp^3-sp^2$ 109.14,111.94
				$sp^3-sp^2-sp^2$ 117.00,118.73
				$sp^2-sp^2-sp^2$ 116.46,122.24
39 GPa				
$Pm\bar{m}n$	3.914	sp^2-sp^2	1.426,1.467	$sp^2-sp^3-sp^2$ 101.59,113.92
		sp^2-sp^3	1.520,1.552	$sp^3-sp^3-sp^3$ 108.02
		sp^3-sp^3	1.538	$sp^3-sp^3-sp^2$ 108.69,111.80
				$sp^3-sp^2-sp^2$ 117.07,119.01
			$sp^2-sp^2-sp^2$ 116.05,121.50	
126 GPa				
$Cmcm$	4.788	sp^2-sp^2	1.462	$sp^3-sp^2-sp^2$ 111.58
		sp^2-sp^3	1.461	$sp^2-sp^3-sp^3$ 111.58
			1.458,1.460	$sp^3-sp^3-sp^3$ 107.19, 109.41
		sp^3-sp^3	1.502,1.517	$sp^3-sp^3-sp^3$ 110.78, 111.06
				$sp^2-sp^3-sp^2$ 109.26
			$sp^3-sp^2-sp^3$ 109.26	

¹ Kittel, C. *Introduction to solid state physics* (8th edition, 2004).

Supplementary Table S3. The calculated values of the density of states at the Fermi level $N(0)$, phonon logarithmic average ω_{\log} , electron-phonon interaction λ , and superconducting transition temperature T_c of CaC_6 for the $Pmmn$ structure at different pressures. μ^* is the Coulomb pseudopotential parameter.

Space group	P (GPa)	$N(0)$ (/Ry)	ω_{\log} (K)	λ	T_c ($\mu^*=0.11$)	T_c ($\mu^*=0.115$)	T_c ($\mu^*=0.12$)
<i>Pmmn</i>	39	15.645	642	0.445	3.87	3.48	3.12
	51	15.58	663	0.587	12.7	11.9	11.2
	78	15.378	623	0.638	15.4	14.7	13.9
	117	14.830	728	0.461	5.23	4.74	4.28

Figure captions

Supplementary Fig. S1. Projected density of states of *Pmmn* structure at phase transition pressure. One can see that there is strong hybridization between Ca-*d* electrons and *p* electrons of C3 and C4 (forming sp^2 hybridized zigzag chains along y-axis direction). sp^3 hybridized C1 and C2 have very little contribution to Fermi level because of their saturated chemical bonds (see also Fig. S3).

Supplementary Fig. S2. Projected density of states of *Cmcm* structure at phase transition pressure. Weak hybridization between Ca-*d* electrons and C-*p* electrons is observed at Fermi level. Pressure brings about *d* energy level move to valence bands, which does good to decrease the total energy of system so as to compensate pressure-induced energetic increase, making it be favorable one energetically.

Supplementary Fig. S3. Electronic local functional (ELF) and structural schematic diagram for *Pmmn* and *Cmcm* phases at phase transitional pressure. Electronic local functional (ELF)² for *Pmmn* structure (a), nanofoam schematic diagram constructed by carbon atoms in *Pmmn* phase, centered with Ca ion chains (b) (unit: Å), *Cmcm* structure along (100) direction (c), and ELF for *Cmcm* structure (d) and (e). In (b) and (c), inequivalent carbon atoms depicted using different color balls, and calcium atoms shown using green balls. For *Pmmn* structure at 39 GPa, sp^2 were connected by sp^3 with C-C chemical bonding (length is 1.552 or 1.520 Å). Along y-axis direction, for sp^2 carbons, there are two types of C-C bonding with different lengths in zigzag chain (1.467 and 1.426 Å). While for sp^3 carbons, the bonding length in zigzag chains along y direction is 1.538 Å. For *Cmcm* structure at 126 GPa, the bonding length could be seen in Table S1.

Supplementary Fig. S4. Phonon dispersion of *Pmmn* phase at zero pressure obtained using Phonopy code. Obviously, it is stable dynamically.

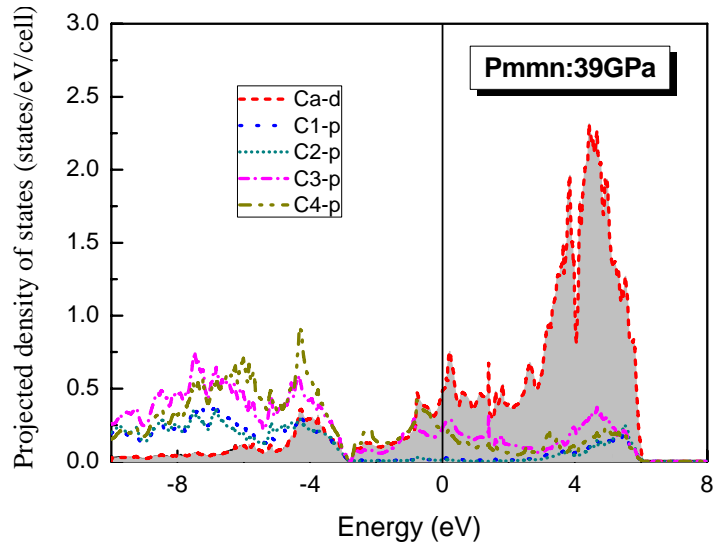
Supplementary Fig. S5. Phonon dispersion of *Cmcm* carbon nanoporous (appeared in phase III) at zero pressure obtained using Phonopy code³.

Supplementary Fig. S6. At zero GPa, band structure of carbon foam (space group *Cmcm*) obtained by removing Ca atoms of phase III (space group, *Pmmn*). The band gap is about 0.49 eV.

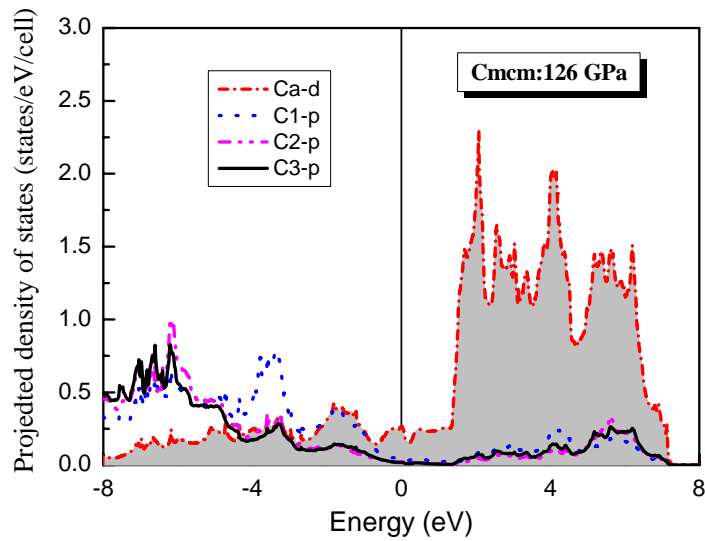
Supplementary Fig S7. Four types of nanoform structures of intercalation compounds under compression. LiC_6 (a), CaC_6 (b), and KC_8 (c and d).

Supplementary Fig S8. Total density of states (TDOS) for *R-3m* at ambient pressure, *C222*, *Pmmn*, and *Cmcm* phases at phase transition pressure. Obviously, pressure leads to more expanded electronic distribution compared to that of *R-3m*, which leads to lower TDOS in *C222* and *Cmcm*. However, *Pmmn* phase has comparative TDOS with that of *R-3m* phase.

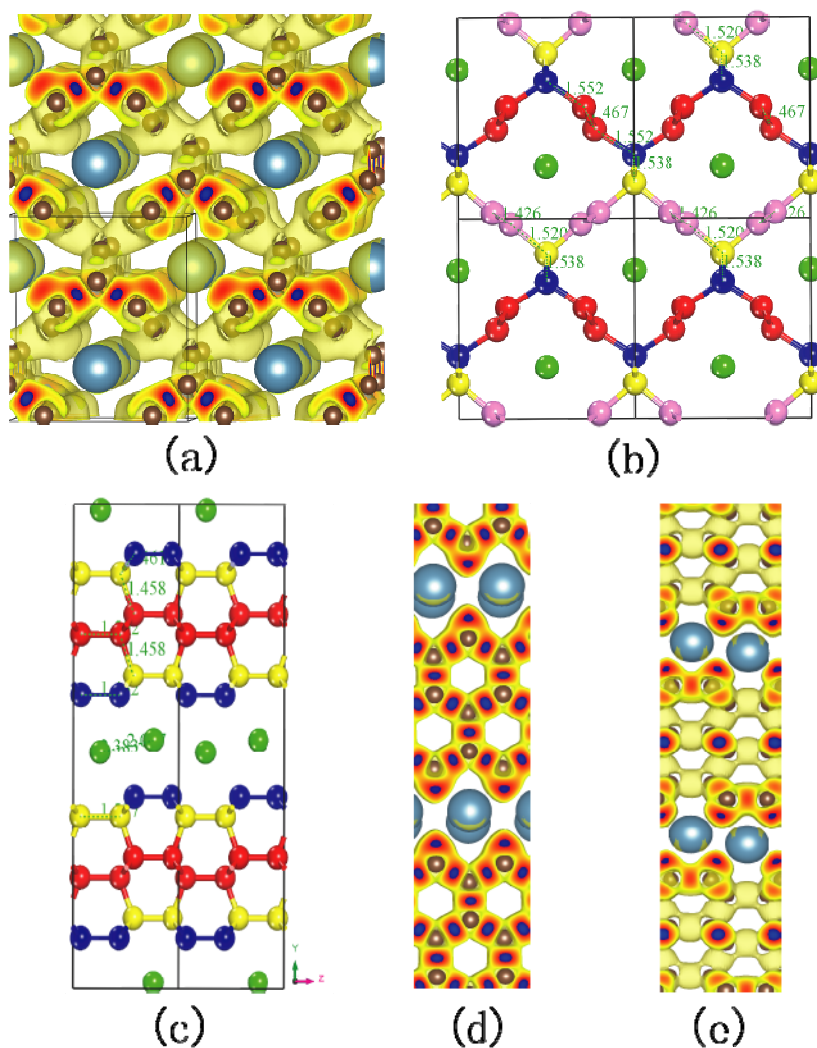
Supplementary Fig S9. Phonon spectrum of three high pressure phases for CaC_6 at cold pressure calculated using Phonopy code. There are soft modes in peculiar direction for *Pmmn* and *Cmcm* phases.



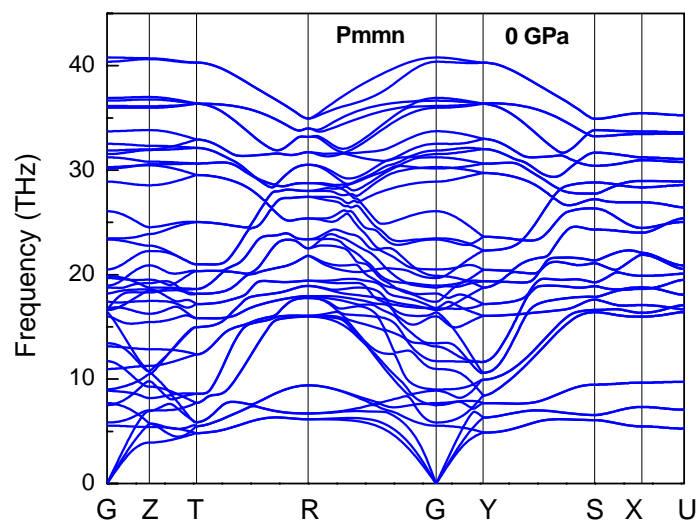
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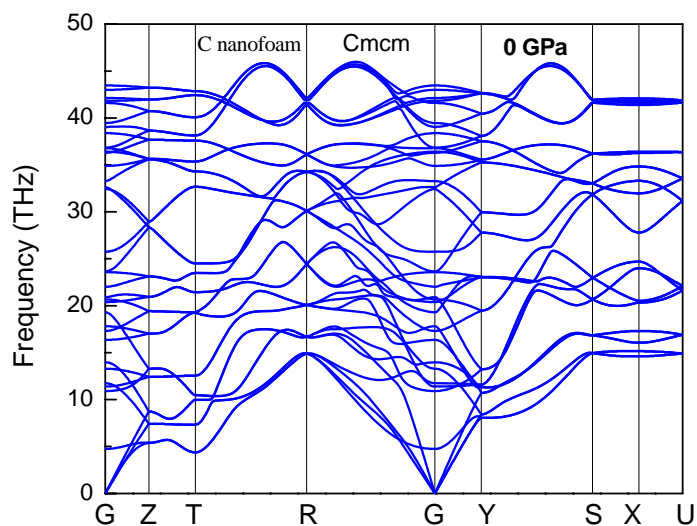
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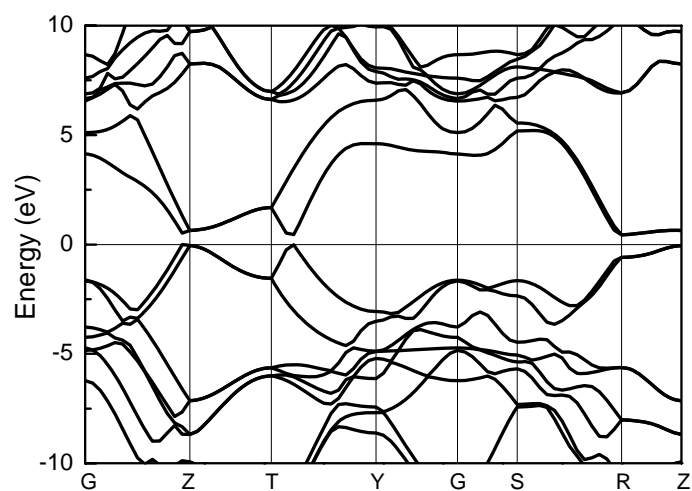
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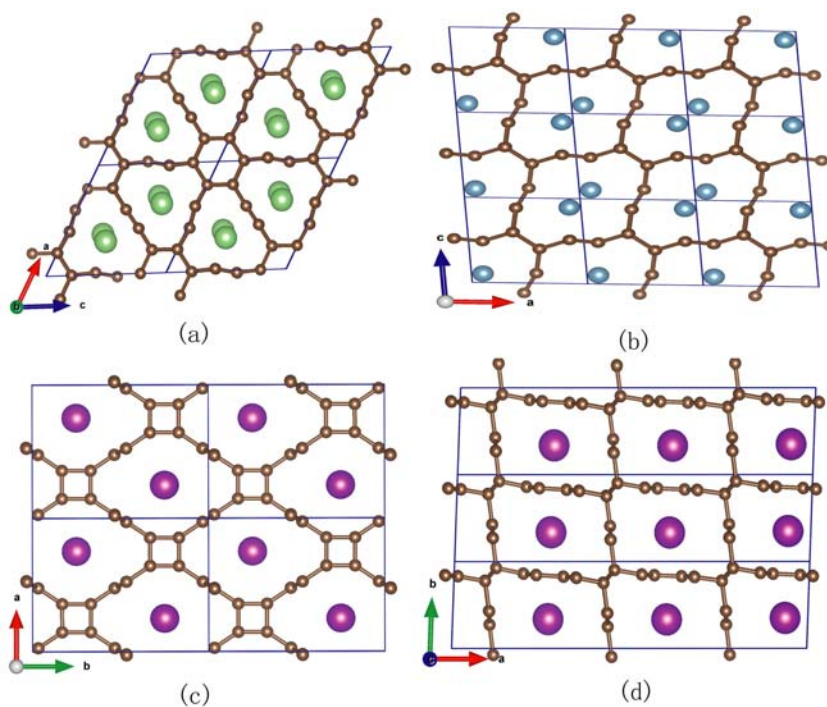
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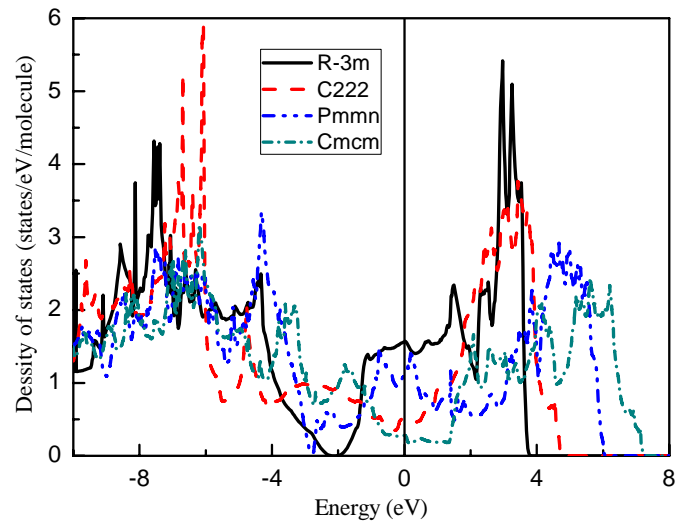
Supplementary Fig. S5. Phonon dispersion of Cmcm carbon nanoporous at zero pressure obtained using Phonopy code³.



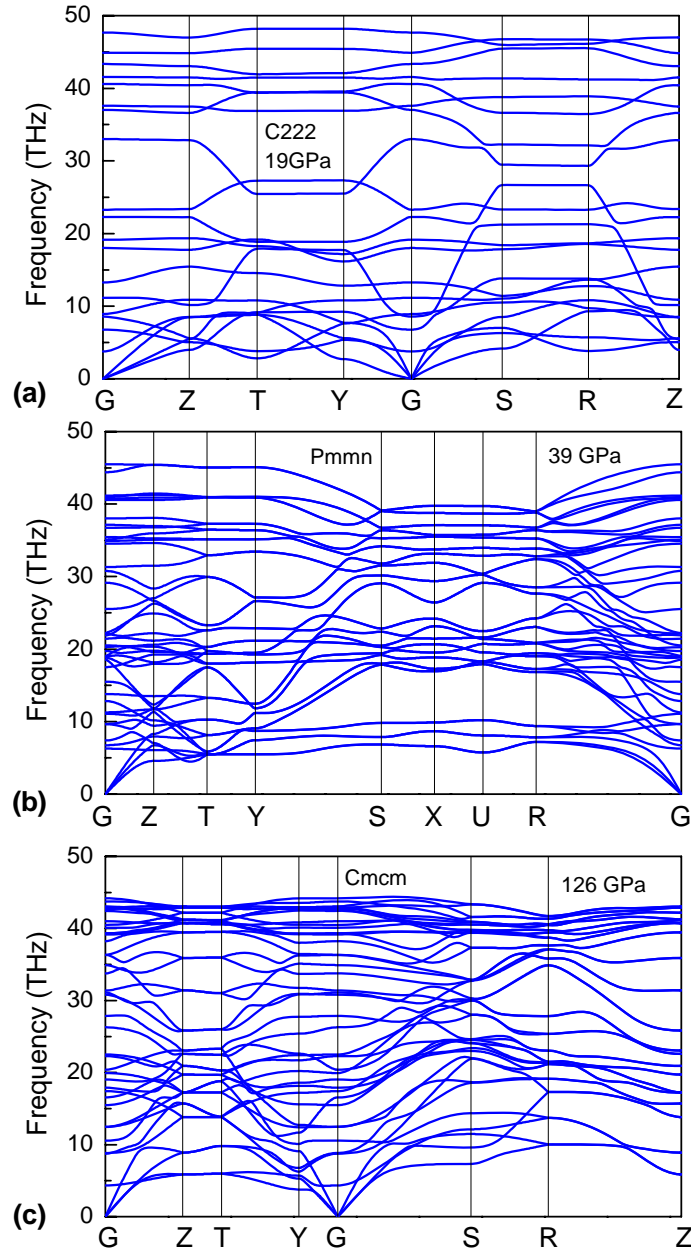
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Reference:

1. Kittel C. *Introduction to solid state physics* (8th edition, 2004).
2. Momma K, and Izumi F (2011) VESTA 3 for three-dimensional visualization of crystal, volumetric and morphology data. *J. Appl. Crystallogr.* **44**, 1272-1276.
3. Togo A, Obe F, and Tanaka I (2008) First-principles calculations of the ferroelastic transition between rutile-type and CaCl_2 -type SiO_2 at high pressures. *Phys. Rev. B* **78**, 134106.