

# 1 Supplemental information

## 1.1 Nonlinear Finite Element Method implementation

The implementation of the finite element method used in the simulations of cellular geometries in this paper is based on the minimization of the strain energy functional. For this purpose we write virtual work equation describing the variation of the work due to the virtual deformation as

$$\delta W = \int_{\omega} \sigma : \delta \mathbf{d} \, dv - \int_{\omega} \mathbf{f} \cdot \delta \mathbf{v} \, dv - \int_{\partial\omega} \mathbf{t} \cdot \delta \mathbf{v} \, da = 0, \quad (1)$$

where  $\sigma$  is Cauchy's stress tensor,  $\mathbf{f}$  and  $\mathbf{t}$  are body forces and tractions respectively,  $\delta \mathbf{d}$  is the virtual rate of deformation tensor and  $\delta \mathbf{v}$  is virtual velocity. We obtain the solution of this problem by linearization and iterative steps with respect to trial deformation solution  $\phi_k$  by use of the Newton-Raphson procedure. The equilibrium equations linearized in the direction of increment  $\mathbf{u}$  in  $\phi_k$  can be then written as:

$$D\delta W(\phi, \delta v)[\mathbf{u}] = \int_{\omega} \delta \mathbf{d} : c : \epsilon \, dv + \int_{\omega} \sigma : [(\nabla \mathbf{u})^T \nabla \delta \mathbf{v}] \, dv + \int_{\omega} \mathbf{f} \cdot \delta \mathbf{v} \, dv + \int_{\partial\omega} \mathbf{t} \cdot \delta \mathbf{v} \, da, \quad (2)$$

where  $c$  is a tangent modulus of elasticity contributing to constitutive component of stress,  $\epsilon$  is a strain tensor. In finite element method this equation is solved by discretization of  $\mathbf{u}$  and  $\delta \mathbf{v}$  with respect to a local support basis for the problem (so called shape functions) which specific form depends on the choice of the finite element discretization. We have used quadrilateral shell elements with extensible director formulation which geometry can be described by the following relations:

$$\mathbf{X}(\xi_i) = \bar{\mathbf{X}}(\xi_\alpha) + \mathbf{D}(\xi_i) \quad (3)$$

$$\bar{\mathbf{X}}(\xi_\alpha) = \sum_{a=1}^n N_a(\xi_\alpha) \bar{\mathbf{X}}_a \quad (4)$$

$$\mathbf{D}(\xi_i) = \sum_{a=1}^n N_a(\xi_\alpha) z_a(\xi_3) \bar{\mathbf{D}}_a \quad (5)$$

$$z_a(\xi_3) = N_+(\xi_3) z_a^+ + N_-(\xi_3) z_a^- \quad (6)$$

$$N_+(\xi_3) = \frac{1}{2}(1 + \xi_3), \quad N_-(\xi_3) = \frac{1}{2}(1 - \xi_3), \quad (7)$$

where Latin and Greek indices are assumed to span from 1 to 3 and from 1 to 2 respectively. The 3D position of the body particle  $\mathbf{X}$  is described by the point on the reference surface  $\bar{\mathbf{X}}$  and the vector  $\mathbf{D}$  called the director. The reference surface and the director are described by the two dimensional shape functions  $N_a$  and nodal values  $\bar{\mathbf{X}}_a$  and  $\bar{\mathbf{D}}_a$ . The function  $z_a$  describes the thickness of the element in terms of the distance from the reference surface to the bottom and top surfaces. The director vector in deformed configuration is not required to be of unit length and takes into account the changes in the thickness.

## 1.2 Material description

To represent a pavement cell wall material we use hyperelastic transversely isotropic model. Hyperelasticity means that the stress arises solely from the deformation (and not how this deformation arises) and that there exist a scalar strain energy function  $W$  from which the stress can be derived at any point. The second Piola-Kirchhoff stress  $\mathbf{S}$  and material elasticity  $\mathcal{C}$  tensors can be then obtained as derivatives of the strain energy with respect to strain tensor  $\mathbf{E}$

$$\mathbf{S} = \frac{\partial W}{\partial \mathbf{E}}, \quad (8)$$

$$\mathcal{C} = \frac{\partial \mathbf{S}}{\partial \mathbf{E}}. \quad (9)$$

The transverse isotropy means that there is a single direction in which the properties of the material are different. This direction in the model is represented by the vector  $\mathbf{a}$  which introduces the dependence of the strain energy on the preferred direction. In our simulations we have assumed that this direction is dynamically controlled by the cell and set it at each step of the simulation to the direction of the maximal principal stress in each element. With the assumptions of transversely isotropic elasticity the strain energy can be expressed as a function of the set of invariants of the right Cauchy-Green deformation tensor  $\mathbf{C}$

$$W = F_1(I_1, I_2, I_3) + F_2(I_4, I_5) + F_3(I_1, I_2, I_3, I_4, I_5), \quad (10)$$

where  $I_1 = \text{tr}\mathbf{C}$ ,  $I_2 = \frac{1}{2}[(\text{tr}\mathbf{C})^2 - \text{tr}\mathbf{C}^2]$ ,  $I_3 = \det\mathbf{C} = J^2$ ,  $I_4 = \mathbf{a} \cdot \mathbf{C} \cdot \mathbf{a} = \lambda^2$ ,  $I_5 = \mathbf{a} \cdot \mathbf{C}^2 \cdot \mathbf{a}$ . The function  $F_1$  describes the behavior of the isotropic matrix part of the material,  $F_2$  corresponds to the response of the fibers and  $F_3$  represents the interaction between fibers and matrix. In our case the fibers represented not only cellulose microfibrils, but also their tethering by xyloglucans and other components of the wall and were modeled in average sense by the function  $F_2$  satisfying relations for  $\lambda^2 = I_4$  being the fiber stretch

$$\begin{aligned} \lambda \frac{\partial F_2}{\partial \lambda} &= 0, \quad \lambda < 0, \\ \lambda \frac{\partial F_2}{\partial \lambda} &= C_3 (\exp(C_4(\lambda - 1)) - 1), \quad \lambda < \lambda^*, \\ \lambda \frac{\partial F_2}{\partial \lambda} &= C_5 \lambda + C_6, \quad \lambda \geq \lambda^*. \end{aligned}$$

The  $\lambda^*$  corresponds to the stretch at which the initial exponential response becomes linear and in the simulations it was assumed  $\lambda^* = 1.02$ . The  $C_3$  is a coefficient of exponential stress,  $C_4$  gives the rate of attaining linear response,  $C_5$  is a modulus of fibers and  $C_6$  is calculated by the constrain of the continuity of a fiber stress at  $\lambda^*$ . As a material of the matrix we have tested Saint Venant–Kirchhoff model with strain energy function given with respect to deviatoric invariants  $\tilde{I}_1, \tilde{I}_2$  by

$$F_1(I_1, I_2, I_3) = 0.5(\lambda + 2\mu) \left(0.5(\tilde{I}_1 - 3)\right)^2 - 0.5\mu(\tilde{I}_2 - 2\tilde{I}_1 + 3) + 0.5\kappa(J - 1)^2, \quad (11)$$

where the initial bulk modulus can be calculated from Lamé parameters  $\kappa = \lambda + 2\mu/3$ , and the Mooney-Rivlin model with analogical function in the form

$$F_1(I_1, I_2, I_3) = C_1(\tilde{I}_1 - 3) + C_2(\tilde{I}_2 - 3) + 0.5\kappa(J - 1)^2. \quad (12)$$

Both of the matrix models gave qualitatively equivalent results thus we presented the results with easier to interpret from material coefficients point of view Saint Venant–Kirchhoff model.