## **Supplementary information**

## **Preparation of Few-Layer Bismuth Selenide by Liquid-Phase-Exfoliation and Its Optical Absorption Properties**

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**Supplementary Figure S1 | Schematic of liquid-phase exfoliation. Bulk Bi<sub>2</sub>Se<sub>3</sub> is sonicated** in solvents to produce few-layer Bi<sub>2</sub>Se<sub>3</sub>. In organic solvent, NMP with an appropriate surface tension plays an important role to exfoliate bulk  $Bi<sub>2</sub>Se<sub>3</sub>$  into stabilized nanosheets against reaggregation. In acetic solution of chitosan, chitosan as a surfactant can be adsorbed on the surface of few-layer  $Bi<sub>2</sub>Se<sub>3</sub>$  to avoid its reaggregation.



**Supplementary Figure S2** | Characterization of as-synthesized bulk Bi<sub>2</sub>Se<sub>3</sub>. (a) XRD patterns of bulk Bi<sub>2</sub>Se<sub>3</sub> prepared at different reaction times. The XRD patterns were readily indexed to rhombohedral  $Bi<sub>2</sub>Se<sub>3</sub>$  (JCPDS no. 89-2008). The strong and sharp peaks of samples indicate that the as-synthesized bulk  $Bi<sub>2</sub>Se<sub>3</sub>$  was crystalline. (b) Energy dispersive X-ray spectrum of as-synthesized bulk  $Bi<sub>2</sub>Se<sub>3</sub>$ . The atomic ratio of Bi and Se is 1:1.58 that is close to the theoretically atomic ratio (1.5) in  $Bi<sub>2</sub>Se<sub>3</sub>$ , which is consistent with the result of XRD. (c, d) AFM image and the corresponding height profiles of as-synthesized bulk  $Bi<sub>2</sub>Se<sub>3</sub>$  with several tens of nanometers in thickness.













Supplementary Figure S3 | Exfoliating as-synthesized bulk Bi<sub>2</sub>Se<sub>3</sub> with different solvents. Ten solvents were used to exfoliate  $Bi_2Se_3$  with same ultrasonic time and concentration. These solvents were NMP, styrene, methanol, acetone, isopropanol, CS-HAc, Pluronic® P-123, poly(sodium 4-styrenesulfonate), sodium cholate and deionized water, respectively (numbering them 1-10 from left to right). Bi<sub>2</sub>Se<sub>3</sub> powder was added into 20 ml of cylindrical vial (1 mg·mL<sup>-1</sup>), then sonicating for 30 h. Photographs show the disperse state of Bi<sub>2</sub>Se<sub>3</sub> in different solvents standing for different days after sonication.  $Bi<sub>2</sub>Se<sub>3</sub>$  can be exfoliated in NMP, methanol, isopropanol, CS-HAc and poly(sodium 4-styrenesulfonate). The colors of NMP and CS-HAc are deepest in all solvents. Therefore, NMP and CS-HAc are used in following experiments.





**Supplementary Figure S4│Effect of ultrasonic time on exfoliation of as-synthesized bulk** 

Bi<sub>2</sub>Se<sub>3</sub>. (a) The photograph of Bi<sub>2</sub>Se<sub>3</sub> with different sonication times in NMP. (b) The photograph of  $Bi<sub>2</sub>Se<sub>3</sub>$  with different sonication times in CS-HAc. (c) The produced few-layer  $Bi<sub>2</sub>Se<sub>3</sub>$  with different sonication times in NMP. (d) The produced few-layer  $Bi<sub>2</sub>Se<sub>3</sub>$  with different sonication times in CS-HAc. The calculated absorption coefficient *α* at 574 nm is  $1378$  mL mg<sup>-1</sup> m<sup>-1</sup>.



Supplementary Figure S5 | Typical Tyndall effect of few-layer Bi<sub>2</sub>Se<sub>3</sub> colloidal suspension.

(a) Few-layer  $Bi<sub>2</sub>Se<sub>3</sub>$  dispersed in NMP. (b) Few-layer  $Bi<sub>2</sub>Se<sub>3</sub>$  dispersed in CS-HAc.



Supplementary Figure S6 | Confirmation of few-layer Bi<sub>2</sub>Se<sub>3</sub> exfoliated in CS-HAc. (a) TEM image of few-layer  $Bi<sub>2</sub>Se<sub>3</sub>$ . (b) SAED pattern of few-layer  $Bi<sub>2</sub>Se<sub>3</sub>$ . (c) HRTEM image of few-layer  $Bi<sub>2</sub>Se<sub>3</sub>$ . (d, e) AFM image and the corresponding height profile of few-layer  $Bi<sub>2</sub>Se<sub>3</sub>$ with a thickness of ca.  $3-4$  nm. (f) XRD pattern of few-layer  $Bi<sub>2</sub>Se<sub>3</sub>$ .



Supplementary Figure S7 | Raman spectra of as-synthesized bulk and few-layer Bi<sub>2</sub>Se<sub>3.</sub> The peak of bulk Bi<sub>2</sub>Se<sub>3</sub> at ~72, ~131, and~174 cm<sup>-1</sup>, are assigned to  $A_{1g}^1$ ,  $E_g^2$  and  $A_{1g}^2$ vibrational modes, respectively. The dashed vertical line indicates red shift of  $A_{1g}^1$  mode in few-layer  $Bi<sub>2</sub>Se<sub>3</sub>$  from bulk  $Bi<sub>2</sub>Se<sub>3</sub>$ .



**Supplementary Figure S8│ TEM and AFM images and the corresponding height**  profiles of few-layer Bi<sub>2</sub>Se<sub>3</sub> in CS-HAc collected with different centrifugal speeds. Three few-layer Bi<sub>2</sub>Se<sub>3</sub> solutions were obtained with the same treatment in Fig. 4. The result is similar to that in NMP.



Supplementary Figure S9 | UV-vis absorption spectra of few-layer Bi<sub>2</sub>Se<sub>3</sub> in CS-HAc **collected with different centrifugal speeds.** In the visible absorption spectra, the absorption band of three samples were 603 (red curve), 487 (blue curve) and 440 nm (green curve) corresponding to 2000, 8000 and 13000 rpm, respectively. The result is similar to that in NMP ( Fig. 4).



**Supplementary Figure S10** Experimental setup of few-layer Bi<sub>2</sub>Se<sub>3</sub>-based Q-switched **EDFL.** The fiber ring laser is schematically described here (Inset: the image of few-layer Bi<sub>2</sub>Se<sub>3</sub>-deposited fiber ferrule). A section of 4.1 m erbium-doped fiber (EDF, Nufern EDFC-980-HP) was pumped by a 200 mW/974 nm laser diode (LD, Bookham Inc.) through a 976/1530 nm wavelength division multiplexer (WDM), for providing the laser gain in 1.5 μm waveband. A 10/90 optical coupler (OC, EPTEK Co. Ltd.) operating in the range of 1520-1570 nm was used to output 10% Q-switched laser. An optical isolator (Thorlabs, IO-H-1550) can block any undesired reflection and also ensure the unidirectional operation. A polarization controller (PC, Thorlabs FPC030) was used to optimize the laser operation. Few-layer  $Bi<sub>2</sub>Se<sub>3</sub>$  was inserted into the laser cavity as a passive Q-switcher. The  $Bi<sub>2</sub>Se<sub>3</sub>$ saturable absorber device was fabricated by optically depositing the few-layer  $Bi<sub>2</sub>Se<sub>3</sub>$  or as-synthesized bulk  $Bi<sub>2</sub>Se<sub>3</sub>$  onto a fiber ferrule. As shown in the inset,  $Bi<sub>2</sub>Se<sub>3</sub>$  can be well deposited on the central core of fiber. The insertion losses of few-layer  $Bi<sub>2</sub>Se<sub>3</sub>$  and as-synthesized bulk  $Bi<sub>2</sub>Se<sub>3</sub>$  devices are 2.5 and 2.4 dB, respectively.



**Supplementary Figure S11│Unstable Q-switching operation with as-synthesized bulk**   $\mathbf{Bi}_2\mathbf{Se}_3$ . Unstable Q-switching operation was observed using bulk  $\mathrm{Bi}_2\mathrm{Se}_3$  at the pump power of 66.7 mW. (a) The optical spectrum of Q-switching; (b) Unstable Q-switched pulsed trains recorded at a 1 min interval; (c) Unstable repetition rate recorded at a 1 min interval; (d) The obtained minimum pulse duration of Q-switching.

When as-synthesized bulk  $Bi<sub>2</sub>Se<sub>3</sub>$  was deposited on a fiber ferrule and then inserted into the laser cavity in Fig. S10, an unstable Q-switching operation was observed. Such laser started the CW lasing at the pump threshold of 9.2 mW, but it transmitted to the Q-switching operation at the higher pump power of 22.1 mW. At the pump power of 66.7 mW, we

recorded the typical characteristics of such Q-switching with bulk  $Bi<sub>2</sub>Se<sub>3</sub>$ , as shown in Fig. S11. The laser spectrum of this Q-switching (Supplementary Fig. S11a) has the central wavelength of 1530.2 nm which is same to that of few-layer  $Bi<sub>2</sub>Se<sub>3</sub>$  (Fig. 6c), due to the similar insertion loss  $(\sim 2.5 \text{ dB})$ . As given in Supplementary Fig. S11b, the oscilloscope traces of Q-switching at the same pump power of 66.7 mW were recorded at a 1 min interval. One can clearly see that the pulse intensity gradually declined. Moreover, the pulsed repetition rate became higher from 7.8 to 10.3 kHz as correspondingly measured by the RF spectrum analyzer. These results indicate that this Q-switching operation using bulk  $Bi<sub>2</sub>Se<sub>3</sub>$  is very unstable, having the strong fluctuations of both the pulse intensity and repetition rate. The minmum pulse duration of such Q-switching is 22.8 μs (Supplementary Fig. S11d) and is much broader than that of few-layer  $Bi_2Se_3$  (4.9 μs, Fig. 6f). More importantly, the Q-switching operation is limited to a small range of pump power (22.1~67.5 mW).



Supplementary Figure S12 | The E<sub>g</sub> for different few-layer Bi<sub>2</sub>Se<sub>3</sub> in NMP collected with **different centrifugal speeds.** (a) Plots of absorption *vs.* wavelength of  $Bi<sub>2</sub>Se<sub>3</sub>$  in the UV-Vis-NIR region. (b) Plot of  $(ahv)^2$  *vs.* hv for different size of  $Bi_2Se_3$ . The values of  $E_g$  were determined by extrapolating the straight portion of the plot to the energy axis in (b).







Supplementary Figure S13 | The band gap  $E_g$  for different few-layer Bi<sub>2</sub>Se<sub>3</sub> in CS-HAc collected with different centrifugal speeds. (a) Plots of absorption *vs.* wavelength of Bi<sub>2</sub>Se<sub>3</sub> in the UV-Vis-NIR region. (b) Plot of  $(ahv)^2$  *vs.* hv for different size of Bi<sub>2</sub>Se<sub>3</sub>. The values of  $E<sub>g</sub>$  were determined by extrapolating the straight portion of the plot to the energy axis in (b).







**Supplementary Fig. S14│Schematic for describing the principle of saturable absorption**  of few-layer Bi<sub>2</sub>Se<sub>3</sub> under the light excitation. Under strong light excitation, the electrons in the valence band become depleted while the finial state in the conduction band is partially occupied, and further excitation from the valence band is blocked and no further absorption is induced, leading to a saturable absorption effect.



**Supplementary Fig. S15│The illustration why the optically saturable absorption of few-layer Bi<sub>2</sub>Se<sub>3</sub> is superior to that of as-synthesized bulk**  $Bi_2Se_3$ **.** Bulk  $Bi_2Se_3$  can be exfoliated to many few-layer  $Bi<sub>2</sub>Se<sub>3</sub>$  sheets, in this process the surfaces/edges can be sharply increased. Under light excitation, the surface electrons of few-layer  $Bi<sub>2</sub>Se<sub>3</sub>$  can be transited more readily, because few-layer Bi<sub>2</sub>Se<sub>3</sub> possesses more metallic surfaces/edges in comparison with the bulk one.