Surface Termination Conversion during $SrTiO₃$ Thin Film Growth Revealed by X-ray Photoelectron Spectroscopy

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Supplementary Notes

Supplementary Note 1

To demonstrate the effect extended defects may have on the determination of the cation ratio using equation 7 from the main text, we analyzed 2 extreme cases of extended defects. For this purpose, we analytically calculated the measured cation intensity ratios of the crystals containing the extended defect and fitted the resulting data points with our model for the perfect crystal. As shown in Supplementary Fig. 1a, we started with a stoichiometric $SrTiO₃$ crystal with SrO termination, which can be described accurately using equation 7 of the main text. For the first case of extended defects which might be induced by non-stoichiometric growth, we inserted one additional layer of SrO below the first unit cell (Supplementary Fig. 1b). This could correspond to a 5 nm SrTiO₃ film with 2 % Sr-excess, if the entire Sr-excess were consumed in the formation of this Ruddelsden-Popper-type defect. Alternatively, this can be regarded as a Ruddelsden-Popper-phase film with the stoichiometry $Sr₁₄T₁₃O₄₀$. We chose to incorporate this defect as close to the surface as possible to demonstrate the most severe effect such extended defects my have on the measured film stoichiometry. Note that a SrO double layer at the surface of the SrTiO₃ crystal is an unrealistic scenario, as it is energetically unfavorable.^{[1](#page-1-0)} In a second step, we inserted further SrO layers into the crystal, yielding a Ruddelsden-Popper-phase film with the stoichiometry $Sr₂TiO₄$ (Supplementary Fig. 1c). This film can hardly be regarded as a non-stoichiometric SrTiO₃ film, but we include it in our analysis to demonstrate the maximum effect extended defects like Ruddelsden-Popper-phases can have. For both cases described above, we calculated the relative cation photoemission intensities layer by layer, in a similar way as described in the main text. For the $Sr_{14}Ti_{13}O_{40}$ film we obtain

$$
I_0^{\text{Sr3d}} = I_0^{\text{Sr3d}} \tag{1}
$$

$$
I_1^{\text{Ti2p}} = I_0^{\text{Ti2p}} \exp(-\frac{a}{2\cos\theta\lambda^{\text{Ti2p}}})\tag{2}
$$

$$
I_2^{\text{Sr3d}} = I_0^{\text{Sr3d}} \exp(-\frac{a}{\cos \theta \lambda^{\text{Sr3d}}})
$$
\n
$$
I_2^{\text{Sr3d}} = I_0^{\text{Sr3d}} \exp(-\frac{a}{\cos \theta \lambda^{\text{Sr3d}}})
$$
\n(3)

$$
I_3^{\text{Sr3d}} = I_0^{\text{Sr3d}} \exp\left(-\frac{3a}{2\cos\theta\lambda^{\text{Sr3d}}}\right) \tag{4}
$$

$$
I_{\text{I}}^{\text{Ti2p}} = I_{\text{I}}^{\text{Ti2p}} \exp\left(-\frac{2a}{2\cos\theta\lambda^{\text{Sr3d}}}\right) \tag{5}
$$

$$
I_4^{\text{112p}} = I_0^{\text{112p}} \exp\left(-\frac{2a}{\cos\theta \lambda^{\text{112p}}}\right) \tag{5}
$$

$$
I_5^{\text{Sr3d}} = I_0^{\text{Sr3d}} \exp\left(-\frac{5a}{2\cos\theta \lambda^{\text{Sr3d}}}\right) \tag{6}
$$

$$
\cdots \tag{7}
$$

for the intensity I_k^{Sr3d} or I_k^{Ti2p} k_k^{112p} from layer *k*. Summing over all layers of a 5 nm film, we arrive at the angle-dependent intensities indicated in Supplementary Fig. 1d.

For $Sr₂TiO₄$ we obtain

$$
I_k^{\text{Sr3d},\text{layer 1}} = I_0^{\text{Sr3d}}(-\xi_k^{\text{/sr3d}})
$$
\n(8)

$$
I_k^{\text{Sr3d,layer 2}} = I_0^{\text{Sr3d}} \exp(-\frac{a'/3}{\cos \theta \lambda^{\text{Sr3d}}}) \exp(-\xi_k^{\text{/sr3d}})
$$
\n(9)

$$
I_k^{\text{Ti2p,layer 3}} = I_0^{\text{Ti2p}} \exp(-\frac{2a'/3}{\cos \theta \lambda^{\text{Ti2p}}}) \exp(-\xi_k^{\text{Ti2p}})
$$
\n(10)

for the intensity $I_k^{\text{Sr3d,layer 1}}$ $\binom{1}{k}$, $\binom{1}{k}$ \int_k Sr3d,layer 2 or $I_k^{\text{Ti2p,layer 3}}$ $\frac{f_{112p,12y}}{k}$ from the first and second SrO layer and the TiO₂ layer from unit cell *k*, respectively, if we define the Sr₂TiO₄ out-of-plane lattice parameter $a' \approx 6$ Å and $\xi_k^{T12p} = \frac{a'k}{\cos \theta \lambda^{T12p}}$. Summing over all layers of a 5 nm film, we reach at the angle-dependent intensities indicated in Sup

We now treat these analytically determined cation intensity ratios for the defective $SrTiO₃$ films like the measured XPS data in the main text and fit them using equation 7. For these fits, which are also shown in Supplementary Fig. 1d, the fraction of SrO-termination and the atomic concentration of Sr, $N_0^{S_r}$, were allowed to vary like described in the main text. For each case, we end up with the best fit with 100 % Sr-termination. The stoichiometry is summarized in Supplementary Table 1 along with the nominal stoichiometry for each film. For both highly-nonstoichiometric films with extended defects, we achieve a reasonable fit with a relative error of below 3.5 %. This indicates that extended defects can in fact lead to an overestimation or underestimation of the true cation ratio of a given film, as was indicated in the main text. At the same time, these worst case scenarios yield a comparably small relative error, especially when compared to the experimental uncertainty or the enormous effect pure termination effects can have (up to 17% , compare [2](#page-1-1) and main text). We therefore deem our analytical model as a very useful tool for the simultaneous determination of termination and stoichiometry.

References

- 1. Lee, J. H. *et al.* Dynamic layer rearrangement during growth of layered oxide films by molecular beam epitaxy. *Nat. Mater.* 13, 879–883 (2014).
- 2. Zhang, L., Wett, D., Szargan, R. & Chassé, T. Determination of ZnO(0001) surface termination by x-ray photoelectron spectroscopy at photoemission angles of 0 and 70. *Surf. Interface Anal.* 36, 1479–1483 (2004).

Supplementary Tables

Supplementary Table 1. Nominal film stoichiometry and best fit obtained with equation 7 from the main text.

Supplementary Figures

Supplementary Figure 1. Termination and stoichiometry determination for SrTiO₃ thin films with severe Sr excess. (a) Schematic illustration of a SrTiO₃ single crystals with SrO termination. (b) Schematic illustration of a $Sr₁₄T₁₃O₄₀$ single crystals with SrO termination and a double SrO layer burried one unit cell underneath the surface. (c) Schematic illustration of a Sr2TiO⁴ single crystals with a two-layer-SrO termination. (d) calculated cation intensity ratio (data points, compare Supplementary Note 1) as a function of the photoemission angle and fits (solid lines) according to equation 7 of the main text.