

## Supplementary

### **Unexpected observation of spatially separated Kondo scattering and ferromagnetism in Ta alloyed anatase TiO<sub>2</sub> thin films**

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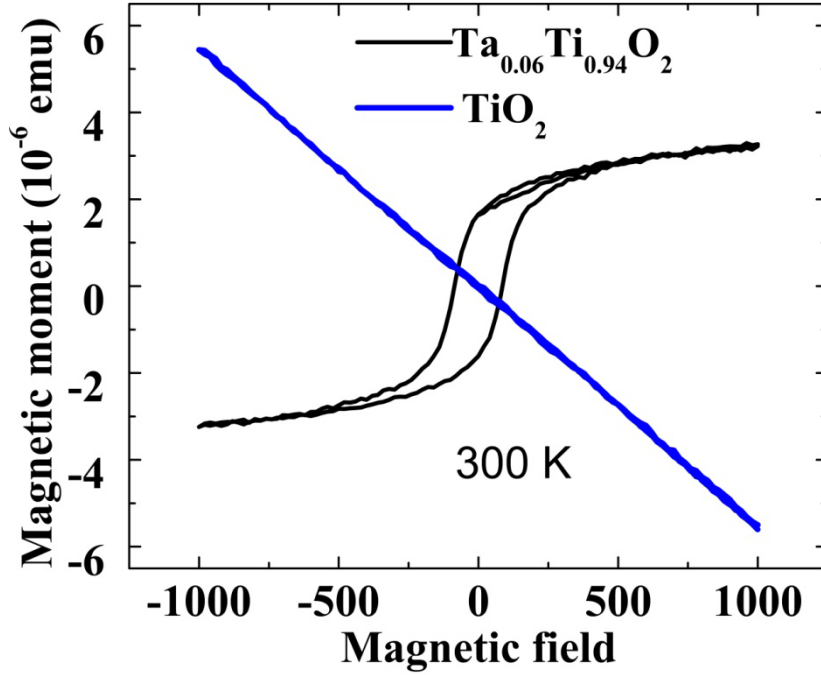
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## Additional results from Ta:TiO<sub>2</sub> samples

### 1. Magnetic moment vs. applied magnetic field characteristics for undoped TiO<sub>2</sub> and Ta:TiO<sub>2</sub> samples



**Figure S1:** The above figure shows the typical room temperature in-plane M-H hysteresis loop of Ta:TiO<sub>2</sub> (black line) and undoped TiO<sub>2</sub> (blue line) grown at the same conditions on SrTiO<sub>3</sub> (001) substrate. From these results, it is clear that there is no signature of ferromagnetism either in the substrate or in undoped TiO<sub>2</sub>.

### 2. Temperature dependent thermopower behavior

An independent validation for Kondo scattering is the observation of a maximum in the Seebeck coefficient close to the Kondo temperature. Figure S2 shows the Seebeck coefficient  $S$  as a function of temperature for 200 nm sample. The comparison between resistivity and Seebeck coefficient versus temperature indicates that as long as the resistivity is metallic (positive temperature coefficient), the Seebeck coefficient is linear in  $T$ , as predicted by Mott law:

$$S = -\frac{\pi^2}{3e} K^2 T \left. \frac{\partial \ln(\sigma(E))}{\partial E} \right|_{E=E_F} \approx -\frac{\pi^2}{3e} K^2 T \left( \left. \frac{\partial \ln(n)}{\partial E} \right|_{E=E_F} + \left. \frac{\partial \ln(\tau)}{\partial E} \right|_{E=E_F} \right) \quad (1)$$

which yields for a degenerate semiconductor in 3D:

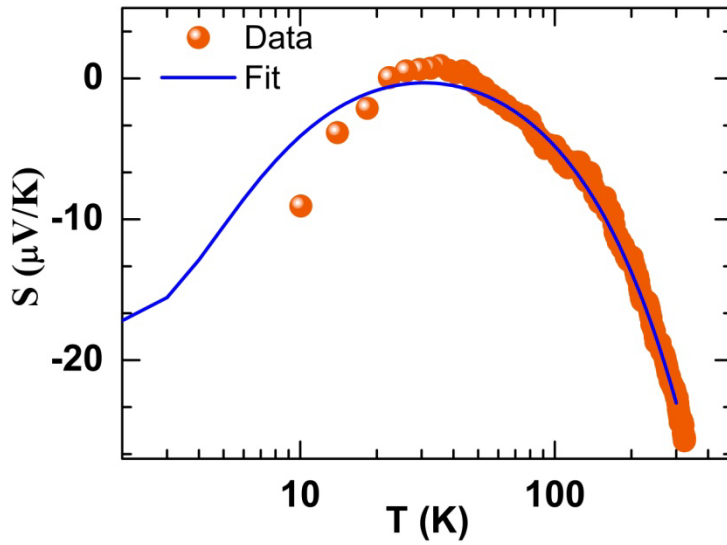
$$S = -\left(\frac{3}{2} + \alpha\right) \frac{8\pi^{8/3} K^2}{3^{5/3} h^2 e} m_{\text{eff}} \frac{T}{n_{3D}^{2/3}} \quad (2)$$

where  $\alpha$  describes the functional dependence of the scattering time on the energy  $\tau \sim E^\alpha$  and its value depends on the dominant scattering mechanism. In many cases  $\alpha \approx 0$  is assumed for simplicity and for lack of a solid theoretical back up,  $\alpha \approx -0.5$  for scattering with acoustic phonons and for most other scattering mechanisms,  $\alpha$  is between 0 and -1. As the carrier concentration measured by Hall effect is virtually constant as a function of temperature in our films, eq. (2) well describes the linear temperature dependence of the Seebeck curves above  $\sim 50$  K, assuming effective masses around  $2m_0$ , where  $m_0$  is the bare electron mass. At lower temperatures there is an onset of resistivity upturn and correspondingly also the Seebeck coefficient changes its temperature derivative sign, consistently with other literature data<sup>1</sup>.

At lower temperatures there is an onset of resistivity upturn and correspondingly also the Seebeck coefficient changes its temperature derivative sign. In the hypothesis of Kondo regime, we can assume that below  $T_K$  a Kondo resonance yields an extra term in the Seebeck coefficient, derived from the Mott eqn. (1)<sup>2,3</sup>. This term can be phenomenologically expressed as :

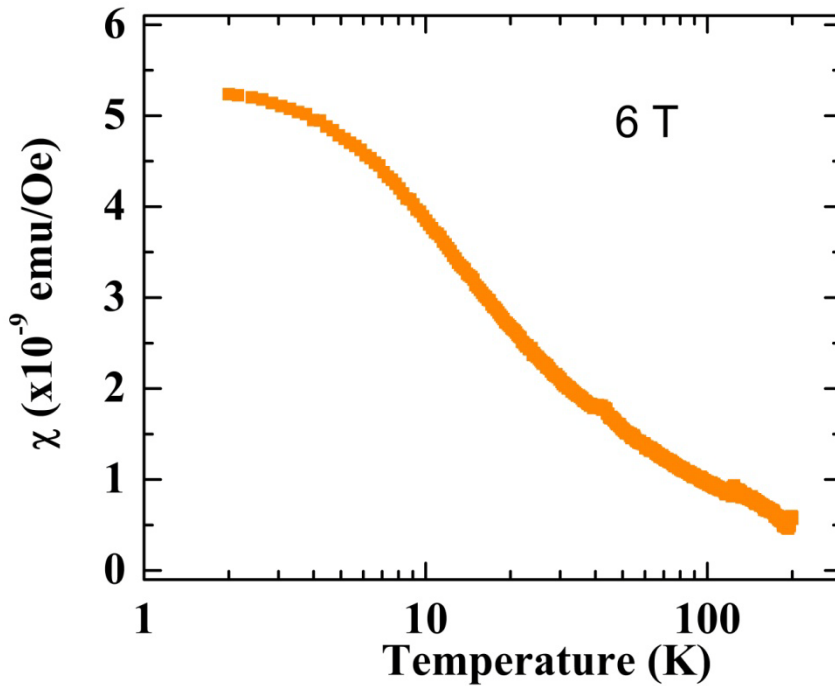
$$S_k = -\frac{aT}{T^2 + b^2} \quad (3)$$

The coefficient  $a$  should be related to the energy position of the Kondo resonance ( $T_K$ ) and to the strength of the magnetic scattering processes, while the coefficient  $b$  should be related to the energy width of the Kondo resonance. We can fit the experimental data in Fig. S2 as a sum of Mott linear Seebeck term (eq. (2)) plus a Kondo Seebeck term (eq. (3)). Associating fitting coefficients  $a$  (0.00015 ) and  $b$  (5) to meaningful physical parameters is not easy as the spin, charge and orbitals are coupled strongly in oxide systems.



**Figure S2:** The thermo power as a function of temperature along with the theoretical fit.

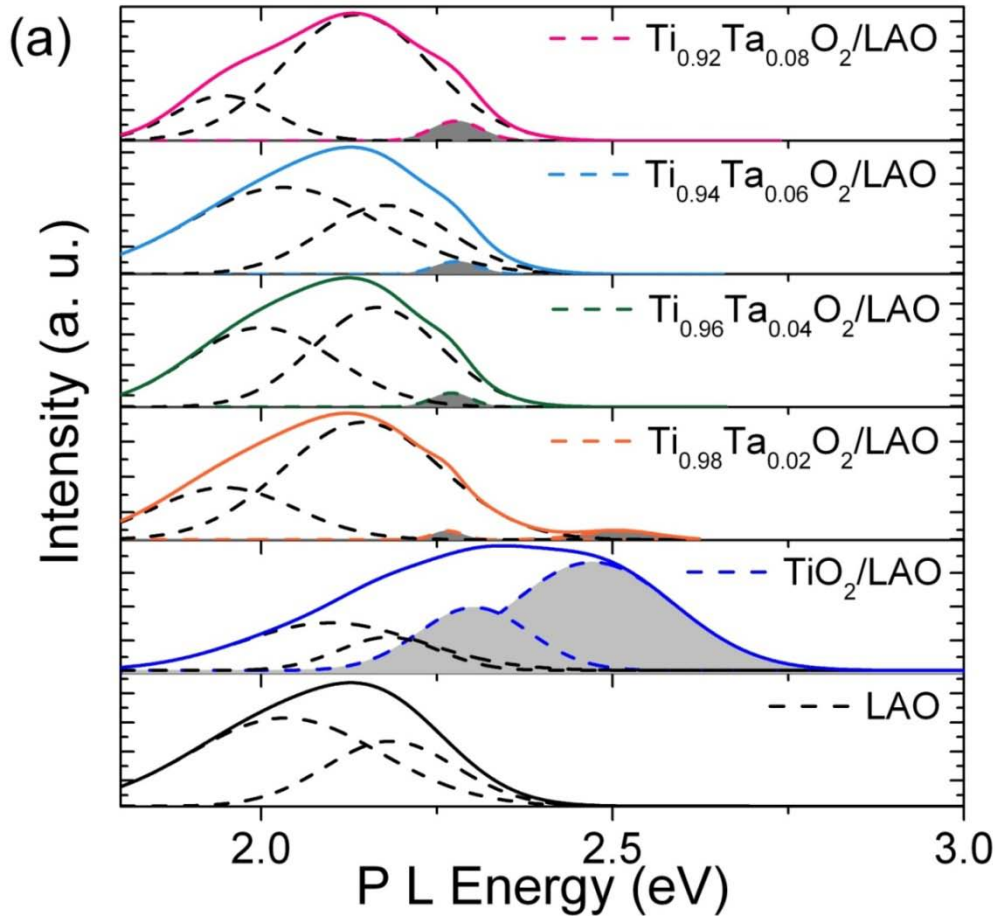
### 3. Temperature dependence of the magnetic susceptibility



**Figure S3:** The temperature dependence of susceptibility ( $\chi$ ) for 200 nm film is shown in the temperature range from 2 to 200 K under an external magnetic field of 6 Tesla. Kondo behavior is identified from the saturation of magnetic susceptibility. However, the high temperature susceptibility could be described by Curie-Weiss  $\chi = C/T - \Theta_p$  where  $C$  is the Curie constant and  $\Theta_p$  is the paramagnetic Curie temperature. Figure S3 shows the

susceptibility below  $\sim 6$  K tends to saturate which deviates from the Curie-Weiss law. This deviation is similar to that reported by other groups<sup>4</sup> which support our Kondo model .

#### 4. Photoluminescence spectra for different concentrations of Ta in $\text{TiO}_2$



**Figure S4:** The normalised PL spectra obtained at 20 K for  $\text{Ti}_{1-x}\text{Ta}_x\text{O}_2$  thin films for different Ta concentrations (0 to 8%) at the low energy range (1.8 eV to 3.0 eV). We observed the Stoke-shifted self-trapped exciton peak centered at 2.35 eV for the undoped  $\text{TiO}_2$  film. This is in accordance with the previous PL studies done on  $\text{TiO}_2$ . The origin of this broad peak has been assigned to oxygen vacancies in the  $\text{TiO}_2$  crystal which form defect levels deep inside the bandgap. The broad PL peak for undoped  $\text{TiO}_2$  thin films was deconvoluted into four Gaussian peaks centered at around 2.05, 2.2, 2.3 and 2.5 eV. Next, the PL spectra from a bare LAO substrate (treated under the same conditions under which the films were grown) were obtained. It was found out that LAO itself has a broad PL peak, which in turn is deconvoluted into two Gaussian components. These two Gaussians matched

pretty closely with the 2.05 and 2.2 eV peaks for the undoped TiO<sub>2</sub> (Figure S4). Hence, we can safely assign these two lower energy peaks to the oxygen vacancies in the LAO substrates. As a result, only the other two peaks at 2.3 and 2.5 eV can then be argued to be coming from TiO<sub>2</sub>. We further observed that the Gaussians at 2.3 and 2.5 eV (due to TiO<sub>2</sub>) were sharply quenched on the slightest Ta incorporation for the Ti<sub>1-x</sub>Ta<sub>x</sub>O<sub>2</sub> thin films. It may be concluded here that the signal obtained for the Ti<sub>1-x</sub>Ta<sub>x</sub>O<sub>2</sub> thin films were actually due to the LAO substrate. Ta incorporation in TiO<sub>2</sub> enhances free donor electrons in the system. Consequently, the self compensating nature of the crystal acts to enhance the formation energy for any electron donor defects in the system. This explains the quenching of the PL peaks (arising due to oxygen vacancies) in Ti<sub>1-x</sub>Ta<sub>x</sub>O<sub>2</sub> thin films. A new peak is found to increase in intensity with Ta concentration at 2.27 eV which may be related to cationic defects which tend to increase with Ta concentration.

## References:

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