**Supplementary Table 1.** Parameters for estimating minimum thermal conductivity in MoS<sub>2</sub> crystal. The three polarizations (TL1, TL2, and TA) are named following the isoenergydecomposition process described in Ref. 1.



### **Supplementary Note 1**

### **Minimum thermal conductivity calculation**

We follow the method described in Ref. [1](#page-7-0) to calculate the minimum thermal conductivity in the *c*-axis and *ab*-plane for MoS<sub>2</sub> crystal. To calculate minimum thermal conductivity in the *c*axis, we used the following equation S(1), equation (S7) from the Supporting Information of Ref. [1,](#page-7-0)

$$
\kappa_{\min-c,Layered} = \sum_{pol} \frac{v_c}{6\pi v_{ab}^2} \frac{k_B^3}{\hbar^2} \left[ \int_0^{x_{D,c}} \frac{T^2 x^3 e^x}{(e^x - 1)^2} dx + \frac{\theta_{D,c}^3}{T} \int_{x_{D,c}}^{x_{D,ab}} \frac{e^x}{(e^x - 1)^2} \left( \frac{\theta_{D,ab}^2 - (Tx)^2}{\theta_{D,ab}^2 - \theta_{D,c}^2} \right)^{\frac{3}{2}} dx \right], \quad S(1)
$$

where  $x = \hbar \omega / k_B T$ . We took the lattice constants (a, b, c), density, and all the elastic constants of MoS<sub>2</sub> crystal from Ref[.2.](#page-7-1) The Debye temperature of the *c*-axis  $\theta_{D,c}$  was calculated using

$$
\theta_{D,c} = \frac{\hbar v_c k_{\text{max},c}}{k_B},
$$
 S(2)

where *h* is the Planck constant,  $k<sub>B</sub>$  is the Boltzmann constant,  $k<sub>max,c</sub>$  is the maximum wave vector in the chain axis direction. The Debye temperatures in the other two directions were calculated similarly. The group velocities in each polarization were estimated from the elastic constants (C<sub>11</sub>=238 GPa, C<sub>33</sub>= 52 GPa, C<sub>44</sub>= 19 GPa, C<sub>66</sub>= 146 GPa, C<sub>13</sub>=13 GPa) and the density  $\rho = 5.06$  g/cm<sup>3</sup> of MoS<sub>2</sub> crystal using  $\sqrt{C_{ij}/\rho}$ . Supplementary Table 1 shows all the parameters we used in this calculation.

Similarly, we used the following equation  $S(1)$ , equation  $(S9)$  from the Supporting Information of Ref. [1](#page-7-0) to calculate the minimum thermal conductivity in the *ab*-plane,

$$
\kappa_{\min-ab,Layered} = \sum_{pol} \frac{1}{6\pi v_c} \frac{k_{B}^{3}}{\hbar^{2}} \left[ \int_{0}^{x_{D,c}} \frac{T^{2}x^{3}e^{x}}{(e^{x}-1)^{2}} dx + \frac{3}{2} \theta_{D,c} \int_{x_{D,c}}^{x_{D,ab}} \frac{Tx^{2}e^{x}}{(e^{x}-1)^{2}} \left( \frac{\theta_{D,ab}^{2} - (Tx)^{2}}{\theta_{D,ab}^{2} - \theta_{D,c}^{2}} \right)^{\frac{1}{2}} dx \right] - \frac{1}{2} \frac{\theta_{D,c}^{3}}{T} \int_{x_{D,c}}^{x_{D,ab}} \frac{e^{x}}{(e^{x}-1)^{2}} \left( \frac{\theta_{D,ab}^{2} - (Tx)^{2}}{\theta_{D,ab}^{2} - \theta_{D,c}^{2}} \right) dx
$$

The minimum thermal conductivity in the *c*-axis for a  $MoS<sub>2</sub>$  crystal is 0.16 W m<sup>-1</sup> K<sup>-1</sup>, with contributions from TL1, TL2, and TA modes 0.004 W m<sup>-1</sup> K<sup>-1</sup>, 0.151 W m<sup>-1</sup> K<sup>-1</sup>, and 0.006 W  $m^{-1} K^{-1}$ , respectively. The minimum thermal conductivity in the *ab*-plane for a MoS<sub>2</sub> crystal is 1.0 W m<sup>-1</sup> K<sup>-1</sup>, with contributions from TL1, TL2, and TA modes 0.5 W m<sup>-1</sup> K<sup>-1</sup>, 0.1 W m<sup>-1</sup> K<sup>-1</sup>, and 0.4 W  $m^{-1} K^{-1}$ , respectively.

## **Supplementary Methods**

### *Thermal Conductivity Measurements*

TDTR was used to measure the thermal conductivity of bulk and thin film intercalated MoS2. In TDTR, a mode-locked Ti:sapphire laser produces a train of pulses at a repetition rate of 80 MHz. The laser beam is split into pump and probe beams. A mechanical delay stage is used to change the optical path difference between the pump and probe beams before they are focused onto the sample surface through an objective lens. The pump beam is modulated at frequency *f* so that the thermoreflectance change at the sample surface can be detected by the probe beam through lock-in detection. The ratio of the in-phase and out-of-phase signal from the lock-in is then fit to a thermal diffusion model. Further details on TDTR can be found elsewhere.<sup>[3,](#page-7-2) [4](#page-7-3)</sup>

Prior to the TDTR measurements, metal thin films (Al or NbV) were deposited on the samples by magnetron sputtering. Samples were exposed to air for only three to five seconds before the process chamber was pumped down. We measured through-plane thermal conductivity of MoS<sub>2</sub> at  $f= 9.8$  MHz, with a  $1/e^2$  radius of the focused laser beams  $w_0 = 11.7 \,\text{\mu m}$ . In the fitting, the heat capacities of Al, NbV, and sapphire are adopted from literature values.<sup>[5,](#page-7-4) [6](#page-7-5)</sup> The thickness of Al thin film was obtained from picosecond acoustics using a longitudinal speed of sound 6.42 nm ps<sup>-1</sup>.<sup>[7](#page-7-6)</sup> The thickness of the MoS<sub>2</sub> thin films was measured by atomic force microscopy (AFM). The thermal conductivity of the Al thin film was calculated using the Wiedemann-Franz law and the electrical resistance of the same transducer layer deposited on a  $\approx$ 315 nm SiO<sub>2</sub> on Si reference sample. The thermal conductivity of sapphire substrate, 32 Wm  ${}^{1}K^{-1}$ , was measured using the same Al transducer. The thermal conductivity of MoS<sub>2</sub> thin films reported in this work is the apparent (or effective) thermal conductivity of the thin film,

including the two interfacial thermal resistances between  $MoS<sub>2</sub>$  and the neighboring materials, besides the intrinsic thermal resistance of the film. The sensitivity of the TDTR data to the inplane thermal conductivity of  $Li_xMoS_2$  thin film is small, which makes this in-plane thermal conductivity challenging to measure.

We measured through-plane thermal conductivity of bulk  $MoS<sub>2</sub>$  at  $f= 9.8$  MHz with  $w_0$ = 11.7 µm. The thermal conductivity of bulk MoS<sub>2</sub> and the interfacial thermal conductance between Al and  $MoS<sub>2</sub>$  were fitted. The in-plane thermal conductivity of bulk  $MoS<sub>2</sub>$  was measured using the beam-offset TDTR method as detailed in ref. [8](#page-7-7) , at *f*= 1.1 MHz with  $w_0 \approx 2.7 \text{ }\mu\text{m}$ . NbV transducer was used in this measurement, whose thermal properties were characterizedin ref. <sup>9</sup>. The total uncertainties of the measured thermal conductivity are calculated by taking into account the systematic errors that propagate from uncertainties in the film thickness, laser spot size, and thermal properties of the transducer film and substrate. We have tried to use a 65 nm-thick NbV thin film as the metal transducer to measure in-plane thermal conductivity  $\Lambda$  of MoS<sub>2</sub> thin film by TDTR method. However, due to the relatively low thermal conductance of the film, i.e.,  $\Lambda d$ , where  $d$  is the thickness of the thin film the in-plane heat flow in metal transducer and the Sapphire substrate instead of the  $MoS<sub>2</sub>$  thin film dominates the lateral heat flow which lead to a low sensitivity to the thin film in plane thermal conductivity in TDTR measurement. The thermal conductivity measurements are performed at different locations on our samples to confirm the homogeneous distribution of lithium.

#### *Elastic constants measurements*

The elastic constants of the  $Li<sub>x</sub>MoS<sub>2</sub>$  thin films were measured using pump-probe techniques. The polycrystalline  $MoS<sub>2</sub>$  thin film with vertically-aligned basal planes is transverse

isotropic, which has five effective independent averaged elastic constants:  $C'_{11}$ ,  $C'_{12}$ ,  $C'_{13}$ ,  $C'_{33}$ , and  $C'_{44}$ .  $C'_{33}$  is calculated from  $\rho v_L^2$ , where the longitudinal speed of sound  $v_L = 2d/t$ , where *t* was measured using the longitudinal acoustic echoes in a TDTR measurement. Film thickness *d* was measured by atomic force microscope (AFM). The film density was calculated by  $\rho = nM/(dN_A)$ , where *M* is the molecular weight of Li<sub>x</sub>MoS<sub>2</sub> and *N<sub>A</sub>* is the Avogadro constant. We used Rutherford backscattering spectrometry (RBS) measurement to determine the areal atomic density  $n_1$  (atoms m<sup>-2</sup>) of Mo and S atoms in  $Li_xMoS_2$  thin films and calculated the total areal atomic density *n* (atoms  $m^{-2}$ ) of  $Li<sub>x</sub>MoS<sub>2</sub>$ .

The  $C'_{44}$  of Li<sub>x</sub>MoS<sub>2</sub> thin films is determined from the velocity of surface acoustic wave (SAW)  $v_{SAW}$ , which is generated and detected in the pump-probe experiment using an elastomeric phase-shift mask made of PDMS.[10](#page-7-9) The thickness of the Al transducer used in the SAW measurements is ≈85 nm. We calculated  $v_{SAW}$  of the tri-layer structure  $(A1/MoS<sub>2</sub>/Sapphire)$  numerically using a Green's function method.<sup>[11](#page-7-10)</sup> The inputs to the calculation are elastic constants, density and thickness of each layer. The Al layer ( $C_{11} = 108 \text{ GPa}$ ,  $C_{12}$  = 63 GPa,  $C_{44}$  = 28.3 GPa,  $\rho$  = 2.7 g cm<sup>-3</sup>)<sup>[10](#page-7-9)</sup> is a (111) textured film and the sapphire substrate ( $C_{11} = 497 \text{ GPa}$ ,  $C_{12} = 163 \text{ GPa}$  $C_{12} = 163 \text{ GPa}$  $C_{12} = 163 \text{ GPa}$ ,  $C_{33} = 501 \text{ GPa}$ ,  $C_{44} = 147 \text{ GPa}$ ,  $\rho = 3.95 \text{ g cm}^{-3}$ )<sup>12</sup> has its basal plane facing up. Under the assumption that the  $c$ -axis of  $MoS<sub>2</sub>$  was randomly oriented in plane and the grains were much smaller than spot size  $(\sim 10 \mu m)$ , we calculated the effective elastic constants of the transverse isotropic  $MoS<sub>2</sub>$  film by averaging from the elastic constants of bulk crystal<sup>[13](#page-7-12)</sup> using derivation from Ref. [14.](#page-7-13) For our typical sample structure Al(100nm)/MoS<sub>2</sub>(200nm)/Sapphire, the sensitivity of  $v_{SAW}$  to  $C'_{33}$  and  $C'_{44}$  are 0.19 and 0.12,

respectively, while the sensitivity to  $C'_{11}$ ,  $C'_{12}$ ,  $C'_{13}$  are 0.02, -0.001 and 0.01, respectively. The MoS2 film is modeled using the calculated  $C'_{11}$ ,  $C'_{12}$ ,  $C'_{13}$ , and measured  $C'_{33}$ .  $C'_{44}$  was adjusted in the calculation to match the calculated  $v_{SAW}$  to the measurement data. The error bars, approximately 20%, is calculated by taking into account the experimental errors and the systematic errors that propagate from uncertainties in the Al film thickness,  $Li<sub>x</sub>MoS<sub>2</sub>$  film density, and the input elastic constants.

Similarly, the elastic constants of bulk  $Li_xMoS_2(C_{33})$  were calculated from  $\rho v_L^2$ , where  $ρ$  is calculated based on literature values of bulk MoS<sub>2</sub> samples (5.06 g/cm<sup>3</sup>) and the degree of lithiation x,  $\rho = 5.06(160 + 7x)/160$ . We deposited ≈10 nm NbV on the bulk Li<sub>x</sub>MoS<sub>2</sub> samples and used picosecond interferometry<sup>[15](#page-7-14)</sup> to determine the longitudinal speed of sound  $v<sub>L</sub>$ . In the picosecond interferometry, the Brillouin scattering frequency *f<sup>B</sup>* is related to the longitudinal speed of sound *v* by  $f_B = 2nv_L/\lambda$ , where *n* is the index of refraction of the sample and  $\lambda$  is the laser wavelength. We used the literature value of  $n \approx 4.7$  at  $\lambda$ =785 nm in this calculation.<sup>[16](#page-7-15)</sup> This measurement of picosecond interferometry uses the same experiment setup as TDTR.

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