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# Supporting Information

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## **Tailored Near-Infrared Photoemission in Fluoride Perovskites through Activator Aggregation and Super-Exchange between Divalent Manganese Ions**

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#### Supporting Information

#### **Tailored near-infrared photoemission in fluoride perovskites through activator**

**aggregation and super-exchange between divalent manganese ions**

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#### **Supplementary Methods**

**Chemicals and materials.** Reagents:  $Yb_2O_3(99.998\%)$ ,  $C_4H_6MgO_4.4H_2O(99.98\%)$ ,

 $C_4H_6MnO_4\cdot 4H_2O$  (99.99%),  $C_4H_6ZnO_4\cdot 2H_2O$  (99.99%),  $C_4H_6CdO_4\cdot 2H_2O$  (99.99%), KF (99.9%), NaF

(99.9%), CsF (99.9%), RbF (99.9%), HNO<sub>3</sub> (analytical grade) and oleic acid (90%). Yb<sub>2</sub>O<sub>3</sub> was

supplied by Alfa Aesar Reagent Company, other chemicals were supplied by Aladdin Industrial

Corporation. All of the chemicals were used as received without further purification.

**Synthesis of KMgF<sub>3</sub></sub>:Yb<sup>3+</sup>/Mn<sup>2+</sup>.** A series of KMg<sub>(1-*x*)F<sub>3</sub>: Mn<sup>2+</sup><sub>*x*</sub> (0.01 ≤ *x* ≤ 1.0) and KMg<sub>(1-*x*</sub>-</sub>  $y_y$ F<sub>3</sub>:Yb<sup>3+</sup><sub>y</sub>/Mn<sup>2+</sup><sub>x</sub> (0.01≤*x*≤0.40; *y* = 0.005) samples were synthesised using a facile hydrothermal method. The typical synthesis procedures used in this study were as follows.

Firstly, a Yb(NO<sub>3</sub>)<sub>3</sub> solution (0.02 mol/L) was obtained by dissolving Yb<sub>2</sub>O<sub>3</sub> in hot HNO<sub>3</sub> solution. Then, 12 mmol of KF was completely dissolved in 10 mL of distilled water, and 15 mL of an aqueous solution containing stoichiometric amounts of  $C_4H_6MgO_4.4H_2O$ ,  $C_4H_6MnO_4.4H_2O$  and  $Yb(NO<sub>3</sub>)<sub>3</sub>$  were dropped into the KF solution under vigorous stirring. After stirring for 20 minutes, the resulting transparent solution was transferred to a 50-mL Teflon-lined autoclave, which was held at 220 °C for 24 h. After reaction, the system was naturally cooled down to room temperature, and the products were collected and centrifuged several times with distilled water. Finally, absolute ethanol was used to remove any soluble residues, and the products were dried at 60  $^{\circ}$ C for 10 h.

**Synthesis of**  $AMnF_3$  **(** $A = K$ **, Na and Cs). Here, 15 mL of ethanol and 5 mL of oleic acid were mixed** under magnetic stirring to form a homogeneous solution. Then, 5 mL of an aqueous solution containing 4 mmol of  $C_4H_6MnO_4.4H_2O$  were added to the solution under vigorous stirring. Finally, 5 mL of an aqueous solution containing 12 mmol of KF (or NaF or CsF) were added to the complex, still stirring. The mixture was agitated for 30 min and transferred to a 50-mL autoclave. The autoclave was then sealed, and the mixture hydrothermally treated at 220 °C for 24 h. After the reaction, the system was naturally cooled back to room temperature. The obtained product was centrifuged, washed and dried as above.

Synthesis of  $AB_{(1-x-y)}F_3$ :  $Yb^{3+y}/Mn^{2+x}$  (x = 0.05 and 0.20; y = 0.005)(A = K, Rb and Cs; B = Mg, Zn **and Cd).** Again, 15 mL of ethanol and 5 mL of oleic acid were mixed and magnetically stirred to form a homogeneous solution. Then, 5 mL of an aqueous solution containing stoichiometric amounts of  $C_4H_6MgO_4 \cdot 4H_2O$ ,  $C_4H_6MnO_4 \cdot 4H_2O$  and  $Yb(NO_3)$ , were added to the solution under vigorous stirring. Finally, 5 mL of an aqueous solution containing 12 mmol of KF was added to the complex. The mixture was agitated for 30 min and transferred to a 50-mL autoclave. The autoclave was then sealed,

and the mixture hydrothermally treated at 220 °C for 24 h. After the reaction was completed, the system was naturally cooled down to room temperature The obtained product was centrifuged, washed and dried as above.

**Theoretical calculations.** In the CASTEP code, the wave functions of valence electrons were expanded in an ultrasoft pseudopotential plane-wave basis with the specific cut-off energy E*cut*. The exchange-correlation functional employed was the Perdew, Burke and Ernzerhof (PBE) implementation of the spin-polarized Generalized Gradient Approximation (GGA).<sup>[1,2]</sup> It is reported that the GGA would result in more exact data than the Local Density Approximation (LDA) for spin polarized system.<sup>[1-3]</sup> Six different  $2 \times 2 \times 4$  supercells (80 atoms) with two Mg<sup>2+</sup> ions substituted by two  $Mn^{2+}$  ions according to the ICSD card No. 94089 structure model were used as initial models. The initial supercell parameters were  $a = b = 7.9794$  Å,  $c = 15.9588$  Å. Each geometry optimization by Broyden-Fletcher-Goldfarb -Shannon (BFGS) alogrithm<sup>[4]</sup> was performed to the model. Both E<sub>cut</sub> and k-point mesh were tested and were set to 310 eV and a  $3 \times 3 \times 3$  Monkhorst-Pack grid, which were sufficient for energy convergence. The structures were considered to be converged when the force on each ion was less than ∼0.05 eV/Å, with a maximum displacement of ∼0.002 Å and a convergence in the total energy of about  $2 \times 10^{-6}$  eV/atom. To understand the coupling interaction between Mn<sup>2+</sup> ions, the electronic spin alignment for  $Mn^{2+}$  ions in the KMgF<sub>3</sub> host were set to be FM and AFM. That is, possible FM coupling and AFM coupling for  $Mn^{2+}$  aggregates were adopted in the geometry optimization. The energy difference of the two coupling modes for each model were also calculated  $(\Delta E = E_{\text{AFM}} - E_{\text{FM}})$ .

#### **Supplementary Tables and Table captions:**



**Table S1.** Initial lattice constants for a  $2 \times 2 \times 4$  supercell of KMgF<sub>3</sub>.

**Table S2.** Optimized cell parameters and forming energy (*E*) of six possible substitution geometry modes (denoted as M1, M2, M3, M4, M5 and M6) for a  $2 \times 2 \times 4$  supercell of KMgF<sub>3</sub>:Mn<sup>2+</sup> with two  $Mg^{2+}$  ions substituted by two antiferromagnetic coupling (AFM)  $Mn^{2+}$  ions.

Parameters	M1	M <sub>2</sub>	M <sub>3</sub>	M <sub>4</sub>	M <sub>5</sub>	M6
a(A)	8.2874	8.2865	8.2867	8.2901	8.2853	8.2856
b(A)	8.2874	8.286548	8.284929	8.290169	8.2853	8.2856
c(A)	16.5722	16.5661	16.5629	16.5561	16.5676	16.5563
$\alpha$ (degree)	89.9999	89.9999	90.0000	90.0000	90.0000	90.0000
$\beta$ (degree)	90.0000	90.0000	90.0000	90.0000	90.0002	89.9999
$\gamma$ (degree)	89.9999	90.0000	89.9999	89.9999	89.9998	90.0000
$V(\AA^3)$	1138.2076	1137.5432	1137.1298	1137.8466	1137.3173	1136.6289
$E_{\text{AFM}}$ (eV/mol)	-59330.9750	-59330.9671	-59330.9896	-59330.9701	-59330.9837	-59330.9744
$\Delta E_a$ (eV/mol)	0.0000	0.0079	$-0.0146$	0.0049	$-0.0087$	0.0006
Two shortest Mn-	4.1437	8.2865	5.8596	9.2686	7.1755	10.1474
Mn distances $(\AA)$	8.2874	8.2865	8.2867	8.2901	8.2853	8.2856

 $\Delta E_a = E_{AFM}(Mi) - E_{AFM}(M1)$  (i = 1, 2, 3, 4, 5 and 6).

**Table S3.** Optimized cell parameters and forming energy (*E*) of six possible substitution geometry modes (denoted as M1, M2, M3, M4, M5 and M6) for a  $2 \times 2 \times 4$  supercell of KMgF<sub>3</sub>:Mn<sup>2+</sup> with two  $Mg^{2+}$  ions substituted by two ferromagnetic coupling (FM)  $Mn^{2+}$  ions.

Parameters	M1	M <sub>2</sub>	M <sub>3</sub>	M <sub>4</sub>	M <sub>5</sub>	M6
a(A)	8.2892	8.2867	8.2873	8.2895	8.2864	8.2903
b(A)	8.2892	8.28678	8.2859	8.2912	8.2864	8.2903
c(A)	16.5712	16.5677	16.5625	16.5582	16.5656	16.5557
$\alpha$ (degree)	90.0000	90.0000	90.0000	90.0000	89.9999	90.0000
$\beta$ (degree)	90.0000	90.0000	90.0000	90.0000	90.0000	90.0000
$\gamma$ (degree)	90.0000	90.0000	89.9999	90.0000	90.0000	90.0000
$V(A^3)$	1138.6469	1137.72135	1137.3273	1137.7748	1137.4788	1138.8743
$E_{FM}$ (eV/mol)	-59330.9353	-59330.9644	-59330.9888	-59330.9709	-59330.9833	-59330.9649
$\Delta E_b$ (eV/mol)	0.0000	$-0.0291$	$-0.0535$	$-0.0356$	$-0.0480$	$-0.0296$
Two shortest Mn-	4.1446	8.2867	5.8573	9.2585	7.1752	10.1433
Mn distances $(\AA)$	8.2892	8.2867	8.2873	8.2895	8.2864	8.2903

 $\Delta E_b = E_{\text{FM}}(\text{Mi}) - E_{\text{FM}}(\text{M1})$  (i = 1, 2, 3, 4, 5 and 6).

Table S4. Forming energy difference between  $E_{AFM}$  and  $E_{FM}$  of the six possible substitution geometry modes (denoted as M1, M2, M3, M4, M5 and M6) for a  $2 \times 2 \times 4$  supercell of KMgF<sub>3</sub>:Mn<sup>2+</sup> with two  $Mg^{2+}$  ions substituted by two  $Mn^{2+}$  ions

Parameters	M1	M <sub>2</sub>	M3	M4	M <sub>5</sub>	M6
$E_{\rm AFM}$ (eV/mol)			-59330.9750 -59330.9671 -59330.9896 -59330.9701 -59330.9837			-59330 9744
$E_{\text{FM}}$ (eV/mol)		-59330.9353 -59330.9644 -59330.9888		-59330.9709 -59330.9833		-59330.9649
$\Delta E$ (eV/mol)	$-0.0397$	$-0.0027$	$-0.0008$	0.0008	$-0.0004$	$-0.0095$

*∆E*= *E<sup>A</sup>*FM- *E*FM.

#### **Supplementary Figures and Figure captions:**



Figure S1. Six possible substitution geometry models (denoted as M1, M2, M3, M4, M5 and M6) for a  $2 \times 2 \times 4$  supercell of KMgF<sub>3</sub>:Mn<sup>2+</sup> with two Mg<sup>2+</sup> ions substituted by two Mn<sup>2+</sup> ions. The green ball, red ball, indigo **+** and yellow **+** denote as Mg, Mn, F and K ions, respectively.



**Figure S2. (a)** XANES spectra of  $KMg_{(1-x)}F_3: Mn^{2+}(0.01 \le x \le 0.20)$  (fluorescence mode). **(b)** XANES spectra of  $KMg_{(1-x)}F_3: Mn^{2+}(x=0.40 \text{ and } 1.0)$  and Mn foil (transmission mode). The spectra of  $KMg_{(1-x)}$  $x_1F_3$ :Mn<sup>2+</sup><sub>x</sub> (0.01≤ *x* ≤1.0) are mutually quite similar, but distinctly different from the reference spectrum of the Mn foil, confirming the presence of  $Mn^{2+}$  dopants in KMgF<sub>3</sub> host, rather than secondary phases. For all  $Mn^{2+}$ -containing perovskite samples, the characteristic peak in the first derivative of the edge spectra was observed at  $\sim 6549$  eV, and a very weak pre-edge feature was consistently observed at ~6539 eV (**arrow in Fig. S2**), indicating the valence state of Mn in this system is +2.



**Figure S3.** X-ray powder diffraction patterns of as-recieved  $KMnF_3$ ,  $NaMnF_3$  and  $CsMnF_3$ nanocrystals. Joint Committee on Powder Diffraction Standards (JCPDS) files number 17-0116, 18- 1224 and  $75-2034$  are included for comparison. The compound  $KMnF<sub>3</sub>$  exhibits a typical cubic perovskite structure with Pm-3m space group (No. 221) and the lattice constant  $a = 4.189$  Å. The compound  $NaMnF<sub>3</sub>$  is an orthorhombic perovskite with Pnma space group (No. 62) and lattice constants  $a = 5.748$  Å,  $b = 8.004$  Å and  $c = 5.551$  Å. The compound CsMnF<sub>3</sub> is a hexagonal perovskite structure with P63/mmc space group (No. 194) and lattice constants  $a = b = 6.225$  Å and  $c = 5.551$  Å.



**Figure S4.** Excitation spectra of samples  $KMnF_3(a)$ ,  $NaMnF_3(b)$  and  $CsMnF_3(c)$  at 10 and 300 K. The six typical excitation peaks between 300 and 570 nm can be assigned to the electronic transitions of Mn<sup>2+</sup> from the ground state <sup>6</sup> $A_1(S)$  to the excited states <sup>4</sup> $T_1(P)$ , <sup>4</sup> $E(D)$ , <sup>4</sup> $T_2(D)$ , [<sup>4</sup> $A_1(G)$ , <sup>4</sup> $E(G)$ ], <sup>4</sup> $T_2(G)$ and  ${}^4T_1(G)$ .



Figure S5. Temperature-dependent UC emission spectra and integrated intensity of KMg<sub>(1-x-</sub>  $y_yF_3: Yb^{3+}y_yMn^{2+}x$  (*x* = 0.01; *y* = 0.005) (**a** and **b**) and KMg<sub>(1-*x*-*y*)</sub>F<sub>3</sub>:Yb<sup>3+</sup><sub>y</sub>,Mn<sup>2+</sup><sub>*x*</sub> (*x* = 0.20; *y* = 0.005) upon excitation with a 976- nm LD at a power density of 10  $Wcm<sup>-2</sup>$ .



**Figure S6. P**lots of the upconversion emission intensity versus 976 nm laser pumping power for the two emission bands (VIS/NIR) of Mn<sup>2+</sup> in KMg<sub>(1-*x*-*y*)</sub>F<sub>3</sub>:Yb<sup>3+</sup><sub>*y*</sub>,Mn<sup>2+</sup><sub>*x*</sub> (0.01≤ *x* ≤ 0.20; *y* = 0.005). (**a**) (*x*)  $y = 0.01$ ;  $y = 0.005$ ), (**b**) ( $x = 0.025$ ;  $y = 0.005$ ), (**c**) ( $x = 0.05$ ;  $y = 0.005$ ), and (**d**) ( $x = 0.20$ ;  $y = 0.005$ ).



**Figure S7.** Room-temperature Stokes emission spectra of  $KMg_{(1-x-y)}F_3$ :  $Yb^{3+}y$ ,  $Mn^{2+}x$  (0.01≤  $x \le 0.40$ ; *y*  $= 0.005$ ) upon 396 nm light excitation. (a) Mn<sup>2+</sup>-emission in the range of 550-850 nm and Yb<sup>3+</sup>emission in the range of 850-1200 nm (**b**). It can be observed that the Stokes emission spectra of KMg<sub>(1-*x*-*y*)F<sub>3</sub>:Yb<sup>3+</sup><sub>*y*</sub>,Mn<sup>2+</sup><sub>*x*</sub> (0.01≤ *x* ≤ 0.40; *y* = 0.005) are similar to those of KMg<sub>(1-*x*)F<sub>3</sub></sub>: Mn<sup>2+</sup><sub>*x*</sub> (0.01≤ *x*)</sub>  $\leq$  0.40) (Fig.2a) upon 396 nm light excitation. The bands centered at 595 and 760 nm can be ascribed to the <sup>4</sup>T<sub>1</sub>(<sup>4</sup>G)→<sup>6</sup>A<sub>1</sub>(<sup>6</sup>S) transitions of isolated Mn<sup>2+</sup> ions and <sup>6</sup>A<sub>1</sub>(<sup>6</sup>S)<sup>4</sup>T<sub>1</sub>(<sup>4</sup>G)→<sup>6</sup>A<sub>1</sub>(<sup>6</sup>S)<sup>6</sup>A<sub>1</sub>(<sup>6</sup>S) transitions of  $Mn^{2+}$ - $Mn^{2+}$  dimers, respectively. Additionally, the Yb<sup>3+</sup>-emission centered at 972 nm  $({}^{2}F_{5/2} \rightarrow {}^{2}F_{7/2}$  ) is also observed in Yb<sup>3+</sup>-Mn<sup>2+</sup> codoped KMgF<sub>3</sub> samples through Mn<sup>2+</sup> excitation, indicating that  $Mn^{2+}$  and  $Yb^{3+}$  formed some unique emission centers in this system since  $Mn^{2+}$  and  $Yb^{3+}$  ions have no spectral overlap.



**Figure S8.** X-ray powder diffraction patterns of  $AB_{(1-x-y)}F_3$ :  $Yb^{3+}/Mn^{2+}$   $(x = 0.05$  and 0.20;  $y = 0.005$ )  $(A = K, Rb$  and Cs;  $B = Mg$ , Zn and Cd) and corresponding reference data (JCPDS or inorganic crystal structure database, ICSD).  $Yb^{3+}/Mn^{2+}$  codoped (a)  $KMgF_3$ , (b)  $KZnF_3$ , (c)  $KCdF_3$ , (c)  $RbCdF_3$ and (c) CsCdF<sub>3</sub>.



**Figure S9.** Room-temperature upconversion emission spectra of  $AB_{(1-x-y)}F_3: Yb^{3+}y/Mn^{2+}x$  ( $x = 0.05$ ,  $y =$ 0.005)  $(A = K, Rb$  and Cs;  $B = Mg$ , Zn and Cd) upon excitation with a 976-nm laser diode at a power density of 10 Wcm<sup>-2</sup>.

#### **Supplementary References**

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