

## Supplementary Materials for **Lowering coefficient of friction in Cu alloys with stable gradient nanostructures**

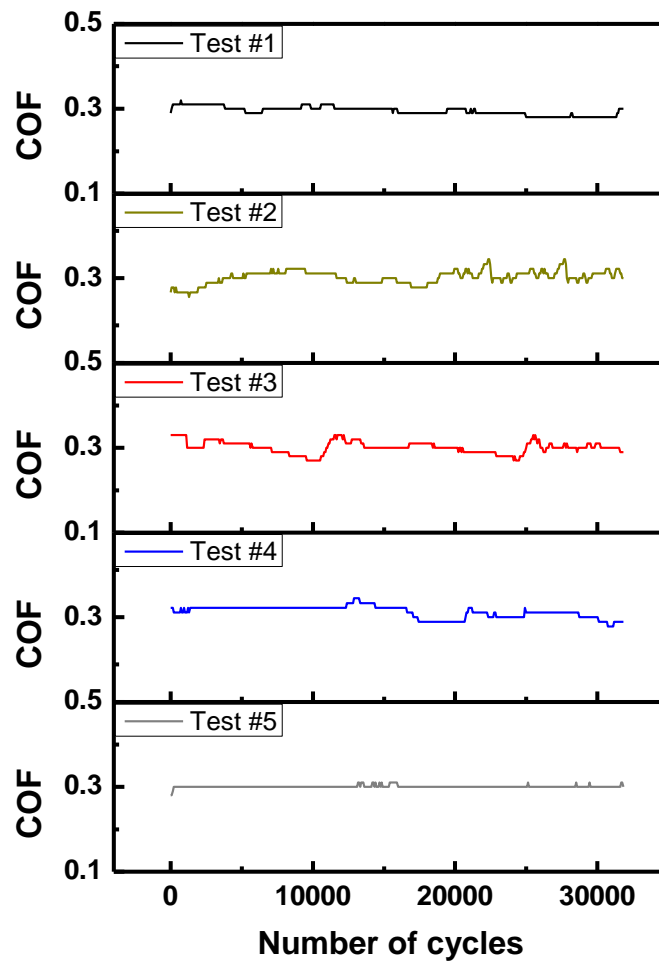
Xiang Chen, Zhong Han, Xiuyan Li, K. Lu

Published 9 December 2016, *Sci. Adv.* **2**, e1601942 (2016)

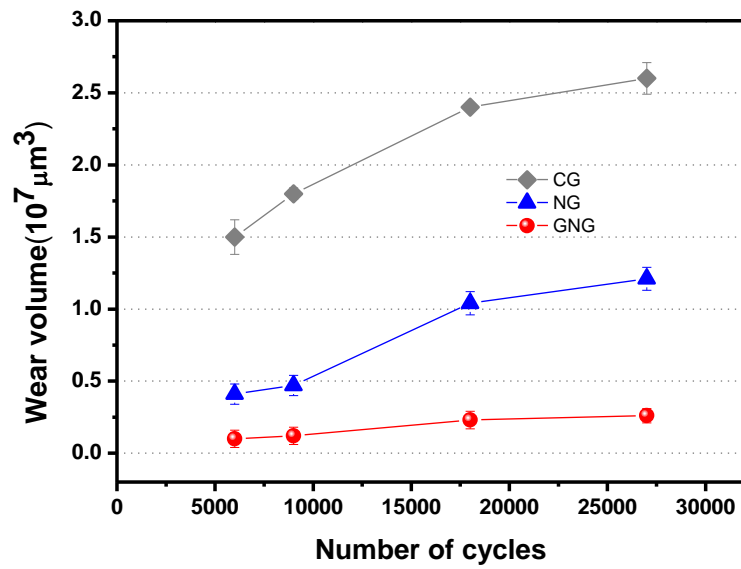
DOI: 10.1126/sciadv.1601942

### **This PDF file includes:**

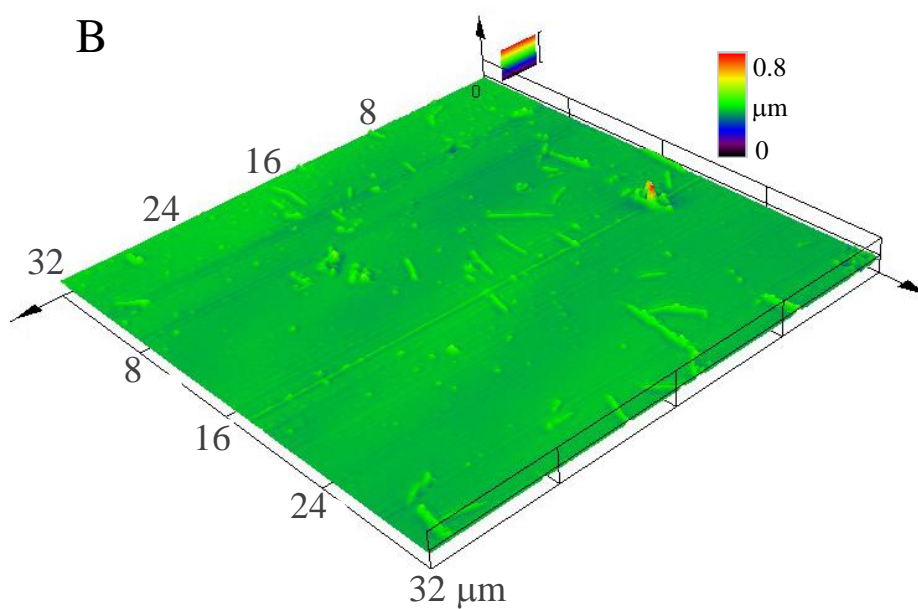
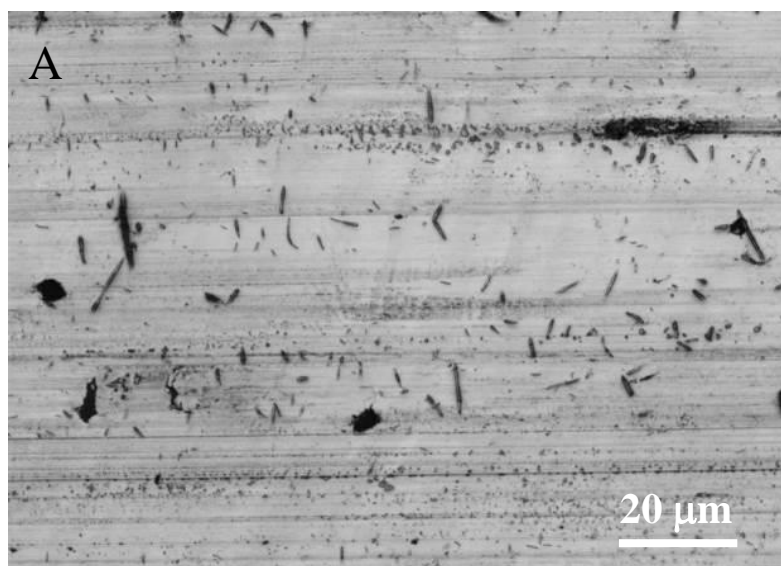
- fig. S1. Measurement repeatability of COF.
- fig. S2. Measurement results of wear rates.
- fig. S3. Surface profiles and morphology of the CG sample under low-load single sliding.
- fig. S4. Effect of Ag addition on COF reduction—Measurement results in pure Cu samples.
- fig. S5. Counter surface analysis.
- fig. S6. COF measurement on the NG, GNG, and CG samples subsequently using exactly the same contact surface of a WC-Co ball.
- fig. S7. Stability of the subsurface microstructure in the GNG samples against sliding.
- fig. S8. Chemical analysis of the topmost NG surface layer.
- fig. S9. Subsurface microstructures in the CG Cu under sliding in the steady state.
- table S1. Surface roughness change after dry sliding for 18,000 cycles.



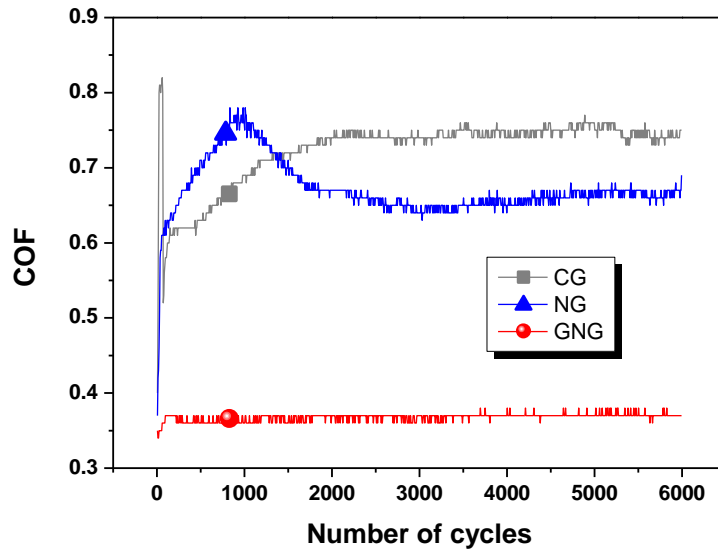
**fig. S1. Measurement repeatability of COF.** Measurement results of COFs with sliding cycles (more than 30000 cycles) in five tests on different GNG Cu samples using WC-Co balls under a load of 50 N, a slide stroke of 1 mm and a velocity of 10 mm/s.



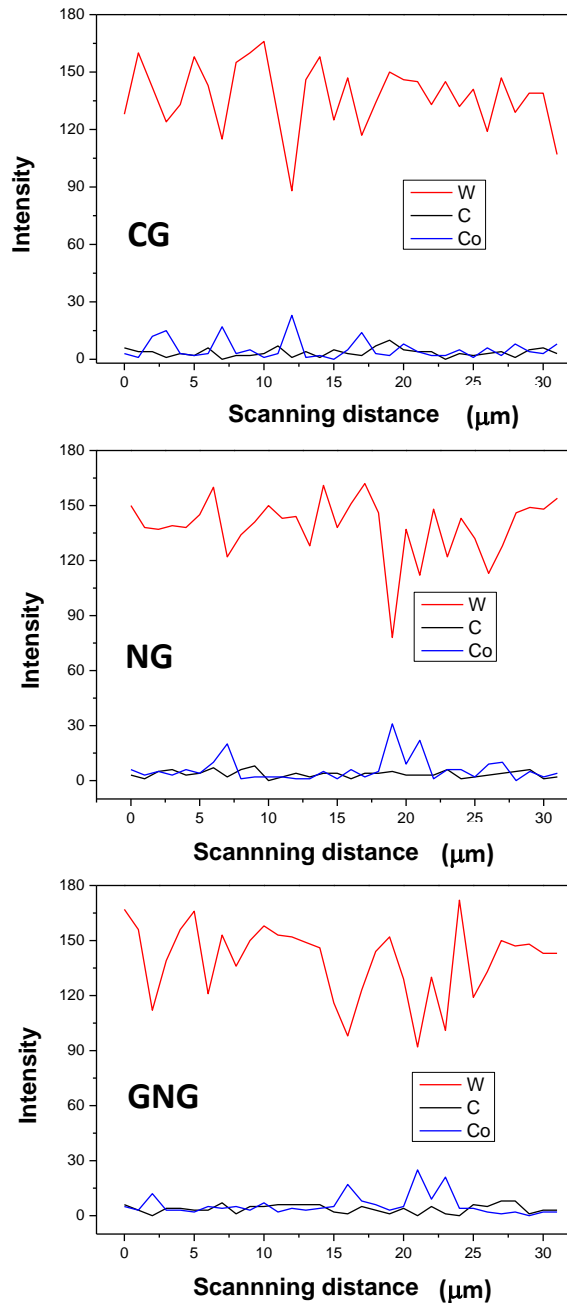
**fig. S2. Measurement results of wear rates.** Variation of wear volume with sliding cycles for the CG, NG, and GNG Cu-Ag samples sliding against WC-Co balls under a load of 50 N, a slide stroke of 1 mm and a velocity of 10 mm/s.



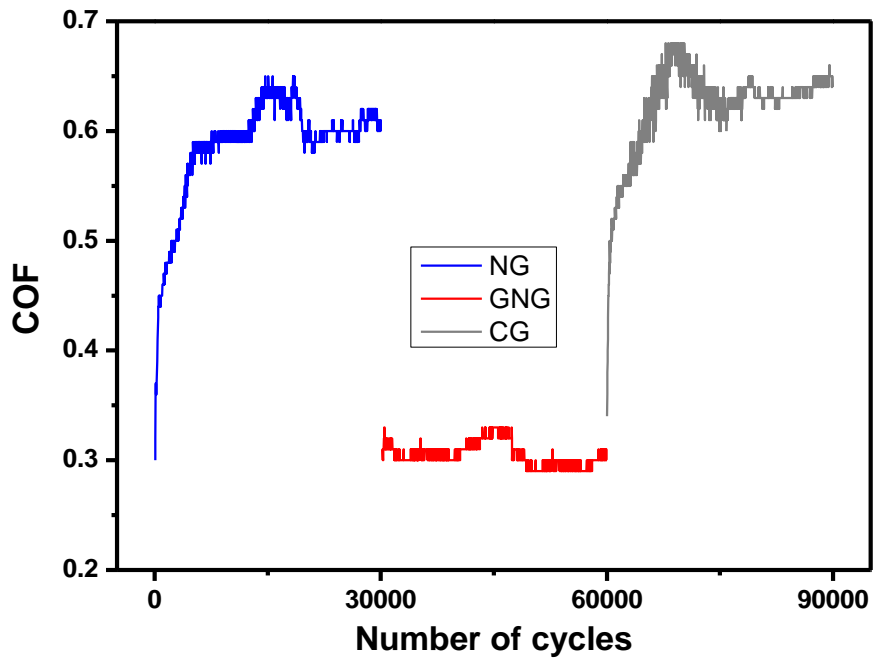
**fig. S3. Surface profiles and morphology of the CG sample under low-load single sliding.** Confocal laser microscopy image (A) and 3-D surface profiles (B) for surface morphology of a CG Cu-Ag sample after a single sliding at a load of 30 N, a slide stroke of 1 mm and a velocity of 10 mm/s.



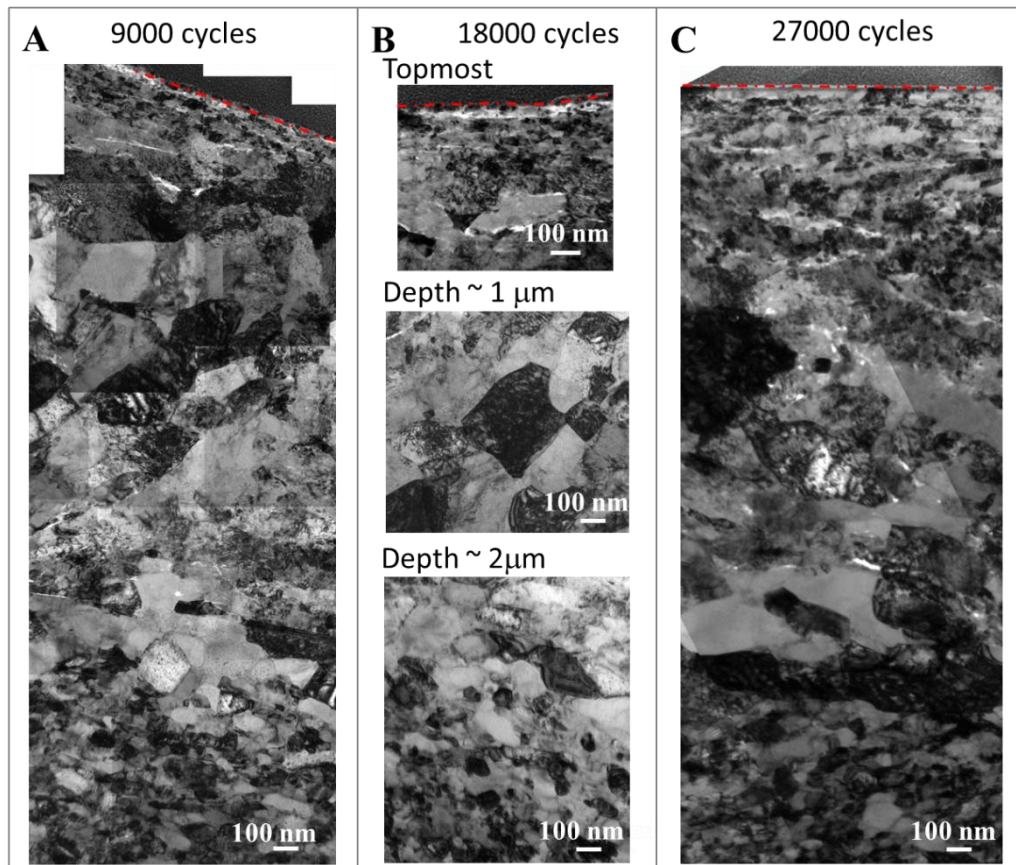
**fig. S4. Effect of Ag addition on COF reduction—Measurement results in pure Cu samples.** Variation of COFs with sliding cycles for the CG, NG, and the GNG pure Cu samples (purity better than 99.99%) sliding against WC-Co balls under a load of 30 N, a slide stroke of 1 mm and a velocity of 10 mm/s. The GNG and NG samples were prepared by using the same techniques as in the Cu-Ag samples. Analogous to the Cu-Ag samples, obvious COF reduction is obtained in the GNG Cu samples.



**fig. S5. Counter surface analysis.** Chemical composition analysis of the WC/Co ball counter surface after sliding for 18000 cycles on the CG, NG, and GNG samples under a load of 50 N, a slide stroke of 1 mm and a velocity of 10 mm/s. No obvious difference in the compositions of W, C, and Co is seen among the three samples. Elements of Cu and O have not been detected on the counter surfaces for the three tests, implying no transfer occurred (from and to the ball) during sliding.

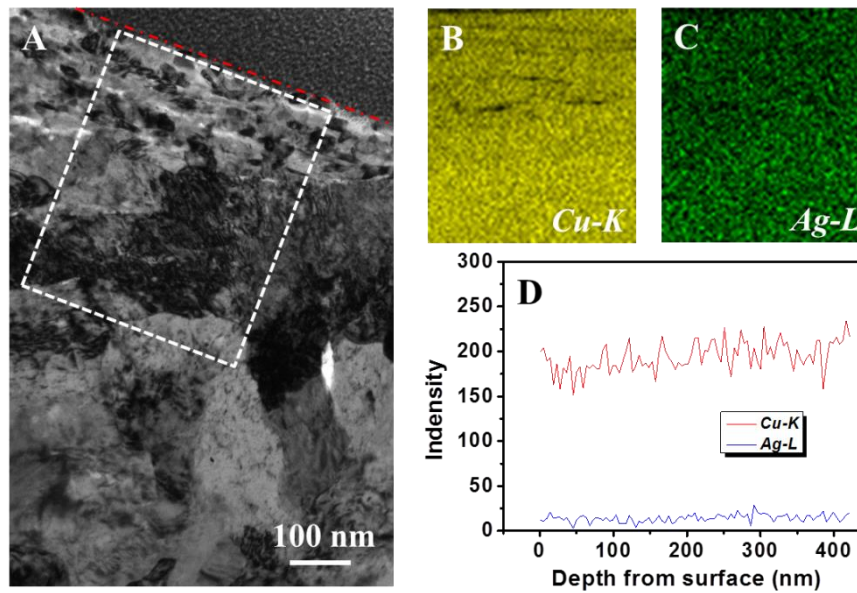


**fig. S6. COF measurement on the NG, GNG, and CG samples subsequently using exactly the same contact surface of a WC-Co ball.** Each sample was measured for 30000 cycles under a load of 50 N, a slide stroke of 1 mm and a velocity of 10 mm/s. Obviously, the effect of counter surface is negligible on the observed COF reduction in the GNG sample.

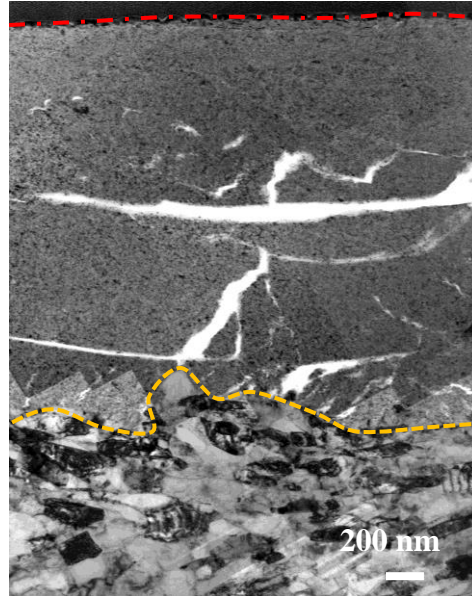


**fig. S7. Stability of the subsurface microstructure in the GNG samples against sliding.** (A-C) Typical TEM images of the subsurface microstructure in the GNG sample after sliding for 9000, 18000 and 27000 cycles at a load of 50 N, respectively. The worn surface is outlined by dash-dotted lines. It is clear that the subsurface microstructures remain basically unchanged against sliding in the steady-state stage.





**fig. S8. Chemical analysis of the topmost NG surface layer.** (A) A cross-sectional TEM image below the sliding surface of the GNG sample after 9000 cycles. (B to D) Corresponding elemental maps analysis (B: Cu, C: Ag) and EDS line scanning along the depth (D) in the topmost region (as indicated in A) below the worn surface. The worn surface is outlined by dash-dotted lines. No detectable variation in Cu and Ag is observed across the topmost surface layer and the grain-coarsened structures. It means the sliding-induced contamination in the topmost surface layer with nano-sized grains is negligible.



**fig. S9. Subsurface microstructures in the CG Cu under sliding in the steady state.** A cross-sectional TEM image below the sliding surface of the CG sample after 18000 cycles at load of 50 N, a slide stroke of 1 mm and a velocity of 10 mm/s. The sliding surfaces are outlined by dash-dotted lines and the tribolayer/recrystallization interfaces by dashed lines. Cracks (white lines) are seen in the top tribolayer.

**table S1. Surface roughness change after dry sliding for 18,000 cycles.** Measured surface roughness (Ra and Rz, in  $\mu\text{m}$ ) for different Cu-Ag samples before sliding and after sliding for 18000 cycles.

	CG	NG	GNG
Original			
Ra ( $\mu\text{m}$ )	$0.034 \pm 0.012$	$0.021 \pm 0.011$	$0.025 \pm 0.013$
Rz ( $\mu\text{m}$ )	$0.178 \pm 0.058$	$0.125 \pm 0.041$	$0.145 \pm 0.051$
18000 cycles			
Ra ( $\mu\text{m}$ )	$0.19 \pm 0.017$	$0.21 \pm 0.018$	$0.022 \pm 0.011$
Rz ( $\mu\text{m}$ )	$1.70 \pm 0.41$	$1.20 \pm 0.31$	$0.15 \pm 0.03$