



## Supporting Information

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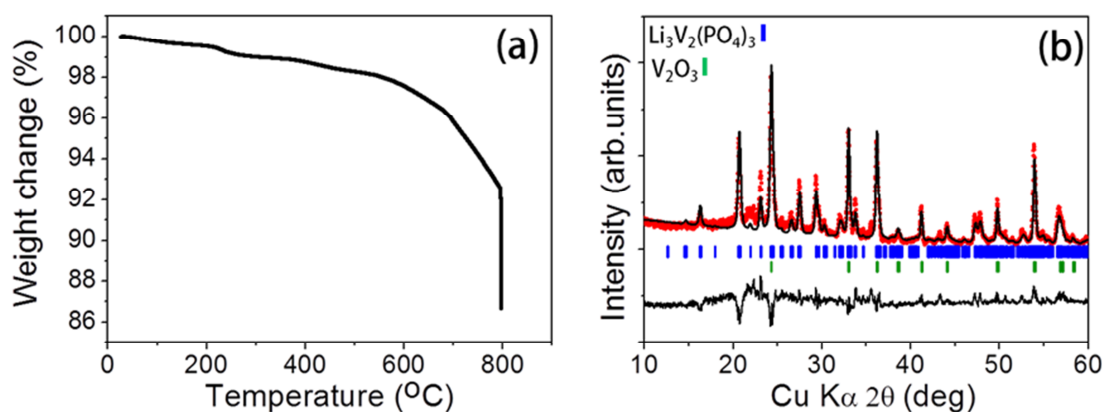
Fast-Rate Capable Electrode Material with Higher Energy Density than  $\text{LiFePO}_4$ : 4.2V  $\text{LiVPO}_4\text{F}$  Synthesized by Scalable Single-Step Solid-State Reaction

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## Supporting Information

## Fast rate capable electrode material with higher energy density than LiFePO<sub>4</sub>: 4.2V LiVPO<sub>4</sub>F synthesized by scalable single-step solid-state reaction

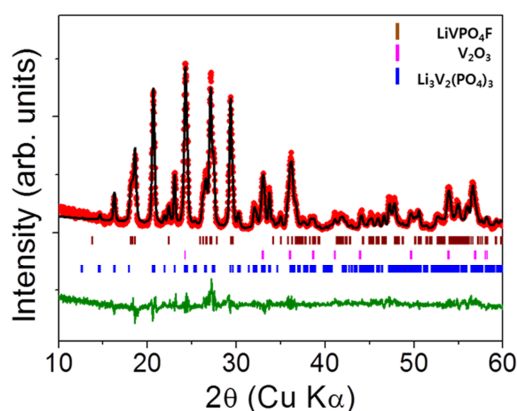
Minkyung Kim, Seongsu Lee and Byoungwoo Kang\*



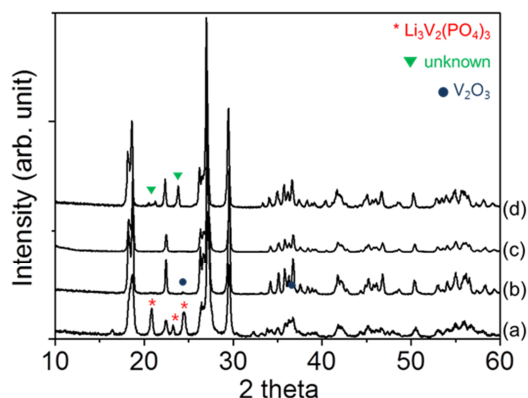
**Figure S1.** (a) TGA analysis of as-prepared LiVPO<sub>4</sub>F up to 800 °C at 5 Kmin<sup>-1</sup> heating rate and hold at 800 °C for one hour under argon atmosphere (b) XRD pattern of the decomposed product after TGA test.



The weight change of LiVPO<sub>4</sub>F in TGA was ~ 13wt% that was similar with that of the reaction (1).

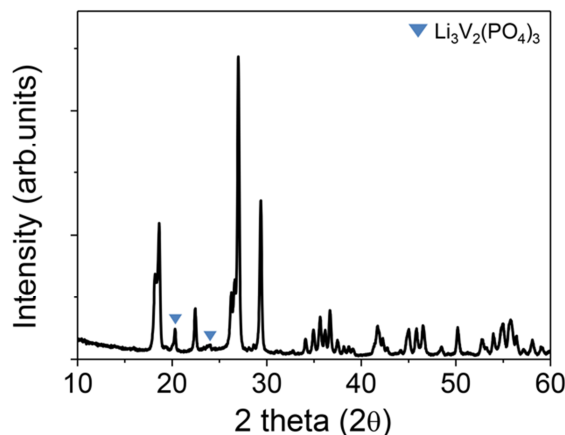


**Figure S2.** XRD pattern of the sample synthesized by the single-step solid-state reaction at 700 °C with the mixture of precursors (LiF, V<sub>2</sub>O<sub>5</sub>, and NH<sub>4</sub>H<sub>2</sub>PO<sub>4</sub>). The resulting sample was composed of LiVPO<sub>4</sub>F and impurity phases, Li<sub>3</sub>V<sub>2</sub>(PO<sub>4</sub>)<sub>3</sub> and V<sub>2</sub>O<sub>3</sub>.



**Figure S3.** X-ray diffraction patterns of the samples synthesized with different amount of PTFE in the mix of precursors (a) 5 wt% (b) 15 wt% (c) 25 wt% (d) 100 wt%. (700 °C for 1 h under argon)

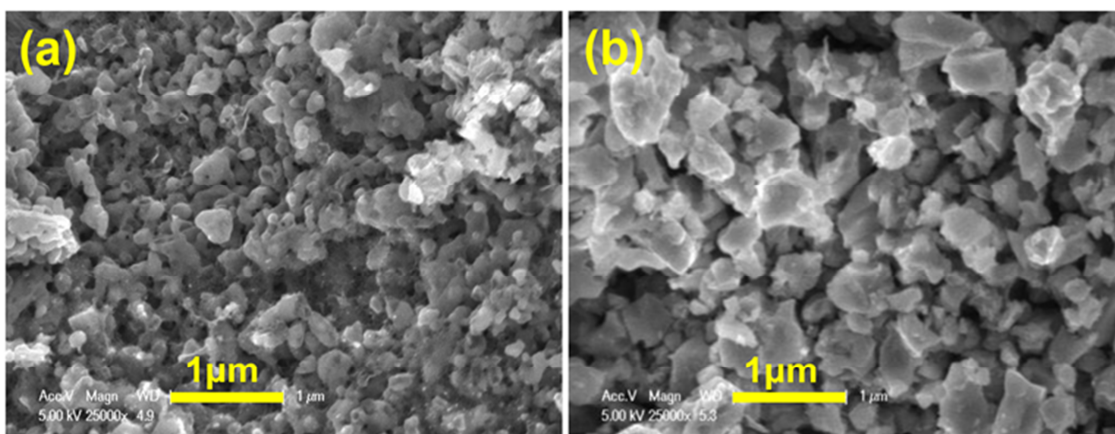
Phase purity of  $\text{LiVPO}_4\text{F}$  strongly depends on the amount of PTFE in the sample. 5 and 15 wt% of PTFE in the samples was not enough to suppress formation of impurity phases and 100 wt% of PTFE was too much to obtain phase pure  $\text{LiVPO}_4\text{F}$  leading to unidentified impurity phases that can be byproducts of PTFE decomposition.



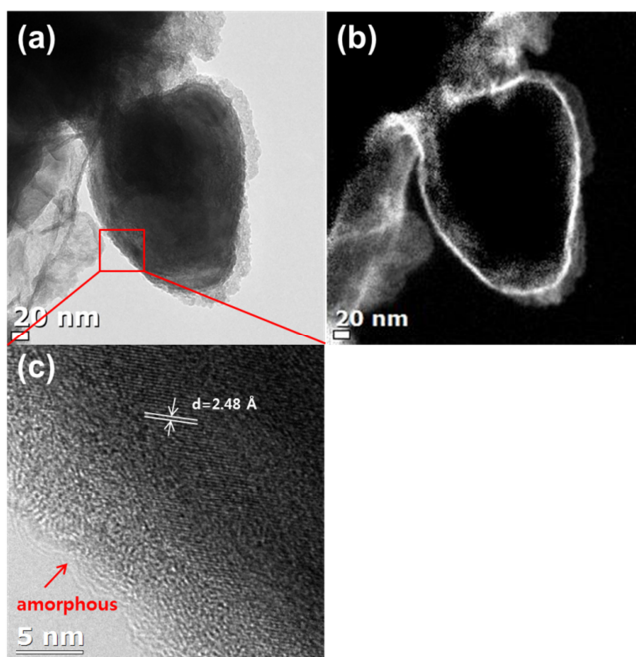
**Figure S4.** X-ray diffraction patterns of  $\text{LiVPO}_4\text{F}$  synthesized by CTR process.  $\text{Li}_3\text{V}_2(\text{PO}_4)_3$  impurity exists.

**Table S1.** Structural parameters obtained from Rietveld refinement of neutron diffraction pattern from  $\text{LiVPO}_4\text{F}$  synthesized by single-step process with PTFE (25 wt%). (700 °C for 1 h under argon)

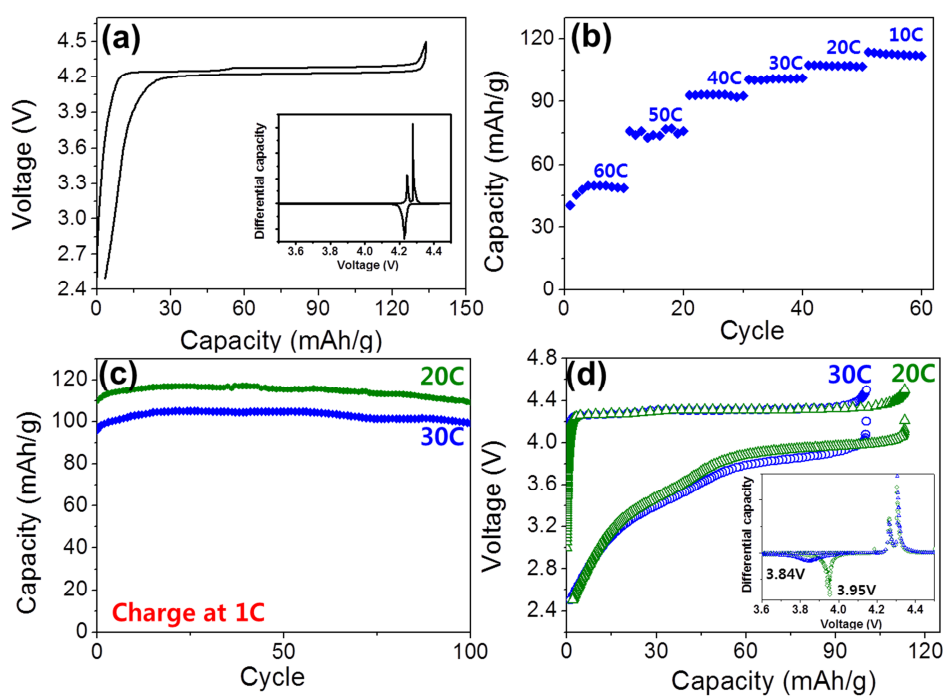
Space group: P-1, Z=2			$R_f = 1.36$
$a = 5.1726 (4) \text{ \AA}$ , $b = 5.3082 (3) \text{ \AA}$ , $c = 7.2612 (5) \text{ \AA}$			$R_{\text{braag}} = 2.28$
$\alpha = 107.5971 (6)^\circ$ , $\beta = 107.9643 (2)^\circ$ , $\gamma = 98.4061 (5)^\circ$			$\text{Chi}^2 = 5.92$
$V = 174.365 \text{ \AA}^3$			
	x	y	z
V1	0.00000(0)	0.00000(0)	0.00000(0)
V2	0.00000(0)	0.00000(0)	0.50000(0)
P	0.32320(87)	0.64468(82)	0.25487(71)
O1	0.37052(60)	0.23868(55)	0.58512(46)
O2	0.10648(72)	-0.33062(68)	0.36174(49)
O3	0.69842(71)	0.65649(64)	-0.13517(56)
O4	0.27216(78)	0.79624(73)	0.09621(49)
F	-0.11003(80)	0.08701(65)	0.24565(55)
Li	0.70790(270)	0.35916(358)	0.22114(234)



**Figure S5.** SEM images of the sample synthesized at different amount of stearic acid (a) carbon-coated  $\text{LiVPO}_4\text{F}$  using stearic acid 5wt% (b) bare  $\text{LiVPO}_4\text{F}$  without stearic acid. Both of them were synthesized by scalable single-step process with PTFE (25 wt%). After performing the carbon coating, particle size decreases from 200 - 600 nm (bare) to 50 - 200 nm.

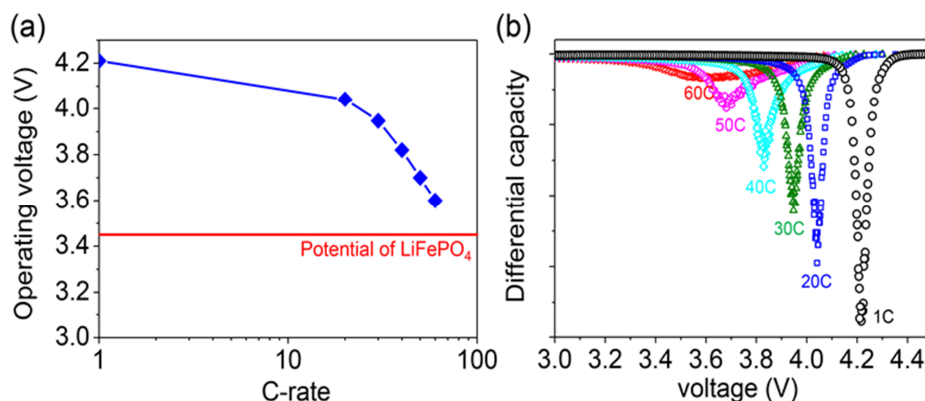


**Figure S6.** (a) TEM image of carbon-coated  $\text{LiVPO}_4\text{F}$  synthesized by PTFE process (b) EELS mapping image of carbon layer. Carbon is well coated on the surface of submicron particles. (c) HR-TEM image. It shows amorphous carbon layer and lattice fringe of  $\text{LiVPO}_4\text{F}$ .

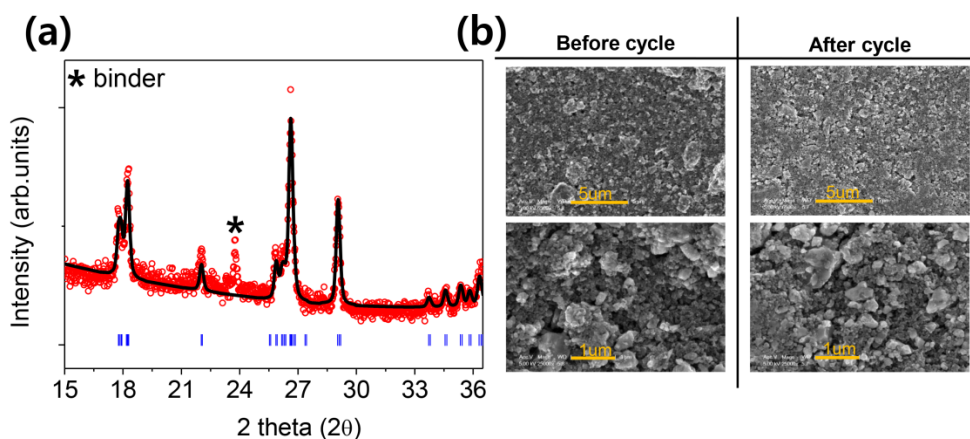


**Figure S7.** (a) Voltage profile of bare- $\text{LiVPO}_4\text{F}$  at C/10; (Inset: differential capacity ( $dQ/dV$ ) plot of the voltage profile) (b) Rate capability at various discharge rates of bare- $\text{LiVPO}_4\text{F}$  (the cell was charged at 1C rate and discharged at various rates) (c) Capacity retention of bare- $\text{LiVPO}_4\text{F}$  at high discharge rates such as 20 C and 30 C rate when the cell was charged at 1C rate. (d) The voltage profiles of 2<sup>nd</sup> cycle in capacity retention test. Cut-off voltage was 2.5 V-4.5 V. (Inset: differential capacity ( $dQ/dV$ ) plot of the voltage profiles at 20C and 30C rate).

Average working potentials at 20C and 30C discharge rate were higher than 3.45V,  $\sim 3.95$  V for 20C and  $\sim 3.84$  V at 30 C rate. For fabrication of the electrode, the ratio of the electrode was  $\text{LiVPO}_4\text{F}$ : carbon black (super P): binder (PVDF) = 80 :15 :5 (wt%). Loading density of electrode was  $\sim 2$  mg/cm<sup>2</sup>.



**Figure S8.** (a) Average operating voltages of carbon-coated  $\text{LiVPO}_4\text{F}$  sample synthesized by PTFE process depending on different discharge rates. (b) Differential capacity ( $dQ/dV$ ) plots of the discharge voltage profiles of carbon-coated  $\text{LiVPO}_4\text{F}$  sample at various C-rates. The operating voltage ( $\sim 3.6\text{V}$ ) of carbon-coated  $\text{LiVPO}_4\text{F}$  even at 60 C rate (1 min discharge) was much higher than 3.45V, the redox potential of  $\text{LiFePO}_4$ . This high operating potential with comparable capacity enables  $\text{LiVPO}_4\text{F}$  to have higher energy density than  $\text{LiFePO}_4$  at the same rate. (Voltage profiles from rate capability test in Fig. 6c)



**Figure S9.** Cycle retention test of C-coated  $\text{LiVPO}_4\text{F}$  at 10 C charge/10 C discharge. Other electrode was tested and it is confirmed stable cycle retention with similar capacity (a) XRD pattern and (b) SEM images of the electrode before and after 500 cycles.