## **Supplementary Information**



**Supplementary Figure 1.** Photographs show the titration experiments by dropwise adding  $\sim$  5 times number of moles of (a) LiOH and LiOH+H<sub>2</sub>O, (b) H<sub>2</sub>O<sub>2</sub> and H<sub>2</sub>O<sub>2</sub>+LiOH, (c) Li<sub>2</sub>O<sub>2</sub> and Li<sub>2</sub>O<sub>2</sub>+H<sub>2</sub>O into the I<sub>3</sub> solution in DME. The I<sub>3</sub> solution was prepared by mixing 0.25 mmole  $I_2$  and 0.25 mmole LiI in 5 mL DME solvent.



**Supplementary Figure 2**. The synchrotron X-ray diffraction patterns of Li<sub>2</sub>O, Li<sub>2</sub>O<sub>2</sub>, LiOH, and LiOOH⋅H<sub>2</sub>O. The wavelength of X-ray source is 0.117418 Å. The X-ray diffraction pattern of LiOOH is clearly different from any other Li compounds tested here. In addition, the sharp peaks indicate the good crystallinity of LiOOH. The high-resolution XRD pattern of the presumed LiOOH⋅H<sub>2</sub>O was also measured (Figure 1b). Based on the 28 primary peaks located between  $3^\circ$  to 16<sup>o</sup> (2 theta angle), the pattern could be well indexed using **DICVOL06** with a small zero-shift. A subsequent LeBail fit confirmed the indexing by achieving  $R_p$ : 5.66%,  $R_{wp}$ : 7.13%,  $R_{exp}$ : 4.20%, Chi<sup>2</sup>: 2.88, i.e. the difference between the observed data and a the fitted lattice is very small. More precise lattice parameters and more detailed microstructural information of the powder samples could be derived from a Lebail fit of the high resolution XRD pattern below 2 Theta =22°. The results are a= 6.36647 Å, b= 6.08798 Å, c= 3.20656 Å,  $\alpha$ =79.59°,  $\beta$ = 101.86°,  $\gamma$ = 102.31°, volume= 117.61 Å<sup>3</sup>, which is very close to the fitting result derived from the lower two theta angle.



**Supplementary Figure 3.** LeBail Fitting of the high-resolution X-ray diffraction pattern of the chemically synthesized phase LiOOH⋅H<sub>2</sub>O. The wavelength is  $\lambda = 0.41423 \pm 0.00004$  Å.



**Supplementary Figure 4.** XRD patterns of LiOOH·H2O after exposed in air or under vacuum condition for different durations. The freshly made wet powder of LiOOH·H<sub>2</sub>O was tested firstly. Then, the powder was left in air for 10 min and 2 hours, and tested again to monitor its stability. The formations of LiOH·H<sub>2</sub>O, LiOH, and Li<sub>2</sub>CO<sub>3</sub> are clearly seen as time evolves, of which  $Li_2CO_3$  could be a reaction product of LiOOH·H<sub>2</sub>O or LiOH with  $CO_2$  from air. In order to rule out the influence of CO<sub>2</sub>, we dried LiOOH·H<sub>2</sub>O in vacuum condition and did another XRD measurement, for which the main products become LiOH and LiOH·H2O, validating the above assumption.



**Supplementary Figure 5.** Raman spectra of LiOOH·H2O exposed in air for different durations. The LiOOH·H2O was initially in the form of wet paste. The Raman spectra of LiOOH·H2O evolved gradually and the characteristic peaks of LiOH and  $Li_2CO_3$  become obvious even after 5 min exposure.<sup>1, 2</sup> The LiOOH was entirely converted to LiOH and  $Li_2CO_3$  after 45 min, indicating LiOOH·H2O is not stable in air. This result is consistent with the XRD measurements.



**Supplementary Figure 6.** Total (black) and partial (colored) phonon density of state for crystalline LiOOH⋅H2O from DFT calculations. For the partial phonon DOS contributions relating to chemically equivalent atoms are added up. It has been confirmed by visualization that the peak calculated at 894 cm<sup>-1</sup> peak corresponds to O-O stretching in the OOH<sup>-</sup> group (superimposed by a H<sub>2</sub>O rocking mode at 874 cm<sup>-1</sup>), while the peak at 1643 cm<sup>-1</sup> peak corresponds to H-O-H bending.



**Supplementary Figure 7.** Snapshots of the vibrations at (a) 838 cm<sup>-1</sup> for LiOOH molecules, and (b) 828 cm<sup>-1</sup> for Li<sub>2</sub>O<sub>2</sub> molecules based on normal mode analysis. Both molecules were calculated at B3LYP/aug-cc-pVQZ level. LiOOH shows a strong Raman peak at 838 cm<sup>-1</sup> which was in good accordance with the experimental data of 860 cm<sup>-1</sup>. Based on the normalmode analysis, this peak is assigned to be the O-O bond stretching. Similarly, Li<sub>2</sub>O<sub>2</sub> demonstrated a Raman active O-O bond stretching at 828 cm<sup>-1</sup> while the experiment value is at 790 cm<sup>-1</sup>. Note that calculations presented in (a and b) are based on isolated molecules. In any case both these simplifications and the limited level of the DFT calculations may account for the error between the calculation and experimental data.



**Supplementary Figure 8.** Raman spectrum of  $H_2O_2$  solution (35 *wt.*% in  $H_2O$ ). The substrate is a piece of glass plate. In the measurement,  $H_2O_2$  solution was dropped onto the glass substrate. A strong peak at around 877 cm<sup>-1</sup> was observed, which is assigned to the O-O bond stretching of  $H_2O_2$ . In comparison, the O-O bond stretching of LiOOH⋅H<sub>2</sub>O is located at around 860 cm<sup>-1</sup>, distinct from that of  $H_2O_2$ .



**Supplementary Figure 9.** (a) ATR-FTIR spectra of Li<sub>2</sub>O<sub>2</sub>, LiOH·H<sub>2</sub>O, LiOOH·H<sub>2</sub>O. Li<sub>2</sub>O<sub>2</sub> presents characteristic peaks at around 524 and 432 cm<sup>-1</sup>.<sup>3</sup> LiOH·H<sub>2</sub>O exhibits a series of peaks from ~1000 to 610 cm<sup>-1</sup>.<sup>4</sup> The FTIR spectrum of LiOOH·H<sub>2</sub>O shows a characteristic peak at around  $1642 \text{ cm}^{-1}$ . LiOOH·H<sub>2</sub>O was extracted from DME/H<sub>2</sub>O solution.



**Supplementary Figure 10.** (a) The exploded structure of static Li-O<sub>2</sub> battery used in this work. (b) The photograph of a piece of LAGP ceramic membrane. (c) The photograph of a redox flow lithium oxygen battery (RFLOB), in which the same battery stack was used for static cells except that the catholyte inlet/outlet on the cathode were converted into O<sub>2</sub> inlet/outlet.



**Supplementary Figure 11.** The charge/discharge curve of a Li cell containing 0.5 mL catholyte consisting of 50 mM LiI in 0.5 M LiTFSI/DME. The cell has the same structure as the static  $Li-O<sub>2</sub>$  battery in Figure S7. The anolyte was 0.5 M LiTFSI in DEGDME. The test was conducted in Ar atmosphere. The current was  $0.1 \text{ mA/cm}^2$ . The effective ceramic area is  $1 \text{ cm}^2$ . The overall charging time is 6.8 h, nearly reaching the theoretical capacity of  $\Gamma/I_3$  and  $I_3/I_2$ . Therefore, the capacity from the redox mediators in static Li-O2 batteries could be reliably estimated.



**Supplementary Figure 12.** The differential pulse voltammetry (DPV) measurements of 5 mM LiI in 0.5 M LiTFSI/DME electrolyte with different water contents. The working electrode was Pt disc electrode and the counter electrode was a piece of Pt plate. The reference electrode was Ag/AgNO<sub>3</sub>. The potential is converted into  $Li/L<sup>+</sup>$  scale for convenience. The potential range is from -0.4 to 0.3 V. The step potential is 0.005 V and the modulation amplitude is 0.025 V. Based on the DPV measurement, the redox potential of I/I<sub>3</sub> has broadly a positive shift with the increase of water content in the aprotic electrolyte, while in contrast that of  $I_3/I_2$  stays nearly constant. Precaution should be taken that as the Ag/AgNO<sub>3</sub> reference electrode was not calibrated when water content in the electrolyte bulk was increased, the changes of junction potential of the reference electrode may impose some error on the measurement.



**Supplementary Figure 13.** ATR-FTIR spectra of the electrolyte and the ORR reaction products by  $EV^+$  in the presence of 9.1  $vol$ %  $H_2O$  in the electrolyte.

To substantiate the presence of LiOH and LiOOH in the discharge (ORR) product, ATR-FTIR measurements were conducted. Here the ORR reaction was promoted by  $EV^+$ , which reduces  $O_2$  in the presence of  $Li^+$  and forms LiOH and LiOOH. As such, we firstly obtained  $EV^+$  electrolyte by reducing 0.1 M  $EV^{2+}$  in 0.5 M LiTFSI / (DMSO:DME, 1:1) electrolyte with Li metal. 9.1 *vol.*% water was then added into the  $EV^+$  electrolyte which was subsequently injected into a flow cell holder in  $O_2$ atmosphere for ATR-FTIR measurement. As shown in Figure S13, the characteristic peaks of LiOH (910-1070 cm<sup>-1</sup>, partly overlapped with that of the electrolyte) and LiOOH·H<sub>2</sub>O (1641 cm<sup>-1</sup>) appeared after 5 min reaction and became more pronounced after 10 min. This further verifies the formation of LiOH and LiOOH from the ORR reaction in the presence of 9.1 vol.% water in the electrolyte.



**Supplementary Figure 14.** (a) UV-Vis spectra of  $I_3$  in DME/H<sub>2</sub>O (10/1) before and after adding excessive LiOOH. (b) UV-Vis spectra of standard  $0.1 \text{ mM } I_3$  solution and the reacted  $I_2$  solution with LiOH. The inset of (b) shows the photos of 6.6 mM  $I_3$  (1), 10 mM reacted  $I_2$  (2, DME/H<sub>2</sub>O = 10/1), 10 mM  $I_2$  (3), and 10 mM reacted  $I_2$  (4, DME/H<sub>2</sub>O = 1/1) solutions.

As the UV-Vis spectra shown in Figure S14a, the characteristic absorption peak of  $I_3$  at 364 nm (extended to the visible region) vanished after mixing with LiOOH·H2O. As a result, the solution became nearly colourless. After adding excessive LiOH and stirring for 1 hour, the colour of  $I_2$ solution in DME/H<sub>2</sub>O (10:1) became slightly lighter (inset of Figure S14b). As the UV-Vis spectra shown in Figure S13b, after reacting with LiOH, the absorption of the  $I_2$  solution became fairly identical to that of the standard solution of I<sub>3</sub>. That is, the absorption at 400-500 nm was greatly attenuated with only the characteristic peak of  $I_3$  presented at around 364 nm, which confirms the existence of  $I_3$  in the reacted  $I_2$  solution. The absorption peak broadening of  $I_2$  solution after titration is presumably attributed to the scattering effect of small particles of LiOH, which has limited solubility in  $DME/H<sub>2</sub>O$  (10:1).



**Supplementary Figure 15.** Oxygen evolution recorded from the reactions of (a) LiOOH with I<sub>3</sub> and (b) LiOH with I<sub>2</sub> monitored by mass spectrometry. The reactions of LiOH with  $I_2$  were performed in solutions with different concentration of  $H<sub>2</sub>O$  as labelled in the figure.

The mass spectrometric measurement in Figure S15a shows that  $O<sub>2</sub>$  evolves instantaneously upon mixing I<sub>3</sub> with LiOOH·H<sub>2</sub>O. The mass spectrometric measurement of the reaction between LiOH and I2 was conducted in two different solutions. As shown in Figure S15b, oxygen evolution was observed instantaneously after  $I_2$  was injected into the 2 M LiOH suspension in DME/H<sub>2</sub>O (10:1). Therefore,  $O_2$  evolution is deemed part of the reaction between  $I_2$  and LiOH in the DME/H<sub>2</sub>O (10:1) electrolyte system, which is the same electrolyte used in the  $Li-O<sub>2</sub>$  cell. In contrast, no oxygen was detected when the solvent was changed to  $DME/H<sub>2</sub>O (1:1)$ .



**Supplementary Figure 16.** Scanning electron microscopic images of the discharging products formed on the carbon felt of a static Li-O<sub>2</sub> battery in the case of (a) dry catholyte; (b) and (c) 9.1 *vol*.% H<sub>2</sub>O in catholyte. Images (b) and (c) were taken on the different location of the same sample. The scale bar, 500 nm.

## **Crystal Structure of LiOOH**⋅**H2O from Rietveld refinement of synchrotron XRD data**

**Space group**: *P*-1

**Lattice constants**:  $a = 6.36610(4)$  Å,  $b = 6.08770(4)$  Å,  $c = 3.20629(2)$  Å, α = 79.5953(4)° β = 101.8572(4)° γ = 102.3084(4)°

**Cell volume** = 117.591(2)  $A^3$  **Density**: 1.637 g/cm<sup>3</sup>

*Supplementary Table 1 1Atomic coordinates and atomic displacement parameters* 



$$
R_{wp} = 7.86\%, R_p = 6.1\%, \chi^2 = 2.93
$$

Global instability index  $GII = 0.077$ 

## **Crystal Structure of LiOOH**⋅**H2O from DFT geometry optimisation**

**Space group**: *P*-1

**Lattice constants**: *a* = 6.3178 Å, *b* = 6.1433 Å, *c* = 3.2120 Å, α = 80.5404° β = 99.8294° γ = 103.3629°

**Cell volume** = 118.49  $A^3$  **Density**: 1.625 g/cm<sup>3</sup>

*Supplementary Table 2 Atomic coordinates and atomic displacement parameters* 



## **Supplementary References**

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