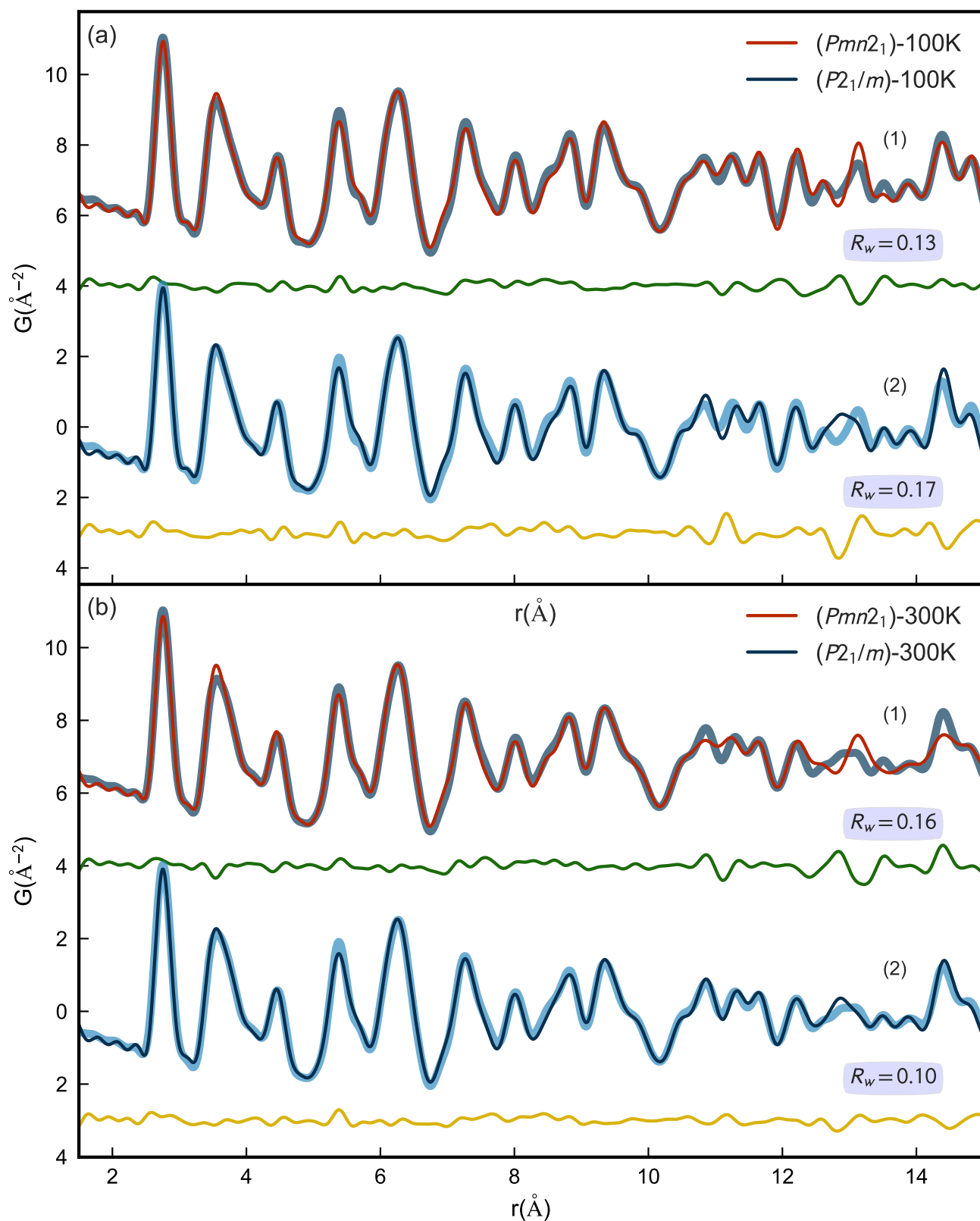
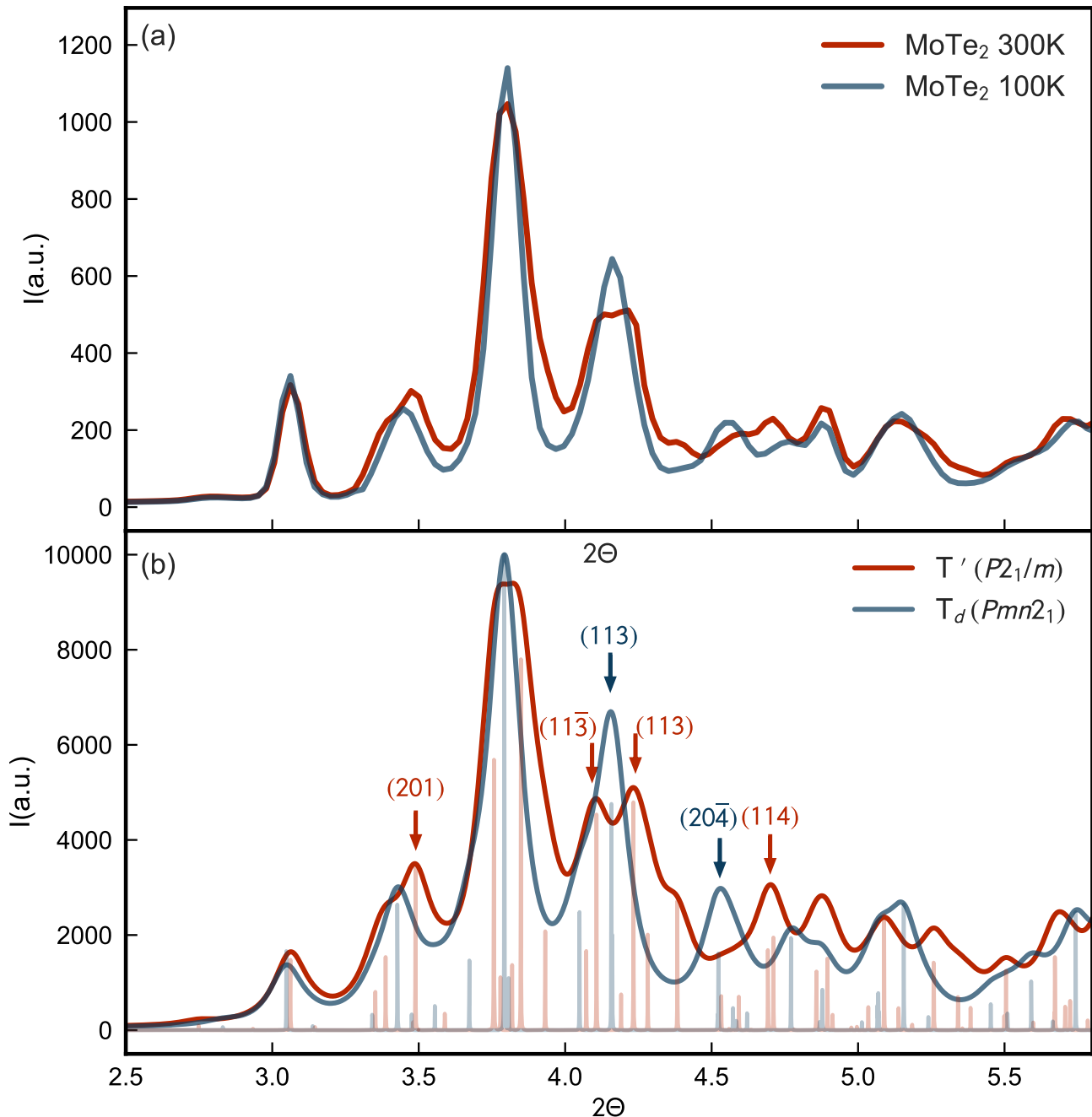


Supplementary Figure 1. **Powder diffraction patterns of MoTe<sub>2</sub>**. Patterns recorded at 300K (a) and 100K (b) plotted as a function of  $Q$  and truncated where Bragg peaks are no longer distinguishable. The insets show the total scattering structure function  $F(Q)$  where the full range used for the PDF transformation is provided. Small vertical lines in red mark the upper limit of  $I(Q)$  given in (a,b), past which a significant amount of scattered intensity is seen in the structure function. Fine green lines correspond to calculated Bragg peaks from candidate models T' (a) and T<sub>d</sub> (b). To the right we show the centrosymmetric monoclinic structure plotted above the non-centrosymmetric orthorhombic structure, from references discussed in the text. The MoTe<sub>6</sub> octahedral units are shaded in blue. Note that in order to provide a useful projection to compare T' and T<sub>d</sub>, the unit cell orientation for the orthorhombic structure is changed from the standard (abc) setting for  $Pmn2_1$  to (ba-c) for the symmetry equivalent  $Pnm2_1$  space group.



Supplementary Figure 2. **Pair Distribution Function Analysis of MoTe<sub>2</sub> data.** (a) PDF local structure refinements of MoTe<sub>2</sub> measured at 100K fit to the T<sub>d</sub> structure (1) and the T' structure (2). (b) an analogous comparison for the 300K measurement. Thick curves are the raw PDF data, and the calculated PDF from model is overlaid as thin solid lines, and labeled in the legend. Offset difference curves in green and yellow are simply the calculated PDF subtracted from the measured PDF. The goodness-of-fit parameter ( $R_w$ ) is provided for each of the four refinements, which confirm that the 100 K data is better described as T<sub>d</sub> and the 300K data as T'.



Supplementary Figure 3. **Comparison of high and low- $T$  diffraction patterns.** (a) Measured diffraction patterns at 300K and 100K using an incident wavelength of  $\lambda = 0.18470$  Å. (b) Bragg profile calculated from unmodified candidate structures. Solid curves in red and blue are the Bragg peaks calculated from the reflections (sharp lines), broadened uniformly with an FWHM=0.1 in  $2\theta$  ( $=0.059\text{\AA}^{-1}$  in  $Q$ ). The orthorhombic (113) reflection measured at 100K, splits into the (113) and (113) at 300K, which was also reported for WTe<sub>2</sub> [1]

### I. SUPPLEMENTARY NOTE 1: CONFIRMATION OF THE ORTHORHOMBIC ( $T_d$ ) TO MONOCLINIC ( $T'$ ) PHASE TRANSITION IN MOTE<sub>2</sub>

Total scattering measurements were carried out on the XPD (28-ID-2) beamline at the National Synchrotron Light Source II (NSLS-II), Brookhaven National Laboratory. A finely ground powder of MoTe<sub>2</sub> was prepared in an inert Argon chamber, and sealed in 1.02mm (OD) polyimide capillaries. Diffraction patterns were collected in a Debye-Scherrer geometry with an X-ray energy of 67.127 keV ( $\lambda = 0.18470$  Å) using a large-area 2D PerkinElmer detector (2048<sup>2</sup> pixels with 200  $\mu\text{m}^2$  pixel size). The detector was mounted with a sample-to-detector distance of 344.79 mm, to achieve a balance between  $q$ -resolution and  $q$ -range. The sample was measured at 100K and 300K using an Oxford CS-700 cryostream, allowing ample time for the material to thermalize. The experimental geometry,  $2\theta$  range, and detector misorientations were calibrated by measuring a crystalline nickel powder directly

prior to data collection at each temperature point, with the experimental geometry parameters refined using the PyFAI program [2].

Raw 2D diffraction patterns are azimuthally integrated to obtain the 1D scattering intensity  $I(Q)$  where  $Q = 4\pi \sin\theta/\lambda$ . Standardized corrections are then made to the data and  $I(Q)$  is normalized by dividing by the total scattering cross-section of the sample, resulting in the structure function  $S(Q)$

$$S(Q) - 1 = \frac{I_c(Q) - \sum c_i |f_i(Q)|^2}{\sum c_i |f_i(Q)|^2} \quad (1)$$

where  $I_c$  is the coherent scattering intensity as a function of the momentum transfer  $Q$  and  $c_i$  and  $f_i$  are the concentration and X-ray scattering factor, per atom type  $i$ . Because the scattering cross section becomes very small at high- $Q$ , an important result of this normalization is that the high angle data are significantly amplified [3]. This is illustrated in Figure 1(a,b) where we show the standard integrated XRD patterns  $I(Q)$  versus the reduced total scattering structure function  $F(Q)$  (insets), where  $F(Q) = Q[S(Q) - 1]$ . While the raw diffraction pattern appears to have almost no distinguishable peaks near the upper limit of the shortened  $Q$ -range, the form factor normalized  $F(Q)$  contains well resolved, high amplitude peaks extending to a  $Q$ -range twice the maximum value shown for  $I(Q)$ . The high degree of structural coherence observed for MoTe<sub>2</sub> is in contrast to diffraction studies of similar layered TMDs, where diffuse scattering from turbostratic disorder and other defects often dominates the high angle signal[4].

The full  $Q$ -range where structure can be resolved, including both Bragg and diffuse components, is shown in the insets of Figure 1. The structure function is Fourier transformed to obtain the pair distribution function (PDF), using PDFgetX3 [5] within xPDFsuite [6]. The experimental PDF,  $G(r)$ , is the truncated Fourier transform of  $F(Q)$

$$G(r) = \frac{2}{\pi} \int_{Q_{\min}}^{Q_{\max}} F(Q) \sin(Qr) dQ, \quad (2)$$

For both MoTe<sub>2</sub> measurements (100K and 300K), the  $Q_{\max}$  was chosen to be  $22.5 \text{ \AA}^{-1}$  to give the best tradeoff between statistical noise and real-space resolution.

The PDF is a histogram of interatomic distances that gives the probability of finding pairs of atoms in a material, separated by a distance  $r$ .  $G(r)$  can be calculated from a structure model according to

$$G(r) = 4\pi r [\rho(r) - \rho_0], \quad (3)$$

$$\rho(r) = \frac{1}{4\pi r^2 N} \sum_i \sum_{j \neq i} \frac{f_i f_j}{\langle f \rangle^2} \delta(r - r_{ij}).$$

Here,  $\rho_0$  is the atomic number density of the material and  $\rho(r)$  is the atomic pair density, which is the mean weighted density of neighbor atoms at distance  $r$  from an atom at the origin. The sums in  $\rho(r)$  run over all atoms ( $N$ ) in the model, with periodic boundary conditions.  $f_i$  is the scattering factor of atom  $i$ ,  $\langle f \rangle$  is the average scattering factor and  $r_{ij}$  is the distance between atoms  $i$  and  $j$ . Supplementary Eqs. 3 are used to fit the PDF calculated from a model to the experimentally measured PDFs. The PDFGUI program was used to construct unit cells from reference structures, carry out the refinements, and determine the agreement between simulated PDFs and data, quantified by the residual function  $R_w$  [7].

We included three candidate structure models for local structure refinements of MoTe<sub>2</sub> PDFs measured at 100K and 300K. The monoclinic 1T'-MoTe<sub>2</sub> (SG:  $P2_1/m$ ) structure reported by B.E. Brown [8], the orthorhombic 1T<sub>d</sub>-MoTe<sub>2</sub> (SG:  $Pmn2_1$ ) structure, also referred to as the  $\gamma$  form, and a hexagonal 2H polytype (SG:  $P6_3/mmc$ ) reported by D. Puotinen and R.E. Newnham [9]. The 2H structure was ruled out as both 1T structures gave significantly better fits for both low-temperature and high-temperature datasets. The T' and T<sub>d</sub> PDF refinements were performed conservatively by adjusting just a few parameters in the structure models: lattice ( $a, b, c$ ), scale, delta-2 (a parameter for correlated motion effects), one isotropic atomic displacement parameter (ADP) for the 4 Mo atoms, and one ADP for the 8 Te atoms, with both structures containing 12 atoms in the unit cell. For the monoclinic phase, an additional symmetry allowed  $\beta$  angle was refined. Atomic positions were not refined. The instrumental resolution parameters  $Q_{\text{damp}} = 0.0217$  and  $Q_{\text{broad}} = 0.009$  were determined through refinements of the nickel calibrant, and kept fixed for all other refinements.

PDFs for MoTe<sub>2</sub> are refined over an  $r$ -range from  $1.5 < r < 15 \text{ \AA}$ , slightly larger than the  $c$ -axis length for both T' and T<sub>d</sub> models. The PDF at low- $r$  contains structural information from the full  $F(Q)$  range and as a result, refinements are very sensitive to deviations from the average structure. This is advantageous when characterizing the local atomic environment for MoTe<sub>2</sub>, with both candidate models containing layers of edge-sharing, distorted MoTe<sub>6</sub> octahedra (Figure 1) within the refinement range selected. The approach is different from traditional powder XRD Rietveld refinements of similar materials, where a substantial amount of diffuse scattering is neglected, and phase identification is restricted to a small number of low angle Bragg reflections, thus increasing the number of possible structures that may fit equivalently to the same data.

The results of the PDF analysis are summarized in Figure 2. For MoTe<sub>2</sub> measured at 100K there is a clear

improvement in the agreement factor ( $R_w$ ) when the data is refined to the orthorhombic  $T_d$  model, with smaller residuals indicating better fits. The difference in refinement quality is more significant above the orthorhombic to monoclinic phase transition ( $T_s \sim 250K$ ) and the 300K PDF is in good agreement with the  $T'$  model. By varying a small number of structural parameters we ensure that these differences in  $R_w$  accurately represent a structural transformation. Additionally, the refined lattice constants differ minimally from the starting values provided in the reference  $T'$  and  $T_d$  models. The best-fit PDF at 100K using the  $T_d$  structure, yields refined lattice parameters of  $a = 3.474\text{\AA}$ ,  $b = 6.328\text{\AA}$ ,  $c = 13.884\text{\AA}$ , and small ADPs,  $U_{\text{iso}}(\text{Mo}) = 0.0043$ ;  $U_{\text{iso}}(\text{Te}) = 0.0056$ . For  $\text{MoTe}_2$  at 300K, the monoclinic structure refines  $a = 6.335\text{\AA}$ ,  $b = 3.471\text{\AA}$ ,  $c = 13.853\text{\AA}$ ;  $\beta = 93.886^\circ$  and  $U_{\text{iso}}(\text{Mo}) = 0.0058$ ;  $U_{\text{iso}}(\text{Te}) = 0.0064$ .

In Figure 3, we index the Bragg profile to support the results from the more quantitative PDF refinements. Here we simply compare the raw diffraction patterns measured at 100K and 300K to powder diffraction patterns calculated from the candidate models using the program VESTA [10]. We plot the data over a  $2\theta$  range where the  $T'$  and  $T_d$  models have Bragg peaks that are well resolved from one another to highlight distinguishable features that differ between the phases in the measured XRD pattern.

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