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Supplementary Information for

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Ultrafast probes of electron-hole transitions between two atomic layers

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Wen et al.

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12 **Supplementary Information for**

13 **Ultrafast probes of electron-hole transitions between two atomic layers**

14 Xiewen Wen^{2†}, Hailong Chen^{4†}, Tianmin Wu^{6†}, Zhihao Yu^{1†}, Qirong Yang¹, Jingwen
15 Deng¹, Zhengtang Liu¹, Xin Guo¹, Jianxin Guan¹, Xiang Zhang², Yongji Gong²,
16 Jiangtan Yuan², Zhuhua Zhang², Chongyue Yi⁵, Xuefeng Guo¹, Pulickel M. Ajayan²,
17 Wei Zhuang^{3*}, Zhirong Liu^{1*}, Jun Lou^{2*}, Junrong Zheng^{1*}

18 1 College of Chemistry and Molecular Engineering, Beijing National Laboratory
19 for Molecular Sciences, Peking University, Beijing 100871, China

20 2 Department of Materials Science and NanoEngineering, Rice University, 6100
21 Main Street, Houston, Texas 77005-1892, United States

22 3 State Key Laboratory of Structural Chemistry, Fujian Institute of Research on the
23 Structure of Matter, Chinese Academy of Sciences, Fuzhou, Fujian 350002, China

24 4 Beijing National Laboratory for Condensed Matter Physics, CAS Key Laboratory
25 of Soft Matter Physics, Institute of Physics, Chinese Academy of Sciences,
26 Beijing 100190, China

27 5 Department of Chemistry, Rice University, 6100 Main Street, Houston, Texas
28 77005-1892, United States

29 6 Department of Chemical Physics, University of Science and Technology of China,
30 Hefei, Anhui 230026, China

31 †These authors contributed equally to the work

32 *To whom correspondence should be addressed: junrong@pku.edu.cn,

33 zhengjunrong@gmail.com, jlou@rice.edu, wzhuang@fjirms.ac.cn,

34 liuzhirong@pku.edu.cn

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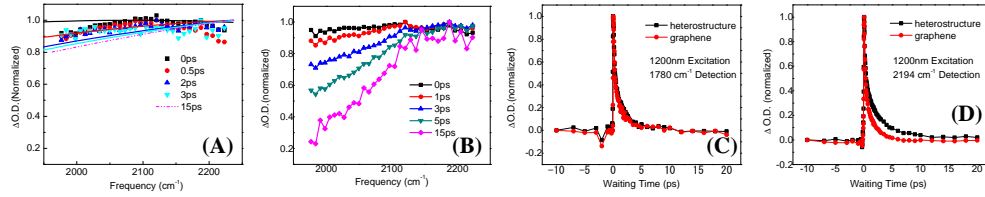
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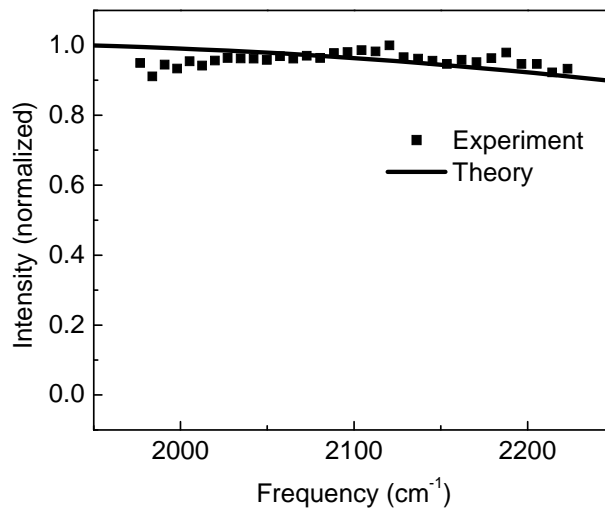
Supplementary Figures



57

58 **Supplementary Figure 1.** *Waiting time dependent IR spectra of monolayer graphene*59 *and heterostructure after excitation with 1.03 eV photons. (A) Graphene; (B)*60 *heterostructure; (C) pump/probe data detected at 1780 cm⁻¹; (D) pump/probe data*61 *detected at 2194 cm⁻¹. Dots are data, and lines are calculations. Calculation*62 *parameters: graphene Fermi level $\mu = -0.19$ eV, phonon fraction $f_{SCOPs} = 0.15$,*63 *photon flux absorbed $F = 0.0069$ J/m², pump/probe response time $t = 170$ fs, and*64 *electron-phonon coupling parameter $\eta = 6.0 \frac{eV}{\text{\AA}}$. These parameters are used for all*65 *calculations for graphene monolayer excited with 1.03 eV.*

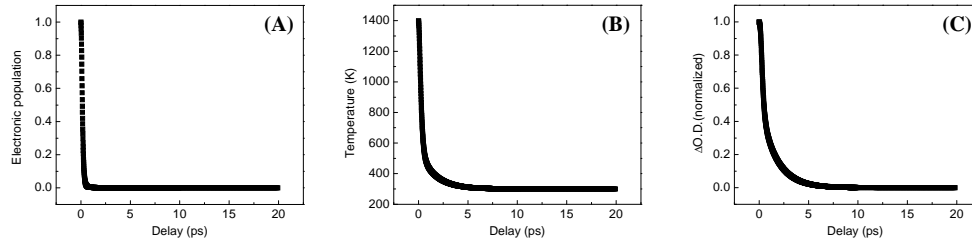
66



67

68 **Supplementary Figure 2.** *Spectrum of heterostructure at $t = 0$ ps after excitation*

69 with 1.03 eV photons. Dots are data, and the line is calculation. Calculation
 70 parameters: graphene Fermi level $\mu = -0.17$ eV, phonon fraction $f_{SCOPs} = 0.15$,
 71 photon flux absorbed $F = 0.0046$ J/m², response time $t = 170$ fs, and
 72 electron-phonon coupling parameter $\eta = 6.0 \frac{eV}{\text{\AA}}$.
 73

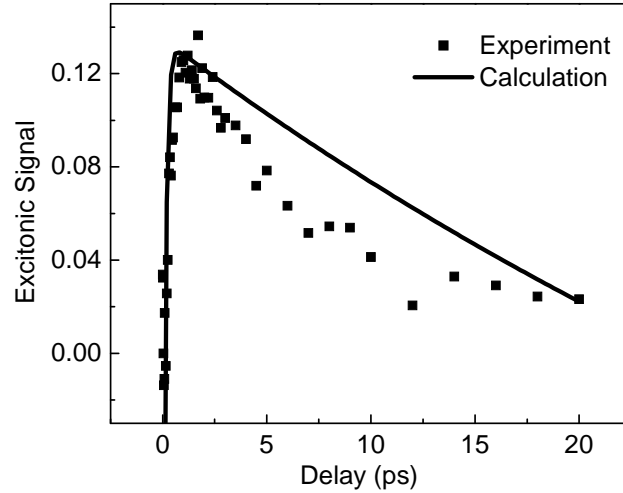


74

75 **Supplementary Figure 3.** (A) Calculated electronic population relaxation in
 76 heterostructure after 1.03 eV excitation. (B) Calculated temperature of graphene in
 77 heterostructure. (C) Calculated transient IR signal (2185 cm⁻¹) of graphene in
 78 heterostructure after 1.03 eV excitation. Calculation parameters: graphene Fermi
 79 level $\mu = -0.17$ eV, phonon fraction $f_{SCOPs} = 0.15$, photon flux absorbed
 80 $F = 0.0046$ J/m², pump response time $t = 170$ fs, and electron-phonon coupling
 81 parameter $\eta = 6.0 \frac{eV}{\text{\AA}}$. Subtracting (C) from the experimental signal of
 82 heterostructure (Supplementary Figure 3E) yields the excitonic signal (Supplementary
 83 Figure 3F). Here the temperature increase from energy released from exciton
 84 formation is ignored. The signal reduction because of exciton formation is not
 85 considered either. The effects of these two factors on the graphene signal are opposite.
 86 Because of this, we estimate the uncertainty of the exciton formation time introduced
 87 by the approximations to be <20% (on average around 25% free carriers form

88 excitons). The exciton formation rate is slightly underestimating by this method.

89



90

91 **Supplementary Figure 4.** Excitonic signal of heterostructure after 1.03 eV excitation.

92 The data (dots) are obtained by directly subtracting the graphene signal from the

93 heterostructure signal. Calculations (line) show that the excitonic formation time is

94 325 fs, which is essentially the same as 364 fs of Supplementary Figure 3F. Both time

95 constants are not normalized to the absorption cross section. The line is a

96 biexponential with $t_1=120$ fs (electronic decay time constant from graphene

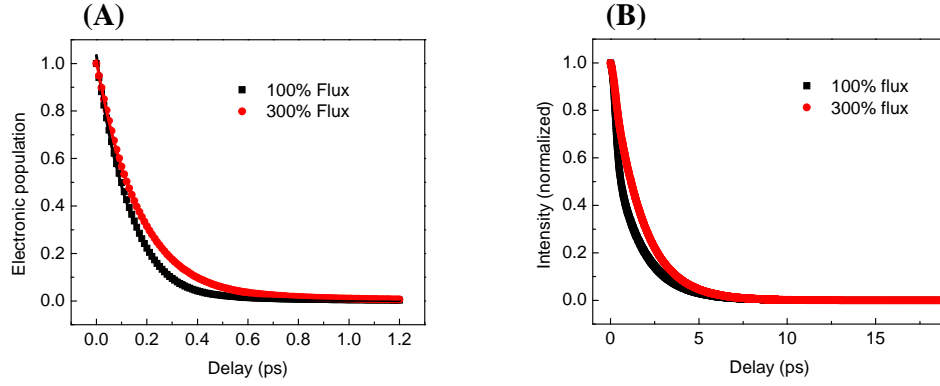
97 calculations) with factor -0.37, and $t_2=55$ ps with factor 0.37. In Supplementary

98 Figure 3F, the calculation for excitonic signal uses parameters: $t_1=120$ fs (electronic

99 decay time constant from graphene calculations) with factor -0.33, and $t_2=65$ ps with

100 factor 0.33.

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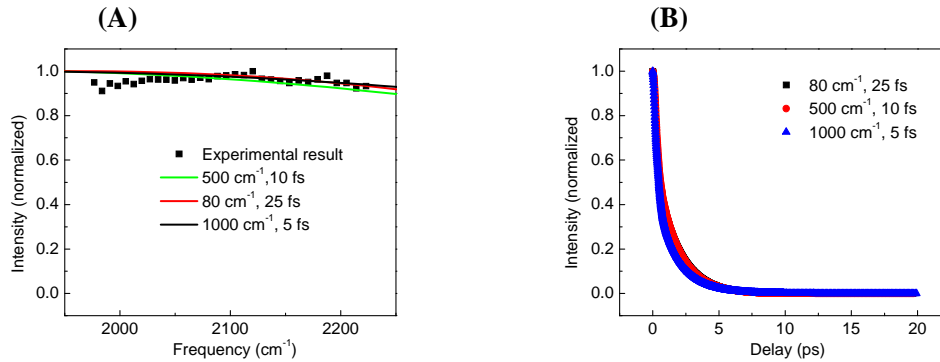
103 **Supplementary Figure 5.** *Calculated graphene 400nm-excitation/2185cm⁻¹-detection*
 104 *electronic dynamics (A) and signals (B) with two different excitation fluxes.*

105 *Calculation parameters: graphene Fermi level $\mu = -0.17$ eV , phonon*
 106 *fraction $f_{SCOPs} = 0.15$, photon flux absorbed*

107 *$F = 0.0087 \frac{J}{m^2}$ (100%flux) and $0.0261 \frac{J}{m^2}$, pump response time $t = 170$ fs, and*
 108 *electron-phonon coupling parameter $\eta = 6.0 \frac{eV}{\text{\AA}}$. The electronic decay time is 130fs*

109 *with 100% and 170fs with 300% flux.*

110

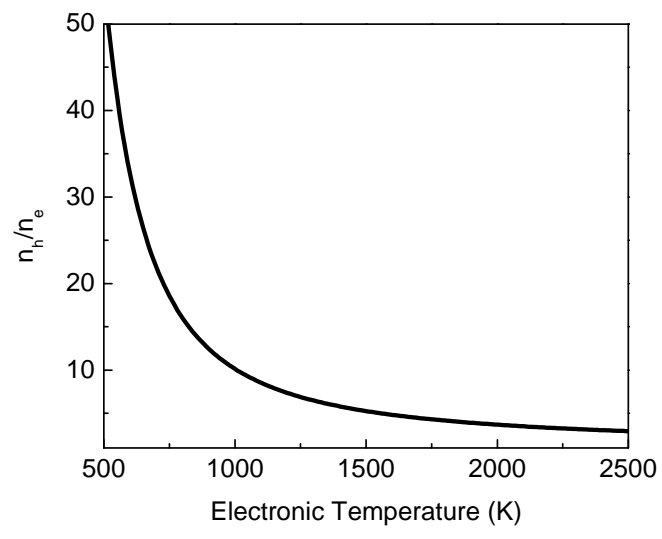


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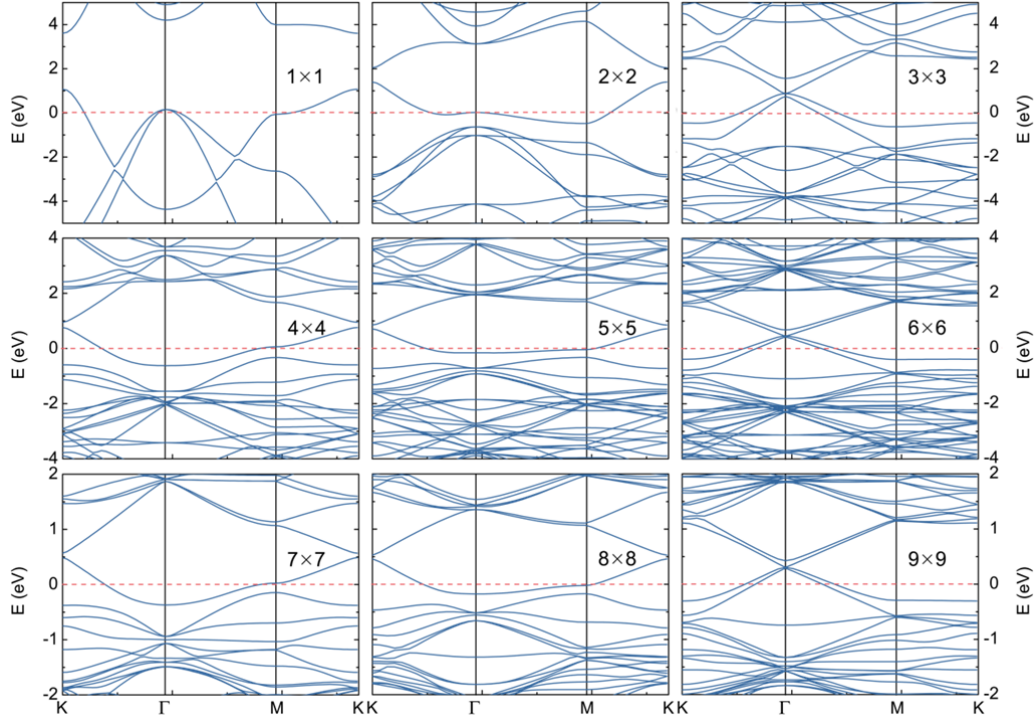
112 **Supplementary Figure 6.** *Comparisons of calculations with different electronic*
 113 *dephasing linewidths and times of graphene. Calculation parameters: graphene*

114 *Fermi level $\mu = -0.17$ eV, phonon fraction $f_{SCOPs} = 0.20$, $f_{SCOPs} = 0.15$, $f_{SCOPs} =$*
 115 *0.1 for calculations with $\Gamma = 80$ cm⁻¹ and $\tau_e = 25$ fs, $\Gamma = 500$ cm⁻¹ and $\tau_e = 10$ fs, $\Gamma =$*

116 1000 cm^{-1} and $\tau_e = 5 \text{ fs}$, respectively; photon flux absorbed
117 $F = 0.0087 \frac{\text{J}}{\text{m}^2}$ (100% flux), pump response time $t = 170 \text{ fs}$, and electron-phonon
118 coupling parameter $\eta = 6.0 \frac{\text{eV}}{\text{\AA}}$. The parameters only slightly affect the linshape and
119 dynamics.
120



121
122 **Supplementary Figure 7.** Temperature dependent n_h/n_e
123

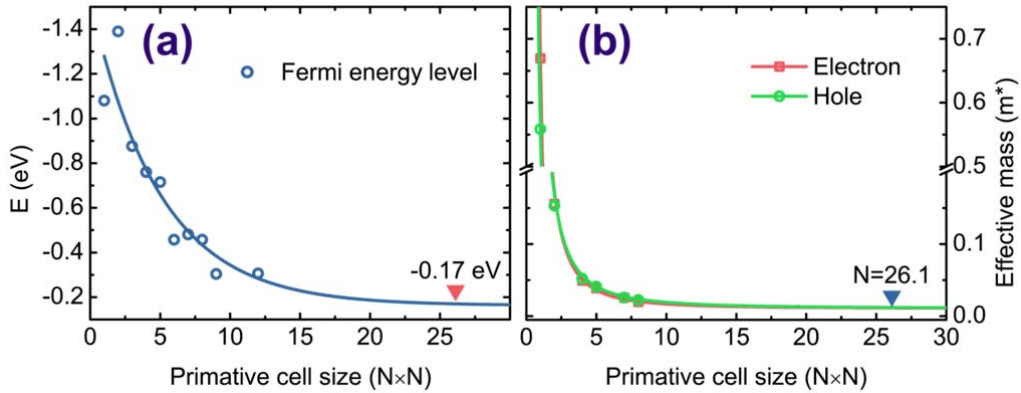


124

125 **Supplementary Figure 8.** Band structure. Electronic band structures of boron-doped

126 graphene with different primitive cell.

127



128

129 **Supplementary Figure 9.** (a) Fermi energy level of boron-doped graphene with

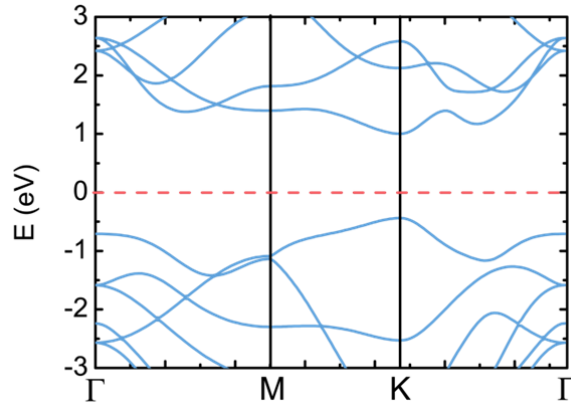
130 different primitive cell. (b) The effective mass of B-doped system for electron and hole

131 transport with different primitive cell. As the Fermi energy level is -0.17 eV, the

132 primitive cell size is 26.1×26.1 , and the corresponding hole and electron effective

133 mass is 0.0121 and 0.0116, respectively.

134

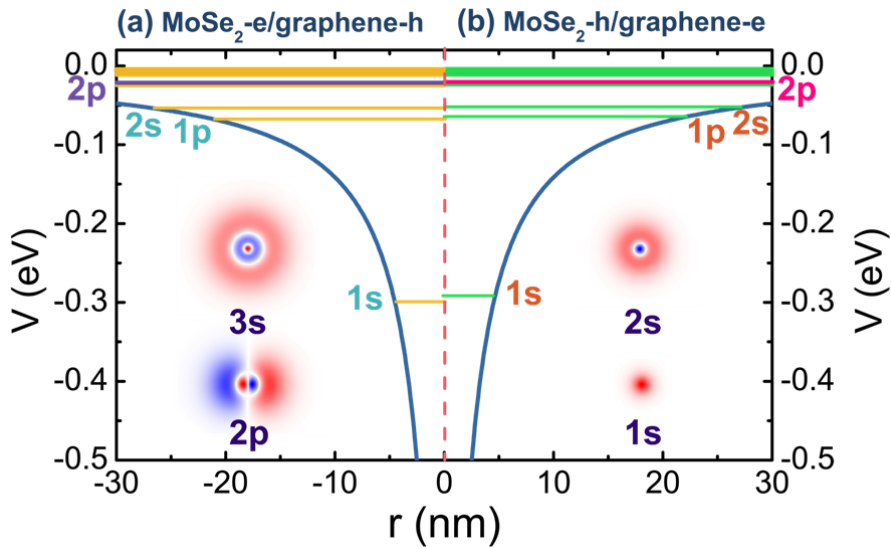


135

136 **Supplementary Figure 10.** The calculated Electronic band structures of monolayer

137 *MoSe₂*, underestimates the band gap by around 0.5 eV.

138



139

140 **Supplementary Figure 11.** Also Supplementary Figure 3D in main text. A few

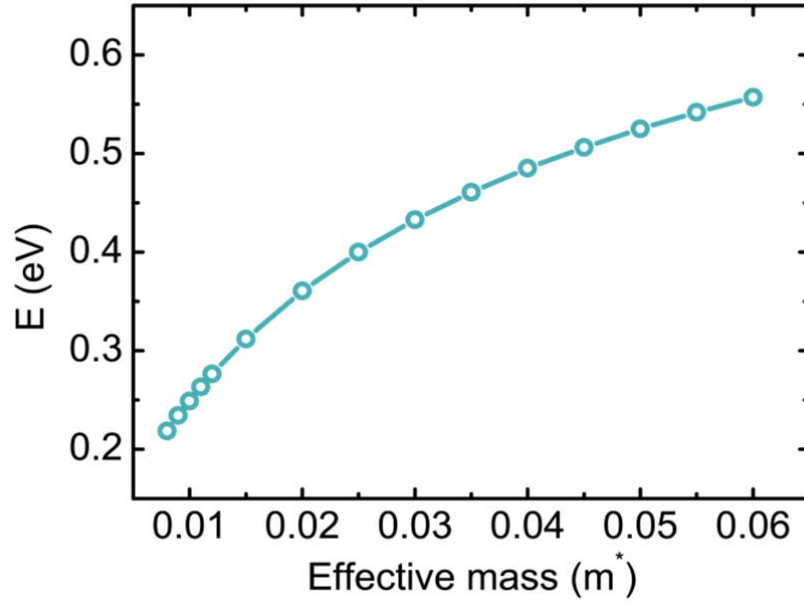
141 eigenvalues are shown on the Model Coulomb potential as a function of in-plane

142 radius for an electron-hole pair across the graphene/*MoSe₂* heterojunction van der

143 Waals interface. (a) Hole is in graphene and electron is in the *MoSe₂* monolayer. (b)

144 *Hole is in MoSe₂ and electron is in graphene. Wave functions of several states are*
 145 *also shown. (red/blue: positive/negative.)*

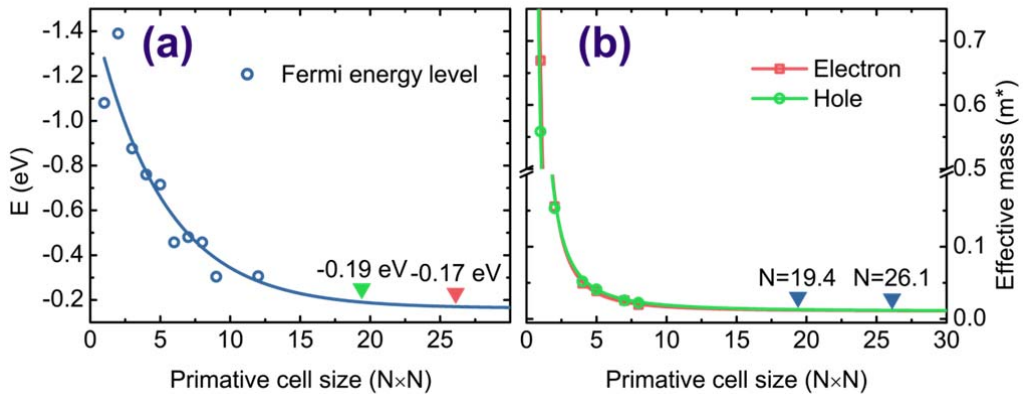
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147

148 **Supplementary Figure 12.** *The exciton transition energy (1s-2p) of graphene/MoSe₂*
 149 *correlated with effective mass of the excitonic quasi particle.*

150

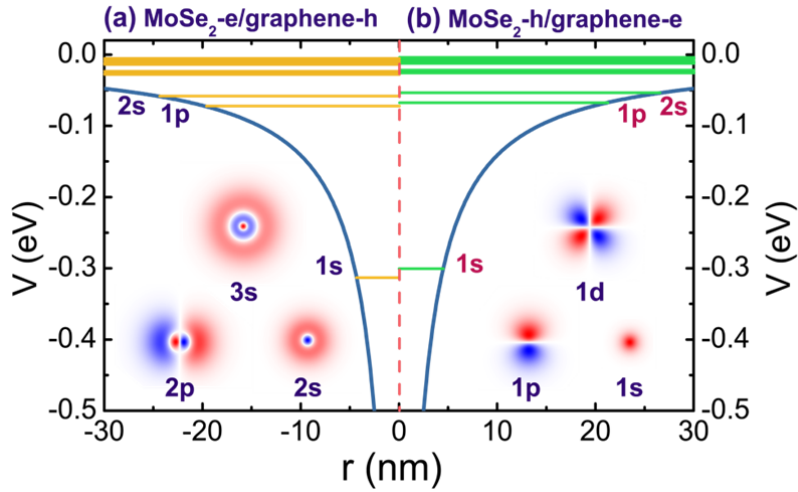


151

152 **Supplementary Figure 13.** *(a) Fermi energy level of boron-doped graphene with*
 153 *different primitive cell. (b) The effective mass of B-doped system for electron and hole*

154 transport with different primitive cell. When the Fermi energy level is -0.19 eV, the
 155 primitive cell size is 19.4×19.4 , and the corresponding hole and electron effective
 156 mass is 0.0132 and 0.0122 , respectively.

157



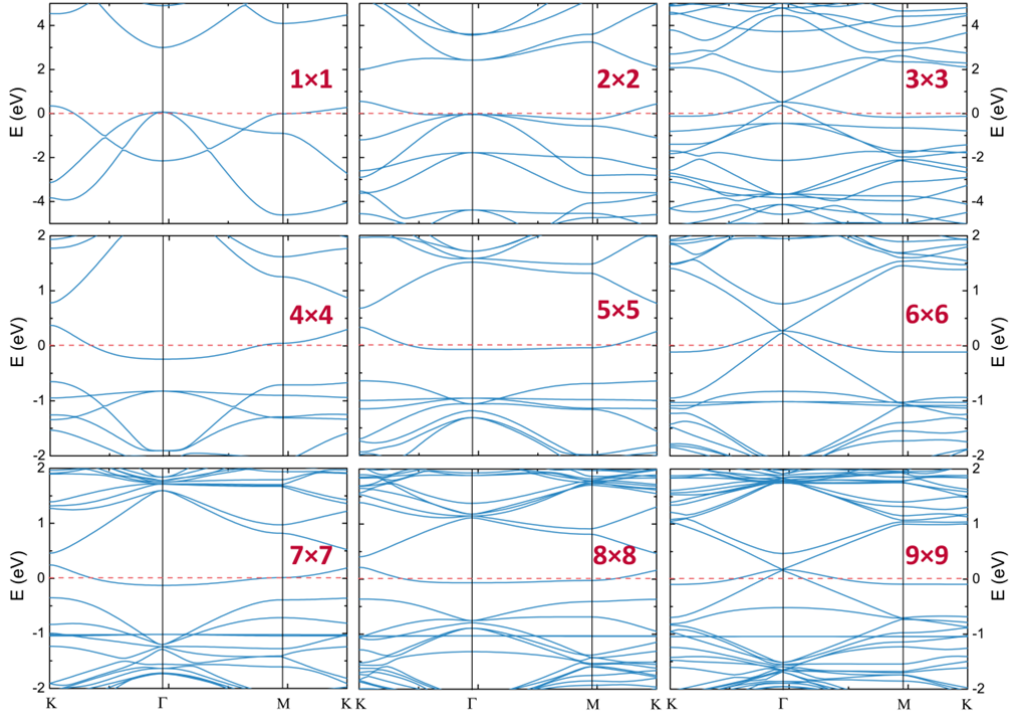
158

159 **Supplementary Figure 14.** A few eigenvalues are shown on the Model Coulomb
 160 potential as a function of in-plane radius for an electron-hole pair across the
 161 graphene/MoSe₂ heterojunction van der Waals interface. **(a)** Hole is in graphene and
 162 electron is in MoSe₂ monolayer. **(b)** Hole is in MoSe₂ slab and electron is in graphene.
 163 Several wave functions are also shown. (red/blue: positive/negative.) Fermi level
 164 -0.19 eV.

165 graphene_hole/MoSe₂_electron: $E_{1s-2p} = 0.287$ eV, $\langle \rho_{CT_{1s}} \rangle = 43.1$ Å

166 graphene_electron/MoSe₂_hole: $E_{1s-2p} = 0.276$ eV, $\langle \rho_{CT_{1s}} \rangle = 45.1$ Å

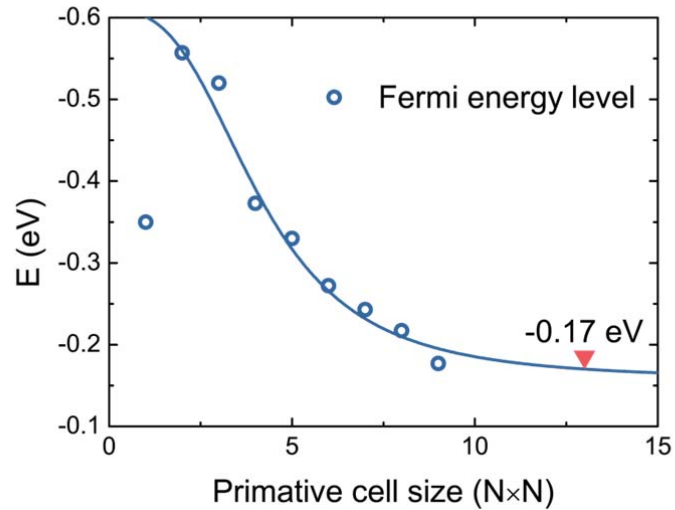
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169 **Supplementary Figure 15.** *Band structure. Electronic band structures of*
 170 *aluminum-doped graphene with different primitive cell.*

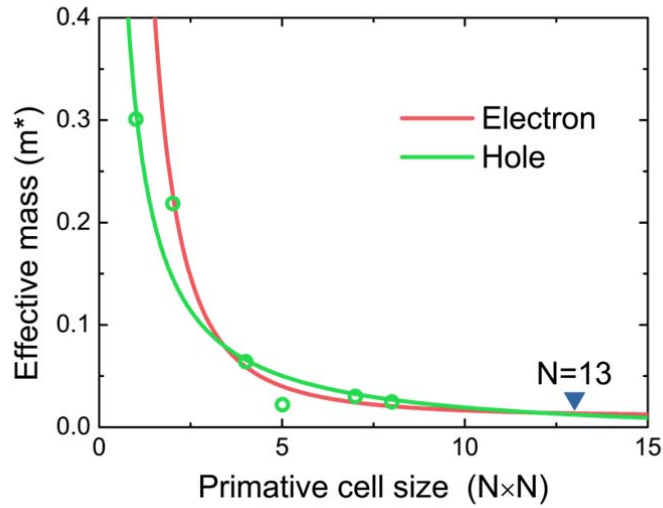
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173 **Supplementary Figure 16.** *Fermi energy level of aluminum-doped graphene with*
 174 *different primitive cell.*

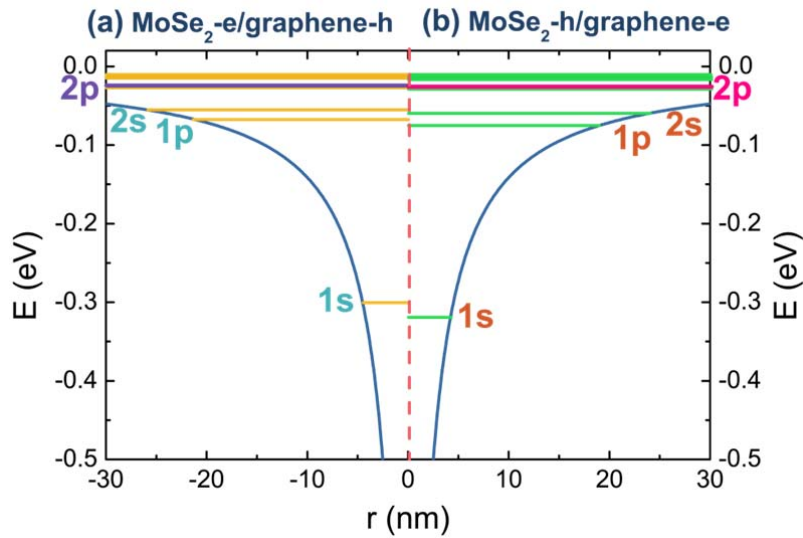
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176

177 **Supplementary Figure 17.** The effective mass of B-doped system for electron and
178 hole transport with different primitive cell. As the Fermi energy level is -0.17 eV, the
179 primitive cell size is 13×13 , and the corresponding hole and electron effective mass is
180 0.0122 and 0.0135 , respectively.

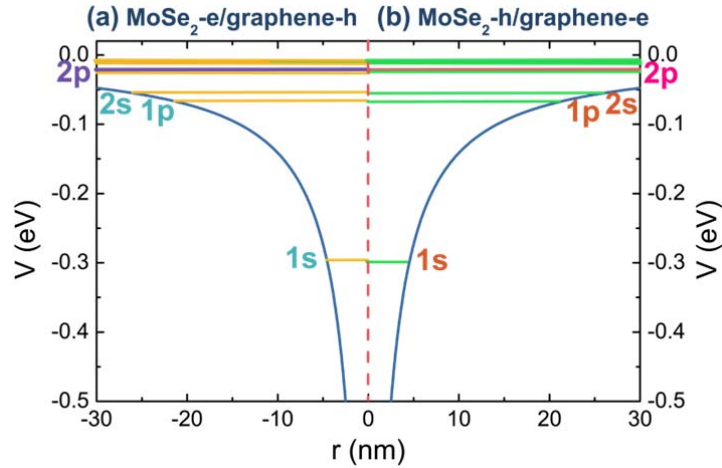
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182

183 **Supplementary Figure 18.** A few eigenvalues are shown on the Model Coulomb

184 potential as a function of in-plane radius for an electron-hole pair across the
 185 graphene/MoSe₂ heterojunction van der Waals interface. (a) Hole is in graphene and
 186 electron is in the MoSe₂ monolayer. (b) Hole is in MoSe₂ and electron is in graphene.
 187 Wave functions of several states are also shown. (red/blue: positive/negative.)
 188

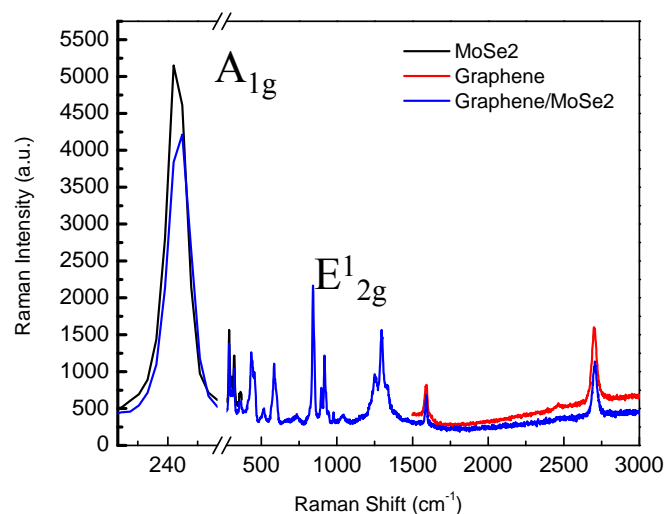


189
 190 **Supplementary Figure 19.** A few eigenvalues are shown on the Model Coulomb
 191 potential as a function of in-plane radius for an electron-hole pair across the
 192 graphene/MoSe₂ heterojunction van der Waals interface. (a) Hole is in graphene and
 193 electron is in MoSe₂ monolayer. (b) Hole is in MoSe₂ slab and electron is in graphene.
 194 Several wave functions are also shown. (red/blue: positive/negative.)

195 graphene_hole/MoSe₂_electron: $E_{1s-2p} = 0.273 \text{ eV}$, $\langle \rho_{CT_{1s}} \rangle = 46.4 \text{ \AA}$

196 graphene_electron/MoSe₂_hole: $E_{1s-2p} = 0.274 \text{ eV}$, $\langle \rho_{CT_{1s}} \rangle = 45.4 \text{ \AA}$

197



198

199 **Supplementary Figure 20.** Raman spectroscopy of graphene, MoSe₂ and

200 graphene/MoSe₂ vdW heterostructure. The frequencies of vibrational modes A_{1g} (242

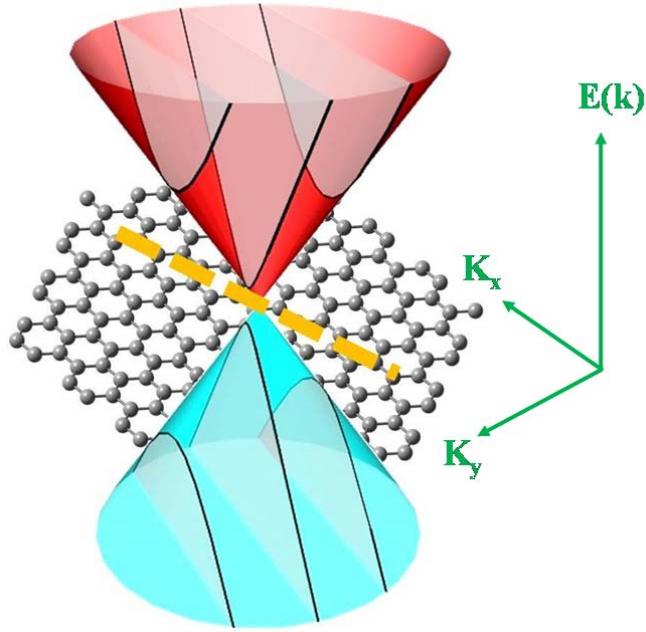
201 cm⁻¹) and E_{2g}¹ (288 cm⁻¹) of the standalone MoSe₂ sample and the heterostructures

202 match previously reported data of monolayer MoSe₂²⁶. The characteristic 2D (2690

203 cm⁻¹) and G (1580 cm⁻¹) peaks of single-layered graphene are present in both the

204 graphene/MoSe₂ heterostructures and the standalone graphene sample.

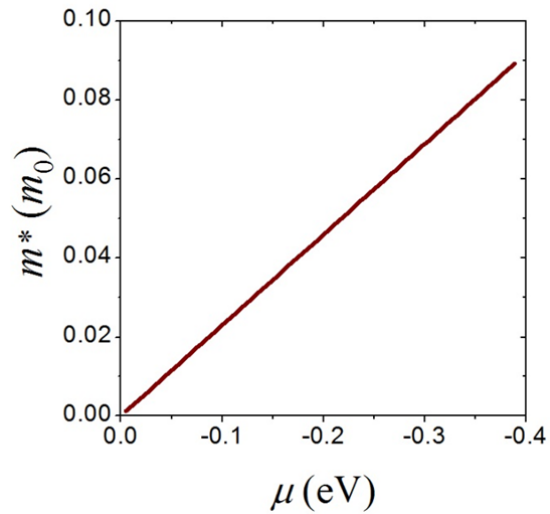
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206

207 **Supplementary Figure 21.** *Schematic illustration of the finite effective mass when k*
 208 *deviates away from the Dirac point of graphene. Some k -lines on the Dirac cones*
 209 *along a specified direction were shown as solid lines, which clearly have finite*
 210 *curvature (effective mass).*

211

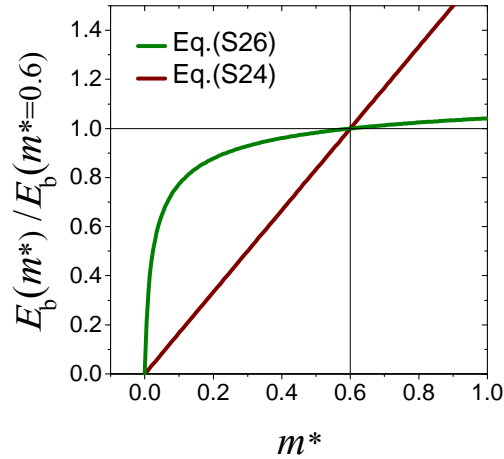


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213 **Supplementary Figure 22.** *The effective mass m^* (in units of m_0 , the mass of the free*
 214 *electron) at the level of chemical potential μ along the normal direction of Dirac cone*

215 in doped graphene. Supplementary Equation (24) is used in the plot.

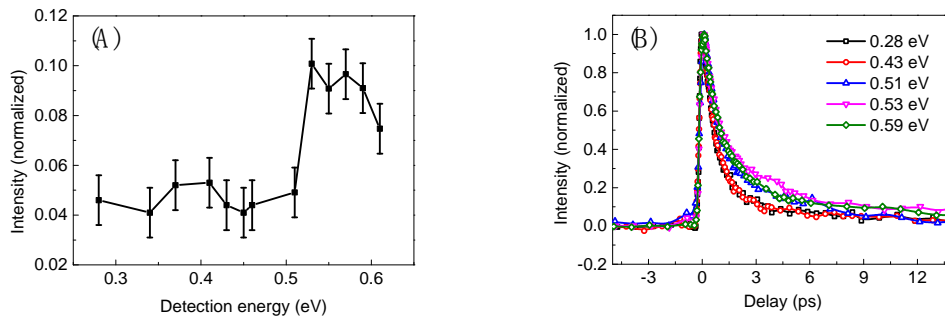
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217

218 **Supplementary Figure 23.** The exciton binding energy as a function of the effective
219 mass m^* (in units of m_0 , the mass of the free electron). The binding energy was
220 measured with respect to that of MoSe_2 ($m^* = 0.6$). Supplementary Equation (26)
221 and (28) were used in plotting the curves.

222

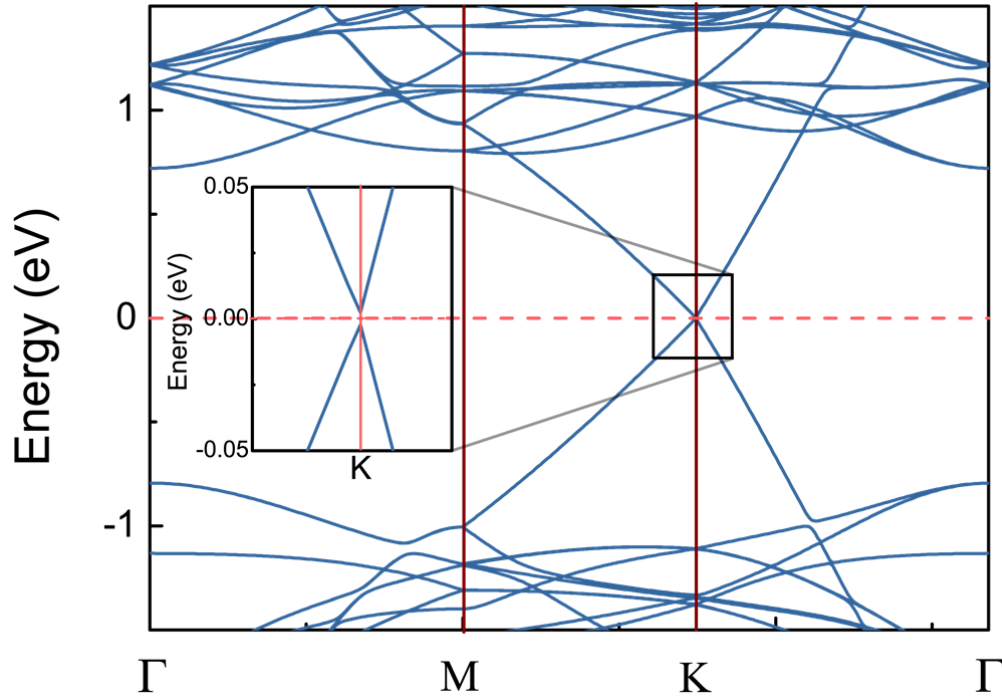


223

224 **Supplementary Figure 24.** (A) Detection frequency dependent (0.28-0.59 eV) photo
225 responses of monolayer MoSe_2 at 10 ps (averaged between 8-12 ps to minimize noise)
226 after excited with 400nm light. A resonance $1s$ - $2p$ transition peak appears at 0.55 eV.
227 The results indicate that the excitonic binding energy in monolayer MoSe_2 is about 0.6
228 eV. (B) Time dependent dynamics detected at different energy values. The transition
229 between different dynamics starts at 0.51eV. The DFG midIR output from Topas OPA

230 combined with HgCdTe detector was used for detection frequencies below 0.47eV. The
231 Idler output of a Palitra OPA combined with an InSb detector was used for detection
232 frequencies above 0.47 eV.

233



234

235 **Supplementary Figure 25.** The band structure of a graphene/MoSe₂ heterostructure.

236 The symmetry of graphene is broken, and its band structure is opened with a gap of ~

237 4.4 meV (the insert). This generates a carrier mass around 0.0004 m_0 . This

238 phenomenon was also previously observed in calculations on a graphene/MoSe₂

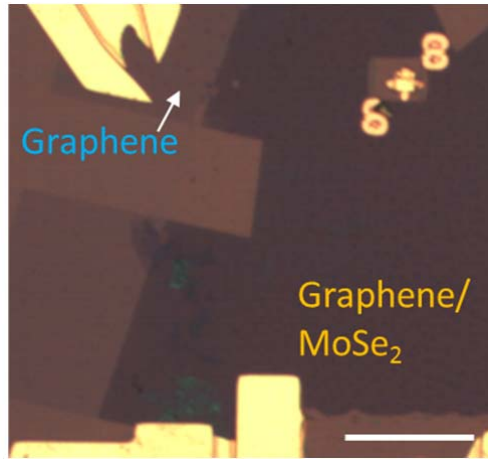
239 heterostructure (JPCC, 115, 20237, 2011). However, the gap is significantly smaller

240 than the thermal energy (26 meV) at room temperature, the effects it induced at room

241 temperature under which our experiments were conducted are expected to be very

242 small.

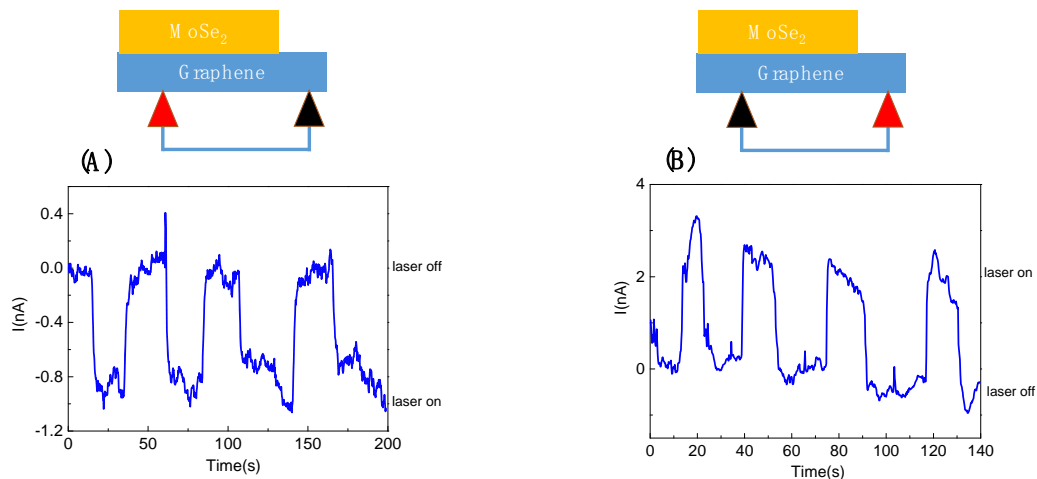
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244

245 **Supplementary Figure 26.** *MoSe₂/Graphene device used in the measurements, scale*
 246 *bar: 10 μm. MoSe₂/graphene vdW heterostructure device for photocurrent*
 247 *measurements. A monolayer of graphene (the lighter image in the center) is*
 248 *transferred on the top of A monolayer of MoSe₂ which is on a silicon wafer. Cr/Au*
 249 *(8nm/120nm) electrodes were deposited using E-beam lithography and lift-off*
 250 *technique. Both electrodes are on the top of graphene. One is on the top of graphene*
 251 *underneath which is Si (with a thin layer of SiO₂). The other electrode is on the top of*
 252 *graphene underneath which is MoSe₂.*

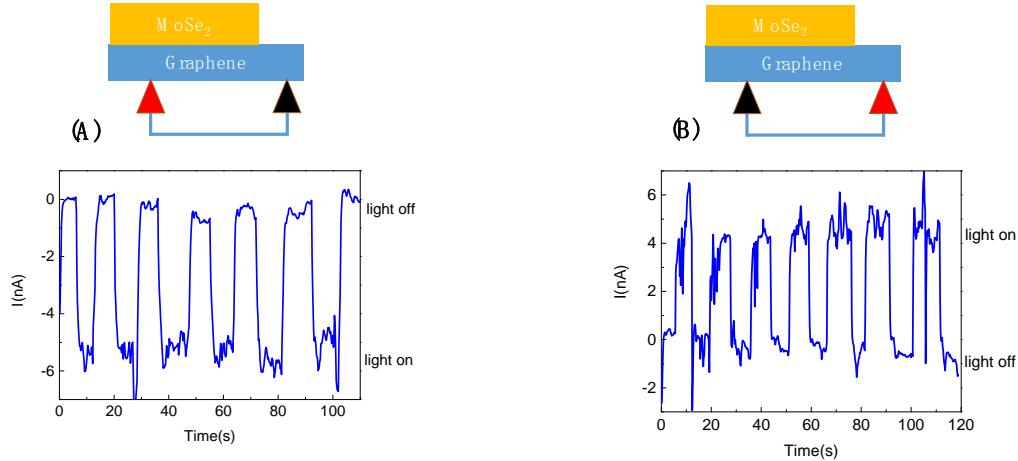
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254

255 **Supplementary Figure 27.** *Photocurrent ($I_{ph}=I_{light}-I_{dark}$) of MoSe₂/Graphene vdW*
 256 *heterostructure under 980nm excitation. The photons only excite graphene but not*
 257 *MoSe₂. Top panel: measurement configurations (A) in which the external current*
 258 *flows from graphene on the top of MoSe₂ to graphene on the top of Si (with a thin*
 259 *layer of SiO₂), and (B) the external current direction is flipped. The background*
 260 *current is subtracted. Both results unambiguously demonstrate that the*
 261 *photoexcitation generates an overall electron flow from MoSe₂ to graphene inside the*
 262 *heterostructure. After photoexcitation, electrons move from MoSe₂ to graphene inside*
 263 *the heterostructure. Therefore, more electrons exist in the area of graphene directly on*
 264 *the top of MoSe₂ than in the area of graphene on the top of Si (with a thin layer of*

265 SiO_2). This results in external current flowing from graphene on the top of Si to
 266 graphene on the top of MoSe_2 . The photocurrent measurements were conducted with
 267 a semiconductor parameter analyzer (Agilent 4155C) and a probe station. The
 268 source/drain bias $V_{sd} = 0$ and the gate voltage $V_g = 0$ in all tests. The two
 269 measurements were conducted under slightly different laser focus sizes.
 270



271
 272 **Supplementary Figure 28.** Photocurrent ($I_{ph} = I_{light} - I_{dark}$) of $\text{MoSe}_2/\text{Graphene}$ vdW
 273 heterostructure under 405nm excitation. The photons excite both graphene and MoSe_2 .
 274 Top panel: measurement configurations (A) in which the external current flows from
 275 graphene on the top of MoSe_2 to graphene on the top of Si (with a thin layer of SiO_2),
 276 and (B) the external current direction is flipped. The background current is subtracted.
 277 Both results unambiguously demonstrate that the photoexcitation generates an overall
 278 electron flow from MoSe_2 to graphene inside the heterostructure. After
 279 photoexcitation, electrons move from MoSe_2 to graphene inside the heterostructure.
 280 Therefore, more electrons exist in the area of graphene directly on the top of MoSe_2
 281 than in the area of graphene on the top of Si (with a thin layer of SiO_2). This results in
 282 external current flowing from graphene on the top of Si to graphene on the top of
 283 MoSe_2 . The photocurrent measurements were conducted with a semiconductor
 284 parameter analyzer (Agilent 4155C) and a probe station. The source/drain bias V_{sd}
 285 $= 0$ and the gate voltage $V_g = 0$ in all tests. Similar photocurrent phenomena was also
 286 observed in $\text{MoS}_2/\text{Graphene}$ vdW heterostructures (ref: Scientific Reports 4, 3826
 287 (2014)).
 288

289

Supplementary Notes

290 **Supplementary Note 1: Optical conductivity of graphene**

291 The tight-binding Hamiltonian for the cone-like band structure of graphene is
292 written as

$$293 \quad H = \begin{bmatrix} 0 & \hbar v_F (k_x + ik_y) \\ \hbar v_F (k_x - ik_y) & 0 \end{bmatrix}, \quad (1)$$

294 where v_F is the Fermi velocity, and k_x and k_y are 2D components of the electronic
295 wave vector \mathbf{k} . With the Hamiltonian, the optical conductivity $[\sigma(\omega)]$ is written as the
296 sum of the interband conductivity $[\sigma_{\text{inter}}(\omega)]$ and the intraband conductivity $[\sigma_{\text{intra}}(\omega)]$
297 given below^{1, 2, 3, 4}:

$$298 \quad \begin{aligned} \sigma(\omega) &= \sigma_{\text{inter}}(\omega) + \sigma_{\text{intra}}(\omega) \\ \sigma_{\text{inter}}(\omega) &= i \frac{e^2 \hbar \omega}{\pi \hbar} \int_0^{+\infty} d\varepsilon \frac{1}{(2\varepsilon)^2 - (\hbar\omega + i\Gamma)^2} [f_{\text{FD}}(\varepsilon - \mu) - f_{\text{FD}}(-\varepsilon - \mu)], \\ \sigma_{\text{intra}}(\omega) &= i \frac{e^2 / \pi \hbar}{\hbar\omega + i\hbar/\tau_e} \int_0^{+\infty} d\varepsilon [f_{\text{FD}}(\varepsilon - \mu) + 1 - f_{\text{FD}}(-\varepsilon - \mu)] \end{aligned} \quad (2)$$

299 where f_{FD} is the Fermi-Dirac distribution function, μ is the chemical potential (Fermi
300 energy), and e is the elementary charge. Γ is the broadening of the interband
301 transitions, and τ_e is the relaxation time due to intraband carrier scattering. In
302 literature^{5, 6}, Γ lies between 0.01 and 0.06 eV while τ_e is in the range of 5 ~ 40 fs.
303 Here, considering that electronic motions typically occur within a few fs rather tens of
304 fs, we use the fast values $\Gamma = 0.062$ eV (500 cm^{-1}) and $\tau_e = 10$ fs throughout the study.
305 Nonetheless, effects of these two parameters are also tested and listed in
306 Supplementary Figure 6. It turns out that the results are not very sensitive to the
307 parameter selections.

308 Applying the Fresnel equations, the change of optical transmission due to the

309 existence of graphene is given as

$$310 \quad \frac{\Delta T_s}{T_0} \approx -\frac{2}{\cos \theta + n_{\text{sub}} \cos \theta''} \sqrt{\frac{\mu_0}{\epsilon_0}} \text{Re}[\sigma(\omega)] \quad (3)$$

311 for s-polarized light. θ is the incident angle, and θ'' is the incident angle in the
 312 substrate. n_{sub} is the refractive index of the substrate. ϵ_0 and μ_0 are vacuum
 313 permittivity and permeability, respectively. For p-polarized light, the change of optical
 314 transmission is

$$315 \quad \frac{\Delta T_p}{T_0} \approx -\frac{2 \cos \theta \cos \theta''}{n_{\text{sub}} \cos \theta + \cos \theta''} \sqrt{\frac{\mu_0}{\epsilon_0}} \text{Re}[\sigma(\omega)]. \quad (4)$$

316 In our experiments, $\theta = 0^\circ$, so we have

$$317 \quad \frac{\Delta T}{T_0} \approx -\frac{2}{1 + n_{\text{sub}}} \sqrt{\frac{\mu_0}{\epsilon_0}} \text{Re}[\sigma(\omega)], \quad (5)$$

318 where $n_{\text{sub}} = 1.39$ for the used substrate (CaF₂).

319

320 **Supplementary Note 2: Heat transfer between electrons and phonons in**
 321 **graphene**

322 Strongly coupled optical phonons (COPs) are in-plane optical phonons for which
 323 intra- and inter-valley carrier scattering can simultaneously conserve energy and
 324 momentum⁷. The considered SCOPs include phonons near the Γ -point with energy \sim
 325 200 meV (for intra-valley carrier scattering) and those near the K -point with energy
 326 \sim 150 meV (for inter-valley carrier scattering). The interaction between phonons and
 327 electrons/holes in graphene is described by a deformation potential theory,^{8,9} and the
 328 transition matrix element is determined to be

329
$$M_{\mathbf{k}',\mathbf{k}}^{(\text{TO}\&\text{LO})} \approx 3\eta \sqrt{\frac{\hbar}{4M_C \omega_{\text{phonon}}}}, \quad (6)$$

330 where η is the electron-phonon coupling parameter, and M_C is the mass of a carbon
 331 atom. The energy dispersion of the transverse optical (TO) and longitudinal optical
 332 (LO) phonon modes is ignored, and Supplementary Equation (6) accounts for the total
 333 contribution from TO and LO at Γ - or K -points. Both emission and absorption of the
 334 phonons are considered under the second quantization. The probability of scattering
 335 from \mathbf{k} to \mathbf{k}' by the SCOPs is

336
$$\begin{aligned} W_{\mathbf{k}',\mathbf{k}}^{(\text{TO}\&\text{LO})} &= \frac{2\pi}{\hbar} \left| M_{\mathbf{k}',\mathbf{k}}^{(\text{TO}\&\text{LO})} \right|^2 \left[N_{\mathbf{q}} \delta(\varepsilon_{\mathbf{k}} - \varepsilon_{\mathbf{k}'} + \hbar\omega_{\text{phonon}}) + (N_{\mathbf{q}} + 1) \delta(\varepsilon_{\mathbf{k}} - \varepsilon_{\mathbf{k}'} - \hbar\omega_{\text{phonon}}) \right] \\ &= \frac{9\pi\eta^2}{2M_C \omega_{\text{phonon}}} \left[N_{\mathbf{q}} \delta(\varepsilon_{\mathbf{k}} - \varepsilon_{\mathbf{k}'} + \hbar\omega_{\text{phonon}}) + (N_{\mathbf{q}} + 1) \delta(\varepsilon_{\mathbf{k}} - \varepsilon_{\mathbf{k}'} - \hbar\omega_{\text{phonon}}) \right] \end{aligned} \quad (7)$$

337 where $N_{\mathbf{q}}$ is the phonon occupation number described by the Bose-Einstein
 338 distribution. The first and second terms in the square brackets correspond to the
 339 phonon emission and absorption processes, respectively. Integrating \mathbf{k}' gives the
 340 scattering rate of \mathbf{k} as

341
$$\begin{aligned} R_{\mathbf{k} \rightarrow * } &= \int W_{\mathbf{k}',\mathbf{k}}^{(\text{TO}\&\text{LO})} \frac{A}{(2\pi)^2} d^2\mathbf{k}' \\ &= \frac{9\eta^2}{2\hbar^2 \rho v_F^2 \omega_{\text{phonon}}} \left[(\varepsilon_{\mathbf{k}} + \hbar\omega_{\text{phonon}}) N_{\mathbf{q}} + (\varepsilon_{\mathbf{k}} - \hbar\omega_{\text{phonon}}) (N_{\mathbf{q}} + 1) \right] \end{aligned} \quad (8)$$

342 where ρ is the 2D mass density of graphene. Therefore, the rate of phonon emission
 343 per unit area is

344
$$\begin{aligned} \frac{dN_{\text{emission}}}{dt} &= \int \frac{9\eta^2}{2\hbar^2 \rho v_F^2 \omega_{\text{phonon}}} (\varepsilon_{\mathbf{k}} - \hbar\omega_{\text{phonon}}) (N_{\mathbf{q}} + 1) f_{\text{FD}}(\varepsilon_{\mathbf{k}}) (1 - f_{\text{FD}}(\varepsilon_{\mathbf{k}} - \hbar\omega_{\text{phonon}})) \frac{1}{(2\pi)^2} d^2\mathbf{k} \\ &= \frac{9\eta^2}{4\pi(\hbar v_F)^4 \rho \omega_{\text{phonon}}} (N_{\mathbf{q}} + 1) \int \varepsilon_{\mathbf{k}} (\varepsilon_{\mathbf{k}} - \hbar\omega_{\text{phonon}}) f_{\text{FD}}(\varepsilon_{\mathbf{k}}) (1 - f_{\text{FD}}(\varepsilon_{\mathbf{k}} - \hbar\omega_{\text{phonon}})) d\varepsilon_{\mathbf{k}} \end{aligned}$$

346 (9)

347 and the rate of phonon absorption is

$$348 \quad \frac{dN_{\text{adsorption}}}{dt} = \frac{9\eta^2}{4\pi(\hbar v_F)^4 \rho \omega_{\text{phonon}}} N_q \int \epsilon_{\mathbf{k}} (\epsilon_{\mathbf{k}} + \hbar \omega_{\text{phonon}}) f_{\text{FD}}(\epsilon_{\mathbf{k}}) (1 - f_{\text{FD}}(\epsilon_{\mathbf{k}} + \hbar \omega_{\text{phonon}})) d\epsilon_{\mathbf{k}},$$

349 (10)

350 which are used to simulate the heat transfer between electrons/holes and SCOPs.

351 The heat capacity of the SCOPs is described by the Einstein model, and the
 352 fraction of Brillouin zone filled by the SCOPs, f_{SCOPs} , is assumed to be temperature
 353 independent for simplicity⁴. The energy stored in the SCOP subsystem relaxes at a
 354 rate of $1/\tau_{\text{ph}}$ to lower energy phonons^{4,10}. The value of f_{SCOPs} and τ_{ph} are determined
 355 by fitting the experimental probe signal profiles.

356

357 **Supplementary Note 3: Fermi level of graphene in heterostructure**

358 Graphene used in our experiments are p-doped with Fermi level -0.19 eV
 359 determined by fitting the transient spectra and dynamics in Supplementary Figure 1
 360 and Supplementary Figure 3E. MoSe₂ is n-doped. When the two monolayers were
 361 placed together to form a heterostructure, some electrons transfer from MoSe₂ to
 362 graphene¹¹, raising the graphene Fermi level to -0.17 eV. The Fermi level of graphene
 363 in heterostructure is obtained by fitting the transient IR spectrum of heterostructure at
 364 time 0 with 1.03 eV excitation (Supplementary Figure 2). The carrier density of
 365 graphene is

$$366 \quad n = \int \frac{1}{(2\pi)^2} f(\epsilon - \mu) d^2\mathbf{k} = \int \frac{k}{2\pi} f(\epsilon - \mu) dk = \int_0^{+\infty} \frac{\epsilon}{2\pi(\hbar v_F)^2} f(\epsilon - \mu) d\epsilon, \quad (11)$$

367 When $|\mu| \gg k_B T$, Supplementary Equation 11 can be simplified into (degenerate spin
368 and KK'):

$$369 \quad n \approx \frac{\mu^2}{\pi(\hbar v_F)^2}. \quad (12)$$

370 According to Supplementary Equation 11&12, the Fermi level increase indicates
371 that about $5 \times 10^{11} \frac{e}{cm^2}$ have transferred from MoSe₂ to graphene after they form a
372 heterostructure.

373

374 **Supplementary Note 4: Calculated IR response of graphene with Fermi level**
375 **-0.17 eV**

376 In the heterostructure excited with 1.03 eV photons, the initial response is only
377 from graphene. Because of fast interlayer charge transfers, some excited carriers
378 move to MoSe₂ very rapidly (< 50 fs) and effectively reduce the electronic
379 temperature in the graphene layer. In our calculations, we consider this problem
380 equivalent to the reduction of excitation flux because the charge transfer and
381 electronic thermalization are extremely fast (< 50 fs), much faster than the exciton
382 formation and electron/phonon couplings. Under ideal fast equilibrium, at most 50%
383 of the excitation energy is transferred to MoSe₂ by the transferred carriers. However,
384 in our experiments, this is obviously not likely because very few electrons can reach
385 CBM of MoSe₂. Therefore, we set the upper limit of flux reduction as 1/3. Calculation
386 results for the exciton formation time with no flux reduction to 1/3 reduction are all
387 within experimental uncertainty.

388

389 **Supplementary Note 5: Excitonic signal from direct experimental data**
390 **subtraction**

391 The excitonic signal presented in Supplementary Figure 3F is obtained by
392 considering pump flux and Fermi level changes. Since these two parameters don't
393 change significantly, we also directly subtract the signal of monolayer graphene from
394 that of heterostructure (both presented in Supplementary Figure 3E) to obtain the
395 excitonic signal. Kinetic analyses on the result by this method are plotted in
396 Supplementary Figure 4. The results from this method and that from the more
397 accurate method in Supplementary Figure 3F turn out to be essentially the same.

398

399 **Supplementary Note 6: Larger flux results in slower graphene signal decay**

400 MoSe₂ absorbs 14% and graphene absorbs 4.14% of 3.1 eV photons. Under ideal
401 fast equilibrium, ~9% of excitation photons would be in either layer of the
402 heterostructure. However, because the graphene Fermi level is between MoSe₂ VBM
403 and CBM, more carriers are expected to flow into graphene. From photoluminescence,
404 we can see that around 15% of free carriers that are originally excited in MoSe₂
405 remain in MoSe₂. Estimated from the results with 1.03 eV photons, about 33% free
406 carriers form interlayer excitons. These two portions together account for 6% of the
407 18.14% total absorption. Therefore, the excitation flux the graphene layer gains from
408 interlayer charge transfers must not exceed 12.5% (300% of 4.14%) besides direct
409 absorption. The charge transfers results in higher electronic temperature and slower
410 dynamics in graphene.

411

412 **Supplementary Note 7: MoSe₂ bandgap**

413 According to STM measurements¹², the bandgap of MoSe₂ monolayer samples
414 on different substrates varies. On graphite, it is 1.94 eV, whereas on bilayer graphene
415 it is 2.18 eV. The PL peak central frequency also varies from 1.67 eV to 1.63 eV at
416 77K. The determined binding energy is 0.55 eV, slightly higher than the theoretical
417 value 0.47 eV¹³. The PL central frequency of our sample at room temperature is 1.56
418 eV (on graphene). 1.56 + 0.47 = 2.03 eV. Considering all results from literature and
419 our own experiments, we adopt around 2.0 eV as the bandgap of MoSe₂ in the
420 heterostructures measured in our experiments.

421

422 **Supplementary Note 8: Ratio of electron/hole transferred from graphene to**

423 **MoSe₂**

424 The population of electrons in graphene that have higher energy than the CBM
425 value of MoSe₂ and thus are capable of interlayer transferring is calculated as

426
$$n_e(T; E_{\text{CBM}}) = \int_{E_{\text{CBM}}}^{+\infty} \frac{\mathcal{E}}{2\pi(\hbar v_F)^2} f(\mathcal{E} - \mu) d\mathcal{E}. \quad (13)$$

427 Similarly, the population of holes being capable of interlayer transferring is

428
$$n_h(T; E_{\text{VBM}}) = \int_{-\infty}^{E_{\text{VBM}}} \frac{\mathcal{E}}{2\pi(\hbar v_F)^2} f(-\mathcal{E} + \mu) d\mathcal{E}. \quad (14)$$

429 By subtracting the populations before pumping ($T_0=300\text{K}$), the ratio of n_h/n_e
430 determined is plotted in Supplementary Figure 7. With 100% flux of 1200nm
431 excitation, the electronic temperature reaches 1583K. At this temperature, n_h/n_e is 4.9.

432 With 2/3 flux, the electronic temperature reaches 1399 K and n_h/n_e is 5.8. Because
433 charge transfers result in a lower temperature, the exact n_h/n_e should lie between these
434 two values. Interlayer charge transfers and electronic thermalization in graphene are
435 extremely fast (<50 fs)^{4, 14, 15}. Thus, they can be treated as semi-instantaneous,
436 compared to the electron/hole recombination (100~200 fs) in graphene and the
437 formation of interlayer excitons (~500 fs). Therefore, the transfer of photo-excited
438 charge carriers between two layers forms a quasi-equilibrium in which the
439 electron/hole population ratio in MoSe₂ is approximately equal to that in graphene.
440 Once the charge carriers form interlayer excitons, the interlayer quasi-equilibrium is
441 perturbed and more carriers are transferred to MoSe₂ as the charge carriers in
442 graphene are continuously adjusting their population distributions. This process
443 makes sure that even though the hole population in graphene below the VBM of
444 MoSe₂ is only a small portion of the total number photo-generated, the total number
445 of transferred holes is still substantial. The process results in that the majority of
446 interlayer excitons have their holes in the MoSe₂ layer and electrons in graphene. The
447 estimated exciton density is about $0.45 \times 10^{12} \text{ cm}^{-2}$ and $0.58 \times 10^{12} \text{ cm}^{-2}$ under the
448 excitation photon energy 1.03 eV and 3.1 eV, respectively. Another possible
449 mechanism can also lead to that electrons stay in graphene and holes prefer MoSe₂.
450 According to literature¹¹, charged impurities in MoSe₂ and graphene can produce an
451 effective electric field that is from graphene to MoSe₂. Such an electric field can
452 provide driving force for holes to move to MoSe₂ and for electrons to go to graphene.
453

454 **Supplementary Note 9: Kinetic analysis**

455 To analyze the electron/hole gas transformation dynamics, we create a kinetic
 456 model. In the model, interlayer charge transfers and electronic thermalization in
 457 graphene are assumed to be semi-instantaneous. Free carriers (y) in graphene decay
 458 with a total rate constant k_3 . The interlayer exciton (x) formation rate constant is k_1
 459 and the exciton decay rate constant is k_2 . The model can be expressed as



461 The solution of the kinetic model is

462
$$\begin{aligned} \frac{dx}{dt} &= k_1[y] - k_2[x] \\ \frac{dy}{dt} &= -k_3[y] \end{aligned} \quad , \quad (16)$$

$$x(t) = \frac{k_1}{k_3 - k_2} (1 - e^{-(k_3 - k_2)t}) e^{-k_2 t}$$

463 where k_3 is obtained from the electronic decay of graphene. k_1 and k_2 are
 464 obtained from fitting the experimental excitonic signal.

465 In Supplementary Figure 3F, $1/k_1 = 364 \text{ fs}$, $1/k_2 = 65 \text{ ps}$, $1/k_3 = 120 \text{ fs}$.

466 In Supplementary Figure 4F, $1/k_1 = 379 \text{ fs}$, $1/k_2 = 55 \text{ ps}$, $1/k_3 = 170 \text{ fs}$.

467 The binding energy of MoSe₂ intralayer exciton is >0.4 eV, which exceeds the
 468 photon energy of our IR probe. In addition, the optical response of free carriers in
 469 MoSe₂ is only 1/6 of the free carriers in graphene. Therefore, in the kinetic model the
 470 contribution from free carriers in MoSe₂ is ignored. The response ratio 1/6 is
 471 calculated based on experimental results: with the same 3.1 eV excitation, 14%
 472 photons are absorbed by MoSe₂, producing a signal size of -0.006, whereas 4.14%

473 photons are absorbed by graphene producing a signal size of -0.011.
 474 $0.011/0.006*14\%/4.14\% \sim 6$.

475

476 **Supplementary Note 10: Exciton and free carrier optical responses**

477 The free carrier optical conductivity from the Drude model¹⁶ is

$$478 \quad \sigma_{free} = \frac{ne^2\tau}{\mu(\omega^2\tau^2 + 1)} = \frac{n\hbar e^2\Gamma_D}{\mu[(\hbar\omega)^2 + \Gamma_D^2]}, \quad (17)$$

479 where n is the carrier population, μ is the reduced mass and Γ_D is the scattering
 480 (dephasing) line width. The 1s-2p transition optical conductivity¹⁷ is

$$481 \quad \sigma_{1s-2p} = \frac{n_X \hbar e^2 \Gamma_{1s-2p} \cdot (\hbar\omega)^2 f_{1s-2p}}{\mu \left[((\hbar\omega)^2 - E_{res}^2)^2 + (\hbar\omega \Gamma_{1s-2p})^2 \right]}, \quad (18)$$

482 where E_{res} is the resonant energy. The optical conductivity ratio between the 1s-2p
 483 transition and free carrier is therefore:

$$484 \quad \frac{\sigma_{1s-2p}}{\sigma_{e-h}} \approx \frac{2n_X f_{1s-2p}}{\Gamma_{1s-2p}} \bigg/ \frac{n_{e-h} \Gamma_D}{(\hbar\omega)^2} \approx \frac{2f_{1s-2p} (\hbar\omega)^2}{\Gamma_D \Gamma_{1s-2p}}, \quad (19)$$

485 where f_{1s-2p} is the oscillator strength that can be calculated in the following¹⁸:

$$486 \quad \begin{aligned} f_{1s-n} &= \frac{2\mu a^2}{\hbar^2} (E_n - E_0) \frac{\left(\frac{n+1}{2}\right)^5 n^{2n-3}}{(n+1)^{2n+5}} \\ &= \frac{2\mu a^2}{\hbar^2} \cdot \frac{e^2 \lambda}{8\pi \epsilon_0 \epsilon_r a} \left[\frac{1}{(1/2)^2} - \frac{1}{(n+1/2)^2} \right] \frac{\left(\frac{n+1}{2}\right)^5 n^{2n-3}}{(n+1)^{2n+5}} \\ &= \frac{\mu e^2 \lambda a}{4\pi \epsilon_0 \epsilon_r \hbar^2} \left[\frac{1}{(1/2)^2} - \frac{1}{(n+1/2)^2} \right] \frac{\left(\frac{n+1}{2}\right)^5 n^{2n-3}}{(n+1)^{2n+5}} \\ &= \frac{\mu e^2 \lambda}{4\pi \epsilon_0 \epsilon_r \hbar^2} \cdot \frac{4\pi \hbar^2 \epsilon_0 \epsilon_r}{e^2 \mu \lambda} \left[\frac{1}{(1/2)^2} - \frac{1}{(n+1/2)^2} \right] \frac{\left(\frac{n+1}{2}\right)^5 n^{2n-3}}{(n+1)^{2n+5}} \\ &= \left[\frac{1}{(1/2)^2} - \frac{1}{(n+1/2)^2} \right] \frac{\left(\frac{n+1}{2}\right)^5 n^{2n-3}}{(n+1)^{2n+5}} \end{aligned} \quad (20)$$

487 Based on Supplementary Equation 18, $f_{1s-2p} = 0.21$, slightly different from the
 488 value 0.32 derived from a non-hydrogen model¹⁷. Γ_{1s-2p} is experimentally determined

489 to be 280 cm^{-1} . Using the ratio $\Gamma_D / \Gamma_{1s-2p} = 6$ adopted in literature¹⁸, $\frac{\sigma_{1s-2p}}{\sigma_{e-h}}$ is
490 calculated to be 3.6. However, $\frac{\sigma_{1s-2p}}{\sigma_{e-h}}$ calculated here is for exciton and free carrier in
491 the same material. In our experiments, the interlayer exciton is within two very
492 different materials of which the free carrier response ratio is $1/6 = \text{MoSe}_2/\text{graphene}$. It
493 is reasonable to expect that σ_{1s-2p} of the $\text{MoSe}_2/\text{graphene}$ interlayer exciton is 1~3.5
494 times (it doesn't go to 6 because the 1s-2p transition is a bound-bound transition and
495 graphene cannot affect it in a way like free carriers) of σ_{1s-2p} of MoSe_2 , giving
496 $\frac{\sigma_{1s-2p\text{-interlayer}}}{\sigma_{e-h\text{-graphene}}} = 0.6 \sim 2.1$. Normalized with the ratio, the excitonic formation time
497 constants obtained from fitting the experimental results in Supplementary Figure
498 3F&4F are 218 fs~764 fs (Supplementary Figure 3F) and 227 fs~796 fs
499 (Supplementary Figure 4F), respectively.

500

501 **Supplementary Note 11: Exciton 1s-2p transition energy calculations**

502 Methods:

503 First-principle calculations in this work are conducted using DFT methods
504 implemented in the Vienna *ab initio* simulation package (VASP)¹⁹. The
505 projector-augmented-wave (PAW) pseudopotentials and the generalized gradient
506 approximation of Perdew, Burke, and Ernzerhof (PBE) for exchange-correlation
507 functional are adopted in our simulations²⁰. In order to study the influence of Fermi
508 level on the electronic properties for hole-doped graphene system, we conducted the
509 first-principle investigation on the electronic band structures of graphene sheets

510 ($N \times N \times 1$ primitive unit cell) doped by the Boron atoms (which can effectively
511 modulate the position of Fermi level but minimally alter other physical properties of
512 graphene). For Brillouin zone (BZ) integrations, a Monkhorst-Pack k-point mesh
513 scheme with $7 \times 7 \times 1$ are adopted²¹. Meanwhile, for graphene and monolayer MoSe₂
514 system, we use a $9 \times 9 \times 1$ k-meshes for Brillouin zone sampling in the
515 Monkhorst-Pack scheme. In order to simulate the graphene/MoSe₂ 2D
516 heterostructures, we impose a commensurability condition between them, where a
517 4×4 supercell of the graphene is used to match a 3×3 supercell of the MoSe₂
518 monolayer. The lateral lattice parameter for the triangular lattice of graphene/MoSe₂
519 nanocomposite is set to be $a(\text{graphene/MoSe}_2) = (4 \cdot a(\text{graphene}) + 3 \cdot a(\text{MoSe}_2)) =$
520 9.912 \AA , in which $a(\text{graphene}) = 2.468 \text{ \AA}$ and $a(\text{MoSe}_2) = 3.317 \text{ \AA}$ are optimized for
521 isolated graphene and monolayer MoSe₂, respectively. Since there is only a very small
522 lattice mismatch of $\sim 0.4\%$ between the graphene and the MoSe₂ monolayers, the
523 approximation used here is reasonable. Boron-doped graphene/MoSe₂ nanocomposite
524 in which one random carbon atom is replaced with a boron atom is also used to study
525 the influence of doping on the thickness of graphene/MoSe₂ heterojunction (along z
526 direction). Since the heterostructure contains one graphene and one MoSe₂ monolayer,
527 van der Waals (vdW) interactions are taken into account by using the semi-empirical
528 DFT-D3 method²². For the pristine as well as the boron-doped graphene/MoSe₂
529 heterojunction, a $5 \times 5 \times 1$ Monkhorst-Pack k-point mesh is used. A cutoff of 500 eV
530 is used for the plane-wave expansion of the wave function to converge the relevant
531 quantities. The structure relaxations are carried out until all the atomic forces on each

532 ion are less than $0.01 \text{ eV}/\text{\AA}$, enforcing a total energy convergence of $1 \times 10^{-7} \text{ eV}$. To
533 avoid spurious interactions with replicas, a vacuum slab larger than 15 \AA is added in
534 the z direction for all system.

535 Results:

536 As demonstrated in Supplementary Figure 8, the Fermi energy levels are adjusted,
537 by boron doping, to the valence bands and the whole doping system acts as a p-dopant.
538 Supplementary Figure 9(a) shows the Fermi energy levels of the doped graphene with
539 different $N \times N$ primitive cells. With the decrease of dopant concentration, the Fermi
540 energy level gradually decreases to the experimental value. As the Fermi energy level
541 of boron-doped graphene reaches the experimental value (-0.17 eV), the primitive cell
542 size is 26.1×26.1 , and the calculated effective mass of B-doped system for electron
543 and hole is 0.0116 and 0.0121, respectively. (Supplementary Figure 9(a))

544 The optimized thickness between the two layers (C atom - Mo atom) of
545 Boron-doped graphene/MoSe₂ heterojunction is 5.178 \AA , which is very close to the
546 value of undoped heterojunction (5.183 \AA), thereby low concentration of boron doping
547 does not alter the thickness between the two layers significantly. Furthermore, the
548 in-plane dielectric constant of B-doped graphene is almost equal to the value of
549 pristine graphene when the primitive cell size is larger than 8×8 . We set the dielectric
550 constant of graphene/MoSe₂ heterojunction as $\bar{\epsilon} = 4.969$, which is the average value
551 of the calculated in-plane dielectric constant $\epsilon_{xx} = 3.651$ and 6.287 for graphene and
552 monolayer MoSe₂, respectively.

553 To obtain the potential experienced by an electron at (ρ, z) due to the presence of

554 a hole at $(0, z_0)$, we use a field method described by Smythe²³ and extended by
 555 Sritharan¹³:

$$\begin{aligned}
 556 \quad & V(\rho) = \\
 557 \quad & -\frac{e^2}{4\pi\epsilon_0\bar{\epsilon}} \left\{ \frac{1}{\sqrt{(z-z_0)^2+\rho^2}} + \sum_{n=0}^{\infty} \left(\frac{\beta_N}{\beta_P} \right)^n \left[\frac{\beta_N/\beta_P}{\sqrt{(z-z_0+2c+2nc)^2+\rho^2}} + \frac{\beta_N/\beta_P}{\sqrt{(z_0-z+2c+2nc)^2+\rho^2}} + \right. \right. \\
 558 \quad & \left. \left. \frac{(K_2-K_1)/(K_2+K_1)}{\sqrt{(z+z_0-2a+2nc)^2+\rho^2}} + \frac{(K_2-K_3)/(K_2+K_3)}{\sqrt{(2b-z-z_0+2nc)^2+\rho^2}} \right] \right\}, \quad (21)
 \end{aligned}$$

559 where $\beta_P = (\bar{\epsilon} + 1)^2$ and $\beta_N = (\bar{\epsilon} - 1)^2$, $K_1=K_3=1$ since the graphene/MoSe₂
 560 heterojunction is approximated as a single dielectric slab sandwiched between
 561 vacuum. When the electron and hole are at different layers, they are assumed to be at
 562 the center of the graphene and MoSe₂ monolayer along the z direction with a fixed
 563 distance of 5.183 Å which is the thickness of heterojunction. The exciton is then
 564 simplified as a two dimensional quasi-particle²⁴.

565 When the hole is in graphene and the electron is in the MoSe₂ layer, the effective
 566 mass of exciton can be obtained from $\frac{1}{\mu} = \frac{1}{m_h^*} + \frac{1}{m_e^*}$, where the effective mass of
 567 monolayer MoSe₂ conduction band $m_e^* = 0.502m_0$ (Supplementary Figure 10) and
 568 the effective mass of B-doped graphene valence band $m_h^* = 0.0121m_0$, m_0 is the
 569 free electron mass. Based on the finite element method, we then solved the
 570 Schrödinger equation on a 2×10^3 Å circle plane, with the effective mass of the
 571 excitonic quasi particle and aforementioned Coulomb potential. The eigenvalues and
 572 selected eigenfunctions calculated are presented in Supplementary Figure 11a. The
 573 exciton transition energy between 1s and 2p is $E_{1s-2p} = 0.274$ eV, and the means
 574 radius of wave functions for 1s CT is $\langle \rho_{CT_{1s}} \rangle = 45.3$ Å. The 1s-1p transition energy is
 575 also larger than 0.20 eV. Both 1s-1p and 1s-2p transition energy values are close to the

576 experimental value 0.27 eV.

577 When electron is in graphene and hole is in MoSe₂, the hole and electron
578 effective masses should be the effective masses of the monolayer MoSe₂ valence
579 band (0.583 m_0) and B-doped graphene conduction band (0.0116 m_0), respectively.
580 The eigenvalues and selected eigenfunctions are presented in Supplementary Figure
581 11b. The exciton transition energy between 1s and 2p is $E_{1s-2p} = 0.267$ eV, and the
582 mean radius of wave functions for 1s CT is $\langle \rho_{CT_{1s}} \rangle = 46.6$ Å. There is a 0.007 eV
583 difference between the binding energies in two conditions. In our experiments, the
584 difference is too small to distinguish.

585 Note that the energies are calculated for the Fermi level of -0.17 eV, for Fermi
586 level of -0.19 eV, the calculated values are provided at the end of this summary.

587 In principle, the exciton 1s-2p transition energy can be estimated by performing
588 the quasiparticle GW and Bethe-Salpeter equation (BSE) calculations. To model the
589 experimental condition herein for the boron doped graphene/MoSe₂ heterojunction,
590 however, the system constructed should contain more than 1200 atoms (the supercell
591 of graphene should be larger than 26×26). On the other hand, to achieve well-converged
592 quasiparticle energies, especially when there is a transition metal element (Mo) in the
593 system, a dense enough ($21 \times 21 \times 1$) k-grid, a large energy cutoff (~ 40 Ry) and a large
594 number (> 10000) of unoccupied states is needed^{12, 25}. The calculation cost is
595 therefore far beyond the computing resource we can afford.

596

597 **Supplementary Note 12: Al-doped Graphene**

598 To explore the possible effect of dopant rather than Fermi level on the binding
599 energy, we also compute the interlayer exciton binding energy with Al-doped
600 graphene.

601 As demonstrated in *Supplementary Figure 15*, the Fermi energy levels are
602 adjusted, by aluminum doping, to the valence bands and the whole doping system also
603 acts as a p-dopant. *Supplementary Figure 15* shows the Fermi energy levels of the
604 doped graphene with different $N \times N$ primitive cells. With the decrease of dopant
605 concentration, the Fermi energy level gradually decreases to the experimental value.
606 As the Fermi energy level of aluminum-doped graphene reaches the experimental
607 value (-0.17eV), the primitive cell size is 13×13 (*Supplementary Figure 16*), and the
608 calculated effective mass of Al-doped system for electron and hole is 0.0122 and
609 0.0135, respectively. (*Supplementary Figure 17*)

610 When the hole is in graphene and the electron is in the MoSe_2 layer, the effective
611 mass of exciton can be obtained from $1/(\mu) = 1/(m_h^*) + 1/(m_e^*)$, where the effective
612 mass of monolayer MoSe_2 conduction band $m_e^* = 0.502m_0$ (*Supplementary Figure 18*)
613 and the effective mass of Al-doped graphene valence band $m_h^* = 0.0122m_0$. m_0 is
614 the free electron mass. Based on the finite element method, we then solved the
615 Schrödinger equation on a $2 \times 10^3 \text{ \AA}$ circle plane, with the effective mass of the
616 excitonic quasi particle and aforementioned Coulomb potential. The eigenvalues and
617 selected eigenfunctions calculated are presented in *Supplementary Figure 18A*. The
618 exciton transition energy between 1s and 2p is $E_{1s-2p} = 0.275 \text{ eV}$, and the means radius
619 of wave functions for 1s CT is $\langle \rho_{CT_{1s}} \rangle = 45.7 \text{ \AA}$.

620 When electron is in graphene and hole is in MoSe₂, the hole and electron
621 effective masses should be the effective masses of the monolayer MoSe₂ valence band
622 0.583 m_0 and Al-doped graphene conduction band 0.0135 m_0 , respectively. The
623 eigenvalues and selected eigenfunctions are presented in Supplementary Figure 18b.
624 The exciton transition energy between 1s and 2p is $E_{1s-2p} = 0.292$ eV, and the means
625 radius of wave functions for 1s CT is $\langle \rho_{CT_{1s}} \rangle = 42.3 \text{ \AA}$.

626

627 **Supplementary Note 13: Both electron and hole effective masses of graphene are**
628 **0.012 m_0**

629 Using 0.012 m_0 reported for a CVD-grown graphene (JAP, 113, 043708, 2013),
630 we also calculate the interlayer binding energy.

631 In summary, the binding energy is only slightly affected by the nature of dopant.
632 In three different calculations, the binding energy is between 0.265~0.292 eV. All are
633 very close to the experimental value.

634

635 **Supplementary Note 14: Origin of carrier reduced mass in graphene and large**
636 **interlayer binding energy**

637 It is well known that the effective mass of the pristine graphene is zero, i.e., the
638 effect mass of the state at $\mathbf{k} = 0$ (Dirac point, at which the Fermi level lie in the
639 pristine graphene) is zero. However, when the \mathbf{k} point deviates from the Dirac point,
640 e.g., the Fermi level shifts due to the effect of doping, the effective mass is no longer
641 zero. We can approximately evaluate this effect based on a tight-binding (TB) model.

642 Under a TB approach, the energy dispersion of graphene near the Fermi points is
 643 expressed as Dirac cones:

$$644 \quad E(\mathbf{k}) = \frac{3}{2} t_0 a_0 |\mathbf{k}|, \quad (22)$$

645 where $a_0 = 2.42 \text{ \AA}$ is the equilibrium C-C bond length, and $t_0 \approx 2.7 \text{ eV}$ is the hopping
 646 energy between the nearest C atoms. The effective mass at \mathbf{k} is defined as

$$647 \quad m^*(\mathbf{k}) = \frac{\hbar^2}{\frac{\partial^2 E(\mathbf{k})}{\partial \mathbf{k}^2}}. \quad (23)$$

648 When $\mathbf{k} \neq 0$, the energy dispersion along the normal direction has a finite curvature
 649 (effective mass) as shown as the solid lines in Supplementary Figure 21. Based on
 650 Supplementary Equation (22) and (23), the effective mass of electrons at the Fermi
 651 level (chemical potential μ) along the normal direction is given as

$$652 \quad m^* \Big|_{E(\mathbf{k})=\mu} = \frac{4\hbar^2}{9t_0^2 a_0^2} |\mu|. \quad (24)$$

653 The dependence of the effective mass on the chemical potential is plotted in
 654 Supplementary Figure 22. It can be seen that $m^* = 0$ for pristine graphene ($\mu = 0$), but
 655 it linearly increases with increasing $|\mu|$. For $\mu = -0.17 \text{ eV}$, $m^* \approx 0.039m_0$. If we
 656 roughly regard the effective mass along the radical direction as unchanged (equal to 0
 657 as that in pristine graphene), a direct average on both the normal and radical
 658 directions give $m^* \approx 0.02m_0$ under $\mu = -0.17 \text{ eV}$. This estimated value is actually very
 659 close to $m^* \approx 0.0116m_0$ from the ab initio calculations assuming B-doped.

660 In 3D systems, the electric potential of a point charge under the screening of a
 661 dielectric surrounding with a relative dielectric constant ϵ is described by a screened
 662 Coulomb potential (in atomic units)

663
$$V_{3D}(\mathbf{r}) = \frac{1}{\epsilon r} \quad (25)$$

664 by solving the Poisson's equation, which gives the exciton binding energy of the
 665 hydrogen-like model:

666
$$E_{b,3D}(\mathbf{r}) = \frac{m^*}{2\epsilon^2}, \quad (26)$$

667 where m^* is the reduced effective mass. Supplementary Equation (26) suggests that E_b
 668 is proportional to m^* .

669 However, for a 2D dielectric sheet embedded into vacuum such as single-layer
 670 MoSe₂ or other 2D systems considered in experiments, the 2D screened potential is
 671 not given as a screened Coulomb potential:

672
$$V_{2D}(\mathbf{r}) \neq \frac{1}{\epsilon r}, \quad (27)$$

673 but is given as a complicated function in terms of the Struve function and the second
 674 kind Bessel function [refer to: P. Cudazzo, I. V. Tokatly, and A. Rubio, *Phys. Rev. B* **84**,
 675 085406 (2011)], which was also listed as Supplementary Equation (19). As a result,
 676 the binding energy of 2D exciton is not described by Supplementary Equation (26),
 677 but is given as [refer to: T. Olsen, S. Latini, F. Rasmussen, and K. S. Thygesen, *Phys.*
 678 *Rev. Lett.* **116**, 056401 (2016)]

679
$$E_{b,2D} = \frac{8m^*}{\left(1 + \sqrt{1 + \frac{32\pi\alpha m^*}{3}}\right)^2}, \quad (28)$$

680 where α is the 2D polarizability calculated as $\alpha = L \frac{\epsilon - 1}{4\pi}$ where ϵ is the dielectric
 681 constant of bulks and L is the thickness of the 2D sheet. A remarkable property in
 682 Supplementary Equation (28) is that E_b is no longer linearly proportional to m^* . When

683 am^* is large enough, we have $E_{b,2D} \approx \frac{3}{4\pi\alpha}$ which is independent on m^* .
684 Supplementary Equation 26 is applicable to both undoped and doped 2D systems. An
685 exciton is formed between an electron and a hole no matter whether the system is
686 doped or undoped, and the influence of doping just lies in its screening effect. As the
687 Fermi energy level of boron-doped graphene reaches the experimental value (-0.17
688 eV), the primitive cell size is 26.1 (refer to page 17 of SI), and the doping ratio is
689 actually small (less than 0.2%). Our ab initio calculation shows that the in-plane
690 dielectric constant of B-doped graphene is almost equal to the value of pristine
691 graphene when the primitive cell size is larger than 8 (refer to page 19 of SI).
692 Therefore, the screening effect in doped graphene is very weak.

693 Take the graphene-MoSe₂ as an example, $\bar{\epsilon} \approx 4.97$ and $L \approx 10.4 \text{ \AA}$, so we have α
694 $= 6.17$ (in atomic units). E_b is plotted in Supplementary Figure 23 as a function of m^*
695 based on Supplementary Equation (28), measured with respect to that of $m^* = 0.6$
696 (the value for MoSe₂). The result of Supplementary Equation (26) is also plotted to
697 provide a comparison. It can be seen that the dependence of E_b on m^* is highly
698 nonlinear for the 2D result in Supplementary Equation (28). For $m^* = 0.01m_0$, the 3D
699 result in Supplementary Equation (26) gives $E_b(m^* = 0.01)/E_b(m^* = 0.6) = 1/60$, i.e.,
700 the binding energy for $m^* = 0.01$ of graphene-MoSe₂ is 60 times smaller compared to
701 that of MoSe₂ as indicated by the reviewer, but the 2D result in Supplementary
702 Equation (28) gives $E_b(m^*=0.01)/E_b(m^*=0.6) \approx 1/3.1$. In other words, with the 2D
703 equation, the binding energy for $m^* = 0.01$ of graphene-MoSe₂ is merely 3.1 times
704 smaller compared to that of MoSe₂. Taking the 0.4-0.6 eV binding energy for the latter,

705 one may expect a binding energy of 0.13-0.19 eV for the graphene-MoSe₂ exciton,
706 which is close to the results from our measurements. In our work, we used a more
707 sophisticated theory that has been tested with other 2D systems (JACS 2015, 137,
708 8313-8320; Nat. Commun. 2016, 7, 12512) to calculate the binding energy and it
709 turns out to be very close to the experimental value. The dependence of calculated
710 1s-2p transition energy on reduced mass is plotted in Supplementary Figure 12.
711

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